

Magnetic structural unit with convex geometry: A building block hosting an exchange-striction-driven magnetoelectric coupling

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We perform a combined experimental and theoretical study of a magnetic-field (B)-induced evolution of magnetic and ferroelectric properties in an antiferromagnetic material $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$, whose structure is characterized by a staggered array of Cu_4O_{12} magnetic units with convex geometry known as square cupola. Our experiments show a B -induced phase transition from a previously reported low- B linear magnetoelectric phase to a high- B magnetoelectric phase, which accompanies a 90° flop of electric polarization and gigantic magnetodielectric effect. Moreover, we observe a B -induced sign reversal of ferroelectric polarization in the high- B phase. Our model and first-principles calculations reveal that the observed complex magnetoelectric behavior is well explained in terms of a B -dependent electric polarization generated in each Cu_4O_{12} unit by the so-called exchange-striction mechanism. The present study demonstrates that the materials design based on the magnetic structural unit with convex geometry deserves to be explored for developing strong magnetoelectric couplings.

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I. INTRODUCTION

Magnetoelectric multiferroics, in which magnetic and ferroelectric orders coexist, are an important class of materials because their unique and strong magnetoelectric couplings provide numerous potential applications such as novel magneto-optical devices and antiferromagnetic spintronics devices [1–9]. Recently, designing magnetoelectric multiferroic materials based on structural units such as specific molecules or transition-metal ion clusters has been extensively studied. Experimentally, many molecular-based multiferroic materials have been found particularly in metal-organic hybrid systems [10–12]. In most of them, their magnetic order is provided by magnetic moments of transition-metal ions while their ferroelectric order is associated with an order-disorder transition of organic molecular units. Because of this different origin of magnetic and ferroelectric orders, however, their magnetoelectric coupling is generally weak [13] and hence a drastic response of an electric polarization (magnetization) to an external magnetic (electric) field has been merely observed.

A promising way to enhance magnetoelectric couplings is already known from extensive studies on magnetoelectric multiferroic inorganic oxides, where magnetic order itself generates an electric polarization [3]. Three types of mech-

anisms for an induced polarization are well established: the spin-current mechanism [14] (or the inverse Dzyaloshinskii-Moriya (DM) interaction mechanism [15]), the metal-ligand d - p hybridization mechanism [16], and the exchange-striction mechanism [17]. The former two mechanisms are associated with a weak relativistic spin-orbit interaction, which usually yields a small polarization. On the other hand, the exchange-striction mechanism, not involving the spin-orbit interaction, potentially generates a much larger polarization, as observed in perovskite manganites such as pressurized RMnO_3 ($R = \text{Tb}$, Dy , and Gd) [18–20]. Therefore, the use of magnetic structural units, where the exchange-striction mechanism is active, is a key to designing a material with a strong magnetoelectric coupling. Theoretically, a system consisting of magnetic trimer molecules was proposed to exhibit an exchange-striction-driven strong magnetoelectric coupling [21]. However, this proposal has not been confirmed experimentally.

In this paper, we consider a magnetic structural unit with convex geometry known as square cupola, depicted in Fig. 1. It consists of four corner-sharing MX_4 plaquettes, where M is a magnetic ion carrying a spin and X is an anionic ligand. Notably, this unit can be found in a wide variety of systems ranging from minerals [22] and salt-inclusion compounds [23] to metal-organic hybrid systems [24]. A peculiar noncoplanar spin arrangement was recently found in the family of square-cupola-based antiferromagnets $A(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ ($A = \text{Ba}$, Sr , and Pb) (see Fig. 2) [25–27]. The ab -plane spin component

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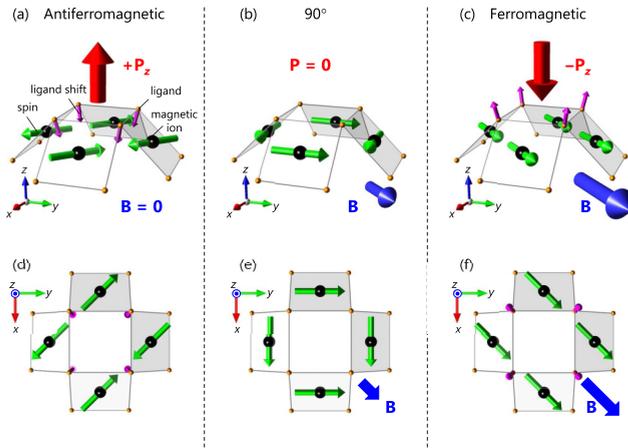


FIG. 1. Proposal for a magnetically controllable electric polarization via the exchange striction using a convex square-cupola spin cluster. The black and orange balls represent the magnetic ion and ligand (anion), respectively. The green arrows denote a spin. The pink arrows indicate a ligand shift due to an exchange-striction effect. This shift induces an electric dipole along the opposite direction. The electric polarization (\mathbf{P}) appears parallel and antiparallel to the convex direction for (a) antiferromagnetic (AFM) and (c) ferromagnetic (FM) cases, respectively, while no \mathbf{P} appears for (b) the 90° arrangement case. Therefore, when the spin arrangement is changed from AFM to FM by a magnetic field (\mathbf{B}), the direction of \mathbf{P} should be reversed. For clarity, the top view illustrations of (a)–(c) are shown in (d)–(f), respectively.

can be regarded as a magnetic quadrupole moment which is a source of linear magnetoelectric effects—linear induction of electric polarization (magnetization) by a magnetic field (electric field). In fact, in $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$, the macroscopic electric polarization *normal* to the convex direction (parallel to the ab plane) was observed in a magnetic field applied along the ab plane [27]. However, the value of the polarization is as small as $\sim 60 \mu\text{C}/\text{m}^2$ at a magnetic field of 9 T and the microscopic mechanism has not been identified. Here, we propose a different magnetoelectric response of the square-cupola spin cluster; that is, an electric polarization due to the exchange-striction mechanism emerges *along* the convex direction (c axis), whose sign can be controlled by a magnetic field. Targeting $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ as a model material, we successfully verify this proposal by high-field measurements of magnetization and electric polarization, collaborated with the analysis of an effective spin model and first-principles calculations. We thus obtain the following finding: a magnetic structural unit with convex geometry is a promising building block hosting an exchange-striction-driven magnetoelectric coupling.

The rest of the paper is organized as follows. In Sec. II, we explain our proposal for the magnetically controllable electric polarization due to the exchange-striction mechanism. In Sec. III, we describe the crystal structure and the previously reported magnetic properties of the target material $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$. In Sec. IV, we describe the experimental and theoretical methods used in the present study. We present our experimental and theoretical results in Sec. V and summarize our findings in Sec. VI.

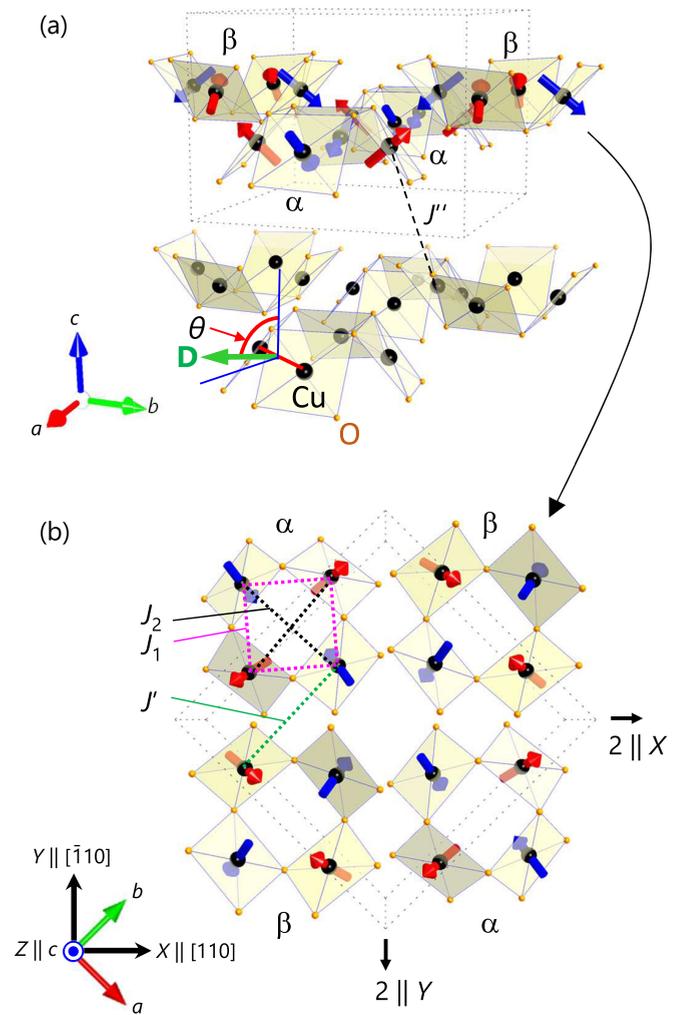


FIG. 2. Crystal and magnetic structure of $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$. (a) The bird's-eye view of the structure illustrating a layered crystal structure composed of two types of Cu_4O_{12} square cupolas labeled α (upward) and β (downward). The black and orange balls denote Cu and O ions, respectively. The spin arrangement without an external magnetic field is illustrated only for the upper layer, where the red and blue arrows indicate spins with positive and negative c -axis ($\parallel Z$ -axis) components, respectively. The same spin arrangement appears in all other layers. The gray dotted line represents a unit cell. The interlayer coupling J'' is indicated by a black dashed line. The DM vector (\mathbf{D}) considered in the theoretical model is shown in the lower layer, which makes an angle θ from the c axis (see text). (b) The top view of the upper layer. The definition of the X , Y , and Z axes used in the text is illustrated. Each layer is characterized by the staggered arrangement of α and β . When the spin arrangement is not considered (i.e., in the paramagnetic phase), α and β are mutually converted by the twofold rotational symmetry along the X and Y axes ($2 \parallel X$ and $2 \parallel Y$). The three dominant exchange couplings, intraclusters J_1 and J_2 , and intercluster J' are indicated.

II. PROPOSAL FOR EXCHANGE-STRICTION-DRIVEN ELECTRIC POLARIZATION

Our proposed idea is summarized in Fig. 1. It should be noted that because of the convex geometry of the square cupola, there is a structurally fixed electric dipole along the

convex direction (direction of the $+z$ direction in Fig. 1). This electric dipole is not controllable by an external field and therefore out of our scope.

To explain the onset of an electric dipole induced by the exchange-striction mechanism [17], let us consider two magnetic ions i and j connected by a ligand with a bond angle ϕ , whose spins \mathbf{S}_i and \mathbf{S}_j are coupled via a superexchange interaction expressed as $J\mathbf{S}_i \cdot \mathbf{S}_j$. Because the exchange constant J strongly depends on ϕ , the ligand would shift in such a way to gain the superexchange energy in response to a given spin arrangement, where the shift should be dominated by $\mathbf{S}_i \cdot \mathbf{S}_j$. This is known as exchange striction. Then, this shift of the negatively charged ligand gives rise to an electric dipole along its counterdirection. According to the Goodenough-Kanamori-Anderson rules, J becomes more strongly antiferromagnetic for ϕ closer to 180° , while it becomes ferromagnetic for ϕ closer to 90° . This indicates that the sign of the ligand shift (and thus electric dipole) is reversed between the antiferromagnetic spin arrangement and the ferromagnetic one. Consequently, neglecting higher-order contributions, the electric dipole \mathbf{p}_{ij} generated by the neighboring two magnetic ions i and j is proportional to $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle$, the expectation value of the inner product of their spin operators \mathbf{S}_i and \mathbf{S}_j .

Now, let us apply this exchange-striction mechanism to a simple case, where the antiferromagnetic spin arrangement shown in Fig. 1(a) emerges on the square cupola. By summing \mathbf{p}_{ij} for all the nearest-neighbor spin pairs, we obtain the electric polarization \mathbf{P}^{SC} from the square-cupola unit, formulated as

$$\mathbf{P}^{\text{SC}} = \sum_{(i,j)} \mathbf{p}_{ij} = A \sum_{(i,j)} \mathbf{e}_{ij} \langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle. \quad (1)$$

Here, $A > 0$ is the constant depending on microscopic details of superexchange interactions and \mathbf{e}_{ij} is the unit vector that determines the direction of \mathbf{p}_{ij} . \mathbf{e}_{ij} points from the ligand site shared by magnetic ions i and j to the center of the bond between these magnetic ions. Significantly, owing to the convex geometry of the square cupola, every spin pair cooperatively generates \mathbf{p}_{ij} pointing to the convex direction with a small tilting. The tilted components normal to the convex direction cancel out with each other. As a result, a finite \mathbf{P}^{SC} appears along the convex direction.

It is immediately predicted from Eq. (1) that when the spins align ferromagnetically [Fig. 1(c)], the induced \mathbf{P}^{SC} is antiparallel to the convex direction. No \mathbf{P}^{SC} is expected at the intermediate state with a 90° spin arrangement [Fig. 1(b)]. As a result, \mathbf{P}^{SC} can be continuously controlled from the positive to negative direction by changing the spin arrangement with an applied magnetic field. These considerations suggest that the square-cupola unit is a promising structural unit carrying a magnetically controllable polarization due to the exchange-striction mechanism. It is therefore expected that a material consisting of square-cupola units deserves to be explored for a large magnetoelectric coupling.

Although we explained our proposal using the very simple example of spin arrangements shown in Fig. 1, this proposal can be easily extended to more complex cases such as a noncoplanar spin arrangement. Moreover, not only ordered components of spins (i.e., $\langle \mathbf{S}_i \rangle \neq 0$), but also quantum spin fluctuations can induce \mathbf{P}^{SC} . An extreme example is a quan-

tum mechanical nonmagnetic singlet state which has $\langle \mathbf{S}_i \rangle = 0$, but $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle \neq 0$. Therefore, the present idea can be applied to various kinds of spin states.

III. TARGET MATERIAL $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$

In order to verify our proposal, we have targeted $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ as a model material [27]. The crystal structure belongs to a tetragonal nonpolar space group $P42_12$, which consists of a two-dimensional staggered array of upward and downward magnetic square-cupola clusters Cu_4O_{12} , as shown in Figs. 2(a) and 2(b). The upward and downward square cupolas, which we call α and β , respectively, are mutually converted by symmetry operation 2 (twofold rotation) along the $[110]$ and $[\bar{1}10]$ axes depicted in Fig. 2(b). In the following, we refer to the $[110]$, $[\bar{1}10]$, and $[001]$ axes as X , Y , and Z axes, respectively. The intercluster interaction (J') within the array is expected to be weaker than the nearest neighbor J_1 , so that the material can be regarded as a weakly coupled Cu_4O_{12} system. Of the various systems with square cupolas [22–27], this material benefits from an availability of sizable single crystals. Moreover, an effective spin Hamiltonian developed for isostructural $\text{Ba}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ [28] would be applicable to the present material. These benefits enable a detailed comparison between experiments and theory that is crucial for microscopic understanding of magnetoelectric couplings.

$\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ undergoes a magnetic ordering at $T_N \approx 7$ K without an external magnetic field. As illustrated in Figs. 2(a) and 2(b), the four spins of each square cupola form a peculiar “two-in, two-out” arrangement, where the Z -axis components of spins align in the antiferromagnetic up-down-up-down manner, while the XY -plane components rotate by 90° . As mentioned above, the XY -plane spin components can be regarded as a magnetic quadrupole moment providing a source for the linear magnetoelectric effect. The magnetic-field-induced electric polarization was indeed observed in the material and its direction is parallel to the XY plane (e.g., the electric polarization appears along the Y axis when a magnetic field is applied along the X axis).

When Eq. (1) is applied to the spin arrangement, one can expect that the Z -axis spin components induce a finite \mathbf{P}^{SC} along the Z axis in each square cupola. \mathbf{P}^{SC} for α and β is defined as \mathbf{P}^α and \mathbf{P}^β , respectively. In $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$, however, a macroscopic electric polarization $\mathbf{P} = \mathbf{P}^\alpha + \mathbf{P}^\beta$ was not observed because a relation $\mathbf{P}^\alpha = -\mathbf{P}^\beta$ is enforced due to the staggered arrangement of square cupolas α and β . Since the resultant staggered antiferroelectric polarization cannot be measured directly, it is insufficient to verify our present proposal. In the present study, we discover macroscopic \mathbf{P} along the Z axis in a different magnetic-field-induced phase, where the relation $\mathbf{P}^\alpha = -\mathbf{P}^\beta$ is broken, as we will see below.

IV. EXPERIMENTAL AND THEORETICAL METHODS

Single crystals of $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ were grown by the slow-cooling method [27]. Powder x-ray diffraction (XRD) measurements on crushed single crystals confirmed a single phase. The crystal orientation was determined by the Laue

x-ray method. A superconducting magnet system up to 18 T and down to 1.6 K at the Tohoku University was used for measurements of dielectric constant ε and \mathbf{P} . For the measurements of ε and \mathbf{P} , single crystals were cut into thin plates and, subsequently, an electrode was formed by painting silver pastes on a pair of the widest surfaces. Using an LCR meter (Agilent E4980), ε was measured at an excitation frequency of 100 kHz. \mathbf{P} was obtained by integrating a pyroelectric current measured with an electrometer (Keithley 6517). The measurements of magnetization \mathbf{M} and \mathbf{P} up to ~ 56 T were performed using a multilayer pulse magnet installed at the International MegaGauss Science Laboratory of the Institute for Solid State Physics at The University of Tokyo. \mathbf{M} was measured by the conventional induction method using coaxial pickup coils. \mathbf{P} was obtained by integrating the polarization current [29]. Multifrequency electron spin resonance (ESR) measurements (600–1400 GHz) in pulsed magnetic fields were performed to obtain the g values for the field directions along the [100], [110], and [001] directions. The g values were found to be isotropic within the experimental accuracy: $g \sim 2.20(5)$ for the three directions. This value is similar to that of the isostructural compound $\text{Ba}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ [28]. The crystal structures displayed in this paper were drawn using VESTA software [30].

To understand magnetoelectric properties obtained by the experiments, we carried out cluster mean-field (CMF) calculations of an effective spin model. In the analysis, we consider an effective spin model associated with $S = 1/2$ degree of freedom of a Cu^{2+} ion which was previously constructed in Ref. [28]. The model and the parameter setting are described in detail in Sec. V C. Even though the effective spin model is simple, an unbiased treatment is still difficult. Accordingly, we analyze the model using the CMF approximation. In the CMF treatment, the intracupola interactions are dealt with by the exact diagonalization so that the quantum effects within a cupola are fully taken into account, while the intercupola interactions are dealt with by the conventional mean-field approximation; that is, $\mathbf{S}_i \cdot \mathbf{S}_j$ is decoupled as $\mathbf{S}_i \cdot \mathbf{S}_j \simeq \langle \mathbf{S}_i \rangle \cdot \mathbf{S}_j + \mathbf{S}_i \cdot \langle \mathbf{S}_j \rangle - \langle \mathbf{S}_i \rangle \cdot \langle \mathbf{S}_j \rangle$. This approximation is suitable for cluster-based magnetic insulators with weaker intercluster interactions. Indeed, we successfully reproduced the magnetization curve and the dielectric anomaly observed in an isostructural of our target material, $\text{Ba}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ [28].

Density functional theory (DFT) calculations were also performed to estimate the magnitude of the magnetically induced electric polarization. The Vienna *ab initio* simulation package (VASP) [31] was used with a projector-augmented wave basis set. The electronic exchange and correlation were described by the Perdew-Burke-Ernzerhof generalized gradient approximation (PBE-GGA) [32]. The DFT + U method [33] was used for the correction of strongly correlated Cu-3d states, where the on-site Coulomb repulsion U_{eff} was set to 4 eV [27]. We first fully optimized the crystal structure of $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ starting from the experimental structure and then optimized atomic coordinates at each spin configuration given by the model calculations under the magnetic fields. The magnetically induced polarization was finally evaluated as the change of the polarization calculated by the Berry phase method [34,35].

V. RESULTS AND DISCUSSION

A. Experiments in 18 T superconducting magnet

Figure 3 summarizes the dielectric properties in the magnetic field applied along the X axis (B_X) below 18 T. As seen in Figs. 3(a) and 3(g), the application of B_X induces a sharp peak in ε along the Y axis (ε_Y), accompanying the onset of a finite \mathbf{P} along the same direction (P_Y). This behavior is consistent with the previous report [27], which results from the above-mentioned linear magnetoelectric effect due to the quadrupole-type spin arrangement. As we will see in the following sections, the exchange-striction mechanism given by Eq. (1) is able to reproduce the B_X -induced P_Y .

By further increasing B_X above 12 T, the ε_Y peak is suppressed and then completely disappears at $B_X = 18$ T, as shown in Fig. 3(a). Correspondingly, a different anomaly appears in ε along the Z axis (ε_Z). These results show that a phase transition between two different magnetoelectric states is induced by the application of B_X . The B_X dependence of ε_Y and ε_Z [Figs. 3(d) and 3(e)] reveals that the transition from the low-field (LF) to the field-induced (FI) phase occurs at the critical field $B_X^{\text{cl}} = 16.4$ T at $T = 2$ K. Notably, the B_X dependence of ε_Z reveals a remarkably large magnetodielectric effect, defined by $[\varepsilon_Z(B) - \varepsilon_Z(0)]/\varepsilon_Z(0)$, with the highest value of 180% at 2.4 K, as shown in the inset of Fig. 3(e). This value is comparable to the ‘‘colossal’’ magnetodielectric effect in some magnetoelectric multiferroic materials, e.g., $\sim 100\%$ for DyMn_2O_5 [36] and $\sim 500\%$ for DyMnO_3 [37], which indicates a very strong magnetoelectric coupling in the present material, $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$. The B_X versus T phase diagram constructed from ε_Z anomalies is drawn in Fig. 3(k).

Measurements of B_X effects on \mathbf{P} have elucidated the origin of the gigantic anomalies in ε . Figures 3(g) and 3(h) show the T profiles of \mathbf{P} along the Y (P_Y) and Z axes (P_Z), respectively, at B_X below and above B_X^{cl} . These data were taken on warming without an applied electric field (E) after the sample was cooled with $E = 0.67$ MV/m. As shown in Fig. 3(g), by applying $B_X = 18$ T $> B_X^{\text{cl}}$, P_Y is strongly suppressed to almost zero. In sharp contrast, P_Z appears [Fig. 3(h)] and its onset T is fully consistent with the phase diagram [Fig. 3(k)]. In addition, no anomaly associated with the transition is seen in ε and \mathbf{P} along the X axis (ε_X , P_X), as shown in Figs. 3(c), 3(f) and 3(i). These results demonstrate that the direction of \mathbf{P} is flopped from the Y to the Z axis by applying B_X . Notably, the direction of \mathbf{P} in the FI phase coincides with the one predicted by our proposal. However, as discussed in Sec. III, it is not straightforward to understand an origin for the emergence of the finite P_Z when considering the staggered arrangement of square cupolas α and β that are related by the twofold rotational operation about the X and Y axes (Fig. 2). As we will see later, a finite P_Z originates from breaking of this symmetry due to a spin arrangement that appears in the FI phase.

In order to examine whether the FI phase is ferroelectric, we have measured a PE hysteresis curve at $T = 2$ K. As shown in Fig. 3(j), P_Z can be reversed by applying E , evidencing a ferroelectricity. Note that the hysteresis curve is highly asymmetric with respect to E . It is known that this behavior is observed in ferroelectric thin films, which is called an imprint effect [38]. Consistently, a finite P_Z is observed

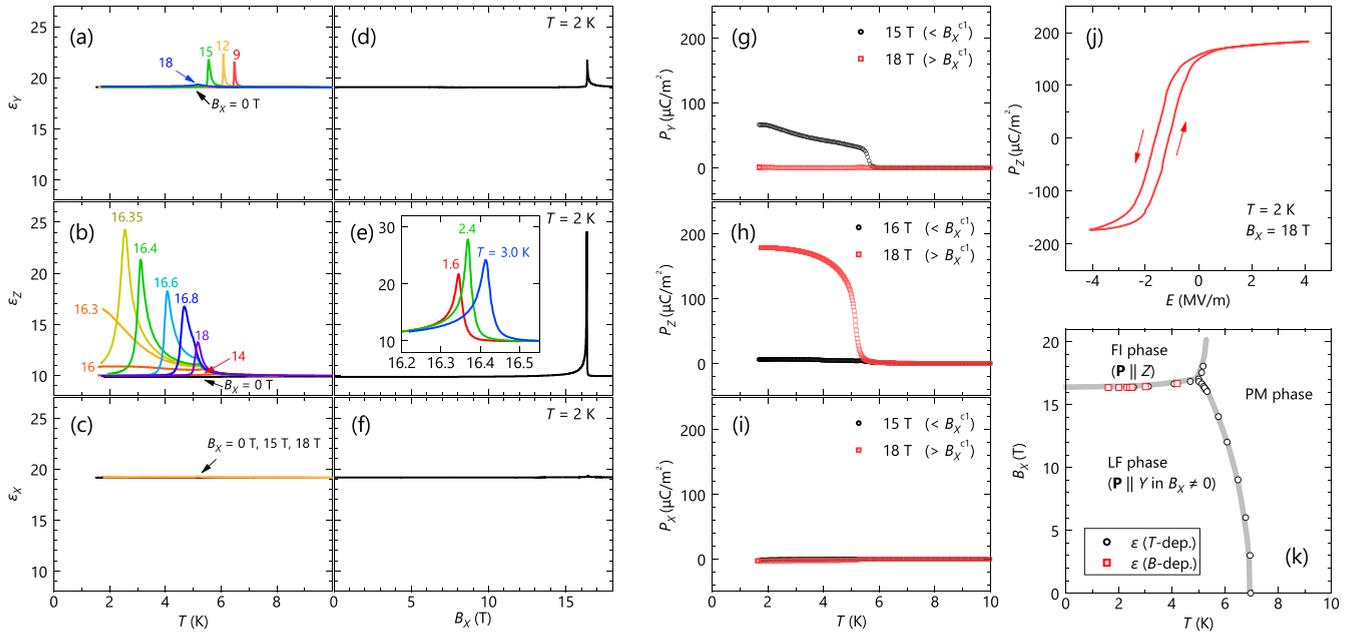


FIG. 3. Dielectric constant (ϵ) and electric polarization (\mathbf{P}) in a magnetic field applied along the X axis (B_X). (a)–(c) The temperature (T) dependence of ϵ along the (a) Y (ϵ_Y), (b) Z (ϵ_Z), and (c) X (ϵ_X) axes at various strengths of B_X . (d)–(f) The B_X dependence of (d) ϵ_Y , (e) ϵ_Z , and (f) ϵ_X at $T = 2$ K. The B_X -induced phase transition occurs at a critical field $B_X^{c1} \sim 16.4$ T, as indicated by the sharp peaks in (d) and (e). The inset in (e) shows ϵ_Z at selected T s near B_X^{c1} . (g)–(i) The T dependence of \mathbf{P} along the (g) Y (P_Y), (h) Z (P_Z), and (i) X (P_X) axes at $B_X < B_X^{c1}$ (black circle) and $B_X > B_X^{c1}$ (red square). The measurements were performed without an applied electric field (E) after the sample was cooled with $E = 0.67$ MV/m. (j) PE hysteresis loop along the Z axes at $B_X = 18$ T and at $T = 2$ K. The linear contribution $\epsilon_Z E$, where $\epsilon_Z \approx 10$ obtained from the data in (e) at $B_X = 18$ T and at $T = 2$ K, is subtracted. (k) The B_X vs T phase diagram determined by the anomalies seen in the T and B_X dependence of ϵ_Z .

even without the E -cooling procedure (not shown). The origin of the imprint effect is unclear and left for future work.

B. Experiments using a pulse magnet up to 56 T

To confirm the link between the magnetism and the observed ferroelectricity, we performed high-field magnetization measurements up to 56 T using a pulse magnet. We show in Fig. 4(a) the magnetization curve in B_X (M_X) at $T = 1.4$ K. The M_X curve initially exhibits a jump at 16.4 T, which coincides well with the critical field B_X^{c1} . This indicates that the observed flop of \mathbf{P} is associated with the B_X -induced magnetic phase transition. Then, the M_X curve gradually increases and shows a saturation above $B_X^{c2} \approx 45$ T, which corresponds to a transition to a fully polarized (FP) phase.

We have also measured magnetization curves for the field applied along the $[100]$ ($M_{[100]}$) and Z (M_Z) axes as they provide critical information for constructing an effective spin model, as shown later. We find that $M_{[100]}$ and M_Z also show an abrupt jump at $B_{[100]}^{c1} = 14.8$ T and $B_Z^{c1} = 12.3$ T, respectively. The saturation fields are $B_{[100]}^{c2} = 47.2$ T and $B_Z^{c2} = 43.4$ T. Importantly, the M_Z curve shows another weak anomaly at around 28 T, which is seen as a broad hump in its field derivative [Fig. 4(b)]. These characteristic features provide a critical test to check the validity of the spin model.

Now, we turn to pulse magnet measurements of \mathbf{P} up to 56 T in order to examine whether the B_X dependence of P_Z in the FI phase follows our proposed idea in Fig. 1. As shown

in Fig. 5(a), after showing a broad maximum around 25 T, P_Z gradually decreases and exhibits a sign reversal at around 37 T. Then, P_Z vanishes in the FP phase above $B_X^{c2} \approx 45$ T. No finite component of P_X and P_Y is seen in the FI phase for $B_X^{c1} < B_X < B_X^{c2}$. This means that the continuous P_Z reversal at around 37 T is not due to a \mathbf{P} rotation, but to a change

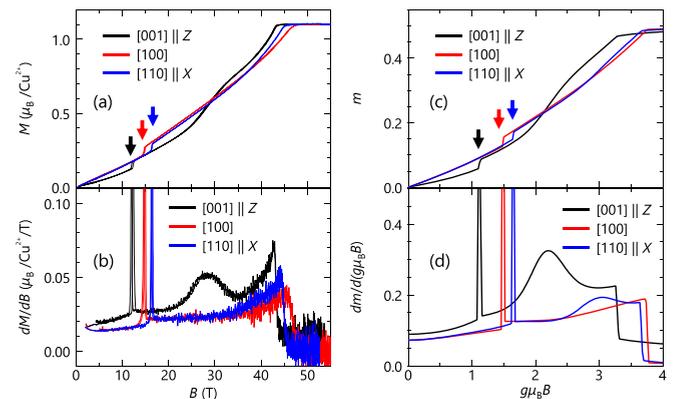


FIG. 4. (a), (b) Magnetic-field (\mathbf{B}) dependence of (a) magnetization (\mathbf{M}) and (b) its field derivative dM/dB in the experiment at $T = 1.4$ K for \mathbf{B} applied along the Z , $[100]$, and X axes. (c), (d) The magnetic-field ($g\mu_B B$) dependence of (c) magnetization (m) and (d) its field derivative $dm/d(g\mu_B B)$ in the theory for the model in Eq. (2). dM/dB in (b) and $dm/d(g\mu_B B)$ in (d) for the Z axis are shifted by 0.01 and 0.05, respectively, for clarity.

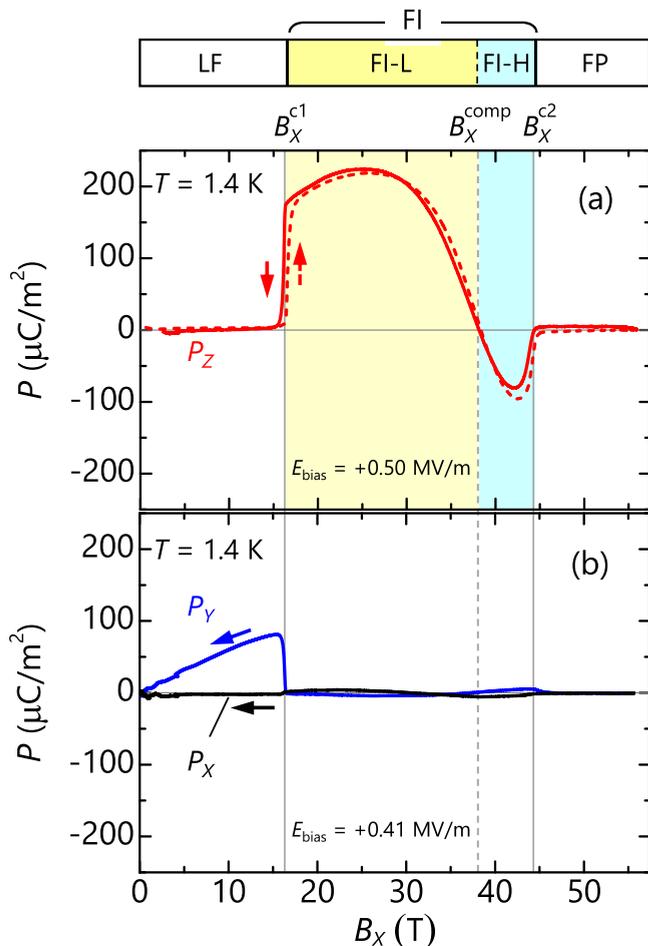


FIG. 5. Electric polarization along the X (P_X), Y (P_Y), and Z (P_Z) axes as a function of a magnetic field applied along the X axis (B_X) measured using a pulse magnet. (a) P_Z measured with an electric field $E = 0.50$ MV/m during the B_X -increasing (red dashed curve) and B_X -decreasing (red solid curve) processes. (b) P_Y (blue solid curve) and P_X (black solid curve) measured with $E = 0.41$ MV/m during the B_X -decreasing process. LF, FI, and FP mean the low-field phase, field-induced phase, and fully polarized phase, respectively. FI-L and FI-H denote the regions in the FI phase below and above the compensation field B_X^{comp} , respectively.

of the magnitude passing through $P_Z = 0$. Qualitatively, this behavior is in agreement with our proposal in Fig. 1. Here, we define $B_X \sim 37$ T for the P_Z reversal as a compensation magnetic field (B_X^{comp}), in analogy with a so-called compensation temperature for ferrimagnets at which a temperature-induced continuous magnetization reversal occurs.

It should be noted that no anomaly is seen in the M_X curve at $B_X \sim B_X^{\text{comp}}$ [Fig. 4(a)]. This indicates that the observed continuous P_Z reversal is associated with neither a phase transition nor domain switching, and thus accompanies no intrinsic hysteresis. Although the B -increasing and -decreasing data [Fig. 5(a)] do not perfectly collapse on top with each other, it must be due to a fast \mathbf{B} sweep in the pulse field measurements. In sharp contrast, a \mathbf{P} reversal in most of the magnetically induced ferroelectrics is associated with either a metamagnetic transition or domain switching. The resultant

large energy barrier between different magnetoelectric states causes a large hysteresis, which causes an undesirable energy loss in devices such as magnetoelectric sensors and oscillators. Therefore, a nonhysteresis feature of the \mathbf{P} reversal in the present material may be useful for these applications.

C. Quantum spin-1/2 model

1. Model and parameter setting

The effective quantum spin-1/2 model was previously developed for the isostructural material $\text{Ba}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$, which quite well reproduces the experimental magnetization curves [28]. Therefore, it is expected that the model also has ability to explain the experimental results of the present material, $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$. In this model, we take into account four dominant symmetric exchange interactions: the intracupola exchange interactions J_1 and J_2 , together with the two intercupola interactions within a layer J' and between neighboring layers J'' (Fig. 2). In addition, we also take into account an antisymmetric DM interaction at J_1 bonds [Fig. 2(a)]. The Hamiltonian can be written as

$$\mathcal{H} = \sum_{\langle i,j \rangle} [J_1 \mathbf{S}_i \cdot \mathbf{S}_j - \mathbf{D}_{ij} \cdot (\mathbf{S}_i \times \mathbf{S}_j)] + J_2 \sum_{\langle\langle i,j \rangle\rangle} \mathbf{S}_i \cdot \mathbf{S}_j + J' \sum_{(i,j)} \mathbf{S}_i \cdot \mathbf{S}_j + J'' \sum_{[(i,j)]} \mathbf{S}_i \cdot \mathbf{S}_j - g\mu_B \sum_i \mathbf{B} \cdot \mathbf{S}_i, \quad (2)$$

where \mathbf{S}_i represents $S = 1/2$ spin at site i . The sums for $\langle i, j \rangle$, $\langle\langle i, j \rangle\rangle$, (i, j) , and $[(i, j)]$ run over J_1 , J_2 , J' , and J'' bonds, respectively. For the intracupola exchange coupling constants, we adopt the estimates from the first-principles calculation: $J_1 = 3.0$ meV and $J_2 = 0.43$ meV [27]. We set J_1 as the unit of energy, namely, $J_1 = 1$ and $J_2 = 1/7$. On the other hand, for the intercupola J' , we set a larger value, $J' = 3/4$, than the first-principles estimate of $J' \simeq 0.14$ because a small $J' \lesssim 0.4$ leads to a nonmagnetic singlet state in the CMF approach; we set $J'' = -1/100$ for ferromagnetic coupling between layers [27]. The last term in Eq. (2) represents the Zeeman coupling where g and μ_B are the isotropic g factor and the Bohr magneton, respectively.

In the second term in Eq. (2), referring to the Moriya rules [39], we take the DM vector \mathbf{D}_{ij} in the plane perpendicular to the J_1 bond connecting i and j sites with the angle θ_{ij} from the Z axis [see the green arrow in Fig. 2(a)]. Note that the convex geometry of the square-cupola cluster induces the in-plane component of \mathbf{D}_{ij} . The sign of \mathbf{D}_{ij} is reversed between the upward (α) and downward (β) cupolas from the symmetry. We assume uniform $\theta = \theta_{ij}$ and $D \equiv |\mathbf{D}_{ij}|$, and tune the values of θ and D so as to reproduce the magnetization curves obtained experimentally.

We calculate the magnetic properties of this model by a standard CMF method, in which the intracupola interactions are treated by the exact diagonalization, while the intercupola interactions are treated by the MF approximation. The details for the calculation procedure were described in the previous report [28]. Figures 4(c) and 4(d) show the B profiles of magnetization per site m and its field derivative $dm/d(g\mu_B B)$, respectively, obtained by the CMF calculations with $\theta = 80^\circ$ and $D = 1.1$. The results well reproduce the experimental data in Figs. 4(a) and 4(b) in the following points. (i) m

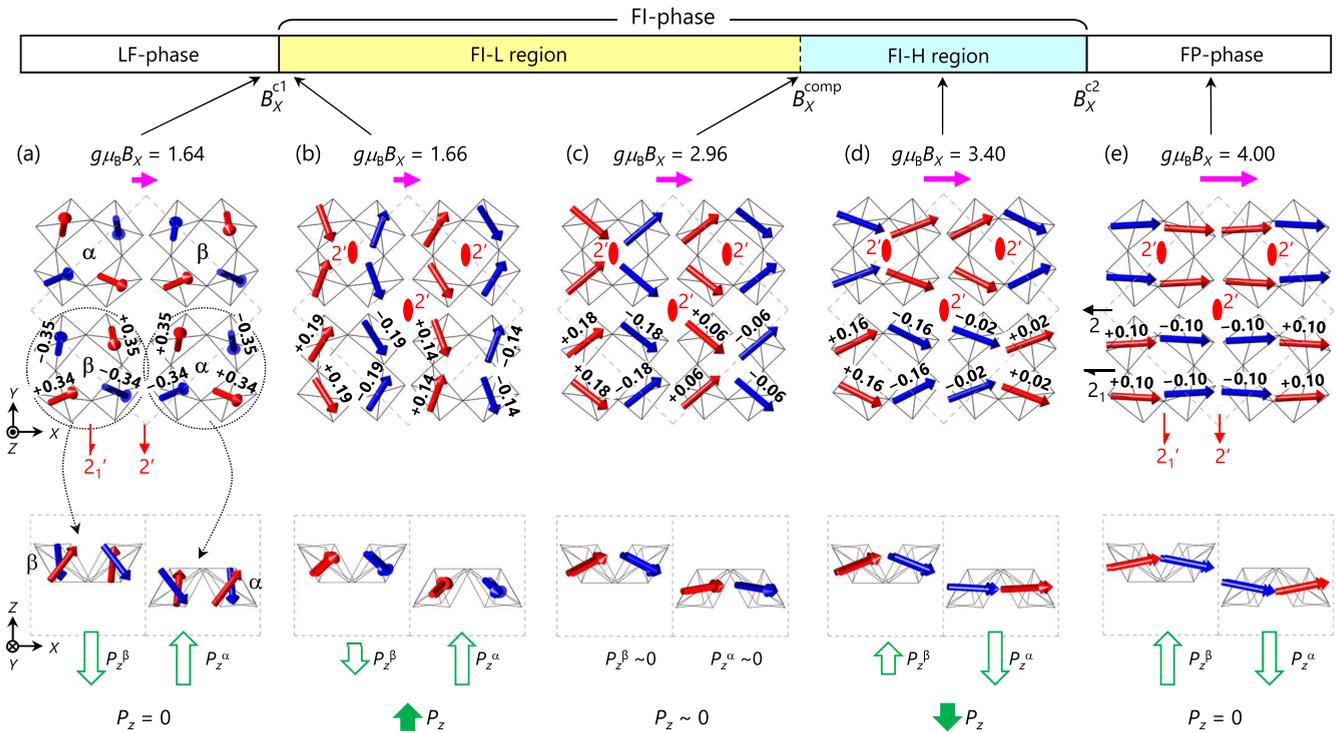


FIG. 6. Calculated B_X -induced evolution of the ordered spin arrangement at the selected strength of B_X . (a) $g\mu_B B_X = 1.64$ just below the metamagnetic transition field B_X^{c1} , (b) $g\mu_B B_X = 1.66$ just above B_X^{c1} , (c) $g\mu_B B_X = 2.96$ proximate to B_X^{comp} , (d) $g\mu_B B_X = 3.40$ in-between B_X^{comp} and B_X^{c2} , and (e) $g\mu_B B_X = 4.00$ in the fully polarized phase. The upper and lower panels are the Z- and Y-axis view of the spin arrangement, respectively. The dashed gray line denotes the unit cell. In the Y-axis view, only the square cupolas marked with the dotted circles in the Z-axis view are indicated. The red and blue thick arrows represent spins with positive and negative Z-axis components, respectively. In the upper panels, values of the Z-axis component of each spin are given. The representative symmetry operations are also indicated. At the bottom of each panel, the electric polarization of square cupola α (P_Z^α) and β (P_Z^β) and their sum (P_Z), expected from the exchange-striction mechanism, is schematically indicated by thick green arrows (see text for details).

shows a jumplike anomaly, whose magnetic field depends on the field direction. The critical field is consistent with the experimentally observed relation, namely, $B_X^{c1} > B_{[100]}^{c1} > B_Z^{c1}$. (ii) The saturation fields satisfy the observed relationship $B_{[100]}^{c2} > B_X^{c2} > B_Z^{c2}$. (iii) $dm/d(g\mu_B B)$ exhibits a hump at an intermediate field for B_Z . Thus, our effective spin model successfully explains the experimental magnetization curves, strongly supporting the validity of our model analysis.

2. Electric polarization

Before elucidating the microscopic mechanism, we discuss the onset of the net \mathbf{P} in terms of symmetry of the spin arrangement calculated using the effective spin model. In Figs. 6(a)–6(e), we summarize a B_X -induced evolution of the ordered spin arrangement at selected strength of B_X . Here, the vectors at each Cu site represent $(\langle S_i^X \rangle, \langle S_i^Y \rangle, \langle S_i^Z \rangle)$, where each component $\langle S_i^\mu \rangle$ is the ordered moment along the μ axis ($\mu = X, Y$, and Z). Figure 6(a) shows the spin arrangement in the LF phase at B_X just below B_X^{c1} . It is found that the only allowed symmetry operations are $2'$ and $2'_1$ along the Y axis. The magnetic point group is therefore $2'$, which allows for the onset of P_Y , consistent with the experimental result [Fig. 5(b)]. Figure 6(b) shows the spin arrangement in the FI phase just above B_X^{c1} . It can be seen that all the spins change their orientations upon the metamagnetic transition to

the FI phase. The magnetic point group is $2'$, where $2'$ along the Z axis is present at the center of each square cupola. Because of this symmetry, the polarization in each square cupola is allowed to emerge only along the Z axis (P_Z^α and P_Z^β). Critically, the angles made by the XY-plane components ($\langle S_i^X \rangle$ and $\langle S_i^Y \rangle$) of neighboring spins in square cupola α are different from those in β . Moreover, as denoted in the upper panel of Fig. 6(b), the absolute value of $\langle S_i^Z \rangle$ in α (0.14) is also different from that in β (0.19). As a result, all the symmetry operations that mutually convert α and β ($2'$ and $2'_1$ along the Y axis) are broken; that is, α and β are no longer symmetrically equivalent. Therefore, the magnitude of P_Z^α and P_Z^β must be different from each other, which can explain the onset of the net P_Z observed in experiments [Fig. 5(a)]. By further increasing B_X , the spins are forced to point along the $+X$ direction, while keeping the magnetic point group $2'$ [Figs. 6(c) and 6(d)]. In the FP phase [Fig. 6(e)], the magnetic point group is changed to $2'2'_2$, where the symmetry operations that relate α and β are recovered. This does not allow for \mathbf{P} along any directions. Therefore, from the symmetry point of view, the calculated spin arrangement fully agrees with the onset and direction of the net \mathbf{P} observed in experiments.

Now, we analyze the B_X dependence of P_Z on the basis of the exchange-striction mechanism. Because the magnetism of $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ is provided by quantum spins $S = 1/2$, not only the ordered components of the spins but also

quantum spin fluctuations give significant contributions to \mathbf{P} . We consider these two contributions separately. To this end, we decompose $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle$ in Eq. (1) as $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle = \langle S_i^X \rangle \langle S_j^X \rangle + \langle S_i^Y \rangle \langle S_j^Y \rangle + \langle S_i^Z \rangle \langle S_j^Z \rangle + \langle \Delta \mathbf{S}_i \cdot \Delta \mathbf{S}_j \rangle$. The first three terms are ‘‘classical’’ contributions to \mathbf{P} , while the last term is a ‘‘quantum’’ contribution, where $\Delta \mathbf{S}_i \equiv \mathbf{S}_i - \langle \mathbf{S}_i \rangle$ describes quantum spin fluctuations. Considering electric dipoles induced by J_1 bonds, we can evaluate electric polarization for each square cupola using the following formula:

$$\mathbf{P}^k = \sum_{\langle i,j \rangle^k} \mathbf{e}_{ij} [\langle S_i^X \rangle \langle S_j^X \rangle + \langle S_i^Y \rangle \langle S_j^Y \rangle + \langle S_i^Z \rangle \langle S_j^Z \rangle + \langle \Delta \mathbf{S}_i \cdot \Delta \mathbf{S}_j \rangle], \quad (3)$$

where $k = \alpha$ and β , and the sum $\sum_{\langle i,j \rangle^k}$ runs over all the J_1 bonds in a square cupola labeled k . The sum of these two gives the total electric polarization in the unit cell, $\mathbf{P} = \mathbf{P}^\alpha + \mathbf{P}^\beta$. [We ignored the coefficient A in Eq. (1) for simplicity.] The calculated results of the B_X dependence of \mathbf{P}^α , \mathbf{P}^β , and \mathbf{P} are summarized in Figs. 7(a)–7(e), which will be described below.

Let us first consider the classical contributions to $P_Z = P_Z^\alpha + P_Z^\beta$ in the FI phase. As shown in Figs. 6(b)–6(d), the Z -axis components $\langle S^Z \rangle$ of four spins within each square cupola align in the up-up-down-down manner and their absolute values are identical to each other in the entire B_X range. According to Eq. (3), the sum of $\langle S_i^Z \rangle \langle S_j^Z \rangle$ is always zero and does not induce a finite P_Z , as shown in Fig. 7(a). Therefore, at the classical level, only the XY -plane spin components contribute to P_Z . Notably, the B_X -induced evolution of the XY -plane components in each square cupola [upper panels of Figs. 6(b)–6(d)] is quite similar to the one in our proposal (Fig. 1). This leads to a continuous sign reversal of P_Z^α (P_Z^β) from positive to negative (negative to positive), as shown in Fig. 7(b). Importantly, the above-mentioned difference between the magnitude of P_Z^α and P_Z^β gives rise to a net P_Z . Furthermore, in agreement with the experimental results, the net P_Z exhibits a continuous sign reversal [red solid line in Fig. 7(b)] due to the following reasons. In the lower-field region of the FI phase (FI-L region) [Fig. 6(b)], the arrangement of the XY -plane components is closer to antiparallel (more antiferromagnetic) in α than in β . The magnitude of P_Z^α is therefore larger than that of P_Z^β , which gives rise to a net P_Z along the $+Z$ direction [see the bottom of the panel in Fig. 6(b)]. On the other hand, in the higher-field region of the FI phase (FI-H region) [Fig. 6(d)], the arrangement of XY -plane components is closer to parallel (more ferromagnetic) in α than in β . As a result, a net P_Z appears along the $-Z$ direction, opposite to P_Z in the FI-L region. In the intermediate-field region of $g\mu_B B_X = 2.96$ [Fig. 6(c)], the XY -plane components of the neighboring spins are roughly perpendicular to each other, resulting in $P_Z \sim 0$. The continuous sign reversal of P_Z thus occurs. These results clearly demonstrate that the classical contributions of the exchange-striction mechanism can explain the onset of P_Z and its B_X -induced continuous sign reversal.

To investigate the effects of quantum spin fluctuations, we plot in Fig. 7(c) the quantum contribution ($\langle \Delta \mathbf{S}_i \cdot \Delta \mathbf{S}_j \rangle$) term) to P_Z^α , P_Z^β , and P_Z . In the FI-L region, the quantum contribution to P_Z is found to be as large as the classical

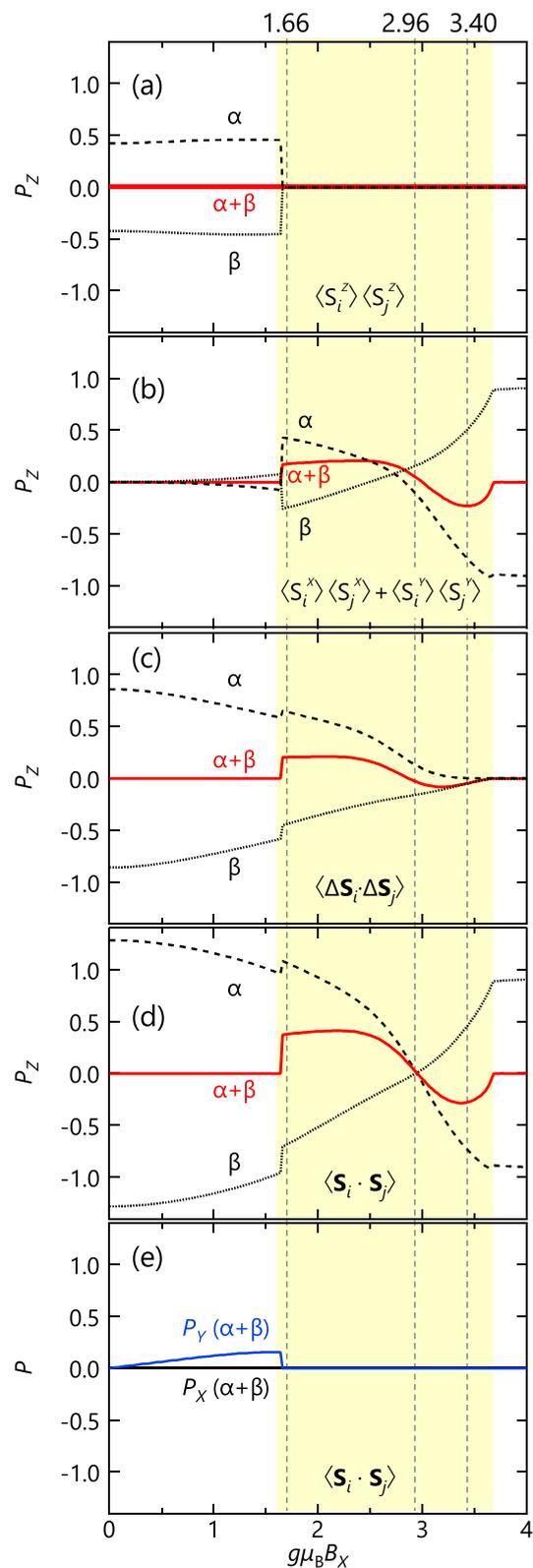


FIG. 7. B_X dependence of the calculated electric polarization P_Z originated from (a) $\langle S_i^Z \rangle \langle S_j^Z \rangle$, (b) $\langle S_i^X \rangle \langle S_j^X \rangle + \langle S_i^Y \rangle \langle S_j^Y \rangle$, (c) $\langle \Delta \mathbf{S}_i \cdot \Delta \mathbf{S}_j \rangle$, and (d) $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle$ terms, and (e) P_X and P_Y from the $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle$ term. In (a)–(d), the contributions from square cupolas α and β and their sum ($\alpha + \beta$) are shown by black dashed, black dotted, and red solid lines, respectively. The dashed vertical lines denote B_X for the spin arrangements shown in Figs. 6(b)–6(d).

contribution, which demonstrates that quantum fluctuations largely enhance the ferroelectricity in $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$. On the other hand, in the FI-H region, the quantum contribution becomes less significant, most likely due to the suppression of the quantum fluctuations by the applied strong magnetic field.

Figure 7(d) shows the total contributions to P_Z of the exchange-striction mechanism. The calculation result quite well reproduces the experimental B_X profile of P_Z in Fig. 5(a). As shown in Fig. 7(e), the calculation result also well reproduces B_X profiles of P_X and P_Y in Fig. 5(b). The B_X -induced P_Y in the LF phase can be understood as follows. In the case of the zero-field spin arrangement (Fig. 2), the ab -plane component of \mathbf{p}_{ij} from all the bonds cancels out completely. On the other hand, when the spin arrangement is deformed by the applied B_X [Fig. 6(a)], the cancellation becomes incomplete, which results in the net P_Y . Therefore, the present model calculations demonstrate that the exchange-striction mechanism provides an excellent description of the ferroelectricity in $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$.

We note that the maximum value of $P_Z \sim 220 \mu\text{C}/\text{m}^2$ observed in experiments is rather small and comparable to the observed value of typical magnetically induced ferroelectric polarization via spin-orbit couplings in inorganic oxides (order of 1–100 $\mu\text{C}/\text{m}^2$ [40]). However, this small value is obviously due to the cancellation of the sublattice polarization P_Z^α and P_Z^β . If the square cupolas were arranged in a uniform manner, rather than the staggered manner as in $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$, the combined electric polarization [= $P_Z^\alpha - P_Z^\beta$ in Fig. 7(d)] would amount to $\sim 1000 \mu\text{C}/\text{m}^2$, which is comparable to the typical polarization induced by the exchange-striction mechanism (e.g., Refs. [41,42]). This estimate, together with the gigantic magnetodielectric effect [Fig. 3(e)], shows that the square-cupola units can be considered as a building block with strong magnetoelectric couplings.

D. Density functional calculations

To unambiguously establish that the observed ferroelectric polarization dominantly arises from the nonrelativistic exchange-striction mechanism, we examine the effects of relativistic spin-orbit coupling (SOC) on the electric polarization. To this end, we have performed DFT calculations of P_Z including SOC self-consistently, whereas we have also performed extra calculations excluding SOC. The former and the latter correspond to a total P_Z and P_Z due to only the exchange-striction mechanism, respectively. The difference between the two corresponds to the effects of SOC on P_Z . In the calculations, the spin arrangement obtained by the CMF analysis of the model given by Eq. (2) was used.

Figure 8(a) shows the results of the B_X dependence of the total P_Z (red open diamonds), P_Z due to the exchange-striction mechanism (red filled diamonds), and P_Z from the effects of SOC (red crosses). It is found that the total P_Z well reproduces the B_X dependence of P_Z observed in experiments including the sign reversal. Importantly, P_Z due to the exchange-striction mechanism dominates the overall B_X dependence, while the effects of SOC give only a minor contribution to P_Z and cannot reproduce the sign reversal. This means that any mechanisms associated with

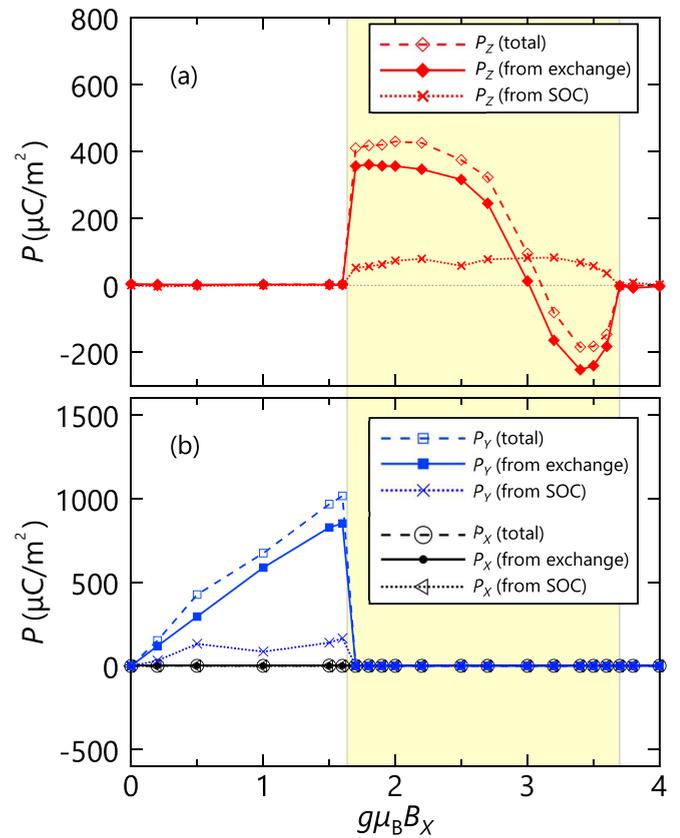


FIG. 8. Density functional calculations of the electric polarization as a function of the magnetic field applied along the X axis (B_X). (a) The electric polarization along the Z axis (P_Z). To extract the effect of the spin-orbit coupling (SOC), calculations including and excluding SOC were performed. The former and the latter correspond to the total P_Z (red open diamonds) and P_Z from the exchange striction (red filled diamonds), respectively. The red crosses denote the difference between the two, which corresponds to P_Z arising from any mechanism involving SOC. (b) The corresponding electric polarization along the X axis (P_X) and the Y axis (P_Y). The total P_Y , P_Y from the exchange striction, and P_Y from SOC are denoted by blue open squares, blue filled squares, and blue crosses, respectively. The total P_X , P_X from the exchange striction, and P_X from SOC are shown by black open circles, black filled circles, and black triangles, respectively, which are zero in the entire B_X range.

SOC, such as the spin-current mechanism [14,15] and the metal-ligand d - p hybridization mechanism [16], do not give an important contribution. Therefore, our DFT result clearly demonstrates that the exchange-striction mechanism is dominant for the observed magnetically induced ferroelectricity in $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$.

The maximum value of the total P_Z amounts to $\sim 400 \mu\text{C}/\text{m}^2$. This value is in fairly good agreement with the maximum value of $P_Z \sim 220 \mu\text{C}/\text{m}^2$ observed in experiments [Fig. 5(a)]. On the other hand, the calculated value is two orders of magnitude smaller than the exchange-striction-driven polarization calculated for other systems (e.g., $\sim 60\,000 \mu\text{C}/\text{m}^2$ in HoMnO_3 [43]). As already mentioned, this is attributable to the cancellation of the sublattice polarization P_Z^α and P_Z^β .

We also calculated the B_X dependence of P_X and P_Y , the result of which is shown in Fig. 8(b). It captures the qualitative features observed in experiments; that is, the onset of P_Y in the LF phase and the absence of P_X in the entire B_X range. However, in contrast to the quantitative agreement between the calculated value of the total P_Z and the experimental value of P_Z in the FI phase, the calculated value of the total P_Y ($\sim 1000 \mu\text{C}/\text{m}^2$) in the LF phase is found to be much larger than the experimental value ($< 100 \mu\text{C}/\text{m}^2$; see Fig. 5). This discrepancy may be explained by the difference between the spin arrangement in the calculations and experiments: The spin arrangement in the LF phase calculated by Eq. (2) [Fig. 6(a)] is more collinear along the Z axis than that proposed by the previous neutron-diffraction experiments in Fig. 2(a) [27], which gives a larger $\langle S_i \rangle \cdot \langle S_j \rangle$ and thus larger calculated P_Y .

E. Electric-field control of domains

Using $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ as a model material, we have successfully demonstrated that the magnetic square-cupola units exhibit the exchange-striction-driven electric polarization, whose sign can be continuously reversed by sweeping a magnetic field. So far, the \mathbf{B} -induced continuous \mathbf{P} reversal, not originating from a phase transition or domain switching, was observed only in a very limited number of materials, including $AE_2\text{TMGe}_2\text{O}_7$ ($AE = \text{Ba}, \text{Sr}$, $\text{TM} = \text{Co}, \text{Mn}$) [44,45] and RMn_2O_5 ($R = \text{Tb}$ or Bi) [46,47]. Little is known about unique ferroelectric properties associated with the \mathbf{B} -induced continuous \mathbf{P} reversal. Therefore, we move to investigate ferroelectric properties of $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ in more detail, in particular a response of ferroelectric domains to an external electric field in the FI phase. Technically, measurements of PE hysteresis loops at a constant B_X using a pulse magnet are rather difficult. We have instead carried out a polarization measurement on sweeping B_X with a various bias electric field E_{bias} in the range of -2.5 to $+3.0$ MV/m.

Before presenting experimental results, let us describe the ferroelectric domains in $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$. In our model calculations, two types of energetically equivalent spin arrangements are obtained in the FI phase. These spin arrangements are mutually converted by the twofold rotation operation about the X axis, with the rotation axis passing through the center of the unit cell [see Figs. 6(b)–6(d)]. They correspond to ferroelectric domains exhibiting the opposite sign of the B_X dependence of P_Z to each other. Here, we define the domain which shows the positive to negative B_X dependence of P_Z , corresponding to Fig. 7(d) as $\text{D}[+0-]$, while another domain with the negative to positive B_X dependence as $\text{D}[-0+]$. The spin arrangements of these domains in the FI-L and FI-H regions are schematically illustrated in Figs. 9(a) and 9(b).

The results of P_Z measured during the B_X increasing and subsequent decreasing processes are separately shown in Figs. 9(a) and 9(b), respectively. They reveal a very complicated behavior. To understand this, we first focus on the $E_{\text{bias}} = 0$ MV/m data (black line). In both of the B -increasing and B -decreasing processes, positive and negative P_Z appear in the FI-L and FI-H regions, respectively. This means that the above-mentioned imprint effect [see Fig. 3(j)] always stabilizes the ferroelectric domain $\text{D}[+0-]$ over the entire

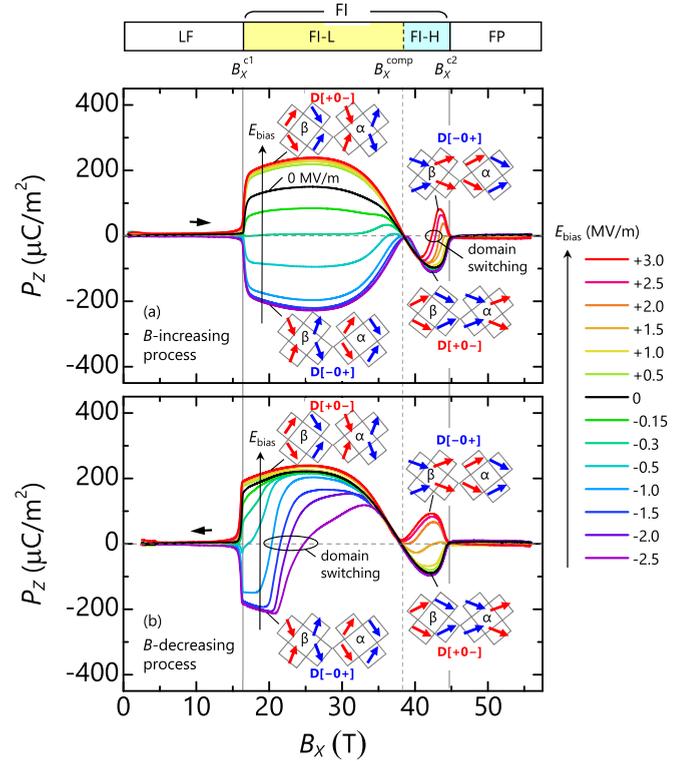


FIG. 9. The B_X dependence of P_Z measured with various bias electric fields (E_{bias}) during (a) a B_X -increasing and (b) a subsequent B_X -decreasing process. In the FI phase, there are two types of domains labeled as $\text{D}[+0-]$ and $\text{D}[-0+]$; the former (latter) shows $+P_Z$ and $-P_Z$ ($-P_Z$ and $+P_Z$) in the FI-L and FI-H regions, respectively. The spin arrangements of square cupolas α and β of these domains in the FI-L and FI-H regions are schematically illustrated. The red and blue arrows represent spins with positive and negative Z -axis components, respectively. The measurements were performed at the base temperature of the measurement system ranging from 1.4 to 1.5 K.

region of the FI phase. Therefore, the imprint effect does not directly couple to the sign of P_Z , but to the domain state. In this sense, $\text{D}[+0-]$ can be regarded as the imprint-stabilized domain, while $\text{D}[-0+]$ as the imprint-destabilized domain.

The imprint-destabilized $\text{D}[-0+]$ state can be stabilized by the application of a strong enough E_{bias} across the transition from the paraelectric (LF or FP) to ferroelectric FI phases. For example, in the FI-L region during the B -increasing process [Fig. 9(a)], the negative P_Z is induced by a negative E_{bias} . It is saturated at $E_{\text{bias}} < -1.0$ MV/m, indicating the emergence of a single $\text{D}[-0+]$ state. Likewise, in the FI-H region during the B -decreasing process [Fig. 9(b)], the positive P_Z is induced by positive E_{bias} and becomes nearly saturated at $E_{\text{bias}} > +2.5$ MV/m, indicating a nearly single $\text{D}[+0-]$ state. Therefore, the ferroelectric domains in $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ are highly responsive to an external electric field, which allows for the unique control of the ferroelectric polarization by the combination of magnetic and electric fields.

Finally, we consider the domain-switching behavior within the ferroelectric FI phase. By the application of the positive E_{bias} in the B -increasing process [Fig. 9(a)], a single domain state of $\text{D}[+0-]$ must be present at $B_X = B_X^{\text{comp}}$. As B_X

further increases into the FI-H region, the D[+0−] domain with $-P_Z$ is switched to the D[−0+] domain with $+P_Z$ by the positive $E_{\text{bias}} > +0.5$ MV/m. However, this domain switching occurs only at B_X far above B_X^{comp} . In other words, a coercive electric field (i.e., E_{bias} for the domain switching) increases as B_X approach to B_X^{comp} . Then, nearby B_X^{comp} , the domain switching finally becomes impossible by E_{bias} applied in the present study ($E_{\text{bias}} \leq 3$ MV/m). A similar behavior is observed for the domain switching in the FI-L region during the B -decreasing process with a negative E_{bias} [Fig. 9(b)]. As in the case of the B -increasing process with the positive E_{bias} , the single domain state of D[+0−] must be present at $B_X = B_X^{\text{comp}}$. As B_X further decreases into the FI-L region, the D[+0−] domain is switched to D[−0+] by a negative $E_{\text{bias}} \leq -1.0$ MV/m, but it occurs only at B_X far below B_X^{comp} . Combined these results, it is expected that the coercive electric field becomes maximum at B_X^{comp} . This is consistent with the absence of a driving force for the domain switching ($P_Z \times E_{\text{bias}} = 0$) at B_X^{comp} since $P_Z = 0$. This feature of the coercive electric field resembles well the temperature dependence of the coercive magnetic field in ferrimagnets showing a temperature-induced magnetization reversal. In memory devices composed of ferrimagnetic materials, this compensation behavior is quite useful for robust data storage and low-field writing. Therefore, the \mathbf{B} -induced compensation of the electrically switchable \mathbf{P} discovered in the present study may open a possibility of unique magnetoelectric devices. Clearly, it is essential to make a more detailed characterization of the B_X dependence of the coercive electric field, such as by means of PE hysteresis loop measurements with a flat pulse magnetic field.

VI. CONCLUSION

In conclusion, we propose a convex-shaped square-cupola spin cluster as a promising structural unit hosting a large magnetoelectric coupling due to the exchange-striction mechanism. Targeting $\text{Pb}(\text{TiO})\text{Cu}_4(\text{PO}_4)_4$ as a model material, which consists of a staggered array of Cu_4O_{12} square cupolas, our joint experimental and theoretical studies successfully

verify this idea by observing the gigantic magnetodielectric effect ($\sim 180\%$) and ferroelectricity that originates from the staggered arrangement of large electric polarization with the different magnitude. We also discover a B -induced continuous reversal of the ferroelectric polarization, which enables an unusual control of the domains by combination of electric and magnetic fields. Here we emphasize that our proposal in the present study does not require a complicated, delicate balance among frustrated magnetic interactions, which are usually required in most magnetically induced ferroelectric materials. Owing to this simplicity, our proposed idea can be extended to other types of spin clusters with convex geometry. The present result therefore demonstrates that materials with convex-shaped magnetic structural units deserve to be explored and synthesized to achieve strong magnetoelectric couplings, which would open the door for future magnetoelectric device applications.

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