Fréedericksz-Like Transition in a Biaxial Smectic-*A* Phase

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The two main classes of liquid-crystal (LC) phases of rodlike molecules are nematics, where the rods align in the same direction (the nematic director **n**), and smectics, where the rods not only are aligned but also form layers. The electro-optic effects in LC devices that are a backbone in today's display industry mainly use the Fréedericksz transition, which is the bulk reorientation of a surface-anchored nematic by an electric field. Conventional (uniaxial) smectics do not present a Fréedericksz transition, because, due to their layered structure, the director reorientation would distort the layers, which would cost too much energy. In a worldwide ongoing effort to extend the variety of LC compounds suitable for applications in the display industry, bent-shaped molecules have recently raised much attention, since they present multiple new LC phases with unusual properties. In this paper, we report on a structural and electro-optic study of the LC phases of a bent-shaped dimer. On cooling from the isotropic liquid, this compound shows a usual nematic (N), a twist-bend nematic (N_{TB}), and a biaxial smectic-A phase ($\text{Sm}A_b$). Quite surprisingly, contrary to usual smectics, $\text{Sm}A_b$ presents a remarkable electro-optic response, with low (<4 V) voltage threshold, no reorganization of the smectic layers, and low (<1 ms) response time (i.e., 30 times faster than the N phase at higher temperature). We interpret this unexpected electro-optic effect as a Fréedericksz transition affecting the secondary director **m** of the $\text{Sm}A_b$, and we model it by analogy with the usual Fréedericksz transition of the **n** director of the uniaxial N phase. Indeed, a Fréedericksz transition affecting only **m** in this biaxial fluid smectic does not alter its layered structure and costs little energy. From the point of view of applications, thanks to its low relaxation time, this "biaxial" Fréedericksz transition could be exploited in electro-optic devices that require fast switching.

Subject Areas: Condensed Matter Physics, Soft Matter

I. INTRODUCTION

Liquid crystals (LCs) are intermediate phases (mesophases) observed, upon melting, between the crystalline state and the usual (isotropic) liquid state. By definition, LC phases are both fluid and anisotropic. The vast majority of LC compounds known today are rodlike organic molecules, and they offer two main types of mesophases called nematics and smectics. Nematic phases are the most disordered ones, as they have only long-range orientational order of the rodlike molecules; i.e., the molecules tend to align along a common direction called the director, **n**. The most common nematic (N) phase used in electro-optic displays has uniaxial symmetry around **n**, but there are other kinds of nematics, such as chiral ones (the cholesteric and blue phases), the still elusive biaxial nematic, which has lower orientational symmetry, and the spontaneously modulated nematics where the orientational order is periodic in space. Smectics have lower symmetry than usual nematics, because they have, in addition to orientational order, long-range positional order of the centers of mass of the LC molecules in at least one direction of space, resulting in a layered structure. The most common smectic, called smectic A ($\text{Sm}A$), is a uniaxial phase where the director is parallel to the normal to the layers and the molecules have no long-range positional order within the layers, which are, therefore, fluid. In other smectic phases, the molecules may be tilted with respect to the normal to the layers ($\text{Sm}C$), or they may have long-range positional order within the layers ($\text{Sm}B$) or even both ($\text{Sm}G$).

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The Fréedericksz transition (FrTr) is arguably the most important electro-optic feature of nematic liquid crystals, as it is a basis of their widespread applications in display technology. This transition, discovered long ago [1], involves reorientation of the nematic optic axis (the director) by an applied electric field above a threshold value. It results from the competition of the electric torque acting on the bulk anisotropic material with the surface torque induced, at the sample boundaries, by the anchoring of **n** in a direction perpendicular to that imposed by the field. A similar FrTr has not yet been reported in smectics, because the twist and bend deformation of the director is topologically forbidden due to their layered structure. Instead, a small electro-optic effect, called a "ghost transition" [2,3], can barely be observed and is of no practical use for applications. Therefore, the FrTr is usually regarded as a distinctive property of the nematic LC phase.

The search for new LC compounds for electro-optic applications, in the course of the past 20 years, has led to the development of bent-shaped, bananalike, molecules that present a rich variety of LC phases, including many different new types of smectics showing spontaneous symmetry breaking and ferroelectricity [4–15]. Among these compounds, bent-shaped "dimer" molecules have recently gained particular interest worldwide, because they often display new types of nematic phases, namely, the "twist-bend" nematic [16–26] (N_{TB}) and the still elusive "splay-bend" nematic (N_{SB} [17,24,27–31]). In addition to these modulated nematics, bent-shaped dimer molecules also show very original smectic phases with structures characterized by the intercalation of the monomer moieties of the molecules [7,9–13,32–42].

In this context, a bent-shaped dimer, 1,7-bis[6-(4-hexyloxybenzoyloxy)naphthalene-2-yl]heptane (labeled BNA-76), is reported [26] to present a very peculiar LC phase, called M_X, whose structure and properties could not be determined because it is monotropic and easily crystallizes, thus precluding any electro-optic and structural investigations. In this work, using mixtures of BNA-76 with a small fraction of a rodlike nematic compound as a means to hinder crystallization, we show that the M_X phase is actually a biaxial Sm*A* (SmA_b) comprised of completely intercalated layers. Moreover, quite surprisingly, this biaxial smectic phase shows a nematiclike electro-optic effect, with rather fast (<1 ms) relaxation times and low (approximately 4 V) field threshold values. We interpret and model these observations as the signature of a FrTr of the secondary director **m** that defines the direction of the width of the lathlike molecule. Indeed, a transition affecting only the secondary director of a biaxial Sm*A* phase, called "biaxial Fréedericksz transition" (BFrTr) hereafter, can take place, since it does not distort the layered structure.

II. EXPERIMENTAL RESULTS

As many other bent-shaped dimers, BNA-76 (Fig. 1) shows a transition to the N_{TB} phase on cooling from the
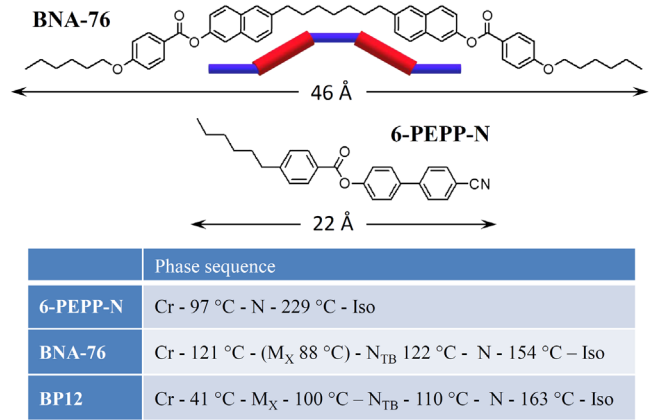

FIG. 1. Molecular structure and phase sequence of the LC compounds. For the BNA-76 bent-shaped dimer, a simplified sketch of the average conformation of the molecule is also shown, with the electron-rich conjugated parts in red and the alkyl and alkoxy chains in blue.

	Phase sequence
6-PEPP-N	Cr - 97 °C - N - 229 °C - Iso
BNA-76	Cr - 121 °C - (M_X 88 °C) - N_{TB} 122 °C - N - 154 °C – Iso
BP12	Cr - 41 °C - M_X - 100 °C – N_{TB} - 110 °C - N - 163 °C - Iso

N phase [26]. On further cooling, when the N_{TB} phase is supercooled by about 40 °C, another transition takes place, to a mesophase, labeled M_X, whose structure remains so far unknown. Its optical textures indicate that it is either a smectic or a modulated nematic (N_{TB} or N_{SB}) phase. However, because the M_X phase is strongly supercooled, it crystallizes very fast in bulk samples, preventing its identification by x-ray scattering. To avoid crystallization, we prepare mixtures of BNA-76 with the rod-shaped nematogen 4′-cyano[1, 1′-biphenyl]-4-yl 4-hexylbenzoate (6-PEPP-*N*). This compound is selected for its high *N-I* transition temperature, good miscibility with BNA-76, high dielectric anisotropy, and molecular size similar to that of the BNA-76 "monomer" (Fig. 1). Here, we focus on the "BP12" mixture (with 12 wt % of 6-PEPP-*N*) whose phase sequence (Fig. 1) is similar to that of pure BNA-76 but with an N_{TB}-M_X transition temperature higher by 30 °C and, most importantly, a much lower crystallization temperature (see the Appendix A). Therefore, the M_X phase of BP12 is thermodynamically stable at temperatures between 80 °C and 100 °C, thus allowing for the x-ray and electro-optic investigation of its structure and properties.

A. Structural study

X-ray scattering patterns (Fig. 2) of a BP12 sample in the M_X phase, with the director aligned by a magnetic field, show two sharp reflections at small scattering angles, at $q_0 = 0.28$ Å^{-1}. These resolution-limited reflections reveal the existence of a layered structure, i.e., a smectic phase. As expected, these reflections are replaced by wider and weaker diffuse scattering spots in the *N* and N_{TB} phases (see Appendix B, Fig. 15). The smectic period of the M_X phase, $d_0 = 2\pi/q_0 = 22.5$ Å, is approximately half the length ($L = 46$ Å) of the conformation of the BNA-76 molecule estimated by the HyperChem package using
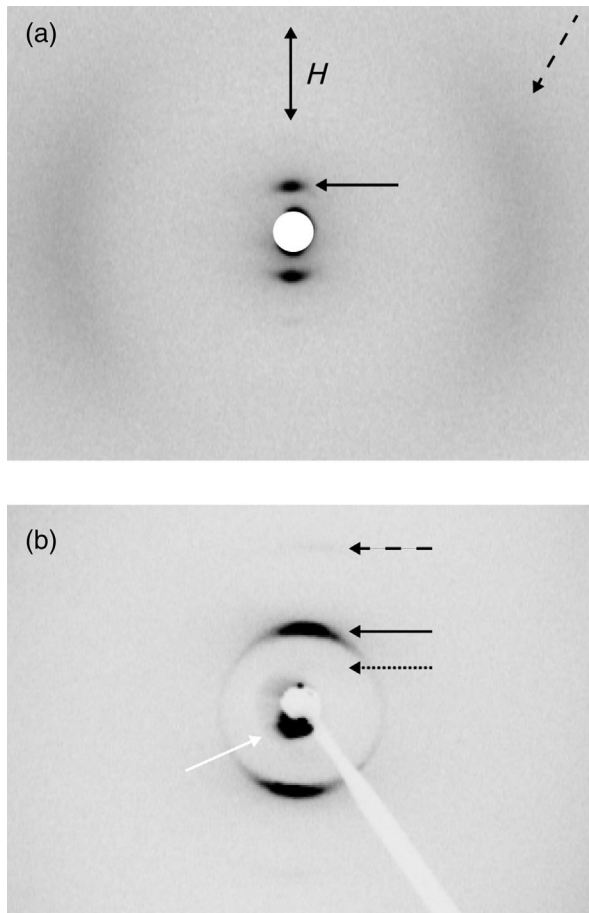
FIG. 2. X-ray scattering patterns of a BP12 sample in the M_X phase ($T = 97\,°C$) aligned by a 1.7 T magnetic field H (double-headed arrow). (a) With a 60 mm sample-to-detection distance. The solid arrow points to one of the smectic reflections at small angles, whereas the dashed arrow points at the wide-angle diffuse ring. (The white disk at the center represents the beam stop.) (b) With a 120 mm sample-to-detection distance. The solid arrow points to one of the smectic reflections, the dashed arrow to the barely observable second-order smectic reflection, and the dotted arrow to the absence of any scattering at wave vector $q_0 = 2\pi/L$, where L is the BNA-76 molecular length. (The white arrow points to parasitic scattering around the beam stop.)
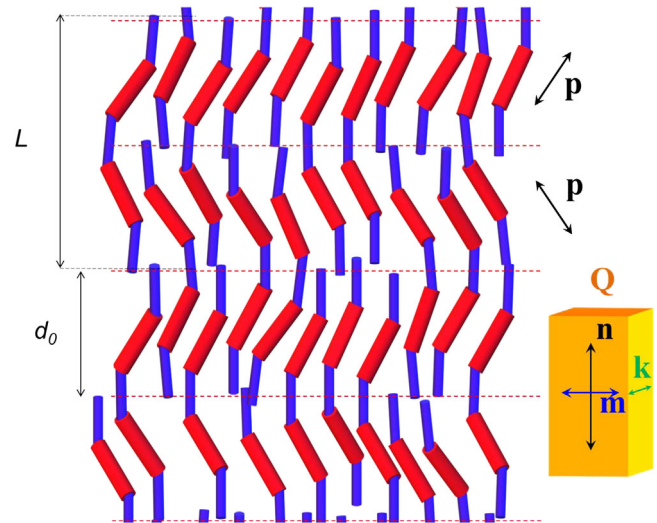

FIG. 3. Sketch of the structure of the M_X phase. The monomer units form smectic liquid layers with thickness $d_0 \approx L/2$. In each layer, the main axes **p** of the monomers are tilted in the same direction, and the sign of the tilt alternates from one layer to the next. The dimers span two adjacent layers and form an orthogonal intercalated smectic phase. The orientational order tensor **Q** of the dimers (shown in orange) is biaxial, with primary director **n** oriented along the layers normal and secondary director **m** parallel to the projection of **p** on the layer plane. (**k** is the third director, which is perpendicular to both **n** and **m**.)

the AM1 Hamiltonian [43]. No scattering is detected around $q_0/2 = 0.14$ Å^{-1}, which would correspond to the whole length of the molecule, and, therefore, the dimers are completely intercalated. Higher-resolution measurements of unaligned samples confirmed this conclusion (see Appendix B, Fig. 16).

In addition, a diffuse scattering ring, observed at wide angles, reveals the liquidlike character of the smectic layers. The maximum of this ring lies in the direction perpendicular to that of the smectic modulation, which suggests that the director is parallel to the normal to the layers, as in a smectic-A (SmA) phase. However, the angular extension of the ring is very large, and, moreover,

it increases with decreasing temperature (see Appendix B, Fig. 17). This last feature is quite unexpected for a common SmA phase and suggests a tilt of the mesogenic cores in the layers, like in SmC or smectic C_A (SmC_A) phases.

Nevertheless, because $d_0 = L/2$, the long axis of the dimer is perpendicular to the layers, as confirmed by optical microscopy observations of the mutual orientation of the optic axis and the layers (see below). Consequently, considering the dimer molecules, the phase is smectic A, with **n** parallel to $\mathbf{q}_0$. However, the x-ray scattering results clearly indicate that the M_X phase is an intercalated smectic, with the monomers tilted with respect to the normal to the layers, as in the intercalated phases already reported for other bent-shaped dimers [7,9–13,32–36,38–42]. There are several structures, differing by the azimuthal correlations of the monomers, which are consistent with the x-ray data (see Appendix F), but the one that agrees best with our whole set of results (see the next sections) is illustrated in Fig. 3. The layer thickness (i.e., the period of the mass density and electron density waves) is $d_0 \approx L/2$, meaning that the repeat unit of the smectic structure is not the dimer molecule but its mesogenic subunit (or "monomer"). Therefore, each dimer molecule must span two adjacent monomer layers. This intercalated structure is similar to that proposed before for the intercalated smectic phase of several bent-shaped dimers [9–11, 13,33,34,41,44–49]. The monomers have their main axes **p** tilted in the same way in each layer, but the tilt direction

alternates from one layer to the next, as in an anticlinic smectic C (SmC_A) [4,42,49,50] (see Sec. III and Appendix F).

B. Optical and electro-optical study

1. Texture observations with or without a field

For optical and electro-optical experiments, we use sandwich cells with substrates treated to provide planar alignment (see Appendix A for details). The cell is filled by capillarity with the LC mixture in the isotropic phase and then cooled to the nematic phase, resulting in uniform planar alignment of the nematic director **n** parallel to the rubbing direction **r**. Slowly cooling the sample across the N-N_{TB} sharp transition, we observe at $T = 108.9\,°C$ the growth of large monochiral N_{TB} domains with optical axis **N**, also parallel to **r**, and alternating handedness of the chirality (Figs. 18 and 19). Upon further cooling, to avoid the typical stripe instabilities of the N_{TB} phase [see Fig. 20(a)], the sample is annealed using temperature oscillations and simultaneous voltage bursts (see Appendix C for details).

The N_{TB}-M_X phase transition starts at $T \approx 103.5\,°C$ and, contrary to the sharp N-N_{TB} transition, has a very large (about $5\,°C$) temperature range of biphasic coexistence. Starting from a well-annealed N_{TB} texture and slowly decreasing the temperature (approximately $0.1\,°C/$min), we are able to grow highly uniform single domains of the M_X phase [Fig. 4(a)]. These domains are birefringent, and their slow (higher-index) axis **N** has the same orientation as that in the surrounding N_{TB} regions, **N**‖**r**. However, they can be easily recognized in polarized light [Figs. 4(a) and 4(b)], because their birefringence is larger
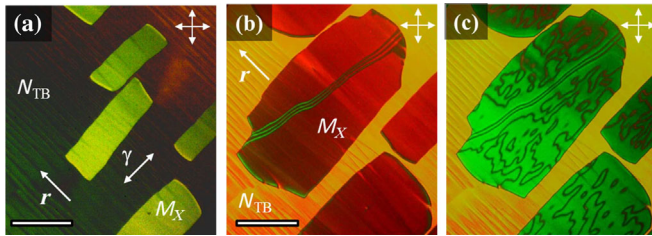

FIG. 4. Coexistence of uniform single domains of the M_X and N_{TB} phases of the BP12 mixture (planar cell, $d = 9.8\ \mu m$, scale bar $100\ \mu m$). The optic axes of both phases are approximately parallel to the rubbing direction **r**. The cell is viewed between crossed polarizers, using a Berek compensator with slow axis $\gamma \perp \mathbf{r}$. (a) Growth of uniform M_X domains within the N_{TB} at $T = 102.4\,°C$. (b),(c) M_X and N_{TB} domains at $T = 100.7\,°C$, without a field (b) and when a biaxial Fréedericksz transition (c) is induced by an ac field ($U_{rms} = 9$ V, $f = 6$ kHz) applied along the cell normal. The birefringence of the N_{TB} [yellow regions in (b) and (c)] is independent of the field. The birefringence of the M_X phase varies with the field, from its low zero-field value [brown-red domains in (b)] to a high value above the transition [green domains in (c)]. The defect walls that separate the twin domains in (c) appear as thick dark lines.

than that of the N_{TB} phase. In the biphasic temperature range, the M_X single domains progressively grow as the temperature is slowly decreased, until the whole sample turns into the M_X phase. At constant temperature, the coexisting M_X and N_{TB} regions are at equilibrium (Fig. 21), and the proportions of the two phases depend only on the temperature, in a reversible way. The analysis of the textures shows that the primary director **n** of the M_X phase is parallel to the helix axis **h** of the N_{TB} phase and that the smectic layers of the M_X domains remain parallel to the N_{TB} pseudolayers (and perpendicular to **r**).

Surprisingly enough for a smectic phase, the birefringence of the M_X domains changes when an electric field is applied to the cell [Figs. 4(b) and 4(c)]. Several features of this electro-optic effect are reminiscent of the classical FrTr in the N phase (i.e., the transition observed in exactly the same cell above the N_{TB}-N transition temperature). The transition has a well-defined voltage threshold $U_c \sim 4$ V rms, which is independent of the cell thickness. U_c is rather low and is not much higher than the FrTr threshold (approximately 2 V rms) in the nematic phase. Apart from the variation in birefringence, there is no texture change at $U > U_c$, which clearly shows that there is no reorientation of the smectic layers at the transition. This result is confirmed by the reversibility and absence of hysteresis of the effect upon field removal. Finally, the orientation of the slow axis of the M_X domain does not change when the field is applied [Figs. 22(b) and 22(d)], which is also similar to the nematic FrTr.

However, the voltage dependence of the birefringence is strikingly different in the M_X phase because the birefringence *increases* with U [for quantitative measurements, see Fig. 6(a) in the next section], whereas it *decreases* toward zero with increasing voltage for the nematic FrTr. Therefore, qualitatively, this behavior proves that, unlike the nematic case, the electro-optic transition in the M_X phase is not related to a rotation of the primary director **n** but is instead due to a rotation of the order-parameter tensor **Q** around **n**. In other words, this behavior is a clear sign that the M_X phase is *biaxial* and that the birefringence variation is related only to an out-of-plane rotation of the secondary macroscopic director **m**.

To study the field-induced transition in the M_X phase in more detail, the cell is observed between crossed polarizers, with a wideband interferential filter centered at $\lambda = 546$ nm and a Berek compensator adjusted to give optimal contrast between the different coexisting phases and textures [Figs. 4(b) and 4(c)]. In these conditions, the N_{TB} phase appears yellow, the low-birefringence (LB) M_X state, with in-plane orientation of both **n** and **m**, appears brown-red, and the high-birefringence (HB) M_X state, with out-of-plane tilt of **m**, appears green. Without the field [Fig. 4(b)], the coexisting N_{TB} and M_X regions are uniform, both with the slow axis parallel to **r** but with slightly different birefringence. The M_X region is mainly in the LB state,

except for occasional defects that are walls with π reorientation of **m** on the boundary surfaces. The π walls separate two domains with antiparallel surface anchoring of **m**. This doubly degenerate orientation of **m**, which is parallel to the surface and perpendicular to **r**, corresponds to anchoring-energy minima. The color in the middle of the π wall is green, because **m** is perpendicular to the surface (in the whole bulk of the wall) and the birefringence is higher, as in the HB state observed under the field.

When the field is applied [Fig. 4(c)], the N_{TB} regions remain unchanged, but the M_X domains undergo a transition to the HB state which is recognized by its green color. In the HB state, in the middle of the cell, **m** is tilted away from the surface plane. Because of the double degeneracy of the tilt, there appear two topologically different HB states with the same energy. This situation is similar to the twin domains observed above the FrTr in nematic samples with no pretilt [51]. In the walls between the domains, the tilt angle of **m** decreases to zero in the middle of the wall and then increases again with the opposite sign. Therefore, the birefringence in the wall is lower (in the middle, it is the same as in the LB state), and the walls are easily distinguished in the images as thick darker lines separating the twin domains. When the field is suddenly applied, the fast reorientation process creates twin domains. Then, due to the higher energy of the walls, the smaller twin domains progressively shrink and disappear, and the texture of the M_X region coarsens (see Supplemental Material [52] for video of the coarsening process). At the end of this coarsening process, only a few twin domains remain in the M_X region. They are stabilized by the occasional pinning of the walls on the cell surfaces and the N_{TB}-M_X interface.

The most direct way to prove the biaxiality of the M_X phase consists in observing and measuring a finite birefringence in a homeotropically aligned sample (i.e., with **n** parallel to the cell normal and to the observation direction). Despite the notorious difficulty of producing a surface-induced homeotropic alignment of bent-shaped dimers, we achieve a quasihomeotropic alignment in our cells by applying a strong field ($E = 17$ V/μm; see Appendix C for details). Figure 5 shows the growth under a field of large and uniform quasihomeotropic M_X domains within a quasihomeotropic N_{TB} region. These domains are indeed birefringent, and their slow axis is parallel to **r**. Their birefringence ($\Delta n \approx 0.02$) is much smaller than that of a planar M_X domain ($\Delta n \approx 0.14$) but is significantly larger than that of the adjacent quasihomeotropic N_{TB} region. The birefringence is independent of the temperature and the field strength (as long as it remains strong enough to maintain the quasihomeotropic texture). During the growth of the M_X domains under a strong field, we do not observe any texture change or instability, which suggests that **n** in the M_X phase is parallel to the helix axis of the quasihomeotropic N_{TB} phase surrounding it. Then, the M_X smectic layers are parallel to the cell surfaces and keep the
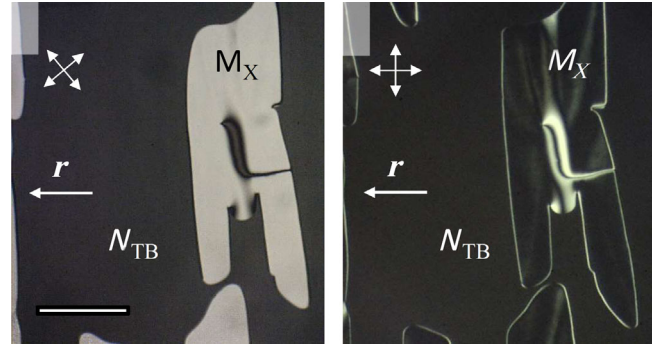

FIG. 5. Growth of a homeotropic M_X domain within the N_{TB} under an electric field ($U_{rms} = 25$ V, $d = 1.4$ μm, $T = 100\,°C$). The observation of birefringence while rotating the crossed polarizers (white double-headed arrows) reveals that the domain is biaxial, with **m** parallel to **r**. (Scale bar 100 μm.)

orientation of the N_{TB} pseudolayers. The rotation of the cell between crossed polarizers reveals (Fig. 5) that **m** is parallel to the cell surface (and the smectic layers) and to the rubbing direction in the bulk of the M_X phase. To conclude on this point, the observation of birefringence while examining the cell along **n** is a direct proof of the lack of revolution symmetry around **n**, i.e., that the M_X phase is biaxial. This conclusion is further confirmed by additional analyses of texture instabilities (see Appendix C).

2. Birefringence measurements

These qualitative proofs of the biaxiality of the M_X phase are quantitatively confirmed by the measurement (see Appendix A) of the birefringence with or without applied fields. The temperature dependence of the birefringence measured in all the phases of the BP12 mixture is shown in Fig. 6(a). For $T > 140\,°C$, the birefringence has the usual behavior expected for a nematic phase. Upon decreasing the temperature, it increases rapidly and follows the Haller law [53] $\Delta n = \Delta n_0 (1 - T/T^*)^\beta$, with $\Delta n_0 = 0.253$, $\beta = 0.20$, and $T^* = 160.9\,°C$, corresponding to the N-I transition temperature. However, below $140\,°C$, still in the nematic phase, the values of Δn are significantly smaller than those extrapolated from the Haller-law fit [red line in Fig. 6(a)]. Such deviation is typical for the bent-shaped dimers that present the N_{TB} phase [31,54–57]. Δn decreases further in the N_{TB}, which is due to the temperature variation of the heliconical tilt angle, as already reported for other N_{TB}-forming compounds [31,54,55,57].

When the sample is cooled into the M_X-N_{TB} coexistence range, the birefringence of the N_{TB} regions remains unchanged, as expected. Without the field, the birefringence of the M_X domains, $\Delta n = n_{nn} - n_{mm}$, is significantly higher than that in the N_{TB} phase, $\Delta n = n_\parallel - n_\perp$. It is closer to the extrapolated Haller fit, but it also remains practically constant throughout the whole temperature range of coexistence (the weak slope of the curves is an
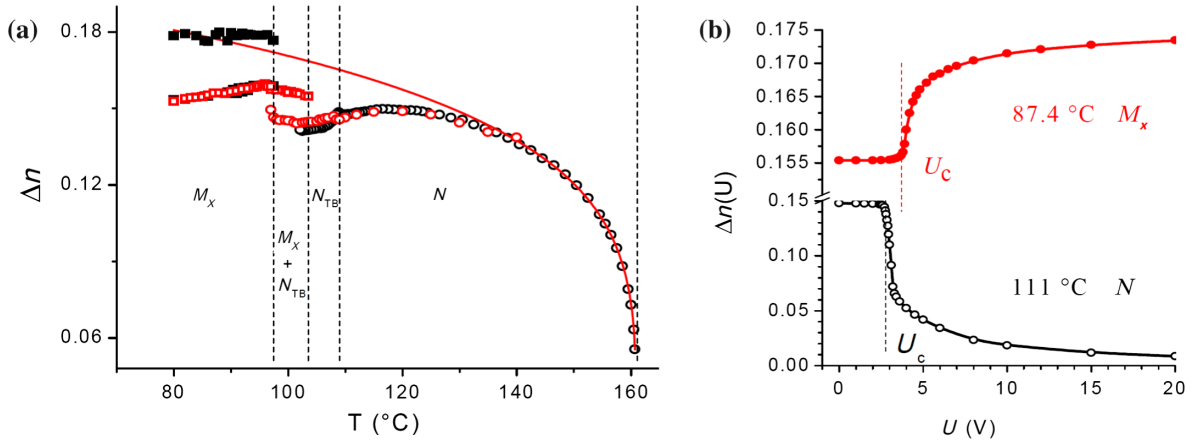
FIG. 6.   Temperature and field dependence of the birefringence of the BP12 mixture. (a) $\Delta n(T)$ measured in different phases and for different sample alignments. The black symbols show the data obtained using PMT detection, whereas the red symbols show the data measured with the image-mapping technique. The circles show the data $\Delta n = n_{\parallel} - n_{\perp}$, obtained in the uniaxial N and N_{TB} phases or regions, whereas the open squares show the data $\Delta n = n_{nn} - n_{mm}$, obtained in the M_X phase without a field. The full squares show the data $\Delta n = n_{nn} - n_{kk}$ in the M_X phase, obtained by extrapolation of the results measured under a field. The red line shows the birefringence expected by extrapolation of the Haller fit [53] of the data in the range 140 °C–160 °C. (b) $\Delta n(U)$ in the N (black symbols) and M_X (red symbols) phases.

artifact due mainly to light scattering by the heterogeneous biphasic texture). On further cooling, when the last N_{TB} domains disappear, the birefringence of the M_X phase decreases slowly, indicating a weak temperature dependence of its biaxial order-parameter tensor.

Under the field, in the N and N_{TB} phases, the only possible reorientation is the out-of-plane rotation of the optic axis $\mathbf{N}$, which decreases the birefringence measured at normal light incidence [Fig. 6(b), black symbols]. The field effect is more subtle in the biaxial M_X phase. When cooling the sample from the N to the M_X phase under a strong field, the M_X domains grow with $\mathbf{n}$ parallel to the field (see Fig. 5) and the smectic layers parallel to the surface, and the

observed birefringence, $\Delta n = n_{mm} - n_{kk}$, is small. On the contrary, when planar M_X domains are grown without the field, $\mathbf{n}$ is oriented parallel to $\mathbf{r}$, and this orientation defines that of the smectic layers. Under realistically strong fields, $\mathbf{n}$ keeps its initial orientation, but, due to the out-of-plane reorientation of $\mathbf{m}$, the birefringence increases with the field. At high applied voltage U, $\mathbf{m}$ aligns parallel to the field and $\Delta n(U)$ saturates to $n_{nn} - n_{kk}$ [Fig. 6(b)]. The extrapolated birefringence value is close to that expected by the Haller fit [Fig. 6(a)], and its temperature variation is very weak.

The characteristic times of the optical response of the BFrTr in the M_X phase are compared with those of the usual
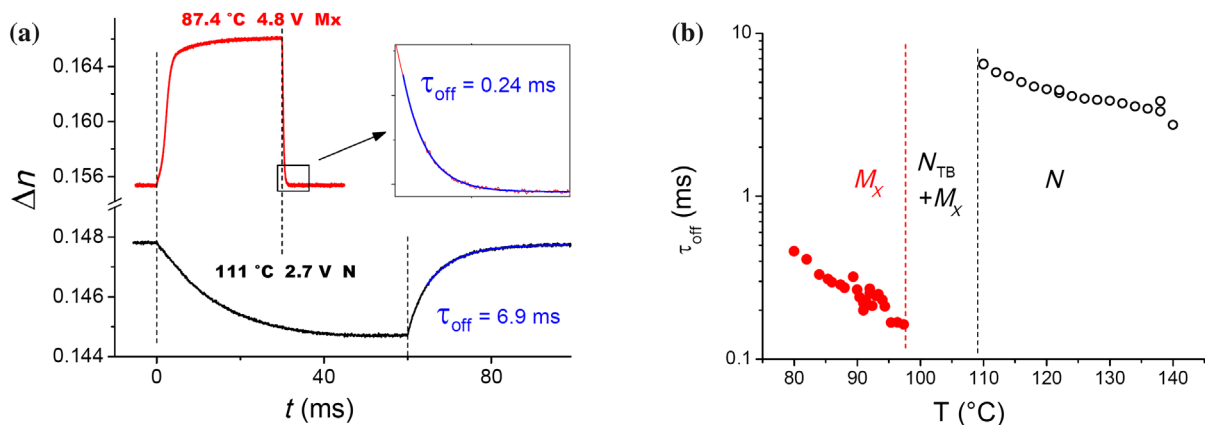

FIG. 7.   Time evolution of the optical response during the Fréedericksz transition in the N (FrTr, black lines and symbols) and M_X (BFrTr, red lines and symbols) phases of the BP12 mixture (cell thickness $d = 1.4$ μm). (a) Optical response to dc pulses ($U > U_c$; the dashed straight lines mark the start and the end of the pulse). The relaxation time τ_{off}, which depends only on the cell thickness, is much faster in the M_X phase. Fits of the data with an exponential relaxation law are shown in blue. (b) Temperature dependence of τ_{off} in the N (open black symbols) and M_X (full red symbols) phases.
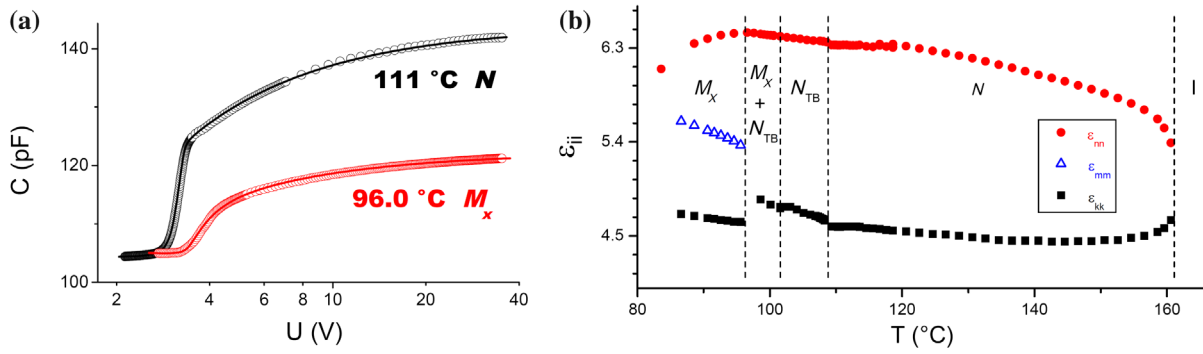
FIG. 8. Dielectric data of the BP12 mixture (cell thickness 9.8 μm). (a) Cell capacitance versus applied voltage in the N phase (black symbols) and in the M_X phase (red symbols). The solid lines show the best fits of the data with the theoretical model (see Appendix E). (b) Temperature dependence of the dielectric tensor eigenvalues measured in the uniaxial N and $N_{\rm TB}$ phases and in the biaxial M_X phase.

FrTr in the N phase in Fig. 7(a). The on time $\tau_{\rm on}$ is proportional to E^{-2} and, therefore, becomes very small for large fields. On the contrary, the off time $\tau_{\rm off}$ of the response when the field is switched off is independent of the field. The temperature dependence of $\tau_{\rm off}$ in the N phase [Fig. 7(b)] is typical, as it just increases from approximately 3 ms at high temperature to approximately 8 ms close to the $N_{\rm TB}$ phase. However, quite strikingly, $\tau_{\rm off}$ of the BFrTr in the M_X phase is much faster, by almost 2 orders of magnitude when extrapolated to the same temperature.

C. Dielectric experiment

The dielectric technique is complementary to the birefringence measurements. The cell capacitance $C(U)$, measured with various surface- and field-induced alignments (see Appendix C), provided all three eigenvalues, ε_{kk}, ε_{nn}, and ε_{mm}, of the dielectric tensor ε (see Appendix E). The capacitance, measured in the N and M_X phases, varies smoothly and reversibly with the voltage [Fig. 8(a)], through the field-induced FrTr and BFrTr, respectively. Fits of these data with theoretical models provide two of the ε components: $\varepsilon_\perp$ and $\varepsilon_\parallel$ in the N phase [58–60] and ε_{kk} and ε_{mm} in the M_X phase. The third component, ε_{nn}, of the M_X phase is measured in a field-induced quasihomeotropic domain. In the $N_{\rm TB}$ phase, the components $\varepsilon_\perp = \varepsilon_{kk} = \varepsilon_{mm}$ and $\varepsilon_\parallel = \varepsilon_{nn}$ are measured at fixed voltages because of the absence of FrTr in this phase. The temperature dependence of the dielectric tensor components across the three mesophases is presented in Fig. 8(b), which provides further direct evidence for the biaxiality of the M_X phase: ε has only two different components, $\varepsilon_\perp$ and $\varepsilon_\parallel$, in the uniaxial N and $N_{\rm TB}$ phases, whereas $\varepsilon_\perp$ is split in two different components, $\varepsilon_{kk} \neq \varepsilon_{mm}$, in the M_X phase. The dielectric data are further exploited in the next section.

III. DISCUSSION

A. Nature of the M_X phase

The whole set of x-ray scattering, dielectric, and electro-optic data shows that the M_X phase is an orthogonal smectic with liquidlike layers, i.e., a smectic A, with primary director $\mathbf{n}$ parallel to the smectic wave vector $\mathbf{q}_0$ (see Appendix F for more details). During the $N_{\rm TB} - M_X$ phase transition, $\mathbf{q}_0$ appears parallel to the helix axis $\mathbf{h}$ of the $N_{\rm TB}$ modulation. In the M_X single domains, $\mathbf{n}$ remains parallel to $\mathbf{h}$ and, therefore, parallel to $\mathbf{q}_0$. Finally, since the order tensor of the N and M_X phases is uniform in the absence of a field, the slowest axis $\mathbf{N}$ in these phases has the same orientation and is parallel to $\mathbf{n}$. Qualitatively, the BFrTr also confirms that the M_X phase is SmA and not SmC. Indeed, the variation of $\Delta n(U)$ occurs without reorientation of the slow axis $\mathbf{N}$, because the order tensor $\mathbf{Q}$ rotates around its main axis $\mathbf{n}$, which remains parallel to $\mathbf{q}_0$. Moreover, the variation of both $\Delta n(U)$ and $\Delta\varepsilon(U)$ during the transition clearly indicates that the order tensor $\mathbf{Q}$ is biaxial.

In principle, various microscopic structures differing by the in-layer order of the tilted monomers are possible for the intercalated SmA phase of bent-shaped dimers. If the tilt direction were continuously degenerated, the monomers would form a de Vries–SmA-like structure, but it is uniaxial and, therefore, cannot explain the observed biaxiality of the M_X phase. Another possible structure has doubly degenerated tilt of the monomers within each smectic layer (Fig. 25). However, although this structure is biaxial, it should be ruled out, because it does not minimize the interaction energy of the monomers (see Appendix F for more details). Consequently, we identify the structure of the M_X phase as that shown in Fig. 3, which is an intercalated biaxial SmA phase of the bent-shaped dimers with anticlinic SmC_A organization of the monomer units. One may find this structure quite reminiscent of that of the antiferroelectric Sm$A_{\rm PA}$ phase [14,15,42]. However, the Sm$A_{\rm PA}$ phase is not intercalated, and its fluid smectic layers are just formed by the bent-shaped molecules oriented with their main axis normal to the layer and their transverse dipoles parallel to one another. Therefore, the phase is a biaxial smectic A with polar in-layer order. The direction of the polarization alternates from one layer to the next, and, hence, the phase is antiferroelectric. Consequently, in each

layer, the polarization vector aligns parallel to an applied moderate in-plane electric field, resulting in the macroscopic polarization of the phase.

In contrast, the M_X phase displays no polar response under the applied field. Moreover, the polar electroclinic effect [23] observed in the N_{TB} completely disappears in the M_X phase, which behaves as a purely dielectric material. This result is due to the complete intercalation of the dimers that locks the polarization of any monomer layer to those of its first neighbors, thus effectively suppressing any ferroelectric response, as already recognized in Ref. [10]. So, hereafter, following the notations of Ref. [42], we refer to the M_X phase as SmA_b. (Note that the SmA_b phase is expected and has been reported [12] for strongly biaxial boardlike molecules; it is sometimes also labeled SmC_M [51], in honor of William McMillan.)

B. Elasticity of the SmA_b phase

Our explanation of the striking electro-optic behavior of SmA_b relies on its biaxial physical properties and elastic response. As for any smectic phase, there are two contributions to the elastic energy of SmA_b. The first one is related to the orientational order of the phase. For a uniaxial phase, this "nematiclike" elasticity is given by the usual Frank expansion of the distortion energy in the director derivatives (here and in the following, we neglect the surfacelike terms, which are not relevant in our case) [61]:

$$f^n = \frac{1}{2}[K_{11}^n(s^n)^2 + K_{22}^n(t^n)^2 + K_{33}^n(b^n)^2]. \quad (1)$$

Here, the vectors $s^n = n(\nabla \cdot n)$ and $b^n = n \times (\nabla \times n)$ and the pseudoscalar $t^n = n \cdot (\nabla \times n)$ describe the main distortion modes, respectively, splay, bend, and twist, of the (primary) nematic director n and the K_{ii}^n are the Frank elastic constants related to the derivatives of n. When the orientational order is biaxial, as in the SmA_b phase (and in biaxial nematics), the distortion energy is much more complex, because it has additional terms, which are related to the derivatives of the secondary director m:

$$f^m = \frac{1}{2}[K_{11}^m(s^m)^2 + K_{22}^m(t^m)^2 + K_{33}^m(b^m)^2], \quad (2)$$

where $s^m = m(\nabla \cdot m)$, $b^m = m \times (\nabla \times m)$, and $t^m = m \cdot (\nabla \times m)$ describe the splay, bend, and twist of m, respectively, and the K_{ii}^m are the corresponding Frank elastic constants. Additional cross terms between m and n and their gradients result in a very complex biaxial-nematic elastic energy containing 12 independent terms (for achiral molecules) [62,63].

The second contribution to the elastic energy of the SmA_b phase (as of any other smectic) is a "smecticlike" term related to the distortion of the positional ordering, i.e., to the compression and deformation of the layers [51]. Actually, this energy is prohibitively large compared to the

moderate electric energy involved in our experiments, and such distortion of the layers should be negligible. Indeed, our observations confirm that no reorganization of the layers occurs in the SmA_b phase, even when strong fields are applied.

The rigid smectic elasticity of SmA_b substantially simplifies the elastic response to the electric field. Indeed, the bend and twist of n are forbidden, because SmA_b is an orthogonal smectic [51]. Then, the only possible distortion of n is the splay. However, it is coupled with the bend of the layers, and its variation requires reorganizing the layers at large scale, which is energetically costly. No such reorganization is observed in the SmA_b domains that we study, which confirms that n remains uniform during the BFrTr. Consequently, the elastic energy of SmA_b just reduces to Eq. (2), which provides an analogy between the elastic distortions of the n director in the N phase and those of the m director in the SmA_b phase (Fig. 9).

Moreover, the boundary conditions $m \perp r$ are uniform in the plane of the cell, and the field is applied along the cell normal, $E \| z$. Therefore, the distortion of m is one-dimensional [$m = m(z)$] and planar [$m(z) \perp r$], with only splay and bend but no twist (Fig. 10). This distortion geometry is again an exact analog of the usual FrTr in the splay geometry [51]. Under the field, the free energy in the N and SmA_b phases becomes, respectively,

$$f^n = \frac{1}{2}\left\{[K_{11}^n \sin^2\theta + K_{33}^n \cos^2\theta]\left(\frac{d\theta}{dz}\right)^2 + D \cdot E\right\}, \quad (3)$$

$$f^m = \frac{1}{2}\left\{[K_{11}^m \sin^2\theta + K_{33}^m \cos^2\theta]\left(\frac{d\theta}{dz}\right)^2 + D \cdot E\right\}, \quad (4)$$

where $\theta = \theta(z)$ is the angle between the relevant director (respectively, n or m) and the z axis, $E = [0, 0, E(z)]$ is the electric field in the cell, which varies with z due to the director distortion, and the relevant D_z component of the dielectric displacement D is $D_z = \varepsilon_{zz}E_z = $ const, where $\varepsilon_{zz} = \varepsilon_\perp \sin^2\theta + \varepsilon_\| \cos^2\theta$ in the uniaxial N case and $\varepsilon_{zz} = \varepsilon_{kk}\sin^2\theta + \varepsilon_{mm}\cos^2\theta$ in the SmA_b case.

Thanks to this analogy, we can easily describe the behavior of the SmA_b phase under a field by using the well-known solutions [58–60,64–66] for the FrTr in the N phase. The voltage threshold of the transition, U_c^i, is given by

$$U_c^n = \pi\sqrt{\frac{K_{11}^n}{\varepsilon_0(\varepsilon_\| - \varepsilon_\perp)}}; \qquad U_c^m = \pi\sqrt{\frac{K_{11}^m}{\varepsilon_0(\varepsilon_{mm} - \varepsilon_{kk})}} \quad (5)$$

(where $i = n, m$ indicates the director involved in the transition).

Here, ε_0 is the vacuum permittivity, and $(\varepsilon_\| - \varepsilon_\perp)$ and $(\varepsilon_{mm} - \varepsilon_{kk})$ are the effective dielectric anisotropies in the N and SmA_b phases, respectively. Up to $U = U_c^i$, the field-induced torque on the relevant director is weaker
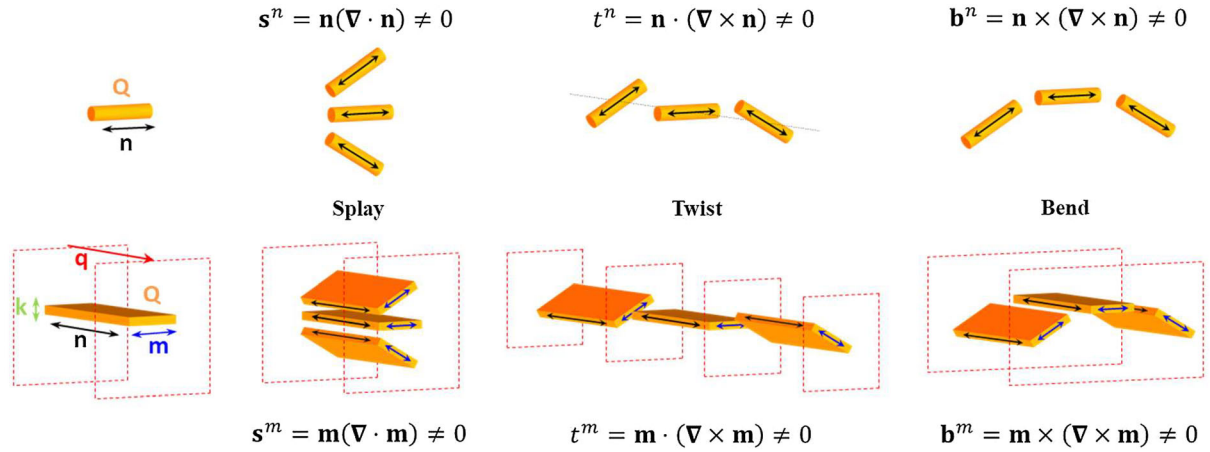
FIG. 9. Schematic representation of the nematic directors (**n**, **m**, and **k**), the order parameter tensor (**Q**), and the main distortion modes in the N and SmA_b phases. In the uniaxial N phase (top row), the only relevant distortions are those of the main director **n**. In the SmA_b phase (bottom row) and at fixed structure of the smectic layers (shown in red), the only relevant distortions are those of the secondary director **m**.

than the surface-anchoring torque, and the cell texture remains undistorted. Above the threshold, the director field is distorted and the field dependence of its tilt angle, $\theta(z)$, is described by the same expressions in the SmA_b as in the N phase but involving the material constants related to **m** instead of those related to **n**.

The analysis of the dielectric data provides a wealth of information about the elastic and surface-anchoring properties of the N and SmA_b phases of the BP12 mixture (see Appendix E for details). The excellent fit of the data with the theoretical model [Fig. 8(a)] gives U_c^i for both phases (Fig. 11). In the N phase, upon cooling, we observe the usual weak variation of $U_c^n(T)$, which increases from approximately 2 to approximately 3 V through the 50 °C

nematic range of BP12. In SmA_b, the threshold is only slightly higher [3.5 V < $U_c^m(T)$ < 4 V], which is quite counterintuitive, because the structure of this phase is expected to be much more rigid than the nematic one. However, this low threshold value can be explained by the fact that the BFrTr induces only a distortion of **m**, which does not involve the rigid smectic order.

The temperature dependence (Fig. 12) of the splay elastic constants K_{11}^n and K_{11}^m (in the N and SmA_b phases, respectively) is obtained directly from Eq. (5) and from the data in Figs. 8 and 11. K_{11}^n increases significantly with decreasing temperature, which is the usual behavior for the nematic phase. As for many other N_{TB}-forming compounds [55,56,67,68], this behavior is not influenced by the
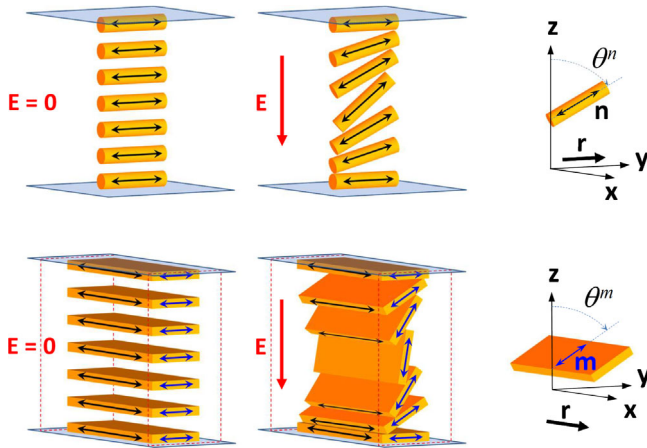

FIG. 10. Geometry of the distortions of **n** in the N phase (top row) and **m** in the SmA_b phase (bottom row); in the left and middle columns, the applied voltage is, respectively, below and above the Fréedericksz transition threshold. **r** is the rubbing direction, $(\mathbf{x}, \mathbf{y}, \mathbf{z})$ is a reference frame, θ^n (respectively, θ^m) is the angle of the **n** (respectively, **m**) director with the normal to the cell (**z** axis). The smectic layers are shown in red.
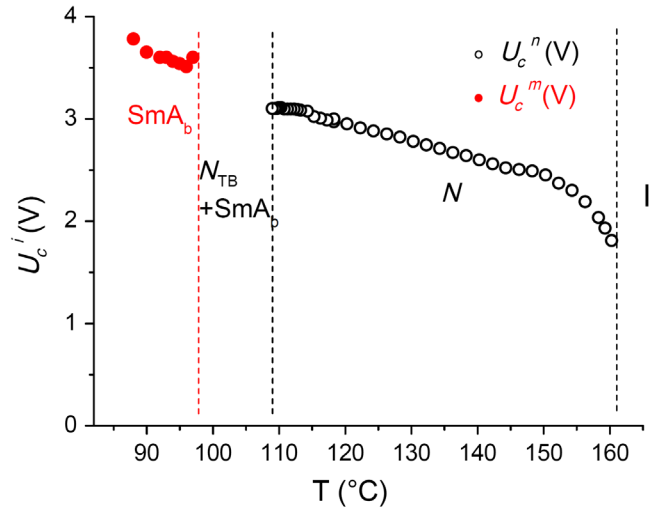

FIG. 11. Temperature dependence of the Fréedericksz transition threshold in the N and SmA_b phases. (There is no Fréedericksz transition in the N_{TB} phase, and no data are available for the biphasic N_{TB}/SmA_b range due to the nonuniform cell texture.)
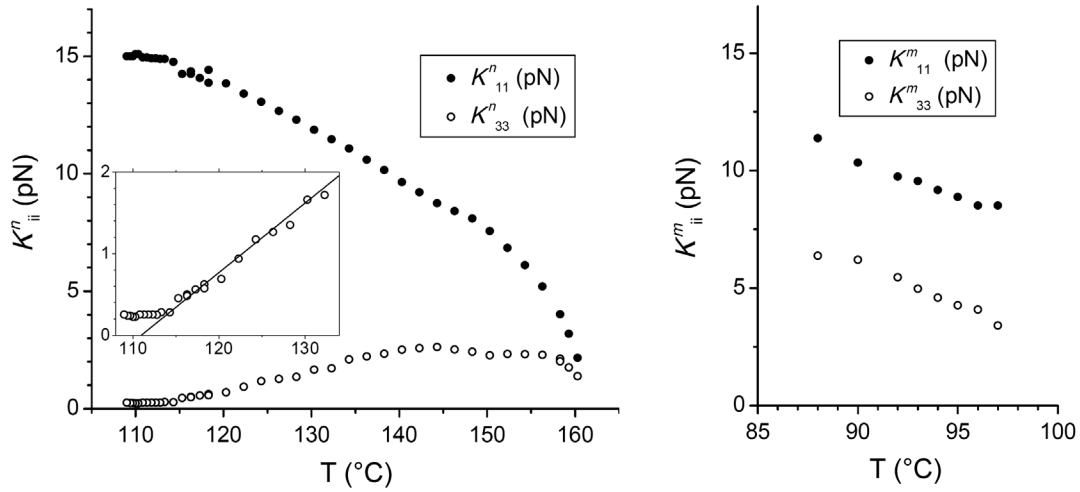
FIG. 12. Temperature dependence of the splay and bend elastic constants in the N (a) and SmA_b (b) phases of the BP12 mixture. [The inset in (a) shows a magnification of the K_{33}^n curve.]

presence of the N_{TB} phase. The order of magnitude of K_{11}^n, in the piconewton range, is also typical for both rodlike and bent-shaped nematics. In contrast, the bend elastic constant K_{33}^n decreases strongly and almost linearly with decreasing temperature, reaching values much smaller than K_{11}^n, which is the theoretically predicted [17,24] and experimentally observed [55,56,67,68] pretransitional behavior in the N phase preceding the N_{TB} phase. This feature is related to the pathological bend elasticity of the bent-shaped nematogen molecules, which leads to the spontaneous bend of **n** in the N_{TB} phase [17].

The behavior of K_{11}^m and K_{33}^m in the SmA_b phase is less peculiar than that of K_{33}^n. Their values are slightly smaller than the usual K_{ii}^n values in the N phase, and they increase with decreasing temperature. K_{33}^m is smaller than K_{11}^m, but they remain of the same order of magnitude. In brief, K_{33}^m in the SmA_b phase has a much more regular behavior than that of K_{33}^n in the N phase. The reason why is that the elasticity of the **m** director is not related to the bent shape of the molecules but essentially to their biaxiality (indeed, a SmA_b phase formed of lathlike molecules should have a similar elastic behavior). Moreover, the elasticity of the **m** director in the SmA_b phase is drastically different from that of the **n** director in usual SmA phases. Indeed, since **n** is parallel to the layer normal, its bend is forbidden because of its incompatibility with the layered structure, resulting in the divergence of K_{33}^n. In contrast, K_{11}^n remains finite but is larger than the splay modulus of the N phase, because the splay is coupled to the curvature of the layers, which costs additional elastic energy. However, in the SmA_b phase, **m** is parallel to the layers instead of their normal, and the curvature of **m** is then possible without any layer distortion; i.e., the (re)orientation of **m** is decoupled from the smectic order. This result explains why the elasticity of the **m** director is nematiclike in SmA_b and why, contrary to the usual SmA, the BFrTr is then possible.

C. Reorientation dynamics

One of the most striking features of the BFrTr is its very short relaxation time. This fast relaxation rate in the SmA_b phase is due to the collective rotation of the molecules *around their long axes* rather than around their short axes, as is usually the case in the N phase. The theoretical description of the rotational viscosities γ_1^n and γ_1^m for reorientation, respectively, of the primary **n** and secondary **m** directors in a biaxial phase is a complex task [63,69–71]. However, one can qualitatively expect that $\gamma_1^m \ll \gamma_1^n$, owing to the significantly different molecular cross sections involved in the two rotations. Figure 13 shows the temperature dependence of γ_1^n and γ_1^m deduced from the measured relaxation times and elastic moduli by a procedure well known for the FrTr case [72] and extended here for the BFrTr case (see Appendix D). Both viscosities increase

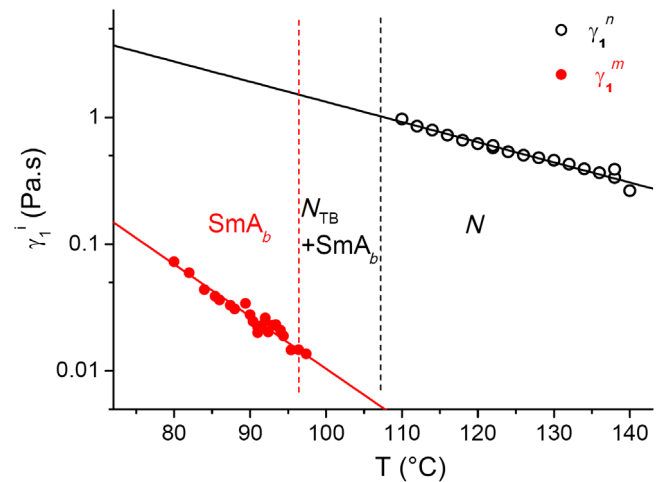FIG. 13. Rotational viscosities of the BP12 mixture for reorientation of **n** in the N phase (open black symbols) and **m** in the SmA_b phase (full red symbols). The solid straight lines show exponential decay fits of the viscosities (Arrhenius-law behavior).

exponentially with decreasing temperature, but γ_1^m is much smaller than γ_1^n (by about 2 orders of magnitude when extrapolated to the same temperature). We note that, although some anisotropy of the rotational viscosities is theoretically expected for biaxial nematics [70,71], the ratio $\gamma_1^n/\gamma_1^m \sim 100$ observed here is much larger than the predicted one (<10). Furthermore, counterintuitively, the activation energy for γ_1^m is twice as large as that for γ_1^n despite the much smaller steric hindrance for rotation of the molecule around its long axis (which is involved in the γ_1^m case). However, the rotational viscosity describes the collective rotation of all the molecules and, for this reason, depends strongly on the relevant order parameter components $\mathbf{Q}_{mm}$ and $\mathbf{Q}_{nn}$. Further detailed studies of the temperature dependence of the biaxial order tensor are, therefore, needed to clarify the observed temperature dependence of the rotational viscosities.

D. Comparison with other smectics

The field-induced transition in the $\mathrm{Sm}A_b$ phase is an exact analog of the Fréedericksz transition in nematics. Its existence is closely related to the biaxiality of this smectic phase. Indeed, in the uniaxial $\mathrm{Sm}A$ phase, the reorientation of the **n** director is impossible without dilation of the smectic layers, which is energetically very costly. For this reason, the field-induced transition in $\mathrm{Sm}A$ is only a "ghost" transition [3]; i.e., it is practically unobservable because of its negligible amplitude. A similar behavior is also expected for the N_{TB} phase, whose macroscopic symmetry is the same as that of the chiral $\mathrm{Sm}A$ [73], as it is pseudolayered and uniaxial. The case of the $\mathrm{Sm}C$ phase is more complex. Because of its biaxiality, two types of field-induced transitions are possible in different geometries [3]: ghost transitions with negligible amplitude due to the dilation of the layers, as in the $\mathrm{Sm}A$, and nematiclike transitions with finite amplitude but with rotation of the **n** director confined on a cone. Although the physical mechanism in this second case is similar to the FrTr in nematics, its optical behavior is very different, and this effect has not yet found any practical application. Optically, $\mathrm{Sm}C$ is approximately uniaxial with optic axis **n**. Therefore, during the transition, the optic axis rotates on a cone instead of in a plane for the nematic and $\mathrm{Sm}A_b$ phases. Hence, the main optical effect during the transition is not the variation of the birefringence but instead a twist of the texture and in-layer rotation of the optic axis by a limited angle, smaller than the $\mathrm{Sm}C$ tilt angle. Moreover, the relaxation time of the FrTr in the $\mathrm{Sm}C$ phase should be slow, as in the nematic phase, because the transition involves the reorientation of the long molecular axis. In contrast, the very fast reorientation of the short molecular axis, observed here in the BFrTr of the $\mathrm{Sm}A_b$ phase, is much more promising for practical applications.

Fast field-induced electro-optic effects have been reported earlier in the uniaxial N and $\mathrm{Sm}A$ phases of bent-shaped liquid crystals. They are due either to a field-induced biaxiality of the microscopic order parameter of the phase [74] or to a field-induced alignment of cybotactic (smectic) clusters in the N phase [75] (i.e., a field-induced biaxiality of the macroscopic order parameter). In contrast to the FrTr reported here, in both these cases, the field induces a biaxial order instead of reorienting a preexistent biaxial order parameter. Faster and thresholdless on and off switchings are expected in these cases under strong fields (approximately $10\ \mathrm{V}/\mu\mathrm{m}$), but, surprisingly, the reported relaxation times (approximately 1 ms) are 4 times slower than the relaxation time $\tau_{\mathrm{off}} \sim 240\ \mu\mathrm{s}$ of the BFrTr in the $\mathrm{Sm}A_b$ phase [Fig. 7(a)].

The BFrTr observed here with the intercalated $\mathrm{Sm}A_b$ is also expected with other orthogonal biaxial smectics, e.g., with nonintercalated $\mathrm{Sm}A_b$ phases [12]. Although here we observe the BFrTr in a $\mathrm{Sm}A_b$ phase formed by bent-shaped dimer molecules and preceded by the N_{TB} phase, the same electro-optic effect should be expected for $\mathrm{Sm}A_b$ phases formed by a much larger class of mesogenic compounds, e.g., for bent-core or even boardlike molecules. Moreover, some modulated nematic phases, like the elusive N_{SB} phase [17,24,31] and the recently discovered splay nematic [27,76–78], are biaxial and pseudolayered and should present a BFrTr identical to that reported here.

IV. CONCLUSION

By adding a small amount of a suitable rodlike nematogenic molecule to the BNA-76 bent-shaped dimer, we stabilize the M_X phase against crystallization, which allows for its structural and electro-optic investigation. The x-ray scattering study shows that the M_X phase is a smectic phase with liquidlike layers, resulting from the complete intercalation of the dimer molecules, a structure already reported for other bent-shaped dimers [7,9–11,32–36,38–42]. Moreover, multiple observations of the textures and the electro-optic response of single domains reveal that the M_X phase is biaxial and lacks ferroelectric order. We therefore identify it as a $\mathrm{Sm}A_b$ phase. The monomer mesogenic moieties of the dimer molecules form smectic layers with each dimer molecule spanning over two neighboring layers. In each layer, the monomer moieties are tilted in the same way, as in a smectic C, but the tilt direction alternates in adjacent layers, as in the $\mathrm{Sm}C_A$ phase. The primary director **n** of the phase is normal to the smectic layers, whereas the secondary director **m** is parallel to the layers.

Surprisingly, the $\mathrm{Sm}A_b$ phase displays a counterintuitive, nematiclike, electro-optic effect, with moderate threshold voltage and fast relaxation time. We show that this effect is an analog of the Fréedericksz transition in the N phase, but with reorientation of **m**, at fixed orientation of **n**. Because of the smectic constraints imposed to **n**, the elastic behavior of $\mathrm{Sm}A_b$ is here much simpler than the complex one of biaxial nematics. By analogy with the N phase, we show that the nematic elastic energy of $\mathrm{Sm}A_b$ is Frank-like but with elastic moduli K_{ii}^m related to the splay, twist, and bend

distortions of **m**. Using this analogy for the interpretation of our experimental results, we show that the bend and splay moduli for **m** in the SmA_b phase are of the same order of magnitude as their analogs for **n** in the nematic phase (far from the N-N_{TB} transition). Similarly, we deduce from the dynamic behavior of the BFrTr that the rotational viscosity constant for reorientation of **m** is almost 2 orders of magnitude smaller than that in the uniaxial nematic.

We present here an original experimental investigation of the elastic and electro-optic response of the biaxial order parameter tensor in a thermotropic liquid crystal. It opens the way for similar studies of other biaxial smectics, of the biaxial nematic, and also of the splay-bend nematic. Indeed, this latter, pseudolayered, phase also has biaxial symmetry, and, even though its pitch should be larger than the dimer length, it should nevertheless show exactly the same electro-optic behavior as that of SmA_b.

In this work, we study a dimer molecule with a markedly bent shape, leading to an N_{TB} phase in addition to the SmA_b one. However, the remarkable biaxial elasticity of SmA_b is not due to the bent shape of the molecules. Therefore, similar elastic and electro-optic behavior should also be expected with phases formed by weakly bent-shaped dimers, bent-core molecules, and even strongly biaxial board-shaped molecules. We hope that our work will, therefore, stimulate further interest in the synthesis of new molecules presenting biaxial mesophases and in the investigation of their elastic properties and electro-optic effects.

The biaxial Fréedericksz transition is very promising for practical applications. Indeed, the synthesis of the BNA-76 compound is not particularly involved or expensive, and it may readily be up-scaled. However, several important improvements are still required for applications: better surface-anchoring alignment methods for easy and reproducible production of large uniform domains; wider temperature range of the phase, centered around room temperature; new optical geometries taking advantage of the increase of the birefringence under a field, as opposed to its decrease in the nematic phase, and of the optical biaxiality of the SmA_b phase; and optimization of the switching characteristics, such as the threshold voltage and the relaxation time. This exciting roadmap is certainly worth exploring, because the threshold voltage of the biaxial Fréedericksz transition is similar to the usual values in the nematic phase and its response times are much faster. Therefore, the biaxial electric-field response of SmA_b could be a good candidate for replacing the traditional uniaxial-nematic electro-optic effects in fast-switching devices. We note that even at the present stage, before any expected improvement of the materials is achieved, the SmA_b phase of the BP12 mixture can already be useful in thermostated, nondisplay, electro-optic devices for which fast response is a key feature, such as modulators, switches, wave plates with continuous electric control of the retardation, light-beam deflectors, devices for adaptive polarization control, etc.

Note added in proof.—Recently, another paper reporting a new, quite convincing, example of the SmA_b phase was published [79]. This paper also describes some electro-optic experiments that show an electric-field-induced increase of the birefringence in that phase. However, in Ref. [79] this effect was not investigated and analyzed in detail as it was not the main focus of this paper.

APPENDIX A: MATERIALS AND METHODS

1. Liquid crystal materials

The bent-shaped dimer 1,7-bis[6-(4-hexyloxybenzoy-loxy)naphthalene-2-yl]heptane (BNA-76) is prepared in 74% yield, following a synthetic route described earlier [26].

The rodlike nematogen 4′-cyano[1, 1′-biphenyl]-4-yl 4-hexylbenzoate (6-PEPP-N) is commercially available (Xi'an Ruilian, China) and is used directly as supplied.

The binary mixture BP12 used in the present study is prepared by dissolving the appropriate quantities of BNA-76 and 6-PEPP-N in chloroform. The solvent is evaporated, and the remaining solid is heated above the clearing point. The phase transition temperatures and enthalpies of both compounds and the BP12 mixture are presented in Table I. They are measured on cooling with a Perkin-Elmer DSC differential scanning calorimeter operated at a scanning rate of $5\,°C/\text{min}^{-1}$ (see Fig. 14 for the DSC scan of the BP12 mixture).

2. X-ray scattering

X-ray scattering patterns of aligned samples are recorded with an already described apparatus [80]. Briefly, the x rays ($\lambda_{\text{CuK}\alpha} = 1.541$ Å) produced by a copper fixed tube are monochromatized and point-focused by a doubly curved pyrolytic graphite monochromator and a 0.5-mm-diameter collimator. The sample is filled into a 1-mm-diameter Lindemann glass capillary (WJM-Glas Müller GmbH, Germany) which is flame-sealed. The capillary is placed in an oven, with ± 0.5 K temperature stability, which is

TABLE I. Phase transition temperatures in °C, enthalpies in kJ mol^{-1}, and dimensionless value of $\Delta S/R$ (in square brackets) of BNA-76, 6-PEPP-N, and the binary mixture BP12.

Material	Cr		M_X		N_{TB}		N		Iso
6-PEPP-N	•	50 *19.20*[7.15]					•	227 *0.56*[0.13]	•
BNA-76	•	74 *32.69*[11.33]	•	81 *1.29*[0.44]	•	122 *0.15*[0.05]	•	154 *0.68*[0.19]	•
BP12	•	41 *19.91*[7.62]	•	100 *2.44*[0.78]	•	110 *0.10*[0.03]	•	163 *0.62*[0.17]	

calibrated with test substances. The oven lies between the poles of a 1.7 T electromagnet, sitting at the center of an evacuated camera. The scattering pattern is recorded on an image plate which is read by a Molecular Dynamics scanner. The sample-to-detection distance is usually 60 mm, but a camera extension provides a 120 mm distance, which gives us access to smaller scattering angles. In terms of scattering vector modulus $q = (4\pi \sin\theta)/\lambda$, where 2θ is the scattering angle, the whole accessible q range is 0.1–1.8 Å^{-1}, and the instrumental resolution corresponds to $\Delta q = 0.023$ Å^{-1}. From the patterns, various scattered intensity profiles can be extracted, for example, along the director (i.e., $q_\parallel$) or at fixed q, through the wide-angle diffuse ring, versus the azimuthal angle.

Unaligned samples (without an applied field) are also studied using a higher-resolution laboratory setup based on a copper rotating anode generator (RU-200BEH, Rigaku Ltd., Japan). The x-ray beam is filtered and focused by a confocal system consisting of two perpendicular graded multilayer mirrors (CMF-12-38Cu6 from Osmic Inc., USA). The samples are contained in glass capillaries as above and mounted in an LTS350 hot stage (Linkam Scientific Instruments, Waterfield, United Kingdom). The scattering signal is measured by a CCD detector cooled down to 30 °C (Photonic Science, United Kingdom). We use 4×4 binning, resulting in an effective pixel size of 96 μm. The sample-to-detector distance is 102.5 mm, yielding an accessible scattering vector range of 0.1–1.9 Å^{-1}. The images are integrated, corrected, and calibrated using the Nika suite [81] (version 1.74) running in Igor Pro 7.08.

3. Electro-optic experiment

For the electric and electro-optic experiments, we use sandwich-type cells with ITO transparent electrodes deposited on the inner faces of the cell glass plates to apply an electric field along the cell normal. The electrodes are
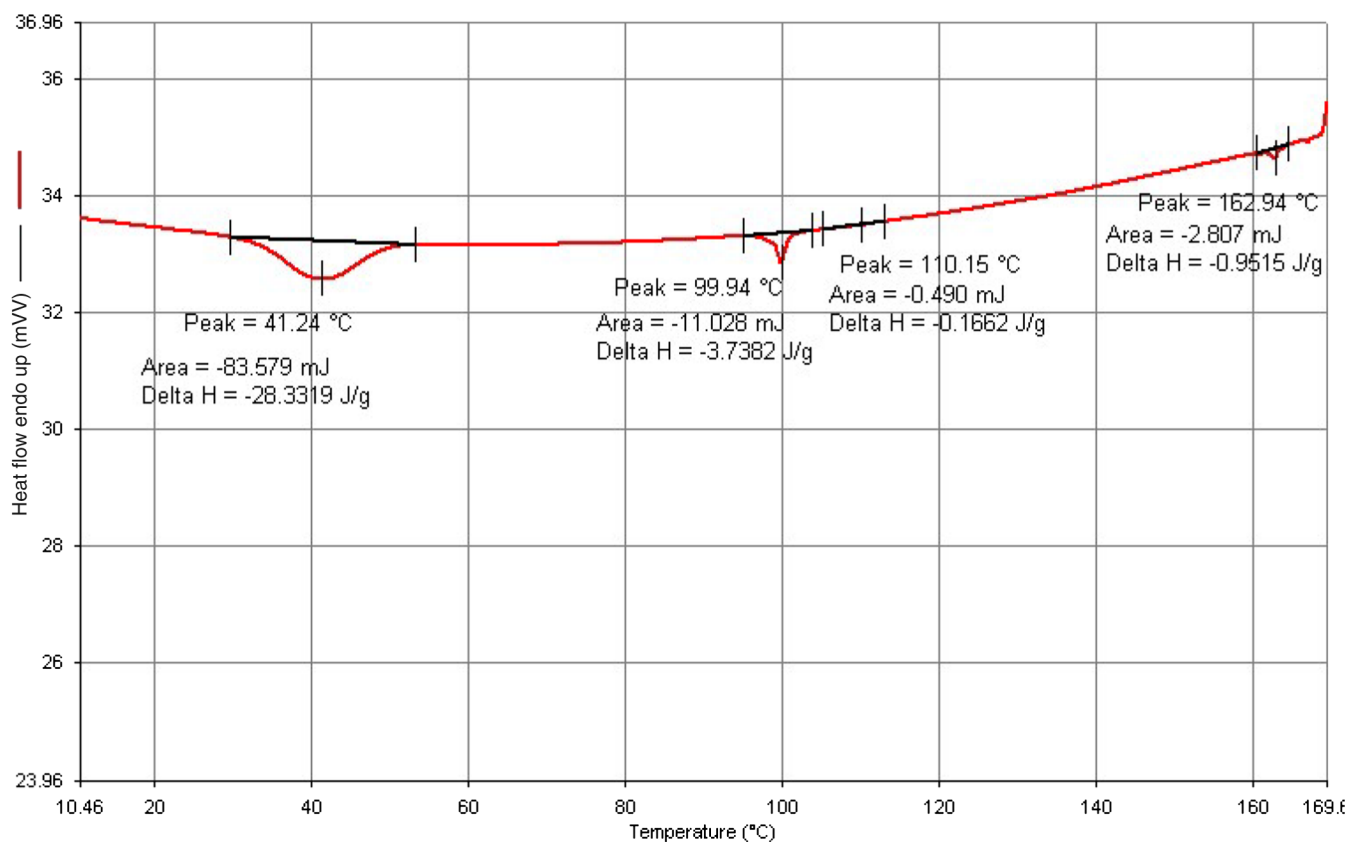

FIG. 14. DSC scan on cooling of the BP12 mixture.

covered with rubbed polymer alignment layers providing homogeneous planar alignment of the nematic director **n** parallel to the rubbing direction **r**. These cells with different gaps d are either of commercial origin (Instec, USA, $d \approx 5$ μm, and MUT, Poland, $d \approx 10$ μm) or prepared in our laboratories ($d \approx 1.4$ μm). The cells are filled by capillarity with the liquid crystal in the isotropic phase and then cooled to the nematic, N_{TB}, and M_X phases for the measurements. The temperature of the cell, placed on a heating stage (HS82, Mettler), is controlled with 10 mK accuracy using a homemade temperature controller.

All the polarized-light optical microscopy (POM) observations and measurements are made with a Leitz Ortholux microscope equipped with a digital camera (Dino-lite Pro AM423X). For precise quantitative measurements, we use a highly sensitive photodetector system mounted on the microscope. The latter consists of a photomultiplier tube (PMT) and an optical system which allows for the precise measurement of the intensity of the transmitted light in a small rectangular window in the image plane. For each experiment, we adapt the size of the window to that of the well-aligned single domain under study. The transmitted intensity is measured by the voltage drop U_L of the PMT anode current on a load resistance R_L. The value of R_L is chosen high (1 MΩ) for experiments with slow dynamics, $\tau > 1$ ms, to ensure high sensitivity of the measurements. When fast system response is required, $\tau \ll 1$ ms, we use $R_L = 1$ kΩ, which provides a response time of approximately 0.4 μs. The amplitude and the time dependence of U_L are measured with a digital oscilloscope (DSO-X 2004A, Agilent) which allows averaging the signal over up to 64 000 acquisitions and, thus, significantly improving the signal-to-noise (S/N) ratio. (To avoid any drift and oscillations of the incident light intensity, we use a stabilized voltage source for the microscope lamp.)

The signal applied to the cell electrodes is adapted to the mesophase properties and the kind of electric-field effect under study. The dynamics of the FrTr is investigated by applying square direct current (dc) voltage pulses or square envelope bursts of a sinusoidal alternating current (ac) with a frequency in the range 10–100 kHz. The pulse or burst duration is varied in the range 100 μs–1 s, depending on the response time of the investigated effect. Additionally, for the preparation of homogeneous planar or quasihomeotropic domains, an ac field with variable frequency and amplitude is applied for up to several hours.

The electric signal is produced by an arbitrary waveform generator (TGA12101, TTI) with an amplitude $U \leq 10$ V and is then amplified by a wideband amplifier (Krohn-Hite 7402M) up to an amplitude of $U \leq 400$ V. To avoid cell deterioration and short circuits, the voltage is limited, depending on the cell gap, to produce a rms field in the cell, E_{rms}, lower than 20 V/μm.

4. Birefringence measurements

All the birefringence measurements are performed with uniform single domains prepared by temperature-oscillation annealing of the sample, as described in Appendix C. Depending on the experimental conditions, we measure the cell phase shift ΔL using different techniques.

For qualitative estimation of the birefringence Δn, we use a Berek tilting compensator (Olympus). This classical technique [82] is direct and fast, but it is not precise enough for our purposes because of the typical error bars of 10 nm on ΔL. (This poor precision comes from the low sensitivity of the human eye in detecting the minimum of the transmitted light intensity at the compensation.) Therefore, we use this technique to check the sign of ΔL and to calibrate the Sénarmont data when the measured phase shift is larger than λ (as the Sénarmont technique gives ΔL, modulo λ).

For precise ΔL measurements, we use a Sénarmont compensator [82] (Leitz). In this case, ΔL is derived from the value of the analyzer decrossing angle α_0 that minimizes the transmitted intensity $I(\alpha)$ of the incident monochromatic light ($\lambda = 546$ nm). To improve the precision, $I(\alpha)$ is measured using the PMT with $R_L = 1$ MΩ, integrating over the area of the single domain. In this way, an excellent S/N ratio is obtained, and it is even further improved by averaging the signal over up to several hundreds of oscilloscope acquisitions. For static experiments, e.g., to measure $\Delta L(T)$ at $U = 0$ or $\Delta L(U)$ at fixed temperature T and continuously applied voltage U, we further improve the precision by measuring $I(\alpha)$ for several α values around the minimum and obtaining α_0 from a fit of the $I(\alpha)$ curve with a parabola. With this "parabola-fit" technique, the precision is better than 0.1 nm.

For dynamic experiments, i.e., to measure the time evolution of ΔL when the sample is submitted to square voltage pulses or bursts, we use a small (1 kΩ) load resistor for the PMT, providing a response time of the setup much smaller than the response times of our cells. In this case, we rotate the analyzer at 45° with respect to the transmission minimum, $\alpha = \alpha_0 - \pi/4$. In this geometry and for small values of $\Delta L(t)$, the transmitted intensity is approximately [82]

$$I(t) = \frac{1}{2}I_0 + \frac{1}{2}I_0 \sin\left(2\pi \frac{\Delta L(t)}{\lambda}\right) \approx \frac{1}{2}I_0 + I_0\pi \frac{\Delta L(t)}{\lambda},$$

(A1)

where I_0 is the intensity of the incident light. The PMT signal is measured in real time during each pulse or burst, amplified, and accumulated in the memory of the digital oscilloscope. Averaging the results over a sufficiently large number of acquisitions, we measure $\Delta L(t)$ with a precision better than 0.01 nm and a time resolution better than 1 μs.

The techniques described above require a uniform single domain in the PMT window. In the N_{TB}-M_X biphasic coexistence range, this condition is difficult to satisfy, and we often use an "image mapping" technique to measure ΔL. With the Sénarmont compensator, multiple images of the sample are acquired at various values of the analyzer angle α but at constant sample orientation and incident light intensity. Then, the variation with α of the light level recorded in each pixel (or group of pixels) is obtained by computer treatment of the images. Each $I(\alpha)$ curve is fitted with a parabola, giving a map of the angle α_0 that minimizes $I(\alpha)$, and a map of ΔL. Note that, in the present study, this technique is used to analyze a texture consisting of small single domains of the two N_{TB} and M_X phases, which have lowest light transmission at quite different values of α. The large contrast between the two phases and the light scattering by the domain walls increase the noise so that we could use this technique only by averaging over relatively large surfaces (a few hundreds of pixels).

5. Dielectric measurements

For dielectric measurements, we use 10-μm-thick commercial cells treated for planar alignment (MUT, Poland). The active area of the cell, which is submitted to the field, shows perfect planar alignment in the N phase. To obtain good planar alignment over the whole active area in the N_{TB} phase and, subsequently, in the M_X phase, we use the annealing technique described in Appendix C. No dielectric measurements are performed in the biphasic region due to the intrinsic impossibility of obtaining a single domain. Finally, when needed, quasihomeotropic orientation of the whole active area is obtained by slow cooling of the sample under 70 V ac voltage (10 kHz).

In all cases, the dielectric constants are obtained from the voltage dependence of the cell capacity, $C(U)$, measured in a single domain. To measure $C(U)$, we develop an original setup and an experimental procedure that will be described in more detail elsewhere. In brief, the cell is connected in a passive RC circuit in series with a variable measurement resistor R_m. An ac voltage with variable frequency f and rms voltage $U_g < 7$ V is generated using a NI PCIe-6251 data acquisition card from National Instruments, then amplified using a wideband amplifier (Krohn-Hite 7402M) up to a much larger rms value, $U_a < 280$ V, and applied to the circuit. The voltage drop U_m on the measurement resistor and its phase shift δ_m with respect to U_a are measured using the same DAQ card and a suitable voltage divider. The measurement process is controlled by a dedicated proprietary software which provides full control of the voltage range, voltage steps, duration, and time sequence of the data acquisitions. The analysis of the U_m and δ_m curves, measured as a function of U_a and f, and their comparison with the theoretical model of the equivalent circuit of the setup provides the $C(U)$ values.

APPENDIX B: ADDITIONAL X-RAY SCATTERING RESULTS

The diffractogram of the crystalline phase (black curve) shows, at wide angles, diffraction lines that are most probably broadened by the small size of the crystallites. In contrast, the diffractograms of the M_X, N_{TB}, and N phases show no diffraction lines at wide angles, which demonstrates their fluid nature. In addition, three sharp diffraction lines are also visible at small angles for the crystalline phase. The line at $q = 0.15$ Å^{-1} corresponds to a lattice spacing of 42 Å and is, therefore, related to the average length of the mixture components. The diffractogram of the M_X phase (blue curves) shows only a single sharp reflection, at $q = 0.28$ Å^{-1}, so that the phase must have an interdigitated structure (since optical measurements rule out any director tilt; see the main text). The diffractograms of both the N_{TB} and N phases (green and red traces, respectively) show no sharp diffraction lines at all, as expected for these mesophases that lack any kind of long-range positional order of the molecules.

Azimuthal profiles of the scattered intensity, at fixed scattering vector modulus at the maximum of the wide-angle diffuse ring, are extracted from the scattering patterns shown in Fig. 15. Surprisingly, the width of these profiles
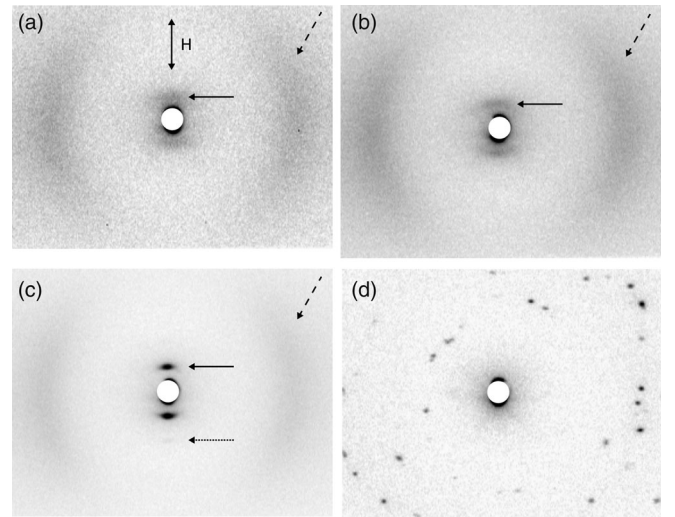

FIG. 15. X-ray scattering patterns of a BP12 sample aligned by a 1.7 T magnetic field H [double-headed arrow in (a)] recorded with a 60 mm sample-to-detection distance. (a) In the N phase ($T = 117\,°C$); (b) in the N_{TB} phase ($T = 107\,°C$); (c) in the M_X phase ($T = 97\,°C$); and (d) in the crystalline phase (room temperature). White disks at the pattern centers represent the beam stop. In (a)–(c), the dashed arrow points to the wide-angle diffuse ring. In (a) and (b), the solid arrow points to small-angle diffuse streaks. In (c), the solid arrow points to one of the two first-order smectic reflections, while the dotted arrow points to one of the two second-order smectic reflections. In (d), the pattern shows only sharp Bragg reflections arising from several large crystallites in reflection position in the x-ray beam.
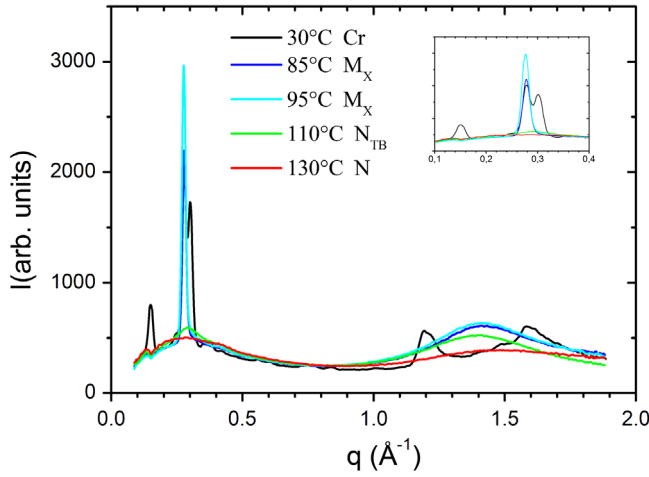
FIG. 16.   Powder x-ray diffractograms of a BP12 sample in the crystalline, M_X, N_{TB}, and N phases. The inset shows a magnification of the low-angle region.
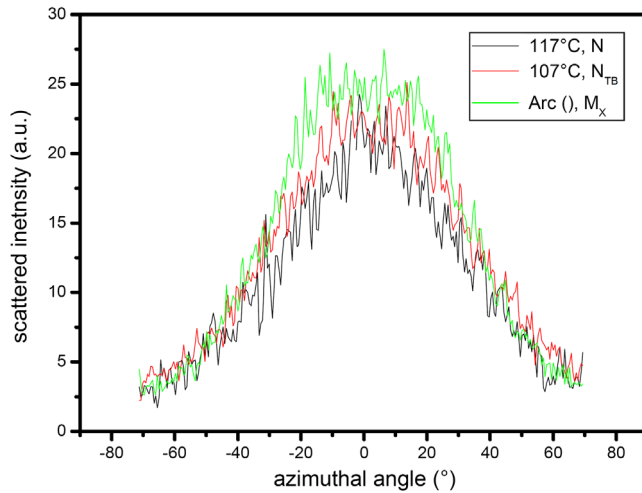

FIG. 17.   Azimuthal profiles of the scattered x-ray intensity of a BP12 sample aligned in a magnetic field in the N, N_{TB}, and M_X mesophases.

increases with decreasing temperature, in particular, in the M_X phase. This unusual feature suggests that the mesogenic cores are actually tilted with respect to the normal to the smectic layers, as in SmC- and SmC_A- type phases.

APPENDIX C: PREPARATION OF LARGE UNIFORM DOMAINS FOR THE MEASUREMENTS

1. Growth of planar single domains of the N_{TB} and M_X phases

To measure ΔL, we need large enough single domains with uniform planar or homeotropic alignment. In the N phase, excellent planar orientation (with a small uniform pretilt of approximately 2°) of the whole sample is induced
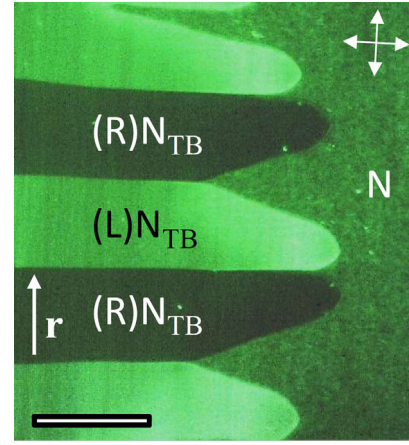

FIG. 18.   Growth of N_{TB} monochiral domains (on the left side) in the N phase (on the right side) at the N-N_{TB} transition of BP12 ($T = 108.9\,°C$, $d = 9.8\ \mu$m). To evidence the alternating sign of the chirality of the domains, they are observed in monochromatic light, $\lambda = 546$ nm, between crossed polarizers (white double-headed arrows) rotated at 3° with respect to the rubbing direction **r**. (Scale bar 100 μm.)

by the alignment layers. Slowly cooling the cell through the N-N_{TB} transition, in a small horizontal thermal gradient, produces large N_{TB} monochiral domains, with uniform planar orientation of the helix axis **h** oriented as the director in the previous N texture (Fig. 18). Close to the transition, these domains are perfectly uniform, with **h**∥**r**, and monochiral, with the sign of the chirality alternating from one domain to the next (Fig. 19). On further cooling, however, we observe the usual stripe instabilities [19] of the N_{TB} phase [Fig. 20(a)]. This result hinders precise bire-fringence measurements and prevents growing large single domains of the M_X phase on further cooling.

To solve this issue, we impose temperature oscillations, with a typical amplitude of ±1 °C and 1–3 min period while cooling the sample. This temperature cycling, together with periodic application of weak electric fields (approximately 0.5 V/μm), allows us to keep the monochiral N_{TB} domains uniform [Fig. 20(d)]. Finally, starting from well-annealed and uniform N_{TB} single domains and slowly (approximately 0.1 °C/ min) decreasing the temperature, we achieve the growth of the required highly uniform single domains of the M_X phase (Fig. 4). As the temperature is slowly decreased, the M_X single domains grow, in a reversible way, until the whole sample turns into the M_X phase. At constant temperature, the coexisting M_X and N_{TB} regions remain at equilibrium (Fig. 21).

However, the azimuthal orientation of the slow axis **N** of the M_X domains is still not perfect. When the sample is observed with the rubbing direction **r** parallel to one of the crossed polarizers, the texture observed is smecticlike, consisting of subdomains with perfectly uniform azimuthal orientation, separated by sharp grain boundaries. Rotating the sample by a small angle, between crossed polarizers,
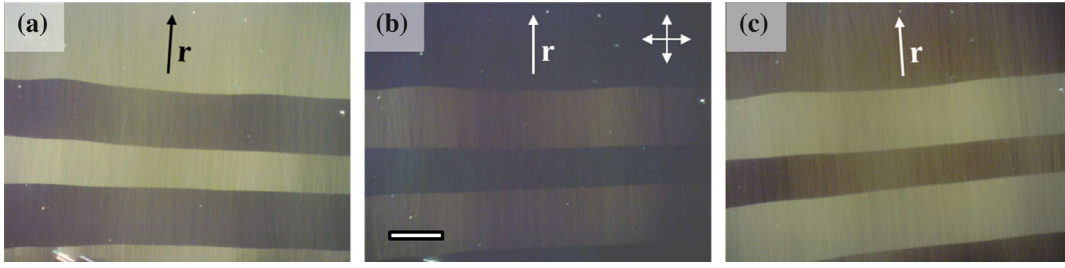
FIG. 19. Alternating monochiral N_{TB} domains of BP12 observed between crossed polarizers ($T = 108.8\,°C$, $d = 9.8\ \mu m$). In both kinds of domains, the optic axis **N** is parallel to the rubbing direction **r**. When **r** is parallel to the input polarizer (b), both kinds of domains appear dark. When the sample is rotated by $\pm 4°$ [(a) and (c)], a strong transmittance contrast appears between the two kinds of domains. The inversion of this contrast with the sign of the rotation indicates the different signs of the chirality of adjacent domains. (Scale bar 100 μm.)
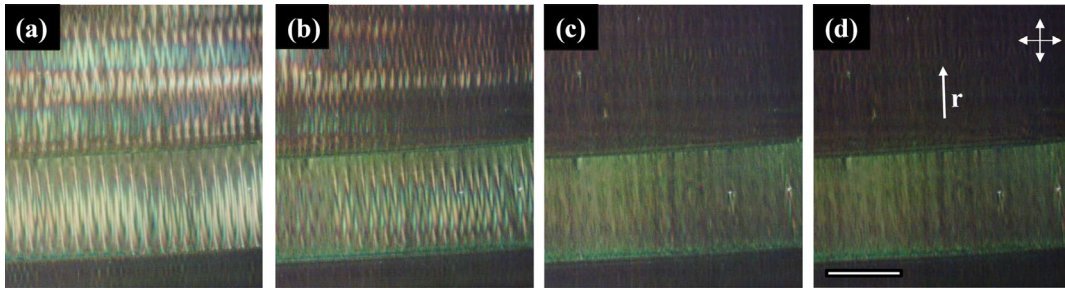

FIG. 20. Stripe instabilities in the monochiral N_{TB} domains of BP12 (a) and their annealing after thermal cycling and repeated application of a weak electric field ($T = 108.0\,°C$, $d = 9.8\ \mu m$). After annealing (b)–(d), the domains are again uniform and their optic axis **N** is parallel to the rubbing direction **r**. The weak residual heterogeneities after annealing (d) are due to the surface memory of the alignment layers. (Scale bar 100 μm.)

reveals that **N** deviates from **r** by a few degrees in these subdomains, typically less than $\pm 3°$ [Figs. 22(a)–22(c)]. Note that these small deviations do not significantly affect the precision of the birefringence measurements.

As described above, the birefringence experiment requires single domains of area typically larger than $20\ \mu m \times 20\ \mu m$ and uniform planar alignment, both
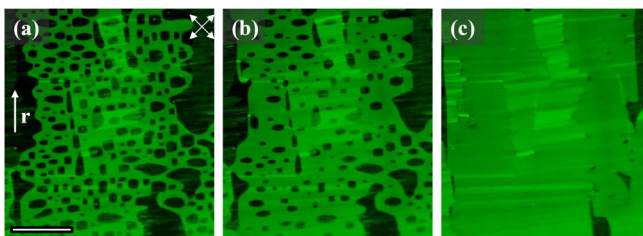

FIG. 21. Biphasic coexistence of the M_X (bright) and N_{TB} (dark) phases in a $d = 1.4\ \mu m$ planar-alignment cell at different temperatures: $100\,°C$ (a), $98\,°C$ (b), and $97\,°C$ (c). To visualize the different birefringence of the two phases, the cell is observed under polarized monochromatic light ($\lambda = 546$ nm) with a Sénarmont compensator and with an analyzer rotated by an angle optimizing the contrast between the two phases. Each image is recorded after 15 min of relaxation at fixed temperature to reach equilibrium between the two phases. (Scale bar 100 μm.)

zenithal and azimuthal, of the slow axis **N**. The requirements for the dielectric experiment are in a way more stringent, because it averages the signal arising from the whole electrode-covered area of the cell. In that case, we need good planar alignment over this square active area of typically 5 mm size. Fortunately, the azimuthal uniformity of the alignment is not so relevant for dielectric measurements, because they are not sensitive to in-plane rotation of the directors. Therefore, we align the whole active area of the cell in the N_{TB} and M_X phases using the same annealing treatment described above. In contrast, a large enough single domain is impossible to produce at the N_{TB}-M_X coexistence, which prevents any dielectric measurements in the biphasic coexistence range.

2. Growth of quasihomeotropic domains of the three phases

To prove directly the biaxiality of the M_X phase, we need a homeotropically aligned sample, i.e., with primary director **n** oriented parallel to the cell normal and to the observation direction. The same orientation is also needed for the measurement of the complete set of the dielectric tensor eigenvalues in the different mesophases. Unfortunately, it is notoriously difficult to achieve the homeotropic alignment of bent-shaped dimers by surface
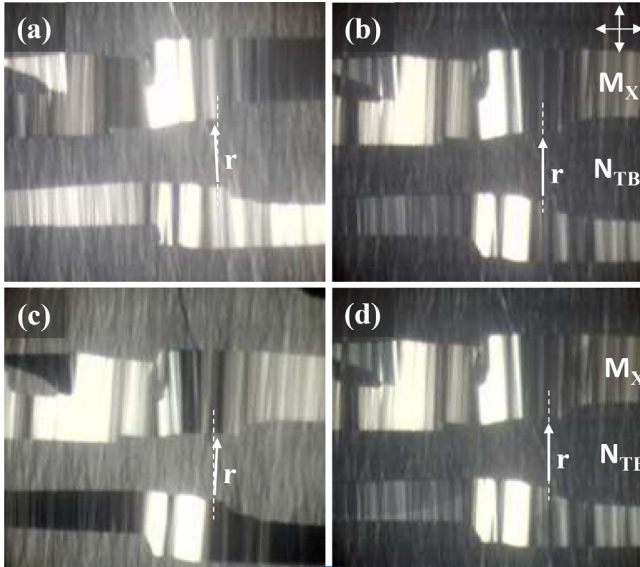
FIG. 22.   Azimuthal alignment of the M_X domains coexisting with the N_{TB} phase at 101.8 °C. The sample is observed between crossed polarizers and rotated at angle $\psi = -2°$, $0°$, $+2°$, and $0°$ in (a), (b), (c), and (d), respectively. Here, ψ is the angle between the rubbing direction **r** and the input polarizer. The slow axis **N** of the M_X regions is approximately parallel to **r**. The azimuthal deviations of **N** from **r**, at angles typically smaller than $\pm 3°$, are due to slight disorientations of the smectic single domains of the M_X phase. The analysis of the transmitted light and the smectic textures shows that **N** is vertical in the pictures, perpendicular to the smectic layers that are horizontal, which confirms the Sm*A* (and not Sm*C*) nature of the M_X phase. In (d), an electric field, larger than the threshold of the Freédericksz-like transition, is applied to the sample ($f = 6$ kHz, $U_{rms} = 7$ V).

surface layers with strong director distortion, which is due to the competition of the aligning torques of the surface and the field. The resulting texture is quasihomeotropic, with a small residual phase shift of about 3 nm and slow axis parallel to **r**. When the sample is cooled slowly under the field to the N_{TB} phase, this quasihomeotropic texture is preserved, with some decrease of the residual birefringence. In the bulk of the sample, the helix axis is homeotropic, and the N_{TB} pseudolayers are parallel to the surfaces, like the layers of a homeotropically aligned smectic phase. Finally, cooling further the sample under the field, we observe the growth of large and uniform quasihomeotropic domains of the M_X phase (Fig. 5). These domains are birefringent, which directly confirms that the M_X phase is biaxial.

Although the growth of quasihomeotropic monodomains is possible only under a strong electric field, the orientation of the layers parallel to the cell surface is kept when the field is removed at a low enough temperature ($T < 85$ °C). Because of the "frozen" layer structure, the **n** director remains homeotropic, while **m** is parallel to the surface. In this case (Fig. 23), we observe texture instabilities with characteristic zebra patterns between crossed polarizers, due to the very weak (or nonexistent) anchoring of **m** parallel to **r**. In the zebra pattern, **m** spontaneously rotates by a few quarter turns in the surface plane. The transmitted intensity varies in the field of view as $\sin^2 2\varphi$, where φ is the angle between **m** and one of the crossed polarizers. This variation again confirms the lack of revolution symmetry around **n** and, therefore, the biaxiality of the M_X phase.

treatments in any of the N and N_{TB} phases. We try several classic surface treatments that usually provide homeotropic alignment for most of the nematic and smectic A-forming compounds: grafted silane surfactant layers, polyimide layers developed for homeotropic alignment, and even bare glass or ITO surfaces. Without surprise, we also fail to obtain a homeotropic alignment of either BP12 or other mixtures of BNA-76 in any phase. In all these attempts, we obtain in the nematic phase a poor, very inhomogeneous planar alignment with azimuthal orientation varying at random on the two cell surfaces and dominated by the anchoring memory. In the N_{TB} and M_X phases, this memorized alignment (incompatible with the pseudolayered or layered structure) turns in a fan-shaped texture, which is unsuitable for our experiments.

However, we produce a quasihomeotropic alignment in our cells treated for planar alignment by taking advantage of the positive dielectric anisotropy of BP12, $\Delta\varepsilon \approx 2$. We apply continuously a strong ac electric field ($E_{rms} = 17$ V/μm, $f = 10$ kHz) throughout the thermal history of the sample. In the nematic phase, this application yields a perfect homeotropic alignment of the bulk of the cell, leaving only two thin (approximately $10 - $nm-thick)

FIG. 23.   Homeotropically aligned M_X phase after field removal ($d = 1.4$ μm, $T = 85$ °C). The smectic layers are parallel to the image plane, **n** is perpendicular to it, and **m** is parallel to the layers, in the image plane. The sample is birefringent, with $\Delta n \sim 0.02$, and the slow axis is parallel to **m**. In the striped areas, **m** rotates in the plane, and each band corresponds to a 90° rotation. (Scale bar 50 μm.)

APPENDIX D: INTERPRETATION OF THE BIREFRINGENCE DATA

In the absence of a field, the birefringence measured in the different phases of BP12 depends on the values of the components of the order parameter tensor **Q**, on its surface-imposed orientation in the single domains under study, and on the structure of the phase (i.e., homogeneous or heliconical). In principle, all of these parameters are sensitive to applied fields, leading to variation of the birefringence. However, for realistically strong fields, such as those applied in our experiment, only the reorientation of **Q** is significant and should be taken into account.

In the birefringence experiments, the optical path difference ΔL between the different normal modes of light propagating in the anisotropic medium is measured. In our experimental geometry, the rubbing axis **r** lies in the cell plane, oriented at 45° with respect to the polarizer, and monochromatic light propagates along the cell normal. Without a field, the optical properties of all three phases (N, N_{TB}, and M_X) are uniform, with the major axis of the ellipsoid of indices, **N**, oriented parallel to **r**. Note that, despite the strong nematic director distortion in the N_{TB} phase, it is optically uniaxial with **N** parallel to the helix axis, $\mathbf{h}(\mathbf{N}\|\mathbf{h}\|\mathbf{r})$ [54]. Therefore, without a field, we have $\Delta L = (n_\| - n_\perp)d$ in the uniaxial N and N_{TB} phases, where the subscripts of the refraction indices refer to the polarization direction with respect to **N**. In the biaxial M_X phase, in the absence of a field, we have $\Delta L = (n_{nn} - n_{mm})d$, where the subscripts of the refraction indices now refer to the polarization direction parallel to, respectively, either **n** or **m**. Similarly, under an infinitely strong field ($U = \infty$), all three phases are again uniform, but with the major axis of the ellipsoid of indices oriented along the cell normal. Therefore, $\Delta L = 0$ in the uniaxial phases and $\Delta L = (n_{mm} - n_{kk})d$ in the biaxial M_X phase.

When a sufficiently large but finite voltage is applied to the N or M_X phase, the cell undergoes a FrTr and the orientation of **N** is no longer uniform but varies along the z axis. (We do not consider here the more complex N_{TB} case, where the reorientation of **h**, and, therefore, of **N**, is a nonequilibrium defect-mediated process.) In the N case, **n** rotates away from **r** but still remains in the same zenithal plane. In the M_X case, **n** remains orthogonal to the rigid smectic layers, but **m** tilts away from its initial orientation while remaining in the same zenithal plane. Because the distortion of the respective director is planar (without twist) in both cases, there is no rotation of the polarization plane of light during its propagation in the cell, and the phase shift in the N phase is

$$\Delta L(U) = \int_0^d \left[\frac{n_\| n_\perp}{\sqrt{n_\|^2 \cos^2\theta + n_\perp^2 \sin^2\theta}} - n_\perp \right] dz, \quad \text{(D1)}$$

where $\theta = \theta(z, U)$ is the position- and voltage-dependent angle between the field and **n** [51].

By analogy, we obtain for the M_X case

$$\Delta L(U) = \int_0^d \left[n_{nn} - \frac{n_{mm} n_{kk}}{\sqrt{n_{mm}^2 \cos^2\theta + n_{kk}^2 \sin^2\theta}} \right] dz, \quad \text{(D2)}$$

where $\theta = \theta(z, U)$ now refers to **m**.

The field-induced tilt of the director, $\theta(z, U)$, during the FrTr in the N phase, can be obtained as a function of the applied voltage and the material constants (elastic moduli, dielectric tensor components, etc.) by integrating the Euler-Lagrange equation corresponding to the free energy of Eq. (3) [58,59,65]. This classical approach allows one to interpret the $\Delta L(U)$ data acquired in the N phase [58,59,83] and to extract the threshold field of the FrTr, U_c^n, and the nematic birefringence, $\Delta n = n_\| - n_\perp$. Moreover, based on the similarity of the elastic behaviors of the N and M_X phases, we use here the same approach to interpret the $\Delta L(U)$ data acquired during the BFrTr in the M_X phase and to extract the threshold field of the BFrTr, U_c^m, and the variation of the refraction index, $n_{mm} - n_{kk}$, related to the reorientation of **m**.

When the voltage is applied as dc pulses or ac bursts, the director orientation in the cell and, therefore, the optical phase shift become time dependent. For the nematic phase, the reorientation of **n** during the FrTr depends on the applied voltage, the cell thickness, the rotational and translational viscosities, and the involved distortion modes [72]. The precise description of the time-dependent optical response of the cell requires a numerical simulation. However, a simple analytical description is possible in two important particular cases for which the time dependence of the tilt angle of **n** in the middle of the cell, $\theta_{\min}(t)$, follows a simple exponential law: $\theta_{\min}(t) \sim \exp(-t/\tau)$. Here, $\tau = \tau_{on}$ and τ_{off} stands for the characteristic response time when the field is switched on or off, respectively. This approximate description, well known for the FrTr of the N phase [72], is extended here, by analogy, to the case of the BFrTr for the M_X phase.

The first case is that of τ_{on} when the applied voltage is much higher than the threshold field $U \gg U_c$; τ_{on} is voltage dependent and is approximately given by

$$\tau_{on}^i \approx \frac{\gamma_1^i}{K_{11}^i} \left(\frac{dU_c^i}{\pi U} \right)^2, \quad \text{(D3)}$$

where the superscript $i = n$, m indicates the director involved in the Fréedericksz transition for the N or M_X phase, respectively, and γ_1^i is the rotational viscosity for reorientation of that director. The strong voltage dependence of τ_{on} allows achieving very fast on response of nematic devices by simply applying high enough voltage.

The second simple case is related to the relaxation process upon field removal. Then, multiple relaxation modes are excited, corresponding to different wavelengths, $2d/(2\nu + 1)$, where $\nu = 0, 1, \ldots$ is an integer, of the director distortion. The off-relaxation times of the different modes decrease as $1/(2\nu + 1)^2$, but, after a fast initial relaxation, the process is dominated only by the slowest time, corresponding to $\nu = 0$:

$$\tau_{\text{off}}^i \approx \frac{\gamma_1^i}{K_{11}^i}\left(\frac{d}{\pi}\right)^2. \tag{D4}$$

This time depends only on the cell thickness and not on the applied field.

Note that the on- and off-relaxation times of the director, defined above, are related in a simple way to the optical relaxation times $\tau_{\text{on/off}}^{\text{opt},i}$ ($i = n, m$), which are measured in our experiment [see Eqs. (D1) and (D2)]: $\tau_{\text{on/off}}^i = 2\tau_{\text{on/off}}^{\text{opt},i}$.

APPENDIX E: INTERPRETATION OF THE DIELECTRIC DATA

The cell capacitance $C(U)$ is measured as a function of the voltage applied to the LC layer in the dielectric experiment. Using various surface- and field-induced alignments, all three eigenvalues of the dielectric tensor $\boldsymbol{\varepsilon}$ can be measured in the three mesophases under study. In the uniaxial N and N_{TB} phases, the components $\varepsilon_\perp = \varepsilon_{kk} = \varepsilon_{mm}$ and $\varepsilon_\parallel = \varepsilon_{nn}$ are obtained by extrapolating the $C(U)$ results (measured in a cell with planar surface alignment) to $U = 0$ and $U \to \infty$, respectively. (In these extreme cases, **n** is uniform and oriented either perpendicular or parallel to the field, respectively.) In the M_X phase, the components ε_{kk} and ε_{mm} are obtained by extrapolating the $C(U)$ data, measured in a sample with planar orientation of **n**, to U values, respectively, much smaller or much higher than the threshold voltage of the BFrTr. Indeed, in these two limit cases, the **Q** tensor is oriented with either the **k** or **m** director parallel to the field. The last component of the

dielectric tensor, ε_{nn}, is obtained from the $C(U)$ data measured in a quasihomeotropic sample, produced by crossing the N-N_{TB}-M_X transitions under a strong field (see Appendix C).

In the N and M_X phases, $C(U)$, measured in a planar cell, varies smoothly and reversibly during the FrTr. The well-known theoretical description of the FrTr in the N phase [64,65] gives $C(U)$ as a function of the elastic, electric, and anchoring properties of the liquid crystal. Thus, the fit of the $C(U)$ data with the theory provides [58–60] the values of $\varepsilon_\perp$, $\varepsilon_\parallel$, and the elastic moduli of the primary director, K_{11}^n and K_{33}^n. Using a proprietary numerical simulation program, we measure precisely these parameters, as well as the pretilt angle ψ^n, and (approximately) the surface anchoring energy W_s^n for **n** [see Fig. 8(a) for an example of the excellent fit of the data with the theory].

We apply a similar approach for the treatment of the dielectric data of the biaxial M_X phase thanks to the analogy of the BFrTr with the usual FrTr in the nematic. The fit of the data with the numerical simulation [Fig. 8(a)] gives precisely the dielectric tensor components ε_{kk} and ε_{mm} and the elastic moduli of the secondary director, K_{11}^m and K_{33}^m.

The pretilt angle ψ^m and the surface anchoring energy W_s^m are also estimated from the fit with the theoretical curves, as they are independent fitting parameters. The experimental data are compared in Fig. 24 (left) with the theoretical curves calculated for three different values of ψ^m and for the best-fit values of all the other parameters [the region around $U = U_c$ is shown only because the curvature of $C(U)$ there is very sensitive to the value of ψ^m]. Because of the symmetry of the unidirectionally rubbed polyimide layers, we expect $\psi^m = 0$ (contrary to the pretilt of the primary director, $\psi^n \neq 0$, because the inversion symmetry of the surface director is broken by the rubbing process). The measured small, but finite, value $\psi^m = 2.2 \pm 0.7°$ is probably due to easy axis gliding [84] under the strong electric torques applied during the experiments. The significant deviation of the experimental curve
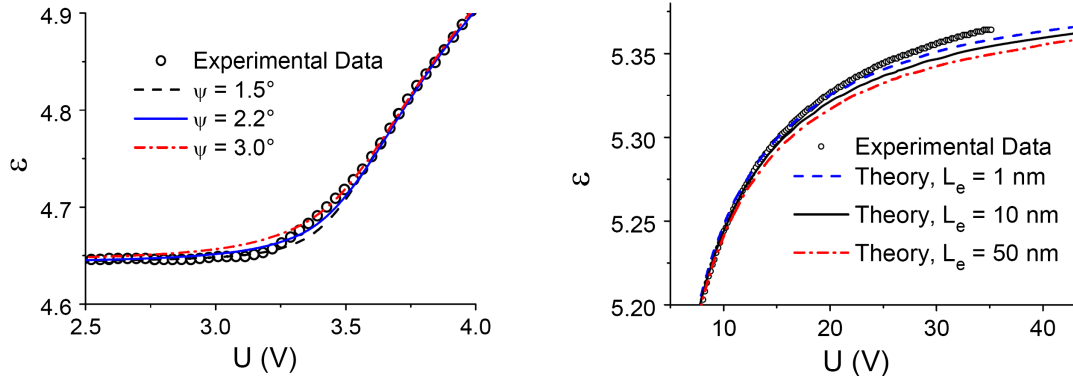

FIG. 24.   Estimations of the pretilt angle ψ^m (left) and the anchoring extrapolation length L_e^m (right) for the secondary director in the M_X phase ($T = 95\,°C$).

from the theoretical ones indicates some local variations of ψ^m on the cell surfaces.

To take into account the finite anchoring energy W_s^m of the **m** director, we use as a fitting parameter the corresponding de Gennes extrapolation length [51] $L_e^m = K_{11}^m/W_s^m$. Figure 24 (right) shows the deviation of the experimental data from the best-fit theoretical curves calculated with different fixed values of L_e^m (this parameter influences mainly the high-voltage region). Qualitatively, the large deviation of the data from the $L_e^m = 50$ nm curve indicates that $L_e^m \ll 50$ nm, i.e., that $W_s^m \gg 0.2$ mJ/m, which is a rather strong value. Surprisingly, the experimental curve is even higher than the curve calculated with infinite anchoring energy. This feature is most probably due to a small field-induced increase of the biaxial order, leading to a higher ε_{mm} value and to an increase of the cell capacitance $C(U)$.

Another way to estimate L_e^m is to compare the values of U_c measured in cells with significantly different gaps (this approach uses only low-U data, thus avoiding high-field artifacts). In fact, for weak anchoring, the threshold value is decreased by a factor of $(1 + 2L_e^m/d)$ [85]. The systematic difference between the BFrTr thresholds measured at the same temperature for the $d = 1.4$ μm- and $d = 9.8$ μm-thick cells is less than 1% (it is, in fact, hidden in the statistical noise of about 2%). This result corresponds to $L_e^m \approx 10$ nm, i.e., again to a very strong anchoring energy $W_s^m \approx 1$ mJ/m.

In contrast to the birefringence measurements, the dielectric technique does not allow for the investigation of the dynamics of the FrTr and BFrTr.

APPENDIX F: IN-LAYER ORIENTATIONAL ORDER OF THE M_X PHASE

The x-ray scattering data clearly show that the M_X phase is smectic and that the bent-shaped BNA-76 dimers are completely intercalated in the smectic layers. Indeed, the smectic period d_0 is about half of the dimer length L, $d_0 \approx L/2$, which indicates that the smectic layers are not formed by the dimer molecules but by the monomer mesogenic units. Each monomer in a layer belongs to a dimer that spans, with equal probability, over one of the two adjacent layers. (Strictly speaking, this description is valid for the M_X phase of the pure BNA-76 compound. In the case of the BP12 mixture that is investigated here, a part of the monomers in each layer is replaced by the 6-PEPP-N molecules, which have approximately the same length as the monomer. For simplicity and because of the small amount of 6-PEPP-N, we neglect this detail here.)

Moreover, the intensity distribution in the wide-angle diffuse scattering ring and the relation $d_0 \approx L/2$ both suggest that the M_X phase is an orthogonal smectic phase of the bent-shaped dimers, i.e., that the (primary) director **n** of the phase, defined by the average orientation of the long axes of the dimers, is perpendicular to the smectic layers. This feature is unambiguously confirmed by our electro-optic observations: Without a field, the slow axis of the sample is along the normal to the layers. When the field is applied, the cell birefringence increases, but the slow axis keeps its orientation, which indicates that **n** is not reoriented by the field and remains perpendicular to the layers. Moreover, all our optical and electric experiments indicate that the M_X phase is biaxial. Therefore, the M_X phase of the bent-shaped dimers is an orthogonal biaxial smectic phase. This phase is already reported in the literature for several compounds under different names: SmA_b [42], SmC_M [51], etc. Here, we use the name SmA_b for this orthogonal biaxial smectic phase.

However, the orientational order of the monomers in each smectic layer also matters, since their para-axes are tilted with respect to $\mathbf{q}_0$, due to the bent shape of the dimers. Depending on the in-layer orientational order of the monomers, different structures of this phase are possible.

One of them, which has been already proposed for the intercalated smectic phase of bent-shaped dimers [9–11,13,32–34,41,44–49], is illustrated in Fig. 3. In this case, the para-axes **p** of all the monomers in a layer are tilted in the same direction, resulting in a SmC layer with tilted director $\langle \boldsymbol{p} \rangle$. Because of the bent shape of the dimer molecules, the tilt of $\langle \boldsymbol{p} \rangle$ alternates from one layer to the next, as in an anticlinic smectic C, SmC_A [4,42,49,50]. Statistically, due to the intercalation, the monomers of a given layer belong with the same probability to dimers oriented in opposite directions. Therefore, the average polarization of the layer vanishes, and the phase is not ferroelectric despite the similarity of its structure (except for the intercalation) with that of the antiferroelectric SmA_{PA} phase [14,15]. Then, the in-layer orientational order of the monomers is described by a biaxial tensor with a tilted primary director. We refer to this kind of intercalated smectic as SmA_b(SmC_A), where we specify the in-layer orientational order of the monomers in brackets.

Another possible structure is sketched in Fig. 25. In that case, the tilt of the monomers in each layer is doubly degenerate, with their para-axes lying in the same plane but tilted (with equal probability) one way or the other. The in-plane order parameter tensor of the monomers is again biaxial, but the primary director $\langle \boldsymbol{p} \rangle$ is perpendicular to the layer. As the in-plane order of the monomers is SmA_b, we refer to this structure as SmA_b(SmA_b). (We note that in this case the monomer layer has the same D_{2h} symmetry as in the classic SmA_b phase formed by boardlike molecules.)

Yet another in-layer structure of the phase, with degenerate azimuthal orientation of the tilted monomers, is compatible with our x-ray scattering data. In this case, the para-axes of the monomers are randomly distributed on a cone, forming a "de Vries SmA" structure [86]. However,
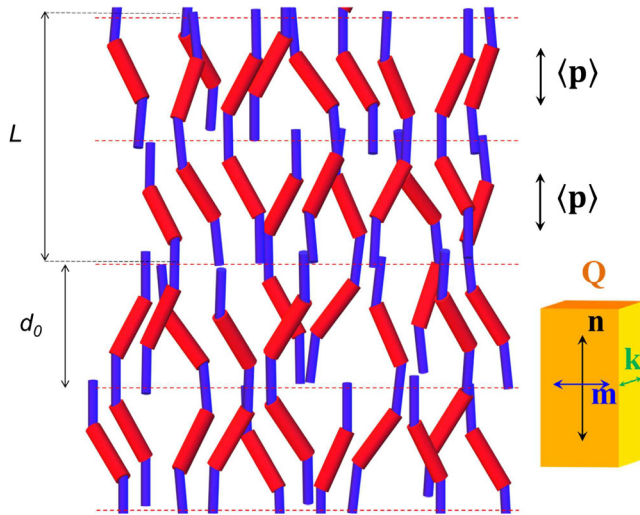
FIG. 25. Sketch of an M_X phase with a $SmA_b(SmA_b)$ structure. As for the $SmA_b(SmC_A)$ structure presented in Fig. 3, the monomer units form smectic liquid layers with thickness $d_0 \approx L/2$. The dimers span again two adjacent layers and form an orthogonal intercalated smectic phase with biaxial orientational order parameter tensor **Q**. However, in each layer, the average orientation of the monomer axis **p** is parallel to the layers normal, because the tilt of the monomers is doubly degenerated, which is quite unlikely due to a strong energetic penalty.

this highly symmetric phase is uniaxial and is, therefore, ruled out by our optical and electric data for the M_X phase.

Our experimental data do not actually give any clear evidence about which of the $SmA_b(SmC_A)$ or $SmA_b(SmA_b)$ microscopic structures applies to the M_X phase. Both structures should present similar elastic, optic, and dielectric properties. However, there are clear theoretical arguments in favor of the $SmA_b(SmC_A)$ structure. Indeed, the relative stability of the two structures is defined by the balance between the energy and entropy terms, both positional and orientational, in the free energy. The condensation of the smectic order of the monomers indicates that the positional energy gain outweighs the positional entropy loss. The energy gain for parallel in-layer orientation of the monomers favors their uniform tilt within each layer, i.e., their SmC-like in-layer organization. Because the monomers are very anisotropic and may form mesophases when nondimerized, this energy term must be very large, of the order of several kT per monomer [87–89]. Note that, for N_{TB}-forming compounds like BNA-76, the angle between two neighboring monomers tilted in opposite ways is approximately 60°, resulting in a prohibitively large energy cost. However, the orientational entropy term favors the SmA_b organization of the monomers in the layer. Nevertheless, the tendency of the monomers to form uniaxial nematic phases strongly indicates that the orientational entropy term, which is even larger in the N phase due to its higher orientational disorder, cannot balance the unfavorable interaction energy of the

$SmA_b(SmA_b)$ structure. Therefore, we conclude that the organization of the monomer layers in the M_X phase is that shown in Fig. 3. Additional factors not discussed here, for example, the close packing of the monomers in the layer, should also favor the SmC_A-like organization of the monomers.

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