



## Electron doped layered nickelates: Spanning the phase diagram of the cuprates

Antia S. Botana,<sup>1</sup> Victor Pardo,<sup>2,3</sup> and Michael R. Norman<sup>1,\*</sup>

<sup>1</sup>Materials Science Division, Argonne National Laboratory, Argonne, Illinois 60439, USA

<sup>2</sup>Departamento de Física Aplicada, Universidade de Santiago de Compostela, E-15782 Santiago de Compostela, Spain

<sup>3</sup>Instituto de Investigaciones Tecnológicas, Universidade de Santiago de Compostela, E-15782 Santiago de Compostela, Spain

(Received 3 May 2017; revised manuscript received 15 June 2017; published 20 July 2017)

$\text{Pr}_4\text{Ni}_3\text{O}_8$  is an overdoped analog of hole-doped layered cuprates. Here we show via *ab initio* calculations that Ce-doped  $\text{Pr}_4\text{Ni}_3\text{O}_8$  ( $\text{Pr}_3\text{CeNi}_3\text{O}_8$ ) has the same electronic structure as the antiferromagnetic insulating phase of parent cuprates. We find that substantial Ce doping should be thermodynamically stable and that other  $4+$  cations would yield a similar antiferromagnetic insulating state, arguing this configuration is robust for layered nickelates of low-enough valence. The analogies with cuprates at different  $d$  fillings suggest that intermediate Ce-doping concentrations near  $1/8$  should be an appropriate place to search for superconductivity in these low-valence Ni oxides.

DOI: [10.1103/PhysRevMaterials.1.021801](https://doi.org/10.1103/PhysRevMaterials.1.021801)

Cuprate-like electronic structures have been pursued in other oxides since high temperature superconductors were discovered [1–4]. Cuprates are layered materials with an underlying  $\text{CuO}_2$  square lattice. The parent compounds ( $\text{Cu}^{2+}$ :  $d^9$ ,  $S = 1/2$ ) have  $d_{x^2-y^2}$  as the active orbitals, with strong antiferromagnetic correlations, as well as a substantial hybridization of Cu- $d$  and O- $p$  states. Upon doping the  $\text{CuO}_2$  planes away from a stoichiometric  $\text{Cu}^{2+}$  oxidation state, the antiferromagnetic state is suppressed, then superconductivity appears, and at higher dopings a Fermi liquid phase arises [2]. A schematic phase diagram with respect to filling of the  $d$  levels for hole doping is shown in Fig. 1.

Given the proximity of Ni and Cu in the periodic table, an interesting conjecture is whether these phenomena could also occur in nickelates since, if so, this could tell us a lot about the nature of high temperature superconductivity.  $\text{LaNiO}_3$ -based heterostructures [5,6] and doped single layer nickelates ( $\text{La}_{2-x}\text{Sr}_x\text{NiO}_4$ ) [7,8] have been intensively studied, but the challenge is finding a nickelate that has a similar electron count ( $\text{Ni}^{1+}$  and  $\text{Cu}^{2+}$  being isoelectronic) with Ni in a square planar environment.

In this context, the recently synthesized low-valence layered nickelates ( $\text{R}_4\text{Ni}_3\text{O}_8$ , R438, with R = La or Pr) with an average Ni valence of 1.33, are promising candidates to investigate [9–16]. Particularly, Pr438, a  $1/3$  self hole doped (relative to  $d^9$ ) line compound, stands among the closest bulk analogs to high- $T_c$  cuprates. It shows metallic behavior and a large orbital polarization with holes in the planar  $d_{x^2-y^2}$  states, as well as a high degree of O- $p$ -Ni- $d$  hybridization [17]. Figure 1 illustrates where metallic Pr438 lies within the cuprate phase diagram, sitting in the overdoped Fermi liquid region.

Electron doping this new phase is an obvious strategy to explore the analogy with cuprates. If we consider as a starting point that the physics of high- $T_c$  superconductivity in cuprates is brought about by doping a two-dimensional (2D) Mott insulator with strong antiferromagnetic correlations, a natural question is whether this type of “parent phase” arises at  $d^9$  filling for electron-doped Pr438 [18]. As was pointed out almost 20 years ago by Anisimov *et al.*, [19] “only if the

Ni ions are forced into a planar coordination with the O ions can a  $S = 1/2$  magnetic insulator be realized with the difficult  $\text{Ni}^{1+}$  oxidation state.” Here, we show using density functional theory-based (DFT) calculations that doping  $\text{Pr}_4\text{Ni}_3\text{O}_8$  with a  $4+$  ion allows going from a Fermi liquid phase at a nominal  $d^{8.67}$  filling to a parent antiferromagnetic insulating phase at  $d^9$  filling, effectively spanning the hole-doped cuprate phase diagram (see Fig. 1).

Our electronic structure calculations were performed within DFT using the all-electron, full potential code WIEN2K based on the augmented plane wave plus local orbitals (APW + lo) basis set [20,21]. For the structural relaxations the Perdew-Burke-Ernzerhof version of the generalized gradient approximation (GGA) has been used [22]. Volume and  $c/a$  optimizations, as well as a full relaxation of all the internal atomic coordinates, were carried out. To deal with strong correlation effects, the LDA +  $U$  scheme that incorporates an on-site Coulomb repulsion  $U$  and Hund’s rule coupling strength  $J_H$  has been applied [23]. We use typical values:  $U = 5$  eV for Ni- $3d$ ,  $U = 8$  eV for Pr- $4f$ , and  $U = 8$  eV for Ce- $4f$ .  $J_H$  has been set to 0.7 eV for Ni- $d$  and 1 eV for Pr and Ce- $f$ . Experience with the cuprates and other transition metal oxides shows LDA +  $U$  to be a reliable method to predict electronic structures [24–26].

Pr438 crystallizes in a tetragonal unit cell [17] with space group  $I4/mmm$  (no. 139) and lattice parameters  $a = 3.9347$  Å,  $c = 25.4850$  Å. The structure, pictured in Fig. 2, consists of three  $\text{NiO}_2$  infinite-layer (IL) planes separated by layers of Pr ions. On either side of this trilayer lies a fluorite (F) Pr- $\text{O}_2$ -Pr block. With an average Ni valence of  $1.33 + (d^{8.67})$  Pr438 is isoelectronic with (hole) overdoped cuprates involving a mixture of  $\text{Ni}^{1+}$  and  $\text{Ni}^{2+}$  ions. The electronic structure of Pr438 (described in Ref. [17]) is found to be metallic, with Ni- $d_{x^2-y^2}$  bands hybridized with O- $p$  states that cross the Fermi level giving rise to a cuprate-like Fermi surface (see Fig. 3).

By substituting trivalent Pr with a tetravalent ion in Pr438, the electron count on the  $\text{NiO}_2$  layers can be changed, moving the material towards the left of the phase diagram shown in Fig. 1. In principle, the ionic count required for reaching optimal doping within the superconducting phase would need  $\sim 1/8$  substitution. We do not describe calculations for this

\*norman@anl.gov

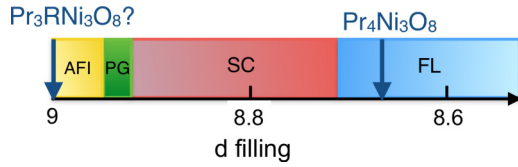


FIG. 1. A schematic diagram representing some of the phases of the cuprates as a function of nominal filling of the  $d$  levels in the hole-doped region: antiferromagnetic insulating (AFI), pseudogap (PG), superconducting (SC), and Fermi liquid (FL).  $\text{Pr}_4\text{Ni}_3\text{O}_8$  and  $\text{Pr}_3\text{RNi}_3\text{O}_8$  (R representing a 4+ cation, namely Ce) are represented at the appropriate  $d$  fillings.

doping since the complicated band structure of the resulting supercell would not be instructive, but as shown in Fig. 3, a rigid-band shift of Pr438 corresponding to such a substitution (equivalent to 1/6 electron doping,  $d^{8.83}$ ) results in a Fermi surface similar to that of optimal doped cuprates [2].

Reaching the parent insulating state (with  $\text{Ni}^{1+}$ :  $d^9$ ) would instead need 25% substitution, giving rise to  $\text{Pr}_3\text{RNi}_3\text{O}_8$ , R = Ce, Th. It is important to remark that low oxidation state  $\text{Ni}^{1+}$  is hard to stabilize. The bulk synthesis of the  $\text{Ni}^{1+}$  infinite-layer compound  $\text{LaNiO}_2$  is difficult and because of these materials issues, it is unclear whether an insulating state is achieved [27–29]. LDA +  $U$  calculations give a stable solution with antiferromagnetic ordering of  $S = 1/2$  ions [19], but experimentally there is no evidence for long-range order. In principle, the on-site energy difference between the  $p$  and  $d$  levels is smaller for  $\text{Cu}^{2+}$  than for  $\text{Ni}^{1+}$ , favoring larger hybridizations in the former. However, we will see that substantial oxygen hybridization can occur for the layered nickelates studied here.

In Pr438 there are two different Pr sites: Pr1 within the  $\text{NiO}_2$  trilayer and Pr2 within the fluorite blocks that separate  $\text{NiO}_2$  trilayers (see Fig. 2). Both types of substitutions were tried,

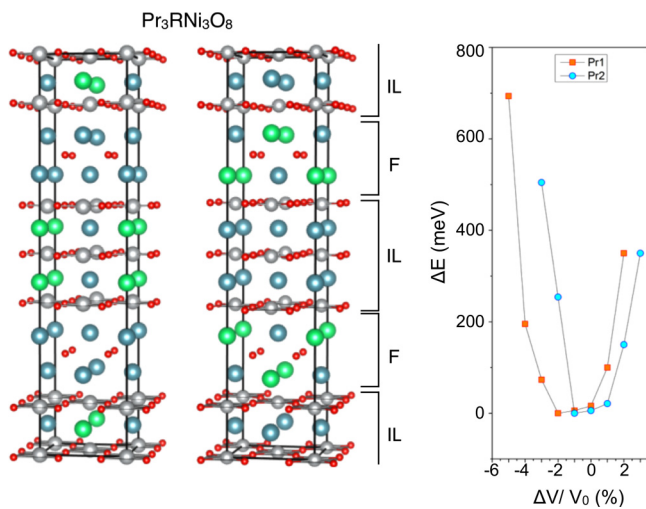


FIG. 2. Left panels: Structure of the  $\sqrt{2}a \times \sqrt{2}a \times c$  cells for  $\text{Pr}_3\text{CeNi}_3\text{O}_8$  with Ce substitution on both Pr1 and Pr2 sites (infinite layer (IL) and fluorite (F) blocks, respectively). Ce atoms are in green, Pr atoms in blue, Ni atoms in gray, and O atoms in red. Right panel: Energy versus volume curves for both structures obtained within GGA-PBE.  $V_0$  represents the experimental volume of Pr438.

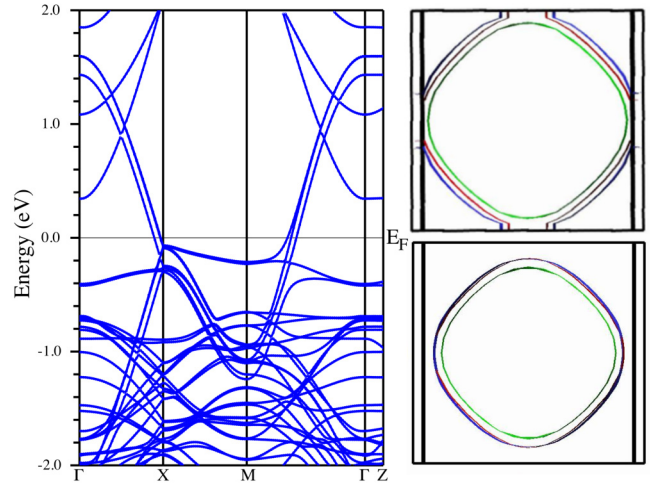


FIG. 3. Left panel: GGA band structure of Pr438. Right panel: Fermi surface of Pr438 (top) and 1/6 electron doped Pr438 (bottom), the latter obtained by a rigid band adjustment of the Fermi energy of Pr438. The center of the plot is at  $(\pi/a, \pi/a)$ .

giving rise to a very similar outcome for the electronic structure (which will be described in detail below). Concerning the most appropriate 4+ dopant atom, Ce is the most obvious choice given its proximity to Pr in the periodic table and because Ce doping has been realized in  $T'$ -cuprates giving rise to electron-doped phases. For instance,  $T'$ - $\text{Pr}_{2-x}\text{Ce}_x\text{CuO}_4$  shows superconductivity with an optimum doping level  $x = 0.135$  and  $T_c = 27$  K.[30–32] In  $T'$ - $\text{Nd}_2\text{CuO}_4$  superconductivity develops by doping with both Ce and Th [33,34].

To test the possibility of in-plane checkerboard magnetic (AFM) ordering in  $\text{Pr}_3\text{CeNi}_3\text{O}_8$ , a  $\sqrt{2}a \times \sqrt{2}a \times c$  cell was used. After the substitution of Pr by Ce in a 3:1 ratio, the optimized volume results in a reduction with respect to that of Pr438 of 1 to 2% (see Fig. 2)—the ionic radius of  $\text{Ce}^{4+}$  in an 8-fold coordination is 0.97 Å vs the ionic radius of  $\text{Pr}^{3+}$  of 1.13 Å, so a decrease in lattice constants can be anticipated [35]. Small distortions of the Ni-O and Pr/Ce-O distances take place after the structural relaxation as shown in Table I. Ce substitution within the F block is energetically favorable

TABLE I. Nearest neighbor Pr-O, Ce-O, and Ni-O distances for  $\text{Pr}_3\text{CeNi}_3\text{O}_8$  in Å for Ce substitution on the Pr1 and Pr2 sites. The Pr/Ce environment is square antiprismatic with eight oxygen ions as nearest neighbors. The R ion is positioned asymmetrically inside the polyhedra. For this reason, two different R-O distances are listed in the table. In undoped Pr438 the Pr-O distances are (in Å) 2.48, 2.59 for Pr1 (IL), and 2.32, 2.74 for Pr2 (F). The Ni is in a square planar environment of oxygens. For undoped Pr438 the Ni-O distance is 1.97 Å.

	$\text{Pr}_3\text{CeNi}_3\text{O}_8$ (IL)	$\text{Pr}_3\text{CeNi}_3\text{O}_8$ (F)	
Ce–IL	2.39, 2.54	Ce–F	2.30, 2.63
Pr–IL	2.46, 2.59	Pr–F	2.40, 2.63
Pr–F	2.33, 2.74	Pr–IL	2.53, 2.59
Pr–F	2.32, 2.79	Pr–IL	2.55, 2.58
Ni	1.96	Ni	1.96

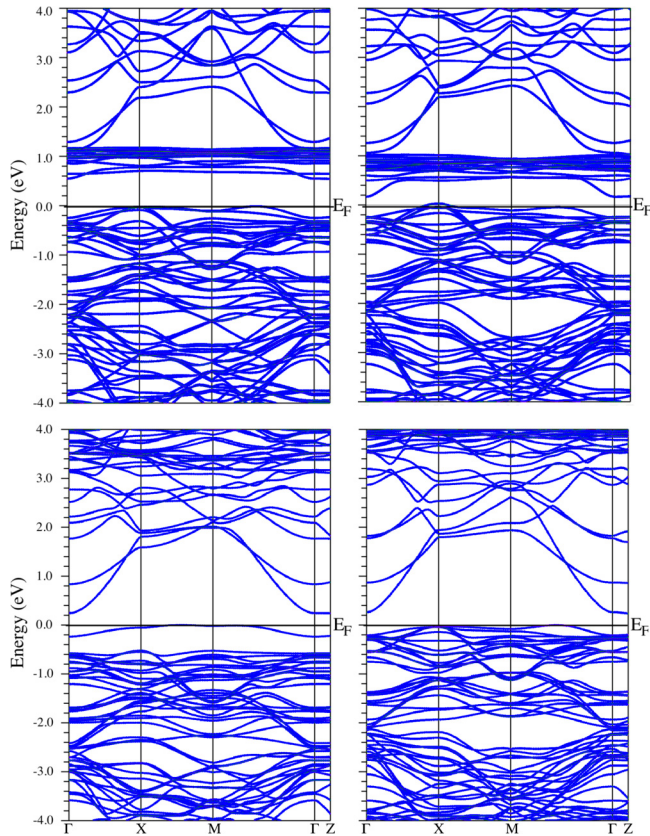


FIG. 4. Band structures for  $\text{Pr}_3\text{RNi}_3\text{O}_8$  within LDA +  $U$  ( $R = \text{Ce}$ , top panels;  $R = \text{Th}$ , bottom panels).  $R^{4+}$  doping within the IL (left panels) and F blocks (right panels) with an in-plane AFM ordering of Ni ions.

by 96 meV/unit cell with respect to substitution in the IL block. The standard enthalpy change  $\Delta H$  can be obtained at  $T = 0$  from the calculated total energies:  $E(\text{Pr}_3\text{CeNi}_3\text{O}_8) + \frac{3}{4}E(\text{O}_2) - \frac{3}{2}E(\text{Pr}_2\text{O}_3) - E(\text{CeO}_2) - 3E(\text{NiO})$  giving a value of -814 kJ/mol [36].

Within GGA, the band structure is metallic with an in-plane AFM coupling within the  $\text{NiO}_2$  planes. Application of a Coulomb  $U$  to the Ni- $d$ , Pr- $f$ , and Ce- $f$  states opens a gap of  $\sim 0.6$  eV for doping in the IL block and  $\sim 0.3$  eV in the F block (see Fig. 4). The gap is smaller for substitution in the F block due to the unoccupied Ni- $d$  states from the outer planes being closer in energy to the Ce- $f$  bands, leading to a level repulsion effect (see Fig. 4). The magnetic moments inside the Ni muffin-tin sphere are approximately  $\pm 0.9 \mu_B$  with checkerboard (AFM) ordering within the  $\text{NiO}_2$  planes as well as between planes. All in all, the LDA +  $U$  band structure, shown in Fig. 4, exhibits a correlated  $S = 1/2$  AFM insulator analog of the parent cuprate materials. Holes would hence go into the Ni- $d$  band of  $x^2-y^2$  character that is highly hybridized with O- $p$  states (see below). Calculations confirm that this state comprised of AFM-ordered Ni layers is more stable than a ferromagnetic one, giving rise to a superexchange interaction  $J \sim 1200$  K, comparable to that of cuprates.

In general, Ce can be trivalent as well as tetravalent. Using the same structural parameters and  $U$  values described above, as well as a checkerboard AFM configuration within the  $\text{NiO}_2$

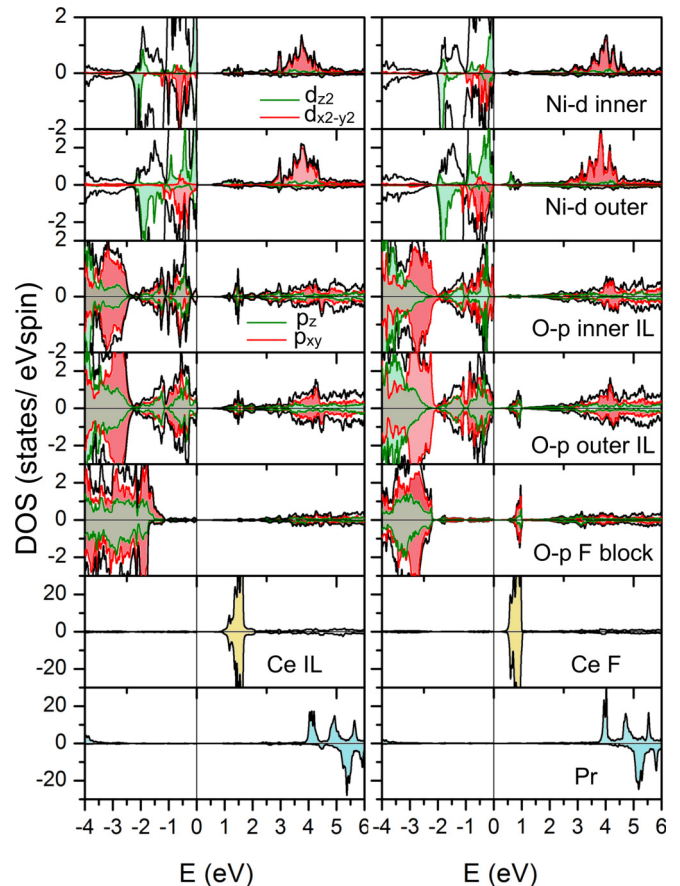


FIG. 5. Partial density of states for  $\text{Pr}_3\text{CeNi}_3\text{O}_8$  within LDA +  $U$  (top parts of panels are for spin up, bottom for spin down).  $\text{Ce}^{4+}$  doping within the IL block (left) and within the fluorite block (right). We find that the Ce cation slightly bonds with neighboring O ions. The unoccupied Ni- $d$  bands lying at about 3 eV above the Fermi level have substantial O character, as a result of the strong  $\sigma$ -bond with neighboring O anions.

planes, a solution where Ce is 3+ was obtained. However, this state (in which a magnetic moment of  $1 \mu_B$  develops on the Ce) turns out to be energetically unfavorable by 0.34 eV/unit cell with respect to that where Ce is tetravalent. An in-trimer charge ordered solution results with AFM ordering in the  $\text{NiO}_2$  planes:  $\text{Ni}^{1+}$  in the outer layers, and  $\text{Ni}^{2+}$  (high spin) in the inner layers [37].

As can be observed in Fig. 5, for  $\text{Ce}^{4+}$  substitution, the top of the valence band consists of a mixture of  $d_{x^2-y^2}$  and  $d_{z^2}$  states, whereas the Ni- $d$  unoccupied bands are  $x^2-y^2$  only in character. At around 1 eV above the Fermi level lie the Ce- $f$  states whereas Pr- $f$  states appear mostly above  $\sim 4$  eV. It is clear from the density of states (DOS) that both inner and outer  $\text{Ni}^{1+}$ :  $d^9$  ( $S = 1/2$ ) ions have a single hole in the minority-spin  $d_{x^2-y^2}$ -like band. Hybridization between the Ni- $d$  and O- $p$  states is clear in the DOS, being most noticeable for the planar O ions as a result of the strong  $pd\sigma$  bond. The lobes of the  $x^2-y^2$  orbital point directly to the  $p$  orbital of the neighboring oxygen, forming a strong covalent bond with a large hopping integral  $t_{pd} = 0.3$  eV. Let us recall that bond covalency sets the unusually high energy scale for the exchange interaction in cuprates. Small moments are induced on the oxygen ions as a

consequence of hybridization along with low-site symmetry:  $\mu_{O1} = -0.05 \mu_B$  (inner NiO<sub>2</sub> plane),  $\mu_{O2} = -0.07 \mu_B$  (outer NiO<sub>2</sub> plane),  $\mu_{O3} = -0.02 \mu_B$  (F block). Importantly, the Ce cation bonds with neighboring O ions (as implied in the DOS, clearly reflecting the hybridization between O-*p* and Ce-*f* states). Previous results show that Ce substitution in R<sub>2</sub>CuO<sub>4</sub> (R = Pr, Nd) enhances the phase stability given that tetravalent Ce has a firmer grip on surrounding O<sup>2-</sup> ions than trivalent rare earth ions [34,38,39].

To check the robustness of the obtained electronic structure, Th substitution on the Pr site was also tried as well as smaller 4+ cations like Zr and Te. For Th substitution, the same type of calculations give rise to a smaller volume reduction of only 0.5–1% (the ionic radius of tetravalent Th<sup>4+</sup> in an 8-fold environment is 1.06 Å, larger than that for Ce<sup>4+</sup>) [35]. The electronic structure is very similar except for the *f* states missing at the tail of the unoccupied Ni-*d* states (see Fig. 4) arguing that the derived insulating AFM state is robust in Pr<sub>3</sub>RNi<sub>3</sub>O<sub>8</sub>. Zr and Te are smaller ions [35] so substitution with these gives rise to bigger distortions in the structure. However, the main features of the electronic structure obtained for Th doping remain for both Zr and Te doping.

If the likelihood of promoting superconductivity in these layered nickelates is to be analyzed, the similarities with cuprates should be characterized. The state we obtained in

Pr<sub>3</sub>CeNi<sub>3</sub>O<sub>8</sub> (for a Ni<sup>1+</sup> oxidation state) has insulating, AFM planar order in common with the cuprates, as well as obvious similarities in structure with common MO<sub>2</sub> (M = Ni, Cu) square lattice planes and a similar superexchange interaction. Even though we are dealing with hypothetical structures, our calculations show that substantial Ce doping should be thermodynamically stable and that several other 4+ cations would yield a very similar antiferromagnetic insulating state, suggesting this configuration is robust in layered nickelates of low-enough valence. Our results show that planar nickelates directly analogous to the high-*T<sub>c</sub>* superconducting cuprates are a realistic possibility with intermediate Ce-doping concentrations near 1/8 being the appropriate place to search for superconductivity.

The authors acknowledge fruitful discussions with Peter Blaha, Warren Pickett, and John Mitchell. Work at Argonne was supported by the Materials Sciences and Engineering Division, Basic Energy Sciences, Office of Science, U.S. DOE. VP thanks MINECO of Spain for financial support through project No. MAT2016-80762-R. We acknowledge the computing resources provided on Blues, the high-performance computing clusters operated by the Laboratory Computing Resource Center at Argonne National Laboratory.

- 
- [1] M. R. Norman, *Rep. Prog. Phys.* **79**, 074502 (2016).  
 [2] W. E. Pickett, *Rev. Mod. Phys.* **61**, 433 (1989).  
 [3] B. Keimer, S. A. Kivelson, M. R. Norman, S. Uchida, and J. Zaanen, *Nature* **518**, 179 (2015).  
 [4] E. Dagotto, *Rev. Mod. Phys.* **66**, 763 (1994).  
 [5] J. Chaloupka and G. Khaliullin, *Phys. Rev. Lett.* **100**, 016404 (2008).  
 [6] S. Middey, J. Chakhalian, P. Mahadevan, J. Freeland, A. Millis, and D. Sarma, *Annu. Rev. Mater. Res.* **46**, 305 (2016).  
 [7] R. Kajimoto, K. Ishizaka, H. Yoshizawa, and Y. Tokura, *Phys. Rev. B* **67**, 014511 (2003).  
 [8] R. Zhong, B. L. Winn, G. Gu, D. Reznik, and J. M. Tranquada, *Phys. Rev. Lett.* **118**, 177601 (2017).  
 [9] A. S. Botana, V. Pardo, W. E. Pickett, and M. R. Norman, *Phys. Rev. B* **94**, 081105 (2016).  
 [10] J. Zhang, Y.-S. Chen, D. Phelan, H. Zheng, M. R. Norman, and J. F. Mitchell, *Proc. Natl. Acad. Sci. USA* **113**, 8945 (2016).  
 [11] V. Pardo and W. E. Pickett, *Phys. Rev. Lett.* **105**, 266402 (2010).  
 [12] V. V. Poltavets *et al.*, *Phys. Rev. Lett.* **104**, 206403 (2010).  
 [13] P. Lacorre, *J. Solid State Chem.* **97**, 495 (1992).  
 [14] N. ApRoberts-Warren, A. P. Dioguardi, V. V. Poltavets, M. Greenblatt, P. Klavins, and N. J. Curro, *Phys. Rev. B* **83**, 014402 (2011).  
 [15] S. Sarkar, I. Dasgupta, M. Greenblatt, and T. Saha-Dasgupta, *Phys. Rev. B* **84**, 180411 (2011).  
 [16] H.-M. Guo, T. Mendes-Santos, W. E. Pickett, and R. T. Scalettar, *Phys. Rev. B* **95**, 045131 (2017).  
 [17] J. Zhang, A. S. Botana, J. W. Freeland, D. Phelan, H. Zheng, V. Pardo, M. R. Norman, and J. F. Mitchell, *Nat. Phys.*, doi:10.1038/nphys4149.  
 [18] P. A. Lee, N. Nagaosa, and X.-G. Wen, *Rev. Mod. Phys.* **78**, 17 (2006).  
 [19] V. I. Anisimov, D. Bukhvalov, and T. M. Rice, *Phys. Rev. B* **59**, 7901 (1999).  
 [20] P. Blaha, K. Schwarz, G. K. H. Madsen, D. Kvasnicka, and J. Luitz, *WIEN2k, An Augmented Plane Wave Plus Local Orbitals Program for Calculating Crystal Properties* (Vienna University of Technology, Austria, 2001).  
 [21] E. Sjöstedt, L. Nördstrom, and D. Singh, *Solid State Commun.* **114**, 15 (2000).  
 [22] J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).  
 [23] A. I. Liechtenstein, V. I. Anisimov, and J. Zaanen, *Phys. Rev. B* **52**, R5467 (1995).  
 [24] V. I. Anisimov, S. Y. Ezhov, and T. M. Rice, *Phys. Rev. B* **55**, 12829 (1997).  
 [25] V. I. Anisimov, M. A. Korotin, A. S. Mylnikova, A. V. Kozhevnikov, D. M. Korotin, and J. Lorenzana, *Phys. Rev. B* **70**, 172501 (2004).  
 [26] P. Blaha, K. Schwarz, and P. Novk, *Int. J. Quantum Chem.* **101**, 550 (2005).  
 [27] M. Crespín, P. Levitz, and L. Gatinéau, *J. Chem. Soc., Faraday Trans. 2* **79**, 1181 (1983).  
 [28] M. A. Hayward, M. A. Green, M. J. Rosseinsky, and J. Sloan, *J. Am. Chem. Soc.* **121**, 8843 (1999).  
 [29] K.-W. Lee and W. E. Pickett, *Phys. Rev. B* **70**, 165109 (2004).  
 [30] Y. Tokura, H. Takagi, and S. Uchida, *Nature* **337**, 345 (1989).  
 [31] D. J. Scalapino, *Rev. Mod. Phys.* **84**, 1383 (2012).

- [32] Y. Krockenberger, J. Kurian, A. Winkler, A. Tsukada, M. Naito, and L. Alff, *Phys. Rev. B* **77**, 060505 (2008).
- [33] H. Takagi, S. Uchida, and Y. Tokura, *Phys. Rev. Lett.* **62**, 1197 (1989).
- [34] M. Naito, Y. Krockenberger, A. Ikeda, and H. Yamamoto, *Physica C: Superconductivity and Its Applications* **523**, 28 (2016).
- [35] R. D. Shannon and C. T. Prewitt, *Acta Crystallogr. Sect. B* **25**, 925 (1969).
- [36] Preliminary results (Junjie Zhang, private communication) indicate that Ce can be doped into R438 nickelates (R = La, Pr).
- [37] H. Wu, *New J. Phys.* **15**, 023038 (2013).
- [38] Y. Idemoto, I. Oyagi, and K. Fueki, *Physica C: Superconductivity* **195**, 269 (1992).
- [39] H. Kanai, J. Mizusaki, H. Tagawa, S. Hoshiyama, K. Hirano, K. Fujita, M. Tezuka, and T. Hashimoto, *J. Solid State Chem.* **131**, 150 (1997).