

## Electronic structures of quasi-one-dimensional cuprate superconductors $\text{Ba}_2\text{CuO}_{3+\delta}$

Kai Liu,<sup>1,\*</sup> Zhong-Yi Lu,<sup>1,†</sup> and Tao Xiang<sup>2,3,‡</sup>

<sup>1</sup>*Department of Physics and Beijing Key Laboratory of Opto-electronic Functional Materials & Micro-nano Devices, Renmin University of China, Beijing 100872, China*

<sup>2</sup>*Institute of Physics, Chinese Academy of Sciences, Beijing 100190, China*

<sup>3</sup>*Collaborative Innovation Center of Quantum Matter, Beijing 100190, China*



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An intact  $\text{CuO}_2$  plane is widely believed to be a prerequisite for the high- $T_c$  superconductivity in cuprate superconductors. However, an exception may exist in the superconducting  $\text{Ba}_2\text{CuO}_{3+\delta}$  materials where  $\text{CuO}$  chains play a more important role. From first-principles density functional theory calculations, we have studied the electronic and magnetic structures of  $\text{Ba}_2\text{CuO}_{3+\delta}$ . The stoichiometric  $\text{Ba}_2\text{CuO}_3$  and  $\text{Ba}_2\text{CuO}_4$  contain quasi-one-dimensional  $\text{CuO}$  chains and intact two-dimensional  $\text{CuO}_2$  planes, respectively. In comparison with the nonmagnetic metal  $\text{Ba}_2\text{CuO}_4$ ,  $\text{Ba}_2\text{CuO}_3$  is found to be an antiferromagnetic (AFM) Mott insulator. It possesses a nearest-neighbor intrachain AFM coupling and a weak interchain interaction, and its lowest unoccupied band and highest occupied band are contributed by the  $\text{Cu } 3d_{b^2-c^2}$  orbital (or  $d_{x^2-y^2}$  orbital if we denote the  $bc$  plane as the  $xy$  plane) and  $\text{O } 2p$  orbitals, respectively. Total energy calculations indicate that the oxygen vacancies in  $\text{Ba}_2\text{CuO}_{3+\delta}$ , in reference to conventional cuprate superconductors, prefer to reside in the planar sites rather than the apical oxygens in the  $\text{CuO}$  chains, in agreement with the experimental observation. Furthermore, we find that moderate charge doping can reduce the energy differences between the low-lying magnetic states of  $\text{Ba}_2\text{CuO}_{3+\delta}$ , which may induce the spin fluctuations that are beneficial for the appearance of superconductivity. Our results also suggest that the spin fluctuations in  $\text{Ba}_2\text{CuO}_{3+\delta}$  are anisotropic, which deserves further experimental verification.

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### I. INTRODUCTION

The interplay between superconductivity and dimensionality is an interesting issue in condensed matter physics. In the well-known cuprate superconductors [1,2], whose superconducting transition temperature is highest at ambient pressure [3,4], a common feature is that they all contain quasi-two-dimensional  $\text{CuO}_2$  planes [5,6] from which the superconducting pairing emerges while other layers just serve as a charge reservoir. Similarly, superconducting condensation is also believed to take place predominantly in quasi-two-dimensional  $\text{FeX}$  ( $X = \text{As, Se, \dots}$ ) layers in iron-based superconductors [7–11]. In recent years, superconducting transitions have also been discovered in Bechgaard-salts organic superconductors [12,13], molybdenum chalcogenides [14,15] and pnictides [16], chromium pnictides [17–22], bismuth iodide [23], hafnium-vanadium-gallium compound [24,25], nickel-bismuth compounds [26], and other quasi-one-dimensional materials.

More recently, the high-temperature superconductor  $\text{Ba}_2\text{CuO}_{3+\delta}$  has been successfully synthesized under high pressure by Jin and co-workers [27]. There are two possibilities regarding the parent compound of these superconductors. One possibility is that  $\text{Ba}_2\text{CuO}_4$  is the parent compound.

High- $T_c$  superconductivity emerges in  $\text{Ba}_2\text{CuO}_{3+\delta}$  when some oxygens are removed from  $\text{Ba}_2\text{CuO}_4$ . The other is that  $\text{Ba}_2\text{CuO}_3$  is the parent compound and the superconductivity emerges in  $\text{Ba}_2\text{CuO}_{3+\delta}$  when more oxygens are doped to it. For the latter, unlike the parent compound of other cuprate superconductors,  $\text{Ba}_2\text{CuO}_3$  does not contain  $\text{CuO}_2$  planes. Instead, it contains only  $\text{CuO}$  chains and the copper-oxygen bond is compressed along the  $c$  axis but stretched along the  $b$  axis (the  $\text{CuO}$  chain direction).

In order to determine the pairing mechanism of electrons, it is crucial to know where the oxygen vacancies reside in the superconducting  $\text{Ba}_2\text{CuO}_{3+\delta}$  compounds, namely, in reference to conventional cuprate superconductors hereafter. If the oxygen vacancies mainly occupy the apical sites, the  $\text{CuO}_2$  planes remain intact as the other cuprate superconductors. Experimental measurements of neutron powder diffraction [28] and electron diffraction [29] for similar compounds, however, suggested that the oxygen vacancies reside mainly on the  $\text{CuO}_2$  planes and the apical sites are fully occupied by oxygen atoms. In this case, there are no perfect  $\text{CuO}_2$  planes. This implies that intact  $\text{CuO}_2$  planes are not absolutely necessary in achieving high- $T_c$  superconductivity.

For cuprate superconductors, the parent compounds are antiferromagnetic (AFM) Mott insulators [30] and the superconductivity can be induced via hole [31] or electron [32,33] doping with enhanced AFM spin fluctuations [34,35]. It is thus important to investigate the magnetic properties of the parent compound of the  $\text{Ba}_2\text{CuO}_{3+\delta}$  superconductor. Furthermore, the oxygen-enriched  $\text{Ba}_2\text{CuO}_{3+\delta}$  or oxygen-deficient

\*kliu@ruc.edu.cn

†zlu@ruc.edu.cn

‡txiang@iphy.ac.cn

$\text{Ba}_2\text{CuO}_{4-\delta}$  can be regarded as hole doping to  $\text{Ba}_2\text{CuO}_3$  or electron doping to  $\text{Ba}_2\text{CuO}_4$ , respectively. The evolution of magnetism with charge doping is also a key issue for understanding the superconductivity of  $\text{Ba}_2\text{CuO}_{3+\delta}$ .

In this paper, we address the above questions by calculating the electronic and magnetic structures of  $\text{Ba}_2\text{CuO}_{3+\delta}$  using first-principles density functional theory. In Sec. II, we first describe the method, and then discuss the numerical results for both stoichiometric  $\text{Ba}_2\text{CuO}_3/\text{Ba}_2\text{CuO}_4$  and oxygen-doped materials. A summary is given in Sec. III.

## II. RESULTS

We performed the first-principles density functional theory calculations on the electronic and magnetic structures of undoped or doped  $\text{Ba}_2\text{CuO}_3$  or  $\text{Ba}_2\text{CuO}_4$  materials using the projector augmented wave (PAW) method [36] with the Vienna *ab initio* simulation package [37,38]. The generalized gradient approximation (GGA) of Perdew-Burke-Ernzerhof (PBE) type was used in the construction of the exchange-correlation functional [39]. The plane-wave basis set with a kinetic energy cutoff of 520 eV and a  $12 \times 12 \times 4$   $k$ -point mesh for the Brillouin zone sampling of conventional cells were adopted. The Fermi level was broadened by the Gaussian smearing method with a width of 0.05 eV.  $\text{Ba}_2\text{CuO}_{3+\delta}$  can be regarded as a compound either by adding oxygen to  $\text{Ba}_2\text{CuO}_3$  or by removing oxygen from  $\text{Ba}_2\text{CuO}_4$ . Various locations of oxygen vacancies in  $\text{Ba}_2\text{CuO}_{3+\delta}$  were investigated with both the virtual crystal approximation (VCA) and the supercell methods. In the latter case, a  $2 \times 2 \times 1$  supercell with different oxygen vacancy distributions was explored, while the lattice constants were fixed to the experimental values, and all internal atomic positions were allowed to relax until the forces on all atoms were smaller than 0.01 eV/Å. The correlation effect among Cu  $3d$  electrons was incorporated using the GGA+ $U$  formalism of Dudarev *et al.* [40] with a typical effective Hubbard interaction  $U$  of 6.5 eV [41].

### A. Electronic and magnetic structures of stoichiometric $\text{Ba}_2\text{CuO}_3$ and $\text{Ba}_2\text{CuO}_4$

Figures 1(a) and 1(b) show the crystal structures of  $\text{Ba}_2\text{CuO}_3$  and  $\text{Ba}_2\text{CuO}_4$ , respectively. The former is composed of CuO chains along the  $b$  axis, while the latter contains  $\text{CuO}_2$  planes. In either case, Cu-O layers are separated by Ba-O layers. Cu atoms in both  $\text{Ba}_2\text{CuO}_3$  and  $\text{Ba}_2\text{CuO}_4$  form a squarelike lattice in each  $ab$  plane.  $\text{Ba}_2\text{CuO}_4$  has the same lattice structure as  $\text{La}_2\text{CuO}_4$ . However, it is only in  $\text{Ba}_2\text{CuO}_3$  that Cu ions have the same nominal valence +2 as in  $\text{La}_2\text{CuO}_4$ .

The  $a$  and  $b$  axes in  $\text{Ba}_2\text{CuO}_4$  are equivalent because there are no O vacancies in the  $\text{CuO}_2$  planes. However, they are not equivalent in  $\text{Ba}_2\text{CuO}_3$  [Fig. 1(a)]. To determine the ground states of  $\text{Ba}_2\text{CuO}_3$  and  $\text{Ba}_2\text{CuO}_4$ , we calculated the energies of several representative magnetic states formed by  $\text{Cu}^{2+}$  ions and compared with that of the nonmagnetic state. These magnetic ordering states, as partially shown in Fig. 2, include the standard AFM Néel state, the collinear AFM (or stripe) state, the AFM1 state in which Cu spins are antiferromagnetically coupled along the  $b$  axis but ferromagnetically coupled along

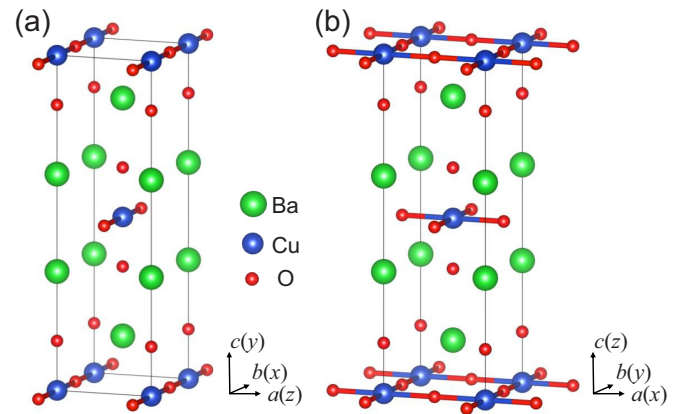


FIG. 1. Crystal structures of (a)  $\text{Ba}_2\text{CuO}_3$  with CuO chains and (b)  $\text{Ba}_2\text{CuO}_4$  with  $\text{CuO}_2$  planes. Note the  $xy$  planes of  $\text{Ba}_2\text{CuO}_3$  and  $\text{Ba}_2\text{CuO}_4$  are defined differently in order to facilitate the discussion on orbitals.

the  $a$  axis, the AFM dimer state [42,43] along the  $b$  axis which is ferromagnetically coupled along the  $a$  axis, and the ferromagnetic (FM) state.

For  $\text{Ba}_2\text{CuO}_3$ , the relative energies of the above magnetic states with respect to the nonmagnetic state are given in Table I. All these magnetic states have lower energies than the nonmagnetic one. Among them, the AFM Néel state is energetically degenerate with the AFM1 state, and the collinear AFM state is energetically degenerate with the FM state. The ground state is found to have either the AFM Néel or the AFM1 order, indicating that there is a strong intrachain AFM coupling along the  $b$  axis but a weak interchain coupling along the  $a$  axis. The AFM dimer state is next to the ground state in energy. In contrast, for  $\text{Ba}_2\text{CuO}_4$ , all the above magnetic states are unstable and the ground state is nonmagnetic. From the calculations, we find that the weak interchain coupling in  $\text{Ba}_2\text{CuO}_3$  is about 0.2 meV, smaller than the corresponding values in  $\text{Ca}_2\text{CuO}_3$  (3.6 meV) and in  $\text{Sr}_2\text{CuO}_3$  (0.8 meV) [44,45].

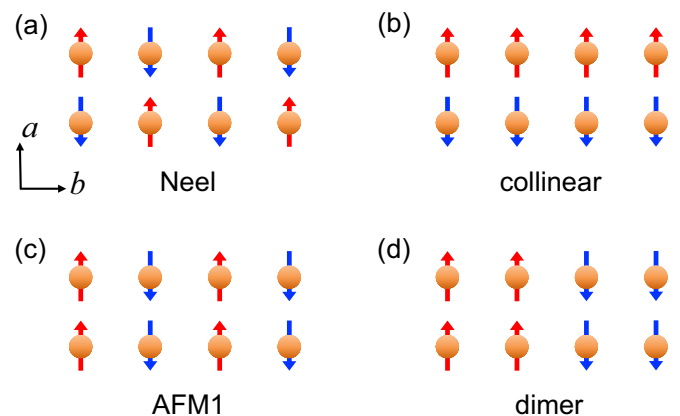


FIG. 2. Spin configurations of (a) AFM Néel state, (b) collinear AFM state, (c) AFM1 state, and (d) AFM dimer state for the  $\text{Cu}^{2+}$  spins in the  $ab$  plane. The brown balls denote Cu atoms. The red and blue arrows represent the up and down spins, respectively.

TABLE I. Relative energies (in unit of meV/Cu) of several magnetic states with respect to the nonmagnetic (NM) state for undoped  $\text{Ba}_2\text{CuO}_3$ .

	Néel	Collinear	AFM1	Dimer	FM
$\text{Ba}_2\text{CuO}_3$	-180	-22	-180	-116	-22

Figure 3 shows the total density of states (DOS) for  $\text{Ba}_2\text{CuO}_3$  in the AFM Néel state and  $\text{Ba}_2\text{CuO}_4$  in the nonmagnetic state, respectively. In contrast to  $\text{Ba}_2\text{CuO}_3$  which is an AFM insulator,  $\text{Ba}_2\text{CuO}_4$  is found to be a nonmagnetic metal. The calculated band gap of  $\text{Ba}_2\text{CuO}_3$  is 1.2 eV, close to the experimental results for  $\text{Ca}_2\text{CuO}_3$  (1.7 eV) and  $\text{Sr}_2\text{CuO}_3$  (1.5 eV) [46]. It should be pointed out that the AFM insulating ground state of  $\text{Ba}_2\text{CuO}_3$  is obtained only when the on-site Hubbard interaction is included in the calculation. Without this interaction, the ground state is metallic. This suggests that  $\text{Ba}_2\text{CuO}_3$  is an AFM Mott insulator [47], similar to  $\text{La}_2\text{CuO}_4$  [30].

We also calculated the partial density of states (PDOS) for  $\text{Ba}_2\text{CuO}_3$  in the AFM Néel state (Fig. 4). For convenience, here we denote the  $b$  and  $c$  axes as the  $x$  and  $y$  axes [Fig. 1(a)]. Due to the nominal valence +2 of Cu ions in  $\text{Ba}_2\text{CuO}_3$ , nine electrons fill the  $3d$  orbitals of the Cu atom. Among these  $3d$  orbitals, the spin-minority  $d_{b^2-c^2}$  orbital (or  $d_{x^2-y^2}$  orbital if we denote the  $bc$  plane as the  $xy$  plane) is mostly unoccupied [Fig. 4(a)]. The occupied bands are mainly contributed by O  $2p$  orbitals. This results from the crystal field effect created by four O atoms surrounding a Cu atom in the  $bc$  plane. The PDOS peaks of unoccupied Cu  $d_{b^2-c^2}$  and O  $2p_b$  states appear at almost the same energy above the band gap. This indicates that there is a strong  $p$ - $d$  hybridization between these orbitals.

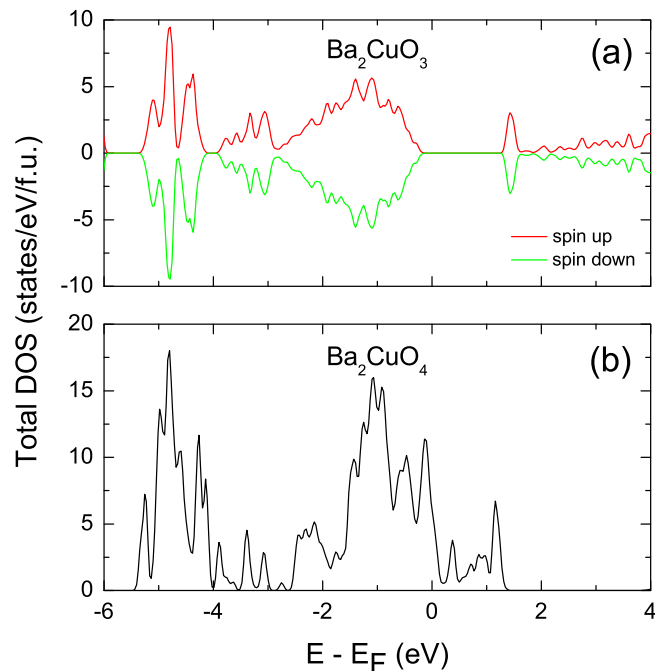


FIG. 3. Density of states (DOS) for (a) the AFM Néel state of  $\text{Ba}_2\text{CuO}_3$  and (b) the nonmagnetic state of  $\text{Ba}_2\text{CuO}_4$ .

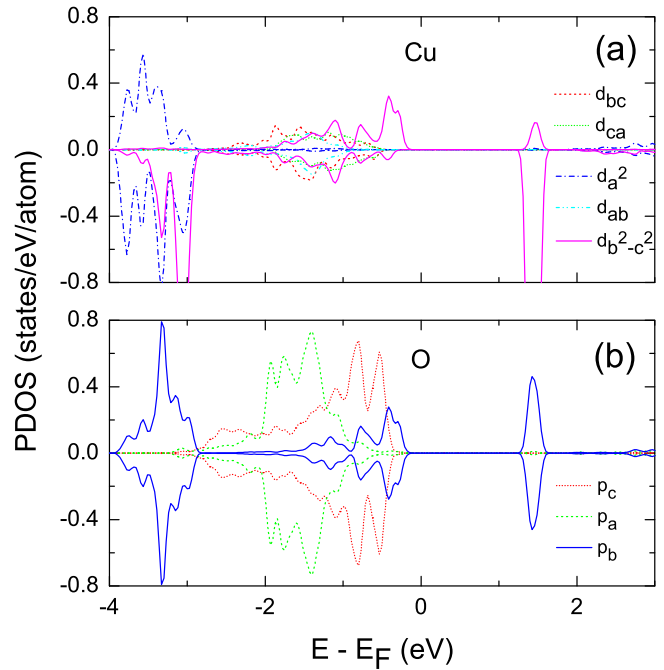


FIG. 4. Partial density of states (PDOS) for the AFM Néel state of  $\text{Ba}_2\text{CuO}_3$  projected on (a) the  $3d$  orbitals of the Cu atom and (b) the  $2p$  orbitals of the O atom along the CuO chain. The up and down parts in each panel represent the spin-up and spin-down channels, respectively.

Figures 5(a) and 5(b) show the calculated band structures of  $\text{Ba}_2\text{CuO}_3$  along high-symmetry paths of the Brillouin zone in the nonmagnetic and the AFM Néel states, respectively. In the AFM state, the valence band maximum is located at the  $M$  point. Since the CuO chain is along the  $b$  axis and the interchain couplings are rather weak, the band dispersions along the  $k_b$  direction are much larger than those along the other two directions.

In oxygen-enriched  $\text{Ba}_2\text{CuO}_{3+\delta}$ , the extra oxygens doped to  $\text{Ba}_2\text{CuO}_3$  introduce hole doping, which is equivalent to shifting the Fermi energy down to the valence bands. In Figs. 5(c) and 5(d), we show the Fermi surface contours when the Fermi level is 0.05 and 0.1 eV below the top of the valence bands in the AFM Néel state, respectively. Upon small hole doping, a hole pocket emerges around the  $M$  point. When more holes are doped into the system, the hole pockets merge together to form flat Fermi surface sheets in the  $k_a$ - $k_c$  plane.

## B. Electronic and magnetic structures of $\text{Ba}_2\text{CuO}_{3+\delta}$

In high- $T_c$  cuprates, superconductivity emerges when charge carriers, either holes or electrons, are doped into the parent compounds. It is commonly believed that AFM spin fluctuations play a very important role in pairing electrons [30,34,35]. One can dope holes to  $\text{Ba}_2\text{CuO}_3$  either by introducing extra oxygen atoms or by substituting Ba atoms with alkali atoms [48,49]. The superconducting material  $\text{Ba}_2\text{CuO}_{3+\delta}$  [27,50–52] can be regarded as a compound either by introducing more oxygens to  $\text{Ba}_2\text{CuO}_3$  or by removing oxygens from  $\text{Ba}_2\text{CuO}_4$ .

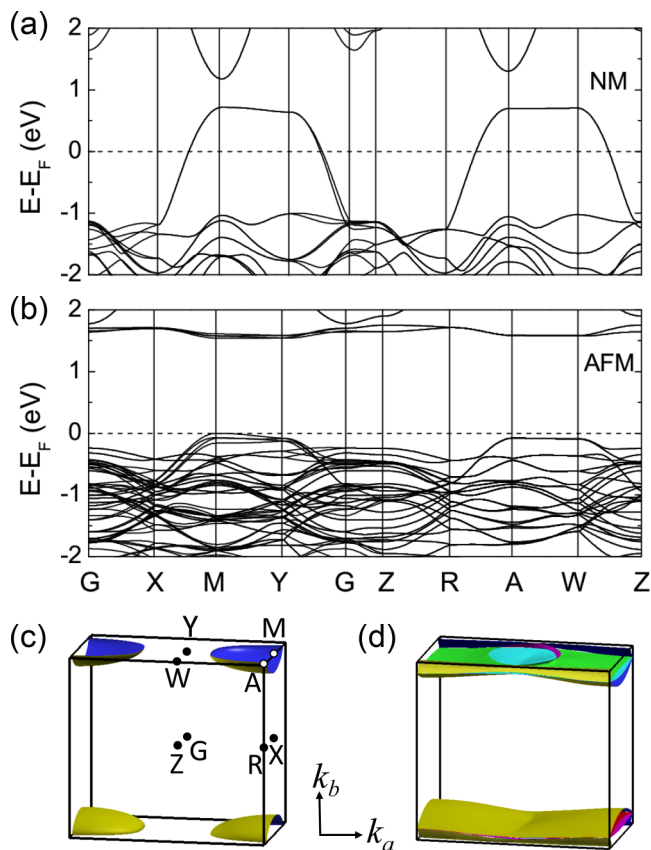


FIG. 5. Band structures of  $\text{BaCuO}_3$  in (a) the nonmagnetic state with a conventional cell and (b) the AFM Néel state with a  $2 \times 2 \times 1$  supercell. (c) and (d) show the Fermi surface contours in the AFM Néel state upon hole doping by rigidly shifting the Fermi level 0.05 and 0.1 eV below the top of the valence bands, respectively.

To know the electronic and magnetic structures of this material, it is important to know accurately where the oxygen vacancies are located. We investigated this problem using both the virtual crystal approximation approach and the supercell approach in the framework of density functional theory.

In the virtual crystal approach, we considered three different kinds of distributions of oxygen vacancies: (1) the vacancies are all located in the Cu-O layers, (2) the vacancies are all located in the Ba-O layers, and (3) the vacancies are equally partitioned into each O site in both kinds of layers. For a nominal component  $\text{Ba}_2\text{CuO}_{3.2}$ , the relative energies of the second and third cases with respect to the first case are found to be +1.88, and +92.58 eV/cell, respectively. This suggests that the oxygen vacancies prefer to reside in the Cu-O layers.

TABLE II. Relative total energies (in unit of eV) for all 26 structures of  $\text{Ba}_2\text{CuO}_{3.25}$  shown in Figs. 6 and 7 with oxygen vacancies in a  $2 \times 2 \times 1$  supercell which contains 16 Ba atoms, 8 Cu atoms, 26 O atoms, and 6 O vacancies.

	BaO-1	BaO-2	82-1	82-2	82-3	73-1	73-2	64-1	64-2	64-3	64-4	64-5	64-6
$\Delta E$	-1.871	0	-5.168	-4.450	-4.480	-6.350	-6.190	-8.173	-7.319	-7.069	-7.614	-7.380	-7.142
	55-1	55-2	55-3	55-4	64-1*	64-3*	64-4*	64-5*	64-6*	55-1*	55-2*	55-3*	55-4*
$\Delta E$	-7.779	-7.736	-7.779	-7.711	-8.201	-7.042	-7.606	-7.401	-7.206	-7.729	-7.657	-7.704	-7.683

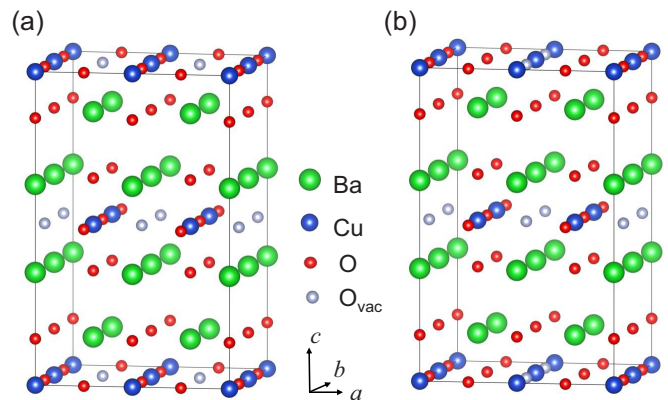


FIG. 6. Crystal structures with (a) the lowest energy state and (b) the second-lowest energy state for  $\text{Ba}_2\text{CuO}_{3.25}$ . They correspond to the 64-1\* and 64-1 structures in Table II, respectively. The label  $\text{O}_{\text{vac}}$  represents O vacancy.

In the supercell approach, we calculated the energies for 26 possible structures of  $\text{Ba}_2\text{CuO}_{3.25}$  (Figs. 6 and 7) in a  $2 \times 2 \times 1$  supercell with six oxygen vacancies. By comparison, we find that the two configurations shown in Figs. 6(a) and 6(b) have the lowest and the second-lowest energies, respectively. In both cases, we also find that the Ba-O layer stays intact, while the oxygen vacancies also prefer to locate in the Cu-O layers rather than occupying the apical oxygen positions. The relative total energies of these 26 structures are listed in Table II.

The results obtained from both approaches indicate that the oxygen vacancies are located in the Cu-O layers rather than in the Ba-O layers. This is consistent with the experimental data of neutron powder diffraction for  $\text{Ba}_2\text{CuO}_{3+\delta}$  [53], and is also in line with neutron powder diffraction [28] and electron diffraction [29] measurements for  $\text{Sr}_2\text{CuO}_{3+\delta}$ .

To see how the magnetic order is changed by the doping to  $\text{Ba}_2\text{CuO}_3$ , we evaluated the energies of several different kinds of magnetic states. Since the interchain coupling is very weak, we ignore the difference of different magnetic ordered states along the  $a$  axis and assume that the interchain coupling along the  $a$  axis is ferromagnetic. Table III shows how the energies of three typical magnetic states (i.e., AFM1, AFM dimer, and FM states) vary with the electron or hole doping. As expected, the energy difference between the lowest-energy (AFM1) state and the second-lowest-energy (dimer) state is reduced with increasing doping. This suggests that the doping tends to induce magnetic frustrations [54] and enhance spin fluctuations, favoring the formation of superconducting Cooper pairs.

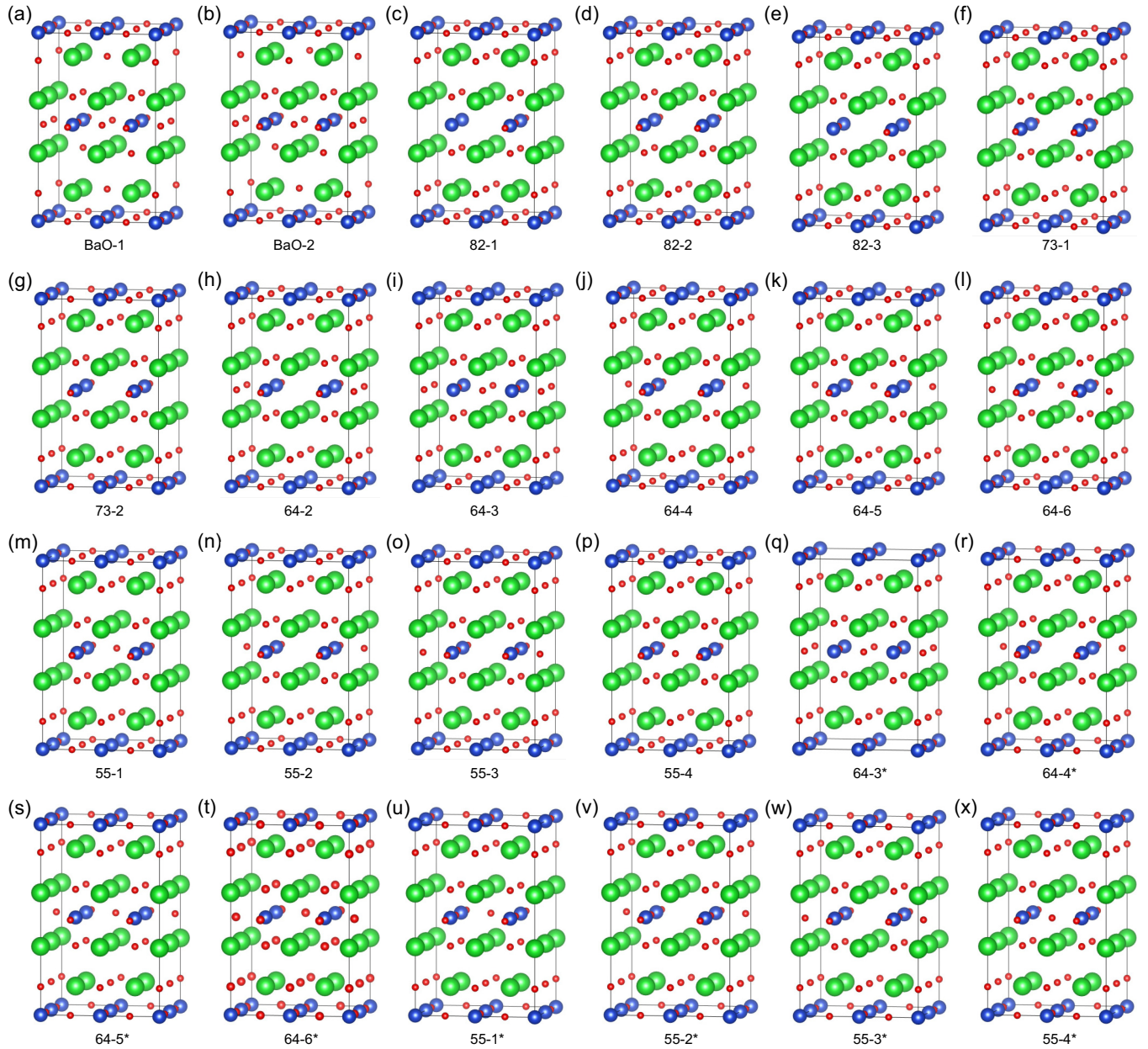


FIG. 7. Twenty-four possible oxygen vacancy structures of  $\text{Ba}_2\text{CuO}_{3.25}$ . For BaO-1 and BaO-2 structures, O vacancies are in the Ba-O layers. For  $mn-i$  structures, there are  $m$  and  $n$  O atoms in the bottom and middle Cu-O layers of the supercell, respectively. The green, blue, and red balls represent the Ba, Cu, and O atoms, respectively.

The GGA+ $U$  method may capture the electronic and magnetic properties of cuprates if the value of Hubbard  $U$  is properly selected [41,55,56]. Here, we chose a typical

TABLE III. Relative energies (in unit of meV/Cu) of the AFM1 state and the AFM dimer state with respect to that of the FM state for  $\text{Ba}_2\text{CuO}_3$  under different electron ( $e$ ) and hole ( $h$ ) doping concentrations (per Cu atom).

Doping	0	$0.1e$	$0.15e$	$0.1h$	$0.15h$	$0.2h$	$0.3h$
AFM1	-158	-62	-14	-136	-90	-70	-24
Dimer	-94	-35	-11	-80	-47	-34	-10

Hubbard  $U$  of 6.5 eV for Cu  $3d$  electrons [41]. To examine the influence of  $U$  on the magnetism, we performed supplementary calculations with different  $U$  values (5.5 and 7.5 eV). The results indicate that the variation of Hubbard  $U$  in the usual range for Cu  $3d$  electrons does not change the conclusion of our work.

The above discussion suggests that it is  $\text{Ba}_2\text{CuO}_3$  rather than  $\text{Ba}_2\text{CuO}_4$  that is the parent compound of superconducting  $\text{Ba}_2\text{CuO}_{3+\delta}$  materials [27,50–52]. This can be more clearly seen by comparison with the other parent compounds of cuprate superconductors, such as  $\text{La}_2\text{CuO}_4$ . First, like  $\text{La}_2\text{CuO}_4$ ,  $\text{Ba}_2\text{CuO}_3$  is an AFM insulator. Second, Cu ions in  $\text{Ba}_2\text{CuO}_3$  have the same +2 nominal valences as in  $\text{La}_2\text{CuO}_4$ . Third, the highest occupied and the lowest unoccupied states

in both compounds derive from the strongly hybridized O  $2p$  orbitals and Cu  $3d_{x^2-y^2}$  orbitals (for  $\text{Ba}_2\text{CuO}_3$ , the  $bc$  plane is defined as the  $xy$  plane and  $d_{x^2-y^2}$  is just the  $d_{b^2-c^2}$  orbital).

### III. SUMMARY

We have studied the electronic and magnetic structures of  $\text{Ba}_2\text{CuO}_{3+\delta}$  with  $\delta$  varying from 0 to 1 by first-principles density functional theory calculations. Unlike  $\text{Ba}_2\text{CuO}_4$  whose ground state is a nonmagnetic metal, the ground state of  $\text{Ba}_2\text{CuO}_3$  is found to be a quasi-one-dimensional AFM Mott insulator. The lowest unoccupied and highest occupied states in  $\text{Ba}_2\text{CuO}_3$  are mainly contributed by Cu  $3d_{b^2-c^2}$  orbitals and O  $2p_b$  orbitals, respectively. By comparison of the total energies of  $\text{Ba}_2\text{CuO}_{3+\delta}$  with different kinds of oxygen vacancy structures, we find that the oxygen vacancies reside mainly in the Cu-O layers (planar sites) rather than in the Ba-O layers (apical sites), in agreement with the experimental observation [53]. Furthermore, doping of charge carriers to

$\text{Ba}_2\text{CuO}_3$  can reduce the energy differences between different low-energy magnetic states and thus enhance the spin fluctuations. All these suggest that  $\text{Ba}_2\text{CuO}_3$  is the parent compound of oxygen-doped  $\text{Ba}_2\text{CuO}_{3+\delta}$  superconductors, and the AFM fluctuations in CuO chains play an important role in the superconducting pairing of electrons in these materials.

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