## Raman Spectroscopic Investigation of H<sub>2</sub>, HD, and D<sub>2</sub> Physisorption on Ropes of Single-Walled, Carbon Nanotubes

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We have observed the S- and Q-branch Raman spectra of  $H_2$ , HD, and  $D_2$  adsorbed at 85 K and pressures up to 8 atm on single-walled, carbon nanotubes (SWNT). Comparative data for  $H_2$  on graphite and  $C_{60}$  were also collected. Frequency-downshifted and upshifted features were observed in the Q-branch spectra of  $H_2$  on  $C_{60}$  and SWNT. These shifts are small and are therefore inconsistent with charge transfer. An  $H_2$ -surface potential with van der Waals and electrostatic terms was developed and used to estimate the shifts in the frequency of the Q(0) transition of  $H_2$  adsorbed in two types of sites. These calculations corroborate the experimental findings and indicate physisorption in multiple sites of the SWNT ropes.

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Single-walled, carbon nanotubes (SWNT) are nanoporous, all-surface macromolecules which are being widely investigated as storage media for H<sub>2</sub> [1,2]. Desorption experiments suggest the presence of a high-energy binding site for H<sub>2</sub> in purified and sonicated SWNT material, which has been proposed to indicate charge transfer [3]. On the other hand, transport data for SWNT suggest that H<sub>2</sub> physisorbs [4]. To probe the interaction on the molecular level, we have collected *S*- and *Q*-branch Raman spectra of H<sub>2</sub>, HD, and D<sub>2</sub> adsorbed on SWNT at 85 K and pressures up to 8 atm. The magnitude of the shifts in the *Q*-branch frequencies of the adsorption potential; for physisorption, the shifts should be small, while in the event of charge transfer, the shifts should be  $\sim 2000 \text{ cm}^{-1}$  per electron [5].

Using a two-step, oxidative, and acid reflux technique, we obtained SWNT samples containing minimal amorphous and multishell carbon (<15 wt%) and residual metal catalyst (<10 wt %, <2 at. %), as determined by temperature programmed oxidation (IGA-3, Hiden Inc.). This material was annealed in a vacuum of better than  $10^{-7}$  Torr at 1200 K for 24 h; such high-temperature "degassing" was recently shown to remove the carboxylic and related functional groups occluding the SWNT ends and side walls [6]. The resulting material was pressed against In foil and inserted in an optical pressure cell, which was placed in a cryostat (Janis Corp. Model VPF 700, 77-700 K) with a gas line to the cell. The cell was valved to a diffusion pump and to the cylinders of H<sub>2</sub> (ultrahigh purity, 99.999%, MG Industries), HD (97.4 mole%, with 1.4% H<sub>2</sub> and 1.2% D<sub>2</sub>, Isotec Inc.), and D<sub>2</sub> (ultrahigh purity, 99.999%, MG Industries). In separate experiments, samples of C<sub>60</sub> powder (Alfa Aesar, 99.9%, packed under Ar) and freshly cleaved, pyrolytic graphite (Alfa Aesar) were loaded into our cell.

Raman spectra were collected in the backscattering geometry, with  $\lambda_{air} = 514.53$  nm excitation provided by a mixed-gas, Ar/Kr-ion laser (Coherent Inc., Innova Spectrum). The incident beam was focused through the windows of the cryostat and pressure cell onto the sample surface. Backscattered Rayleigh light was rejected by a holographic "supernotch" filter (Kaiser, Inc.), and Raman light was focused through a 25  $\mu$ m slit into a single-grating monochromator (Instruments S.A. HR460, grating: 1800 grooves/mm) equipped with a CCD. To calibrate our spectra and correct for instrumental nonlinearity, one Hg and several pairs of Ne emission lines close to the *Q*-branch lines of H<sub>2</sub>, D<sub>2</sub>, and HD were used. The Ar, Hg, and Ne frequencies were taken from Refs. [7] and [8].

For all data we report, the sample temperature was 85  $\pm$  5 K, and the pressure of the adsorbate gas on the sample was increased from vacuum to as high as 8 atm, in the case of H<sub>2</sub> and D<sub>2</sub>, and to 5 atm in the case of HD. Typical *S*- and *Q*-branch spectra are shown in Figs. 1 and 2; we use the notation *S*(*J*) and *Q*(*J*) to denote transitions, in which *J* is the initial state. The frequencies,  $\nu_{ads}$ , of the observed *S*- and *Q*-branch transitions are given in Table I; these values were determined by fitting Voigt-profile functions to



FIG. 1. S-branch data for  $H_2$ , HD, and  $D_2$  on SWNT at 8, 5, and 8 atm, respectively. Asterisks indicate unrelated lines due to external calibration sources.



FIG. 2. *Q*-branch data for  $H_2$ , HD, and  $D_2$  on SWNT at 8, 5, and 8 atm, respectively. For comparisons of these peak positions with the frequencies of the free molecules, see Table I.

the spectra using Humlíček's algorithm [9]. Also provided in Table I are the frequencies,  $\nu_{\rm free}$ , of these transitions for free molecules, from Ref. [5].

Our S-branch data suggest freely rotating molecules in the gas phase, and are in good agreement with recent inelastic neutron scattering data [10]. Small, downshifted shoulders are evident in the S-branch lines, but our experimental resolution is insufficient to quantify these effects. The Q-branch data for HD and D<sub>2</sub> contain prominent bands which are slightly upshifted from the free-molecule frequencies (see Table I); due to the small splitting of the J = 0 and J = 1 bands, the peak substructure could not be resolved. Data for H<sub>2</sub> in SWNT are considerably more illu-

TABLE I. Frequencies  $\nu_{ads}$  of the observed *S*- and *Q*-branch transitions of H<sub>2</sub>, HD, and D<sub>2</sub> adsorbed on SWNT at 8, 5, and 8 atm, respectively. Also indicated are the free-molecule frequencies ( $\nu_{free}$ ) from Ref. [5]; the difference is indicated by  $\Delta \nu$ .

Adsorbate species	Transition	$ \frac{\nu_{\rm free}}{({\rm cm}^{-1})} $	$rac{ u_{ m ads}}{( m cm^{-1})}$	$\Delta \nu$ (cm <sup>-1</sup> )
H <sub>2</sub>	Q(0)	4160.5	4162.3	+1.8
	~ ~ /		4161.1	+0.6
			4159.3	-1.2
	Q(1)	4154.4	4156.4	+2.0
			4155.2	+0.8
			4153.5	-0.9
	S(0)	354.4	354.8	+0.4
	S(1)	587.0	587.5	+0.5
HD	Q(0)	3629.8	3632.5	+2.7
	Q(1)	3625.8	3628.8	+0.6
	Q(2)	3620.5	3620.9	+0.4
	S(0)	267.1	267.7	+0.6
	S(1)	443.1	444.1	+1.0
D <sub>2</sub>	Q(0)	2993.5	2993.8	+0.3
	$\tilde{Q}(1)$	2991.4	2991.9	+0.5
	Q(2)	2987.2	2987.5	+0.3
	$\tilde{S}(0)$	179.1	178.5	-0.6
	<i>S</i> (1)	297.5	297.8	+0.3

minating, and are shown in expanded form in Fig. 3 at two pressures, along with comparative data for  $C_{60}$  and highly oriented pyrolytic graphite (HOPG) collected at 85 ± 5 K and 4 atm. Best fits (solid lines) to the data in Fig. 3 were computed using a sum of Voigt components (dashed lines).

The surface area of HOPG is very small, so the signal strength from adsorbed molecules should be negligible. Therefore, we assign the bands observed at 4161.3 and 4155.4 cm<sup>-1</sup> to gas-phase H<sub>2</sub> near the surface; these frequencies are in reasonable agreement with the those reported in Ref. [11]. The gas-phase components should also be present in the data on  $C_{60}$  and SWNT, and so our analysis of these data proceeded by inserting components of frequency and width equal to those observed on HOPG, and then introducing separate Voigt components to describe the additional bands. Consistent with this approach, we noted an approximately twofold increase in the relative intensities of the bands attributed to gas-phase species, as the pressure was increased from 4 to 8 atm. In our analysis of the data on SWNT at 8 atm, the positions of the bands attributed to the gas phase were allowed to vary, and the frequencies were found to be  $\sim 0.1 \text{ cm}^{-1}$  lower, again consistent with Ref. [11]. In addition to the gas-phase components, the data for C<sub>60</sub> and SWNT contain downshifted and upshifted bands absent in the HOPG data. We ascribe these components to adsorption in the interstices, internal pores, or corrugated outer surfaces of the respective lattices. Neutron powder diffraction work by Fitzgerald et al., has indicated the existence of an adsorbed phase in the octahedral sites of fcc- $C_{60}$  [12]; presumably, these sites are partially filled under the conditions of our study and account for the lower-frequency component present in our *Q*-branch data for  $H_2$  on  $C_{60}$ . The number of fitting parameters was reduced by setting equal the linewidth parameters of equally spaced pairs of Q(0) and Q(1) peaks. All parameters were then refined until  $\chi^2$  was minimized and the frequencies were obtained. Our confidence in the



FIG. 3. *Q*-branch data for  $H_2$  on SWNT at 8 and 4 atm, together with data for  $C_{60}$  and HOPG at 4 atm.

number of Voigt components introduced for each data set is affirmed by the relative intensities between "pairs" of Voigt components:  $I_{Q(1)}/I_{Q(0)} = 2.2-2.5$ , similar to the equilibrium orthopara ratio at 85 K. Also, the Q(1)-Q(0) spacing we find between these pairs (5.8–6.0 cm<sup>-1</sup>) is consistent with the value (~5.97 cm<sup>-1</sup>) reported in Ref. [11].

Based on shifts observed in the tangential Raman bands of SWNT material previously dosed with  $H_2$ , partial electron transfer from SWNT to the  $H_2$  adsorbate has been proposed [3]. Our experiment is uniquely suited to looking for such effects, given the extreme sensitivity of the  $H_2$ stretching frequency on the molecular charge state. Contrary to reports on certain zeolites and oxides [13], no strongly shifted *Q*-branch lines were observed on SWNT or C<sub>60</sub>. This would seem to rule out charge transfer under our experimental conditions.

To predict the magnitudes of the stretching-mode shifts which arise from purely physical adsorption of  $H_2$ , we have constructed an interaction potential and estimated the frequency shifts in two types of adsorption geometries in SWNT ropes. In one geometry, the surface is taken to be flat — representative of the outer and inner surfaces of individual SWNT. In the other, "confined" geometry, the  $H_2$ is positioned between three surfaces, which is the situation in the interstitial channels of a SWNT rope. In both cases, the component surfaces were taken to be flat, an approximation which should not greatly affect the result.

Our potential is a sum of C-H van der Waals interactions,  $U_{LJ}$ , and the electrostatic interaction,  $U_{el}$ , of the H<sub>2</sub> static multipole moments with the static screening charges induced on the graphene surface [14]. Because the lowfrequency dielectric response of graphene is metallic [15], the screening charges may be represented by full image charges displaced from the graphene sheet by a distance  $z_0$ . Using a result for  $U_{el}$  which depends on the angular orientation  $(\theta)$  of the molecule axis with respect to the surface normal, the electrostatic interaction may be expressed in terms of the H<sub>2</sub> quadrupole  $(\Theta)$  and hexadecapole  $(\Phi)$  moments [14]. The holding potential may then be written  $V(z_c, \cos\theta) = U_{el} + U_{LJ}(z_1) + U_{LJ}(z_2)$ , where the  $z_i$  are the perpendicular distances of the H atoms from the surface, and the  $U_{LJ}(z_i)$  are the corresponding, pairwise, van der Waals interactions, which are assumed to be of Lennard-Jones (12-6) form. For simplicity, we smear out the C atoms along the surface. In this approximation,

$$U_{LJ}(z) = 2\pi \vartheta \epsilon_{\text{C-H}} \sigma_{\text{C-H}}^2 \left[ \frac{2}{5} \left( \frac{\sigma_{\text{C-H}}}{z} \right)^{10} - \left( \frac{\sigma_{\text{C-H}}}{z} \right)^4 \right],$$
(1)

where  $\vartheta = 0.38 \text{ Å}^{-2}$  is the surface density of the C atoms. We adopted the values  $\epsilon_{\text{C-H}} = 2.26 \text{ meV}, \sigma_{\text{C-H}} = 2.76 \text{ Å}$  [16]. The most uncertain parameter in our potential is  $z_0$  [17]. To estimate this parameter, we computed the minimum of the C-H<sub>2</sub> potential at the equilibrium distance  $z_{\text{eq}}$ , where H<sub>2</sub> is preferentially oriented flat

against the surface. In the case of graphene, the isotropic C-H<sub>2</sub> holding potential has the same functional form as Eq. (1), with a minimum at  $z_{eq} = \sigma_{C-H_2}$ . Using  $\epsilon_{C-H_2} = 42.8$  K and  $\sigma_{C-H_2} = 2.97$  Å [18], the resulting well depth is 46.6 meV and  $z_0 = 1.36$  Å, consistent with previous *ab initio* and empirical estimates for this system [19].

For the study of  $H_2$  in the ICs, we adapted a potential developed for a single nanotube [20]. The dispersion part of this potential ignores many-body effects, considers the C atoms to be smeared on the surface, and is isotropic. The total potential for an IC was obtained by summing this potential over an assembly of three nanotubes and azimuthally averaging the result.

The electrostatic portion of the potential in the ICs contains two contributions, denoted  $U_1$  and  $U_2$ , which arise, respectively, from the interaction of the H<sub>2</sub> quadrupole moment with the local electrostatic field of the SWNT, and the interaction of the H<sub>2</sub> static multipole moments with the screening charges induced on the surface.  $U_1$  is given by the usual relation for the electrostatic energy of a quadrupole in a nonuniform field, and was calculated from first principles for "zigzag" SWNT with radius R = 6.9 Å [21]. To evaluate  $U_2$ , we included the  $H_2$  quadrupole moment (the first nonzero permanent moment of an axial quadrupole), and the hexadecapole moment. The problem is quite complicated, so we introduce three simplifications. Because of the small size of H<sub>2</sub>, we consider its multipoles to be "pointlike." Next, to simplify the geometry of the IC, we represent the SWNT walls with graphene planes. The locations of the image charges are determined by the boundary condition requiring that the potential be zero at each plane. Finally, we assumed that the graphene planes are perfectly conducting, so that the static and dynamic responses are represented by images. We adopted the same value for  $z_0$  and the same form for the interaction as in the previous case. To compute the vibrational frequency shift, the total holding potential U must be expressed as a function of the relative bondlength change,  $\xi = (r - r_e)/r_e$ , relative to the equilibrium value,  $r_e$ , and also of the configuration  $\tau$  of the molecule. Given a sufficiently small surface interaction, one can treat U and the anharmonic terms in the potential energy function of free H<sub>2</sub> as perturbations to the harmonic oscillator Hamiltonian [22]. Using first- and second-order perturbation theory, the change in the frequency of the fundamental  $(n = 0 \rightarrow n = 1)$  transition due to U is [22,23]:

$$\Delta \omega = \Delta \omega_{1 \leftarrow 0} = \frac{B_e}{\hbar \omega_e} \langle U'' - 3aU' \rangle_{\tau} + O\left(\frac{B_e}{\hbar \omega_e}\right)^2.$$
(2)

in which  $B_e$  is the equilibrium rotational constant, *a* is the anharmonicity, primes denote derivatives with respect to  $\xi$ , and the angled brackets denote the average over all  $\tau$ . The expressions for  $\Theta(\mathbf{r})$  and  $\Phi(\mathbf{r})$  were derived via a model in which the electron cloud is concentrated between the nuclei rather than about each individual nucleus. Assuming an H<sub>2</sub> bondlength of 0.71 Å, and placing the electrons symmetrically on the molecular axis, separated by 0.48 Å, the experimental value of the  $H_2$ quadrupole moment is reproduced. From this model, we find  $\Theta \sim r^2$  and  $\Phi \sim r^4$ , for small changes in  $\xi$  about  $r_e$ . Regarding the orientational dependence, U' and U'' are functions of the z displacement of molecular centerof-mass, and of the polar  $(\theta)$  and azimuthal  $(\phi)$  angles. U depends on  $z_c$ , so the translational and rotational parts of  $U'(z_c; \theta, \phi)$ ,  $U''(z_c; \theta, \phi)$  do not decouple exactly. To separate the variables, we treat the interaction of translational and rotational modes in a mean-field manner [24]. The effective interaction was computed by averaging the derivatives of Eq. (2) over the center-of-mass vibrations of the molecule. Treating these as modes of a simple harmonic oscillator, the rms deviations in the z position are found to be  $\delta z_{\rm rms} = 0.298$  Å on a single graphene surface, and  $\delta z_{\rm rms} = 0.15$  Å for H<sub>2</sub> in an IC. Next, to compute  $\langle U' \rangle_{\tau}$  and  $\langle U'' \rangle_{\tau}$ , variations were performed with respect to r in the adsorption potential for different  $\tau$ . Finally, we averaged over all orientations  $(\theta, \phi)$  of the molecular axis.

Using the values a = -1.6 and  $B_e/\hbar\omega_e = 0.0138$ [23], we find that for the case of H<sub>2</sub> adsorbed on graphene, our procedure predicts an upshift of  $\Delta\omega_{1\leftarrow0} = 1.4 \text{ cm}^{-1}$  $\langle\langle 3aU'\rangle = -126.2 \text{ cm}^{-1}, \langle U''\rangle = -28.3 \text{ cm}^{-1}$ ). In contrast, for H<sub>2</sub> adsorbed in an IC, we predict a *downshift* of  $\Delta\omega_{1\leftarrow0} = -2.9 \text{ cm}^{-1}$  ( $\langle 3aU'\rangle = 168.0 \text{ cm}^{-1}, \langle U''\rangle =$  $-42.2 \text{ cm}^{-1}$ ). We expect similar results for D<sub>2</sub>: the same potential parameters and quadrupole moment may be assumed, though *a* and  $B_e/\hbar\omega_e$  are smaller and should produce smaller shifts. In the case of HD, the center of mass and charge do not coincide, leading to a larger Lennard-Jones contribution to the potential and a larger upshift.

In summary, we have observed several lines within the S- and Q-branch Raman spectra of H<sub>2</sub>, HD, and D<sub>2</sub> adsorbed on SWNT; the observed frequency shifts are small and consistent with physisorption. Both down- and upshifted features are evident in the Q branch, an effect we ascribe to H<sub>2</sub> adsorbed in two different types of sites in a SWNT rope. An interaction potential was developed to estimate the frequency shifts of the adsorbate stretching modes in two kinds of geometries similar to those found in SWNT ropes. In agreement with our experimental data, downshifted and upshifted modes are predicted, depending on the geometry of the adsorption site, and the shifts are small. Our calculations demonstrate that the experimental data are consistent with purely physical adsorption, without charge transfer.

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