Discovery of quasi-six-coordinated layered phase of PBr₃ at high temperature and pressure

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Layered trihalides, mainly formed by metal elements and halogens, are promising candidates for spintronic devices and van der Waals heterostructures, but are rarely found in nonmetal halides. Motivated by the purpose of searching for the compelling layered configuration in a covalent system, we performed systemically high-pressure experiments to explore the crystal structures in typical PBr₃ with diamond anvil cell technique. At room temperature, PBr₃ crystallizes into a molecular crystal with orthorhombic symmetry above 0.9 GPa and is found to persist up to at least 43.9 GPa as confirmed by *in situ* Raman and x-ray diffraction measurements. Strikingly, a novel layered phase with a space group $P2_1/c$ is observed at ~27.0 GPa and ~1800 K, which is characterized by quasi-six-coordination of P atoms with nearby Br atoms. In addition, first-principles calculations indicate that the unusual coordination of the P atom in $P2_1/c$ phase is highly correlated with the pressure-induced generation of pnictogen bonds. Our current findings not only expand the understanding of the phase diagram of PBr₃, but also show an interesting transformation from molecular crystal to an extended layered phase achieved in a broad range of nonmetal trihalides.

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I. INTRODUCTION

Molecular materials, characterized by strong intra- and weak intermolecular bonds, will tend to form extended threedimensional geometries under extreme pressures, leading to intriguing phenomena such as phase transition [1–3], metallization [4,5], decomposition [6], disproportionation [7], and polymerization [8,9], thus providing fundamental knowledge in physics, chemistry, and planetary science. Recent reports reveal that high temperatures are pivotal for observing the above transformations owing to the kinetic barriers between different phases and chemical reactions during compression at room temperature, i.e., disproportionation of SO₂ [8] and phase transitions of CF₄ [10] and SrOsO₃ [11] were reported at high-pressure and high-temperature (HPHT) conditions, providing unequivocal experimental evidence for pursuing accurate energetic landscapes in simple molecular systems.

Group V elemental trihalides AX_3 (A = N, P, As, Sb, Bi; X = F, Cl, Br, I) are a family of simple four-atom compounds, which possess versatile applications in chemistry and materials fields [12–14]. Among them, the nitrogen, phosphorus, arsenate, and selenium trihalides usually adopt trigonal

pyramidal geometry with C_{3v} point group symmetry, where A atom is sp^3 hybridized and overlaps with three halogen atoms to form covalent compounds AX₃ [15–21]. In contrast, ionic type compounds AX₃ such as BiI₃, where the metallic bismuth atom donates three p electrons to iodine atoms, tend to form a six-coordinated layered structure which is characterized by layers of bismuth ions that are octahedrally surrounded by six iodine ions so that they share edges with neighboring octahedra [21,22], the layers are stacked on top of one another with weak van der Waals bonding between adjacent layers. In addition CrI3 and VI3, which are isostructural to BiI₃, possess fascinating properties such as magnetic ordering and anisotropic electrical conductivity [23,24]. In previous reports, pressure has been confirmed to tune nonmetal into metal [25-28], which raises a scientific question about whether the trigonal pyramidal covalent compounds AX₃ could be transformed into a layered six-coordinated ionic structure under external compression.

In this paper, we present a pressure-induced structural evolution of PBr₃ by experimental and theoretical approaches up to 45.0 GPa. The *Pnma* phase of PBr₃ is formed at a freezing pressure of about 0.9 GPa and persists up to at least 43.9 GPa. Moreover, a layered phase (space group $P2_1/c$) with quasisix-coordinated phosphorus is generated after laser heating at 27.0 GPa and is stable down to 1.5 GPa during decompression. Further first-principles calculations reveal that the generation of pnictogen bonds in $P2_1/c$ phase results in increased coordination of P atoms. The discovery broadens physical properties of PBr₃ and extends its potential applications as well as those of other two-dimensional functional materials. This study will pave a new avenue towards understanding the

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chemical bonding and compression mechanisms in AX_3 and expand the realm of two-dimensional functional materials in nonmetal trihalides.

II. METHODS

Phosphorus tribromide (PBr₃) (Alfa Aesar, purity 99%) was loaded into symmetric diamond anvil cells with a culet diameter of 200 µm. T-301 gasket was preindented to 20 µm thickness and a sample hole of 100 µm was drilled. The liquid sample and ruby pieces were loaded into the chamber in a glove box filled with argon. Pressures were determined using Raman shift of diamond anvil [29] and ruby [30] luminescence method. In situ x-ray diffraction (XRD) experiments were performed at the 4W2 beamline of the Beijing Synchrotron Radiation Facility (BSRF). Using a pair of Kirkpatrick-Baez mirrors to focus monochromatic synchrotron radiation x-ray beam with a wavelength of 0.6199 Å and spot size of $10 \times 30 \,\mu\text{m}^2$. The two-dimensional (2D) xray diffraction patterns at different pressures were recorded by 2D imaging plate detector (MAR-3450). The instrument parameters were calibrated using a CeO₂ standard. The acquisition time of each 2D XRD pattern was 300 s. The integration and processing of 2D XRD patterns were carried out by using DIOPTAS program [31]. Rietveld refinements of XRD patterns were performed using GSAS II package [32]. Additionally, pressure dependencies of the unit-cell volume were fitted by the third-order Birch-Murnaghan equation of state (EOS) [33]. Raman spectra were collected using a homemade Raman instrument (Acton Standard Series SP-2556 Imaging Spectrograph and PyLoN:100BR eXcelon CCD) with 632.8 nm excitation in backscattering geometry, and the Raman signals were collected using a 20× Mitutoyo long working distance objective and dispersed by a 1200 mm^{-1} grating. Laser-heating experiments were excited by a Nd: YAG laser (1064 nm) with a power of 5 W. A Mitutoyo $10 \times$ (N.A. = 0.26) NIR objective was used to focus the laser and the typical focal spot diameter is about 10 µm. The temperature was determined using the emission spectrum of the black body radiation within the Planck's radiation law [34].

First-principles calculations based on density functional theory (DFT) [35,36] were performed within the generalized gradient approximation (Perdew-Burke-Ernzerhof functional) [37], and the projector augmented-wave method [38,39] as implemented in the VASP code [40,41]. The electron-ion interaction was described with $3s^23p^3$ and $4s^24p^5$ configurations considered as the valence electrons of P and Br, respectively. Setting the electronic wave functions cutoff energy to 500 eV and Sampling Brillouin zone by Γ -centered k-point meshes with a value of $2\pi \times 0.03 \text{ Å}^{-1}$ are to ensure that the convergence criteria for structural optimizations was set to about 1 meV/atom.

III. RESULTS AND DISCUSSION

A. High-pressure experiments of molecular phase

PBr₃ is a colorless liquid at ambient condition and its Raman spectrum is characterized by two stretching modes $\omega_1(A)$ and $\omega_3(E)$ and two deformation modes $\omega_2(A)$ and $\omega_4(E)$, which are located at 396 cm⁻¹, 378 cm⁻¹, 160 cm⁻¹,



FIG. 1. High-pressure Raman spectra of PBr₃, where ω represents the Raman mode.

and 115 cm⁻¹, respectively [42]. Among them, ω_3 and ω_4 are twofold degenerate, and the detailed Raman peaks assignments are given in Table S1 [43,44]. In order to know the pressure response of PBr₃, high-pressure Raman measurements of PBr₃ were performed up to 43.9 GPa shown in Fig. 1. It can be seen that a liquid-solid phase transition occurs at about 0.9 GPa evidenced by a sudden narrowing in the full wide at half maximum (FWHM) and the appearance of lattice modes, indicating the crystalized PBr₃ possesses a relatively uniform molecular bonding length and angle than that of liquid phase. This transition is also confirmed by the visual observation of crystallization in the chamber (see Supplemental Material, Fig. S1 [44]). With increasing pressure, except ω_3 band, all other Raman peaks shift to high wavenumber region, suggesting the reduced bond lengths of their corresponding normal vibrations are shortened under external compression. As changes in response of the Raman mode with pressure could provide useful information about phase transitions and chemical changes et al. [45-47], which are characterized by missing and splitting of vibrational modes, appearance of new modes, and sudden changes in the slope of the frequency-pressure curve at certain pressure. The Raman frequency-pressure relationships were plotted and given in Fig. S2 [44]. Consistent with the above analysis about the liquid-solid phase transition, sudden slope changes in the curves and appearance of lattice modes are observed at 0.9 GPa. Furthermore, the twofold-degenerated bands ω_3 and ω_4 start to split when the pressure approaches 5.0 and 10.0 GPa, respectively, indicating a pressure-induced lowering of molecular point group symmetry occurs. From the high-pressure Raman spectra and frequency-pressure relationships of PBr₃, taken



FIG. 2. (a) Rietveld refinement of the crystal structure of *Pnma* phase at 1.5 GPa with a wavelength of 0.6199 Å. The black circles, red curves, and blue curves correspond to the experimental data, Rietveld refinement fits, and residues, respectively. (b) EOS and crystal structure of *Pnma* phase where the square symbols and solid line represent the experimental data and fit, respectively. Brown and beige balls represent the P atoms and Br atoms, respectively.

together, no obvious phase transition could be concluded in the pressure range of 0.9–43.9 GPa.

In order to clarify the crystal structure and the compression behavior of the solid phase, we performed synchrotron XRD measurements of PBr₃ up to 42.1 GPa. The XRD patterns were indexed in terms of an orthorhombic unit cell (space group *Pnma*) with lattice parameters of a = 7.658 Å, b =9.723 Å, and c = 6.218 Å at 1.5 GPa [Fig. 2(a)]. The *Pnma* phase was characterized as four formula units per unit cell and each P atom is coordinated by three Br atoms. From the evolution of XRD (see Supplemental Material, Fig. S3 [44]), it is obvious that the *Pnma* phase is stabilized to 42.1 GPa, which is consistent with the Raman results. Furthermore, we fitted the equation of state (EOS) of the *Pnma* phase [Fig. 2(b)] and obtained the bulk modulus B₀ = 15.4 GPa and pressure derivative B'₀ = 4.2 by using V₀ = 129.4 Å³ [18].

B. High-pressure experiments of quasi-six-coordinated layered phase

Under high pressure, enthalpy is the main parameter that determines whether phase transitions or chemical reactions occur. However, due to the kinetic barrier, they cannot always be observed in the room-temperature compression process [1]. High temperature can increase the kinetic energy of atoms, thus inducing the system to surpass the energy barrier and stabilize at a relatively lower enthalpy condition. In recent years, laser-heating diamond-anvil-cell technique is an efficient means to provide a high-temperature and pressure environment [48,49]. Using this technique, phenomena such as phase transition and disproportionation could be observed. In this report, PBr₃ was laser heated to about 1800 K at 27.0 GPa, and the corresponding Raman spectrum is given in Fig. 3. Compared to the Raman spectrum of Pnma -structured PBr₃ at 27.0 GPa before laser heating, the Raman spectrum of that after laser heating has more and sharper Raman peaks. To qualify the width of the Raman peaks, the FWHM of each Raman peak of PBr3 before and after laser heating at 27 GPa are shown in Fig. S4 [44]. Except the newly arisen peaks located at 228, 316, and 359 cm⁻¹ in $P2_1/c$ phase (after laser heating), the FWHM values of *Pnma* phase are larger than that of $P2_1/c$ phase. The *Pnma* phase of PBr₃ has a character of molecular crystal, and the optical phonon modes are 3N-6=6 where N=4 is the atomic number in each molecular PBr₃. While after laser heating, the number of Raman peaks is almost doubled, suggesting an extended structure or a molecular unit containing more than 4 atoms is built. In addition, P-Br stretching modes ω_1 and ω_3 shift to the low wavenumber region, indicating the P-Br bond is elongated in the newly formed phase. Moreover, the Raman measurements were performed at each step of decompression and shown in Fig. S5 [44]. Compared to the *Pnma* phase, a



FIG. 3. Raman spectra of PBr_3 before and after laser heating at 27.0 GPa.



FIG. 4. (a) Rietveld refinement of the crystal structure of phase $P2_1/c$ at 27.0 GPa with a wavelength of 0.6199 Å. The black circles, red curves, and blue curves correspond to the experimental data, Rietveld refinement fits, and residues, respectively. (b) EOS of the $P2_1/c$ phase where the square symbols and solid line represent the experimental data and fit, respectively. Crystal structure of phase $P2_1/c$. Brown and beige balls represent the P atoms and Br atoms, respectively. Red dashed represents longer bonding than that of bicolor line.

softening behavior of a few Raman modes is observed in the new-formed phase, which should be attributed to the existence of stronger intermolecular interaction. Meanwhile, it is obvious that the newly formed phase is stable down to 1.8 GPa and then transforms back to the Pnma phase at 1.2 GPa.

In order to know the crystal structure of the new-formed phase of PBr₃, we performed synchrony XRD measurements, and the XRD pattern was collected and indexed as a monoclinic structure (space group $P2_1/c$) with lattice parameters of a = 5.481 Å, b = 8.018 Å, c = 6.714 Å, and $\beta = 107.5375^{\circ}$ at 27.0 GPa [Fig. 4(a)], which is isostructural with SbI₃ [50,51]. As shown in Fig. 4(b), the $P2_1/c$ phase consists of a single layer of P atoms sandwiched between two layers of Br atoms, and each P atom is coordinated by three nearest-neighboring and three next-nearest-neighboring Br atoms, forming a PBr₆ octahedra. From the previous high-pressure results, it is undoubtedly demonstrated that pressure could induce bandgap closure and metallization in elements and compounds [1,2,4,5,52] and could decrease electronegativities of elements in periodic table (except Tb, Dy, and Ho) [53]. Moreover,

(a)

for ionic type compounds AX₃, metal element A donates three p electrons to X atoms forming an ideal six-coordinated layered structure [21,23–25]. Therefore, it is reasonable that the metallicity of nonmetal element P increases and the electronegativities decrease under high pressure leading to the discovery of quasi-six-coordinated layered structure of PBr₃. The decompressed XRD patterns of $P2_1/c$ phase are shown in Fig. S6 [44] suggesting the $P2_1/c$ phase can be stable down to 1.5 GPa, which is consistent with the previous Raman analysis. In addition, we fitted the EOS of $P2_1/c$ phase in Fig. 4(b) and yield the bulk modulus $B_0 = 14.4$ GPa, pressure derivative $B'_0 = 4.4$, and $V_0 = 123.8$ Å³. Compared to the Pnma phase, the increase in the bulk modulus and decrease in V_0 indicates the $P2_1/c$ phase is a close-packed phase.

C. First-principles calculations To gain a deeper insight into the mechanism of phase

transition of PBr₃, we investigated electronic structures of

the *Pnma* and $P2_1/c$ phases at 30 GPa (see Fig. 5). Results

(b) 3 P-s P-s Energy (eV) Energy (eV P-p **P-***p* Br-Br-Br-p Br-p Br-a Br-a U PDOS нс EM 1A PDOS R τ х Z D

FIG. 5. Calculated electronic band structures (left panel) and partial density of state (right panel) of (a) Pnma structure and (b) $P2_1/c$ structure at 30 GPa.



FIG. 6. Electron localization functions within (a) the (h k l) (3.569 0 1) plane of *Pnma* phase and (b) the (h k l) (0 1 0) plane of $P2_1/c$ phase at 30 GPa.

indicate both are indirect gap semiconductors and show a continuous decrease in band gap as the pressure increase (see Supplemental Material, Fig. S7 [44]). The images of the sample chamber (Pnma phase) at high pressure were given in Fig. S8 [44]. It can be clearly seen that the color of the sample gradually deepens, which is consistent with the theoretical results of the band gap. Furthermore, the band gap of $P2_1/c$ phase is larger than that of *Pnma* phase in the pressure range of this study (see Supplemental Material, Fig. S7 [44]). The projected density of states (PDOS) of Pnma -PBr₃ [see Fig. 5(a) indicates that valence band is occupied mostly by Br-4 p orbital, while the Br-4 p and P-3 p orbitals show strong hybridization as shown in conduction band. The $P2_1/c$ phase adopts a similar p - p hybridization with the *Pnma* phase [see Fig. 5(b)]. Moreover, we calculated Bader charges of the two phases, respectively (see Supplemental Material, Table S2 [44]), which indicates that there is no obvious difference between the two phases.

We have calculated electron localization functions (ELFs) of *Pnma* phase [Fig. 6(a)] and $P2_1/c$ phase [Fig. 6(b)], respectively. As shown in Fig. 6(a), it is clear that the localized charge distribution is seen within the molecule PBr₃, which is attributed to the strong P-Br covalent bond. Figure 6(b) displays ELF in the (0 1 0) plane of $P2_1/c$ phase, as a result of the layered atomic distribution, electrons are mainly distributed within the sandwiched layers of P and Br atoms. In addition, P-Br bonds can be divided into three strong and three weak bonds with the bond lengths ranging from

2.26 Å to 2.30 Å and 2.68 Å to 2.80 Å, respectively (see Supplemental Material, Fig. S9 [44]), resulting in a quasi-sixcoordinated layer structure of PBr₃. Among them, the strong bonds are relative to primitive P-Br covalent bonds as the Pnma phase, and the weak bonds are attributed to noncovalent interactions-pnictogen bonds, which is the result of Coulomb attraction between sites of unequal charge density [54–58]. Recently, scientists found that there are three σ holes that exist in molecule PBr₃, leading to Coulomb attraction between σ holes and atoms characterized by nucleophilicity such as halogen atoms, thus resulting in the generation of pnictogen bonds [59]. In conclusion, the generation of layered phase $P2_1/c$ in PBr₃ is related for two reasons: firstly, high temperature induces an increase in kinetic energy of PBr₃ molecules, leading to greater freedom of movement of the molecules; Secondly, intrinsic σ holes in PBr₃ molecules attract adjacent Br atoms of other PBr3 molecules by Coulomb force, leading to generation of quasi-six-coordinated layer phase $P2_1/c$.

IV. CONCLUSIONS

To sum up, we systematically investigate the phase transitions of PBr₃ up to 43.9 GPa by a joint high-pressure experimental (Raman scattering, synchrotron XRD, and laser-heating diamond-anvil-cell technology) and theoretical approaches. Upon room temperature compression, a Pnma phase is solidified from the liquid PBr₃ at 0.9 GPa and is found to be stable up to 43.9 GPa. Intriguingly, a novel layered phase (space group $P2_1/c$) is generated from the *Pnma* phase at ~ 27.0 GPa and ~ 1800 K, which is characterized by quasi-six-coordination of P and Br atoms. Furthermore, the electronic band structures PDOS and ELFs of Pnma and $P2_1/c$ phases are analyzed, which indicated that the bonds of $P2_1/c$ phase can be divided into covalent bonds and pnictogen bonds. This report realizes the phase diagram of PBr₃, paves a new avenue towards the synthesis of layered vdW trihalides from nonmetal elements and halogens and provides a reference for generation of three-dimensional extension structure from molecules under the influence of pnictogen bonds.

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chamber at 0.2 and 0.7 GPa; Raman frequency-pressure curves of *Pnma* phase of PBr₃; High-pressure XRD patterns of the *Pnma* phase up to 42.1 GPa; The FWHM of each Raman peak of PBr₃ before and after laser heating at 27 GPa; Raman spectra of the *P*2₁/*c* PBr₃ in decompression; Evolution of XRD patterns for phase *P*2₁/*c* as a function of pressure; The band gap of the *Pnma* and *P*2₁/*c* phases as a function of pressure; Images of the sample chamber (*Pnma* phase) at high pressure; Bader charges of *Pnma* and *P*2₁/*c* phases at 25 GPa; ELF within the intralayer plane of *P*2₁/*c* phase at 25 GPa.

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