### Processing effect on spin-orbit torque switching and efficiency characterization in perpendicularly magnetized pillar devices

Wei-Bang Liao,<sup>1</sup> Chun-Yi Lin,<sup>1</sup> Tung-Yue Cheng,<sup>1</sup> Chao-Chung Huang<sup>0</sup>,<sup>1</sup> Tian-Yue Chen,<sup>1,\*</sup> and Chi-Feng Pai<sup>0</sup>,<sup>2,3,†</sup>

<sup>1</sup>Department of Materials Science and Engineering, National Taiwan University, Taipei 10617, Taiwan <sup>2</sup>Center of Atomic Initiative for New Materials, National Taiwan University, Taipei 10617, Taiwan <sup>3</sup>Center for Quantum Science and Engineering, National Taiwan University, Taipei 10617, Taiwan

(Received 26 May 2023; revised 5 September 2023; accepted 6 October 2023; published 24 October 2023)

Spin-orbit torque (SOT)-induced magnetization switching can be explained by either a single-domain coherent switching scenario or a domain-wall dynamics scenario, depending on the device size. In this study, we systematically investigate the processing effect on magnetic and SOT properties across a broad range of device sizes and geometries, from 5-um Hall bars to submicrometer-sized pillars and magnetic tunnel junctions (MTJs), with an identical W/CoFeB/MgO magnetic heterostructure. We first examine the impact of the fabrication process and device size on the measured magnetic properties, where coercivity  $H_c$  enhances while reducing the pillar size. Next, the current-driven hysteresis loop-shift measurement is utilized to characterize the dampinglike SOT efficiency  $\xi_{DL}$ , which is found to be fairly size independent. In contrast, current-induced SOT switching shows that the critical switching current density  $J_{sw}$  increases significantly with the device size reduction, suggesting a strong correlation between  $J_{sw}$  and  $H_c$ . Nevertheless, the current-driven loop-shift phase diagram and the domain-wall depinning model provide a relatively consistent estimation of  $J_{sw}$ , verifying the applicability of these two methods in the studied pillar size range. Last, we compare the SOT switching results from a micrometer-sized Hall bar device and a micrometer-sized three-terminal MTJ device, where a consistent estimated dampinglike SOT efficiency can be reached. Our results give insight into the processing and size effects on magnetic devices and provide proper protocols for SOT efficiency and critical switching current density estimations.

DOI: 10.1103/PhysRevMaterials.7.104409

#### I. INTRODUCTION

Spin torque-based magnetic random access memory (MRAM) is now fast developing for embedded applications and has the potential to serve as a last-level cache due to its nonvolatile nature, fast operation speed, and low power consumption. The spin-orbit torque (SOT) MRAM prototype [1-4] proposed a decade ago leverages the spin Hall effect in heavy transition metals to form a three-terminal magnetic tunnel junction (MTJ) device architecture that can separate read path and write path without flowing a tremendous amount of current through the tunnel barrier. Therefore, compared to the conventional spin-transfer torque MRAM with two-terminal geometry [5–7], SOT MRAM can potentially give rise to more efficient magnetization switching and higher endurance. SOT MRAM devices can be further categorized into various types depending on the relative orientation between the anisotropy axis and the spin-polarization directions. For example, type-zSOT MRAM with perpendicular magnetic anisotropy (PMA) has the advantage of reaching higher storage density and shorter switching time (subnanosecond) compared to those with type-y configuration (in-plane anisotropy) [6,8,9]. Detailed research and development of type-z SOT MRAM devices is therefore critical for realizing energy-efficient SOT MRAM products.

However, the fabrication of a fully functional threeterminal MTJ device with high tunneling magnetoresistance (TMR) could be challenging and time-consuming. In the semiconductor industry, it is common to employ suitable test vehicles or testkey devices for preliminary property screening during the early stage of development. Micrometer-sized Hall bar devices with normal metal-ferromagnet magnetic heterostructures are therefore widely adopted as the test vehicle for preliminary SOT characterizations due to their more straightforward fabrication process and broader applicability. In an MTJ device, the magnetic state readout comes from the current perpendicular-to-plane resistance  $R_{\rm MTI}$  through TMR. In contrast, the magnetic state of a Hall bar device with PMA is detected through the transverse voltage arm employing the anomalous Hall effect (AHE) measurement [10–17], which can streamline the study of SOT properties and improve device development efficiency. However, it is essential to note that using Hall bar devices as test vehicles requires careful consideration, such as the device dimensions, which could significantly impact the switching dynamics [4,18–28]. To bridge the gap between the Hall bar device and MTJ, a magnetic pillar can be fabricated or patterned at the Hall-cross region. The size effects can then be investigated by defining various sizes of pillars on the Hall bar device. It is also crucial to know that, for micrometer-sized devices in the multidomain regime, magnetization reversal occurs due

<sup>\*</sup>Present address: Department of Physics, New York University, NY 10012, USA.

<sup>&</sup>lt;sup>†</sup>cfpai@ntu.edu.tw



FIG. 1. (a) Illustration of pillar-shaped device on Hall bar. (b) SEM images of photolithography-defined pillar on 5- $\mu$ m-wide Hall bar (left) and e-beam lithography-defined pillar on 1- $\mu$ m-wide Hall bar (right). (c)  $H_c$  of pillar device vs etching time. (d)  $H_c$  as function of pillar size. Dashed line represents value for extended Hall bar.

to domain nucleation and domain-wall depinning (motion) nature [12,25,26]. In contrast, the magnetization reversal process is coherent rotation if the device size is less than 80 nm, which follows a single-domain scenario [4,20–23,27]. Thus, choosing an appropriate model to analyze the results obtained from various-sized pillar devices is critical.

In this work, we systematically characterize the SOT efficiency and perform current-induced SOT switching measurements on a wide range of devices, from 5-µm Hall bar to submicrometer-sized pillars (on the Hall cross) and MTJs with identical W-based magnetic heterostructures. First, we analyze the resistance and coercivity  $(H_c)$  change during the fabrication (Ar ion-beam etching) process, and we observe a noticeable increase in both when the magnetic pillar is formed out of the Hall bar. Next, SOT characterization results from current-induced hysteresis loop-shift measurement show a fairly constant dampinglike SOT (DL SOT) efficiency  $|\xi_{\rm DL}| \sim 0.35$  across all pillar sizes, suggesting the universality of the adopted technique. Current-induced SOT switching measurement shows that the critical switching current density  $(J_{sw})$  increases with decreasing the pillar size, corresponding to the enhancement of  $H_c$ . We further utilize two methods: (1) domain-wall depinning model and (2) loop-shift phase diagram approach to estimate  $J_{sw}$ . By carefully taking the in-plane field and the Joule heating effect on  $H_c$  into account, we find consistent results that affirm the strong correlation between the estimated  $|\xi_{DL}|$  and  $J_{sw}$ . Additionally, we adopt the domain-wall depinning model to estimate  $|\xi_{DL}|$  from a W-based Hall bar device and a three-terminal MTJ device with an identical film-stack design, which yields a consistent DL SOT efficiency of  $|\xi_{DL}| \sim 0.3$ . These results highlight the validity of using a Hall bar as test vehicle for early-stage SOT study and the importance of using suitable models that take different effects acting on coercivity into account for analysis.

### II. DEVICE FABRICATION AND SIZE DEPENDENCE OF $H_c$

Thin-film W(4)/CoFeB(1.4)/MgO(1.5)/Ta(2) (units in nanometer) magnetic heterostructures are prepared by magnetron sputtering. The multilayer stacks are deposited on a  $Si/SiO_2$  substrate with a base pressure of  $2 \times 10^{-8}$  Torr. The metallic and oxide layers are deposited by DC and rf sputtering with Ar working pressure of 3 and 10 mTorr, respectively. The thin films are first postannealed at 300 °C in the vacuum chamber for 1 h to induce PMA. Next, the films are patterned into Hall bar devices with a current channel width of 5 µm through photolithography and subsequent Ar ion-beam etching (IBE). Next, we define a square pillar at the center of the Hall cross through either photolithography or e-beam lithography, as illustrated in Fig. 1(a). Pillar sizes  $\ge 2 \,\mu m$  are defined by photolithography, whereas sizes  $< 2 \,\mu m$  are defined using e-beam lithography. A scanning electron microscope (SEM) image of pillar devices fabricated using photolithography is shown in Fig. 1(b). Note that as pillar size shrinks, the Hall signal decreases and makes the readout difficult (see Supplemental Material, S1 [29]). Therefore, for pillar size  $< 1 \,\mu m$ , the width of the current channel and voltage arm are further



FIG. 2. (a) Schematic illustration of current-induced hysteresis loop measurement on pillar device. Top inset shows nucleated domain in CoFeB layer with domain-wall moments aligned parallelly to bias field  $H_x$ . Red (blue) dots represent -z(+z) moments and arrows represent domain-wall moments. (b) Representative shifted loops under  $I_{DC} = \pm 2.5$  mA and  $H_x = 1500$  Oe for 1.3-µm pillar device. (c) Extracted switching field  $H_{sw}$  of up-to-down (black squares), down-to-up (red circles), and current-induced effective field  $H_z^{eff}$  (blue triangles) as functions of  $I_{DC}$  at  $H_x = 1500$  Oe. Open circle and open square indicate estimated switching currents extrapolated from quadratic fittings of switching fields. (d)  $H_z^{eff}$  as function of  $I_{DC}$  at  $H_x = \pm 1500$  Oe for representative pillar device. (e) DL SOT efficacy  $\chi$  as function of  $H_x$ . (f) DL SOT efficiency as function of nominal pillar size. Dashed line represents value estimated from Hall bar device.

narrowed down to 1  $\mu$ m using e-beam lithography and subsequent IBE to enhance the detected Hall signal, as shown in Fig. 1(b).

After either photolithography or e-beam lithography process, we employ IBE to pattern the pillars. Figure 1(c) summarizes  $H_c$  of the pillars obtained via AHE measurement as a function of the etching time.  $H_c$  initially increases with the etching time and then gradually saturates after the device is etched for 90 s, indicating the reduction of domain nucleation sites and/or domain numbers in the CoFeB layer. Notably, a precise etching method monitored by secondary ion mass spectrometer is developed to ensure the complete etching of CoFeB, which is essential for the quantitative characterization in the following sections (see more IBE details in Supplemental Material, S2 [29]).

To investigate the size effect on  $H_c$ , we perform the abovementioned etching method to fabricate pillar devices with various lateral dimensions. As summarized in Fig. 1(d),  $H_c$ strongly enhances from ~ 63 Oe for the original Hall bar to as large as 800 Oe as the pillar size is reduced down to 800 nm, which again can be attributed to the reduction of domain nucleation sites and/or number of domain within the pillar [29]. Previous works pointed out that the increase of  $H_c$  in decreasing device size could also be correlated to the change of demagnetization factor and effective perpendicular magnetic anisotropy [19,27].

# III. SIZE DEPENDENCE OF CHARACTERIZED SOT EFFICIENCY $\xi_{DL}$

To characterize the dependence of SOT efficiency on device size, we conduct hysteresis loop-shift measurement [10,11,30,31] on a series of pillar devices, as illustrated in Fig. 2(a). A DC current  $(I_{DC})$  is injected into the current channel and an in-plane bias field  $(H_x)$  is applied along the x axis, and the Hall resistance is detected along the transverse y direction. With increasing  $H_x$ , the domain-wall moments in any nucleated domains start to align with this field [31]. The DL SOT then exerts on the domain-wall moments, serving as an out-of-plane effective field  $H_z^{\text{eff}}$  that further facilitates domain expansion [26,31]. Note that when  $H_x$  is sufficiently large and comparable to the Dzyaloshinskii-Moriya interaction effective field  $(H_{\text{DMI}})$ , the domain-wall moments are parallelly aligned with  $H_x$  (full Néel configuration), resulting in a maximized SOT efficiency (more details in Refs. [26,32-34]). Representative results obtained from a 1.3-µm pillar device are shown in Figs. 2(b)-2(d). Figure 2(b) shows the shifted hysteresis loops with  $I_{DC} = \pm 2.5$  mA and  $H_x = 1500$  Oe. The sharp magnetization reversal of the hysteresis loop is a feature of small-sized pillar devices. By extracting the shift of hysteresis loops under various currents, the switching fields  $(H_{sw})$ and current-induced effective fields  $(H_z^{eff})$  are summarized in Fig. 2(c), where  $H_z^{\text{eff}}$  is found linearly proportional to the



FIG. 3. (a) Representative current-driven switching loops of Hall bar, 1.3-µm pillar, and 860-nm pillar device. (b) Critical switching current densities  $J_{sw}$  as function of pillar size. Estimated results are extracted from switching field extrapolation [as shown in Fig. 2(c)]. Dashed line presents  $J_{sw}$  of Hall bar. (c)  $H_c$  vs in-plane bias-field  $H_x$ . (d)  $H_c$  vs. applied current  $I_{DC}$  under  $H_x = 1500$  Oe.

applied  $I_{DC}$ . The opposite sign of the slopes at  $H_x = \pm 1500 \text{ Oe}$  shown in Fig. 2(d) corresponds to the opposite orientation of the domain-wall moment. The extracted DL SOT efficacies  $\chi = H_z^{\text{eff}}/J_{DC}$  vs  $H_x$  for a Hall bar and a 1.3-µm-wide pillar device are summarized in Fig. 2(e), where the magnitude of  $\chi$  saturates at  $\approx 90 \text{ Oe}/10^{11} \text{ A m}^{-2}$  for both devices. The fields to saturate  $\chi$  are extracted to be  $|H_{DMI}| \approx 1000 \text{ Oe}$  (the maximum among various nucleated domains) for the Hall bar and  $|H_{DMI}| \approx 1500 \text{ Oe}$  for the 1.3-µm pillar device [10]. It is noted that the  $H_{DMI}$  extracted by hysteresis loop-shift measurement only presents a rough estimation of the upper bound of  $H_{DMI}$  due to the multidomain nature of the tested devices. Also note that the current shunting effect has been considered using a parallel circuit model to estimate  $J_{DC}$ .

The DL SOT efficiency  $(\xi_{DL})$  can be further estimated through [10,32]

$$|\xi_{\rm DL}| = \frac{2e}{\hbar} \left(\frac{2}{\pi}\right) \mu_0 M_{\rm s} t_{\rm CoFeB} \chi, \qquad (1)$$

where  $M_s$  represents the saturation magnetization of CoFeB  $\approx 1200 \text{ emu/cc}$  (obtained by a vibrating sample magnetometer). Figure 2(f) summarizes the DL SOT efficiency of pillar devices with various sizes. It shows that the DL SOT efficiencies are fairly consistent (~0.35) among Hall bar devices, micrometer-sized and submicrometer-sized pillar devices. This result implies that even though the current-induced hysteresis loop-shift measurement is based on a multidomain scenario, it is still feasible to estimate the DL SOT efficiency

from micrometer-sized Hall bar devices and pillar-shaped devices with dimensions down to 800 nm. Note that another popular SOT characterization technique that employs Hall bar as test vehicle, the harmonic Hall voltage approach, is usually performed in the saturated state. Under such circumstance, the magnetization remains in a quasiuniform configuration and can be explained by a macrospin model to potentially simplify the characterization protocol. However, the predicament of calibrating the planar Hall correction and the thermal effects inherent in this method can potentially lead to incorrect estimation. The plight of these inconsistent SOT efficiency estimations still remains unresolved [35–40]; therefore, in this work we focus on only the loop shift and the switching approaches.

## IV. SIZE DEPENDENCE OF SOT SWITCHING CURRENT DENSITY $J_{sw}$

We further perform current-induced SOT switching measurements on pillar devices with various sizes. The measurement scheme is similar to Fig. 2(a) with applying different amplitudes of pulsed currents (pulse width  $t_{pulse} = 50 \text{ ms}$ ) and an in-plane bias field  $H_x$  on the device. Representative current-induced switching loops of a Hall bar, a 1.3-µm-wide pillar device, and an 860-nm-wide pillar device under  $H_x =$ 1500 Oe are summarized in Fig. 3(a).  $J_{sw}$  is enhanced when the device size is scaled down. We further outline the  $J_{sw}$ under  $H_x =$  1500 Oe as a function of pillar size in Fig. 3(b). The observed increase of  $J_{sw}$  while reducing pillar size is consistent with previous studies and can be well explained by the reduction of the domain nucleations and domain-wall propagations [21,23,28].

In our micrometer-sized and submicrometer-sized devices, the magnetization reversal process is typically governed by domain nucleation and the domain-wall depinning (see Supplemental Material, S3 [29]). To further examine the correlation between  $J_{sw}$  and previously estimated DL SOT efficiency, it has been proposed that the relation between  $J_{sw}$  and  $\xi_{DL}$  in this regime can be evaluated through [12,25,26,41]

$$J_{\rm sw} = \frac{2e}{\hbar} \left(\frac{2}{\pi}\right) \mu_0 M_{\rm s} t_{\rm CoFeB} \left(\frac{H_{\rm c}}{|\xi_{\rm DL}|}\right). \tag{2}$$

Here, we emphasize that the  $H_c$  employed in this estimation must first account for its in-plane bias-field dependence since  $H_c$  reduces when increasing  $H_x$  [42,43], as shown in Fig. 3(c).  $H_{\rm c}$  shows a significant reduction with increasing  $H_{\rm x}$  for the 1.3-µm-wide pillar device while it decreases less but not negligibly for the Hall bar. Moreover, as presented in Fig. 3(d), the Joule heating effect is also pronounced in the pillar device.  $H_c$  of the 1.3-µm-wide pillar device under  $H_x = 1500 \text{ Oe}$ with various currents is estimated from  $\frac{H_{sw}^{\text{UP-to-DN}} - H_{sw}^{\text{DN-to-UP}}}{2}$  in Fig. 2(c). The obvious quadratic relation between such  $H_c$  and  $I_{\rm DC}$  suggests that the Joule heating should also be accounted for to estimate the depinning field accurately [4,44–46]. In contrast, the Joule heating effect is less noticeable, and  $H_c$ reduces only slightly for the Hall bar device (see more details in Supplemental Material, S4 [29]). Failure to consider these effects would result in overestimating  $J_{sw}$  from  $\xi_{DL}$  (or  $\xi_{DL}$ from  $J_{sw}$ ). To show the validity of Eq. (2), we estimate  $J_{sw}$ with the  $\xi_{DL}$  obtained from loop-shift measurement and the  $H_{\rm c}$  obtained from the quadratic fitting in the loop-shift phase diagram [Fig. 2(c)] under  $H_x$  and  $I_{sw}$ . As shown in Fig. 3(b), the  $H_c/\xi_{DL}$ -estimated  $J_{sw}$  is consistent with the measured  $J_{sw}$ , confirming the strong correlation between DL-SOT efficiency and critical switching current density. It is noted that we also measure  $J_{sw}$  under various  $H_x$ , and  $\xi_{DL}$  estimated from a much simpler macrospin model shows an unreasonably large SOT efficiency ( $\sim$ -2.88), suggesting the assumption of coherent switching is inapplicable to pillar devices in both micrometersized and submicrometer-sized regimes (see Supplemental Material [29] for details).

Notably, Zhu *et al.* [25] have reported that the domainwall depinning model fails to estimate the  $J_{sw}$  (or  $\xi_{DL}$ ) even considering the decrease of  $J_{sw}$  due to  $H_x$ . However, here we highlight the importance of  $H_c$  reduction due to both  $H_x$  and the Joule heating. Alternatively, one can also directly use the loop-shift phase diagram [Fig. 2(c)] to estimate the critical switching current density without using Eq. (2). The critical switching current density can be directly extracted by performing quadratic fittings to the up-to-down and down-to-up switching fields under various currents and then extrapolating them to the *x* axis. The extracted  $J_{sw}s$  are denoted in Fig. 2(c) as the open symbols. The loop-shift estimated  $J_{sw}$  summarized in Fig. 3(b) also shows a consistent trend compared to the directly measured results.

### V. SOT SWITCHING: HALL BAR VS THREE-TERMINAL MTJ DEVICE

Based on previous analysis, we expand the scope of the study to three-terminal MTJ devices. The tested thin film has the structure of Ta(2)/W(4)/CoFeB (1.8)/MgO(1.5)/CoFeB(2)/W(0.5)/Ta(4), which is further postannealed at 400 °C for 20 min in vacuum to induce PMA. W(4) serves as the SOT source. The thinner CoFeB(1.8) is the free layer, and the thicker CoFeB(2) is the fixed layer. Multiple thin films with the same layer design are then fabricated into a Hall bar device and a three-terminal MTJ device for SOT switching measurement. Figure 4(a) shows the AHE result of a Hall bar device, in which the two-step hysteresis loop consists of magnetization reversal of the free and the fixed CoFeB layers. Current-induced SOT switching measurement is then performed on the Hall bar device, as shown in Fig. 4(b). The magnetization reversal is confirmed from the free layer through the Hall resistance change. The switching polarities are opposite under  $H_x = \pm 1000 \,\text{Oe}$ , which follows the SOT symmetry.

Figure 4(c) shows the structure and the measurement scheme of a three-terminal MTJ device. The bottom Ta(2)/W(4) layers are patterned into a 30-µm-wide strip, and the rest of the layers are etched with IBE to form a micrometer-sized magnetic pillar with lateral dimensions of  $10 \times 10 \,\mu\text{m}^2$ . The MTJ resistance  $R_{\text{MTJ}}$  is measured in response to an out-of-plane field  $(H_z)$  or pulsed currents under  $H_x$ . Figure 4(d) shows both major and minor loops of  $R_{\rm MTJ}$  vs  $H_z$  with a TMR ratio  $\approx 14\%$ . The switching of the free (fixed) layer is observed in the minor (major) loop at  $|H_z| \approx 100 \,\text{Oe}$ (230 Oe). Compared to Fig. 4(a), the switching fields of the free layer and the fixed layer are higher in MTJ than those in the Hall bar device, which is attributed to the reduced lateral dimensions of the ferromagnetic layer in MTJ. It echoes the results in the previous section that the domain-wall number is less in the patterned pillar-shaped device.

Current-induced SOT switching is then performed on the three-terminal MTJ, as shown in Fig. 4(e). Compared to the Hall bar device [Fig. 4(b)], the higher  $J_{sw}$  of the MTJ device is consistent with the larger  $H_c$  observed. The critical switching current densities under various  $H_x$  are extracted from the Hall bar and the MTJ device, as shown in Fig. 4(f). Both devices follow a similar  $H_x - J_{sw}$  dependence, except that a higher  $J_{sw}$ is observed in the MTJ device under the same  $H_x$ , which results from the higher  $H_c$  in the MTJ device. Finally, to directly estimate the DL SOT efficiency from MTJ devices, the domain-wall depinning analysis should be applied, where  $\chi = H_c/J_{sw}$ . It shows that  $\chi$  for both the Hall bar and the MTJ device is  $\approx 80 \text{ Oe}/10^{-11} \text{ A m}^{-2}$ , corresponding to a DL SOT efficiency  $\approx 0.3$ , which is close to the results of Figs. 2(e) and 2(f). Again, the validity of the domain-wall depinning analysis for estimating DL SOT efficiency is verified by using a three-terminal MTJ device.

It is finally noted that in this work we delve into W-based Hall bar, pillar-shaped, and fully patterned MTJ devices, since W is one of the most commonly employed spin Hall source materials for contemporary CoFeB/MgO-based MTJ. However, heterostructures or devices based on different spin Hall materials might behave differently. We hope our present work



FIG. 4. (a) Representative AHE loop obtained from Hall bar device with MTJ layer stack Ta(2)/W(4)/CoFeB(1.8)/MgO(1.5)/ CoFeB(2)/W(0.5)/Ta(4). (b) Representative current-induced switching loops under  $H_x = \pm 1000$  Oe of Hall bar device. (c) Structure and measurement scheme for three-terminal MTJ device with pillar size  $10 \times 10 \,\mu\text{m}^2$  and current channel width 30  $\mu$ m. (d) TMR major and minor loops. (e) MTJ resistance as function of  $J_{\text{pulse}}$  under  $H_x = \pm 1000$  Oe. (f) Switching-phase diagram ( $J_{\text{sw}}-H_x$ ) from Hall bar device and three-terminal MTJ device. Solid (open) symbols represent  $J_{\text{sw}}$  extracted from Hall bar (MTJ). Dashed lines represent guidelines for eyes.

could spark future studies to explore other materials systems using the proposed characterization protocols.

### **VI. CONCLUSION**

In conclusion, we systematically analyze the fabrication process and the size dependence of both magnetic and SOT characteristics of various types of devices.  $H_c$  shows a substantial enhancement when the device size is reduced, which can be attributed to the reduction of the domain nucleation sites or the hindered domain-wall motion. The DL SOT efficiency of pillar devices is characterized through hysteresis loop-shift measurement, where the value of  $\xi_{DL}$  remains fairly constant across varying pillar sizes. In addition, currentinduced SOT switching experiments reveal that  $J_{sw}$  exhibits a noticeable increase with decreasing pillar size, corresponding to a similar trend from  $H_c$ . The switching-phase diagrams of submicrometer-sized devices show trends that cannot be quantitatively described by the macrospin model. Therefore, a domain-wall depinning model is required to accurately estimate  $J_{sw}$  from the characterized SOT efficiency, in which the  $H_c$  reduction from both in-plane bias-field  $H_x$  and the Joule heating effect are also considered. Additionally, the  $J_{sw}$ estimated from the loop-shift phase diagram are consistent with those obtained directly from switching measurements. These findings highlight the strong correlation between  $\xi_{DL}$ and  $J_{sw}$ , even in micrometer-sized devices that do not follow

a single-domain behavior. Finally, we point out that Hall bar devices can serve as effective test vehicles for characterizing SOT efficiency and  $J_{sw}$  before investing heavy resources in MTJ (three-terminal SOT device) fabrication, as long as the processing-correlated factors are correctly taken into account while performing analysis.

#### ACKNOWLEDGMENTS

This work is supported by the National Science and Technology Council (NSTC) under Grant No. NSTC-112– 2636-M-002–006 and by the Center of Atomic Initiative for New Materials (AI-Mat) and the Advanced Research Center of Green Materials Science and Technology, National Taiwan University from the Featured Areas Research Center Program within the framework of the Higher Education Sprout Project by the Ministry of Education (MOE) in Taiwan under Grant No. NTU-112L9008.

W.-B.L. conceived the experiments, fabricated the devices, performed the analysis of the data, and drafted the manuscript; C.-Y.L. performed the hysteresis loop-shift measurement and current-induced SOT switching measurement; T.-Y. Cheng fabricated the magnetic tunnel junction; C.-C.H. performed the current-induced SOT switching measurement; T.-Y. Chen performed the analysis of the data; and C.-F.P. proposed and supervised the study.

The authors declare no competing interests.

- L. Liu, C.-F. Pai, Y. Li, H. W. Tseng, D. C. Ralph, and R. A. Buhrman, Spin-torque switching with the giant spin Hall effect of tantalum, Science 336, 555 (2012).
- [2] C.-F. Pai, L. Liu, Y. Li, H. W. Tseng, D. C. Ralph, and R. A. Buhrman, Spin transfer torque devices utilizing the giant spin Hall effect of tungsten, Appl. Phys. Lett. 101, 122404 (2012).
- [3] M. Cubukcu, O. Boulle, M. Drouard, K. Garello, C. O. Avci, I. M. Miron, J. Langer, B. Ocker, P. Gambardella, and G. Gaudin, Spin-orbit torque magnetization switching of a three-terminal perpendicular magnetic tunnel junction, Appl. Phys. Lett. 104, 042406 (2014).
- [4] K. Garello, C. O. Avci, I. M. Miron, M. Baumgartner, A. Ghosh, S. Auffret, O. Boulle, G. Gaudin, and P. Gambardella, Ultrafast magnetization switching by spin-orbit torques, Appl. Phys. Lett. 105, 212402 (2014).
- [5] T. Kawahara, K. Ito, R. Takemura, and H. Ohno, Spin-transfer torque RAM technology: Review and prospect, Microelectron. Reliab. 52, 613 (2012).
- [6] S. Ikeda, K. Miura, H. Yamamoto, K. Mizunuma, H. D. Gan, M. Endo, S. Kanai, J. Hayakawa, F. Matsukura, and H. Ohno, A perpendicular-anisotropy CoFeB–MgO magnetic tunnel junction, Nat. Mater. 9, 721 (2010).
- [7] W. G. Wang, M. Li, S. Hageman, and C. L. Chien, Electricfield-assisted switching in magnetic tunnel junctions, Nat. Mater. 11, 64 (2011).
- [8] E. Grimaldi, V. Krizakova, G. Sala, F. Yasin, S. Couet, G. Sankar Kar, K. Garello, and P. Gambardella, Single-shot dynamics of spin-orbit torque and spin transfer torque switching in three-terminal magnetic tunnel junctions, Nat. Nanotechnol. 15, 111 (2020).
- [9] I. M. Miron, K. Garello, G. Gaudin, P. J. Zermatten, M. V. Costache, S. Auffret, S. Bandiera, B. Rodmacq, A. Schuhl, and P. Gambardella, Perpendicular switching of a single ferromagnetic layer induced by in-plane current injection, Nature (London) 476, 189 (2011).
- [10] C.-F. Pai, M. Mann, A. J. Tan, and G. S. D. Beach, Determination of spin torque efficiencies in heterostructures with perpendicular magnetic anisotropy, Phys. Rev. B 93, 144409 (2016).
- [11] T. Dohi, S. Fukami, and H. Ohno, Influence of domain wall anisotropy on the current-induced hysteresis loop shift for quantification of the Dzyaloshinskii-Moriya interaction, Phys. Rev. B 103, 214450 (2021).
- [12] T.-Y. Chen, C.-T. Wu, H.-W. Yen, and C.-F. Pai, Tunable spinorbit torque in Cu-Ta binary alloy heterostructures, Phys. Rev. B 96, 104434 (2017).
- [13] W.-B. Liao, T.-Y. Chen, Y. Ferrante, S. S. P. Parkin, and C.-F. Pai, Current-induced magnetization switching by the high spin Hall conductivity  $\alpha$ -W, Phys. Status Solidi RRL **13**, 1900408 (2019).
- [14] T.-Y. Chen, Y.-C. Hsiao, W.-B. Liao, and C.-F. Pai, Tailoring neuromorphic switching by CuN<sub>x</sub>-Mediated orbital currents, Phys. Rev. Appl. 17, 064005 (2022).
- [15] J. Kim, J. Sinha, M. Hayashi, M. Yamanouchi, S. Fukami, T. Suzuki, S. Mitani, and H. Ohno, Layer thickness dependence of the current-induced effective field vector in Ta|CoFeB|MgO, Nat. Mater. 12, 240 (2013).
- [16] K. Garello, I. M. Miron, C. O. Avci, F. Freimuth, Y. Mokrousov, S. Blugel, S. Auffret, O. Boulle, G. Gaudin, and P. Gambardella,

Symmetry and magnitude of spin-orbit torques in ferromagnetic heterostructures, Nat. Nanotechnol. **8**, 587 (2013).

- [17] J. Kim, P. Sheng, S. Takahashi, S. Mitani, and M. Hayashi, Spin Hall magnetoresistance in metallic bilayers, Phys. Rev. Lett. 116, 097201 (2016).
- [18] H. Sato, M. Yamanouchi, K. Miura, S. Ikeda, H. D. Gan, K. Mizunuma, R. Koizumi, F. Matsukura, and H. Ohno, Junction size effect on switching current and thermal stability in CoFeB/MgO perpendicular magnetic tunnel junctions, Appl. Phys. Lett. **99**, 042501 (2011).
- [19] J. W. Chenchen, M. A. K. B. Akhtar, R. Sbiaa, M. Hao, L. Y. H. Sunny, W. S. Kai, L. Ping, P. Carlberg, and A. K. S. Arthur, Size dependence effect in MgO-Based CoFeB tunnel junctions with perpendicular magnetic anisotropy, Jpn. J. Appl. Phys. 51, 013101 (2012).
- [20] K.-S. Lee, S.-W. Lee, B.-C. Min, and K.-J. Lee, Threshold current for switching of a perpendicular magnetic layer induced by spin Hall effect, Appl. Phys. Lett. **102**, 112410 (2013).
- [21] C. Zhang, S. Fukami, H. Sato, F. Matsukura, and H. Ohno, Spin-orbit torque induced magnetization switching in nano-scale Ta/CoFeB/MgO, Appl. Phys. Lett. 107, 012401 (2015).
- [22] C. Zhang, S. Fukami, K. Watanabe, A. Ohkawara, S. DuttaGupta, H. Sato, F. Matsukura, and H. Ohno, Critical role of W deposition condition on spin-orbit torque induced magnetization switching in nanoscale W/CoFeB/MgO, Appl. Phys. Lett. 109, 192405 (2016).
- [23] B. Jinnai, H. Sato, S. Fukami, and H. Ohno, Scalability and wide temperature range operation of spin-orbit torque switching devices using Co/Pt multilayer nanowires, Appl. Phys. Lett. 113, 212403 (2018).
- [24] A. Kurenkov, C. Zhang, S. DuttaGupta, S. Fukami, and H. Ohno, Device-size dependence of field-free spin-orbit torque induced magnetization switching in antiferromagnet/ferromagnet structures, Appl. Phys. Lett. **110**, 092410 (2017).
- [25] L. Zhu, D. C. Ralph, and R. A. Buhrman, Lack of simple correlation between switching current density and spin-orbittorque efficiency of perpendicularly magnetized spin-currentgenerator–ferromagnet heterostructures, Phys. Rev. Appl. 15, 024059 (2021).
- [26] O. J. Lee, L. Q. Liu, C. F. Pai, Y. Li, H. W. Tseng, P. G. Gowtham, J. P. Park, D. C. Ralph, and R. A. Buhrman, Central role of domain wall depinning for perpendicular magnetization switching driven by spin torque from the spin Hall effect, Phys. Rev. B 89, 024418 (2014).
- [27] W. Stefanowicz, L. E. Nistor, S. Pizzini, W. Kuch, L. D. Buda-Prejbeanu, G. Gaudin, S. Auffret, B. Rodmacq, and J. Vogel, Size dependence of magnetic switching in perpendicularly magnetized MgO/Co/Pt pillars close to the spin reorientation transition, Appl. Phys. Lett. 104, 012404 (2014).
- [28] C. Zhang, S. Fukami, S. DuttaGupta, H. Sato, and H. Ohno, Time and spatial evolution of spin–orbit torque-induced magnetization switching in W/CoFeB/MgO structures with various sizes, Jpn. J. Appl. Phys. 57, 04FN02 (2018).
- [29] See Supplemental Material at http://link.aps.org/supplemental/ 10.1103/PhysRevMaterials.7.104409 for details of Hall signal with respect to pillar size, etching time dependence of coercivity, angle dependence of coercivity, switching-phase diagram, and J<sub>sw</sub> estimation on Hall bar device. It also includes Refs. [47–49].

- [30] L. Liu, O. J. Lee, T. J. Gudmundsen, D. C. Ralph, and R. A. Buhrman, Current-induced switching of perpendicularly magnetized magnetic layers using spin torque from the spin Hall effect, Phys. Rev. Lett. 109, 096602 (2012).
- [31] P. P. Haazen, E. Mure, J. H. Franken, R. Lavrijsen, H. J. Swagten, and B. Koopmans, Domain wall depinning governed by the spin Hall effect, Nat. Mater. 12, 299 (2013).
- [32] S. Emori, E. Martinez, K.-J. Lee, H.-W. Lee, U. Bauer, S.-M. Ahn, P. Agrawal, D. C. Bono, and G. S. D. Beach, Spin Hall torque magnetometry of Dzyaloshinskii domain walls, Phys. Rev. B 90, 184427 (2014).
- [33] K. S. Ryu, L. Thomas, S. H. Yang, and S. Parkin, Chiral spin torque at magnetic domain walls, Nat. Nanotechnol. 8, 527 (2013).
- [34] S. Emori, U. Bauer, S. M. Ahn, E. Martinez, and G. S. Beach, Current-driven dynamics of chiral ferromagnetic domain walls, Nat. Mater. 12, 611 (2013).
- [35] M. Hayashi, J. Kim, M. Yamanouchi, and H. Ohno, Quantitative characterization of the spin-orbit torque using harmonic Hall voltage measurements, Phys. Rev. B 89, 144425 (2014).
- [36] T. Schulz, K. Lee, B. Krüger, R. Lo Conte, G. V. Karnad, K. Garcia, L. Vila, B. Ocker, D. Ravelosona, and M. Kläui, Effective field analysis using the full angular spin-orbit torque magnetometry dependence, Phys. Rev. B 95, 224409 (2017).
- [37] L. Zhu, K. Sobotkiewich, X. Ma, X. Li, D. C. Ralph, and R. A. Buhrman, Strong damping-like spin-orbit torque and tunable Dzyaloshinskii–Moriya interaction generated by low-resistivity Pd<sub>1-x</sub>Pt<sub>x</sub> alloys, Adv. Funct. Mater. **29**, 1805822 (2019).
- [38] S. Karimeddiny, T. M. Cham, D. C. Ralph, and Y. K. Luo, Sagnac interferometry for high-sensitivity optical measurements of spin-orbit torque, Sci. Adv. 9, eadi9039 (2023).
- [39] N. Roschewsky, E. S. Walker, P. Gowtham, S. Muschinske, F. Hellman, S. R. Bank, and S. Salahuddin, Spin-orbit torque and Nernst effect in Bi-Sb/Co heterostructures, Phys. Rev. B 99, 195103 (2019).

- [40] T.-Y. Chen, C.-W. Peng, T.-Y. Tsai, W.-B. Liao, C.-T. Wu, H.-W. Yen, and C.-F. Pai, Efficient spin–orbit torque switching with nonepitaxial chalcogenide heterostructures, ACS Appl. Mater. Interfaces 12, 7788 (2020).
- [41] D. Bhowmik, M. E. Nowakowski, L. You, O. Lee, D. Keating, M. Wong, J. Bokor, and S. Salahuddin, Deterministic domain wall motion orthogonal to current flow due to spin orbit torque, Sci. Rep. 5, 11823 (2015).
- [42] S. Kanai, Y. Nakatani, M. Yamanouchi, S. Ikeda, F. Matsukura, and H. Ohno, In-plane magnetic field dependence of electric field-induced magnetization switching, Appl. Phys. Lett. 103, 072408 (2013).
- [43] C. Grezes, A. Rojas Rozas, F. Ebrahimi, J. G. Alzate, X. Cai, J. A. Katine, J. Langer, B. Ocker, P. Khalili Amiri, and K. L. Wang, In-plane magnetic field effect on switching voltage and thermal stability in electric-field-controlled perpendicular magnetic tunnel junctions, AIP Adv. 6, 075014 (2016).
- [44] K.-S. Lee, S.-W. Lee, B.-C. Min, and K.-J. Lee, Thermally activated switching of perpendicular magnet by spin-orbit spin torque, Appl. Phys. Lett. 104, 072413 (2014).
- [45] D. Li, S. Chen, Y. Zuo, J. Yun, B. Cui, K. Wu, X. Guo, D. Yang, J. Wang, and L. Xi, Roles of Joule heating and spin-orbit torques in the direct current induced magnetization reversal, Sci. Rep. 8, 12959 (2018).
- [46] D. Bedau, H. Liu, J. Z. Sun, J. A. Katine, E. E. Fullerton, S. Mangin, and A. D. Kent, Spin-transfer pulse switching: From the dynamic to the thermally activated regime, Appl. Phys. Lett. 97, 262502 (2010).
- [47] E. C. Stoner and E. P. Wohlfarth, A mechanism of magnetic hysteresis in heterogeneous alloys, IEEE Trans. Magn. 27, 3475 (1991).
- [48] F. Schumacher, On the modification of the Kondorsky function, J. Appl. Phys. 70, 3184 (1991).
- [49] S. Fukami, T. Anekawa, C. Zhang, and H. Ohno, A spin-orbit torque switching scheme with collinear magnetic easy axis and current configuration, Nat. Nanotechnol. 11, 621 (2016).