Mn-induced multiple magnetic ground states in $Sr_3(Ru_{1-x}Mn_x)_2O_7$

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Transition-metal oxides provide an excellent platform to generate ground states by tuning the various degrees of freedom via stimuli such as chemical doping. By investigating the electrical, magnetic, and thermodynamic properties of several compositions of the bilayered $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with $x \ge 0.36$, we complete its x-T phase diagram and unveil a spin-glass regime. Upon Mn doping, the ground state of the system changes from the short-range antiferromagnetically (AFM) coupled metal ($0 \le x \le 0.05$) to the E-type AFM-ordered insulator ($0.05 < x \le 0.20$) to the spin-glass insulator ($0.20 < x \le 0.66$) to the canted AFM-ordered insulator ($x \sim 1.0$). The complex evolution of the magnetic ground state results from the competition between AFM and ferromagnetic (FM) interactions. The peculiar spin-glass state between two long-range magnetically ordered states implies the significance of spin disorder induced by the intermixing of Mn and Ru ions with the lowest transition temperature for Ru : Mn $\sim 1:1$ ($x \sim 0.5$).

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I. INTRODUCTION

Correlated transition-metal oxides (TMOs) have attracted considerable interest due to the abundance of exciting physical phenomena arising from the strong interactions between charge, lattice, orbital, and spin degrees of freedom. Among TMOs, the Ruddleson-Popper strontium ruthenate series with the general formula $Sr_{n+1}Ru_nO_{3n+1}$ $(n = 1, 2, ..., \infty)$ is widely known for intriguing physical properties such as Mott metal-insulator transition (MIT), unconventional superconductivity, inherently competing magnetic interactions, and field-induced quantum criticality [1-5]. In this series, the bilayered $Sr_3Ru_2O_7$ (n = 2) crystallizes in a slightly distorted tetragonal lattice structure, as shown in Fig. 1(a), that consists of two layers of RuO₆ octahedra planes stacked along the c axis. It is a paramagnetic metal developing short-range antiferromagnetic (AFM) correlation $< \sim 18 \text{ K}$ [1,6]. However, it has been reported to possess ferromagnetic (FM) fluctuations as well [7,8]. The presence of both AFM and FM interactions results in a fragile magnetic ground state that shows remarkable response to external stimuli such as magnetic field, pressure, and chemical substitution [1,3,9], for example, the application of magnetic field results in the second-order metamagnetic phase transition [3,4]. By pushing the temperature $T \rightarrow 0$ K, the field-induced quantum critical point (QCP) can be reached, and the electronic nematicity is seen around the QCP [10-12]. Similarly, the application of hydrostatic pressure can stabilize the FM interaction in the system [6,13].

Competing interactions have been unveiled through chemical doping in either the Sr or Ru site. While isovalent doping, a partial substitution of Sr²⁺ by smaller Ca²⁺ results in spin-glass (SG) and AFM (A-type) insulating states in $(Sr_{1-x}Ca_x)_3Ru_2O_7$ [14–17]. As the physical properties of Sr₃Ru₂O₇ are determined by Ru bands [18], a partial substitution of Ru by other transition metals can dramatically tip the inherent interactions in Sr₃Ru₂O₇, resulting in a rich phase diagram [1,19]. For example, introducing 5d Ir leads to initially an AFM metallic phase (0.4 < x < 0.6)and eventually an insulating canted AFM state (x > 0.6) in $Sr_3(Ru_{1-x}Ir_x)_2O_7$ [19]. Such a result is unexpected, given that Ir has more extended d orbitals with more electrons than that of Ru. On the other hand, when Ru ions are substituted by 3d Mn, an equally complex evolution of the magnetic and electronic ground states are observed in $Sr_3(Ru_{1-x}Mn_x)_2O_7$ [1]. Upon Mn doping (x), the electronic ground state becomes insulating with the MIT temperature (T_{MIT}) increasing with x. Simultaneously, the introduction of Mn drives the doped system into long-range AFM ordering at T_N with $T_N < T_{MIT}$. Unlike T_{MIT} , which increases with increasing x, T_N varies nonmonotonically with x in the region 0.05 < x < 0.20, revealing a local maximum at $x \sim 0.16$. Such a doping dependence of $T_{\rm N}$ is extremely intriguing, implying complex interplay between various magnetic interactions. While significant attention has been paid to the low Mn-doping region of this series, information about how the system behaves beyond the long-range AFM order in the region 0.05 < x < 0.20 is lacking in the literature. In this paper, by measuring the electrical and magnetic properties of x = 0.36, 0.55, 0.66, and 1.0 single crystals, we unveil the magnetic interplay that leads to the

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FIG. 1. (a) Crystal structure of $Sr_3(Ru_{1-x}Mn_x)_2O_7$. (b) Temperature dependence of the in-plane resistivity for x = 0.36, 0.55, and 1.0. Top inset: *x* dependence of the activation energy estimated via the Arrhenius fitting of the resistivity data. Bottom inset: Enlarged view of the temperature dependence of resistivity for x = 0.36 between 250 and 375 K with the metal-insulator transition temperature T_{MIT} indicated by an arrow. (c) Temperature dependence of the magnetic susceptibility measured under H = 1 T parallel to the *ab* plane (χ_{ab}) for x = 0.36, 0.55, 0.66, and 1.0. Inset shows the enlarged view of χ_{ab} for x = 1.0. (d) Phase diagram of $Sr_3(Ru_{1-x}Mn_x)_2O_7$. The colored points are taken from Refs. [1,20], and the black points are from this paper.

unusual evolution of the magnetic ground state in the higher Mn-doped region (x > 0.20) and complete the phase diagram of Sr₃(Ru_{1-x}Mn_x)₂O₇. Our experimental observations reveal the presence of the SG ground state in a wide doping range with the lowest transition temperature T_{SG} located at $x \sim 0.5$. As x approaches 1.0, the SG behavior is weakened, while a long-range magnetic order is formed in Sr₃Mn₂O₇ (x = 1.0) below $T_N = 160$ K. Our complete magnetic phase diagram covering $0 \le x \le 1$ allows us to understand the underlying physics of Sr₃(Ru_{1-x}Mn_x)₂O₇ thoroughly by considering multiple magnetic interactions.

II. METHODS

Single crystals of $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with several nominal x_{nominal} values between 0.3 and 1.0 ($x_{\text{nominal}} = 0.3, 0.6, 0.75,$ and 1.0) were grown in a floating zone furnace under 0.9MPa oxygen pressure. The phase purity of the crystals was checked via room-temperature x-ray diffraction (XRD) using a PANalytical Empyrean x-ray diffractometer with Cu Kα radiation (see Fig. 2). The chemical compositions of the crystals were determined using energy dispersive x-ray spectroscopy (EDS) equipped by FEI Quanta 200 under vacuum environment. The actual Mn/Ru ratios of the crystals were found to be x = 0.35, 0.55, 0.66, and 1.0 (see Table I); the actual x values are used in this paper. To avoid oxygen deficiency, the as-grown samples were annealed in an oxygen atmosphere at 550-600 °C for 6 wk. The magnetization measurements were performed in a magnetic properties measurement system (MPMS-7 T, Quantum Design), and the electrical transport and specific heat measurements were performed in a physical properties measurement system (PPMS-14 T, Quantum Design). The high-temperature magnetization measurements were done using a vibrating sample magnetometer option. High-resolution scanning transmission electron microscopy (STEM), elemental electron energy loss spectroscopy (EELS) elemental mapping, and energy dispersive x-ray spectroscopy (EDS) elemental mapping measurements were performed using Thermo Scientific Spectra 300 (S)TEM equipped with X-CFEG. The microscope settings for the STEM images were 200 kV, C2 aperture of 50 μ m, 23 mrads semiconvergence angle, cathodoluminescence of 320 mm EFSTEM with indicated magnification of 7.2 MX and beam current of 129 pA. For the high-resolution (HR) EELS spectra, GIF Continuum HR was utilized with entrance aperture of 5 mm, dispersion of 1.5 eV/channel, semicollection angle of 73.5 mrads, dwell time of 0.005 s per pixel, total acquisition time of 117 s, and high-loss exposure of 4.95 ms. For EDS mapping, the total acquisition time was 1268 s with the probe current of 129 pA.

III. RESULTS AND DISCUSSION

Figure 1(b) shows the temperature dependence of the inplane electrical resistivity ρ_{ab} for x = 0.36, 0.55, and 1.0 samples. Overall, ρ_{ab} increases with increasing x in the entire temperature range measured. For x = 0.36, upon cooling, ρ_{ab} initially decreases, displaying a metallic behavior until $T_{\text{MIT}} \sim 300 \text{ K}$, below which ρ_{ab} increases sharply as the system enters an insulating state. For x = 0.55 and 1.0, the system displays an insulating behavior without sign of MIT below 400 K. By fitting the data to the Arrhenius function $\rho_{ab}(T) = \rho_0 \exp(\Delta/2k_BT)$, we find that the activation energy



FIG. 2. (a) Room temperature x-ray diffraction (XRD) patterns obtained from the flat (0 0 l) surfaces of $Sr_3(Ru_{1-x}Mn_x)_2O_7$ single crystals with x = 0.36, 0.55, 0.66, and 1.0. (b) Magnified view of the intensity profiles of the (0 0 10) peak in the four samples. (c) x dependence of the lattice parameter c in $Sr_3(Ru_{1-x}Mn_x)_2O_7$. The error bars represent the standard errors. Note that the relative shift in the peak position observed in (b) and the decrease in lattice parameter c observed in (c) agree with the difference in the effective ionic radius between Ru and Mn.

 Δ increases from ~0.05 eV for x = 0.36 to ~0.15 eV for x = 1.0 [see the inset of Fig. 1(b)].

Figure 1(c) shows the temperature dependence of the magnetic susceptibility χ_{ab} measured under H = 1 Tesla along the *ab* plane for x = 0.36, 0.55, 0.66, and 1. Note that, for x < 1, the magnitude of χ_{ab} increases with increasing *x*, indicating enhanced magnetic interactions with increasing the Mn content. For $0.36 \le x \le 0.66$, χ_{ab} increases with decreasing temperature until $T_{\rm M}$, indicated by black arrows in Fig. 1(c), below which the zero-field cooled (ZFC) susceptibility is highly suppressed, whereas the field cooled (FC) susceptibility only shows a slight kink. In contrast, χ_{ab} for Sr₃Mn₂O₇ (x = 1) has a distinct characteristic, as highlighted in the inset of Fig. 1(c). Upon decreasing temperature, χ_{ab} initially increases until ~300 K, which then decreases. It increases again below $T_{\rm M} \sim 160$ K. This peculiar behavior is observed in both ZFC and FC curves.

Based on data shown above and previous work [1], we construct a complete phase diagram for $Sr_3(Ru_{1-x}Mn_x)_2O_7$, which is presented in Fig. 1(d). It consists of two phase boundary lines: one MIT line that separates the high-temperature metallic phase (Region I) from the low-temperature insulating phase and another magnetic transition line (T_M) that separates the high-temperature paramagnetic insulating phase (Region II) from the low-temperature magnetic phases. Region III from the low-temperature magnetic phases. Region III represents the short-range AFM region in the low

TABLE I. Nominal Mn/(Mn + Ru) ratio (x) values in the four single-crystal samples investigated along with the x values measured from EDS measurements.

Sample	Nominal x value in $Sr_3(Ru_{1-x}Mn_x)_2O_7$	Measured x value from EDS
1	0.3	0.36 ± 0.01
2	0.6	0.55 ± 0.02
3	0.75	0.66 ± 0.02
4	1.0	_

Mn-doping region. The long-range AFM interactions are stabilized upon further increasing *x*, resulting in an E-type AFM order for 0.06 < x < 0.2 [8,21] with the maximum transition temperature located at $x \sim 0.16$ (Region IV). The dome-shaped $T_{\rm M}$ vs *x* indicates that there are competing magnetic interactions, which reduce $T_{\rm M}$ at x > 0.16. Remarkably, $T_{\rm M}$ reaches the minimum at $x \sim 0.5$, i.e., Ru:Mn = 1:1 (Region V). On the other end of the phase diagram, pure Sr₃Mn₂O₇ (x = 1) was reported to be a G-type AFM below the Néel temperature $T_{\rm N} = 160$ K [20]. Thus, it can be inferred that a long-range AFM ordering is again established in the system with high Mn concentration (x > 0.66, Region VI).

To illuminate the nature of the magnetic ground states, we explore the magnetic properties of x = 0.36, 0.55, 0.66, and 1.0 in detail. Figure 3(a) shows $\chi(T)$ for x = 0.55 by applying H = 1 T along the *ab* (red data) and *c* (blue data) directions, respectively. Overall, the magnitude of $\chi_{ab} > \chi_c$, indicating magnetic anisotropy. The inset of Fig. 3(a) shows the temperature dependence of the inverse magnetic susceptibility $(1/\chi)$ taken under both ZFC and FC conditions. The experimental data at $T \gg T_{\rm M}$ can be described well by the Curie-Weiss formula $\frac{1}{\chi - \chi_0} = \frac{T - \theta_{CW}}{C}$, where χ_0 is the temperature independent susceptibility, θ_{CW} is the Curie-Weiss temperature, and C is the Curie-Weiss constant. From the fit, represented by the solid lines in the inset of Fig. 3(a), we obtained $\chi_0^{ab} =$ -1.03×10^{-3} emu/mol-Oe, $C^{ab} = 3.136(7)$ emu-K/mol-Oe, and $\theta_{CW}^{ab} = -28.4(3) \text{ K}$ for $H \parallel ab$ and $\chi_0^c = -1.05 \times 10^{-3} \text{ emu/mol-Oe}$, $C^{ab} = 3.19(1) \text{ emu-K/mol-Oe}$, and θ_{CW}^c = -40.8(8) K for $H \parallel c$. The negative values of θ_{CW} indicate that the magnetic coupling is dominantly AFM in both directions. Furthermore, from the fitted values of C, we can estimate the effective moment to be $\sim 5 \,\mu_{\rm B}/{\rm f.u.}$

Despite both the in-plane and out-of-plane magnetic interactions being seemingly AFM, the transition temperature $T_{\rm M} < \theta_{\rm CW}^{ab}$ and $\theta_{\rm CW}^c$ and the susceptibility curves do not represent trivial antiferromagnetism. To gain insight into magnetic interactions, we have measured the isothermal $M_{ab}(H)$ at different temperatures, as displayed in Fig. 3(b). Note that



FIG. 3. Magnetic properties of $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with x = 0.55. (a) Temperature dependence of magnetic susceptibility χ measured under magnetic field H = 1 T parallel to the *ab* (red) and *c* (blue) directions. Both zero-field cooled (ZFC) and field cooled (FC) data are presented. Inset: Temperature dependence of the inverse susceptibilities in both directions and the respective Curie-Weiss fits (solid lines). (b) Field dependence of magnetization M(H) at 2, 50, and 100 K for $H \parallel ab$ (main panel) and *c* (inset) directions. (c) Temperature dependence of χ_{ab} under various indicated H. The arrows mark the freezing temperatures T_f . Inset: T_f vs $H^{2/3}$ for both $H \parallel ab$ (red) and $H \parallel c$ (blue) configurations. The linear fits correspond to the de Almeida–Thouless relationship. (d) Time dependence of ZFC M_{ab} under H = 0.1 T at 10 K with a temporary cooling to 5 K. Inset: Data points from t_1 and t_3 plotted together highlighting magnetic memory. (e) Temperature dependence of FC χ_{ab} measured with stops at temperatures indicated by the arrows (black circles) and the memory (red) curve. (f) Comparison of 1 T FC $M_{ab}(H)$ curve with the ZFC $M_{ab}(H)$ curve at 2 K.

 M_{ab} varies linearly with H at 100 K but nonlinearly at 50 K. Below $T_{\rm M}$, a large hysteretic M(H) loop develops in both the *ab* and *c* directions below $T_{\rm M}$. This clearly indicates that, in addition to AFM interaction, there is FM interaction as well. The mixed AFM and FM interactions with unsaturated M up to 7 T suggest the presence of the SG magnetic state. To confirm, we measure the magnetic susceptibility under a different field. As depicted in Fig. 3(c), both T_M and the magnitude of χ decrease with increasing H. In canonical SGs, the freezing temperature $(T_f = T_M)$ and H follow the de Almeida–Thouless relationship [22,23] $H = \alpha (T_{SG} - T_f)^{3/2}$, where T_{SG} is the SG transition temperature for H = 0, and α is a function of magnetic interaction strengths. As shown in the inset of Fig. 3(c), the T_f decreases monotonously with increasing H with $T_f \sim H^{2/3}$ in both the ab and c directions, consistent with the SG behavior. From the linear fits, represented by the solid lines, we obtain $T_{SG}^{ab} \sim 25$ K and $a_{ab} \sim 19 \text{ T}$ and $T_{\text{SG}}^{ab} \sim 22 \text{ K}$ and $\alpha_c \sim 18 \text{ T}$.

An outstanding characteristic of SG is its nonequilibrium phenomena such as time-dependent magnetization, magnetic memory effect, and exchange bias (EB) [24–27]. Figure 3(d) shows the time dependence of ZFC M_{ab} (see the Appendix for detail) performed at $T_0 = 10$ K under H = 0.1 T. At T_0 , M_{ab} evolves with time during $t_1 = 1$ h following a stretched exponential function. Despite the sample being cooled to 5 K for $t_2 = 1$ h, the initial magnetic state is retained when it is warmed back to $T_0 = 10$ K ($t_3 = 1$ h). As illustrated in the inset of Fig. 3(d), the data collected during t_1 and t_3 are not affected by the cooling history during the t_2 duration, demonstrating the magnetic memory effect. Further verification of the magnetic memory can be seen in the FC magnetization (see the Appendix for detail), shown in Fig. 3(f), where the data (black) were collected while cooling under H = 0.1 T with temporary stops of $\Delta t = 1$ h with the removal of the applied field at temperatures indicated by black arrows. Once $\Delta t = 1$ h had elapsed, the field was reapplied, and cooling resumed. After reaching 2 K, the memory curve [red curve in Fig. 3(d)] was measured without any stops in the warming mode. Clearly, the system was "aged" by stopping during the cooling cycle, confirming the magnetic memory effect.

In addition to the magnetic memory effect, we also observed a prominent EB effect below T_{SG} , which is characterized by a shift in the magnetization hysteresis curve [28,29]. Figure 3(f) shows the $M_{ab}(H)$ curve measured at T = 2 K under both the ZFC (black curve) and FC (red) conditions. A leftward shift can be clearly seen in the FC curve compared with the ZFC curve. The exchange field H_E is quantified as $H_E = (H_{c1} + H_{c2})/2$, where H_{c1} and H_{c2} are the left and right coercive fields in the FC curve [indicated in the Fig. 3(f) by arrows], respectively. From the data taken under 1 T field



FIG. 4. Magnetic properties of $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with x = 0.36. (a) Temperature dependence of χ with H = 1 T parallel to *ab* and *c* directions, respectively. Inset: Temperature dependence of inverse susceptibility shifted vertically for clarity and Curie-Weiss fitting (solid lines). (b) Field dependence of isothermal magnetization at various temperatures with $H \parallel ab$ (main panel) and $H \parallel c$ (inset). (c) $\chi_{ab}(T)$ curves at indicated temperatures with arrows for $T_f(H)$. (d) T_f as a function of $H^{2/3}$ for $H \parallel ab$ and $H \parallel c$ with fits represented by solid lines. (e) Magnetic relaxation at 10 K for time durations t_1 and t_3 with a temporary cooling to 5 K for time duration t_2 . Inset: Data at 10 K during t_1 and t_3 showing the magnetic memory effect. (f) ZFC $M_{ab}(H)$ curve at 2 K along with FC $M_{ab}(H)$ curve measured after cooling the sample to 2 K under H = 1 T.

cooling, we obtain $H_E \sim -0.45$ and -0.29 T for $H \parallel ab$ and $H \parallel c$, respectively. The presence of the EB below T_{SG} demonstrates the existence of disparate magnetic orders [28,29].

The SG ground state was observed in the $0.36 \le x \le 0.66$ region of the phase diagram (see Figs. 4 and 5). The emergence of SG in Sr₃(Ru_{1-x}Mn_x)₂O₇ is reminiscent of the cluster-SG state in (Sr_{1-x}Ca_x)₃Ru₂O₇ for 0.08 < x < 0.4 [16]. This is attributed to the variation of the electronic structure caused by the structural distortion, resulting in the enhancement of FM correlation over the AFM fluctuations [16]. In the case of Mn-doped Sr₃(Ru_{1-x}Mn_x)₂O₇, the local structure is also altered by smaller Mn ions [5,30], namely, reduced octahedral distortion [1,5,31]. The puzzling fact is that T_{SG} reaches the minimum when Ru/Mn ~ 1 (i.e., $x \sim 0.5$), implying the SG state is associated with the interplay between Ru and Mn moments.

To understand the origin of the SG state, we examine the evolution of the magnetic properties with *x*. Figure 6(a) shows the *x* dependence of the magnetic susceptibility anisotropy (χ_{ab}/χ_c) at $T = 300 \text{ K} (> T_{\text{SG}})$. Though $\chi_{ab} > \chi_c$ for x = 0.36, the anisotropy nearly vanishes for x = 0.55with $\chi_{ab}/\chi_c \sim 1$. Increasing *x* further restores the magnetic anisotropy, reaching the maximum for x = 1.0 ($\chi_{ab}/\chi_c =$ 1.8). Figure 6(b) shows the variation of the effective magnetic moment (μ_{eff}) obtained from Curie-Weiss fitting of the magnetic susceptibility data (see Figs. 4 and 5) with *x*, where a similar behavior is observed: μ_{eff} is approximately the same obtained from both χ_{ab} and χ_c and reaches the minimum for x = 0.55. It should also be noted that, while there is a slight directional variation in μ_{eff} in x = 0.36 and 0.66, it is approximately isotropic for x = 0.55. The isotropic behavior of μ_{eff} with the lowest T_{SG} for $x \sim 0.5$ signals the importance of intermixing of Ru and Mn.

The random occupation of Ru and Mn can lead to complex magnetic interactions. A preliminary understanding can be achieved from the Curie-Weiss temperature θ_{CW} . As shown in Fig. 6(c), both θ_{CW}^{ab} and θ_{CW}^c are negative for x = 0.36, indicating AFM interaction in both directions. Upon increasing x, both θ_{CW}^{ab} and θ_{CW}^c gradually change toward positive values. The apparent sign change of θ_{CW}^{ab} indicates that there are both AFM and FM interactions with dominant FM interaction at $x \sim 0.66$. The mixed AFM and FM interactions are crucial in the establishment of the SG state. Therefore, it is important to determine the dopant distribution.

Figure 7(a) shows the high-angle annular dark-field (HAADF) STEM image along the [100] zone axis for x = 0.16. As the intensity of the imaged atoms is proportional to the atomic number, the bright spots should correspond to Mn, Ru, and Sr. Indeed, we find that the atomic arrangement is consistent with that shown in Fig. 1(a). In the projected structure shown in Fig. 7(a), white dots represent Sr, and green dots denote Ru/Mn. To detect the Ru and Mn distribution,



FIG. 5. Magnetic properties of $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with x = 0.66. (a) Temperature dependence of χ with H = 1 T parallel to the *ab* and *c* directions. Inset: Temperature dependence of inverse susceptibility shifted vertically for clarity and Curie-Weiss fitting (solid lines). (b) Field dependence of the isothermal magnetization at various temperatures with $H \parallel ab$ (main panel) and $H \parallel c$ (left inset). (c) $\chi_{ab}(T)$ curves with various fields. (d) T_f as a function of $H^{2/3}$ for $H \parallel ab$ and $H \parallel c$ with fits represented by solid lines. (e) Magnetic relaxation at 15 K for time durations t_1 and t_3 with a temporary cooling to 10 K for time duration t_2 . Inset: Data at 15 K during t_1 and t_3 showing the magnetic memory effect. (f) ZFC $M_{ab}(H)$ curve at 2 K along with FC $M_{ab}(H)$ curve measured after cooling the sample to 2 K under H = 7 T.

EDS measurement was carried out to map Ru and Mn along the [100] zone axis for x = 0.16. Figure 7(b) shows the distribution of Ru (orange) and Mn (green), revealing randomness of Mn occupation in the Ru site (also see Fig. 8).

Due to a random distribution of Mn, competing magnetic interactions between Ru and Mn moments lead to a sharp decrease in T_N above x = 0.16. Further increase of x apparently introduces stronger spin disorder. Figure 7(c) illustrates such a trend with competing FM and AFM interactions. At the maximum mixing (i.e., Ru : Mn = 1:1), the strongest FM and AFM competition results in the SG state with the low-

est T_{SG} [27]. As illustrated in Fig. 7(d), the local magnetic landscape is comprised of segregated magnetic regions with competing short-range ordering that prevents the formation of a long-range magnetic order. This resembles the electronic phase separation near T_{MIT} in Sr₃(Ru_{1-x}Mn_x)₂O₇ (x < 0.1) as well as Sr₃(Ir_{1-x}Ru_x)₂O₇ [5,19]. While the MIT occurs at temperatures much higher than T_{SG} for $x \ge 0.36$, the presence of a sizable EB effect observed at $T < T_{SG}$ substantiates the magnetic phase segregation scenario.

As mentioned above, the magnetic behavior of pure $Sr_3Mn_2O_7$ (x = 1) is different from that in the region of



FIG. 6. Doping dependence of (a) χ_{ab}/χ_c at T = 300 K with H = 1 T, (b) effective magnetic moment per formula unit, and (c) Curie-Weiss temperature. The dashed line in (c) delineates the zero value of θ_{CW} .



FIG. 7. (a) High-angle annular dark-field (HAADF) scanning transmission electron microscopy (STEM) image and (b) energy dispersive x-ray spectroscopy (EDS) elemental mapping of Ru and Mn in $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with x = 0.16 taken along the [100] zone axis. Simulated crystal structure is overlaid in (a). (c) Visualization of the local spin randomness within the (Ru/Mn)O₆ plane. Arrows represent magnetic moments of Mn (blue) and Ru (red). The short-range correlation between the Mn and Ru spins creates spin clusters, represented by the shaded region, giving rise to the spin-glass (SG) state. (d) Depiction of the evolution of competing magnetic regions (blue and white areas) in (Ru/Mn)O₆ planes upon increasing *x*.

 $0.36 \le x \le 0.66$. Figure 9(a) shows the temperature dependence of χ_{ab} and χ_c for Sr₃Mn₂O₇ between 2 and 800 K. Note that both χ_{ab} and χ_c show prominent kinks at ~160 K. While the temperature at which the anomaly occurs agrees

with the reported AFM transition at $T_N = 160 \text{ K}$ [20], its nature is atypical: there is a broad hump centered ~ $2T_N$. This indicates the occurrence of strong spin fluctuations at *T* much greater than T_N . Likewise, the inverse susceptibility does not



FIG. 8. (a) High-angle annular dark-field (HAADF) scanning transmission electron microscopy (STEM) image and electron energy loss spectroscopy (EELS) elemental mapping of (b) Ru and (c) Mn in $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with x = 0.16 taken along the [100] zone axis. Simulated crystal structure is overlaid in (a).



FIG. 9. (a) Temperature dependence of magnetic susceptibility in $Sr_3Mn_2O_7$ measured under H = 1 T along the *ab* and *c* directions. (b) Relationship between oxygen deficiency δ and lattice parameter *c* in $Sr_3Mn_2O_{7-\delta}$. The solid points are from Refs. [20,33,34], the solid line represents a linear fit, and the open circles are from this paper. (c) Field dependences of isothermal magnetization with $H \parallel ab$ (main panel) and $H \parallel c$ (inset) measured at indicated temperatures. (d) Temperature dependence of specific heat C_p . Inset: Low-temperature behavior plotted as C_p/T vs T^2 . Solid line represents linear fit.

conform to the Curie-Weiss law until \sim 500 K. The inset of Fig. 9(a) demonstrates the excellent fit of the experimental data by the Curie-Weiss law with $\mu_{\rm eff} \sim 7.56 \,\mu_{\rm B}/{\rm f.u.}$ and $\theta_{\rm CW} = -1413$ K, which are also plotted in Figs. 6(b) and 6(c). Compared with T_N , θ_{CW} is much larger. This may be related to the layered structure, as discussed in Ref. [32]. The obtained $\mu_{\rm eff}$ is also higher than that for Mn⁴⁺ with maximum $S_{\text{Mn4+}} = \frac{3}{2}$ [20,32]. Our μ_{eff} requires that the total spin $S_{Mn} \sim 2$, which corresponds to Mn^{3+} . If this is the case, oxygen deficiency δ is expected with chemical composition Sr₃Mn₂O_{7- δ}. Figure 9(b) shows the δ dependence of the lattice parameter c in Sr₃Mn₂O_{7- δ} obtained from the reports in Refs. [20,33,34], where a linear trend can be observed. The lattice parameter c of both as-grown crystal [c = 20.15(2)Å] and O₂ annealed crystal [c = 20.08(2)Å] used in this paper fit well with this relationship. From the linear fit, we estimate our as-grown and O2 annealed crystals to have the oxygen deficiency $\delta_{as-grown} \sim 0.64$ and $\delta_{O2-annealed} \sim 0.06$, respectively. The prolonged annealing in O₂ pushes the crystal close to stoichiometry, resulting in the same AFM transition temperature as that for $Sr_3Mn_2O_7$ [14]. Based on this, the high effective moment cannot be attributed to Mn^{3+} but rather the larger g factor (>2).

At present, the magnetic properties of $Sr_3Mn_2O_{7-\delta}$ remain unsettled. According to Ref. [20], Sr₃Mn₂O₇ forms a G-type AFM ordering <160 K based on neutron scattering investigation on the polycrystalline sample. In our case, while χ_c deceases, χ_{ab} is enhanced below $T_N = 160 \text{ K}$ [see Fig. 9(a)]. This indicates the prominent FM interaction in the *ab* plane, thus pointing toward a canted spin configuration where the spins are coupled AFM in the c direction but FM in the *ab* plane. This assertion is supported by the field dependence of M shown in Fig. 9(c), where a nonlinear behavior is developed at low temperatures for M_{ab} while M_c depends linearly on H (see the inset of Fig. 9(c)). Such information may be difficult to probe in polycrystalline samples, which is the averaged result in all directions. On the other hand, a canted AFM ordering ($< \sim 250 \,\text{K}$) has been observed for $Sr_3Mn_2O_6$ [34]. We find that the as-grown crystal (Sr₃Mn₂O_{6.36}) does not show any sign of magnetic ordering down to 2 K (see Fig. 10), agreeing with that for $Sr_3Mn_2O_{6.55}$ $(\delta = 0.45)$ [20].

A typical magnetic transition would manifest its signatures in thermodynamic properties as excess entropy is released when spins form a long-range order. To gain further insight into the nature of the transition at T_N in our nearly



FIG. 10. Temperature dependence of the magnetic susceptibility in as-grown $Sr_3Mn_2O_{7-\delta}$ single crystal measured under H = 1 T.

stoichiometric $Sr_3Mn_2O_7$ crystals, we measure its specific heat C_p . Figure 9(d) shows the temperature dependence of C_p for Sr₃Mn₂O_{6.94} under the zero field, demonstrating a smooth variation with temperature without noticeable anomaly near T_N . This suggests that most of the magnetic entropy is released at higher temperatures before $T_{\rm N}$, in agreement with the presence of broad humps in the magnetic susceptibility at a temperature well above T_N . In an AFM system, $C_p =$ $C_{\text{phonon}} + C_{\text{AFM}}$, where $C_{\text{phonon}} = \beta T^3$ is the phonon specific heat, and C_{AFM} varies exponentially with T, which is related to the spin gap [35]. The inset of Fig. 9(d) shows C_p/T as a function of T^2 , with linear behavior $< \sim 12$ K. This implies that $C_{\rm AFM}$ is negligible at low temperatures. While the slope corresponds to $\beta = 0.127 \text{ mJ/mol-} \text{K}^4$, we also observe a nonzero constant ($\sim 5 \text{ mJ/mol-}\text{K}^2$), indicating the presence of a linear term in the specific heat. Given an insulating system, a linear contribution should not be attributed to the electronic specific heat. While several scenarios are proposed [36], most relevant to our system is the existence of a magnetic amorphous phase, as suggested for another manganite [37], and/or contributions from oxygen deficiency [38].

IV. CONCLUSIONS

In summary, through the characterization of high-quality single crystals of $Sr_3(Ru_{1-x}Mn_x)_2O_7$ with $0.36 \le x \le 1.0$, we have demonstrated the intricate FM and AFM interplay between the spins and untangled the peculiar evolution of the magnetic ground state in the series. The intermixing of Mn and Ru leads to spin disorder due to the complexities of the magnetic interactions. Specifically, the SG ground state is observed in the region of $0.36 \le x \le 0.66$. At $x \sim 0.5$ (i.e., Ru : Mn = 1:1), the system exhibits the lowest SG transition temperature, implying the strongest competition between AFM and FM interactions. The Ru and Mn intermixing provides the necessary components for the local magnetic phase segregation leading to the SG with a large EB. As x approaches 1, the in-plane FM and c-direction AFM become less combative but more cooperative. The system thus forms a long-range AFM order with antiparallel spins along the c direction but FM in the *ab* plane, i.e., canted AFM. Our experimental observations also demonstrate that the tuning magnetic interactions and imposing randomness is a viable path to generate exotic spin states.

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APPENDIX: PROCEDURES FOR MEMORY EFFECT EXPERIMENTS

For the ZFC measurement, the sample was cooled from $T \gg T_{SG}$ under H = 0. After reaching $T_0 = 10$ K, H = 0.1 T was applied, and M was measured as a function of time t for a time duration $t_1 = 2$ h. After t_1 , the sample was cooled rapidly to 5 K, and M was measured as a function of t for another time duration $t_2 = 2$ h. After t_2 had elapsed, the sample was warmed back up to $T_0 = 10$ K, and M was measured for time duration $t_3 = 2$ h.

For the FC measurement, the sample was cooled from $T \gg T_{\rm SG}$ under $H = 0.1 \,\mathrm{T}$ to the base temperature of 2 K, and M was measured as a function of T. During the cooling cycle, temporary stops were made at temperatures $T_{\text{stop}} = 30$, 25, 20, 15, and 10 K. During each stop, the magnetic field was removed, and the sample was allowed to "age" by keeping it at that temperature for $t_{stop} = 2$ h. After t_{stop} had elapsed, the magnetic field was turned back on, and cooling was resumed. This temporary stop-and-aging process was performed at each T_{stop} until the base temperature of 2 K was reached. The thusobtained M(T) curve is termed the "stop" curve in this paper and Fig. 3(e). After reaching 2 K, the sample was warmed up to $T \gg T_{SG}$, and M(T) was measured while the field was still on continuously without any stops. The data obtained in the warming cycle demonstrate the memory effect and are termed "mem." in this paper and Fig. 3(e).

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