## Does the Interaction Potential Determine Both the Fragility of a Liquid and the Vibrational Properties of Its Glassy State?

Patrice Bordat, Frédéric Affouard, and Marc Descamps\*

Laboratoire de Dynamique et Structure des Matériaux Moléculaires, UMR 8024, Université Lille I, 59655 Villeneuve d'Ascq cedex, France

K. L. Ngai

Naval Research Laboratory, Washington, D.C. 20375-5320, USA (Received 8 January 2004; published 2 September 2004)

By performing molecular dynamics simulations of binary Lennard-Jones systems with three different potentials, we show that the increase of anharmonicity and capacity for intermolecular coupling of the potential is the cause of (i) the increase of kinetic fragility and nonexponentiality in the liquid state, and (ii) the  $T_g$ -scaled temperature dependence of the nonergodicity parameter determined by the vibrations at low temperatures in the glassy state. Naturally, these parameters correlate with each other, as observed experimentally by T. Scopigno *et al.* [Science **302**, 849 (2003)].

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The structural relaxation time,  $\tau$ , of all glass-forming liquids increases on cooling. It becomes so long at some temperature  $T_g$  that equilibrium cannot be maintained and the liquid is transformed into a glass.  $T_g$  is defined as the temperature at which  $\tau$  reaches some arbitrarily chosen long time, say,  $10^2$  s. Although this behavior is shared by glass formers of diverse chemical and physical structures, the scaled temperature dependence of  $\tau$  in the liquid state can differ greatly from one liquid to another in the degree of departure from the Arrhenius scaled temperature dependence [1,2]. The departure can be characterized by the rapidity of the change of  $\log(\tau)$  with  $T_g/T$  at  $T_g/T = 1$ , which is given by the steepness index or the fragility *m* defined by [3]

$$m = \frac{d\log(\tau)}{d(T_g/T)} \Big|_{T_g/T=1}.$$
 (1)

The values of *m* of glass formers of all kinds vary over a large range, from the least value of about 17 for strong glass formers (such as silica) having nearly Arrhenius scaled temperature dependence of  $\tau$ , to values as high as about 200 found for some glass formers called fragile. Naturally, such large variations observed in *m* beg the question of its microscopic origin. Several attempts have been made in the past to correlate *m* with other dynamic or thermodynamic properties, with the hope that the correlations will lead to the factor or factors that determine m. Examples include (i) the correlation of m with  $n \equiv (1 - \beta)$  at  $T = T_g$ , where  $\beta$  is the stretched exponent in the Kohlrausch function,  $\exp[-(t/\tau)^{\beta}]$ , used to fit the time dependence of the correlation functions such as the intermediate scattering functions; (ii) the correlation of m or  $(1 - \beta)$  with the mean-square displacement  $\langle u^2 \rangle$  obtained [4] from quasielastic neutron scattering measurement of the Debye-Waller factor  $\exp[-\langle u^2 \rangle Q^2/3]$  [glass former with larger m or  $(1 - \beta)$  has a larger  $\langle u^2 \rangle$  at the PACS numbers: 61.20.Ja, 64.60.Ht, 64.70.Pf

same value of  $T/T_g$  and rises more rapidly as a function of  $T/T_g$ , below  $T_g$  as well as near and across  $T_g$  in the liquid states [4]]; (iii) the correlation between m and the slope of the change of the configurational entropy,  $S_c$ , with  $T/T_g$ at  $T_g$  [5]; (iv) the correlation of *m* with the statistics of potential energy minima of the energy landscape [6,7]; and (v) the correlation of m with the temperature dependence of the shear modulus of the liquid [8]. Perhaps the most intriguing of all correlations is (vi) between m and the vibrational properties of the glass at temperatures well below  $T_g$  found recently by Scopigno *et al.* [9]. The nonergodicity parameter, f(Q, T) at  $T \ll T_g$ , is determined by vibrations. From inelastic x-ray scattering data, the temperature dependence of  $f(Q \rightarrow 0, T)$  is well described by  $[1 + \alpha (T/T_g)]^{-1}$ . Scopigno showed that m and  $\alpha$  are proportional for many glass formers. Apparently, this last correlation (vi) seems to be related to (ii) for  $\langle u^2(T/T_g) \rangle$ from neutron scattering at temperatures well below  $T_{g}$ .

In any glass former, it is the interaction potential, V(r), that determines ultimately all dynamic, thermodynamic, and vibrational properties at all temperatures both below and above  $T_g$ . Changes in any of the quantities, m, n,  $\langle u^2(T/T_{\rho})\rangle$ ,  $\alpha$ ,  $S_c$ , free volume  $\nu$  [10], and the degree of dynamic heterogeneity, from one glass former to another, originate from the change in V(r). Thus, correlations found between these quantities are clues for finding out which aspects of V(r) determine them and give rise to the correlations between them. One would like to examine the interaction potentials in real glass formers. However, in such materials, the different kinds of chemical bonding and the different sizes of the basic structural unit make the comparisons ambiguous. For this reason we consider the binary Lennard-Jones (LJ) particles with different choices of interaction potentials V(r) between the particles, and perform molecular dynamics (MD) simulations on them to obtain m, n,  $\alpha$ , and  $\langle u^2(T/T_g) \rangle$ . Correlations are found between all these quantities, thus reproducing the empirical findings from real glass formers. Since the number of particles as well as their density ( $\rho = 1.2$ ) are the same, the changes of these quantities are predominantly due to the change in V(r). The latter is well controlled, and therefore we identify anharmonicity and the capacity of intermolecular coupling of V(r) to be responsible for enhancement of m, n,  $\alpha$ , and  $\langle u^2(T/T_g) \rangle$ , and hence their correlations.

MD simulations were performed on binary LJ particles systems with three different interaction potentials by the MD package DLPOLY [11]. Technical details of the MD simulations are given in [12]. We have performed from  $10^5$  to  $6 \times 10^7$  time steps, depending on temperature. We have investigated 12 temperatures in the range of [0.675-5], [0.416-5], and [0.26-2] for models I, II, and III, respectively. All models are composed of 1500 uncharged particles (1200 species A and 300 species B). The generalized (q, p) LJ potentials have the form  $V(r) = \frac{E_0}{(q-p)} \times [p(\frac{r_0}{r})^q - q(\frac{r_0}{r})^p]$ . The parameters  $r_0$  and  $E_0$  represent the position of the minimum of the well and its depth, respectively. The reduced LJ units [13] are used. The choice of q = 12 and p = 6 corresponds to the standard LJ potential used by Kob and Andersen (KA) [14] and by others for extensive studies by simulation. For the purpose of investigating the change of dynamics with controlled change of V(r), we developed two other models by changing only the exponents, q and p, of the LJ potential for the A-A interactions. They are (q = 8, p = 5) and (q =12, p = 11) and are shown together with the (12,6) LJ potential in Fig. 1. The well depth and the position of the minimum of V(r) are unchanged, and we have kept the standard (12,6) LJ potentials of the KA model for the A-Band B-B interactions in order to retain as much as possible the remarkable ability of the KA model to form a glass upon cooling. The (12,11) LJ potential is more harmonic than the classical (12,6) LJ potential, while the (8,5) LJ



potential is a flat well and exceedingly anharmonic. Whenceforth the (12,11), (12,6), and (8,5) potentials are referred to as models I, II, and III, respectively, reminding us that anharmonicity is increasing in this order. The V(r) potentials are not truncated and shifted according to [14]. This modification does not alter the behavior of the models.

The structure has been briefly studied by the radial A-A distribution functions g(r) of the (12,11) and (8,5) models, which have been calculated and found very similar to that of the (12,6) model in the temperature ranges investigated (see inset of Fig. 1). Respectively for models I, II, and III at T = 0.700, 0.440, and 0.268, the position of the first peak of g(r) is at 1.072, 1.066, and 1.057 and its width at half the maximum is 0.129, 0.147, and 0.191. The g(r) of the three models in the supercooled state is similar to that in the liquid regime, ensuring that the structures have disorder in the middle and long range.

Dynamics have been investigated by computing the self-  $F_S(Q, t)$  and the total- F(Q, t) intermediate scattering functions of particles A at  $Q_0 = 2\pi/r_0$ , close to the maximum of the collective static structure factor S(Q), for the three models. At high temperatures,  $F_{S}(Q_{0}, t)$ decays linear exponentially to zero with a characteristic time of about 0.45 close to crossover time  $t_c \approx 1$  to 2 ps used as a fundamental time in the coupling model [15,16]. When temperature is lowered, the dynamics slows down dramatically and a two-step process appears. This behavior is well described by the mode coupling theory [17]. For the three models, it is indeed possible to draw a master curve with  $F_S(Q_0, t)$  at each of these lower temperatures in the reduced unit  $t/\tau_A$ , where  $\tau_A$  is the relaxation time. From the master curves, we find an independent temperature  $\beta$ , the stretched exponent, equal to 0.84, 0.81, and 0.76 with error bars of 0.1 for models I, II, and III, respectively. Because of the relatively large error bars, we have determined the nonergodicity parameter (height of



FIG. 1. Potential V(r) governing the A-A interaction. The dashed curve is the (12,11) LJ potential for model I, the solid curve is the (12,6) LJ potential for model II, and the dotted curve is the (8,5) LJ potential for model III. Inset: radial distribution function g(r) of species A at  $T = 1.02T_{\text{ref}}$  for the three models (see below for the definition of  $T_{\text{ref}}$ ).

FIG. 2. Self-intermediate scattering function  $F_S(Q_0, t)$  vs scaled time  $t/\tau_A$ . Dashed, solid, and dotted lines are for models I, II, and III, respectively. For all three models,  $\tau_A(T_{\text{ref}}) = 46\,435.8$ . The inset shows the stretched exponent  $\beta = (1 - n)$  as a function of the scaled reciprocal temperature  $T_{\text{ref}}/T$  for the three models: ( $\blacklozenge$ ) model I, ( $\blacklozenge$ ) model II, and ( $\blacksquare$ ) model III. For definitions of  $\tau_A$  and  $\beta$ , see text.

the plateau),  $f_S(Q_0, T)$ , the relaxation time,  $\tau_A$ , and the stretched exponent,  $\beta$ , from the fit to the second step decay of individual  $F_S(Q_0, t)$  (rather than the master curves) by  $f_S(Q_0, T) \exp[-(t/\tau_A)^{\beta}]$ . Shown in Fig. 2 are  $F_S(Q_0, t)$  vs  $t/\tau_A$  of all three models at the reference temperature  $T_{\text{ref}}$  defined by  $\tau_A(T_{\text{ref}}) = 46\,435.8$ , a very long time in our simulations. The values of  $T_{\text{ref}}$  are 0.688, 0.431, and 0.263 for models I, II, and III, respectively.

Shown in Fig. 2 (inset) are  $\beta$  of the three models as a function of  $T_{\rm ref}/T$ . At any  $T_{\rm ref}/T$ ,  $(1 - \beta)$  is least for model I and largest for model III. At  $T = T_{ref}$ ,  $\beta =$ 0.69, 0.65, and 0.60, respectively, for models I, II, and III. Thus  $(1 - \beta)$ , which provides a measure of the intermolecular coupling according to [15], increases with anharmonicity. In Fig. 3,  $\log(\tau_A)$  is plotted against  $T_{\rm ref}/T$  and the data of the three models show systematic change. It can be seen that the slope, fragility index  $m(\tau_A) \equiv [d \log(\tau_A)]/[d(T_{\text{ref}}/T)]$  as  $(T_{\text{ref}}/T) \rightarrow 1$ , increases monotonically in the order of models I, II, and III. The values of  $m(\tau_A)$  determined from this latter relation are 15.07, 18.57, and 26.58 for models I, II, and III. Alternatively, the estimated values of  $m(\tau_A)$  based on the Sastry method [7] are 0.195, 0.241, and 0.405 for models I, II, and III, respectively. Hence,  $m(\tau_A)$  increases with anharmonicity. The fragility index found in the present study for model II is in good agreement with the value determined in an earlier work on the same model [7]. The diffusion coefficient,  $D_A$ , of particles A was calculated from the mean-square displacement  $\langle u^2(t) \rangle$  at long times when  $\langle u^2(t) \rangle$  assumes the linear t dependence. The evolution of  $1/D_A$  is comparable to  $\tau_A$ , leading to another definition of the steepness or fragility index,  $m(D_A) \equiv [d \log(1/D_A)]/[d(T_{ref}/T)]$  as  $(T_{ref}/T) \rightarrow$ 1. Again,  $m(D_A)$  increases monotonically in the order of models I, II, and III, or with anharmonicity.

So far we are concerned for dynamic quantities and their correlations for  $T > T_{ref}$ , as analogues of them in the



FIG. 3. The relaxation times  $\tau_A$  obtained from  $F_S(Q_0, t)$  for the three models as a function of  $T_{\text{ref}}/T$ , where  $T_{\text{ref}}$  is defined as the temperature at which  $\tau_A$  reaches 46 435.8: ( $\blacklozenge$ ) model I, ( $\blacklozenge$ ) model II, and ( $\blacksquare$ ) model III.  $T_{\text{ref}}$  is the analogue of  $T_g$  for simulations when the dynamics of the system slows down to more than 10 ns.

liquid state of real glass formers. Next we examine the vibrational properties of the three models at temperatures lower and much lower than  $T_{ref}$  by first quenching the liquid and then allowing vibrational degrees of freedom to relax. At low temperatures, relaxation of any kind is absent in the simulation time window, and the nonergodicity parameter f(Q, T) determined from F(Q, t) is contributed entirely from vibrations. As performed in [9], we have followed the behavior of  $f(Q \rightarrow 0, T) \equiv f_0(T)$ , which has been obtained by a Q-quadratic extrapolation of f(Q, T) for the lowest temperatures. The results of the three models are shown by a plot of  $f_0(T)^{-1}$  vs  $T/T_{ref}$  in Fig. 4.

In all three cases, the dependence of  $f_0(T)^{-1}$  on  $T/T_{\text{ref}}$  is approximately linear, and  $f_0(T)^{-1}$  has the extrapolated value of unity at the origin. The dependence of  $f_0(T)$  from simulation on  $T/T_{\text{ref}}$  is governed by the parameter,  $\alpha$ , through the expression

$$f_0(T)^{-1} = 1 + \alpha \frac{T}{T_{\rm ref}},$$
 (2)

just like a similar expression used to represent the dependence of  $f_0(T)$  on  $T/T_g$  of real glass formers obtained by inelastic x-ray scattering [9]. We see from Fig. 4 that the increase of  $f_0(T)^{-1}$  with  $T/T_{ref}$  is fastest for model III and slowest for model I. Equivalently stated, the slope  $\alpha$ is largest for model III and smallest for model I.  $\alpha$ increases with anharmonicity of the potential, such as  $m(\tau_A)$  or  $m(D_A)$ , and  $(1 - \beta)$ , as seen before (Figs. 1–3). Hence the interacting potential is the origin of the correlation of  $\alpha$  with  $m(\tau_A)$  or  $m(D_A)$ , and  $(1 - \beta)$  in our simulation, suggesting the same holds for real glass formers. Moreover, in the inset of Fig. 4, we observe that  $m(\tau_A)$  and  $\alpha$  are proportional together, as shown in [9].

Presumably there is no disagreement that the interaction potential is pivotal in determining all dynamic, thermodynamic, and vibrational properties of glass formers at all temperatures. With all these experimentally



FIG. 4.  $f(Q \to 0, T)^{-1} \equiv f_0(T)^{-1}$  vs  $T/T_{ref}$  for the three models:  $(\blacklozenge)$  model I,  $(\blacklozenge)$  model II, and  $(\blacksquare)$  model III.  $f_0(T)^{-1}$  is almost linear relative to  $T/T_{ref}$  with a slope noted  $\alpha$ . The inset shows the correlation of the fragility *m* with  $\alpha$  from the results of the three models.

accessible properties originating from the interaction potential, it is not surprising to find correlations or anticorrelations between them. Making this point is one of the motivations of the work in demonstrating that the various correlations between  $\alpha$ , m,  $(1 - \beta)$ , and  $\langle u^2(T/T_g) \rangle$  observed in real glass formers are reproduced by simulations as the analogues of correlations between  $\alpha$ ,  $m(\tau_A)$ ,  $m(D_A)$ , and  $(1 - \beta)$  by varying the interaction potential V(r). The results confer a bonus in identifying which feature of the interaction potential is responsible for enhancement of  $\alpha$ ,  $m(\tau_A)$ ,  $m(D_A)$ , and  $(1 - \beta)$ . Certainly the anharmonicity of V(r) increases when going from model I to models II and III, but one also can observe from Fig. 1 that the long range interaction becomes more pronounced. The latter trend means that neighboring LJ particles are more coupled in their motions. The increase in interparticle coupling from model I to model III is consistent with the position of the first peak of radial distribution functions g(r) (1.072, 1.066, and 1.057 for models I, II, and III, respectively) and with its width at half the maximum (0.129, 0.147, and 0.191 for models I, II, and III). This insight from simulation, when transferred to real glass formers, suggests that the capacity for intermolecular coupling and the anharmonicity of the interaction potential determine the dynamic, thermodynamic, and vibrational properties of glass formers above as well as below  $T_g$ . Although the thermodynamic variables, configurational entropy  $S_c$ , and free volume  $\nu$ are determined by V(r), they reenter into the dynamics by their influence on molecular mobility. Thus two factors govern dynamics, the capacity for intermolecular coupling directly from V(r), and  $S_c$  and  $\nu$  that come indirectly through V(r). On supercooling a liquid,  $S_c$  and  $\nu$ change, and since the kinetic fragility m is the slope of  $T_g$ -scaled temperature variation of  $\tau$ , it is not surprising that *m* is correlated with the slope of the corresponding change of  $S_c$ , i.e., the thermodynamic fragility. The capacity for intermolecular coupling of V(r) is solely responsible for the shape of the dispersion or the nonexponentiality parameter  $(1 - \beta)$ , and it also determines  $\tau$  in conjunction with  $S_c$  and  $\nu$ . The results of our simulation with the three potentials support this view. Increasing the density of the particles of the binary LJ system with the fixed (12,6) potential effectively forces the particles to be closer to each other and thereby increases intermolecular coupling. The simulation performed in this manner [7] showing that *m* increases with density can be reinterpreted as due to the increase of intermolecular coupling. Intermolecular coupling manifests itself in the dynamic properties in various ways, such as the  $Q^{-2/\beta}$  dependence of  $\tau$  [4]. All these provide evidence for intermolecular coupling that must be taken into consideration in conjunction with  $S_c$  and  $\nu$  for explaining all observed experimental facts in the liquid state. At low temperatures and deep in the glassy state,  $S_c$ 

and  $\nu$  having constant values cannot influence the temperature dependence of the vibrational properties characterized by  $\alpha$ . Hence,  $\alpha$  is controlled by the anharmonicity of V(r), as demonstrated by the simulations.

In summary, we demonstrate by using three different interparticle potentials of binary LJ systems that the capacity for intermolecular coupling and the anharmonicity of the potential are responsible for the correlations between various dynamic, thermodynamic, and vibrational properties of glass formers. The increase of the capacity for intermolecular coupling and anharmonicity has the effects of increasing the kinetic fragility, m, and the nonexponentiality parameter,  $(1 - \beta)$ , in the liquid state, and of increasing in the glassy state the parameter  $\alpha$  that characterizes the  $T_g$ -scaled temperature dependence of the nonergodicity parameter determined by vibrations at low temperatures. The correlations between m,  $(1 - \beta)$ ,  $\alpha$ , and other quantities follow as consequences, and their observations by experiments have now been explained.

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\*Electronic address: bordat@cyano.univ-lille1.fr

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