Periodic Atomic Displacements and Visualization of the Electron-Lattice Interaction in the Cuprate

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Traditionally, x-ray scattering techniques have been used to detect the breaking of the structural symmetry of the lattice, which accompanies a periodic displacement of the atoms associated with charge density wave (CDW) formation in the cuprate pseudogap states. Similarly, spectroscopic imaging scanning tunneling microscopy (STM) has visualized the short-range CDW. However, local coupling of electrons to the lattice in the form of a short-range CDW has been a challenge to visualize; thus, a link between these measurements has been missing. Here, we introduce a novel STM-based technique to visualize the local bond-length variations obtained from topographic imaging with picometer precision. Application of this technique to the high- T_c cuprate superconductor Bi₂Sr₂CaCu₂O_{8+ $\delta}$} reveals a high-fidelity local lattice distortion of the BiO lattice as large as 2%. In addition, analysis of local breaking of rotational symmetry associated with the bond lengths reveals modulations around four-unit-cell periodicity in both B_1 and E representations in the C_{4n} group of the lattice, which coincides with the unidirectional d-symmetry CDW (dCDW) previously identified within the CuO₂ planes, thus providing direct evidence of electron-lattice coupling in the pseudogap state and a link between the x-ray scattering and STM measurements. Overall, our results suggest that the periodic lattice displacements in E representations correspond to a locally frozen version of the soft phonons identified by the x-ray scattering measurements, and a fluctuation of the bond length is reflected by the fluctuation of the dCDW formation near the quantum critical point.

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I. INTRODUCTION

The CuO₂ plane is a common crystallographic ingredient among different cuprate families and is the main playground of the electronic system. It determines both the electronic and magnetic characteristic properties and exhibits instabilities toward states such as antiferromagnetism, *d*-wave superconductivity [1], charge density wave (CDW) [2], electronic nematicity [3–5], and pair density wave [6,7]. Interactions among these states as well as their relationships to the bosonic excitations such as the antiferromagnetic spin fluctuation and the phonon excitations underlie the grand challenge of understanding the mechanism of the cuprate high-temperature superconductivity. Subject Areas: Condensed Matter Physics, Strongly Correlated Materials, Superconductivity

One of the key phenomena commonly associated with the cuprate pseudogap state is CDW order, for which a static modulation of the charge appears below or near the pseudogap opening temperature T^* [8,9]. Such a CDW has been ubiquitously observed in different cuprate families, and the associated period is 3-4 unit cells, depending on the material [2]. Spectroscopic imaging (SI) scanning tunneling microscopy (STM) measurements on $Bi_2Sr_2CaCu_2O_{8+\delta}$ and Ca_{2-r}Na_rCuCl₂O₂ provide evidence for a unidirectional electronic modulation with d-wave symmetry CDW (dCDW), in which the charge modulations primarily occur at in-plane oxygen sites, O_x and O_y , in a fashion such that the changes on these sites are out of phase by π [10]. In contrast, x-ray diffraction on different cuprates provides evidence of periodic modulations of the lattice, from which CDW (or charge-stripe) order is inferred [11,12], while resonant soft-x-ray scattering provides evidence of charge modulations on specific atomic sites [13,14]. Furthermore, neutron scattering and resonant inelastic x-ray scattering measurements show a softening of bond-stretching phonons at the wave vector

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corresponding to charge order [15–17]. An open question is whether it is possible to detect in real space the lattice modulation associated with the *d*CDW in $Bi_2Sr_2CaCu_2O_{8+\delta}$. Here, we demonstrate a new approach to analyzing local bond-length modulations detected by STM topography and show how they correlate with the electronic modulations measured simultaneously by SI STM.

Fundamentally, the CuO₂ plane is formed by a square lattice of Cu, with oxygens O_x and O_y bridging Cu atoms along orthogonal directions. In Bi₂Sr₂CaCu₂O_{8+ δ}, there is an orthorhombic distortion of the lattice along axis at 45° to the Cu–O–Cu bond directions, due to a long-period modulation within the BiO bilayers. The SI STM measurements are performed after cleaving the sample, which leaves a BiO layer at the surface. The long-period modulation of the BiO layers is well characterized and can be mathematically removed from the topographic image, leaving on average a square lattice of Bi atoms whose positions are expected to correlate with the Cu atoms below them.

II. SYMMETRY OF THE CuO₂ SQUARE LATTICE

The CuO_2 square lattice is group-theoretically classified as a C_{4v} group consisting of eight symmetry operations, $\{I, C_4, C_2, C_4^3, \sigma_x, \sigma_y, \sigma_d, \sigma_{d'}\}$, in which I describes a unitary operation, C_n describes an *n*-fold rotation, and σ_i describes a mirror operation along the *i* plane [d(d')] is a mirror plane 45° rotated from the x (y) axis]. Irreducible representations of the C_{4v} group are often denoted by A_1 , A_2 , B_1 , B_2 , and E [18], in which A represents a quantity that is invariant under C_4 rotation and B represents one that changes the sign under the C_4 rotation. E is a twodimensional irreducible representation. Ideally, physical quantities within the CuO₂ plane are invariant under these operations and are represented by the irreducible representations. However, in a real material, there are impurities, missing atoms, and electronic instabilities coupling to the lattice that would break the underlying symmetry of the lattice. By evaluating symmetries for each irreducible representation with respect to bond lengths about each atomic site, one can test for and characterize the spatial correlations of local lattice distortions.

Before such an analysis can be attempted, we must first correct for an experimental artifact. The piezo scanner that controls the STM tip scanning along the x and y directions at a picometer precision shows nonlinear drifts such that atomic modulations observed in a topography $T(\mathbf{r})$ are extrinsically phase disordered. In other words, an observed image of the atoms is not simply described by a sinusoidal wave, but by a combination of cosine and sine waves due to the extrinsic phase disorder. In fact, if the Fourier transform of a raw $T(\mathbf{r})$ is evaluated, the Bragg peaks that correspond to periodicities of the atoms show finite intensities in the imaginary part (the sine component of the modulation)

because of the tip drifts. Similarly, electronic structure images that are simultaneously taken with the $T(\mathbf{r})$, such as current and dI/dV, etc., are phase disordered in the same way.

To overcome this systematic error in the measurement, the Lawler-Fujita algorithm [4] has been widely used to remove the effect of the tip drift from images [19–21]. This algorithm derives a displacement vector field $\mathbf{u}(\mathbf{r})$ (Fig. 5 in Appendix A) that describes local phase shifts of a particular modulation (in the present case, atomic modulations) at a coarse-graining length. For the piezo scanner drift, a coarse-graining length scale is typically much longer than the atomic spacing d_0 (3.83 Å) of the CuO₂ unit cell, since relaxation of the piezo scanner occurs very slowly. Here, we use $\mathbf{u}(\mathbf{r})$ to remove the piezo scanner drift, rendering the $T(\mathbf{r})$ and any simultaneously taken images to be lattice periodic on average over the 30 Å coarse-graining length (see Appendix A). To determine the residual (intrinsic) atomic displacements and extract the bond lengths between adjacent Bi atoms, we apply the Lawler-Fujita algorithm once again to obtain the displacement vector field $\mathbf{u}'(\mathbf{r})$, but now with 6 Å for the coarse-graining length. Now, $\mathbf{u}'(\mathbf{r})$ represents how much individual atoms are displaced from the perfect lattice configuration, allowing us to locate Bi positions and possible intrinsic local lattice distortions (Appendix A). While the Cu atoms within the CuO_2 plane are not directly identified as the STM scans over the BiO surface, as we show in Fig. 4, electronic structures from the CuO₂ layer spatially correlate with structural modulations of the atomic displacements detected at the BiO layer, suggesting that displacements of the Bi atoms reflect those for the Cu atoms within the CuO_2 .

III. RESULTS

We perform SI STM measurements at 11 K on a single crystal of $Bi_2Sr_2CaCu_2O_{8+\delta}$ that is near optimal doping, with a superconducting transition temperature T_c equal to 91 K (see Appendix B). Figure 1(a) shows a typical driftcorrected topograph, $T(\mathbf{r})$, of the Bi₂Sr₂CaCu₂O_{8+ δ} sample in a 50 nm \times 50 nm field of view (FOV). The atoms seen in the $T(\mathbf{r})$ are Bi, and the Cu atoms that form the twodimensional CuO₂ plane where all the cuprate phenomenology occurs are located 6 Å below these Bi atoms. It should be noted that the crystal supermodulation displaces the atoms in the bulk at $\mathbf{q} = \mathbf{Q}_{\text{SM}}$ along the *b* direction. Such an effect is apparently stronger in BiO than in the CuO₂ planes [22], masking intrinsic lattice displacements [23]. It is also found that the amplitude of the modulation of the Bi and Cu atoms is, at most, 2% of the Cu-O-Cu bond length $d_0 \sim 3.83$ Å [24]. SI STM visualizes both the spatial amplitude and phase of the modulation such that the effect of the supermodulation can be effectively removed by Fourier filtration (Fig. 6 in Appendix A), after which we anticipate that the Cu atom positions can be precisely inferred from the Bi positions. Similarly, the symmetry of the lattice in the CuO₂ plane can be obtained by studying the symmetry of the BiO plane that is observed in the $T(\mathbf{r})$. Hence, local lattice distortions of the CuO₂ plane can be accessed by studying the corrected topograph, $T(\mathbf{r})$.

Now that we know where all the Bi atoms are in $T(\mathbf{r})$, after the filtration of the supermodulation, we show in Fig. 1(b) a bond-length map, in which the bond length is represented by a colored bar connecting nearest-neighbor Bi atoms. Small black dots represent the locations of the Bi atoms. It is obvious that the Bi atoms are disordered, exhibiting local displacements from the perfectly periodic positions. Figure 1(c) shows histograms of the bond lengths for both the x (right) and y (left) directions. Both bond lengths show virtually identical distributions; peak positions (or averages) are located at d_0 , and the full width at half maximum is about 2% of d_0 for both directions. Such disorder is somewhat larger than the estimate from neutron scattering measurements for a stripe-ordered system [25].

Next, we separate the bond lengths according to orientation, associating the bond length between neighboring Bi atoms at \mathbf{r}_i and \mathbf{r}_i with the bond center $\mathbf{R}_{ij} = (\mathbf{r}_i + \mathbf{r}_j)/2$. Note that, geometrically, \mathbf{R}_{ij} is approximately the location of the oxygen sites O_x and O_y in the CuO_2 plane. Figure 1(d) [1(e)] is an intensity map of the bond length along the x direction (y direction), $B_{\rm x}({\bf r})$ $[B_{v}(\mathbf{r})]$. It is obvious that the bond lengths for both the x and y directions show strong modulations along the Cu-O-Cu directions. The inset in Fig. 1(d) [1(e)] is a Fourier transform of the $B_x(\mathbf{r}) [B_y(\mathbf{r})]$, which exhibits a strong peak around $\mathbf{Q}_B \sim (\pm 0.2, 0)$ [(0, ± 0.2)] similar to those observed for the electronic density wave in the pseudogap states [26]. The same peaks also appear around the Bragg peaks (marked as "+"), due to the convolution of the lattice and modulations in real space [10]. Line profiles shown in Fig. 1(f) reveal fine structures of these Fourier peaks for both the x and y directions, while the general shape of each line profile can be fitted by a Lorentzian. The fine structure



FIG. 1. Spatial variation of the bond length. (a) Topograph of the Bi₂Sr₂CaCu₂O_{8+ δ} taken within 50 nm × 50 nm FOV. The inset shows an enlarged topograph from the small rectangle region near the center of the FOV. Bi locations are marked up by the red dots, while perfect lattice locations are marked up by the blue dots. (b) Bond-length map, in which adjacent Bi atoms are connected by the straight lines. Line colors represent a bond length in the unit of d_0 (~3.83 Å). The inset shows the same topo as in the inset in (a) superimposed with the bond-length map. (c) Histograms of the bond length for both x and y directions. (d)[(e)] A sublattice representation of the bond length along the x (y) direction. A value of the bond length is assigned at the O_x (O_y) position of the CuO₂ plane. The inset is a magnitude Fourier transform of $B_x(\mathbf{r})$ [$B_y(\mathbf{r})$]. (f) Line profiles of the Fourier transform in (d) and (e) insets extracted from the black dashed lines along the Cu–O–Cu bond directions. The black crosses are locations of the Bragg peaks.

suggests that there would be a domain structure of the bond-length modulations having a discommensuration between the domains [27]. These observations obviously show that the translational symmetry of the lattice is broken, which implies an associated modulation of electronic states.

Now, we introduce a symmetry decomposition of the local lattice to analyze possible local breaking of rotational symmetry in the lattice structure. We define a function

$$S^{\Gamma}(\mathbf{r}_i) = \frac{\sum_j s_{ij}^{\Gamma} |\mathbf{r}_i - \mathbf{r}_j|}{\frac{1}{N} \sum_j |\mathbf{r}_i - \mathbf{r}_j|},\tag{1}$$

where Γ is an irreducible representation of the C_{4v} group (A_1, A_2, B_1, B_2, E) , \mathbf{r}_i and \mathbf{r}_j are positions of the Bi atoms at *i* and *j* sites, respectively, and s_{ii}^{Γ} is a symmetry-dependent structure factor. A summation is taken over the nearestneighbor sites for $\Gamma = A_1, B_1, B_2, E$, so that $|\mathbf{r}_i - \mathbf{r}_i|$ is on the scale of d_0 and the denominator in Eq. (1) is a local average of the bond lengths. [For $\Gamma = A_2$, the summation is taken over the next-nearest-neighbor sites (blue arrows).] For example, when we consider the A_1 representation of the local bonds, $s_{ij}^{A_1} = 1$, since the A_1 representation shows no sign change by the symmetry operations in C_{4v} group (see Appendix C and Table I). Then the summation is taken over the nearest-neighbor sites [red arrows in Fig. 2(a)] with the weight $s_{ii}^{A_1}$. To visualize the irreducible representations of the bonds obtained using Eq. (1), we draw a line over the atomic site i with a color that represents the intensity obtained by Eq. (1). For example, for the B_1 representation [Fig. 2(b)], we draw a line along the direction of the expanded bonds [28].

The B_1 representation is of particular interest, as it corresponds to a bond-stretching quadrupolar mode [15,29]. In Fig. 2(b), we show the B_1 representation of the bond lengths, in the same FOV as Fig. 1; it exhibits a pattern in which blue and orange bonds alternate at a period of fourunit cells. This can be viewed as a bond "nematicity." Moreover, a careful examination of the pattern within regions highlighted by the black rectangles in Fig. 2(b) leads us to identify a modulation similar to the unidirectional *d*CDW that can be seen in the electronic states [Fig. 4(a)]. In fact, the B_1 representation is equivalent to *d* symmetry, suggesting that *d*CDW couples to the lattice and locally alters the bond length in a *d*-wave fashion [29].

Figures 2(c) and 2(d) show the (1, 1) and (2, 2) components of the *E* representation, E_{11} and E_{22} , respectively, as given by a 2 × 2 matrix representation. $S^{E_{11}}(\mathbf{r}_i)$ [$S^{E_{22}}(\mathbf{r}_i)$] is simply the difference of the bond lengths between nearest-neighbor sites of \mathbf{r}_i along only the *x* (*y*) direction, characterizing a unidirectional stretching or shrinking of the bond length. It is intriguing that both the E_{11} and E_{22} representations show spatial modulations of the bond stretching. Similar to the B_1 representation



FIG. 2. Symmetry representations of the in-plane bond length. (a) A schematic image of the CuO₂ plane, in which Cu $3d_{x^2-y^2}$ orbitals and O 2*p* orbitals are represented in blue and yellow, respectively. The red and blue arrows are vectors connecting the nearest- and the next-nearest-neighbor Cu sites. (b) B_1 representation of the bond length obtained from the topography shown in Fig. 1(a). (c) (1,1) component of the *E* representation, E_{11} , of the bond length obtained from the topography shown in Fig. 1(a). (d) (2,2) component of the *E* representation, E_{22} , of the bond length obtained from the topography shown in Fig. 1(a). The solid (dotted) black rectangles in (b)–(d) highlight locations of the canonical *d*-symmetry CDW identified in the electronic structure for the *x* (y) direction (Appendix A, Fig. 8).

shown in Fig. 2(b), a canonical "bond-stretching halfbreathing mode" can be found in regions highlighted within the black rectangles again where unidirectional *d*CDW can be clearly observed. Other representations such as A_1 , A_2 , and B_2 are shown in Fig. 9 in Appendix C.

To analyze the modulated structures of each representation of the bond length, shown in Fig. 2, we replot $S^{\Gamma}(\mathbf{r}_i)$ for $\Gamma = B_1$, E_{11} , and E_{22} in Figs. 3(a), 3(b), and 3(c), respectively, now with amplitude indicated by colored circles. The corresponding Fourier-transform magnitudes are presented in Figs. 3(d), 3(e), and 3(f), respectively. For the case of B_1 symmetry, the magnitude of the Fourier transform shown in Fig. 3(d) exhibits strong peaks near $\mathbf{Q} \sim (\pm 1/4, 0), (0, \pm 1/4)$. Note that there are similar peaks surrounding the Bragg peaks marked by the black cross. Similarly, the magnitude Fourier transforms for E_{11} and E_{22} , shown, respectively, in Figs. 3(e) and 3(f), exhibit strong peaks near the same wave vectors, but only along the \mathbf{Q}_x or \mathbf{Q}_y direction.

In Fig. 4(a), we show $R(\mathbf{r}, 150 \text{ mV}) = [I(\mathbf{r}, +150 \text{ mV})/I(\mathbf{r}, -150 \text{ mV})]$, the ratio of the tunneling current measured



FIG. 3. Spatial modulations of the symmetry. (a) A sublattice representation of Fig. 2(b), $B_1(\mathbf{r})$, in which spatial modulations are clearly resolved for both the *x* and *y* Cu–O–Cu bond directions. (b) A sublattice representation of Fig. 2(c), $E_{11}(\mathbf{r})$, in which unidirectional modulations are clearly resolved for the *x* Cu-O-Cu direction. (c) A sublattice representation of Fig. 2(d), $E_{22}(\mathbf{r})$, in which unidirectional modulations are clearly resolved for the *y* Cu-O-Cu direction. (d) A magnitude Fourier transform of (a), exhibiting peaks along both the *x* and *y* Cu–O–Cu bond directions. The inset shows line profiles extracted from the black lines. (e) A magnitude Fourier transform of (b), exhibiting peaks along the *x* Cu–O–Cu bond direction. An inset shows a line profile extracted from the black line. Data points (open circles) are fitted by the Lorentzian, and a peak position is found at $\mathbf{q}_x = (0.242 \pm 0.03) 2\pi/d_0$. (f) A magnitude Fourier transform of (c), exhibiting peaks along the *y* Cu–O–Cu bond direction. An inset shows a line profile extracted from the black line. Data points (open circles) are fitted by a Lorentzian, and a peak position is $\mathbf{q}_y = (0.238 \pm 0.03) 2\pi/d_0$. The black crosses are locations of the Bragg peaks.

at opposite bias voltages, which is demonstrated to represent the cuprate dCDW without ambiguity, since the ratio can effectively mitigate a systematic error in the measurement related to the junction formation [30]. As shown in Fig. 4(a), unidirectional density wave modulations along both the x and y directions are clearly visualized, and the inset shows the phase-resolved sublattice Fourier transform in a *d*-symmetry channel [10], clearly showing peaks near $\mathbf{q} \sim (\pm 1/4, 0)$; $(0, \pm 1/4)$. This confirms that the electronic degrees of freedom at the O_r and O_v sites are out of phase by π , indicating that the shortrange density wave holds d symmetry even at $p \sim 0.17$. In Fig. 4(b), we show a comparison of the line profiles from $R(\mathbf{q}, 150 \text{ mV}), S^{B_1}(\mathbf{q}), S^{E_{11}}(\mathbf{q}), \text{ and } S^{E_{22}}(\mathbf{q})$ for both the x (left) and y (right) directions, where they all show peaks near $|\mathbf{q}| \sim 0.25$. Note that peak positions of the line profiles for $R(\mathbf{q}, 150 \text{ mV})$ and $S^{B_1}(\mathbf{q})$ are not peaked at exactly $|\mathbf{q}| = 0.25$; this is most likely due to the discommensuration of the dCDW domains [27]. A precise estimation of the wave vectors for B_1 , E_{11} , and E_{22} and a study of their amplitudes and phases in relationship to the electronic structure is an intriguing direction for the future studies.

IV. DISCUSSION

Figures 4(c) and 4(d) summarize the situation occurring in both the electronic and lattice structures within a canonical *d*CDW that is extracted from the black rectangle in Fig. 2. The *d*CDW breaks both the rotational and the translational symmetry as shown in Fig. 4(c) (right) together with the simultaneous topograph (left), in which local atomic displacements are clearly seen (red dots are the locations of atoms). Our observations indicate that the unidirectionality of the *d*CDW couples to the lattice, in which the bond length is periodically stretched and shrunk as revealed in the *E* representation (half-breathing mode) [Fig. 4(d) right], having a relatively shorter bond length at the bond center of the *d*CDW [28]. Simultaneously, a modulated intra-unit-cell rotational symmetry breaking of a *d*-symmetry electronic degree of freedom couples to the



FIG. 4. Direct visualization of the electron-lattice interaction. (a) $R(\mathbf{r}, 150 \text{ mV})$ within 20 nm \times 20 nm FOV exhibiting the d-CDW, in which electronic degree of freedoms reside primarily on the O_{x} and O_{y} modulating out of phase by π . The inset shows the magnitude of the phase-resolved Fourier transform of $R(\mathbf{r}, 150 \text{ mV})$ in the *d*-symmetry channel, exhibiting $\mathbf{Q} = (\pm 1/4, 0); (0, \pm 1/4)$ peaks as encircled by the orange dashed lines. The orange crosses mark locations of the Bragg peaks. (b) Line profiles extracted from the inset in (a) along gray lines and line profiles along the same line trajectories in $B_1(\mathbf{r})$, $E_{11}(\mathbf{r})$, and $E_{22}(\mathbf{r})$ for both x (left) and y (right) Cu–O–Cu bond directions. (c) $T(\mathbf{r})$ and $R(\mathbf{r}, 150 \text{ mV})$ extracted from a region in the solid orange rectangle in (a), where a canonical *d*-symmetry CDW is observed. (d) B_1 and E_{11} maps extracted from the same region of $B_1(\mathbf{q})$ and $E_{11}(\mathbf{q})$ obtained from the same FOV shown in (a).

lattice, displacing the atoms as revealed in the B_1 representation (quadrupolar mode) [Fig. 4(d) left].

From these results, one notices that the bond-length modulations in the B_1 and $E(E_{11} \text{ and } E_{22})$ representations show a strong correspondence to the electronic properties of the dCDW seen in the pseudogap states. First, the dCDW is highly unidirectional, breaking both rotational and translational symmetries [10,26]. Second, it is highly heterogeneous [26]. Third, it is predominantly d symmetry [10,26]. Fourth, the periodicity of the dCDW modulation is close to four unit cells [10,31]. Thus, the modulations in $S^{B_1}(\mathbf{q}), S^{E_{11}}(\mathbf{q}), \text{ and } S^{E_{22}}(\mathbf{q})$ together with the strong correspondence to $R(\mathbf{r}, 150 \text{ mV})$ indicate that there are significant electron-lattice interactions, reflecting the observation that the atoms are displaced in the same symmetrical fashion as those for the density wave in the electronic structure at the pseudogap energy scale. The presence of such coupling indicates that the modulations seen in $R(\mathbf{r}, 150 \text{ mV})$ indeed carry a charge degree of freedom, that of a charge density wave with dCDW, since an electron-lattice interaction requires a local charge density.

In addition, it is natural to speculate that the periodic lattice displacements visualized in E representations would be linked to the phonon dynamics reported in recent resonant x-ray measurement [17], in which the interplay of the bond-stretching phonons and charge order fluctuations are increased with lowering temperature below T_{c} . The wave vector at which the phonon anomalies occur is quite similar to the one observed in the *E* representations, suggesting that the periodic lattice displacements in Erepresentations correspond to a locally frozen version of the soft phonons and consistent with the lock-in mechanism of the dCDW by the electron-phonon coupling [32] for our $Bi_2Sr_2CaCu_2O_{8+\delta}$ sample. The lack of spatial homogeneity of the bond-length modulations, as shown in Fig. 1(b), can be linked to the quantum fluctuations of the density waves inside the superconducting dome that are associated with the corresponding quantum critical point located just beyond the optimal doping [17,33].

In this study, we have introduced a group theoretic symmetry decomposition algorithm to analyze the symmetry of a lattice as well as the local displacements of it in $Bi_2Sr_2CaCu_2O_{8+\delta}$. First, we visualized the heterogeneity of the in-plane bond lengths, for which the fluctuation is about 2% of the unit-cell distance. Second, the bond lengths themselves are spatially modulating at four-unit-cell periodicity. Third, B_1 , E_{11} , and E_{22} representations of the bond length are modulating at similar periodicity. Fourth, a strong correspondence between the dCDW and the displacements of the atoms visualized in the B_1 , E_{11} , and E_{22} representations of the bond length provides a direct realspace observation of the electron-lattice interactions for the first time, indicating that the charge degree of freedom is indeed involved in the dCDW. Finally, the STM bondlength analysis offers a new paradigm for simultaneous studies of both the local lattice and electronic structures in quantum materials.

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APPENDIX A: DETERMINATION OF THE ATOMIC SITES

For the purpose of the symmetry decomposition analysis, first, we apply the Affin transformations to the $T(\mathbf{r})$ and the simultaneously taken electronic structure images, so that the Bragg wave vectors \mathbf{Q}_1 and \mathbf{Q}_2 satisfy

TABLE	[. C]	haracter	table	for	the	C_{4v}	point	group.
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		Matrix representation								
Basic of the representation	Representation	Ι	C_4	<i>C</i> ₂	C_4^3	σ_x	σ_y	σ_d	$\sigma_{d'}$	
z	A_1	1	1	1	1	1	1	1	1	
$xy(x^2 - y^2)$	A_2	1	1	1	1	-1	-1	-1	-1	
$(x^2 - y^2)$	B_1	1	-1	1	-1	1	1	-1	-1	
xy	B_2	1	-1	1	-1	-1	-1	1	1	
$\{x, y\}$	Ē	$\begin{pmatrix} 1 & 0 \\ 0 & 1 \end{pmatrix}$	$\begin{pmatrix} 0 & -1 \\ 1 & 0 \end{pmatrix}$	$\begin{pmatrix} -1 & 0 \\ 0 & -1 \end{pmatrix}$	$\begin{pmatrix} 0 & 1 \\ -1 & 0 \end{pmatrix}$	$\begin{pmatrix} -1 & 0 \\ 0 & 1 \end{pmatrix}$	$\begin{pmatrix} 1 & 0 \\ 0 & -1 \end{pmatrix}$	$\begin{pmatrix} 0 & 1 \\ 1 & 0 \end{pmatrix}$	$\begin{pmatrix} 0 & -1 \\ -1 & 0 \end{pmatrix}$	

 $|\mathbf{Q}_1| = |\mathbf{Q}_2|$ and $\mathbf{Q}_1 \cdot \mathbf{Q}_2 = 0$ after the transformation. Second, we apply the Lawler-Fujita algorithm [4] to remove a STM tip drift in the data. In this algorithm, we obtain a displacement field $\mathbf{u}(\mathbf{r})$, which is associated with the STM tip drift, to render $T(\mathbf{r})$ to be perfectly lattice periodic (see Fig. 5). Finally, Bi positions are determined in the topograph, in which the STM tip drift is removed, but local lattice distortions remain. We describe these procedures step by step in the followings.

An original $T(\mathbf{r})$, in which the STM tip drift is embedded, is given by

$$T(\mathbf{r}) = |A_{\mathbf{Q}_1}(\mathbf{r})| \cos\{\mathbf{Q}_1 \cdot [\mathbf{r} + \mathbf{u}(\mathbf{r})]\} + |A_{\mathbf{Q}_2}(\mathbf{r})| \cos\{\mathbf{Q}_2 \cdot [\mathbf{r} + \mathbf{u}(\mathbf{r})]\} + \cdots, \quad (A1)$$

where $|A_{\mathbf{Q}}(\mathbf{r})|$ is an amplitude of the complex field $A_{\mathbf{Q}}(\mathbf{r})$ for the wave vector \mathbf{Q} and $\mathbf{u}(\mathbf{r})$ is the displacement field



FIG. 5. A schematic of the atomic displacements described by the displacement field $\mathbf{u}(\mathbf{r})$. The blue circles are atoms in the perfect lattice configuration. When there is a local lattice distortion, the lattice is displaced from the perfect lattice configuration, and such a displacement is characterized by the displacement field $\mathbf{u}(\mathbf{r})$.

that distorts the $T(\mathbf{r})$ [30]. A drift-corrected topography $T'(\mathbf{r})$ is then obtained by

$$T'(\mathbf{r}) = T(\mathbf{r} - \mathbf{u}(\mathbf{r})) \tag{A2}$$

(see Fig. 6). Similarly, the same displacement field $\mathbf{u}(\mathbf{r})$ is also used to remove the STM tip drift for simultaneously measured electronic structure images. The Lawler-Fujita algorithm utilizes a two-dimensional lock-in technique to obtain complex fields $A_{\mathbf{Q}}(\mathbf{r})$ for a periodic modulation at \mathbf{Q} . Amplitude $|A_{\mathbf{Q}}(\mathbf{r})|$ and spatial phase shift $\Phi_{\mathbf{Q}}(\mathbf{r})$ are then obtained by the following equations:

$$\mathbf{A}_{\mathbf{Q}}(\mathbf{r}) = \int d\mathbf{R} T(\mathbf{R}) e^{i\mathbf{Q}\cdot\mathbf{R}} e^{-(\mathbf{r}-\mathbf{R})^2/2\sigma^2}, \qquad (A3)$$

$$|A_{\mathbf{Q}}(\mathbf{r})| = \sqrt{[\mathrm{Re}A_{\mathbf{Q}}(\mathbf{r})]^2 + [\mathrm{Im}A_{\mathbf{Q}}(\mathbf{r})]^2}, \qquad (\mathrm{A4})$$



FIG. 6. The same topography as shown in Fig. 1(a) but without the crystal structural supermodulation that is removed by the Fourier filtrations at multiple supermodulation wave vectors. The red dots are locations of the Bi atoms. The inset shows the magnitude Fourier transform of the topography, representing a single sharp Bragg peak.

$$\Phi_{\mathbf{Q}}(\mathbf{r}) = \tan^{-1} \frac{\operatorname{Re}A_{\mathbf{Q}}(\mathbf{r})}{\operatorname{Im}A_{\mathbf{Q}}(\mathbf{r})},$$
 (A5)

where $\sigma(=L/2)$ a coarse-graining length. In the case of a tetragonal lattice that is characterized by two orthogonal Bragg vectors, \mathbf{Q}_1 and \mathbf{Q}_2 , $\mathbf{u}(\mathbf{r})$ is then simply related to the phase shift $\Phi_{\mathbf{Q}}(\mathbf{r})$ as $\mathbf{Q} \cdot \mathbf{u}(\mathbf{r}) = \Phi_{\mathcal{Q}}(\mathbf{r})$ for $\mathbf{Q} = \mathbf{Q}_1$ and \mathbf{Q}_2 , such that $\mathbf{u}(\mathbf{r})$ is obtained by

$$\mathbf{u}(\mathbf{r}) = \begin{pmatrix} \mathbf{Q}_1 \\ \mathbf{Q}_2 \end{pmatrix}^{-1} \begin{pmatrix} \Phi_{\mathbf{Q}_1}(\mathbf{r}) \\ \Phi_{\mathbf{Q}_2}(\mathbf{r}) \end{pmatrix}.$$
 (A6)

For the Bi₂Sr₂CaCu₂O_{8+ δ} in this study, we use L = 30 Å to calculate **u**(**r**) and the topography $T'(\mathbf{r})$, in which the STM tip drift is removed, is obtained by Eq. (A2). Note that the STM tip drift occurs at a much longer length scale than 30 Å, and this value is still much longer than the lattice constant of the Bi₂Sr₂CaCu₂O_{8+ δ}. This means that, while extrinsic distortions in $T'(\mathbf{r})$ due to the STM tip drift are removed by Eq. (A2), intrinsic local lattice distortions still remain in $T'(\mathbf{r})$. Then, we filter out the supermodulations.

Next, in order to locate Bi atoms, \mathbf{r}_{Bi} in $T'(\mathbf{r})$, for which atomic positions are displaced from the perfect lattice configuration, we apply the Lawler-Fujita algorithm [4] once again to $T'(\mathbf{r})$ to get a displacement field $\mathbf{u'}(\mathbf{r})$ with L = 6 Å. Now, the shorter coarse-graining length is used to capture atomic-scale local lattice distortions, and $\mathbf{u'(r)}$ characterizes how much the atoms are displaced from the perfect lattice configuration, which is again obtained by Eq. (A6). Note that the perfect lattice does not have any disorder, such that atomic corrugations are purely described by periodic functions without the phase disorder. Thus, a position of the Bi atoms in the distorted lattice is obtained by

$$\mathbf{r}_{\mathrm{Bi}} = \mathbf{r}_{\mathrm{Bi}}^{\mathrm{P}} + \mathbf{u}'(\mathbf{r}_{\mathrm{Bi}}^{\mathrm{P}}), \qquad (A7)$$

where \mathbf{r}_{Bi}^{P} a vector that represents the position of the Bi atoms in the perfect lattice. \mathbf{r}_{Bi}^{P} is a set of vectors that are 2π modulo in the phases of the perfect lattice,

$$\mathbf{Q}_1 \cdot \mathbf{r}_{Bi}^{P} = 2\pi m, \qquad m = 0, \pm 1, \pm 2, ...,$$
 (A8)

$$\mathbf{Q}_2 \cdot \mathbf{r}_{Bi}^{P} = 2\pi n, \qquad n = 0, \pm 1, \pm 2, ...,$$
 (A9)

and, then, \boldsymbol{r}_{Bi}^{P} is obtained by solving Eqs. (A8) and (A9):

$$\mathbf{r}_{\mathrm{Bi}}^{\mathrm{P}} = 2\pi \begin{pmatrix} \mathbf{Q}_{1} \\ \mathbf{Q}_{2} \end{pmatrix}^{-1} \begin{pmatrix} m \\ n \end{pmatrix}.$$
(A10)

Finally, by plugging Eq. (A10) into Eq. (A7), the Bi atom positions without extrinsic distortions due to the STM tip



FIG. 7. Procedures for calculating $S^{\Gamma}(\mathbf{r}_i)$ at the position \mathbf{r}_i in each representation. The blue (yellow) orbitals represent Cu (O_x and O_y) sites, forming a square lattice. Note that Bi atoms reside 6 Å above the Cu sites. The symmetry-dependent structure factor s_{ij}^{Γ} in Eq. (1) in the main text is represented by the sign (and shadows), the bond length between *i* and adjacent site *j* is weighted with the s_{ij}^{Γ} , and the sum is taken over the adjacent sites. For instance, in the case of the B_1 representation, the bond lengths along the *x* direction are summed and subtracted from those along the *y* direction. It should also be noted that E_{11} and E_{22} representations are modified to include contributions from the nearest-neighbor sites at $\mathbf{r}_i \pm \boldsymbol{\delta}_y$, where $\boldsymbol{\delta}_y$ is a unit-cell vector along the *y* direction. Similarly, for E_{22} , to include contributions from the nearest-neighbor sites at $\mathbf{r}_i \pm \boldsymbol{\delta}_x$ where $\boldsymbol{\delta}_x$ is a unit-cell vector along the *x* direction.

drift but with intrinsic local lattice distortions are obtained in $T'(\mathbf{r})$.

APPENDIX B: SI STM MEASUREMENTS

All the SI STM measurements reported in this study are performed at 11 K on the custom-designed SI STM system at the OASIS complex in Brookhaven National Laboratory. A single crystal of the nearly optimally doped Bi₂Sr₂CaCu₂O_{8+ δ} ($T_c = 91$ K, $p \sim 0.17$) is cleaved by the Kapton tape at room temperature in the preparation chamber at the pressure lower than 3×10^{-10} Torr. Prior to perform SI STM measurements on Bi₂Sr₂CaCu₂O_{8+ δ}, the STM tip is characterized on the single crystal Au(111) surface, which is prepared by several cycles of Ar⁺ sputtering and annealing at 450 °C.

APPENDIX C: $C_{4\nu}$ GROUP

There are eight symmetry operations $\{I, C_4, C_2, C_4^3, \sigma_{x'}, \sigma_{y'}, \sigma_d, \sigma_{d'}\}$ and five irreducible representations $\{A_1, A_2, B_1, B_2, and E\}$ in the C_{4v} point group. As shown in Table I, a character table for the C_{4v} point group shows the corresponding matrix presentation for each irreducible representation [18]. The matrix representation describes how variables are transformed by the symmetry operations for each representation. Figure 7 shows examples of how to calculate s_{ij}^{Γ} for each representation.



FIG. 8. Enlarged B_1 , E_{11} , and E_{22} maps from the rectangle regions in Fig. 1.



FIG. 9. (a)–(c) S^{A_1} (**r**_{*i*}), S^{A_2} (**r**_{*i*}), and S^{B_2} (**r**_{*i*}) obtained from topography shown in Fig. 1(a).

APPENDIX D: SETUP CONDITION DEPENDENCE

In order to verify that the bond map revealed in this study is not affected by the formation of the tunneling junction, which is set by the tunneling current, I_{set} , and V_{bias} voltages, V_{bias} , applied between the STM tip and sample, we systematically measure topographs at different setup conditions in the same FOV [Figs. 11(a), 11(d), and 11(g)]. At each setup condition, bond maps are also calculated from the corresponding topographs measured. Figures 11(a), 11(d), and 11(g) show topographs measured at (I_{set} , V_{bias}) = (150 pA, 150 mV), (100 pA, 100 mV)



FIG. 10. (a) A sublattice representation of $S^{A_1}(\mathbf{r}_i)$ from Fig. 8(a). The inset shows the magnitude Fourier transform of (a). There are apparent modulations in this representation, but no well-defined Fourier peaks are observed. (b) A sublattice representation of $S^{A_2}(\mathbf{r}_i)$ from Fig. 8(b). The inset shows the magnitude Fourier transform of (b). (c) A sublattice representation of Fig. 8(c), $B_2(\mathbf{r})$. The inset shows the magnitude Fourier transform of (c). For A_1, A_2 , and B_2 representations, there are no well-defined peaks as shown in the insets in (a)–(c). Thus, these symmetries are preserved. (d)–(f) show $B_1(\mathbf{r}), E_{11}(\mathbf{r})$, and $E_{22}(\mathbf{r})$ within the whole FOV of Fig. 1(a).

and (50 pA, 50 mV), respectively (1 G Ω junction resistance). The bond maps are extracted from Figs. 11(a), 11(d), and 11(g) and shown in Figs. 11(b), 11(e), and 11(h), respectively. Similarly, corresponding histograms of the bond lengths for both the *x* and *y* directions for each setup condition are shown in Figs. 11(c), 11(f), and 11(i). The topographs, bond maps, and histograms measured at the different setup conditions are virtually identical, indicating that the setup effect is negligible within these conditions.

APPENDIX E: COARSE-GRAINING LENGTH DEPENDENCE

In this section, we show how the local fluctuation of the bond length changes with the size of the coarse-graining length L in Eq. (A3) and explain how we choose L. Figures 12(a)-12(c) show the bond-length map obtained

at 4, 6, and 13 Å, respectively. With increasing L, local lattice distortions are smoothed out and washed away when L > 10 (Å); thus, the local fluctuations of the bond length become undetectable. For $L \leq 5$, an additional effect of the missing atoms that contributes to the bond length becomes obvious as weak horizontal contrasts, since they are identified on top of the remnant supermodulations. Thus, $L \sim 6$ (Å) is the smallest size of the filter for precisely characterizing the local lattice distortion. Figures 12(d)-12(f) show distributions of the bond length for L = 4, 6, and 13 Å, respectively, from Figs. 12(a)-12(c), systematically showing a change in their height and width of the distribution. The half width at half maximum at each L is plotted in Fig. 12(g), indicating that, for L > 10, the distribution becomes very sharp such that the large-scale fluctuations of the bond length are smoothed out and not detectable.



FIG. 11. (a),(d),(g) Topographs measured at $(I_{set}, V_{bias}) = (150 \text{ pA}, 150 \text{ mV})$, (100 pA, 100 mV), and (50 pA, 50 mV), respectively. Note that the supermodulations are removed by the Fourier filtrations. (b),(e),(h) Bond maps $B(\mathbf{r})$ extracted from the topographs in (a), (d), and (g), respectively. (c),(f),(i) Histograms of (b), (e), and (h), respectively, for both the *x* and *y* directions. All these maps and histograms are virtually identical; thus, a systematic error due to the junction formation in these measurements is negligible within these setup conditions.



FIG. 12. (a)–(c) The bond-length map obtained at different coarse-graining lengths L (4, 6, and 13 Å). (d)–(f) The distributions of the bond length for both x and y Cu-O-Cu directions from (a)–(c). (g) The half width at half maximum of the distributions at different L from 4 to 30 Å.



FIG. 13. (a) The topograph measured at $V_{\text{bias}} = -1050 \text{ mV}$, I = 30 pA. (b) The bond-length map $B(\mathbf{r})$ obtained from (a). (c) The distributions of the bond length for both x and y Cu-O-Cu directions. (d) The sublattice representation of the $B(\mathbf{r})$, $B_x(\mathbf{r})$. The inset shows the Fourier transform of $B(\mathbf{r})$, $|B_x(\mathbf{r})|$. (e) The sublattice representation of the $B(\mathbf{r})$, $B_y(\mathbf{r})$. The inset shows the Fourier transform of $B(\mathbf{r})$, $|B_x(\mathbf{r})|$. (e) The sublattice representation of the $B(\mathbf{r})$, $B_y(\mathbf{r})$. The inset shows the Fourier transform of $B(\mathbf{r})$, $B_y(\mathbf{q})|$. (f) Line cuts of the $|B_x(\mathbf{q})|$ and $|B_y(\mathbf{q})|$ along the arrows in (d) and (e).

APPENDIX F: HIGH-BIAS TOPOGRAPHY

In this section, we show a result of the analysis that is performed on a topograph measured at high-bias voltage $(V_{\text{bias}} = -1050 \text{ mV})$ and show that the result is highly consistent with those shown in Fig. 1. Figure 13(a) is the topograph of the $Bi_2Sr_2CaCu_2O_{8+\delta}$ ($p \sim 0.17$) measured at $V_{\text{bias}} = -1050 \text{ mV}$, I = 30 pA (courtesy of Professor J. C. Davis, Cornell University). Figure 13(b) shows the bond-length map $B(\mathbf{r})$ obtained from Fig. 13(a) using the same method as in Fig. 1, and the $B(\mathbf{r})$ exhibits heterogeneous bond lengths as seen in Fig. 1(b). Distributions of the $B(\mathbf{r})$ for both x and y Cu-O-Cu directions are shown in Fig. 13(c), and, by fitting with Gaussian, we obtain that widths of the distributions are about 2% for both x and y bond directions. Figure 13(d) [13(e)] are the x(y) bondlength maps, $B_x(\mathbf{r})$ [$B_y(\mathbf{r})$], segregated from Fig. 13(b), exhibiting spatial modulations that propagate along the x (y) Cu-O-Cu directions. The inset in Fig. 13(d) [13(e)] is the Fourier transform of the $B_x(\mathbf{r})$ [$B_y(\mathbf{r})$] showing peaks that correspond to the modulation seen in the $B_x(\mathbf{r}) [B_y(\mathbf{r})]$. Figure 13(f) shows the line cuts of the Fourier transforms extracted along the arrows in the insets in Figs. 13(d) and 13(e), and a peak for both directions is evident. Overall, all these observations are virtually identical to those in Fig. 1, indicating that, concomitantly to Appendix D, the displacement of the atoms we discover is due to neither the systematic error of the setup nor the electronic contributions, but structural origin.

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Correction: Typographical errors were introduced during the production cycle in Appendix D and in the caption of Fig. 11 and have been fixed.