

Electrostatic modulation of the electronic properties of Dirac semimetal Na₃Bi thin films

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Large-area thin films of topological Dirac semimetal Na₃Bi are grown on amorphous SiO₂:Si substrates to realize a field-effect transistor with the doped Si acting as a back gate. As-grown films show charge carrier mobilities exceeding 7 000 cm²/V s and carrier densities below 3 × 10¹⁸ cm⁻³, comparable to the best thin-film Na₃Bi. An ambipolar field effect and minimum conductivity are observed, characteristic of Dirac electronic systems. The results are quantitatively understood within a model of disorder-induced charge inhomogeneity in topological Dirac semimetals. The hole mobility is significantly larger than the electron mobility in Na₃Bi which we ascribe to the inverted band structure. When present, these holes dominate the transport properties.

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I. INTRODUCTION

Topological Dirac semimetals (TDSs) are three-dimensional analogs of graphene, with linear electronic dispersions in three dimensions [1–3]. Thin films of TDSs are of interest for topological devices; a conventional-to-topological quantum phase transition (QPT) occurs with increasing film thickness [3–5], and gate electrodes can enable an electric field-tuned QPT, realizing a topological transistor [4,5]. Additionally, nanostructured TDSs exhibit unusual transport phenomena due to Fermi arc surface states [6,7]. Field-effect structures are a key step toward realizing topological devices based on TDSs. To date field-effect gating has been realized on individual nanowires [8] or nanoplates [9] of the TDS Cd₃As₂, and electrolyte gating has been used on Cd₃As₂ thin films [10]. However, no large-area solid-state process has been demonstrated for a TDS field-effect device.

Here we report the growth of high-quality thin films of Na₃Bi on amorphous SiO₂ on Si substrates. We have characterized the films using low-temperature magnetotransport *in situ* in ultrahigh vacuum (UHV) [11–13]. As-grown films show charge carrier mobilities exceeding 7 000 cm²/V s and carrier densities below 3 × 10¹⁸ cm⁻³, comparable to the best thin films on Al₂O₃ [12,13]. Using the doped Si substrate as a back-gate electrode, the carrier density can be tuned over 2 × 10¹⁸ cm⁻³. Using a combination of molecular surface transfer doping [13] and field-effect gating, an ambipolar field effect and minimum conductivity are observed, characteristic of Dirac electronic systems near the Dirac point. The results are quantitatively understood within a model of disorder-induced charge inhomogeneity in topological Dirac semimetals. The hole mobility is found to be significantly higher than the electron mobility in Na₃Bi, reflecting the unusual character of the inverted band structure.

II. METHODS

A Na₃Bi thin film of thickness 18 nm was deposited through a stencil mask onto a pre-existing electrode pattern on SiO₂ (1 μm) on Si substrates [14]. Figure 1(a) shows a schematic of the sample and stencil mask used to define the Hall bar geometry. After growth of the thin film on the pre-existing electrical contacts, the entire structure is transferred within UHV to the analysis chamber where electrical measurements are performed at a temperature of 5 K, and scanning tunneling microscopy (STM) is also performed. Figure 1(b) shows a STM image of the film surface. The film is highly oriented with the *c* axis perpendicular to the substrate, showing atomically flat [0001] terraces of width 20 nm or more. Since the as-grown films are found to be *n* type, we employ additional *p*-type doping with the organic molecule 2,3,5,6-tetrafluoro-7,7,8,8-tetracyanoquinodimethane (F4-TCNQ). F4-TCNQ acts as a *p*-type dopant due to its high electron affinity, and physisorbs on the surface of Na₃Bi with little detrimental effect on the charge carrier mobility [13]. The Na₃Bi film thickness of 18 nm is comparable to or less than the Thomas-Fermi screening length at the observed charge carrier densities, indicating that band-bending across the thickness of the film is not significant and hence field-effect and molecular doping should affect the entire film thickness.

III. RESULTS

Figure 2 shows the conductivity σ_{xx} [Fig. 2(a)] and Hall carrier density n_{Hall} [Fig. 2(b)] as a function of gate voltage V_g applied to the silicon, for Na₃Bi films as grown and after doping with 8.8 Å of F4-TCNQ. The Hall carrier density is calculated from the low-field ($B < .05$ T), linear Hall response: $n_{\text{Hall}} = B/(e \rho_{xy})$. The as-grown Na₃Bi film (blue circles) has an *n*-type carrier density of 8.8 × 10¹⁷ cm⁻³ and a conductivity 1.0 × 10³ S/cm at $V_g = 0$ V. This corresponds to a Hall mobility of 7200 cm²/V s. The conductivity decreases with decreasing Hall carrier density as expected for the *n*-type film. The gate-dependent carrier density for the as-grown

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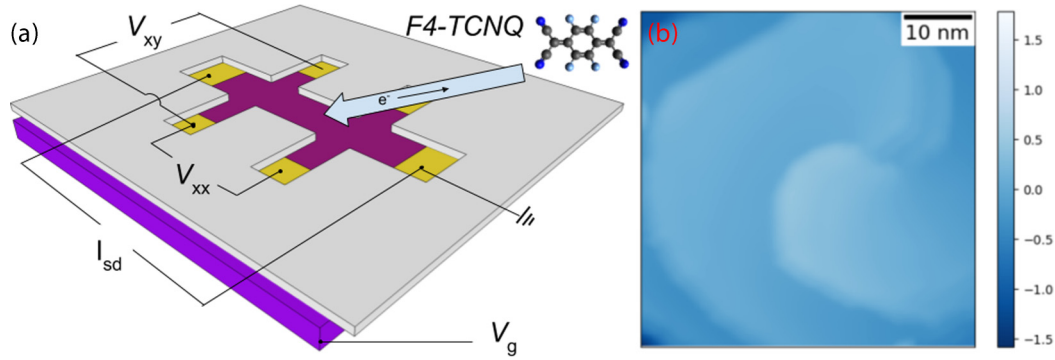


FIG. 1. (a) Device schematic, showing Hall bar film geometry defined by the surface stencil mask affixed on the SiO₂ on Si substrate that serves as a back gate. After growth of the Na₃Bi thin film, the molecular charge acceptor F4-TCNQ is subsequently deposited on the top surface. (b) STM morphology of sample. Scan area is 50 × 50 nm (vertical color scale nm), STM bias voltage $V_{\text{bias}} = -300$ mV, and current $I = 200$ pA. The observable step edges have heights of 4.2 Å, close to the expected 4.83 Å half-unit cell [3].

film in Fig. 2(b) shows nonlinear saturation for negative gate voltages, probably due to charge trapping in the oxide due to degradation by the Na/Bi deposition. Charge trapping is also consistent with the forward/backward gate-dependent

hysteresis observable in the conductivity of the F4-TCNQ doped measurement in Fig. 2(a).

The same Na₃Bi film with an 8.8 Å overlayer of F4-TCNQ shows qualitatively different behavior (red squares). The conductivity of the sample is reduced by approximately a factor of 4, and the gate-dependent conductivity reveals a minimum. The Hall carrier density shows an asymptotic positive (negative) divergence in carrier density at gate voltages roughly above (below) the gate voltage of minimum conductivity. Qualitatively similar behavior is observed in other gapless Dirac semimetals such as graphene [15] and the surface state of the topological insulator Bi₂Se₃ [16], where the minimum conductivity and the change of sign of the Hall carrier density reflect the Fermi energy crossing the Dirac point.

To better understand the physics close to the crossing of the Dirac point, in Fig. 3 we show the detailed longitudinal and Hall resistivity of the F4-TCNQ-doped sample in the region close to the sign change in the carrier density. Figure 3(a) shows the gate-voltage dependence of the zero-field longitudinal resistivity ρ_{xx} , showing a maximum at $V_g = 0$ V. Figure 3(b) shows the magnetic field dependence of the Hall resistivity ρ_{xy} at various gate voltages. There are several notable features in the gate-dependent Hall response. At low field ($B < 0.25$ T) there is a clear transition from electron- to hole-like response as the gate is modulated from positive to negative bias; this results in the sign change of n_{Hall} observed in Fig. 2(b). The smooth change of slope from positive to negative through zero results in the divergence of n_{Hall} observed in Fig. 2(b), as n_{Hall} is inversely proportional to the slope. Furthermore, as the low-field response tends to being hole-like (negative slope), the overall signal becomes nonlinear, which indicates contributions from multiple carrier types [17–19]. Note that the as-grown Na₃Bi sample showed completely linear $\rho_{xy}(B)$ over the entire range of gate voltage, consistent with a single *n*-type carrier.

We first attempted to fit the $\rho_{xy}(B)$ data using a two-carrier model. The change in slope of $\rho_{xy}(B)$ with B requires a hole channel of lower concentration and higher mobility than the electron channel, and individual $\rho_{xy}(B)$ curves can be well fitted to the two-carrier model. However, the implied gate dependence of the total carrier number (e.g., the sum of the concentrations of the two channels, $n + p$) is unphysical,

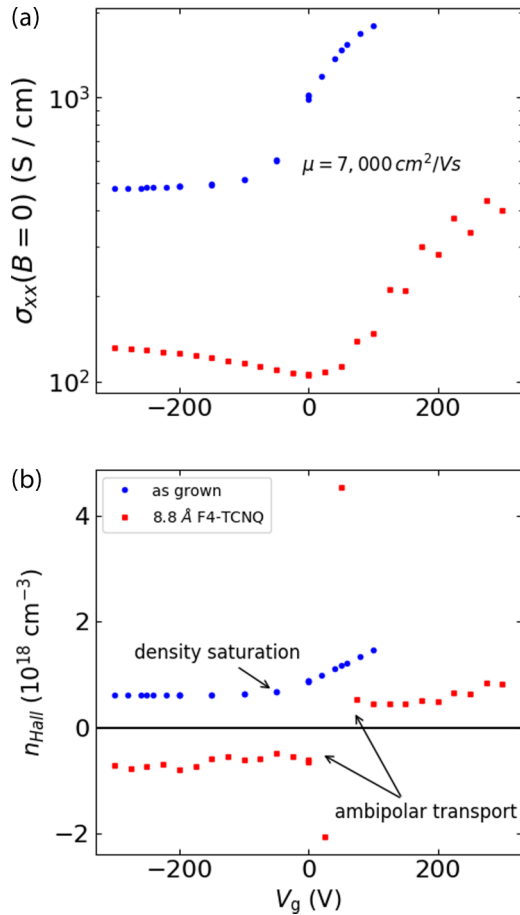


FIG. 2. (a) Zero field conductivity (S/cm) and (b) low-field Hall carrier density (10^{18} cm^{-3}) vs back-gate voltage (V_g), for the Na₃Bi film as grown (blue circles), and after the deposition of 8.8 Å F4-TCNQ on the surface (solid squares).

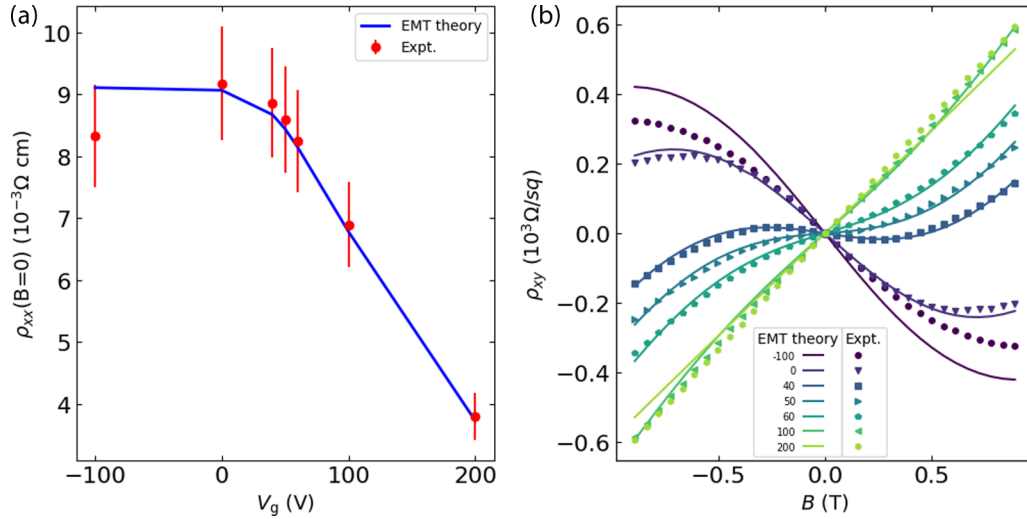


FIG. 3. (a) Gate dependence of the zero-field resistivity ρ_{xx} (Ω cm) after top-gating with F4-TCNQ. The red points are the measured values; the blue line is the theoretical fit. (b) Transverse magnetoresistance ρ_{xy} as a function of magnetic field perpendicular to the film, for back-gate voltages (V_g) indicated in the legend. The markers are the measured data, while the solid lines are fits using the charge inhomogeneity theory described in the text.

indicating, e.g., that the number of electrons in the sample is increasing with more negative applied gate voltage.

Failure of the two-carrier model is not surprising. Transport near the Dirac point is understood to involve spatially inhomogeneous regions of different carrier sign and density, with qualitatively different behavior than the two-carrier model [20]. Therefore to quantitatively understand our results, we have applied the previously developed effective medium theory (EMT) for charge inhomogeneity in graphene [20], modified to apply to Weyl semimetals for the relevant field regime ($\mu B < 1$) [21]. For the data taken following the F4-

TCNQ deposition, we can fit the data in Figs. 3(a) and 3(b) at each applied gate voltage using three global fit parameters. Two parameters, A_e and A_h , are proportional to the electron and hole mobilities, respectively; they are defined as the constants of proportionality between σ_{xx} and $|n|^{4/3}$ (the expected carrier density dependence for a TDS [22]) and depend on the impurity density n_{imp} , the electron and hole Fermi velocities v_e and v_h , and the dielectric constant κ . A third global parameter, n_{rms} , is the measure of the disorder-induced carrier density fluctuations [23]. We take the carrier density n as the only V_g -dependent fit parameter.

Solid lines in Fig. 3 show the results of the fits to the EMT. The EMT fits reproduce the very shallow maximum in $\rho_{xx}(V_g)$, and the nonlinear $\rho_{xy}(B)$. As anticipated, we find a larger hole mobility than electron mobility, as reflected in the parameters $A_h = 1.5 \times 10^{-1}$ and $A_e = 1.4 \times 10^{-2} e^2/h/10^{15} \text{ cm}^{-3}$. The final global fit parameter is the Gaussian width that determines the size of the charge inhomogeneous regime $n_{\text{rms}} = 2.2 \times 10^{17} \text{ cm}^{-3}$.

Figure 4 shows the carrier density extracted from the fits as a function of gate voltage, as well as the measured Hall carrier density for the film before F4-TCNQ doping, expected to be in the homogeneous regime. The overall dependence of the net carrier density n on $V_g > 0$ for the F4-TCNQ doped film determined from fitting (red squares) and as-grown (blue circles) has similar slope in all samples we measured (see Supplemental Material for details [14]).

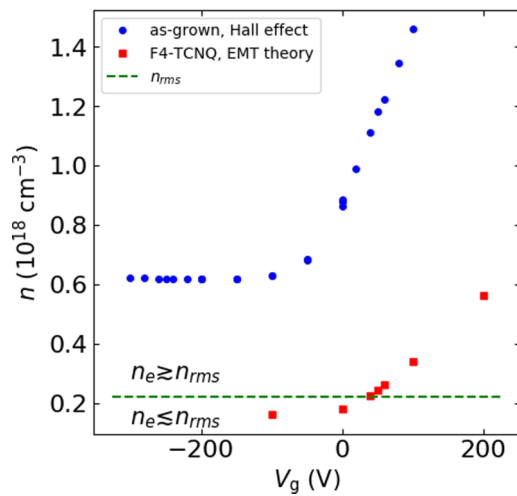


FIG. 4. Carrier density n as a function of gate voltage V_g for the as-grown sample as determined from Hall effect (blue circles) and for the F4-TCNQ-doped sample from the EMT fits in Fig. 3(b) (red squares). The solid green line is the value of the global fit parameter $n_{\text{rms}} = 2.2 \times 10^{17} \text{ cm}^{-3}$. The blue circles are the as-grown carrier density for comparison.

IV. DISCUSSION

Surprisingly, the net carrier density at all gate voltages remains n type even after F4-TCNQ doping. The implication is that the change in sign of the Hall carrier density [Fig. 2(b)] results from minority holes which dominate the Hall signal due to their much higher mobility. The value of $n_{\text{rms}} = 2.2 \times 10^{17} \text{ cm}^{-3}$, is also indicated for reference (green dashed

line). As the net carrier density is reduced and approaches this inhomogeneous regime, the n -type fraction of the sample decreases while the p -type fraction increases. The conductivity minimum at $V_g = 0$ V [Fig. 2(a)] thus does not correspond to a change in carrier sign; this counterintuitive observation is not unexpected in the case of this huge asymmetry between electron and hole conductivities [24].

Analysis of the magnetoresistance (MR) $\rho_{xx}(B)$ is complicated by the large weak antilocalization (WAL) contribution in our thin films, which is not captured by the semiclassical EMT. We have attempted to subtract the WAL component in order to compare the experimental MR to the EMT. We find that EMT still overestimates the magnitude of the WAL-corrected quadratic low-field MR. However, anomalies in high field measurements of the resistivity tensor of Na₃Bi (including $\vec{E} \perp \vec{B}$ comparable to the present geometry) have been previously noted and attributed to a B -dependent transport lifetime τ_{tr} [25]. If the $\tau_{tr}(B)$ dependence is identical for electrons and holes, only the ρ_{xx} component is affected, explaining why we achieve good agreement in fitting only the ρ_{xy} component. MR data and further detail is provided in the Supplemental Material [14].

The magnitude of the charge inhomogeneity n_{rms} can be used to estimate the potential fluctuations in the sample due to charge puddling, E_{rms} . Taking the Fermi velocity to be in the range $v_F = (1.4\text{--}2.4) \times 10^5$ m/s [12,26] we obtain $E_{rms} = 8.9\text{--}15$ meV and $4\text{--}6.9$ meV for the inhomogeneous ($n \ll n_{rms}$) and homogeneous ($n \gg n_{rms}$) limits, respectively; we expect that the actual value lies between these limits as $n \sim n_{rms}$. These values are 1–3 times larger than measured from direct measurements of the charge puddles [27]. This is not surprising because the amorphous SiO₂ substrate used in this case should introduce considerably more structural disorder than the atomically flat substrates used in the STM study, consistent with the larger discrepancy in step height observed in Fig. 1(b) compared to previous work [12].

The hole conductivity coefficient A_h is an order of magnitude larger than the electron coefficient A_e , implying a hole mobility an order of magnitude higher than electron mobility. This is already implied by the nonlinear $\rho_{xy}(B)$: at low field, ρ_{xy} is the mobility-weighted average and is dominated by the high-mobility minority holes, and the slope is negative (p type). At fields higher than the inverse mobility of the holes, the Hall angle for the holes approaches $\pi/2$, and the lower mobility but higher concentration electrons begin to dominate producing a tendency to positive slope (n type). Higher mobility for holes compared to electrons is unusual in conventional semiconductors. However, the TDS band

structure results from a band inversion, in which the shallower maximum of the more massive p band occurs above the Fermi energy, and the deeper minimum of the less massive s band occurs below the Fermi energy. This naturally results in a situation where the hole velocity exceeds the electron velocity [3].

We note that temperature-dependent transport measurements on bulk single crystal Na₃Bi have shown the dominance of hole-like transport above 80 K [28]. Similar hole dominance at a higher transition temperature has recently been observed in thin-film samples [29]. It is possible in that case that thermal activation of high-mobility holes causes a p -type Hall signal in net electron-doped crystals, similar to the effect of inhomogeneity observed here.

V. CONCLUSION

In summary, we have demonstrated the growth of high-quality, large-area thin films of Na₃Bi directly on amorphous SiO₂ on Si realizing a field-effect transistor structure, a significant step toward TDS devices. The gate voltage-dependent conductivity shows a nonzero minimum, and the Hall carrier density shows a transition from electron-like to hole-like, and a divergence to positive (negative) for gate voltages above (below) the minimum conductivity. The gate-voltage- and magnetic-field-dependent longitudinal and Hall conductivity are well described with the recently developed theory of charge inhomogeneity in a Dirac semimetal system using physically realistic parameters [21]. The modeling indicates that holes have higher mobility than electrons, a natural consequence of the inverted band structure of Na₃Bi. As a result, the minimum conductivity and sign change of the Hall effect occur when the sample is still net n type.

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