Chain Conformations and Photoluminescence of Poly(di-*n*-octylfluorene)

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The diverse steady-state spectroscopic properties of poly(di-*n*-octylfluorene) are addressed from a molecular-level perspective. Modeling of representative oligomers support the experimental observation of at least three distinguishable classes of conformational isomers with differing chain torsion angles. One class appears to be populated by a single compact structural isomer, and this conforms to the so-called β phase. A rigorous Franck-Condon analysis of the photoluminescence in conjunction with Frenkel-type exciton band structure calculations is performed. These results accurately reproduce all major spectral features of the photoabsorption and those of singlet exciton emission.

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Recent years have witnessed widespread activity in the field of polymer electronics. Integral to this progress is the development of realistic molecular-level models that can address fundamental issues of photophysics and charge transport. Π -conjugated organics, including small molecules, oligomers, and polymers, that tend to pack into locally well-ordered crystalline structures have received broad attention because of notable advances [1–3] in both theoretical chemistry and physics. These gains have not been matched by parallel progress in the burgeoning area of the functionalized polymers because of sample or processing derived heterogeneities and, more importantly, limited knowledge of the chain conformations. Thus many reports rely on qualitative underpinnings with few explicit references to the molecular-level construction [4–7].

Polyfluorenes (PFs) [8] exemplify the extreme sensitivity of conjugated polymer photophyics to minor changes in molecular architecture and film-forming conditions. Poly(di-*n*-octylfluorene), PF8 as sketched in Fig. 1, is known for polymorphic phase behavior [9,10] and the striking presence of a low-energy emitting " β -type" chromophore. This often cited "phase" always appears as a minority constituent in the photoabsorption (PA) but dominates the optical emission [9,10]. Its photoluminescence (PL), by polymer standards, is also highly irregular in that these features become remarkably sharp at reduced temperatures [7,11,12]. Cadby *et al.* [9] have proposed that this emission originates in regions of enhanced chain planarity. Unusual thermal properties and additional emitting chromophores appear as well [12–14].

This Letter advances a near molecular-level framework for explaining the diverse steady-state optical properties of singlet excitons in PF8. A simple empirical force-field modeling of representative oligomers yields a suggestive trifurcation of the low-energy chain structures to give three nominally distinct classes or families of conformational isomers (referred to as C_{α} , C_{β} , and C_{γ}) with 10°–20° steps in the average fluorene-fluorene torsion angles ($|\phi|$). The C_{β} -type "family" is comprised of just a single isomer. The inferred nature of these families is then used to frame Frenkel-type tight-binding (TB) band structure calculations of the single chain PA and PL spectra. Calculated optical spectra, including a full Franck-Condon (FC) vibronic progression, manifest many key attributes seen in the data. As a general feature PF8 β -type (C_{β}) emission incorporates a power law line shape and not the composite Gaussian/Lorentzian typically assumed [7,15]. This approach also provides a clear microscopic starting point for more quantitative analysis.

All of the PF8 polymers in this study were obtained from American Dye Source and used as received. Various solutions were prepared, typically 1% w/w in solvent, and the specific formulations (and some optical spectra) are already published [12,14]. Thin-film samples were cast onto sapphire substrates. All PA and PL spectra were recorded from the same physical location under steadystate conditions using a dual PA-PL spectrometer [12]. Thus all PL spectra are corrected for self-absorption and, additionally, normalized for detector response.



FIG. 1. Representative PF8 PA and PL spectra from the three claimed conformational isomer families in combination with calculated PL fits to the data using Eq. (1). Inset: sketch of PF8.

The three sets of PL-PA spectra in Fig. 1 demonstrate the wide range of PF8's optical properties. Each PL spectrum has its emission dominated by one of the three claimed conformational isomer families. The most intense "0-0" peak is a superposition of the $\pi - \pi^*$ transition and, on the low-energy tailing edge, vibronic overtones from lowenergy torsional modes [16]. Additional peaks arise from other FC vibronic subbands. The top spectra are representative of PF8 films [14] spin cast from p-xylene and contain a coexistence of C_{α} and C_{β} isomers. Because of large scale exciton energy migration the lowest energy chromophores dominate the PL (here C_{β} isomers). The middle spectra are from a C_{γ} dominated sample and, in this case, the $\pi - \pi^*$ transition energies are intermediate to those of the proposed C_{α} and C_{β} families. This occurs in samples cycled above \sim 430 K [14]. The bottommost spectra, obtained by PF8 casting from a toluene/tetrahydrofuran mixture [12], is dominated by emission from C_{α} -type isomers. This nematic glass sample also includes a broad low-energy PL background due to interchain excitations [17]. Formation of the C_{β} conformer has been suppressed by quenching to -30 °C from a thermotropic liquid crystal state. All PL spectra are notably sharper than the respective PA (again due to energy migration), but the distinct narrowing of the C_{β} emission is quite evident.

Also included in Fig. 1 are nonlinear least-square fits to the PL data using a slightly modified FC formula:

$$PL(\hbar\omega) \propto [n(\omega)\hbar\omega]^3 \sum_{n_1=0}^{\infty} \cdots \sum_{n_p=0}^{\infty} \prod_{k=1}^{p} (1+c) \\ \times \left[\frac{e^{-S_k} S_k^{n_k}}{n_k!} \right] \Gamma \left\{ \delta \left[\hbar\omega - \left(\hbar\omega_0 - \sum_{k=1}^{p} n_k \hbar\omega_k \right) \right] \right\}$$
(1)

where $n(\omega)$ is the index of refraction (with data taken from Ref. [7]), $\hbar\omega_0$ is the $\pi - \pi^*$ transition energy, and $\hbar\omega_k$ are vibrational mode energies for each mode k with $n_k =$ 0, 1, 2, ... vibronic overtones. S_k is the conventional Huang-Rhys (HR) coefficient but c = 0.2(p-1) (for $p \ge 1$ 1) is an *ad hoc* term that increases the relative contribution by higher order modes. Because the 18 K C_{β} -type PL is exceptionally well defined nine separate base ω_k 's were necessary (and extending up to seven overtones). When possible these frequencies were derived from Raman scattering data [18]. The line shape operator, Γ , is actually a composite function with, for example, in the 0-0 case, $(\hbar\omega - \hbar\omega_0)^{3/2} \exp[(\hbar\omega_0 - \hbar\omega)]/k_b T$ when $(\omega > \omega_0)$ and a small Urbach tail (for $\omega < \omega_0$) to account for states extending into the band gap. The first of these two Γ factors is essential for quantitative agreement along the high energy side of the 0-0 PL band in C_{β} -type PL spectra (both at 18 K and higher temperatures). Γ also includes convolution with a narrow Gaussian representing instrumental resolution and residual disorder. The only essentials needed in the following text are the HR parameters, the ω_k 's (for use after the TB calculations), and the displacement parameter, $D_{\text{rel}} = \sum_{i=1}^{9} \hbar \omega_i S_i$. D_{rel} reflects the relative difference between the ground and excited state coordinates.

For structural modeling we used a combinatorial method approach in which a large number of possible single chain conformers were examined using empirically determined molecular mechanics (MM3*) [19] parameters at zero temperature in the gas phase. Recent work in a related PF [20] suggests that this approach, though limited in scope, reveals some basic attributes of the local intrachain structure. In this case over 700 distinct starting PF8 decamers were constructed. The fluorene backbone consisted of a repeating dyad (variants of a 2/1 helix as suggested in Ref. [10]) in which the starting torsion angles were alternately set to $\pm 160^{\circ}$, respectively. The three core alkyl side chain carbon-carbon dihedral angles (see Fig. 1) were then assigned to all possible combinations of the three conventional alkyl low-energy conformers (i.e., anti, gauche, and gauche') within each alkyl chain. Every fluorene monomer received the identical side chain conformer (see Ref. [20] for details). More extensive modeling, to identify viable dyad and larger conformational repeats, and simulations are needed to fully validate this approach. Figure 2 displays the lowest zero-temperature energies of the trial starting conformers in terms of the average backbone torsion angle $|\phi|$. At torsion angles above 120°, one can identify three distinct classes of low-energy chain structures near $|\phi| = 135^\circ$, 150°, and 160° (or C_{α} , C_{γ} , and C_{β} , respectively). Only a single unique isomer represents the C_{β} -type conformer. The Fig. 2 inset depicts the resulting compact molecular construction. The alkyl chains adopt a *trasoid* (i.e., nearly anti), gauche, gauche arrangement of the three innermost carbon-carbon torsion angles (τ_1 , τ_2 , and τ_3 in the Fig. 1 inset). van der Waals interactions between neighboring alkyl side chains and the fluorene backbone work to stabilize this structure. This conformer is not stable if the alkyl chain length is two CH2 units smaller and a similar analysis



FIG. 2. Low-energy PF8 conformers (referenced to the C_{β} isomer). Each symbol corresponds to a single, unique tested conformer. C_{α} , C_{β} , and C_{γ} correspond to the proposed conformer families. Inset: model of C_{β} isomer highlighting side chains.

TABLE 1. Equation (1) parameters from his to PL data in Fig. 1.												
Sample	$\hbar\omega_0 \; (\mathrm{eV})$	$D_{\rm rel}~({\rm eV})$	$\hbar \omega_{i,i=1 \text{ to } 9}$ (eV):	0.0075	~ 0.07	~0.09	~0.11	0.140	0.159	0.167	0.176	0.199
C_{β}	2.806	0.12	HR_i :	0.75	0.02	0.02	0.02	0.11	0.17	0.04	0.05	0.30
C_{γ}	2.839	0.11	HR_i :	0.98	0.03	0.03	0.03	0.06	0.13	0.08	0.05	0.19
C'_{α}	2.922	0.21	HR_i :	2.17	0.03	0.03	0.03	0.14	0.21	0.17	0.05	0.46

of poly(di-2-ethyl-*n*-hexylfluorene) [20] gives only monomodal distributions of helices. We therefore hypothesize the presence of three low-energy classes of chain structures with differing average chain conformations and that C_{β} segments are well ordered.

This suggestive finding is further tested by employing a conventional TB Hamiltonian [21–23] $H = \sum E_n |n\rangle \langle n| +$ $\sum J_{n+1,n} \cos(\phi_{n+1,n}) | n+1 \rangle \langle n |$, where E_n and $J_{n+1,n}$ are on-site excitation energy and nearest-neighbor intersite transfer energies, respectively. In this analysis we assume *no* on-site disorder and both coefficients $(E_n \text{ and } J_{n+1,n})$ are given a single unique value. The second term has been modified by a cosine function to mimic the nature of π conjugation [24] as the backbone torsion angle $\phi_{n+1,n}$ varies. After diagonalization of the matrix the density of states and oscillator strengths were calculated in standard fashion to give an absorption line shape $A(\omega) =$ $\langle \Sigma_j \mu_j^2 \delta(\omega - \omega_j) \rangle$, where μ_j^2 is the transition dipole moment of the *j*th eigenstate and averaged over the entire ensemble. This function was further convoluted with the appropriate FC expression for absorption [16] using HR parameters identified with the conformer dominating the respective PL (as in Table I) and a narrow Gaussian representing the spectrometer resolution plus residual disorder.

Calculations of the PL employed $A(\omega)$ directly. To model emission from excitons in thermodynamic equilibrium at the bottom of the band $A(\omega)$ was multiplied by a Boltzmann factor, $\exp(\hbar\omega_0 - \hbar\omega)/k_bT(\omega \ge \omega_0)$, next convoluted with the composite resolution-residual disorder function and then expressed using a conventional FC formula. Matching the exact energies required a small Stokes shift defined as Δ . Clearly this simple approach cannot address the myriad of processes present in real PFs. There is no accounting for either time dependence or energy migration. In addition there is nothing that reflects an exciton binding energy or interchain excitations. Only singlet exciton emission is modeled.

In all these calculations each phenylene ring (i.e., 1/2 of a fluorene unit) was deemed the basic structural unit so that every other off-site term was set to $-J_{n+1,n}$ because of the chemically imposed planarization of the fluorene unit. A single chain was truncated to 100 monomer units (or 200 sites) and this qualitatively reflects the experimental length of the PF8. To represent conformational isomerism we specified a different average fluorene-fluorene torsion angle, $|\phi|$, for each family and to this we typically added a Gaussian distribution with a standard deviation σ . In most cases only two of these three claimed families were chosen by specifying x and 1 - x relative phase fractions assuming the minority phase segment length adopted Poisson statistics. In this way features of both segmental and worm-like disorder [22] are included. Over 4000 single chain trials were averaged for adequate statistics. $A(\omega)$ was sensitive to even small parameter changes.

Figure 3 displays direct comparisons of both PA and PL in four representative data sets (three are from Fig. 1) with those calculated. Table II lists the associated parameters. These parameters best approximate the leading edge of the PA, the region that is most critical for simultaneously assessing both PA and PL. In particular, the anomalous line shape of the C_{β} 0-0 PL peak is accurately reflected in this calculation. The leading edge of the PA exhibits a near power law rise and it is this feature that gives the atypical line shape on the high energy side of the 0-0 PL peak and also, in regards to peak position and width, the systematic progression in Ref. [12]. Replacing the Poisson distribution of the minority isomer segmental length with a Gaussian worsens these fits. This attribute is washed out in the C_{α} and C_{γ} dominated PL spectra because σ and the residual disorder (see the text on Γ) are much larger.

An important factor for estimating the relative phase fraction is the energy dependence of the oscillator strength. Spectra similar to the top PA in Fig. 1 are assigned up to a 25% C_{β} concentration based on lifetime measurements [7]. If the actual TB results are directly employed, then this fraction drops to under 10%. This phase, as suggested by a kinetic analysis [12], appears to be isolated one-dimensional threads of enhanced structural order em-



FIG. 3. PA and PL data in comparison with curves obtained from the tight-binding model calculations (see the text).

TABLE II. Tight-binding model parameters.

Т	Minority conformer					C_{α} -type region			
(K)	Туре	Conc.	$\langle n \rangle$	$ \phi $	σ	$\langle n_{\alpha} \rangle$	$ \phi_{\alpha} $	σ_{lpha}	(meV)
18	C_{β}	9.5%	5.0	165°	3°	32	136.5°	13°	10
210	C_{β}	9.3	5.1	165	6	33	135	15	24
110	C_{β}	0.7	3.0	172	3	55	129	31 ^a	9
110	$C_{\gamma}^{\prime b}$	9.1	6.2	155	15	29	127	31 ^a	11

^aAssumes a uniform distribution of torsion angles.

^bAdditionally incorporates 0.5% of C_{β} conformers.

bedded in a sea of more disordered conformers. Moreover, these TB calculations obtain relatively short segmental lengths (specified as $\langle n \rangle$ in Table II) for the emitting chromophore unit and, even in the locally well-ordered C_{β} isomer, averages only five monomers.

All calculated $|\phi|$ and σ parameters adopt physically realistic values and, in regards to $|\phi|$, match the MM3^{*} modeling values. Matching the temperature dependence of the C_{β} dominated sample required few parameter changes except for σ (which doubles at 210 K). Increasing σ , because of the oscillator strength weighting effects, enhances the leading edge of the PA. The Boltzmann weighting factor provides for an immediate fit of the PL data although Δ , a Stokes shift, goes from 10 to 24 meV. This may reflect the nature of exciton energy migration. The TB results yield E_n and $J_{n+1,n}$ values of 5.45 and 1.35 eV. These give a $\pi - \pi^*$ threshold energy of 2.75 eV and a bandwidth of 2.70 eV. This threshold approximates those of the planarized ladder-type polymers [25].

There are additional factors that need to be emphasized. Mimicking the slow rise on the leading edge of the PA in samples dominated by C_{α} - or C_{γ} -type emission required very small but finite fractions of C_{β} . Modeling of the broad PA at higher energy in these two cases is imperfect. The quenched (C_{α} -type) sample includes a uniform distribution of torsion angles between 98° and 160°. More complex distributions are necessary to address the failure at high energies. Other shortcomings are present. All FC progressions for the PAs assume fixed HR coefficients regardless of $\hbar\omega$. The implicit heterogeneity of these PF8 models should produce systematic variations in these coefficients. Moreover, the PA HR coefficients are unlikely to be identical to those for PL.

We also resolve an order dependent systematic failure of the FC model. As the number of vibronic overtones rises the PL fit becomes progressively worse. This even occurs on the low-energy side of the 0-0 peak. This effect may, in part, be due to the limited number of the ω_k 's used. However the 0-4 FC vibronic subband structure (near 2.1 eV) can still be resolved experimentally (see Fig. 1 [26]), and this is many times that given by Eq. (1). Given the simplistic assumptions used to specify the HR parameters this failing is not surprising [27].

One unexpected result relates to the D_{rel} values listed in Table I. Because the equilibrium excited state geometry

approaches planarity [28] we naively expect emission from the most planar ground state conformations (i.e., C_{β} -type) to yield the smallest displacement parameter. We actually find that the smallest D_{rel} occurs when the PF8 PL is dominated by the claimed C_{γ} -type family. Quantum chemical calculations are still needed to further verify this result.

As a final comment we note that this discrete three family model is overly abrupt and that one may expect more gradual crossovers in terms of local structure. The modeling procedure itself has merit for identifying prospective chain conformations including the side chains and, with additional effort, conformational disorder as well. Successfully deciphering the single chain properties is one very important step for developing a full microscopic understanding of structure and photophysics.

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