

Bimetal single-molecule magnets supported on benzene with large magnetic anisotropy and unquenched orbital moment

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Single-atom magnets, the ultimate limit of high-density magnetic storage, have attracted widespread attention over recent decades. However, they are far from being applicable as a storage medium because of their low magnetic stability. In this paper, we describe a strategy to induce huge magnetic anisotropy in “bimetal magnets” on a benzene (Bz) substrate based on the electron filling of d orbitals. Our first-principles calculations reveal that OsX-Bz ($X = \text{Fe, Ru, and Os}$) exhibits high structural stability, large unquenched orbital moments ($1.55\text{--}1.59 \mu_B$), and huge perpendicular magnetic anisotropy energy (MAE) above 54.8 meV. The synergistic effect of two transition metal components toward this huge MAE and preserved orbital moment is discussed in depth, which is mainly attributed to the d - d coupling induced energy level rearrangement. Our work may provide insights into the underlying physical mechanism and bimetal molecule magnet design.

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I. INTRODUCTION

The continuous downscale of magnetic storage devices would finally reach the ultimate limit scale—one atom stores one bit of information. Hence, an urgent task is to design the new magnetic materials with high stability and long lifetime of the atom’s magnetization. One of the critical parameters characterizing the magnetic stability is the magnetic anisotropy energy (MAE). MAE originates from the interaction of spin-orbit coupling (SOC) and ligand field (LF). In the past years, a major breakthrough on magnetic materials is the discovery of single-molecule magnets with large MAE, as a result of reduced dimensionality and symmetry. Many multimetallic and bimetallic single-molecule magnets, such as small metal clusters and dimers, were found to exhibit good performance in terms of maintaining a stable magnetic orientation, enabling a MAE value of up to several hundred meV [1–4]. The magnetic centers of single-molecule magnets are generally $3d$ and $5d$ transition metal (TM) atoms. Their advantages are the large spin moment in $3d$ metal atoms and the strong SOC in the heavy $5d$ atoms, respectively. As the smallest chemical species, transition metal dimers possess a significant energy barrier against the reorientation of their magnetization between the low-energy direction (dimer axis) and the high-energy direction (perpendicular to the dimer axis). Błoński and Hafner reported that the MAE of

the gas-phase homonuclear Ir_2 dimer and the heteronuclear Ir-Co dimer can reach 70 meV with large anisotropy of orbital moments of about $1.02 \mu_B$ and $1.2 \mu_B$ [5,6]. For the Ir_2 dimer, Liang *et al.* found its MAE can be further enhanced up to 294 meV by F functionalization which drives orbital rearrangement [7]. Moreover, the formation of chemical bonds would quench orbital moment in multimetallic single-molecule magnets. For dimers, Fritsch *et al.* discovered that free Fe_2 and Co_2 dimers possess MAE above 30 meV and large unquenched orbital moment anisotropy (OMA) of about $1.7 \mu_B$ and $1.67 \mu_B$, respectively [8]; Chen *et al.* found that the Pt_2 dimer has a large OMA of about $2 \mu_B$ [9]. Niemeyer *et al.* demonstrated that the orbital moments are strongly quenched even for clusters of only three atoms [10].

Tuning the local environment of molecular magnets to produce an axial ligand field can have an important consequence on preserving the orbital moment as well as attaining the large MAE. Two-dimensional (2D) materials can be naturally utilized as supporting substrates to provide the appropriate ligand field. A variety of two-dimensional materials have been examined for a long time, such as Au [11], Pt [12], black phosphorus [13], graphyne [14], and graphene [15]. For example, the key magnetic parameters of transition dimers Os-Ru and Co_2 adsorbing on pristine, single-vacancy (SV) or nitrogenized divacancy (NDV) graphene have been reported [15–17]. For the Os-Ru dimer absorbed on pristine (NDV) graphene, the orbital moments of the top Os atom have pronounced changes, which is from $1.44 \mu_B$ ($1.29 \mu_B$) to $-0.02 \mu_B$ ($0.21 \mu_B$). The corresponding MAEs are found to be 176 meV [15] and 54 meV [17], respectively. Co_2 dimer absorption on SV graphene also preserves the OMA and MAE. The MAE is about 53 meV and the orbital moments of the top Co atom along the (001) and (100) axis are about $2.1 \mu_B$ and $0.3 \mu_B$ [16]. To achieve single-molecule magnets

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with high magnetic stability, it is meaningful to pursue the optimal compositions of the transition metal dimers as well as the substrates.

In this study, the magnetic stability of a series of transition metal dimers adsorbed on benzene is explored using first-principles calculations. Benzene (Bz) is the simplest and smallest organic unit with D_{6h} symmetry. Various transition metal dimers (M_1M_2) have been considered by varying the combinations and configurations. After high-throughput screening of many M_1M_2 compositions, the $5d$ element Os combined with same-group of transition metal atoms, labeled as M_1M_2 -Bz ($M_1 = \text{Os}, M_2 = \text{Fe, Ru, and Os}$), are picked up. They have strong SOC effect, partially occupied d orbitals, large atomic magnetic moment, large anisotropic spin and orbital moments, and huge MAE. Both MAE and unquenched orbital moments are along the uniaxial direction. The synergistic effect of transition metal dimers and Bz substrates for regulating magnetic properties have also been further explained by either p - d hybridization between the lower M_2 metal with the Bz or d - d hybridization between M_1 and M_2 atoms. These results provide a feasible strategy to stabilize the magnetization in single-molecule magnets by achieving large MAE and unquenched orbital moment simultaneously.

II. METHODS

Using spin-polarized density functional theory (DFT) implemented in the VASP packages [18,19], first-principles calculations were performed to investigate the stability, electronic structures, and magnetic properties of M_1M_2 -Bz systems ($M = \text{Os, Ru, Fe, Ir, Rh, and Co}$). Based on the Perdew-Burke-Ernzerhof (PBE) parametrization [20], the generalized gradient approximation (GGA) was used to describe the exchange and correlation effects. The projected augmented wave (PAW) [21] method was employed with an energy cutoff of 500 eV to describe the electronic wave functions. The M_1M_2 -Bz systems were placed in a large cubic box, ensuring a distance of no less than 20 Å between neighboring periodic images to avoid any unphysical interaction. Geometry relaxation was performed until the force on each atom was less than 0.01 eV/Å and the total energy converged to 1×10^{-7} eV. The Γ point was used for the large cubic supercell of individual M_1M_2 -Bz.

The binding energy between M_1M_2 and Bz can be defined as

$$E_b = E_{M_1M_2\text{-Bz}} - E_{M_1M_2} - E_{\text{Bz}}, \quad (1)$$

where $E_{M_1M_2\text{-Bz}}$, $E_{M_1M_2}$, and E_{Bz} denote the total energies of the M_1M_2 -Bz, freestanding M_1M_2 dimer, and Bz molecule, respectively.

Based on the equilibrium configurations, the MAE was determined from the self-consistent calculation considering both SOC and magnetic noncollinearity, as follows:

$$\text{MAE} = E(\parallel) - E(\perp), \quad (2)$$

where $E(\parallel)$ denotes the total energies for the magnetization direction along the Bz plane, while $E(\perp)$ represents the total energies for the magnetization direction parallel ($+z$) or antiparallel ($-z$) to the TM dimer. In our systems, we find that the $+z$ and $-z$ directions are identical and give the same

results. For perpendicular magnetic storage device applications, a positive MAE corresponding to the out-of-plane easy magnetization direction is more valuable [22–24].

Moreover, the torque method developed by Wang *et al.* [25] was employed to conduct a thorough and quantitative analysis of the origin of the MAE of M_1M_2 -Bz complexes. In the framework of the torque method [25], the MAE is expressed as

$$E_{\text{MAE}} = \sum_{i \in \text{OCC}} \langle \psi_i | \frac{\partial H_{\text{SO}}}{\partial \theta} | \psi_i \rangle_{\theta=45^\circ}. \quad (3)$$

Here, ψ_i is the i th relativistic eigenvector, and H_{SO} is the SOC Hamiltonian. In this approach, the MAE is independent of the absolute accuracy of two total energy calculations. For the convenience of analysis, MAE was divided into contributions from three types of coupling, namely the majority spin states (uu), the minority spin states (dd), and also the cross-spin coupling ($ud + du$).

Based on the second order perturbation approach proposed by Wang, Wu, and Freeman [26], the MAE contributed by each atom can be approximately determined by matrix elements of the angular momentum operators L_z and L_x , across the unoccupied (u) and occupied (o) states:

$$\text{MAE} \approx \xi_n^2 \sum_{o,u,\alpha,\beta} (2\delta_{\alpha\beta} - 1) \times \left(\frac{\langle o, \alpha | L_z | u, \beta \rangle^2}{E_{u,\alpha} - E_{o,\beta}} - \frac{\langle o, \alpha | L_x | u, \beta \rangle^2}{E_{u,\alpha} - E_{o,\beta}} \right). \quad (4)$$

Here, the SOC constant ξ_n is an element dependent property where n is the effective principal quantum number of the d electron, which is the radial integral of the product of the SOC amplitude and the radial wave functions of the states o and u . We have given a different matrix for each kind of element (including Os, Ru, Fe, C, and H) involving the M_1M_2 -Bz molecules. α and β denote different spin channels. When α and β are in the same spin channel, $\delta_{\alpha\beta} = 1$; otherwise, $\delta_{\alpha\beta} = 0$. We define

$$P = \langle o, \alpha | L_z | u, \beta \rangle^2 - \langle o, \alpha | L_x | u, \beta \rangle^2. \quad (5)$$

Each M_1M_2 -Bz molecule involves $3d$, $4d$, and $5d$ TM atoms. Hence, the strong correlation effect of the localized d orbitals must be treated carefully. To examine the electron-electron correlation effect on MAE, we performed test calculations using the GGA + U method [27,28]. In these TM based systems, a strong correlation among the localized d orbitals is necessary. Thus, the Hubbard U term was tested in the GGA + U formulation of first-principles calculations [29,30]. An effective interaction parameter U_{eff} is used to represent the value of $U - J$ in the subsequent discussion. For Os, Ru, and Fe, the value of U is typically about 1.5 eV. In our work, three values of U_{eff} (0.5, 1.5, and 2.5) were tested, and the results show that magnetic and orbital moment anisotropies do not vanish with the implementation of the U_{eff} term, and increase slightly (see Table S1 of the Supplemental Material [31]).

TABLE I. Equilibrium bond lengths of Os-X and X-Bz, binding energy (E_b), spin magnetic moments on TM atom, and MAE of OsX-Bz ($X = \text{Fe, Ru, and Os}$).

Structure (OsX-Bz)	Bond length (Os-X) (Å)	Bond length (X-C) (Å)	E_b (eV)	M_{Os} (μ_B)	M_X (μ_B)	MAE (meV)
OsFe-Bz	2.06	2.08	-1.81	1.519	0.252	67.2
OsRu-Bz	2.18	2.23	-2.56	1.565	0.168	58.5
Os ₂ -Bz	2.21	2.24	-2.84	1.492	0.173	54.83

III. RESULTS AND DISCUSSION

As stated above, we use benzene as the substrate, and the two transition metal atoms are chosen from all the possible $3d$ - $5d$ transition metals. After full relaxation, the magnetic moments and MAEs of a series of transition metal dimers adsorbed on benzene were determined by noncollinear DFT calculations. As listed in Table S2, the systems of Fe, Co, and Ni groups have large MAE. Especially, Fe and Co groups exhibit giant perpendicular MAE. For example, OsFe-Bz, OsRu-Bz, Os₂-Bz, IrCo-Bz, and IrRh-Bz are about 67.20, 58.5, 54.83, 81.53, and 72.35 meV, respectively. Previously, IrCo-Bz has been studied by Xiao *et al.* and its MAE was about 248 meV calculated by an all-electron full-potential local-orbital scheme [32]. M_1M_2 -Bz molecules by other groups have either a very small MAE or even no magnetism. For Ti and Mn groups, the HfTi-Bz and ReMn-Bz have small MAE of about 2.5 and -5.94 meV with fully quenched orbital angular momentum. V and Cr groups all behave as nonmagnetic. Hence, we will mainly focus on the Fe group to discuss the synergistic effect between two TM atoms in producing the huge MAE in the following contents.

The atomic structures, electronic structures, and magnetic properties of Os-Fe, Os-Ru, and Os₂ dimers adsorbed on Bz are systematically investigated and the related parameters are listed in Table I. First, three possible high-symmetry structures are built. As a representative, Fig. 1 illustrates the structural model and checks the relative energies of configurations I (ΔE_1) and II (ΔE_2) of OsFe-Bz. For Os-Ru dimer and Os₂ dimers, ΔE_1 are 1.08 and 1.38 eV, and ΔE_2 are 0.94 eV and 1.26 eV, respectively. These relative energy calculations clearly prove that all the Fe group dimers prefer an upright position above the center of Bz with C_{6v} symmetry. Besides, we consider two possible situations in configuration III: Os near Bz and away from Bz. The energy calculations show that Os atom prefers to stay away from the Bz in configuration III. Specifically, we will use OsX-Bz ($X = \text{Fe, Ru, and Os}$) to denote the OsX dimer adsorbed on Bz upright, where X is the bottom atom nearest to the benzene. We can see that, naturally, the bond lengths of Os-X and X-C increase gradually from OsFe-Bz to Os₂-Bz because the reducibility to release electrons of Fe is stronger than that of Ru and Os according to the activity series of metals [33]. The binding energies of OsX-Bz systems defined by Eq. (1) are in the range -1.81 to -2.84 eV (Table I), implying the exothermic formation of these dimer-benzene complexes.

To gain deep insight into the interactions of Os-X and X-Bz, the partial density of states (PDOS) of OsFe-Bz, OsRu-Bz, and Os₂-Bz are calculated and displayed in Fig. 2. Clearly, in

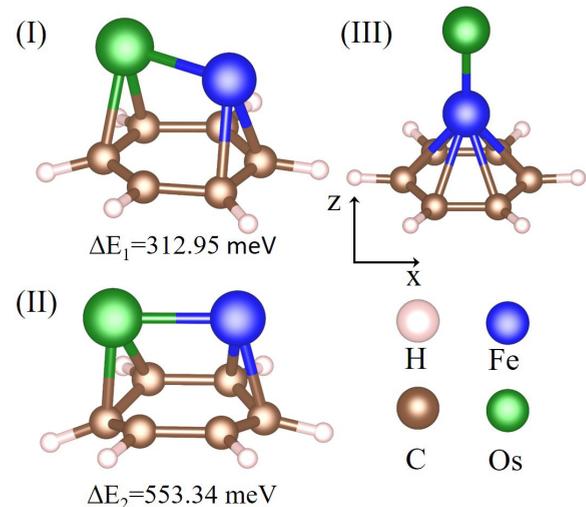


FIG. 1. I, II, III are three possible high-symmetry structures for OsX-Bz complexes. Here, ΔE_1 and ΔE_2 are the energy difference of (I) - (III) and (II) - (III) of OsFe-Bz, respectively. The positive value represents that structure III possesses lowest energy which can be as the ground state structure.

the OsFe-Bz system, the interaction between two TM atoms includes two types of strong hybridization: $5d_{x^2-y^2}-3d_{x^2-y^2}$, $5d_{xz/yz}-3d_{xz/yz}$, and $5d_{z^2}-3d_{z^2}$. The corresponding bonding and antibonding states can be denoted as π/π^* , α/α^* , and σ/σ^* , respectively. The π/π^* and σ/σ^* are fully occupied in the spin-up channels, whereas both spin-up and spin-down channels of α/α^* and the spin-down channels of π/π^* and σ/σ^* are partially occupied. In OsRu-Bz and Os₂-Bz, only σ/σ^* and α/α^* states exist and α/α^* are very weak. Meanwhile, the spin-up and spin-down channels of σ/σ^* states are fully and partially occupied, respectively. Regarding the X-C bonds ($X = \text{Fe, Ru, and Os}$), only p - d hybridization exists at a deep energy level, i.e., $d_{xz/yz}-p_z$. Furthermore, Mulliken population analysis in the CASTEP package [34] reveals that the Fe (Ru,

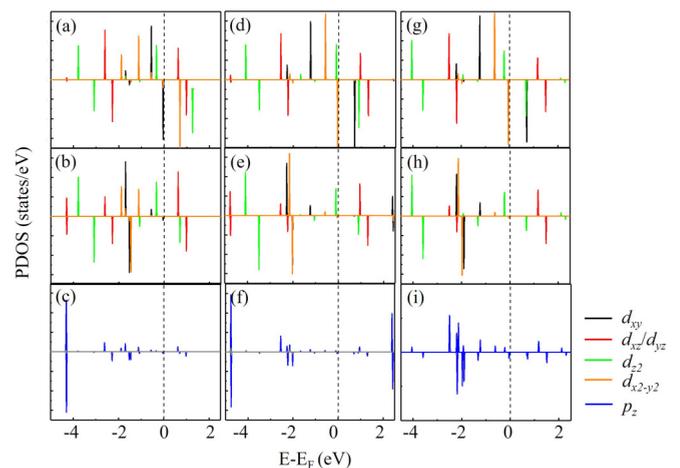


FIG. 2. PDOS of TM atoms of OsFe-Bz and OsRu-Bz. (a)-(c) PDOS of Os, Fe, and Bz for OsFe-Bz. (d)-(f) PDOS of Os, Ru, and Bz for OsRu-Bz. (g)-(i) PDOS of Os upper, Os bottom, and Bz for Os₂-Bz. The Fermi level (E_F) is set to zero.

Os) atom shares nearly 0.24 (0.09, 0.17) electrons with the top Os atom and gives 1.63 (0.92, 0.91) electrons to Bz. Totally, the Fe atom contributes nearly two electrons ($1.87e^-$) to the surrounding atoms, exhibiting a Fe^{2+} chemical state in this molecule. We can see that Fe shares more electrons with the top Os atom compared with Ru and Os because Fe has smaller electronegativity (1.83) than Ru (2.2) and Os (2.2), as given by the Pauling scale. Additionally, bond population analyzed by CASTEP [34] shows that the bond order of the Os-Fe bond (0.77) is stronger than that of Os-Ru (0.46) and Os-Os (0.63). Such strong bonding interactions make a major contribution to the structural stability of OsX-Bz complexes.

Our spin-polarized DFT calculations reveal the coexistence of large spin and orbital magnetic moments in OsX-Bz systems. First, the top 5d transition metal Os atom provides the majority of spin magnetic moments as listed in Table I, which are about $1.52 \mu_B$, $1.57 \mu_B$, and $1.49 \mu_B$ for OsFe-Bz, OsRu-Bz, and Os₂-Bz, respectively. This can be more clearly visualized from the spin charge density in Fig. S1. From the PDOS of TM atoms in Fig. 2, we can understand the origin of the magnetic moment in those systems. For OsFe-Bz complexes, the magnetic moment of Os mainly comes from the partially occupied d_{z^2} and $d_{x^2-y^2}$ orbitals (about $0.5 \mu_B$ and $1 \mu_B$, respectively). The Fe atom only contributes about $0.25 \mu_B$ through d_{z^2} orbitals. Similarly, in Figs. 2(d) and 2(g), the d_{xy} orbital of the top Os atom provides about $1 \mu_B$, and d_{z^2} contributes $0.51 \mu_B$ and $0.44 \mu_B$ for OsRu-Bz and Os₂-Bz, respectively. Here, the Ru atom in OsRu-Bz and the bottom Os atom in Os₂-Bz only provide about $0.17 \mu_B$ through the d_{z^2} orbital. Second, the orbital angular momenta of OsX-Bz are largely preserved, as shown in Table S3. The orbital moments reach up to $1.55 \mu_B$ and $1.59 \mu_B$ along the dimer axis while the in-plane direction is only about $0.15 \mu_B$ and $0.2 \mu_B$, respectively. Besides, we observed the difference of spin moment between the out-of-plane and in-plane directions in a few systems. The anisotropy of spin moment is originating from the SOC induced reordering of the eigenstates close to the Fermi level, which have been explained in detail in the previous work. For our systems, it is larger in OsX-Bz than in the Fe and Ru based systems. Meanwhile, the larger spin moment is found along the out-of-plane (easy-axis) orientation. One can see that the large spin magnetic moment of the top 5d TM atom and the unquenched orbital moment on the TM atoms are protected very well. Thus, this low coordination environment provided by the bottom TM really plays a positive role in preserving the free-atom-like properties of the 5d TM element, as proved by our DFT calculations. A similar large magnetic moment was also observed for the Os containing molecules [35,36].

Considering SOC together with magnetic noncollinearity, the MAE was evaluated by both self-consistent calculations and the torque method. The OsX-Bz ($X = Fe, Ru, \text{ and } Os$) exhibit large MAE values of about 67.2 (59.92), 58.5 (55.54), and 54.83 meV (52.9 meV), respectively, whereas the values from torque model calculations are given in brackets. Obviously, the PAW method used in our study has also been supported by the torque method. The two methods obtain the same results. With the help of the torque method, we decompose the MAE into three coupling terms with respect to the Fermi level (Table II). As seen in Fig. 3, the coupling

TABLE II. Contributions of different spin channels, including majority spin states (uu), minority spin states (dd), and cross spin states ($ud + du$); total MAE of OsX-Bz ($X = Fe, Ru$) given by torque calculations.

Structure	uu (meV)	dd (meV)	$ud + du$ (meV)	Total (meV)
OsFe-Bz	-48.18	99.30	8.79	59.92
OsRu-Bz	-41.98	99.98	-2.46	55.54
Os ₂ -Bz	-33.29	80.82	5.35	52.90

between occupied and unoccupied spin-down d states (dd term) provides the major contribution to the positive MAE compared with other couplings near the Fermi level in those three systems. Combined with the PDOS analysis (Fig. 2), it can be concluded that the large positive MAE of OsX-Bz mainly comes from the dd term, which is composed of the spin-down orbitals of occupied $5d_{xy}/d_{x^2-y^2}$ and unoccupied $5d_{x^2-y^2}/d_{xy}$. Those orbitals have a small energy difference of about 0.71 eV and are coupled by a large coupling coefficient $P = 4$, which is obtain by Eq. (5). As stated by Edström [37], a large MAE might appear if there are many occupied and unoccupied states with energies very close to the Fermi energy. Correspondingly, it is demonstrated directly in Eq. (4) that a smaller energy difference between occupied and unoccupied states in the denominator is generally more beneficial for obtaining a large MAE. Besides, the same picture can be also applied to interpret the small MAE of other groups such as Ti and Mn groups. As shown in Fig. S2, Near the Fermi level, the small MAE values of HfTi-Bz (2.5 meV) and ReMn-Bz (-5.94 meV) are only contributed by the partial occupied d orbitals (d_{xz}/d_{yz} and d_{xy}/d_{xz}) with the coupling coefficient $P = 1$ and -1 , respectively.

To further clarify the composition effect, the difference of MAE between OsFe-Bz and OsRu-Bz is discussed. As stated above, the energy difference between occupied and unoccupied states near the Fermi level also plays a key role in determining the strength of magnetic anisotropy. All three coupling terms are given in Table II. Among them, we find that the $ud + du$ term is the key factor. The $ud + du$ term of OsFe-Bz is 8.79 meV, notably larger than that of OsRu-Bz of -2.46 meV. The orbital energy levels and the coupling coefficients of $ud + du$ terms are listed in Table III. It can be clearly seen that the smaller ΔE between occupied and unoccupied states of Os atom in OsFe-Bz accounts for the greater SOC and MAE with regard to

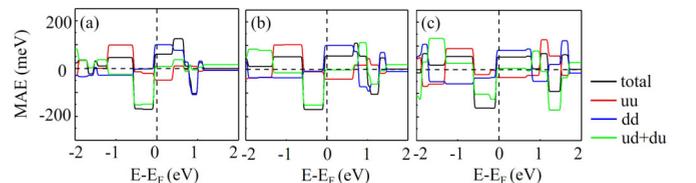


FIG. 3. Fermi level dependent total and decomposed MAEs of OsFe-Bz (a), OsRu-Bz (b), and Os₂-Bz (c). Here, majority (red line), minority (blue line), cross (green line) represent the coupling between spin-up channels, spin-down channels, and spin-up with spin-down channels, respectively. The Fermi level (E_F) is set to zero.

TABLE III. Orbital energy levels, energy level difference (ΔE), and coupling coefficient (P) of $\text{Os}X\text{-Bz}$ ($X = \text{Fe}, \text{Ru}$); Fermi level (E_F) is set to zero.

o/u	OsFe-Bz			OsRu-Bz			P
	o^+	u^-	ΔE	o^+	u^-	ΔE	
d_{z^2}/d_{yz}	-0.34	0.98	1.32	-0.10	1.30	1.4	3
d_{xy}/d_{xz}	-0.56	0.98	1.54	-1.24	1.30	2.54	1
$d_{x^2-y^2}/d_{yz}$	-0.58	0.98	1.56	-0.58	1.30	1.88	1
$d_{xy}/d_{x^2-y^2}$	-0.56	0.70	1.26	-0.58	0.70	1.28	-4

OsRu-Bz. Such an energy level difference is caused by the coordination environment, which is determined by the bottom transition metal atom X . All the results suggest that a lighter base atom may modulate the electron structure of the top metal atom more effectively. Coincidentally, this phenomenon has also been observed in systems with group 9 elements: IrCo-Bz and IrRh-Bz. The MAE of IrCo-Bz is ~ 81.53 meV, somewhat larger than the 72.35 meV of IrRh-Bz.

To illuminate the role of $3d$, $4d$, $5d$ transition metal atoms and Bz, Table S3 presents the M_1 - M_2 bond lengths, spin and orbital magnetic moments on TM atoms, and MAE given by noncollinearity self-consistent (NSC) method and torque method of free homonuclear Os, Fe, Ru dimers (TM_2) and supported on Bz (TM_2 -Bz), free heteronuclear Os-Fe dimers, Os-Ru dimers, and single Os, Fe, Ru adatoms on Bz (TM -Bz). Previously, the MAE values of Fe_2 , Ru_2 , and Os_2 dimers have been reported by Błoński and Hafners [6]. Our results for freestanding Fe_2 and Ru_2 dimers are generally consistent with their results in structure and physical properties, while the MAE value of Os_2 disagrees (see Table S3).

As shown in Fig. 4(a), when comparing the freestanding Os_2 (Fe_2 , Ru_2) and Os_2 -Bz (Fe_2 -Bz, Ru_2 -Bz), or freestanding Os-Ru (Os-Fe) and OsRu-Bz (OsFe-Bz) the ligand field provided by Bz plays an important role in fixing the easy magnetization direction, regardless of whether it is a homonuclear or heteronuclear system. Moreover, the OMA in $\text{Os}X$ -Bz systems is enhanced up to three times that of free dimers. In particular, when an Os-Fe dimer is anchored on Bz, the spin anisotropy and OMA increase concurrently, thus obtaining a large MAE of 67.2 meV. As displayed in Fig. 4(b), the upper atom has two roles in a M_1M_2 -Bz system. First, it induces the out-of-plane easy magnetization direction, as demonstrated by the results of Os-Bz (Ru-Bz, Fe-Bz) and Os_2 -Bz (Ru_2 -Bz, Fe_2 -Bz). Second, it also provides a strong SOC effect when

a $5d$ element is used. For example, when the top Fe atom in Fe_2 -Bz is replaced by Os, the MAE increases significantly from -1.79 to 67.2 meV. For Ru_2 -Bz and OsRu-Bz, the MAE also increases from 9.22 to 58.5 meV. Similar to the above discussions about spin and orbital magnetic moments, the bottom metal atom in the Os-Bz and OsFe-Bz (OsRu-Bz) complexes weakens the influence of the ligand field on the $5d$ TM atom [Fig. 4(c)]. This effect preserves the strong SOC strength of the top atom and makes it behave as a free atom.

According to previous experimental works, many similar molecules have already been synthesized by laser vaporization source, such as Fe_nBz_m , Ni_nBz_m , Rh_nBz_m , Co_nBz_m , and Pt_nBz_m [38–43]. Their magnetic recording properties have not been investigated, but highly anticipated. Although there is no evidence for the preparation of our proposed $M_1M_2\text{Bz}$, we still believe they would be achieved using the same method later.

Presumably for these single-molecule magnets to be useful in magnetic storage devices, we have also investigated their magnetic stability on supporting bases. We put these bimetal $\text{Os}X$ -Bz molecules on the representative 2D surfaces, such as h -BN, black phosphorus (BP), and graphene. To simulate the interactions between supporting bases and $\text{Os}X$ -Bz molecules, the 6×6 supercell for h -BN and graphene, and 4×4 supercell for BP were constructed, which are large enough to avoid interactions among dimers in the neighboring cells. Fortunately, one can see that the $\text{Os}X$ -Bz molecules will not dissociate into adatoms when placed on h -BN, BP, and graphene (Fig. S3). In addition, the large magnetic moments of our studied $\text{Os}X$ -Bz molecules are also preserved well in h -BN and BP (Table S4).

In our studied metal-benzene complexes, the strong coupling between transition metal atoms through d - d direct exchange interaction lead to the large MAE of ~ 54 meV, which could provide a kind of feasible building blocks for designing magnetic recording devices. Together with MAE, exchange interaction is also the critical point to determine the preferred direction of the magnetization. Clearly, the exchange interaction between the different molecule magnets is rather weak, which can be negligible. To investigate the exchange interactions in the molecule magnets, we have calculated the energy difference between a spin-parallel and a spin-antiparallel state. The values are shown in Table S5, which is 180 meV for OsRu-Bz (Os_2 -Bz) and 137 meV for OsFe-Bz, respectively. The corresponding exchange coupling parameters are estimated to be 300 and 153 meV for the OsRu-Bz and Os_2 -Bz systems. One can see that the exchange interactions between two magnetic ions is strong enough to resist a thermally induced spin flipping.

IV. CONCLUSIONS

To summarize, we have investigated the stability and magnetic properties of transition metal dimers adsorbed on benzene using systematical first-principles calculations. Positioning the dimer on the center of a carbon hexagon, arranged upright with the lighter atom nearest to the benzene, is the optimal configuration in terms of both stability and magnetic properties. When the $\text{Os}X$ ($X = \text{Fe}, \text{Ru}$, and Os) dimer is supported on Bz, the large orbital moment anisotropy of top Os atom can be achieved owing to the unquenching orbital

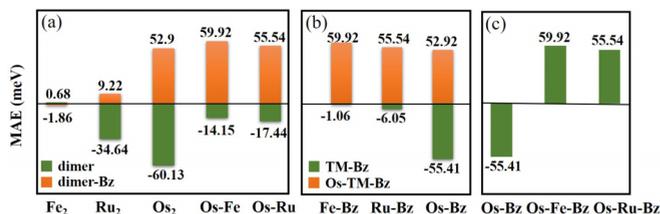


FIG. 4. MAE values of free dimers (homonuclear and heteronuclear dimers) and dimers or single atoms absorbed on Bz.

angular momentum. Large MAEs above 54 meV are obtained in OsX-Bz systems. Furthermore, the origin of large MAE of OsX-Bz complexes has been in depth understood and the roles of each component have been identified.

In the OsX-Bz complexes, the upper Os atom plays an important role in producing large MAE values when it is protected well by the lower transition metal X atom with strong intra-atomic exchange integrals. Compared with 4d or 5d atoms, 3d atoms in the lower metal site display better performance in terms of manipulating the electronic structure of the upper atom near the Fermi level. Such difference is due to stronger *d-d* direct exchange interaction between them than 4d/5d atoms-based systems; the corresponding energy level shifts have been presented in the paper. The benzene substrate not only provides support for the dimer, but also fixes its easy magnetization axis to the perpendicular direction. Furthermore, the OsX-Bz molecules arranged on the bases (*h*-BN, black phosphorus) still show good performance—that

is, their structural stability and large magnetic moments are well retained. Our study provides a further understanding of the underlying physics about the magnetic anisotropy mechanism in bimetal single-molecule magnets. It may give some guidance about the design of multicenter molecular magnets.

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