Electrical studies of Barkhausen switching noise in ferroelectric PZT: Critical exponents and temperature dependence

C. D. Tan,¹ C. Flannigan,¹ J. Gardner,² F. D. Morrison,² E. K. H. Salje,^{3,4} and J. F. Scott^{1,2}

¹*School of Physics and Astronomy, University of St. Andrews, St. Andrews, Fife KY16 9SS, United Kingdom*

²*School of Chemistry, University of St. Andrews, St. Andrews, Fife KY16 9ST, United Kingdom*

³*Deptartment of Earth Sciences, Cambridge University, Cambridge CB2 3EQ, United Kingdom* ⁴*State Key Laboratory for Mechanical Behavior of Materials, Xi'an Jiaotong University, Xi'an 710049, China*

(Received 21 September 2018; revised manuscript received 13 December 2018; published 6 March 2019)

Crackling noise of ferroelectric lead zirconate titanate samples during ferroelectric switching is demonstrated to be compatible with avalanche statistics. The peaks of the slew rate (time derivative of current dI/dt squared), defined as "jerks," were statistically analyzed and shown to obey power laws. The critical exponent obtained is 1.64 ± 0.15 , in agreement with predictions from avalanche theory. The exponent is independent of temperature within experimental error margins.

DOI: [10.1103/PhysRevMaterials.3.034402](https://doi.org/10.1103/PhysRevMaterials.3.034402)

I. INTRODUCTION

A complete analysis of the dynamic properties of ferroelectric domain switching [\[1\]](#page-4-0) showed that the statistical switching variables (switching amplitudes, energy, and interevent times) in BaTiO₃ follow avalanche statistics. The switching exponents are universal and close to values of the field integrated mean field model, which predicts that the stress-integrated distribution of pulse energies scales as [\[2\]](#page-4-0)

$$
p(E) \sim E^{-5/3} = E^{-1.67}.
$$
 (1)

The experimental approach in [\[1\]](#page-4-0) was to measure the acoustic emission (AE) generated by moving 90° boundaries in BaTiO₃. Such AE distributions have been observed previously during ferroelastic phase transitions [\[2–5\]](#page-4-0) and their origin was identified as moving domains [\[6\]](#page-4-0). Here we show that the same criticality is seen by purely electronic measurements in the most common ferroelectric material (lead zirconate titanate, PZT). Small jumps in the ferroelectric hysteresis curve are identified as being due to moving domains. These jumps occur at the steepest region of the polarization hysteresis curve $[7,8]$. When electric fields are applied, the domains initially aligned antiparallel to the field will reverse creating interfaces, *domain walls* [\[9,10\]](#page-4-0), extending through space. When the domain walls move, they may come across regions near defects and become pinned [\[7,8\]](#page-4-0). As the magnitude of the field increases, the domain walls will eventually gain enough energy to overcome the pinning and move forward, causing an increase in polarization [\[7,8\]](#page-4-0). The rapid nucleation of new domain walls and their jerky movements are responsible for the temporal changes in polarization.

Barkhausen noise and many other crepitations are statistically similar: they resemble avalanches and the size distribution of these crackling noises obeys a power-law with characteristic exponents $[6,11-14]$. Due to universality, these exponents allow avalanche systems to be compared with one another, leading to a lot of interesting physics being unveiled [\[6,11,12\]](#page-4-0). For example, Dahmen and Ben-Zion in Ref. [\[12\]](#page-4-0) showed that Barkhausen noise and earthquake events share many statistical similarities. Meanwhile, Barό *et al.* in Ref. [\[15\]](#page-4-0) demonstrated that earthquakes and compressions of porous materials are statistically alike too. Performing these experiments under the safety of the laboratory allows researchers to learn more about seismology and improve earthquake risk assessments [\[6,16,17\]](#page-4-0).

The investigation of purely electrical noise is a very important problem in low electric fields for PZT for medical magnetic resonance imaging [\[18\]](#page-4-0) and in high electric fields [\[19\]](#page-4-0) for switching devices such as ferroelectric memory cards (\$100 million/year product last year for subway fare cards, etc., in Japan and the USA). However, most reviews are limited to the low-field situation [\[20\]](#page-4-0). Two commercial PZT ceramics, PIC 151 and PIC 255, from PI Ceramic Lederhose, Germany [\[21\]](#page-4-0) were categorized using a conventional hysteresis apparatus and the jerky changes in the polarization switching current responses were shown to obey power laws (critical exponents listed in Table [I](#page-1-0) below). If treated as proxies for energy exponents, our results are in agreement with the predictions of avalanche theory $[6,14,22-25]$. The actual compositions of both samples are proprietary and therefore undisclosed, but from the product brochure [\[21\]](#page-4-0), PIC 255 was designed to have a higher coercive field *Ec* compared to PIC 151.

In addition to the basic critical exponent measurement, we have explored the temperature dependence of the crackling noise. We find that as temperature is increased towards the Curie temperature T_c , there is a nonmonotonic evolution of a second exponent, reaching a peak at ∼1.90 at *T* = 413 K. This effect may be related to the dynamical behavior of dislocations. PZT has a large number [\[26–](#page-4-0)[28\]](#page-5-0) of threading dislocations $(10^{13} \text{ cm}^{-2}$ at best, about an order of magnitude greater than in $BaTiO₃$), and empirical evidence from avalanche in other systems (such as porous metals) suggests that exponents near 1.9–2.0 are due to such dislocations [\[29\]](#page-5-0).

TABLE I. Summarized critical exponents ε of PIC 151, PIC 255, and PIC 151b from this study and the field-driven energy exponent from MFT $[6]$.

Systems	PIC 151	PIC 255	PIC 151b	Field-driven MFT
ε		1.73 ± 0.04 1.61 ± 0.04 1.64 ± 0.04		1.67

II. EXPERIMENT DETAILS AND PROCEDURES

The measurements were conducted using an aixACCT TF Analyzer 2000 [\[30,31\]](#page-5-0) with a high voltage setup that allows voltage pulses ranging from 200 V to 40 kV $[31]$ to be applied to the sample via the probes.

In our experiments, triangular electric field/voltage pulses were designed using a manual waveform generator from the TF Analyzer 2000 software by providing the target voltage values at specific time stamps. As the coercive field E_c of ferroelectrics changes with the voltage ramp rate [\[32,33\]](#page-5-0), the time taken for different maximum voltages were calculated and specified to keep the ramp rate constant. The ramp rate was 40Vs^{-1} for PIC 151 and 60Vs^{-1} for PIC 255. A downswitching-pulse was applied through a sample via the probes followed by an equal and opposite up-switching-pulse. The current-time response during the linear ramping region of the up-switching-pulse was then acquired. Due to the limitations of the apparatus, the highest achievable sampling rate was 40 Hz (1000 points 25 s).

Figure 1 shows the current *I* response (black dashed) of PIC 151 against time. The measured current is not smooth but consists of small jumps called jerks and may indicate very small events [\[6\]](#page-4-0). Procedures for extracting the jerks from the measured current *I* were as follows:

FIG. 1. A single measurement from PIC 151. The smooth looking current *I* curve (black dashed line) was jerky, as shown after taking the square of its time derivative, $(dv/dt)^2$ and removing the baseline (blue). The peaks (red labels) were used for statistical analysis. The baseline removed graph is conventionally called the jerk spectrum.

(1) The first time derivative *dI*/*dt* is taken and squared, revealing the jerky nature of the current response with peaks superposed onto a smooth baseline.

(2) The minima were then fitted using the Piecewise Cubic Hermite Interpolating Polynomial function from Matlab®.

(3) The fitted line was subtracted, leaving $\left(\frac{dv}{dt}\right)^2$ with its baseline removed (blue solid line in Fig. 1). This is conventionally called the jerk spectrum [\[6\]](#page-4-0).

(4) The peaks in the jerk spectrum (red labels in Fig. 1) are defined as jerks *J* and are subjected to statistical analyses.

In avalanche studies, jerks can be defined in multiple ways [\[6,23](#page-4-0)[,34\]](#page-5-0). In He *et al.* [\[34\]](#page-5-0) ferroelastic switching simulations, the jerks were defined as the total change in potential energy, the energy drop and the shear stress drop. In crystal plasticity experiments, the jerks were defined as the square of the velocity *v* of *slip* avalanches squared, $\left(\frac{dv}{dt}\right)^2$, which took the form of an energy $[6,14,22,25]$.

In our case, the jerks are defined as peaks of the square of the *slew rate* [\[35\]](#page-5-0), $(dv/dt)^2$. The slew rate dI/dt is a term commonly found in electronics to describe how fast an input waveform can change when it passes through an op-amp before the signal is distorted [\[36\]](#page-5-0). The slew rate is usually defined as the time derivative of voltage *dV*/*dt* [\[36\]](#page-5-0) but it can also be defined as *dI*/*dt* [\[35\]](#page-5-0). Taking the square of the slew rate allows the peaks in our jerk spectrum to closely represent the jerks defined in crystal plasticity and acoustic emission experiments, which is $(dv/dt)^2$ as mentioned. In this study, we successfully show that *J* extracted from the slew rate obeys a power law $[6,11,12]$:

$$
P(J) = \frac{\varepsilon - 1}{J_{\min}} \left(\frac{J}{J_{\min}} \right)^{-\varepsilon},\tag{2}
$$

where ε is an exponent and J_{min} is the lower-bound normalization condition [\[37,38\]](#page-5-0).

III. RESULTS AND DISCUSSION

A total of five measurements were taken for both PIC 151 and PIC 255. For each sample individually, the jerk peaks were extracted, normalized by the mean, and combined for statistical evaluation. A total of 1261 peaks were retrieved for PIC 151 and 1468 peaks for PIC 255.

The peaks for both samples were logarithmically binned and the distributions were fitted with two straight lines (Fig. [2\)](#page-2-0). The slope yields −1.59 for PIC 151 (blue circles) and −1.50 for PIC 255 (red squares).

Following [\[37\]](#page-5-0), a high coefficient of determination R^2 value of the linear fit cannot be trusted. *Non-power-law distributed* histograms *can resemble* a power-law distribution over many orders of magnitude, thus providing a large *R*² value. Also, the jerk data *a priori* could involve two exponents or more (or a whole Gaussian distribution of exponents). Fitting a straight line could lead to a false slope that was due to the averaging effect of exponents [\[38\]](#page-5-0). And lastly, the starting and ending limits of the fitted data should be independently justified. The data in Fig. [2](#page-2-0) clearly demonstrate a straight-line behavior. Nevertheless, it is desirable to show that this is robust, and that the apparent power-law distribution corresponds to a single exponent [\[37,38\]](#page-5-0).

FIG. 2. The log-log probability distribution $P(J_{\text{norm}})$ of normalized jerks J_{norm} from PIC 151 (blue circles) and PIC 255 (red squares) shows a straight-line behavior. The slope is linearly fitted at region ln(*J*norm) > −2 for PIC 151 is −1.59 and for PIC 255 is −1.50.

To accomplish these buttressing arguments we utilize the Maximum Likelihood (ML) method. This is a statistically reliable way of demonstrating a single exponent in ε power-law dependence. The procedure is sketched below and provided in complete detail elsewhere [\[38\]](#page-5-0).

ML estimates how likely a scaling parameter, the exponent ε from the power-law model, had generated an experimental jerk spectrum [\[37,39\]](#page-5-0). The estimated exponent derived [\[37\]](#page-5-0) from maximizing the *likelihood* function with respect to an exponent ε is

$$
\hat{\varepsilon} = 1 + N \left[\sum_{i=1}^{N} \ln \frac{j_i}{j_{\min}} \right]^{-1},\tag{3}
$$

where $\hat{\varepsilon}$ is the convention used to denote that the exponent is an estimate [\[37–39\]](#page-5-0). The standard error of $\hat{\varepsilon}$ is

Standard error,
$$
\sigma = \frac{\hat{\varepsilon} - 1}{\sqrt{N}}
$$
 + higher-order terms, (4)

where the higher-order terms are positive [\[37\]](#page-5-0).

Not all data points are power-law distributed in experimental settings. Due to the limitations of the apparatus, low signals may be undercounted due to saturation effects [\[38\]](#page-5-0), which leads to an unknown lower bound J_{min} in Eq. (3) [\[37,38\]](#page-5-0). By guessing the value of J_{min} and discarding any jerk data J less than J_{min} , one runs the risk of $[37]$

(1) Underestimating *J*min, performing a power-law ML estimate to non-power-law *J* data.

(2) Overestimating *J*min, discarding valid data points and increasing the statistical error of $\hat{\varepsilon}$.

Thus to fully utilize the ML estimation, a computational algorithm of Eq. (3) is iterated on the jerk data by a series of cutoff values J_0 (that replaces J_{min}) [\[37,38\]](#page-5-0). For each J_0 value, data points less than J_0 are discarded and an estimated exponent $\hat{\varepsilon}$ is computed. This results in an array of $\hat{\varepsilon}$ which is plotted against the natural-log transformed $ln(J_0)$ (see Fig. 3). This method is called the *lower-bound estimation* [\[37\]](#page-5-0) but is

FIG. 3. ML fit of PIC 151 (blue solid line) and PIC 255 (red dashed line). PIC 151 displays a plateau of two decades at $-2.0 \le$ $ln(J_0) \le 0.0$ with an average estimated exponent $\hat{\epsilon}$ of 1.73. PIC 255 shows a plateau of three decades at $-3.0 \leq \ln(J_0) \leq 0.0$ with $\hat{\varepsilon}$ of 1.61. Both averaged exponents have a standard deviation σ of ± 0.04 . The lack of definitive kinks at $J_0 = J_{\text{min}}$ for both ML curves was due to lower cutoff effects that arose from saturation effects in the measurement apparatus [\[38\]](#page-5-0).

conventionally understood as performing the Maximum Likelihood Estimation/Method/Fit [\[6,16](#page-4-0)[,38,40\]](#page-5-0). The end result is a graph that has a broad plateau at the most likely exponent, which starts with a kink at the correct $J_0 = J_{\text{min}}$ value and extends for a range of J_0 [\[37,38\]](#page-5-0). The plateau then fluctuates at large J_0 values and the error bar increases due to a lack of statistically relevant data points [\[38\]](#page-5-0). If there is no plateau, the assumption of a single-exponent power law is invalid.

In experimental settings, ML curves and log-log binning plots are the keys to understanding an avalanche system. Many subtleties of a system can be uncovered using ML analysis, which is described in Ref. [\[38\]](#page-5-0). Barkhausen dynamics in ferroelectric domains are not expected to be the same as in magnetic domains. This is because the domain wall thicknesses differ by $\times 100 - \times 1000$, they are more Ising-like in ferroelectrics, and the interaction is very long range.

The jerk peaks *J* for PIC 151 and PIC 255 were subjected to ML analyses (see Fig. 3). PIC 151 (blue solid curve) exhibits a plateau of two decades at $-2.0 \leq \ln(J_0) \leq 0.0$ and the averaged exponent $\hat{\epsilon}$ at this region yields 1.73. The plateau for PIC 255 (red dashed line) extends for three decades at $-3.0 \leq \ln(J_0) \leq 0.0$ with an averaged exponent $\hat{\epsilon}$ of 1.61. The standard deviations σ for both averaged exponents are ± 0.04 .

The values for both $\hat{\epsilon}$ and σ for each ML analysis depended highly on where the plateaus were defined and would be biased. Unlike ideal power laws, there were no definitive kinks to determine the value of J_{min} and where the plateau started due to the lower cutoff effect [\[38\]](#page-5-0). The lower cutoff effect is also observed in the two jerk distributions in Fig. 2. The straight-line behavior extended until $ln(J_{\text{norm}}) \approx -2$ and followed by a smooth cutoff due to saturation effects [\[38\]](#page-5-0).

FIG. 4. *P*(*J*) versus *J* for three different temperatures, showing power-law mixing at temperatures near 413 K (140 °C). An attempt to fit these distributions with a straight line will lead to an averaging effect of two exponents [\[38\]](#page-5-0).

An additional PIC 151 sample (151b) was studied and a total of 2722 peaks were extracted. The jerk spectrum exhibits intense peaks that are at least one magnitude larger than the usual jerk peaks seen in the former two samples and may represent spanning avalanches [\[34\]](#page-5-0). As mentioned by He *et al.* in Ref. [\[34\]](#page-5-0), these peaks do not play a role in avalanche statistics and 41 peaks above an arbitrarily set cutoff value were removed. The ML estimate over the remaining 2681 jerk data yields an exponent $\hat{\epsilon}$ of 1.64 with $\sigma = 0.04$ when averaged over a plateau at region $-2.7 \leq \ln(J_0) \leq 0.0$.

Statistical noise from electronic systems will influence the ML curve, causing an increase in the measured exponent. Thus, based on both the linear fit of the log-log power-law distribution and the ML curve, we estimated that the exponent should be 1.64 ± 0.15 to take the random electrical noise effects into account.

Existing avalanche theory makes no direct prediction about the temperature dependence of Barkhausen pulse dynamics besides the invariance of the mean field results. However, we may expect that these dynamics change at ultra-low temperatures where creep freezes out and is replaced by quantum tunneling of domain walls [\[41\]](#page-5-0). At high temperatures near *T*^c the wall mobility increases dramatically and domain sizes decrease, which may potentially lead to deviations from the mean field behavior.

Figure 4 shows the behavior of Barkhausen dynamics as *T* is increased towards *T*_c. At \sim *T* = 410 K, the straight-line behavior of the power-law exponent 1.65 for $P(J)$ against *J* gives way to a superposition of two separate exponential distributions, one of which is greater at 1.80−2.00.

The maximum likelihood analysis reveals the power-law mixing effect of two simultaneous exponents. From Fig. 5, the initial kinks of the ML curves approach but underestimate the first exponent. They then decrease to the value of the second lower exponent before fluctuating due to a lack of statistically relevant data. In other words, the jerk peaks extracted from

FIG. 5. Maximum likelihood graphs at three different temperatures, verifying the power-law mixing effect from the slopes in Fig. 4. The initial kinks of the ML curves underestimate the first critical exponent, then decrease to the value of the second critical exponent.

higher temperatures were generated from two different powerlaw mechanisms. The second, smaller exponent remains near 1.6 and is independent of temperature.

We suggest an explanation for this: PZT is known to have a very high density of dislocations (\sim 10¹³ cm⁻²) even in very good samples, about an order of magnitude greater than in BaTiO₃, and neighboring dislocations \sim 12 nm apart [\[26](#page-4-0)[,27\]](#page-5-0). These are mostly threading dislocations [\[28\]](#page-5-0).

It is known empirically that depinning from dislocations gives an exponent near 2.0 [\[29\]](#page-5-0). Our tentative hypothesis for the second, higher exponent is the passage through the temperature region in which ferroelectric domain walls depin from threading dislocations. Depinning temperatures significantly below T_c are well known in other ferroelectrics: CsD_2AsO_4 [\[42,43\]](#page-5-0). In magnets, Bohn *et al.* have demonstrated [\[44–46\]](#page-5-0) different universality classes in Barkhausen pulses. These emphasize dimensionality as well as thicknesses of domains and explicitly include exponent crossover as fields or temperature increases, domains become smaller, and depinning becomes different from occasional creep. This could give rise to two exponents. However, the comparison of both hard and soft PZT in the present work makes this hypothesis unlikely in our case. Most importantly, the switching exponent $\varepsilon = 1.64$ remains independent of temperature.

IV. CONCLUSION

From two commercial PZT samples albeit with a low sampling rate, we showed that the jerks obeyed power-law statistics and were consistent with Barkhausen noise [\[6,12\]](#page-4-0). The distribution of Barkhausen noises obeys a power-law and the determined exponents with an estimated value of 1.64 ± 0.15 are in agreement with values from avalanche theory. The present ambient data yield a power exponent of 1.64 ± 0.15 . This is a highly reproducible random error. We alert readers that weak non-Barkhausen pulses due to other noise sources could lower this. We estimate the maximum

systematic error as possibly lowering the exponent to 1.55. In defense of the value 1.64 we note that noise measurements on BaTiO3 by Salje *et al.* [1] give exactly 1.64, and nonnoise antiferromagnetic measurements by Paruch [\[49\]](#page-5-0) give 1.6. Moreover, Salje *et al.* [1] measure six different exponents, and these are interrelated in a self-consistent way compatible with avalanche.

This observation, when compared with the expected mean field theory (MFT) exponent 1.33 $[6,14,22-25]$ for single avalanches and 1.67 for field-integrated models, indicates that the polarization switching follows a field-integrated pathway. This indicates that after each pinning or depinning event an exponential relaxation follows as part of the avalanche. We conclude that the switching processes in our PZT samples take the form of complex avalanches and are consistent with Barkhausen noise and many other crackling systems. The discovery and categorization of these exponents are crucial as they show that the domain switching mechanisms in ferroelectric ceramics demonstrate criticality [2[,47,48\]](#page-5-0). Other exponents for the amplitudes and, most importantly, for the interevent times for domain switches, cannot be obtained from our measurements [1]. Nevertheless, these parameters were measured $[1]$ for the domain switching in BaTiO₃ and we may assume that the values in PZT follow those of $BaTiO₃$ very closely. Future work on the acoustic emission in PZT is highly desirable to identify such full sets of switching parameters and to compare with other ferroelectric materials. Unlike sand grains in sand piles, ferroelectric domains rapidly decrease domain size and rapidly increase mobility as the Curie temperature T_c is approached from below. This might explain why we get two power exponents as *T* is increased, one of which is ∼2.0. We note that Paruch [\[49\]](#page-5-0) also finds two exponents, one approximately 2.0, in some of her atomic force microscopy (AFM) domain wall data.

Finally, there is no evidence in the prior literature that ferroelectric Barkhausen avalanche fits (self-organizing criticality (SOC) universality class, or even that it is SOC. That is premature and why our paper(s) are good beginnings.

Note added in proof. Recently, a talk was given by Prof. P. Paruch of the University of Geneva, which confirms our exponents in an independent way [\[49\]](#page-5-0). She measured the power exponent by AFM imaging without electrical noise or acoustic measurements and recorded jerk movement of ferroelectric domains at ambient temperatures with a power law of 1.6. In addition, under some conditions of field and temperature this separated into two exponents, one of which is near 2.0; in each case these values verify ours.

ACKNOWLEDGMENTS

This work at St. Andrews was supported by the Engineering and Physical Sciences Research Council (EPSRC) Grant No. EP/P024637/1. E.K.H.S. thanks the EPSRC for support.

- [1] E. K. H. Salje, D. Xue, X. Ding, K. A. Dahmen, and J. F. Scott, [Phys. Rev. Mater.](https://doi.org/10.1103/PhysRevMaterials.3.014415) **[3](https://doi.org/10.1103/PhysRevMaterials.3.014415)**, [014415](https://doi.org/10.1103/PhysRevMaterials.3.014415) [\(2019\)](https://doi.org/10.1103/PhysRevMaterials.3.014415).
- [2] P. Bak, C. Tang, and K. Wiesenfeld, [Phys. Rev. A](https://doi.org/10.1103/PhysRevA.38.364) **[38](https://doi.org/10.1103/PhysRevA.38.364)**, [364](https://doi.org/10.1103/PhysRevA.38.364) [\(1988\)](https://doi.org/10.1103/PhysRevA.38.364).
- [3] E. K. H. Salje, X. Ding, Z. Zhao, T. Lookman, and A. Saxena, [Phys. Rev. B](https://doi.org/10.1103/PhysRevB.83.104109) **[83](https://doi.org/10.1103/PhysRevB.83.104109)**, [104109](https://doi.org/10.1103/PhysRevB.83.104109) [\(2011\)](https://doi.org/10.1103/PhysRevB.83.104109).
- [4] R. J. Harrison and E. K. H. Salje, [Appl. Phys. Lett.](https://doi.org/10.1063/1.3460170) **[97](https://doi.org/10.1063/1.3460170)**, [021907](https://doi.org/10.1063/1.3460170) [\(2010\)](https://doi.org/10.1063/1.3460170).
- [5] S. Puchberger, V. Soprunyuk, W. Schranz, A. Tröster, K. [Roleder, A. Majchrowski, M. A. Carpenter, and E. Salje,](https://doi.org/10.1063/1.4979616) APL Mater. **[5](https://doi.org/10.1063/1.4979616)**, [046102](https://doi.org/10.1063/1.4979616) [\(2017\)](https://doi.org/10.1063/1.4979616).
- [6] [E. K. H. Salje and K. A. Dahmen,](https://doi.org/10.1146/annurev-conmatphys-031113-133838) Annu. Rev. Condens. Matter Phys. **[5](https://doi.org/10.1146/annurev-conmatphys-031113-133838)**, [233](https://doi.org/10.1146/annurev-conmatphys-031113-133838) [\(2014\)](https://doi.org/10.1146/annurev-conmatphys-031113-133838).
- [7] V. M. Rudyak, [Sov. Phys. Usp.](https://doi.org/10.1070/PU1971v013n04ABEH004681) **[13](https://doi.org/10.1070/PU1971v013n04ABEH004681)**, [461](https://doi.org/10.1070/PU1971v013n04ABEH004681) [\(1971\)](https://doi.org/10.1070/PU1971v013n04ABEH004681).
- [8] R. Tebble, IOP Sci. **2**, 1017 (1955).
- [9] E. K. H. Salje, [ChemPhysChem](https://doi.org/10.1002/cphc.200900943) **[11](https://doi.org/10.1002/cphc.200900943)**, [940](https://doi.org/10.1002/cphc.200900943) [\(2010\)](https://doi.org/10.1002/cphc.200900943).
- [10] D. Damjanovic, [Rep. Prog. Phys.](https://doi.org/10.1088/0034-4885/61/9/002) **[61](https://doi.org/10.1088/0034-4885/61/9/002)**, [1267](https://doi.org/10.1088/0034-4885/61/9/002) [\(1998\)](https://doi.org/10.1088/0034-4885/61/9/002).
- [11] J. P. Sethna, K. A. Dahmen, and C. R. Myers, [Nature \(London\)](https://doi.org/10.1038/35065675) **[410](https://doi.org/10.1038/35065675)**, [242](https://doi.org/10.1038/35065675) [\(2001\)](https://doi.org/10.1038/35065675).
- [12] K. A. Dahmen and Y. Ben-Zion, *Encyclopedia of Complexity and Systems Science* (Springer, New York, 2009), pp. 5021– 5037.
- [13] K. Dahmen and J. P. Sethna, [Phys. Rev. B](https://doi.org/10.1103/PhysRevB.53.14872) **[53](https://doi.org/10.1103/PhysRevB.53.14872)**, [14872](https://doi.org/10.1103/PhysRevB.53.14872) [\(1996\)](https://doi.org/10.1103/PhysRevB.53.14872).
- [14] K. A. Dahmen, Understanding complex systems, in *Avalanches in Functional Materials and Geophysics*, edited by E. K. Salje, A. Saxena, and A. Planes (Springer International Publishing, Cham, 2017), pp. 19–30.
- [15] J. Baró, A. Corral, X. Illa, A. Planes, E. K. H. Salje, W. Schranz, D. E. Soto-Parra, and E. Vives, [Phys. Rev. Lett.](https://doi.org/10.1103/PhysRevLett.110.088702) **[110](https://doi.org/10.1103/PhysRevLett.110.088702)**, [088702](https://doi.org/10.1103/PhysRevLett.110.088702) [\(2013\)](https://doi.org/10.1103/PhysRevLett.110.088702).
- [16] X. Jiang, H. Liu, I. G. Main, and E. K. Salje, [Phys. Rev. E](https://doi.org/10.1103/PhysRevE.96.023004) **[96](https://doi.org/10.1103/PhysRevE.96.023004)**, [023004](https://doi.org/10.1103/PhysRevE.96.023004) [\(2017\)](https://doi.org/10.1103/PhysRevE.96.023004).
- [17] E. K. H. Salje, [Am. Mineral.](https://doi.org/10.2138/am-2015-5085) **[100](https://doi.org/10.2138/am-2015-5085)**, [343](https://doi.org/10.2138/am-2015-5085) [\(2015\)](https://doi.org/10.2138/am-2015-5085).
- [18] R. L. Goldberg and S. W. Smith, [Ultrason. Imaging](https://doi.org/10.1177/016173469501700202) **[17](https://doi.org/10.1177/016173469501700202)**, [95](https://doi.org/10.1177/016173469501700202) [\(1995\)](https://doi.org/10.1177/016173469501700202).
- [19] N. C. Pisenti, A. Restelli, B. J. Reschovsky, D. S. Barker, and G. K. Campbell, [Rev. Sci. Instrum.](https://doi.org/10.1063/1.4969059) **[87](https://doi.org/10.1063/1.4969059)**, [124702](https://doi.org/10.1063/1.4969059) [\(2016\)](https://doi.org/10.1063/1.4969059).
- [20] M. Sayer, C. V. R. V. Kumar, F. V. Letelier, J. Hazell, M. Watt, and P. Donders, [Integr. Ferroelectr.](https://doi.org/10.1080/10584589408018682) **[5](https://doi.org/10.1080/10584589408018682)**, [79](https://doi.org/10.1080/10584589408018682) [\(1994\)](https://doi.org/10.1080/10584589408018682).
- [21] *Piezoelectric Ceramic Products Brochure*, PI ceramic GmbH. Available: https://www.piceramic.com/en/products/ [piezoceramic-materials. Accessed 21 April 2018.](https://www.piceramic.com/en/products/piezoceramic-materials)
- [22] N. Friedman, A. T. Jennings, G. Tsekenis, J.-Y. Kim, M. Tao, J. T. Uhl, J. R. Greer, and K. A. Dahmen, [Phys. Rev. Lett.](https://doi.org/10.1103/PhysRevLett.109.095507) **[109](https://doi.org/10.1103/PhysRevLett.109.095507)**, [095507](https://doi.org/10.1103/PhysRevLett.109.095507) [\(2012\)](https://doi.org/10.1103/PhysRevLett.109.095507).
- [23] Y. Ben-Zion, K. A. Dahmen, and J. T. Uhl, [Pure Appl. Geophys.](https://doi.org/10.1007/s00024-011-0273-7) **[168](https://doi.org/10.1007/s00024-011-0273-7)**, [2221](https://doi.org/10.1007/s00024-011-0273-7) [\(2011\)](https://doi.org/10.1007/s00024-011-0273-7).
- [24] K. A. Dahmen, Y. Ben-Zion, and J. T. Uhl, [Phys. Rev. Lett.](https://doi.org/10.1103/PhysRevLett.102.175501) **[102](https://doi.org/10.1103/PhysRevLett.102.175501)**, [175501](https://doi.org/10.1103/PhysRevLett.102.175501) [\(2009\)](https://doi.org/10.1103/PhysRevLett.102.175501).
- [25] G. Tsekenis, J. T. Uhl, N. Goldenfeld, and K. A. Dahmen, [EPL](https://doi.org/10.1209/0295-5075/101/36003) **[101](https://doi.org/10.1209/0295-5075/101/36003)**, [36003](https://doi.org/10.1209/0295-5075/101/36003) [\(2013\)](https://doi.org/10.1209/0295-5075/101/36003).
- [26] V. Nagarajan, C. L. Jia, H. Kohlstedt, R. Waser, I. B. Misirlioglu, S. P. Alpay, and R. Ramesh, [Appl. Phys. Lett.](https://doi.org/10.1063/1.1922579) **[86](https://doi.org/10.1063/1.1922579)**, [192910](https://doi.org/10.1063/1.1922579) [\(2005\)](https://doi.org/10.1063/1.1922579).
- [27] C. L. Canedy, H. Li, S. P. Alpay, L. Salamanca-Riba, A. L. Roytburd, and R. Ramesh, [Appl. Phys. Lett.](https://doi.org/10.1063/1.1308531) **[77](https://doi.org/10.1063/1.1308531)**, [1695](https://doi.org/10.1063/1.1308531) [\(2000\)](https://doi.org/10.1063/1.1308531).
- [28] I. Vrejoiu, G. Le Rhun, N. D. Zakharov, D. Hesse, L. Pintilie, and M. Alexe, [Philos. Mag.](https://doi.org/10.1080/14786430600728653) **[86](https://doi.org/10.1080/14786430600728653)**, [4477](https://doi.org/10.1080/14786430600728653) [\(2006\)](https://doi.org/10.1080/14786430600728653).
- [29] P. O. Castillo-Villa, J. Baró, A. Planes, E. K. H. Salje, P. [Sellappan, W. M. Kriven, and E. Vives,](https://doi.org/10.1088/0953-8984/25/29/292202) J. Phys.: Condens. Matter **[25](https://doi.org/10.1088/0953-8984/25/29/292202)**, [292202](https://doi.org/10.1088/0953-8984/25/29/292202) [\(2013\)](https://doi.org/10.1088/0953-8984/25/29/292202).
- [30] *aixACCT TF Analyzer 2000*. Available: http://www.aixacct. [com/html/prod/memprods-tf2000.php, Accessed 23 January](http://www.aixacct.com/html/prod/memprods-tf2000.php) 2018.
- [31] *TF Analyzer 2000 Hysteresis Software*, 2nd ed. (aixACCT Systems GmbH, Aachen), p. 174.
- [32] J. F. Scott, [Integr. Ferroelectr.](https://doi.org/10.1080/10584589608013050) **[12](https://doi.org/10.1080/10584589608013050)**, [71](https://doi.org/10.1080/10584589608013050) [\(1996\)](https://doi.org/10.1080/10584589608013050).
- [33] D. Viehland and Y. H. Chen, [J. Appl. Phys.](https://doi.org/10.1063/1.1325001) **[88](https://doi.org/10.1063/1.1325001)**, [6696](https://doi.org/10.1063/1.1325001) [\(2000\)](https://doi.org/10.1063/1.1325001).
- [34] X. He, X. Ding, J. Sun, and E. K. H. Salje, [Appl. Phys. Lett.](https://doi.org/10.1063/1.4942387) **[108](https://doi.org/10.1063/1.4942387)**, [072904](https://doi.org/10.1063/1.4942387) [\(2016\)](https://doi.org/10.1063/1.4942387).
- [35] [A. Charalambous, X. Yuan, and N. McNeill,](https://doi.org/10.1109/TPEL.2017.2743041) IEEE Trans. Power Electron. **[33](https://doi.org/10.1109/TPEL.2017.2743041)**, [5660](https://doi.org/10.1109/TPEL.2017.2743041) [\(2018\)](https://doi.org/10.1109/TPEL.2017.2743041).
- [36] B. Carter, *Op Amps for Everyone* (Elsevier, Oxford, 2013), pp. 195–223.
- [37] A. Clauset, C. R. Shalizi, and M. E. J. Newman, [SIAM Rev.](https://doi.org/10.1137/070710111) **[51](https://doi.org/10.1137/070710111)**, [661](https://doi.org/10.1137/070710111) [\(2009\)](https://doi.org/10.1137/070710111).
- [38] E. K. H. Salje, A. Planes, and E. Vives, [Phys. Rev. E](https://doi.org/10.1103/PhysRevE.96.042122) **[96](https://doi.org/10.1103/PhysRevE.96.042122)**, [042122](https://doi.org/10.1103/PhysRevE.96.042122) [\(2017\)](https://doi.org/10.1103/PhysRevE.96.042122).
- [39] J. A. Rice, *Mathematical Statistics and Data Analysis*, 3rd ed. (Thomson Brooks/Cole, Belmont, CA, 2007), p. 688.
- [40] M. C. Gallardo, J. Manchado, F. J. Romero, J. Del Cerro, E. K. [Salje, A. Planes, E. Vives, R. Romero, and M. Stipcich,](https://doi.org/10.1103/PhysRevB.81.174102) Phys. Rev. B **[81](https://doi.org/10.1103/PhysRevB.81.174102)**, [174102](https://doi.org/10.1103/PhysRevB.81.174102) [\(2010\)](https://doi.org/10.1103/PhysRevB.81.174102).
- [41] F. Kagawa, S. Horiuchi, and Y. Tokura, [Crystals](https://doi.org/10.3390/cryst7040106) **[7](https://doi.org/10.3390/cryst7040106)**, [106](https://doi.org/10.3390/cryst7040106) [\(2017\)](https://doi.org/10.3390/cryst7040106).
- [42] A. A. Baski, W. F. Oliver, and J. F. Scott, [Ferroelectr. Lett. Sect.](https://doi.org/10.1080/07315178708200547) **[7](https://doi.org/10.1080/07315178708200547)**, [171](https://doi.org/10.1080/07315178708200547) [\(1987\)](https://doi.org/10.1080/07315178708200547).
- [43] A. Sidorkin, N. Burdanina, and L. Kamysheva, Sov. Phys. Solid State **26**, 1910 (1984).
- [44] F. Bohn, G. Durin, M. A. Correa, N. R. Machado, R. D. Della Pace, C. Chesman, and R. L. Sommer, [Sci. Rep.](https://doi.org/10.1038/s41598-018-29576-3) **[8](https://doi.org/10.1038/s41598-018-29576-3)**, [11294](https://doi.org/10.1038/s41598-018-29576-3) [\(2018\)](https://doi.org/10.1038/s41598-018-29576-3).
- [45] G. Z. d. S. Lima, G. Corso, M. A. Correa, R. L. Sommer, P. C. Ivanov, and F. Bohn, [Phys. Rev. E](https://doi.org/10.1103/PhysRevE.96.022159) **[96](https://doi.org/10.1103/PhysRevE.96.022159)**, [022159](https://doi.org/10.1103/PhysRevE.96.022159) [\(2017\)](https://doi.org/10.1103/PhysRevE.96.022159).
- [46] [A. P. Mehta, A. C. Mills, K. A. Dahmen, and J. P. Sethna,](https://doi.org/10.1103/PhysRevE.65.046139) Phys. Rev. E **[65](https://doi.org/10.1103/PhysRevE.65.046139)**, [046139](https://doi.org/10.1103/PhysRevE.65.046139) [\(2002\)](https://doi.org/10.1103/PhysRevE.65.046139).
- [47] P. Bak, C. Tang, and K. Wiesenfeld, [Phys. Rev. Lett.](https://doi.org/10.1103/PhysRevLett.59.381) **[59](https://doi.org/10.1103/PhysRevLett.59.381)**, [381](https://doi.org/10.1103/PhysRevLett.59.381) [\(1987\)](https://doi.org/10.1103/PhysRevLett.59.381).
- [48] P. Bak and K. Chen, [Sci. Am.](https://doi.org/10.1038/scientificamerican0191-46) **[264](https://doi.org/10.1038/scientificamerican0191-46)**, [46](https://doi.org/10.1038/scientificamerican0191-46) [\(1991\)](https://doi.org/10.1038/scientificamerican0191-46).
- [49] P. Paruch, Proceedings of the Conference on Ferroics 2019, Tampa, Florida, 28 January 2019.