

Composition-tunable magnon-polaron anomalies in spin Seebeck effects in epitaxial $\text{Bi}_x\text{Y}_{3-x}\text{Fe}_5\text{O}_{12}$ films

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We have investigated hybridized magnon-phonon excitation (magnon polarons) in spin Seebeck effects (SSEs) in $\text{Bi}_x\text{Y}_{3-x}\text{Fe}_5\text{O}_{12}$ ($\text{Bi}_x\text{Y}_{3-x}\text{IG}$; $x = 0, 0.5,$ and 0.9) films with Pt contact. We observed sharp peak structures in the magnetic field H dependence of the longitudinal SSE (LSSE) voltages, which appear when the phonon dispersions are tangential to the magnon dispersion curve in $\text{Bi}_x\text{Y}_{3-x}\text{IG}$. By increasing the Bi amount x , the peak fields in the LSSE shift toward lower H values due to the reduction of the sound velocities in $\text{Bi}_x\text{Y}_{3-x}\text{IG}$. We also measured the SSE in a nonlocal configuration and found that magnon-polaron anomalies appear with different signs and intensities. Our result shows composition-tunability of magnon-polaron anomalies and provides a clue to further unravel the physics of magnon-polaron SSEs.

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I. INTRODUCTION

Bismuth (Bi) substituted yttrium iron garnet (YIG) has been an important material in magnetics and magneto-optics [1,2]. In spite of Bi^{3+} being a diamagnetic ion, via the spin-orbit interaction, the substitution dramatically increases the Faraday rotation angle of garnets (by a factor of $\sim 10^2$ compared to primitive YIG [1,2]), which led to its versatile magneto-optical application. Another interesting feature of Bi-substituted YIG (Bi:YIG) includes the increased Curie temperature [1,2] and magnetoelastic interaction [3]. Fabrication of strained Bi:YIG ultrathin films with perpendicular magnetic anisotropy and low Gilbert damping has also been reported, offering a new opportunity for their spintronic applications [4–6]. Furthermore, because of the atomic mass of Bi (= 209) much larger than that of Y (= 89), a decrease of the sound velocities of Bi:YIG ($c_{\text{TA,LA}}^{\text{Bi:YIG}}$) has been reported for both the transverse acoustic (TA) and longitudinal acoustic (LA) modes compared to those of YIG ($c_{\text{TA,LA}}^{\text{YIG}}$) [7,8], as schematically shown in Fig. 1(a) (see also Table I).

Lately, in spintronics, magnon-phonon coupled phenomena have renewed attention, where not only magnon but also phonon dispersion relations play an important role [9–25]. Of particular interest is the region of the crossings of their branches, at which magnons and phonons are allowed to

hybridize by the magnetoelastic interaction, giving rise to the magnon-polaron modes having both magnonic and phononic characters [11,13,26]. In thermal magnon-spin transport, or the spin Seebeck effect (SSE) [27–30], the magnon-polaron formation manifests as anomalous peak or dip structures at the onset fields $H_{\text{TA,LA}}$, where the phonon dispersions in a magnet tangentially touch the magnon dispersion gapped by the Zeeman interaction (\propto external magnetic field H) [see Fig. 1(b)] [13,26]. A calculation based on Boltzmann transport theory reveals that, when the scattering rate of magnons τ_{mag}^{-1} is larger (smaller) than that of phonons τ_{ph}^{-1} , magnon polarons may have a longer (shorter) relaxation time than pure magnons. The SSE intensity is therefore enhanced (suppressed) at the touching fields, where the effect of magnon-polaron formation is maximal [26]. So far, through longitudinal SSE (LSSE) experiments, magnon-polaron peaks have been detected for several magnetic films, indicating the situation of $\tau_{\text{mag}}^{-1} > \tau_{\text{ph}}^{-1}$ [13,31–35], while dips were observed for some specific YIG bulk samples (i.e., $\tau_{\text{mag}}^{-1} < \tau_{\text{ph}}^{-1}$) [30,36].

Here, we report SSEs in epitaxially grown $\text{Bi}_x\text{Y}_{3-x}\text{Fe}_5\text{O}_{12}$ ($\text{Bi}_x\text{Y}_{3-x}\text{IG}$ with $x = 0, 0.5,$ and 0.9) films. The $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films may provide an interesting platform to systematically study magnon-polaron anomalies in SSEs, as the films exhibit the enhanced magnetoelastic coupling constant B_{\perp} [3] and the reduced sound velocities $c_{\text{TA,LA}}$ by increasing the Bi amount. The former feature leads to an increase in magnon-phonon hybridized region in momentum space (i.e., increase in the anticrossing gap $\propto B_{\perp}$ [26]), and is thus advantageous

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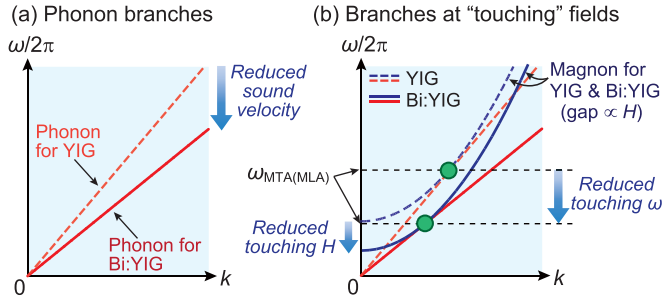


FIG. 1. (a) A schematic illustration of the phonon dispersion relations for pure YIG and Bi-substituted YIG (Bi:YIG). (b) A schematic illustration of the magnon and phonon dispersion relations for pure YIG and Bi:YIG at the touching field. Owing to the reduction of the sound velocity by the Bi substitution, the touching field $H_{TA(LA)}$ and the touching angular frequency $\omega_{MTA(MLA)}$ between the magnon and TA(LA)-phonon branches for Bi:YIG shift toward lower values compared to those for YIG.

for magnon-polaron SSEs. The latter feature decreases the magnon-phonon touching fields, since they scale with the sound-velocity squared, as discussed later [$H_{TA(LA)} \propto c_{TA(LA)}^2$, see Eq. (2)]. This situation can also be understood intuitively through a sketch of the magnon-phonon branches at the touching field; as shown in Fig. 1(b), the external H required for making the touching condition decreases for Bi:YIG having small $c_{TA(LA)}$ compared to YIG. In this paper, we first show structural and magnetic characterization of the epitaxial $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films used in this study and evaluate their sound velocities and magnetoelastic coupling constant (from Sec. III A to Sec. III C). We then show experimental results on the LSSE in the Pt/ $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films, where decreased anomaly fields are indeed observed for Bi:YIG (Sec. III D). We also performed experiments on the SSE in a nonlocal configuration, in which magnon-polaron anomalies appear differently from those in the longitudinal configuration in terms of their sign and intensity (Sec. III E). Our results show composition-tunability of magnon-polaron anomalies and provide a clue to further understand magnon-polaron SSEs.

II. SAMPLE PREPARATION AND EXPERIMENTAL SETUP

We have grown three types of $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films by a liquid phase epitaxy (LPE) method [37–42]: YIG, $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$, and $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ with the thickness of 2.50, 1.16, and 0.78 μm , respectively. The YIG and $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$ films were grown on (111) planes of 0.5-mm-thick GGG substrates, while the $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ film was grown on a (111) plane of a 0.7-mm-thick $(\text{GdCa})_3(\text{GaMgZr})_5\text{O}_{12}$ (substituted-GGG; SGGG) substrate to reduce the lattice mismatch between the film and substrate layers. Here, the lattice constant for cubic YIG (a_{YIG}), $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$ ($a_{\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}}$), $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ ($a_{\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}}$), GGG (a_{GGG}), and SGGG (a_{SGGG}) are 12.376, 12.417, 12.450, 12.383, and 12.508 \AA , respectively [8,43]. The LPE fabrication for the pure YIG film ($x = 0$) was done in $\text{PbO-B}_2\text{O}_3$ flux at the temperature of $T = 1210\text{ K}$ [41], while that for the $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ film with $x = 0.5$ (0.9) was done in $\text{PbO-Bi}_2\text{O}_3\text{-B}_2\text{O}_3$ flux at 1053 K (1023 K). After the growth, the

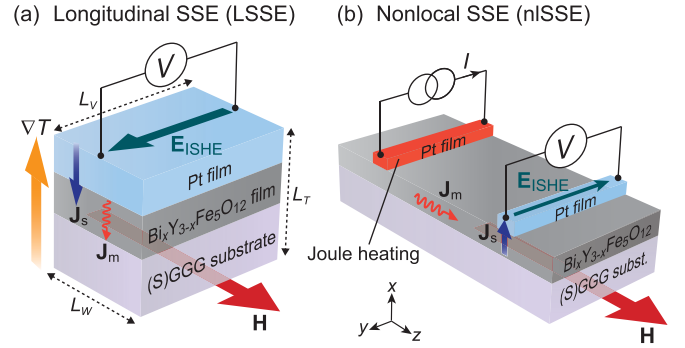


FIG. 2. Schematic illustrations of the (a) LSSE and (b) nlSSE in Pt/ $\text{Bi}_x\text{Y}_{3-x}\text{IG}/(\text{S})\text{GGG}$, where \mathbf{E}_{ISHE} , \mathbf{J}_s , \mathbf{J}_m , ∇T , and \mathbf{H} represent the electric field induced by the ISHE, spin-current injection, magnon-flow direction in $\text{Bi}_x\text{Y}_{3-x}\text{IG}$, temperature gradient, and external magnetic field, respectively. $\mathbf{J}_m \perp \mathbf{H}$ for the LSSE, while $\mathbf{J}_m \parallel \mathbf{H}$ for the nlSSE.

$\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films were cleaned with so-called Piranha etch solution (a mixture of H_2SO_4 and H_2O_2 at a ratio of 1:1) and also with acetone in an ultrasonic bath before Pt deposition. The film composition was characterized by an electron probe micro analyzer in the wavelength dispersive mode. The crystallinity and lattice parameters of the $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films were analyzed by means of high-resolution x-ray diffraction (XRD) and transmission electron microscopy (TEM), from which the film thickness was evaluated. The magnetic properties were measured using VSM (vibrating sample magnetometer) option of Physical Property Measurement System (PPMS), Quantum Design Inc.

For LSSE measurements, we fabricated 5-nm-thick Pt films on the $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films to electrically detect the SSE voltage based on the inverse spin-Hall effect (ISHE) [44–48] [see Fig. 2(a)]. Here, the Pt films were prepared by ex-situ d.c. magnetron sputtering in a 10^{-1} Pa Ar atmosphere [49]. The samples were cut into a rectangular shape whose length L_V , width L_W , and thickness L_T are, respectively, ~ 4 mm, 2 mm, and 0.5 (0.7) mm for the Pt/YIG/GGG and Pt/ $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}/\text{GGG}$ (Pt/ $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}/\text{SGGG}$) structures. To apply a temperature difference, ΔT , the sample was sandwiched between two sapphire plates; the temperature T_L of the bottom sapphire plate in contact with the (S)GGG substrate is varied in the range from 300 to 3 K, while the temperature T_H of the top sapphire plate placed on the Pt layer is increased by an attached heater [50,51]. A uniform magnetic field \mathbf{H} (with the magnitude H) was applied parallel to the film interface [along the z direction in Fig. 2(a)] by a superconducting solenoid magnet. We measured the d.c. electric voltage difference V between the ends of the Pt films. Hereafter, we plot the LSSE coefficient defined as $S \equiv (V/L_V)/(\Delta T/L_T)$.

For nonlocal SSE (nlSSE) measurements, we prepared nanofabricated Pt/ $\text{Bi}_x\text{Y}_{3-x}\text{IG}/\text{Pt}$ devices, where two electrically-separated Pt strips with the distance of $d = 8\ \mu\text{m}$ are formed on the $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films, as schematically shown in Fig. 2(b). Here, Pt strips were prepared by means of electron-beam lithography, followed by Pt sputtering and a lift-off process [52,53], whose dimensions are 200 μm length (l_V), 100 nm width, and 10 nm thickness. In our nlSSE setup,

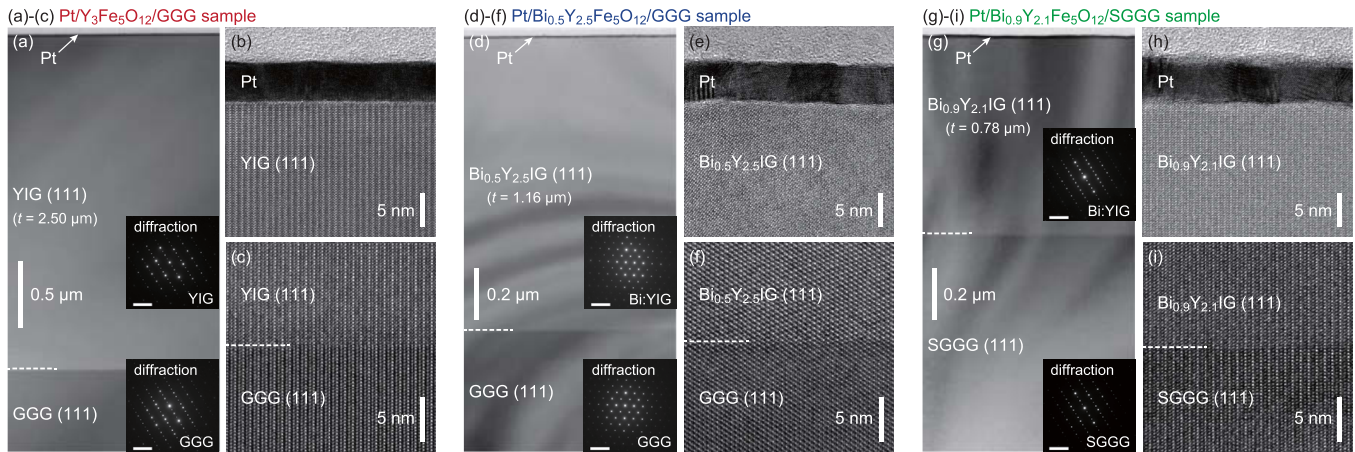


FIG. 3. Cross-sectional TEM images of the [(a)–(c)] Pt(5 nm)/YIG/GGG, [(d)–(f)] Pt(5 nm)/Bi_{0.5}Y_{2.5}IG/GGG, and [(g)–(i)] Pt(5 nm)/Bi_{0.9}Y_{2.1}IG/SGGG samples. The TEM images shown in (a), (d), and (g) provide the overall sample cross sections. The insets in (a), (d), and (g) show the selected area diffraction patterns, showing good agreement of the patterns between the (a) YIG and GGG, (d) Bi_{0.5}Y_{2.5}IG and GGG, and (g) Bi_{0.9}Y_{2.1}IG and SGGG layers. The white scale bars in the diffraction patterns represent 5 nm⁻¹. Note that the cross sections of the TEM lamellae specimens were arbitrary crystalline planes (i.e., the incident electron beam was not directed along specific crystalline axes), so that the different diffraction patterns appear for each sample. High resolution TEM images of the (b) Pt/YIG, (c) YIG/GGG, (e) Pt/Bi_{0.5}Y_{2.5}IG, (f) Bi_{0.5}Y_{2.5}IG/GGG, (h) Pt/Bi_{0.9}Y_{2.1}IG, and (i) Bi_{0.9}Y_{2.1}IG/SGGG interfaces.

the Joule heating of an applied charge current (I) to the one Pt strip generates a magnon spin current in Bi _{x} Y_{3- x} IG. When some of the magnons reach the other Pt strip, they are converted into a conduction-electron spin current and subsequently detected as an ISHE voltage [see Fig. 2(b)]. We measured the nISSE using a lock-in detection technique [52–55]; an a.c. charge current $I = \sqrt{2}I_{\text{rms}} \sin(\omega t)$ at the frequency $\omega/2\pi$ of 13.423 Hz is applied to the injector Pt strip and a resultant second harmonic nonlocal voltage across the detector Pt strip is measured. Henceforth, we plot the nISSE coefficient normalized by the applied current squared and the Pt length: $\tilde{S} \equiv V/(I_{\text{rms}}^2 l_V)$ [56–58]. All the data are obtained in the linear regime [52,53], where $V \propto I_{\text{rms}}^2$.

III. RESULTS AND DISCUSSION

A. Structural characterization

Figure 3 shows the cross-sectional TEM images of the Pt/YIG/GGG, Pt/Bi_{0.5}Y_{2.5}IG/GGG, and Pt/Bi_{0.9}Y_{2.1}IG/SGGG samples. The overall sample cross-section images [Figs. 3(a), 3(d), and 3(g)] show high uniformity and flatness of the films. The high-resolution images at the Bi _{x} Y_{3- x} IG/(S)GGG(111) interfaces [Figs. 3(c), 3(f), and 3(i)] reveal atomically smooth and epitaxial interfaces of our garnet films. Neither macroscopic defects nor misalignment in the lattice planes were observed in TEM images. As shown in the insets of Figs. 3(a), 3(d), and 3(g), the diffraction patterns of the Bi _{x} Y_{3- x} IG films coincide with those of the (S)GGG substrates, indicating that the Bi _{x} Y_{3- x} IG layers are grown as single-crystalline films with the [111] orientation in the out-of-plane direction. We also confirmed clear interfaces between the (polycrystalline) Pt-film and Bi _{x} Y_{3- x} IG-film contacts [see Figs. 3(b), 3(e), and 3(h)], which allows us to investigate interfacial spin-current transport in these sample systems.

Figures 4(a), 4(b), and 4(c) display the 2θ - ω XRD patterns of the Bi _{x} Y_{3- x} IG films around the (444) Bragg peaks from the GGG ($2\theta = 51.058^\circ$) or SGGG ($2\theta = 50.546^\circ$) substrates. For all the samples, we observed the (444) diffraction peaks from the Bi _{x} Y_{3- x} IG layer, confirming the out-of-plane orientation of Bi _{x} Y_{3- x} IG[111]/(S)GGG[111], consistent with the TEM results shown in Fig. 3. The XRD rocking curves exhibit full width at half maximum (FWHM) values of 0.0085°, 0.0093°, and 0.0103° for the YIG, Bi_{0.5}Y_{2.5}IG, and Bi_{0.9}Y_{2.1}IG, respectively, which are close to the values for the substrates (0.0078° for GGG and 0.0099° for SGGG), ensuring high crystalline quality of our films.

To further characterize structural properties of the films, we performed reciprocal space mapping (RSM) [59–61]. Figures 4(d), 4(e), 4(f) show the RSM data around the (664) diffraction for the YIG/GGG, Bi_{0.5}Y_{2.5}IG/GGG, and Bi_{0.9}Y_{2.1}IG/SGGG samples, respectively. For all the samples, the asymmetric diffraction peaks lie on one vertical line, indicating that the films have a pseudomorphic structure with the in-plane lattice constant identical to that of the substrate along the $\langle 11\bar{2} \rangle$ direction [59–61]. The pseudomorphic structure was confirmed also along the $\langle 1\bar{1}0 \rangle$ direction by the RSM data for the (486) diffraction [see Figs. 4(g), 4(h), and 4(i)].

From the XRD and RSM results, the out-of-plane (in-plane) lattice constant was estimated to be 12.376 Å (12.383 Å) for the YIG film on GGG, 12.456 Å (12.383 Å) for the Bi_{0.5}Y_{2.5}IG film on GGG, and 12.446 Å (12.508 Å) for the Bi_{0.9}Y_{2.1}IG film on SGGG. This result shows that the YIG film grown on GGG is nearly free from strain, while the Bi_{0.5}Y_{2.5}IG (Bi_{0.9}Y_{2.1}IG) film grown on GGG (SGGG) has compressive (tensile) epitaxial strain. In Table I, we show the in-plane biaxial strain ϵ_{\parallel} and out-of-plane uniaxial strain ϵ_{\perp} values evaluated according to Ref. [60]. The in-plane stress σ_{\parallel} values [60] are also calculated using the elastic coefficients C_{11} ($= \rho c_{\text{LA}}^2$; ρ : mass density) and C_{44} ($= \rho c_{\text{TA}}^2$) and the relation $C_{11} - C_{12} = 2C_{44}$ [62,63], which we list also in Table I.

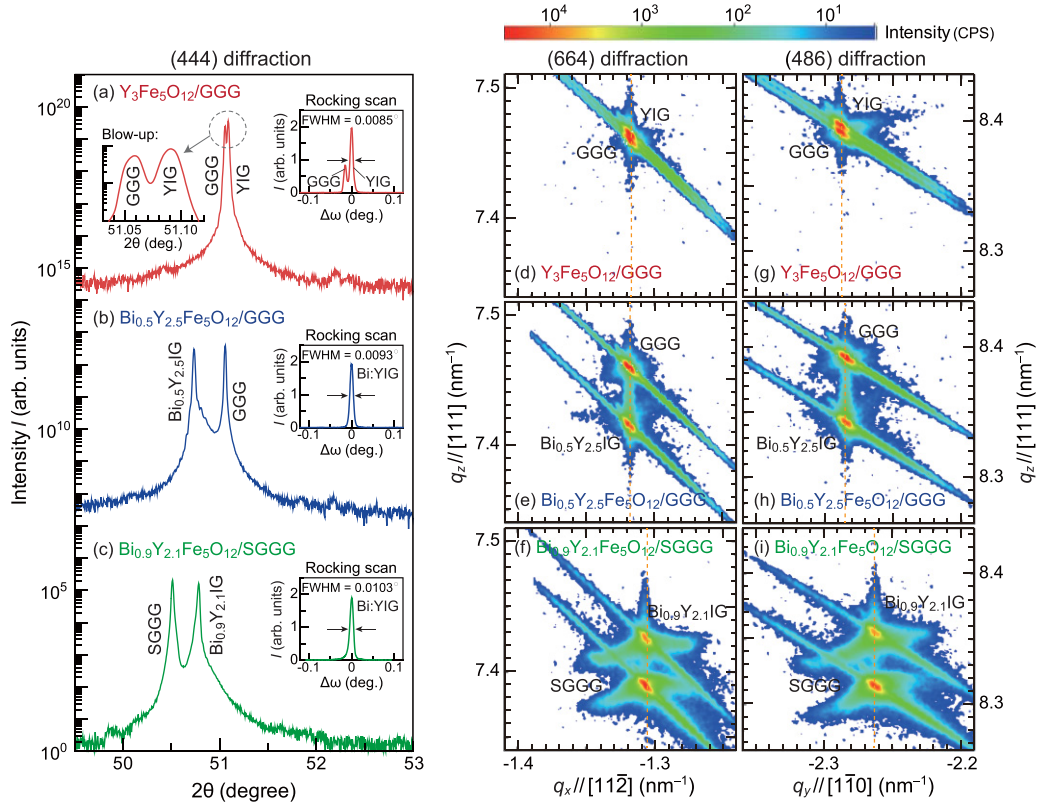


FIG. 4. (a)–(c) 2θ - ω XRD around the (444) diffraction peaks of the (a) YIG(2.50 μm)/GGG(111), (b) Bi_{0.5}Y_{2.5}IG(1.16 μm)/GGG(111), and (c) Bi_{0.9}Y_{2.1}IG(0.78 μm)/SGGG(111) samples. The insets in (a), (b), and (c) show the rocking-scan results around the (444) diffraction peaks of the (a) YIG, (b) Bi_{0.5}Y_{2.5}IG, and (c) Bi_{0.9}Y_{2.1}IG films, from which the FWHM values were estimated to be (a) 0.0085°, (b) 0.0093°, and (c) 0.0103°, respectively. (d), (e), (f) [(g), (h), (i)] RSMs around the (664) [(486)] diffraction peaks of the (d) [(g)] YIG/GGG, (e) [(h)] Bi_{0.5}Y_{2.5}IG/GGG, and (f) [(i)] Bi_{0.9}Y_{2.1}IG/SGGG samples. The RSMs show that the Bi_xY_{3-x}IG films have a coherently strained pseudomorphic structure on the (S)GGG substrates (see the orange dashed lines, which show that the diffraction peak positions for the Bi_xY_{3-x}IG film and substrate layers along the in-plane directions are the same for each sample [59,60]).

B. Magnetic characterization

In Figs. 5(a), 5(b), and 5(c), we show the magnetic field H dependence of the magnetization M of the (a) YIG, (b) Bi_{0.5}Y_{2.5}IG, and (c) Bi_{0.9}Y_{2.1}IG films, respectively. Here, the M - H curves are recorded at 300 K under the applied H parallel to the in-plane $[11\bar{2}]$ (solid lines) and to the out-of-plane $[111]$ (dashed lines) axes. The pure YIG film (nearly free from strain) has an in-plane easy axis due to the shape anisotropy and negligibly-small magnetocrystalline anisotropy [60,64,65]; the saturation field H_s value

for the in-plane $[11\bar{2}]$ field ($\mu_0 H_s^{\parallel} = 3$ mT) is much smaller than that for the out-of-plane $[111]$ field ($\mu_0 H_s^{\perp} = 0.22$ T) [see Fig. 5(a)]. The in-plane saturation field H_s^{\parallel} for the Bi_{0.5}Y_{2.5}IG (Bi_{0.9}Y_{2.1}IG) film is estimated to be $\mu_0 H_s^{\parallel} = 1$ mT (0.17 T), which is much smaller (slightly higher) than that for the out-of-plane $\mu_0 H_s^{\perp} = 0.23$ T (0.12 T) [see Figs. 5(b) and 5(c)]. The different H_s^{\parallel} v.s. H_s^{\perp} features can be interpreted in terms of the stress-induced anisotropy; for a material with a negative magnetostriction constant $\lambda_{111} < 0$ such as Bi_xY_{3-x}IG, the magnetic easy axis tends to lie in

TABLE I. Parameters for the (magneto)elastic properties of YIG, Bi_{0.5}Y_{2.5}IG, and Bi_{0.9}Y_{2.1}IG. $\epsilon_{\parallel(\perp)}$, σ_{\parallel} , $c_{\text{TA(LA)}}$, λ_{111} , and B_{\perp} denote the in-plane biaxial (out-of-plane uniaxial) strain, in-plane stress, TA(LA)-phonon sound velocity, magnetostriction constant, and magnetoelastic coupling constant, respectively. $c_{\text{TA(LA)}}$ values for the YIG film are adopted from Ref. [62]. λ_{111} values were evaluated from the relationship $\lambda_{111} = -2.819 \cdot (1 + 0.75x) \times 10^{-6}$ for Bi_xY_{3-x}IG films [4]. $c_{\text{TA(LA)}}$ values for Bi_{0.5}Y_{2.5}IG and Bi_{0.9}Y_{2.1}IG were measured by means of a time-resolved magneto-optical imaging method, followed by a Fourier transform process [15,17]. B_{\perp} values were calculated with the use of c_{TA} and λ_{111} (see also the main text).

Sample	ϵ_{\parallel} (%)	ϵ_{\perp} (%)	σ_{\parallel} (GPa)	c_{TA} (10 ³ m/s)	c_{LA} (10 ³ m/s)	λ_{111} (10 ⁻⁶)	$B_{\perp}/2\pi$ (THz)
Y ₃ Fe ₅ O ₁₂	0.032	-0.027	0.092	3.84	7.21	-2.82	1.87
Bi _{0.5} Y _{2.5} Fe ₅ O ₁₂	-0.31	0.28	-0.72	3.34	6.42	-3.88	2.05
Bi _{0.9} Y _{2.1} Fe ₅ O ₁₂	0.27	-0.24	0.63	3.30	6.28	-4.72	2.57

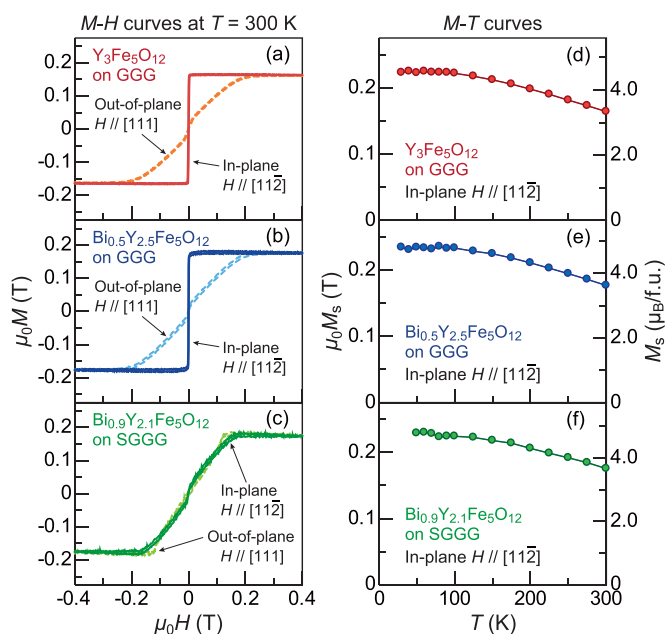


FIG. 5. [(a)–(c)] H dependence of M of the (a) YIG, (b) $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$, and (c) $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ films measured at $T = 300$ K, where the magnetic field \mathbf{H} was applied along the in-plane $[11\bar{2}]$ direction (solid lines) and along the out-of-plane $[111]$ direction (dashed lines). [(d)–(f)] T dependence of the saturation magnetization M_s of the (d) YIG, (e) $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$, and (f) $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ films for $\mathbf{H} \parallel [11\bar{2}]$. The M values of these $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films were extracted by subtracting the contributions from the paramagnetic (S)GGG substrates. Because of the large paramagnetic offset coming from the (S)GGG substrates at a low- T range, the M data were detectable at $T \geq 30$ K for the YIG and $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$ films and at $T \geq 50$ K for the $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ film.

(perpendicular to) the film plane for compressive (tensile) epitaxial strain [1,59,60,66,67]. Therefore, for the $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$ films on GGG with the compressive epitaxial strain, both the shape anisotropy and stress-induced anisotropy make the magnetic easy axis within an in-plane direction ($H_s^{\parallel} \ll H_s^{\perp}$). In contrast, for the $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ film on SGGG with the tensile epitaxial strain, the stress-induced out-of-plane anisotropy appears and prevails against the shape anisotropy, which renders the magnetic easy axis perpendicular to the film plane ($H_s^{\parallel} > H_s^{\perp}$) [4]. The saturation magnetization M_s values were estimated to be $3.34 \mu_B$ (0.164 T), $3.62 \mu_B$ (0.177 T), and $3.66 \mu_B$ (0.175 T) at 300 K, for the YIG, $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$, and $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ films, respectively, in units of the Bohr magneton (Tesla) (see Fig. 5). The M_s value increases by the Bi substitution, which is consistent with the previous reports [1,2]. Figures 5(d), 5(e), and 5(f) show the T dependence of M_s of the (d) YIG, (e) $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$, and (f) $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ films, respectively, for the H direction parallel to the in-plane $[11\bar{2}]$ axis. For all the samples, the M_s value increases with decreasing T and approaches to $\sim 5 \mu_B$ at the lowest T , showing good agreement with the theoretical and previous experimental results [1,2,68,69].

C. Evaluation of sound velocity and magnetoelastic coupling

To determine the sound velocities of the $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films, we employed a pump-and-probe magneto-optical imaging

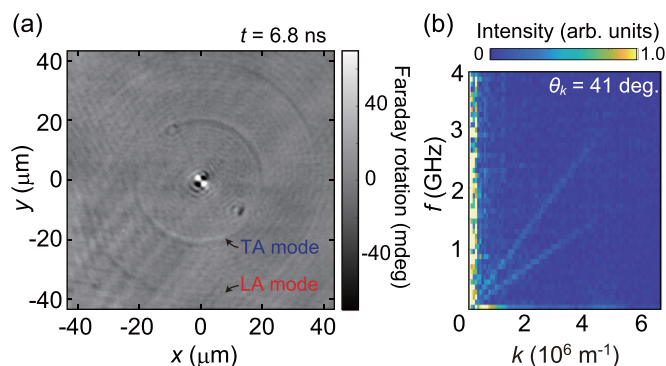


FIG. 6. (a) A time-resolved magneto-optical image for the $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ sample obtained at the time delay of 6.8 ns. The sample is magnetized slightly along the x direction by the in-plane external field of $\mu_0 H = 2.8$ mT, while the out-of-plane component of the magnetization distribution is measured through the Faraday effect of the probe pulse [15,17]. (b) The phonon dispersion relations for the $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ sample reconstructed by Fourier transforming the propagating waveform shown in (a) for $\theta_k = 41^\circ$, where θ_k denotes the angle between the wavevector \mathbf{k} and $\mathbf{H} \parallel \hat{x}$.

method combined with a Fourier transform (FT) process (see Refs. [15] and [17] for the details of the optical system in this experiment). When a pumped laser pulse with a spot radius of $\sim 2.5 \mu\text{m}$ is irradiated to the sample, elastic waves are excited and propagate radially from the excitation point. Via the magnetoelastic interaction, the elastic waves tilt the magnetic moments from an in-plane to out-of-plane direction that can be detected as a change of the Faraday rotation angle of the probe pulse [15,17]. Figure 6(a) shows a snapshot of the temporal evolution of out-of-plane magnetization distribution of the $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ sample obtained at the time delay of $t = 6.8$ ns between the pump and probe pulses. Two ring structures are visible; the inner (outer) ring is attributed to the coupled TA (LA) and spin waves having a smaller (larger) velocity. By Fourier transforming the observed propagating waveform with respect to the time and spatial coordinates, we obtain coupled phonon and spin waves dispersion relations. As shown in Fig. 6(b), two k -linear dispersions with different slopes are discerned; the dispersion showing the lower (larger) slope is assigned to the TA (LA) phonon branch, from which we can estimate the sound velocities of each phonon mode. Measurements were performed also for the $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$ sample. We note that the crossings between phonon and spin wave branches are inaccessible, since their positions in the momentum direction are large and therefore out of range in the present experiment.

Significant decreases in the sound velocities for the Bi:YIG films are indeed confirmed through the above measurements and analysis. As shown in Table I, the TA- and LA-phonon sound velocities of the $\text{Bi}_{0.5}\text{Y}_{2.5}\text{IG}$ films were determined to be $c_{\text{TA}} = 3.34 \times 10^3$ m/s and $c_{\text{LA}} = 6.42 \times 10^3$ m/s, giving values smaller than those of YIG. For the $\text{Bi}_{0.9}\text{Y}_{2.1}\text{IG}$ film, $c_{\text{TA}} = 3.30 \times 10^3$ m/s and $c_{\text{LA}} = 6.28 \times 10^3$ m/s, showing the smallest values among our samples.

Using the magnetostriction constant λ_{111} values for μm -thick $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films [$\lambda_{111} = -2.819 \cdot (1 + 0.75x) \times 10^{-6}$] shown in Ref. [4] and the sound velocities

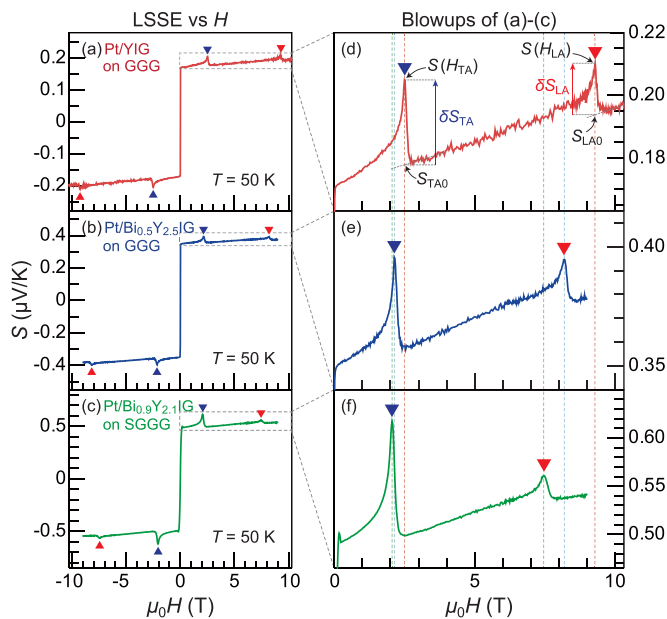


FIG. 7. [(a)–(c)] H dependence of the LSSE coefficient S of the (a) Pt/YIG, (b) Pt/Bi_{0.5}Y_{2.5}IG, and (c) Pt/Bi_{0.9}Y_{2.1}IG samples at $T = 50$ K. [(d)–(f)] Corresponding magnified views of $S(H)$ around the anomaly fields. The S peaks at H_{TA} and H_{LA} are marked by blue and red filled triangles, respectively. The light-red, light-blue, light-green dashed lines represent the peak positions for the Pt/YIG, Pt/Bi_{0.5}Y_{2.5}IG, and Pt/Bi_{0.9}Y_{2.1}IG samples, respectively. $S(H_i)$ and S_{i0} ($i = TA$ or LA) shown in (d) represent the intensity of S and extrapolated background S at the peak position H_i , respectively, from which the anomaly intensity is evaluated as $\delta S_i \equiv S(H_i) - S_{i0}$. The intensity of magnon-polaron anomalies relative to the background S_0 is defined as $\delta S_i/S_0 \equiv [S(H_i) - S_{i0}]/S_0$.

c_{TA} ($= \sqrt{C_{44}/\rho}$) in Table I, the magnetoelastic coupling constant $B_{\perp}/2\pi$ [$= -3 a_{\text{Bi:YIG}}^3 \lambda_{111} C_{44}/(2\pi \hbar)$] ($a_{\text{Bi:YIG}}$: lattice constant) [62,63,70] was estimated to be 1.87, 2.05, and 2.57 THz for the YIG, Bi_{0.5}Y_{2.5}IG, and Bi_{0.9}Y_{2.1}IG films, respectively. The B_{\perp} value monotonically increases by increasing the Bi amount x , which may be attributed to the enhanced spin-orbit coupling induced by the substitution of Bi³⁺ ions [3].

D. Magnon-polaron features in LSSE

Now, we present the experimental results on the LSSE in the Pt/Bi _{x} Y_{3- x} IG samples. Figure 7(a) shows the H dependence of the LSSE coefficient S of the Pt/YIG sample measured at $T = 50$ K. In this conventional system, we observed a clear SSE signal as well as two magnon-polaron-induced spikes on top of the almost flat S background signal [see the blue and red filled triangles in Fig. 7(a) and magnified view shown in Fig. 7(d)]. The peak at the lower (higher) H appears at $\mu_0 H_{TA} = 2.51$ T ($\mu_0 H_{LA} = 9.27$ T), consistent with Ref. [13]. The H_{TA} and H_{LA} values coincide with the H intensities at which the magnon dispersion curve touches the TA- and LA-phonon dispersion curves of YIG, respectively, being the conditions that the magnon and phonon modes can be coupled over the largest volume in momentum space, so the magnon-polaron effects are maximized [13,26].

Next, let us focus on the effect of Bi substitution on the LSSE and magnon-polaron features. Figures 7(b) and 7(c) show the H dependence of S of the Pt/Bi_{0.5}Y_{2.5}IG and Pt/Bi_{0.9}Y_{2.1}IG samples, respectively. We observed SSE signals in these samples, of which the amplitudes are comparable to that for the Pt/YIG shown in Fig. 7(a). The peak features show up also in these Pt/Bi:YIG systems as marked by the blue and red filled triangles in Figs. 7(b) and 7(c). Interestingly, the peak positions shift toward lower H values by increasing the amount of Bi substitution. The peak positions at the lower and higher H for the Pt/Bi_{0.5}Y_{2.5}IG sample are estimated to be $\mu_0 H_{TA} = 2.15$ T and $\mu_0 H_{LA} = 8.18$ T, respectively. For the Pt/Bi_{0.9}Y_{2.1}IG, the peak fields are further decreased: $\mu_0 H_{TA} = 2.06$ T and $\mu_0 H_{LA} = 7.47$ T.

The observed peak shifts are explained in terms of the reduction of the sound velocities and resultant touching fields of Bi:YIG films. The touching field can be evaluated quantitatively by solving the combined equation for the magnon (ω_{mag}) and phonon (ω_{ph}) dispersion relations and their group velocities,

$$\omega_{\text{mag}} = \omega_{\text{ph}}, \quad \partial \omega_{\text{mag}}/\partial k = \partial \omega_{\text{ph}}/\partial k, \quad (1)$$

where the magnon dispersion relation, disregarding the dipolar interaction ($M_s \ll H_{TA,LA}$), reads $\omega_{\text{mag}} = D_{\text{ex}} k^2 + \gamma \mu_0 H$, while the phonon dispersions are given by $\omega_{\text{ph}} = c_{TA,LA} k$. Here, D_{ex} , k , and γ represent the exchange stiffness, wavenumber, and gyromagnetic ratio, respectively. Calculation of Eq. (1) leads to

$$\mu_0 H_{TA(LA)} = \frac{c_{TA(LA)}^2}{4D_{\text{ex}}\gamma}, \quad (2)$$

which shows that the touching field $H_{TA(LA)}$ scales quadratically with the sound velocity $c_{TA(LA)}$, and so does the anomaly field in the SSE. In fact, the observed peak fields $H_{TA,LA}$ are well reproduced by the sound velocities $c_{TA(LA)}$ shown in Table I and reasonable D_{ex} values of 8.0×10^{-6} , 7.1×10^{-6} , and 7.3×10^{-6} m²/s for the YIG, Bi_{0.5}Y_{2.5}IG, and Bi_{0.9}Y_{2.1}IG, respectively. The frequency value for the touching point $\omega_{\text{MTA}}/2\pi$ ($\omega_{\text{MLA}}/2\pi$) between the magnon and TA(LA)-phonon branches at H_{TA} (H_{LA}), depicted as green filled circles in Fig. 1(b), can be evaluated as $\omega_{\text{MTA}}/2\pi = 0.15$ THz ($\omega_{\text{MLA}}/2\pi = 0.52$ THz) for the YIG, 0.13 THz (0.47 THz) for the Bi_{0.5}Y_{2.5}IG, and 0.12 THz (0.43 THz) for the Bi_{0.9}Y_{2.1}IG.

We carried out systematic measurements of the temperature dependence of the LSSE in these Pt/Bi _{x} Y_{3- x} IG samples. Figures 8(a), 8(d), and 8(g) show the S - H curves for the various average sample temperature T_{avg} (from 300 to 3 K) for the Pt/YIG, Pt/Bi_{0.5}Y_{2.5}IG, and Pt/Bi_{0.9}Y_{2.1}IG samples, respectively. In all the samples, the overall T dependencies of S agree with those for the Pt/YIG-film samples reported in Refs. [50] and [71]; when the sample temperature is decreased from 300 K, the magnitude of S increases and shows a broad peak at around $T = 150$ K [see Figs. 9(a)–9(c)]. On decreasing T further, the S signal begins to decrease and eventually goes to zero.

We now discuss the T dependence of the magnon-polaron features in the LSSE. First, let us focus on the results at a low- T range below 50 K, at which characteris-

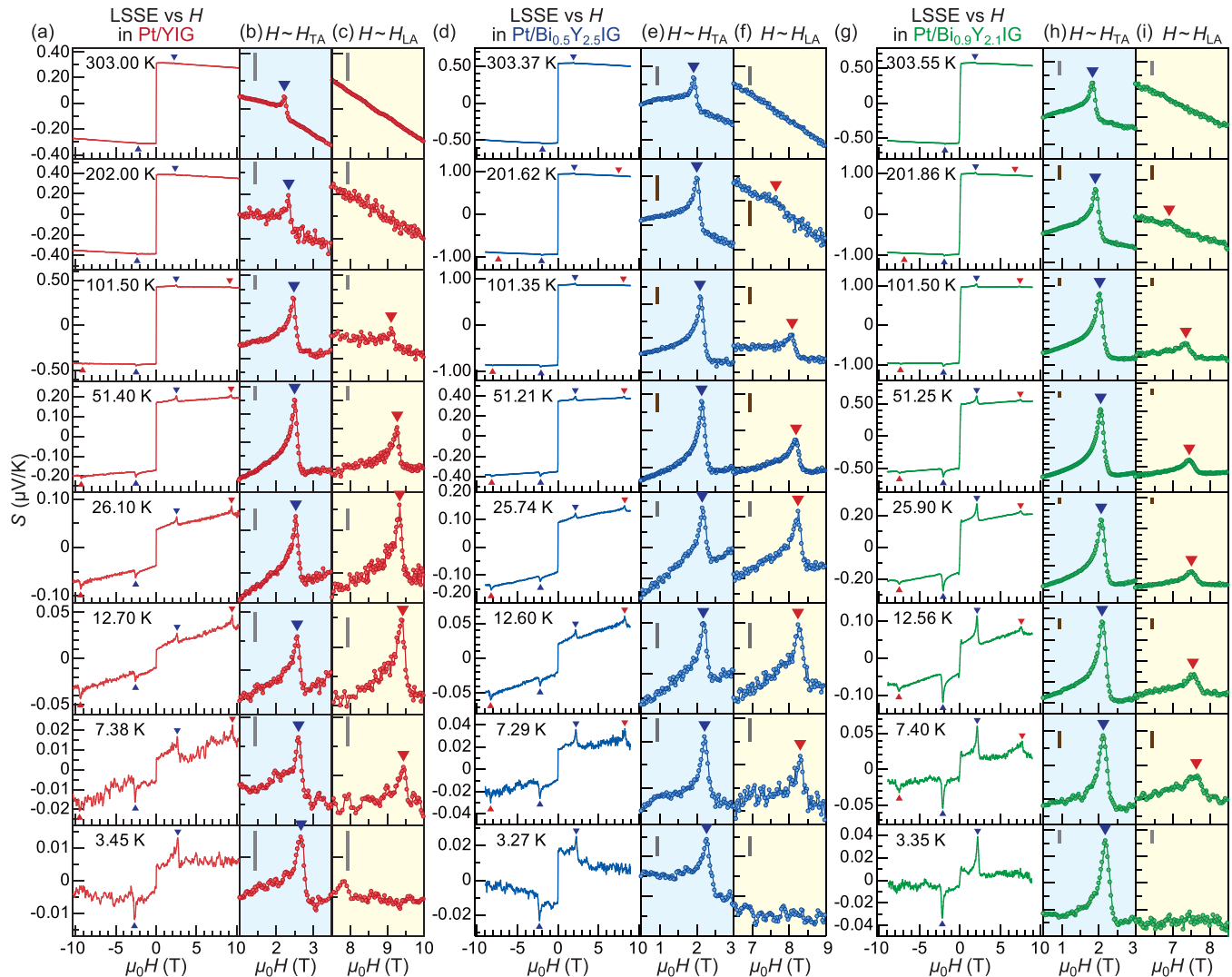


FIG. 8. (a), (d), (g) $S(H)$ of the (a) Pt/YIG, (d) Pt/Bi_{0.5}Y_{2.5}IG, and (g) Pt/Bi_{0.9}Y_{2.1}IG samples for various values of the sample temperature $T_{\text{avg}} = (T_H + T_L)/2$. (b), (e), (h) [(c), (f), (i)] Blowups of S versus H around H_{TA} (H_{LA}) for the (b) [(c)] Pt/YIG, (e) [(f)] Pt/Bi_{0.5}Y_{2.5}IG, and (h) [(i)] Pt/Bi_{0.9}Y_{2.1}IG samples. The gray and brown scale bars represent 0.005 $\mu\text{V/K}$ and 0.010 $\mu\text{V/K}$, respectively. The S peaks at H_{TA} and H_{LA} are marked by blue and red filled triangles, respectively. The global S signal below ~ 30 K increases with increasing H , which may be attributed to the additional spin-current generation by the paramagnetic (S)GGG substrate [13,72–76].

tic T response is expected to occur due to the competition between the thermal energy $k_B T$ and the magnon-polaron's touching energy $\hbar\omega_{\text{MTA,MLA}}$, which governs the T dependence of thermal occupation of magnon-polaron modes [13]. Figures 9(d), 9(e), and 9(f) represent the T dependence of the magnon-polaron peak amplitude $\delta S_{\text{TA,LA}} \equiv S(H_{\text{TA,LA}}) - S_{\text{TA0(LA0)}}$ for the Pt/YIG, Pt/Bi_{0.5}Y_{2.5}IG, and Pt/Bi_{0.9}Y_{2.1}IG samples, respectively, where $S_{\text{TA0(LA0)}}$ is the extrapolated background S at the peak position $H_{\text{TA(LA)}}$ [see also the definition shown in Fig. 7(d) and its caption]. The peak intensities $\delta S_{\text{TA,LA}}$ at $H = H_{\text{TA}}$ and H_{LA} exhibit different T dependencies. We further introduce and hereafter discuss the quantity $\delta S_{\text{TA(LA)}}/S_{\text{TA0(LA0)}}$, where $\delta S_{\text{TA(LA)}}$ is normalized by the background $S_{\text{TA0(LA0)}}$, which allows us to evaluate the T dependence of the magnon-polaron SSE coefficient relative to the background magnonic one at $H = H_{\text{TA(LA)}}$ and also to exclude common T -dependent factors in the magnon-polaron and magnonic SSE voltages such as the interfacial spin con-

ductance, electric resistance, and spin diffusion length in the Pt film [52,77,78]. As shown in Figs. 9(g)–9(i), $\delta S_{\text{TA}}/S_{\text{TA0}}$ monotonically increases with decreasing T and takes a maximum value at the lowest T for all the samples (see also the S - H curves shown in Fig. 8). In contrast, $\delta S_{\text{LA}}/S_{\text{LA0}}$ is gradually suppressed below $T^* \sim 12$ K (~ 7 K) for the Pt/YIG sample (Pt/Bi_{0.5}Y_{2.5}IG and Pt/Bi_{0.9}Y_{2.1}IG samples) [see Figs. 8 and 9(g)–9(i)]. At the lowest T , the S peak at H_{LA} is so small and almost indistinguishable from the background [see Figs. 8(c), 8(f), and 8(i)]. The T -dependent feature can be interpreted in terms of the difference in the frequency values of the branch touching point ω_{MTA} for $H = H_{\text{TA}}$ and ω_{MLA} for $H = H_{\text{LA}}$ [see Fig. 1(b)] [13,26]. Here, in the unit of temperature, the ω_{MLA} values approximately correspond to $T_{\text{MLA}} (= \hbar\omega_{\text{MLA}}/k_B) = 25$ K, 22 K, and 21 K for the YIG, Bi_{0.5}Y_{2.5}IG, and Bi_{0.9}Y_{2.1}IG, respectively, and they are more than three times greater than those of the ω_{MTA} values (7 K, 6 K, and 6 K, respectively). Therefore, for $T < T_{\text{MLA}}$, the excitation

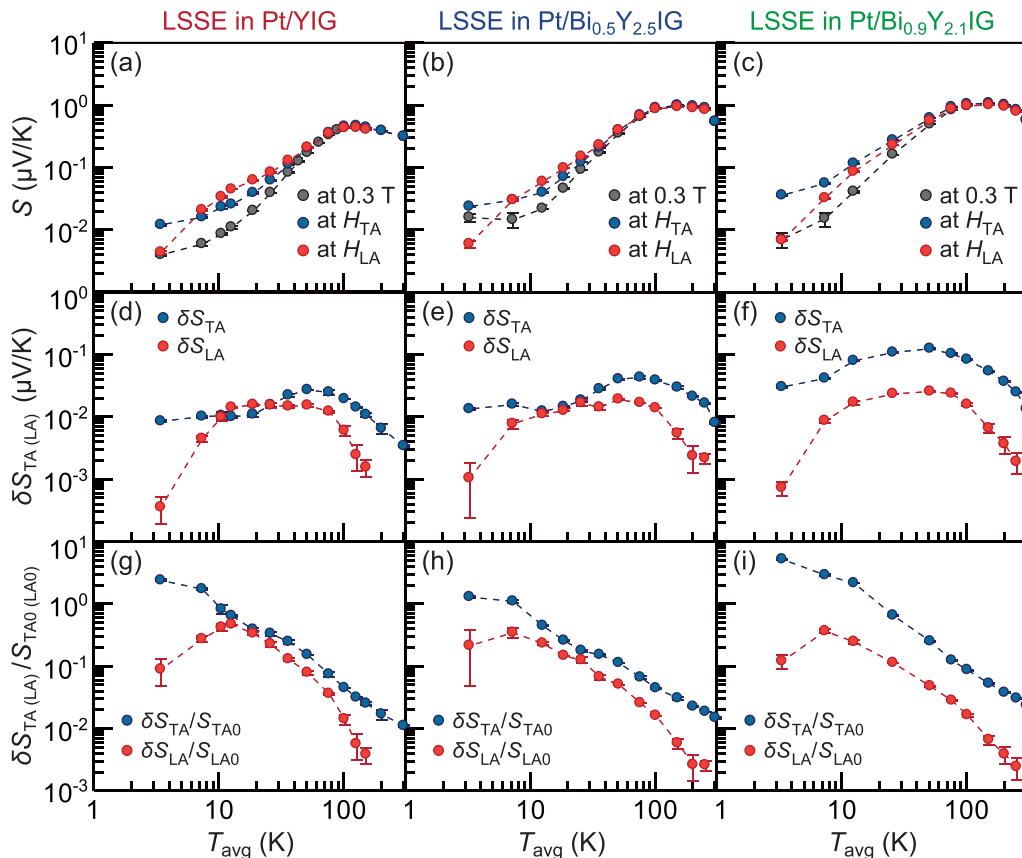


FIG. 9. (a), (b), (c) T_{avg} dependence of S at $\mu_0 H = 0.3 \text{ T}$, $\mu_0 H_{\text{TA}}$, and $\mu_0 H_{\text{LA}}$ for the (a) Pt/YIG, (b) Pt/Bi_{0.5}Y_{2.5}IG, and (c) Pt/Bi_{0.9}Y_{2.1}IG samples. (d), (e), (f) [(g), (h), (i)] T_{avg} dependence of the intensities of magnon-polaron peaks δS_{TA} and δS_{LA} [$\delta S_{\text{TA}}/S_{\text{TA0}}$ and $\delta S_{\text{LA}}/S_{\text{LA0}}$] for the (d) [(g)] Pt/YIG, (e) [(h)] Pt/Bi_{0.5}Y_{2.5}IG, and (f) [(i)] Pt/Bi_{0.9}Y_{2.1}IG samples. The definitions of δS_{TA} , δS_{LA} , $\delta S_{\text{TA}}/S_{\text{TA0}}$, and $\delta S_{\text{LA}}/S_{\text{LA0}}$ are shown in the caption of Fig. 7.

of magnon polarons at the touching frequency $\omega \sim \omega_{\text{MLA}}$ is rapidly suppressed, which leads to the disappearance of the S peak at H_{LA} at the lowest T . Furthermore, the reduction of ω_{MLA} values for Bi_{0.5}Y_{2.5}IG and Bi_{0.9}Y_{2.1}IG, by a value of $\sim 5 \text{ K}$, compared to that for the YIG may be responsible for the difference of the onset T^* values between the Pt/Bi:YIG ($T^* \sim 7 \text{ K}$) and Pt/YIG ($T^* \sim 12 \text{ K}$), at which $\delta S_{\text{LA}}/S_{\text{LA0}}$ starts to be suppressed with decreasing T .

Next, we move on to the results at a higher- T range ($50 \text{ K} \lesssim T_{\text{avg}} \lesssim 300 \text{ K}$). In this T range, we are also able to resolve small but finite peak structures at $H = H_{\text{TA}}$ for all the samples, while, at H_{LA} , the peak structures were detectable below $\sim 150 \text{ K}$ ($\sim 250 \text{ K}$) for the Pt/YIG (Pt/Bi:YIG) sample (see Figs. 8 and 9). As shown in Figs. 9(g)–9(i), the peak amplitudes at H_{TA} and H_{LA} relative to the background SSE signals, $\delta S_{\text{TA}}/S_{\text{TA0}}$ and $\delta S_{\text{LA}}/S_{\text{LA0}}$, monotonically decrease with increasing T , which may be due to the appearance of large contributions of thermally populated (uncoupled) magnons with the energy of $k_{\text{B}}T > \hbar\omega_{\text{MTA,MLA}}$ to the background SSE signal. Furthermore, the observed structures with only peak shapes (no dip shapes) suggest that the scattering rate of magnons τ_{mag}^{-1} with the frequencies relevant to the hybridization (i.e., $\omega \sim \omega_{\text{MTA,MLA}}$) are always larger than that of phonons τ_{ph}^{-1} at all the T range: $\tau_{\text{mag}}^{-1} > \tau_{\text{ph}}^{-1}$ [13,26]. We also found that, in this T range, as the

temperature is increased, the $\delta S_{\text{LA}}/S_{\text{LA0}}$ value decreases much faster than the $\delta S_{\text{TA}}/S_{\text{TA0}}$ value for all the samples [see Figs. 9(g)–9(i)]. We note that with increasing T , dominant scattering sources for magnons and phonons may change from disorders (impurities) to inelastic magnon-magnon and phonon-phonon scatterings due to their increased thermal population [36,77,79–81]. A detailed knowledge on the T and k dependencies of such scattering events for the Bi _{x} Y_{3- x} IG films will thus be important to fully understand the observed behavior.

Finally, we compare the amplitudes of the peak structures in the LSSE in the Pt/Bi _{x} Y_{3- x} IG samples. The magnon-polaron contribution $\delta S_{\text{TA}}/S_{\text{TA0}}$ at $H = H_{\text{TA}}$ for the Pt/YIG, Pt/Bi_{0.5}Y_{2.5}IG, and Pt/Bi_{0.9}Y_{2.1}IG samples at the lowest $T \sim 3 \text{ K}$ ($T = 50 \text{ K}$) was evaluated as 2.4, 1.3, and 5.2 (0.15, 0.11, and 0.25), respectively. This nonmonotonic dependence with respect to the Bi amount is not simply explained by the magnetoelastic coupling constant B_{\perp} , since it monotonically increases by increasing the Bi substitution (see Table I). Although larger B_{\perp} values should be beneficial for the magnon-polaron formation and resultant SSE anomalies, our results suggest that the scattering ratio parameterized by $\eta = \tau_{\text{mag}}^{-1}/\tau_{\text{ph}}^{-1}$ may play a rather important role for the comparison of the magnon-polaron peak intensities between each sample. Furthermore, in the LSSE results, we found that the

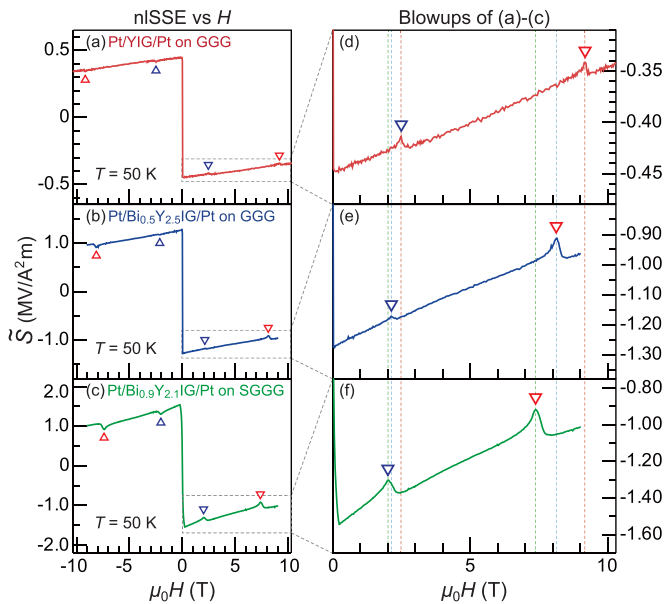


FIG. 10. [(a)–(c)] H dependence of the nISSE coefficient \tilde{S} of the (a) Pt/YIG/Pt, (b) Pt/Bi_{0.5}Y_{2.5}IG/Pt, and (c) Pt/Bi_{0.9}Y_{2.1}IG/Pt samples at $T = 50$ K. [(d)–(f)] Corresponding magnified views of $\tilde{S}(H)$ around the anomaly fields. The \tilde{S} dips at H_{TA} and H_{LA} are marked by blue and red unfilled triangles, respectively. The light-red, light-blue, light-green dashed lines represent the peak positions for the Pt/YIG, Pt/Bi_{0.5}Y_{2.5}IG, and Pt/Bi_{0.9}Y_{2.1}IG samples, respectively. The intensity of magnon-polaron anomalies is evaluated as $\delta\tilde{S}_i \equiv \tilde{S}(H_i) - \tilde{S}_{i0}$ ($i = TA$ or LA), where $\tilde{S}(H_i)$ and \tilde{S}_{i0} represent the intensity of \tilde{S} and extrapolated background \tilde{S} at the peak position H_i , respectively, similar to the case for the LSSE [see also the caption of Fig. 7(d)]. The intensity of the anomaly relative to the background \tilde{S}_{i0} is defined as $\delta\tilde{S}_i/\tilde{S}_{i0} \equiv [\tilde{S}(H_i) - \tilde{S}_{i0}]/\tilde{S}_{i0}$.

magnon-polaron contribution at $H = H_{TA}$ is greater than that at H_{LA} , i.e., $\delta S_{TA}/S_{TA0} > \delta S_{LA}/S_{LA0}$ for all the T range and all the samples [see Figs. 9(g)–9(i)]. This tendency is compared with that for the nISSE results in the next section.

E. Magnon-polaron features in nISSE

We now show the results on the nISSE in the Pt/Bi_xY_{3-x}IG/Pt systems. Figures 10(a), 10(b), and 10(c) show the H dependence of the nISSE coefficient \tilde{S} of the Pt/YIG/Pt, Pt/Bi_{0.5}Y_{2.5}IG/Pt, and Pt/Bi_{0.9}Y_{2.1}IG/Pt samples measured at $T = 50$ K, respectively. The overall signal appears with a sign opposite to that for the LSSE, indicating that thermally-excited magnons accumulate beneath the detector Pt strip. The observed sign is indeed a characteristic in nISSEs for μ m-thick garnet films with a long injector-detector separation distance d ($d = 8 \mu$ m in the present case) [52,82,83]. As marked by the blue and red unfilled triangles in the nISSE signals in Fig. 10, magnon-polaron induced anomalies are also observed at the touching fields $H_{TA,LA}$, which shift toward lower H values by increasing the Bi amount [see also the magnified views for $H > 0$ shown in Figs. 10(d), 10(e), and 10(f)]. The $H_{TA(LA)}$ values are evaluated as $\mu_0 H_{TA} = 2.48$ T ($\mu_0 H_{LA} = 9.17$ T) for the Pt/YIG/Pt, 2.13 T (8.14 T) for the Pt/Bi_{0.5}Y_{2.5}IG/Pt, and 2.02 T (7.38 T) for the Pt/Bi_{0.9}Y_{2.1}IG/Pt, in good agreement

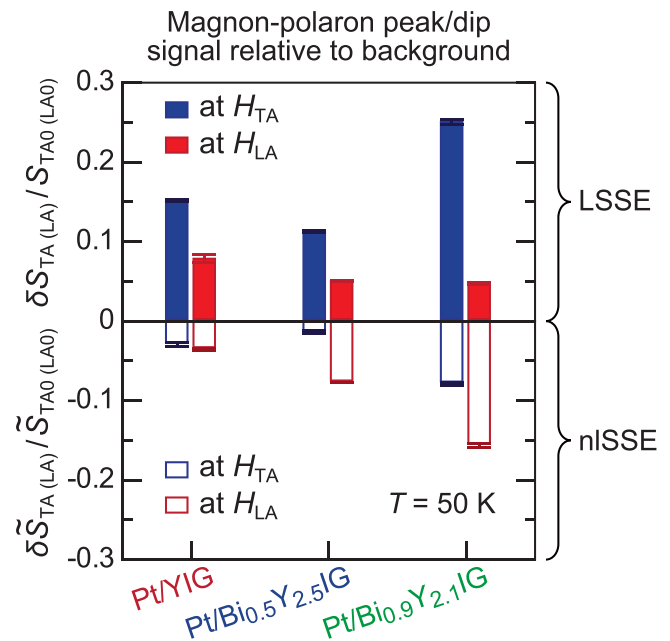


FIG. 11. Comparison between the intensities of magnon-polaron LSSE-peak and nISSE-dip relative to the background, $\delta S_i/S_{i0}$ for the LSSE and $\delta\tilde{S}_i/\tilde{S}_{i0}$ for the nISSE ($i = TA$ or LA), at the touching fields H_{TA} (blue bars) and H_{LA} (red bars) at $T = 50$ K for all the samples.

with those estimated from the LSSE results. For the nISSE, the magnon-polaron formation decreases rather than increases the \tilde{S} signal at the touching fields $H_{TA,LA}$, causing dip structures (note that the background nISSE signal is opposite in sign compared to that for the LSSE [52,53,84]). In Ref. [52], the dip feature is discussed in terms of the competition between a temperature-gradient induced magnon current, $\mathbf{J}_m^T = -(\zeta/T)\nabla T$, from the injector to detector and a diffusive backflow, $\mathbf{J}_m^\mu = -\sigma_m \nabla \mu_m$, induced by the gradient of the magnon chemical potential μ_m [$\mathbf{J}_m^T \parallel +\hat{\mathbf{z}}$ and $\mathbf{J}_m^\mu \parallel -\hat{\mathbf{z}}$ in the coordinate system shown in Fig. 2(b)], which are both increased by the hybridization with phonons when $\tau_{\text{mag}}^{-1} > \tau_{\text{ph}}^{-1}$ [26,52]. In particular, assuming that the magnon spin conductivity σ_m is more strongly increased than the bulk SSE coefficient ζ via the magnon-polaron formation, the increased backflow of magnons toward the injector direction causes the reduction of magnon accumulation beneath the Pt detector, which leads to the suppression of the nISSE at the touching fields [52].

Through a careful look at the magnon-polaron anomalies in Fig. 10, we notice that the intensity of the anomaly at $H = H_{LA}$ is larger than that at H_{TA} in the nISSE for all the samples: $|\delta\tilde{S}_{TA}/\tilde{S}_{TA0}| < |\delta\tilde{S}_{LA}/\tilde{S}_{LA0}|$. The $\delta\tilde{S}_{TA(LA)}/\tilde{S}_{TA0(LA0)}$ values at $H_{TA(LA)}$ are evaluated to be -0.029 (-0.035) for the Pt/YIG/Pt, -0.014 (-0.077) for the Pt/Bi_{0.5}Y_{2.5}IG/Pt, and -0.079 (-0.16) for the Pt/Bi_{0.9}Y_{2.1}IG/Pt at $T = 50$ K, which we summarize in Fig. 11. The observed relationship for the nISSE, $|\delta\tilde{S}_{TA}/\tilde{S}_{TA0}| < |\delta\tilde{S}_{LA}/\tilde{S}_{LA0}|$, holds in a wide T range for $T \lesssim 100$ K, and is opposite to that for the LSSE results, where $\delta S_{TA}/S_{TA0} > \delta S_{LA}/S_{LA0}$ (see Fig. 11 and also Figs. S1 and S2 in the Supplemental Material [85] for the T dependence of the nISSE). In the following, we discuss possible origins of the anisotropic feature. One possible scenario is a spectral nonuniform magnon current $J_m(\omega)$ [50,86] that

may vary depending on the experimental configurations; the frequency-resolved current $J_m(\omega_{\text{MTA}})$ at H_{TA} may be larger (smaller) than $J_m(\omega_{\text{MLA}})$ at H_{LA} for the longitudinal (non-local) configuration. In Refs. [50], [53], and [71], through high- H dependence experiments and its comparison with theory, a spectral nonuniformity in a magnon current is suggested to be present in both LSSEs and nLSSEs due to the frequency-dependent magnon scattering time [86,87]. Moreover, in the present case, as shown in Figs. 7 and 10, the high- H response of the background SSEs in the longitudinal and nonlocal configurations differs with each other. This may be a signature that the frequency-dependent $J_m(\omega)$ contribution is different between the LSSE and nLSSE. Another possibility would be the anisotropy in magnon and magnon-polaron dispersion relations due to the nature of the dipole and magnetoelastic interaction [11,26], which depend on the magnon propagation direction \mathbf{J}_m ($\|\mathbf{k}\| \nabla T$) relative to the external magnetic field \mathbf{H} . In the LSSE, \mathbf{J}_m is essentially perpendicular to \mathbf{H} , while in the nLSSE parallel to \mathbf{H} in the one-dimensional limit (see Fig. 2). Therefore, the LSSE and nLSSE may be affected by the anisotropic dispersion relations in a different manner. In fact, a solution of a Boltzmann transport equation shows that, for $\nabla T \perp \mathbf{H}$ (compatible with the LSSE), the magnon-polaron anomaly at $H = H_{\text{TA}}$ is larger than that at H_{LA} , while, for $\nabla T \parallel \mathbf{H}$ (compatible with the nLSSE in the one-dimensional limit), the anomaly at H_{TA} is smaller than that at H_{LA} [26]. This is consistent with our experimental results, and suggests that the anisotropic nature of magnon and magnon-polaron dispersion relations may affect the SSEs. It is worth mentioning that the profile of the temperature gradient created by the local Pt heater in the nonlocal configuration may be affected by size effects [52,82,83], such as the thickness of the magnetic layer and the size of the local heater. Future systematic measurements of the Bi:YIG thickness, heater size, and injector-detector separation distance (d) dependencies may provide useful information to understand possible roles of the size effects in the anisotropic feature of magnon-polaron signals. Besides, the magnon-polaron formation can also affect the magnon and phonon thermal conductivities [26], which may modify the temperature gradient and resultant spin-current intensity at the onset field of magnon-polaron formation. The process, which has not been considered in analysis so far, would be important for further quantitative argument on the magnon-polaron anomalies in SSEs.

IV. CONCLUSIONS

To summarize, we have prepared LPE-grown single crystalline $\text{Bi}_x\text{Y}_{3-x}\text{Fe}_5\text{O}_{12}$ ($\text{Bi}_x\text{Y}_{3-x}\text{IG}$; $x = 0, 0.5, \text{ and } 0.9$) films

with Pt contact and measured SSEs in both the longitudinal and nonlocal configurations. We observed two anomalous peaks in H -dependent LSSE signals at the onset fields $H_{\text{TA,LA}}$ for the magnon-polaron formation at which the magnon and TA,LA-phonon branches in $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ touch with each other. The anomaly fields shift toward lower values by increasing the Bi amount x , which is attributed to the reduction of the sound velocities of $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ by the Bi substitution. The observed peak behavior at the wide temperature range from 3 to 300 K suggests that the scattering rate of magnons τ_{mag}^{-1} is larger than that of phonons τ_{ph}^{-1} in all our sample systems [26]. In the nLSSE measurements, we found that the magnon-polaron formation suppresses rather than enhances the signal, causing dip structures at the touching fields $H_{\text{TA,LA}}$ for the $\text{Bi}_x\text{Y}_{3-x}\text{IG}$ films, consistent with the previous studies [52,53,84]. We further found that the intensities of magnon-polaron anomalies in the nLSSE appear to be different from those in the LSSE, although the anomaly fields are almost identical for both the configurations. We anticipate that our detailed temperature and magnetic-field dependent SSE results provide useful information to further elucidate the physics of magnon-polaron SSEs.

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- [1] P. Hansen, K. Witter, and W. Tolksdorf, Magnetic and magneto-optic properties of lead- and bismuth-substituted yttrium iron garnet films, *Phys. Rev. B* **27**, 6608 (1983).
 - [2] P. Hansen and J.-P. Krumme, Magnetic and magneto-optical properties of garnet films, *Thin Solid Films* **114**, 69 (1984).
 - [3] R. Kumar, B. Samantaray, and Z. Hossain, Ferromagnetic resonance studies of strain tuned Bi:YIG films, *J. Phys.: Condens. Matter* **31**, 435802 (2019).
 - [4] L. Soumah, N. Beaulieu, L. Qassym, C. Carr  tero, E. Jacquet, R. Lebourgeois, J. Ben Youssef, P. Bortolotti, V. Cros, and A. Anane, Ultra-low damping insulating magnetic thin films get perpendicular, *Nat. Commun.* **9**, 3355 (2018).
 - [5] M. Evelt, L. Soumah, A. B. Rinkevich, S. O. Demokritov, A. Anane, V. Cros, J. Ben Youssef, G. de Loubens, O. Klein, P. Bortolotti, and V. E. Demidov, Emission of Coherent

- Propagating Magnons by Insulator-Based Spin-Orbit-Torque Oscillators, *Phys. Rev. Appl.* **10**, 041002 (2018).
- [6] A. V. Chumak, P. Kabos, M. Wu, C. Abert, C. Adelman, A. O. Adeyeye, J. Åkerman, F. G. Aliev, A. Anane, A. Awad *et al.*, Advances in magnetics roadmap on spin-wave computing, *IEEE Trans. Magn.* **58**, 0800172 (2022).
- [7] P. X. Zhang, D. J. Lockwood, and H. J. Labbé, Magnon and acoustic-phonon light scattering from Bi-doped yttrium iron garnet, *Phys. Rev. B* **48**, 6099 (1993).
- [8] G. G. Siu, C. M. Lee, and Y. Liu, Magnons and acoustic phonons in $Y_{3-x}Bi_xFe_5O_{12}$, *Phys. Rev. B* **64**, 094421 (2001).
- [9] L. Dreher, M. Weiler, M. Pernpeintner, H. Huebl, R. Gross, M. S. Brandt, and S. T. B. Goennenwein, Surface acoustic wave driven ferromagnetic resonance in nickel thin films: Theory and experiment, *Phys. Rev. B* **86**, 134415 (2012).
- [10] N. Ogawa, W. Koshibae, A. J. Beekman, N. Nagaosa, M. Kubota, M. Kawasaki, and Y. Tokura, Photodrive of magnetic bubbles via magnetoelastic waves, *Proc. Natl. Acad. Sci. U.S.A.* **112**, 8977 (2015).
- [11] K. Shen and G. E. W. Bauer, Laser-Induced Spatiotemporal Dynamics of Magnetic Films, *Phys. Rev. Lett.* **115**, 197201 (2015).
- [12] S. C. Guerreiro and S. M. Rezende, Magnon-phonon interconversion in a dynamically reconfigurable magnetic material, *Phys. Rev. B* **92**, 214437 (2015).
- [13] T. Kikkawa, K. Shen, B. Flebus, R. A. Duine, K. Uchida, Z. Qiu, G. E. W. Bauer, and E. Saitoh, Magnon Polarons in the Spin Seebeck Effect, *Phys. Rev. Lett.* **117**, 207203 (2016).
- [14] D. A. Bozhko, P. Clausen, G. A. Melkov, V. S. L'vov, A. Pomyalov, V. I. Vasyuchka, A. V. Chumak, B. Hillebrands, and A. A. Serga, Bottleneck Accumulation of Hybrid Magnetoelastic Bosons, *Phys. Rev. Lett.* **118**, 237201 (2017).
- [15] Y. Hashimoto, S. Daimon, R. Iguchi, Y. Oikawa, K. Shen, K. Sato, D. Bossini, Y. Tabuchi, T. Satoh, B. Hillebrands *et al.*, All-optical observation and reconstruction of spin wave dispersion, *Nat. Commun.* **8**, 15859 (2017).
- [16] J. Holanda, D. S. Maior, A. Azevedo, and S. M. Rezende, Detecting the phonon spin in magnon-phonon conversion experiments, *Nat. Phys.* **14**, 500 (2018).
- [17] Y. Hashimoto, D. Bossini, T. H. Johansen, E. Saitoh, A. Kirilyuk, and T. Rasing, Frequency and wavenumber selective excitation of spin waves through coherent energy transfer from elastic waves, *Phys. Rev. B* **97**, 140404(R) (2018).
- [18] H. Hayashi and K. Ando, Spin Pumping Driven by Magnon Polarons, *Phys. Rev. Lett.* **121**, 237202 (2018).
- [19] R. Yahiro, T. Kikkawa, R. Ramos, K. Oyanagi, T. Hioki, S. Daimon, and E. Saitoh, Magnon polarons in the spin Peltier effect, *Phys. Rev. B* **101**, 024407 (2020).
- [20] K. An, A. N. Litvinenko, R. Kohno, A. A. Fuad, V. V. Naletov, L. Vila, U. Ebels, G. de Loubens, H. Hurdequint, N. Beaulieu, J. Ben Youssef, N. Vukadinovic, G. E. W. Bauer, A. N. Slavin, V. S. Tiberkevich, and O. Klein, Coherent long-range transfer of angular momentum between magnon Kittel modes by phonons, *Phys. Rev. B* **101**, 060407(R) (2020).
- [21] F. Godejohann, A. V. Scherbakov, S. M. Kukhtaruk, A. N. Poddubny, D. D. Yaremkevich, M. Wang, A. Nadzeyka, D. R. Yakovlev, A. W. Rushforth, A. V. Akimov, and M. Bayer, Magnon polaron formed by selectively coupled coherent magnon and phonon modes of a surface patterned ferromagnet, *Phys. Rev. B* **102**, 144438 (2020).
- [22] T. Hioki, Y. Hashimoto, and E. Saitoh, Bi-reflection of spin waves, *Commun. Phys.* **3**, 188 (2020).
- [23] P. Frey, D. A. Bozhko, V. S. L'vov, B. Hillebrands, and A. A. Serga, Double accumulation and anisotropic transport of magnetoelastic bosons in yttrium iron garnet films, *Phys. Rev. B* **104**, 014420 (2021).
- [24] J. Zhang, M. Chen, J. Chen, K. Yamamoto, H. Wang, M. Hamdi, Y. Sun, K. Wagner, W. He, Y. Zhang *et al.*, Long decay length of magnon-polarons in $BiFeO_3/La_{0.67}Sr_{0.33}MnO_3$ heterostructures, *Nat. Commun.* **12**, 7258 (2021).
- [25] R. Schlitz, L. Siegl, T. Sato, W. Yu, G. E. W. Bauer, H. Huebl, and S. T. B. Goennenwein, Magnetization dynamics affected by phonon pumping, *Phys. Rev. B* **106**, 014407 (2022).
- [26] B. Flebus, K. Shen, T. Kikkawa, K. Uchida, Z. Qiu, E. Saitoh, R. A. Duine, and G. E. W. Bauer, Magnon-polaron transport in magnetic insulators, *Phys. Rev. B* **95**, 144420 (2017).
- [27] K. Uchida, H. Adachi, T. Ota, H. Nakayama, S. Maekawa, and E. Saitoh, Observation of longitudinal spin-Seebeck effect in magnetic insulators, *Appl. Phys. Lett.* **97**, 172505 (2010).
- [28] K. Uchida, M. Ishida, T. Kikkawa, A. Kirihara, T. Murakami, and E. Saitoh, Longitudinal spin Seebeck effect: From fundamentals to applications, *J. Phys.: Condens. Matter* **26**, 343202 (2014); **26**, 389601 (2014).
- [29] K. Uchida, H. Adachi, T. Kikkawa, A. Kirihara, M. Ishida, S. Yorozu, S. Maekawa, and E. Saitoh, Thermoelectric generation based on spin Seebeck effects, *Proc. IEEE* **104**, 1946 (2016); **104**, 1499 (2016).
- [30] T. Kikkawa and E. Saitoh, Spin Seebeck effect: Sensitive probe for elementary excitation, spin correlation, transport, magnetic order, and domains in solids, [arXiv:2205.10509](https://arxiv.org/abs/2205.10509).
- [31] H. Wang, D. Hou, T. Kikkawa, R. Ramos, K. Shen, Z. Qiu, Y. Chen, M. Umeda, Y. Shiomi, X. Jin, and E. Saitoh, The bimodal distribution spin Seebeck effect enhancement in epitaxial $Ni_{0.65}Zn_{0.35}Al_{0.8}Fe_{1.2}O_4$ thin film, *Appl. Phys. Lett.* **112**, 142406 (2018).
- [32] R. Ramos, T. Hioki, Y. Hashimoto, T. Kikkawa, P. Frey, A. J. E. Kreil, V. I. Vasyuchka, A. A. Serga, B. Hillebrands, and E. Saitoh, Room temperature and low-field resonant enhancement of spin Seebeck effect in partially compensated magnets, *Nat. Commun.* **10**, 5162 (2019).
- [33] W. Xing, Y. Ma, Y. Yao, R. Cai, Y. Ji, R. Xiong, K. Shen, and W. Han, Facet-dependent magnon-polarons in epitaxial ferrimagnetic Fe_3O_4 thin films, *Phys. Rev. B* **102**, 184416 (2020).
- [34] J. Li, H. T. Simensen, D. Reitz, Q. Sun, W. Yuan, C. Li, Y. Tserkovnyak, A. Brataas, and J. Shi, Observation of Magnon Polarons in a Uniaxial Antiferromagnetic Insulator, *Phys. Rev. Lett.* **125**, 217201 (2020).
- [35] B. Yang, S. Y. Xia, H. Zhao, G. Liu, J. Du, K. Shen, Z. Qiu, and D. Wu, Revealing thermally driven distortion of magnon dispersion by spin Seebeck effect in $Gd_3Fe_5O_{12}$, *Phys. Rev. B* **103**, 054411 (2021).
- [36] Z. Shi, Q. Xi, J. Li, Y. Li, M. Aldosary, Y. Xu, J. Zhou, S.-M. Zhou, and J. Shi, Role of Magnon-Magnon Scattering in Magnon Polaron Spin Seebeck Effect, *Phys. Rev. Lett.* **127**, 277203 (2021).
- [37] S. L. Blank and J. W. Nielsen, The growth of magnetic garnets by liquid phase epitaxy, *J. Cryst. Growth* **17**, 302 (1972).
- [38] Z. Šimša, J. Šimšová, D. Zemanová, J. Čermák, and M. Nevřiva, Correlation between Faraday rotation and bismuth and lead contents in LPE YIG films, *Czech. J. Phys.* **34**, 1102 (1984).

- [39] B. Keszei, Z. Vértesy, and G. Vértesy, Growth of Bi and Ga substituted YIG and LuIG layers by LPE method, *Cryst. Res. Technol.* **36**, 953 (2001).
- [40] T. Kono, T. Machi, N. Chikumoto, K. Nakao, N. Koshizuka, N. Adachi, and T. Okuda, LPE growth of bismuth substituted iron garnet films with in-plane magnetization, *J. Magn. Soc. Jpn.* **30**, 600 (2006).
- [41] Z. Qiu, K. Ando, K. Uchida, Y. Kajiwara, R. Takahashi, H. Nakayama, T. An, Y. Fujikawa, and E. Saitoh, Spin mixing conductance at a well-controlled platinum/yttrium iron garnet interface, *Appl. Phys. Lett.* **103**, 092404 (2013).
- [42] Z. Qiu, D. Hou, T. Kikkawa, K. Uchida, and E. Saitoh, All-oxide spin Seebeck effects, *Appl. Phys. Express* **8**, 083001 (2015).
- [43] M.-Y. Chern and J.-S. Liaw, Study of $\text{Bi}_x\text{Y}_{3-x}\text{Fe}_5\text{O}_{12}$ thin films grown by pulsed laser deposition, *Jpn. J. Appl. Phys.* **36**, 1049 (1997).
- [44] A. Azevedo, L. H. Vilela-Leão, R. L. Rodríguez-Suárez, A. B. Oliveira, and S. M. Rezende, dc effect in ferromagnetic resonance: Evidence of the spin-pumping effect? *J. Appl. Phys.* **97**, 10C715 (2005).
- [45] E. Saitoh, M. Ueda, H. Miyajima, and G. Tatara, Conversion of spin current into charge current at room temperature: Inverse spin-Hall effect, *Appl. Phys. Lett.* **88**, 182509 (2006).
- [46] S. O. Valenzuela and M. Tinkham, Direct electronic measurement of the spin Hall effect, *Nature (London)* **442**, 176 (2006).
- [47] M. V. Costache, M. Sladkov, S. M. Watts, C. H. van der Wal, and B. J. van Wees, Electrical Detection of Spin Pumping due to the Precessing Magnetization of a Single Ferromagnet, *Phys. Rev. Lett.* **97**, 216603 (2006).
- [48] T. Kimura, Y. Otani, T. Sato, S. Takahashi, and S. Maekawa, Room-Temperature Reversible Spin Hall Effect, *Phys. Rev. Lett.* **98**, 156601 (2007).
- [49] T. Nozue, T. Kikkawa, T. Watamura, T. Niizeki, R. Ramos, E. Saitoh, and H. Murakami, Fabrication of yttrium-iron-garnet/Pt multilayers for the longitudinal spin Seebeck effect, *Appl. Phys. Lett.* **113**, 262402 (2018).
- [50] T. Kikkawa, K. Uchida, S. Daimon, Z. Qiu, Y. Shiomi, and E. Saitoh, Critical suppression of spin Seebeck effect by magnetic fields, *Phys. Rev. B* **92**, 064413 (2015).
- [51] N. Ito, T. Kikkawa, J. Barker, D. Hirobe, Y. Shiomi, and E. Saitoh, Spin Seebeck effect in the layered ferromagnetic insulators CrSiTe_3 and CrGeTe_3 , *Phys. Rev. B* **100**, 060402(R) (2019).
- [52] L. J. Cornelissen, K. Oyanagi, T. Kikkawa, Z. Qiu, T. Kuschel, G. E. W. Bauer, B. J. van Wees, and E. Saitoh, Nonlocal magnon-polaron transport in yttrium iron garnet, *Phys. Rev. B* **96**, 104441 (2017).
- [53] K. Oyanagi, T. Kikkawa, and E. Saitoh, Magnetic field dependence of the nonlocal spin Seebeck effect in Pt/YIG/Pt systems at low temperatures, *AIP Adv.* **10**, 015031 (2020).
- [54] N. Vlietstra, J. Shan, B. J. van Wees, M. Isasa, F. Casanova, and J. Ben Youssef, Simultaneous detection of the spin-Hall magnetoresistance and the spin-Seebeck effect in platinum and tantalum on yttrium iron garnet, *Phys. Rev. B* **90**, 174436 (2014).
- [55] T. Kikkawa, D. Reitz, H. Ito, T. Makiuchi, T. Sugimoto, K. Tsunekawa, S. Daimon, K. Oyanagi, R. Ramos, S. Takahashi, Y. Shiomi, Y. Tserkovnyak, and E. Saitoh, Observation of nuclear-spin Seebeck effect, *Nat. Commun.* **12**, 4356 (2021).
- [56] L. J. Cornelissen, J. Liu, R. A. Duine, J. Ben Youssef, and B. J. van Wees, Long-distance transport of magnon spin information in a magnetic insulator at room temperature, *Nat. Phys.* **11**, 1022 (2015).
- [57] L. J. Cornelissen and B. J. van Wees, Magnetic field dependence of the magnon spin diffusion length in the magnetic insulator yttrium iron garnet, *Phys. Rev. B* **93**, 020403(R) (2016).
- [58] J. M. Gomez-Perez, S. Vélez, L. E. Hueso, and F. Casanova, Differences in the magnon diffusion length for electrically and thermally driven magnon currents in $\text{Y}_3\text{Fe}_5\text{O}_{12}$, *Phys. Rev. B* **101**, 184420 (2020).
- [59] M. Kubota, A. Tsukazaki, F. Kagawa, K. Shibuya, Y. Tokunaga, M. Kawasaki, and Y. Tokura, Stress-induced perpendicular magnetization in epitaxial iron garnet thin films, *Appl. Phys. Express* **5**, 103002 (2012).
- [60] M. Kubota, K. Shibuya, Y. Tokunaga, F. Kagawa, A. Tsukazaki, Y. Tokura, and M. Kawasaki, Systematic control of stress-induced anisotropy in pseudomorphic iron garnet thin films, *J. Magn. Magn. Mater.* **339**, 63 (2013).
- [61] C. O. Avci, A. Quindeau, C.-F. Pai, M. Mann, L. Caretta, A. S. Tang, M. C. Onbasli, C. A. Ross, and G. S. D. Beach, Current-induced switching in a magnetic insulator, *Nat. Mater.* **16**, 309 (2017).
- [62] A. G. Gurevich and G. A. Melkov, *Magnetization Oscillations and Waves* (CRC Press, Boca Raton, FL, 1996).
- [63] A. Paoletti, ed., *Physics of Magnetic Garnets - Proceedings of the International School of Physics 'Enrico Fermi' 27 June-9 July 1977* (North-Holland, Amsterdam, 1978).
- [64] T. Kikkawa, K. Uchida, Y. Shiomi, Z. Qiu, D. Hou, D. Tian, H. Nakayama, X.-F. Jin, and E. Saitoh, Longitudinal Spin Seebeck Effect Free from the Proximity Nernst Effect, *Phys. Rev. Lett.* **110**, 067207 (2013).
- [65] J. C. Gallagher, A. S. Yang, J. T. Brangham, B. D. Esser, S. P. White, M. R. Page, K.-Y. Meng, S. Yu, R. Adur, W. Ruane *et al.*, Exceptionally high magnetization of stoichiometric $\text{Y}_3\text{Fe}_5\text{O}_{12}$ epitaxial films grown on $\text{Gd}_3\text{Ga}_5\text{O}_{12}$, *Appl. Phys. Lett.* **109**, 072401 (2016).
- [66] C. Tang, P. Sellappan, Y. Liu, Y. Xu, J. E. Garay, and J. Shi, Anomalous Hall hysteresis in $\text{Tm}_3\text{Fe}_5\text{O}_{12}$ /Pt with strain-induced perpendicular magnetic anisotropy, *Phys. Rev. B* **94**, 140403(R) (2016).
- [67] J. Fu, M. Hua, X. Wen, M. Xue, S. Ding, M. Wang, P. Yu, S. Liu, J. Han, C. Wang, H. Du, Y. Yang, and J. Yang, Epitaxial growth of $\text{Y}_3\text{Fe}_5\text{O}_{12}$ thin films with perpendicular magnetic anisotropy, *Appl. Phys. Lett.* **110**, 202403 (2017).
- [68] M. A. Gilleo and S. Geller, Magnetic and crystallographic properties of substituted yttrium-iron garnet, $3\text{Y}_2\text{O}_3 \cdot x\text{M}_2\text{O}_3 \cdot (5-x)\text{Fe}_2\text{O}_3$, *Phys. Rev.* **110**, 73 (1958).
- [69] T. Kikkawa, M. Suzuki, J. Okabayashi, K. Uchida, D. Kikuchi, Z. Qiu, and E. Saitoh, Detection of induced paramagnetic moments in Pt on $\text{Y}_3\text{Fe}_5\text{O}_{12}$ via x-ray magnetic circular dichroism, *Phys. Rev. B* **95**, 214416 (2017).
- [70] C. Kittel, Physical theory of ferromagnetic domains, *Rev. Mod. Phys.* **21**, 541 (1949).
- [71] H. Jin, S. R. Boona, Z. Yang, R. C. Myers, and J. P. Heremans, Effect of the magnon dispersion on the longitudinal spin Seebeck effect in yttrium iron garnets, *Phys. Rev. B* **92**, 054436 (2015).
- [72] S. M. Wu, J. E. Pearson, and A. Bhattacharya, Paramagnetic Spin Seebeck Effect, *Phys. Rev. Lett.* **114**, 186602 (2015).

- [73] Y. Chen, E. Cogulu, D. Roy, J. Ding, J. B. Mohammadi, P. G. Kotula, N. A. Missert, M. Wu, and A. D. Kent, Spin transport in an insulating ferrimagnetic-antiferromagnetic-ferrimagnetic trilayer as a function of temperature, *AIP Adv.* **9**, 105319 (2019).
- [74] K. Oyanagi, S. Takahashi, L. J. Cornelissen, J. Shan, S. Daimon, T. Kikkawa, G. E. W. Bauer, B. J. van Wees, and E. Saitoh, Spin transport in insulators without exchange stiffness, *Nat. Commun.* **10**, 4740 (2019).
- [75] K. Oyanagi, J. M. Gomez-Perez, X.-P. Zhang, T. Kikkawa, Y. Chen, E. Sagasta, A. Chuvilin, L. E. Hueso, V. N. Golovach, F. S. Bergeret, F. Casanova, and E. Saitoh, Paramagnetic spin Hall magnetoresistance, *Phys. Rev. B* **104**, 134428 (2021).
- [76] K. Oyanagi, S. Takahashi, T. Kikkawa, and E. Saitoh, Mechanism of paramagnetic spin Seebeck effect, [arXiv:2208.03857](https://arxiv.org/abs/2208.03857).
- [77] L. J. Cornelissen, K. J. H. Peters, G. E. W. Bauer, R. A. Duine, and B. J. van Wees, Magnon spin transport driven by the magnon chemical potential in a magnetic insulator, *Phys. Rev. B* **94**, 014412 (2016).
- [78] E.-J. Guo, J. Cramer, A. Kehlberger, C. A. Ferguson, D. A. MacLaren, G. Jakob, and M. Kläui, Influence of Thickness and Interface on the Low-Temperature Enhancement of the Spin Seebeck Effect in YIG Films, *Phys. Rev. X* **6**, 031012 (2016).
- [79] S. M. Rezende, R. L. Rodríguez-Suárez, R. O. Cunha, A. R. Rodrigues, F. L. A. Machado, G. A. Fonseca Guerra, J. C. Lopez Ortiz, and A. Azevedo, Magnon spin-current theory for the longitudinal spin-Seebeck effect, *Phys. Rev. B* **89**, 014416 (2014).
- [80] S. R. Boona and J. P. Heremans, Magnon thermal mean free path in yttrium iron garnet, *Phys. Rev. B* **90**, 064421 (2014).
- [81] R. Schmidt, F. Wilken, T. S. Nunner, and P. W. Brouwer, Boltzmann approach to the longitudinal spin Seebeck effect, *Phys. Rev. B* **98**, 134421 (2018).
- [82] J. Shan, L. J. Cornelissen, N. Vlietstra, J. Ben Youssef, T. Kuschel, R. A. Duine, and B. J. van Wees, Influence of yttrium iron garnet thickness and heater opacity on the nonlocal transport of electrically and thermally excited magnons, *Phys. Rev. B* **94**, 174437 (2016).
- [83] J. Shan, L. J. Cornelissen, J. Liu, J. Ben Youssef, L. Liang, and B. J. van Wees, Criteria for accurate determination of the magnon relaxation length from the nonlocal spin Seebeck effect, *Phys. Rev. B* **96**, 184427 (2017).
- [84] J. Shan, A. V. Singh, L. Liang, L. J. Cornelissen, Z. Galazka, A. Gupta, B. J. van Wees, and T. Kuschel, Enhanced magnon spin transport in NiFe₂O₄ thin films on a lattice-matched substrate, *Appl. Phys. Lett.* **113**, 162403 (2018).
- [85] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevMaterials.6.104402> for details on the temperature dependence of the nISSE results for the Pt/Bi_xY_{3-x}Fe₅O₁₂/Pt samples.
- [86] R. Schmidt and P. W. Brouwer, Theory of the low-temperature longitudinal spin Seebeck effect, *Phys. Rev. B* **103**, 014412 (2021).
- [87] S. Streib, N. Vidal-Silva, K. Shen, and G. E. W. Bauer, Magnon-phonon interactions in magnetic insulators, *Phys. Rev. B* **99**, 184442 (2019).