

Direct Measurement of Sheet Resistance R_{\square} in Cuprate Systems: Evidence of a Fermionic Scenario in a Metal-Insulator Transition

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The metal-insulator transition (MIT) has been studied in $\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{2+x}/\text{CaCuO}_2$ ultrathin cuprate structures. Such structures allow for the direct measurement of the 2D sheet resistance R_{\square} , eliminating ambiguity in the definition of the effective thickness of the conducting layer in high temperature superconductors. The MIT occurs at room temperature for experimental values of R_{\square} close to the 25.8 k Ω universal quantum resistance. All data confirm the assumption that each CaCuO_2 layer forms a 2D superconducting sheet within the superconducting block, which can be described as weak-coupled equivalent sheets in parallel.

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The basic features of the metal-insulator transition (MIT) in high temperature superconductors (HTSs) remain the subjects of ongoing discussion. In ultrathin films of low temperature superconductors (LTSs), the onset of superconductivity was found to occur when the normal-state sheet resistance R_{\square} was smaller than the critical value of $h/4e^2 = 6.45 \text{ k}\Omega$ [1–3]. A theoretical model based on quantum fluctuations of the order-parameter phase difference in a network of conducting clusters coupled by Josephson tunnelling junctions has been proposed [4,5] to explain MIT in ultrathin films. In this model the increase of the normal-state resistance is associated with an increase in quantum fluctuations of the superconducting phase. Eventually, these fluctuations destroy the global phase coherence and lead to an insulating state. It can be shown that a localization of the phase difference across the junction which occurs when the normal-state sheet resistance falls below a threshold value [6–9] R_{\square}^{Tr} of $h/4e^2$ gives rise to the superconductivity [10,11]. Recently it has been unambiguously verified that the MIT transition in these systems is driven by changes in intrinsic dissipation rather than being related to the film disorder [3]. An alternative explanation for the MIT is related to the renormalization of the interelectron interaction in the Cooper channel by the long-range Coulomb repulsion specific to dirty two-dimensional (2D) superconductors. In this case, it can be shown that the MIT occurs when the charge carrier density is so low that the mean free path of the electrons becomes smaller than the interatomic distance ($k_F l \approx 1$). Such a localization phenomenon occurs when R_{\square} approaches the critical value $R_{\square}^{\text{Tr}} = 25.8 \text{ k}\Omega$ [12]. When $R_{\square}^{\text{Tr}} = h/4e^2$ these systems are termed bosonic since the carriers are treated as pairs, and when $R_{\square}^{\text{Tr}} = h/e^2$ these systems are

termed fermionic since the carriers are treated as single electrons.

The normal-state resistance behavior as a function of temperature appears to be quite different for the two classes of superconductors. For LTSs, the derivative $\Delta R/\Delta T$ is positive, though very small in most cases, over the whole temperature range, while, on the contrary, negative values have been measured at low temperatures for many HTSs [13]. The bosonic nature of the MIT for the LTSs is widely accepted. However, the critical normal-state sheet resistance frequently reported in the literature [14,15] for HTSs raised concerns about the nature of the MIT in such systems because it was usually larger than $h/4e^2$. A conclusive picture of the MIT in HTS systems is still missing, even though studies have been done on several different high- T_c cuprates, such as irradiated YBaCuO and BiSrCaCuO [16], Zn-doped LaBaCaCuO [17], Y-doped BiSrCaCuO [18], Ce-doped NdCuO [19], oxygen-deficient YBaCuO single crystals [20], Pr-doped or Zn-doped YBaCuO [21,22], and ultrathin DyBaCuO films [23].

An important uncertainty related to the structural features of all HTSs [24] is the ambiguity in the definition of the effective thickness t of the conducting layer. This value, which is necessary to calculate R_{\square} from the resistivity ρ value, has been variably chosen as half of the c -axis lattice constant ($t = 15 \text{ \AA}$) in $\text{Bi}_2\text{Sr}_2\text{Y}_x\text{Ca}_{1-x}\text{Cu}_2\text{O}_8$ [18], as the whole c -axis lattice constant ($t = 11.8 \text{ \AA}$) in $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ [20], or as the spacing between CuO_2 layers ($t = 6 \text{ \AA}$) in $\text{Nd}_{2-x}\text{Ce}_x\text{CuO}_4$ [19]. This ambiguity is crucial for judging whether the bosonic or fermionic explanation of the 2D MIT is realized in practice. Ultrathin artificial cuprate structures, as discussed here, offer unique

opportunities to define the proper regime. In a previous work on the HTS superlattices $(\text{BaCuO}_{2+x})_m/(\text{CaCuO}_2)_n$ ($\text{Cbcco}-m \times n$) both the dependence of the critical temperature and the electrical transport properties on n for n ranging from 1 to 15 (with $m = 2$) have been investigated [25]. These artificial structures were grown by pulsed laser deposition in order to stack in sequence individual non-superconducting layers of BaCuO_{2+x} and CaCuO_2 . In the $\text{Cbcco}-m \times n$ unit cell, the superconducting block consists of n epitaxial layers of CaCuO_2 while the m epitaxial layers of BaCuO_{2+x} play the role of a charge reservoir block [26]. In the case of samples with ultrathick superconducting blocks [25], the ambiguity in determining the critical normal-state sheet resistance at which the MIT occurs was minimal. To calculate R_{\square} for each superlattice, the formula $R_{\square} = \rho/t$ was used, where t can be either the superlattice modulation length Λ or the thickness of the Ca-based infinite layer (IL) block alone. For the insulating $\text{Cbcco}-2 \times 14$ superlattice, t is equal to 44.8 Å if we consider the IL block alone or 53.6 Å for the whole superlattice cell. This leads to an uncertainty of about 15% on the R_{\square} value. For comparison, in the case of YBCO, the whole unit cell is 3 times larger than the superconducting Y-based block. The critical temperature T_c versus the sheet resistance R_{\square} for Cbcco ultrathick superlattices [25] showed that the MIT takes place for a sheet resistance of about $\approx 26 \text{ k}\Omega$ [27]. However, even if the ultrathick cuprate superlattices have shown a good reliability in the precise determination of R_{\square} values at the MIT, a direct measure of R_{\square} has been never performed on a single-layer HTS film.

Recently it has been shown that high- T_c superconductivity can be achieved in ultrathin CaCuO_2 layers at the surface of $\text{SrTiO}_3/\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{2+x}/\text{CaCuO}_2$ and in the trilayered $\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{2+x}/\text{CaCuO}_2/\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{2+x}$ heteroepitaxial structures [28,29]. In order to obtain a more stable structure for such ultrathin structures, 10% of the Ba atoms are substituted by the trivalent Nd cations, slightly increasing the compensation of the electrical charge. Both the structures consist of only one superconducting Ca-based block of N layers of CaCuO_2 . However, in the case of the bi-layered structures, the Ca-based block is grown on top of a single Ba-based block of M layers of $\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{2+x}$, while in the case of the trilayered structures, the Ca-based block is sandwiched between two Ba-based blocks. From now on, we will refer to these as the $\text{Cbcco}-M/N$ and the $\text{Cbcco}-M/N/M$ structures, respectively. Here we show that such a superconducting ultrathin structure allows a direct measurement of R_{\square} in a HTS system, clarifying some important issues regarding the MIT in HTS systems.

A preliminary argument to be addressed is the distribution of the electrical charge coming from the charge reservoir block within the superconducting block. In $\text{Cbcco}-2 \times n$ structures, the thickness of the superconducting blocks has been varied over a wide range, well beyond the possibility offered by conventional HTS materials. n

varies from 1 to 15 while in standard HTS $n \leq 3$. Among the $\text{Cbcco}-2 \times n$ structures, the $\text{Cbcco}-2 \times 2$ superlattices proved to be optimally doped and showed the highest superconducting transition temperature with zero resistance above 80 K. For thicker CaCuO_2 blocks the critical temperature decreases until for $n > 11$ the artificial structure is no longer superconducting. Such an effect was explained by considering the decrease of the effective carrier concentration per CuO_2 plane in thick Ca-based blocks. This result is consistent with a picture where holes injected from the charge reservoir block do not localize at the interfaces, but rather distribute quite homogeneously over the whole superconducting block. This can be confirmed by plotting T_c vs the number of CuO_2 planes [25] in terms of “carriers per plane”. For a uniform charge distribution within the superconducting block, the carrier concentration per CuO_2 plane is simply inversely proportional to the number of CuO_2 planes (Fig. 1).

Additional support comes from the similarity between the plot of T_c vs the number of CuO_2 planes and the universal phase diagram for the high- T_c superconductors [30–32] (see the dotted line in Fig. 1). In the case of charge localization at the interfaces between the Ba-based and the Ca-based blocks, one would expect a saturation of the critical temperature value for a sufficiently thick IL block, as in the case of twin-planes superconductivity. We find instead the maximum value of $T_c = 80 \text{ K}$ in the case of $(\text{BaCuO}_{x+2})_2/(\text{CaCuO}_2)_2$ (2×2 superlattice), which has the optimal doping value $p = 0.18\text{--}0.19$ holes per CuO_2 plane [33]. In analogy with the $\text{Cbcco}-m \times n$ superlattices, we can drive the ultrathin $\text{Cbcco}-M/N$ heteroepitaxial structures through the MIT by increasing the number of conducting CaCuO_2 , thus decreasing the doping level for a single CaCuO_2 layer. $R(T)$ measurements were performed by the standard four-probe pulse-reverse dc technique. Contacts were made by silver epoxy directly onto the substrate before the film deposition, in order to avoid any

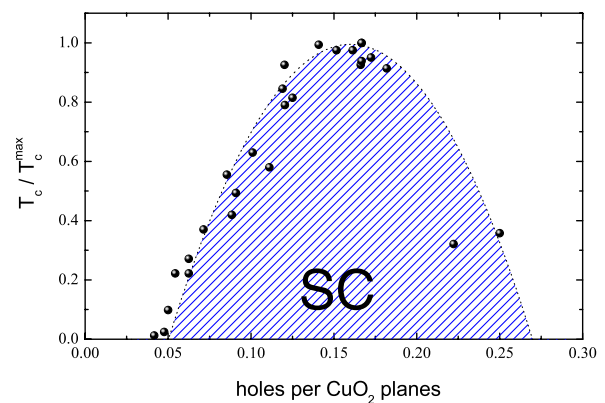


FIG. 1 (color online). Critical temperature T_c behavior (normalized to the maximum value $T_c^{\text{max}} = 80 \text{ K}$) as a function of the carrier (holes) concentration per CuO_2 plane. The universal bell-shaped curve of superconducting (SC) phase vs doping is also plotted as a dotted line.

chemical reaction between the film and the solvent in the silver epoxy. All the ultrathin structures were measured using the Van der Pauw configuration [34]. The R_{\square} was estimated as $(f \cdot \pi R_{\text{mean}} / \ln 2)$, where R_{mean} is the mean value of the Van der Pauw contacts' geometry (see the two configurations in the lower left panel of Fig. 2) and f is a prefactor value depending on R_a/R_b [34]. Because the R_a/R_b ratio has been found smaller than 3–4 for all samples, for sake of simplicity we set $f = 1$ for all the samples, resulting in a maximum overestimate of the absolute value of about 15%. In Fig. 2 the R_{\square} behavior as a function of the temperature is displayed for Cbcco-5/ N with $N = 0.66$ and 1.0. In the case of $N = 0.66$, the number of laser shots on the CaCuO₂ target was 2/3 of those required for $N = 1.0$. The accuracy of the thickness calibration was previously demonstrated by diffraction and reflectivity measurements with synchrotron radiation [35].

Comparison between the temperature dependence of R_{\square} of the Cbcco-5/0.66 and Cbcco-5/1 structures suggests that the contribution to the normal-state resistance of the Ba-based block, which is more than 5 times thicker than the Ca-based block, can be neglected. The two structures only differ by a single CaCuO₂ layer. However, the resistance value measured in the Cbcco-5/1.0 structure is almost one-order of magnitude smaller than in the case of the Cbcco-5/0.66 structure, consisting in unconnected CaCuO₂ islands grown on the Ba-based block. This leads us to conclude that the change in the resistance value is primarily due to the single CaCuO₂ layer, which shows a higher conductivity compared to the Ba-based block.

This result is particularly interesting in view of a comparison between the CaCuO₂ alone and that grown on top of a Ba-based CR block. The R_{\square} vs temperature behavior of the CaCuO₂ parent compound is also reported in the inset of Fig. 2. The electrically well-compensated CaCuO₂ block alone shows a strongly insulating behavior. In con-

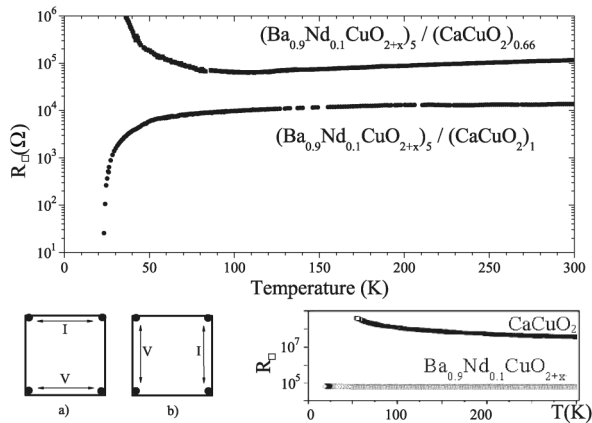


FIG. 2. (upper part) The sheet resistance R_{\square} vs temperature for the $(\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{x+2})_5/(\text{CaCuO}_2)_N$ (with $N = 0.66, 1$). (lower part) On the right, the schematic diagram of the Van der Pauw geometrical configurations of the two pads for current injection is reported; on the left, the $\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{x+2}$ and CaCuO_2 parent compounds' behavior is also reported.

trast, CaCuO₂ doped by a Ba-based block results in the highest electrical conductivity, possibly because of external doping in a well-ordered material.

Here we stress that the $R(T)$ measurements of a single CaCuO₂ layer in the Cbcco-5/1 structure is a direct measurement of 2D- R_{\square} . The $R(T)$ behavior was investigated in the Cbcco-5/ N structures with N varying from 1 to 6, thereby keeping the Ba-based block thickness constant while varying the superconducting block. Assuming a uniform distribution of the carrier concentration, the N CaCuO₂ layers can be considered as N equivalent CuO₂ conducting parallel planes. In analogy with the Cbcco-2 \times n superlattices, the disappearance of the superconductivity for samples with thick enough CaCuO₂ blocks indicates that the carrier density decreases with increasing N . In contrast to all previous measurements, we can evaluate the single CuO₂ sheet resistance by simply multiplying the resistance of the Cbcco-5/ N structures by a factor N . The R_{\square} versus temperature behavior along the Cbcco-5/ N series is shown in Fig. 3.

Again, as a consequence of the decrease of the carrier concentration per single layer, the R_{\square} resistance increases with N . The R_{\square} behavior near room temperature is shown in the inset of Fig. 3. The MIT occurs for a R_{\square} value between the superconducting Cbcco-5/4 and the insulating Cbcco-5/5 structure values. More precisely, taking into account the 15% maximum overestimate on the absolute value of the R_{\square} (previously discussed), the MIT occurs for a R_{\square} value in the range between 25.2 ± 2.0 k Ω and 32.8 ± 2.7 k Ω for the Cbcco-5/4 and the Cbcco-5/5 structures, respectively. Such a value of R_{\square} for MIT is in good agreement with the 2D quantum resistance value $h/e^2 = 25.8$ k Ω .

Further proof of the validity of such an approach is given by the analysis of the R_{\square} for the Cbcco-5/2/5 trilayered structure, investigated both by transport measurements [29] and by scanning squid microscopy [36]. In that case, the effective thickness t is very different if one chooses to

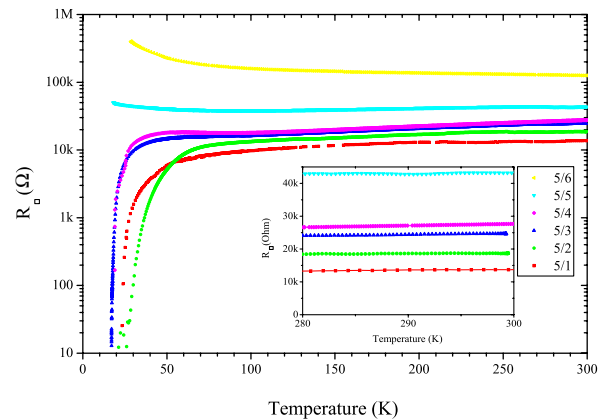


FIG. 3 (color online). R_{\square} vs temperature behavior, along the $(\text{Ba}_{0.9}\text{Nd}_{0.1}\text{CuO}_{x+2})_5/(\text{CaCuO}_2)_N$ series (with N varying from 1 to 6). In the inset, the R_{\square} behavior is magnified in proximity of the room temperature.

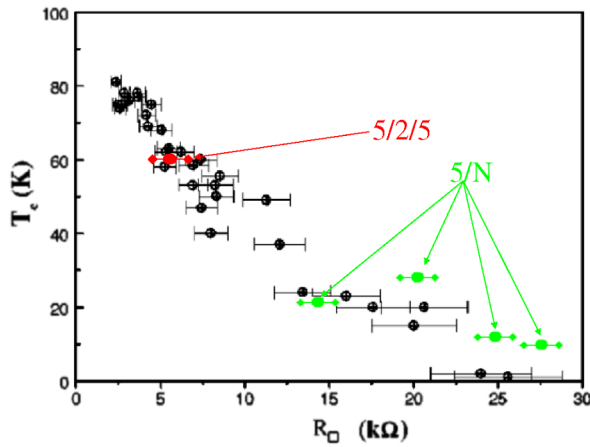


FIG. 4 (color online). Critical temperature T_c versus the sheet resistance R_{\square} for ultrathick Cbcco- $2 \times n$ superlattices [25] (black dots), and for ultrathin Cbcco- $5/N$ [28] (green dots), and Cbcco- $5/2/5$ [29] (red dot) structures.

consider the IL-layer thickness alone (6.4 Å) or the whole structure thickness (48.9 Å). In agreement with the assumptions made for the Cbcco- $5/N$ structures as well as for the trilayered ultrathin structures, we consider just the $(\text{CaCuO}_2)_2$ layer as the truly superconducting block. Based on such an assumption, the R_{\square} value for the Cbcco- $5/2/5$ sample that shows a critical temperature $T_c = 60$ K [29] is found to be about 5.4 kΩ. Such a value is in very good agreement with the general T_c vs R_{\square} behavior found for the bilayered Cbcco- $5/N$ ultrathin structures and the Cbcco- $2 \times n$ ultrathick superlattices (all plotted in Fig. 4).

In conclusion, the metal-insulator transition has been investigated in terms of room temperature 2D R_{\square} values. Because of the complex structural features of HTS materials, the R_{\square} value strongly depends on the thickness chosen for the conducting layer, giving rise to an ambiguity in evaluating the sheet resistance. We have shown that it is indeed possible to directly measure the 2D R_{\square} sheet resistance in the bilayered Cbcco- M/N and trilayered Cbcco- $M/N/M$ ultrathin (overall thickness smaller than 50 Å) heteroepitaxial structures. In agreement with a previous study [25], the metal-insulator transition occurs for a critical value of R_{\square} close to the 2D quantum resistance value h/e^2 , strongly supporting the fermionic explanation for the MIT in the HTS cuprate system. Moreover all data confirm the assumption that each CuO_2 layer forms a 2D superconducting sheet within the superconducting block, which can be described as weak-coupled equivalent CuO_2 sheets in parallel.

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