

Evidence for Mean Free Path Fluctuation Induced Pinning in $\text{YBa}_2\text{Cu}_3\text{O}_7$ and $\text{YBa}_2\text{Cu}_4\text{O}_8$ Films

R. Griessen, Wen Hai-hu, A. J. J. van Dalen, B. Dam, J. Rector, and H. G. Schnack
Faculty of Physics and Astronomy, Free University, 1081 HV Amsterdam, The Netherlands

S. Libbrecht, E. Osquiguil, and Y. Bruynseraede
Laboratorium voor Vaste Stof-Fysika en Magnetisme, Katholieke Universiteit Leuven, 3001-Leuven, Belgium
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The critical current j_c and the pinning energy U_c have been determined for three types of yttrium-based superconducting films from current j_s and dynamic relaxation $Q = d \ln j_s / d \ln (dB/dt)$ data by means of the generalized inversion scheme. For $B < 2$ T and $T < 80$ K the temperature dependence of j_c and U_c for all films is found to be in excellent agreement with a model of single vortices pinned by randomly distributed weak pinning centers via spatial fluctuations of the charge carrier mean free path. Pinning due to spatial fluctuations of T_c is not observed.

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In type II superconductors pinning results from spatial variations of the Ginzburg-Landau (GL) coefficient α associated with disorder in the transition temperature T_c and/or from spatial variations in the charge carrier mean free path l near lattice defects [1-5]. These two sorts of pinning are usually called δT_c pinning and δl pinning [6]. In δT_c pinning spatial variations of T_c lead to spatial modulations of the linear and quadratic terms of the GL free energy functional $\alpha |\Psi|^2 + (\beta/2) |\Psi|^4$ while in δl pinning variations of the mean free path only affect the $|\nabla \Psi|^2$ term through variation of the two components m and M of the effective mass tensor.

For example, for a single vortex along the c axis of a superconductor containing randomly distributed weak pinning centers Blatter *et al.* [1] showed that all quantities characterizing the collective pinning of a single vortex can be expressed in terms of a disorder parameter δ , i.e.,

$$\begin{aligned} L_c &= \xi \delta^{-1/3} \varepsilon^{4/3}, \\ U_c &= \mu_0^{-1} B_c^2 \xi^3 \delta^{1/3} \varepsilon^{2/3}, \\ j_c &= j_0 \delta^{2/3} \varepsilon^{-2/3}, \end{aligned} \quad (1)$$

where ξ is the GL coherence length, B_c the thermodynamic critical field ($B_c = \Phi_0 / 2\sqrt{2}\pi\lambda\xi$ with $\Phi_0 = h/2e$ as flux quantum), $j_0 = 4B_c/3\sqrt{6}\mu_0\lambda$ the depairing current, j_c the true critical current, and ε is the mass anisotropy ratio $\varepsilon^2 = m/M < 1$. The collective pinning length L_c , the critical current j_c , and the collective pinning energy U_c are related via $U_c = j_c \Phi_0 L_c \xi$. In the absence of disorder (and, consequently, of pinning) the relations in Eq. (1) lead to $U_c = 0$, $j_c = 0$, and $L_c = \infty$. For δT_c pinning the disorder parameter designated by δ_a is proportional to ξ while for δl pinning the corresponding disorder parameter varies as $\delta_m \propto \xi^{-3}$ [1]. For both types of pinning one can, however, derive the following δ -independent relation:

$$U_c(T, B) = (B_c^2 \xi^3 \varepsilon / \mu_0 j_0^{1/2}) j_c^{1/2}(T, B) \quad (2)$$

for single vortex pinning [see Eq. (1)]. Similar, δ -independent relations for small and large vortex

(super)bundles can also be derived [7].

In their monumental review article Blatter *et al.* [1] conclude, however, that in high- T_c superconductors it has not been possible so far to prove experimentally which of the two pinning mechanisms (δT_c or δl pinning) is the dominant one. The purpose of this Letter is to demonstrate that for stoichiometric yttrium-based high- T_c superconducting thin films δl pinning is the only important pinning mechanism below ~ 80 K in moderate magnetic fields ($B < 2$ T). As explained below this conclusion is reached without using any explicit model for the disorder parameter δ . Moreover, the independence of the δl -pinning mechanism on the substrates and growth conditions of the films suggests that the source of δl pinning is related to randomly distributed oxygen vacancies [8] which are likely to be present in all studied films.

The samples investigated in this work are of three different types: (i) $\text{YBa}_2\text{Cu}_3\text{O}_7$ films of 180 nm thickness obtained by molecular-beam epitaxy (MBE) codeposition of pure Y and Cu from electron guns and BaF_2 from a Knudsen cell and postannealed in wet oxygen in order to remove the fluorine [9]. From x-ray diffraction patterns it is found that the FWHH of the (005) peak is 0.28° . These films have thus a good crystallinity with the c axis perpendicular to the film plane. The critical temperature is $T_c \cong 91$ K and the transition width $\Delta T_c = 0.5$ K. The substrates are made of SrTiO_3 as in (ii). (ii) $\text{YBa}_2\text{Cu}_4\text{O}_8$ films of 100 nm obtained with a deposition technique [10] similar to that in (i), with $T_c = 78$ K and $\Delta T_c = 0.5$ K. (iii) $\text{YBa}_2\text{Cu}_3\text{O}_7$ films of 450 nm thickness deposited onto $\text{MgO}(100)$ substrates using *in situ* 90° off-axis sputtering [11]. X-ray diffraction as well as Rutherford backscattering channeling measurements show that the films are grown epitaxially on the $\text{MgO}(100)$ substrates, with a channeling minimum yield smaller than 7%. The layers are purely c axis oriented and have narrow rocking curves [$\text{FWHH}(005) \leq 0.3^\circ$]. The as-prepared films have a critical temperature $T_c \cong 89$ K and $\Delta T_c < 2$ K.

The superconducting current j_s induced in the films when the external magnetic field is swept at a constant

rate dB/dt was determined by means of a sensitive capacitance torque magnetometer similar to that described in Refs. [12] and [13]. The torquemeter is mounted in the continuous flow cryostat of a 7 T Oxford Instrument cryomagnetic system. The temperature is stabilized to within 0.1 K. For all measurements the external field is applied at a finite angle close to the c -axis direction since the torque vanishes when B is exactly parallel to the c axis. (For the analysis it is, however, completely justified to consider B as being parallel to the c axis.) The dynamic relaxation rate Q [defined as $Q = d \ln j_s / d \ln (dB/dt)$], which is essentially equal to the conventional relaxation rate $-d \ln j / d \ln t$, is determined from the variation of j_s as a function of the sweep rate [14,15]. At low temperatures it is experimentally found that j_s depends linearly on $\ln(dB/dt)$ over a large range of sweep rates, $10^{-7} \text{ T/s} \leq dB/dt \leq 4 \times 10^{-2} \text{ T/s}$. At higher temperatures a clear curvature develops. In Figs. 1(a) and 1(b) we show the temperature dependence of j_s and Q at 0.5 T for the three different films. The spread in the absolute values of j_s reflects the microstructural differences between films grown under different conditions and/or on different substrates. Similar data were obtained at 1 and 2 T.

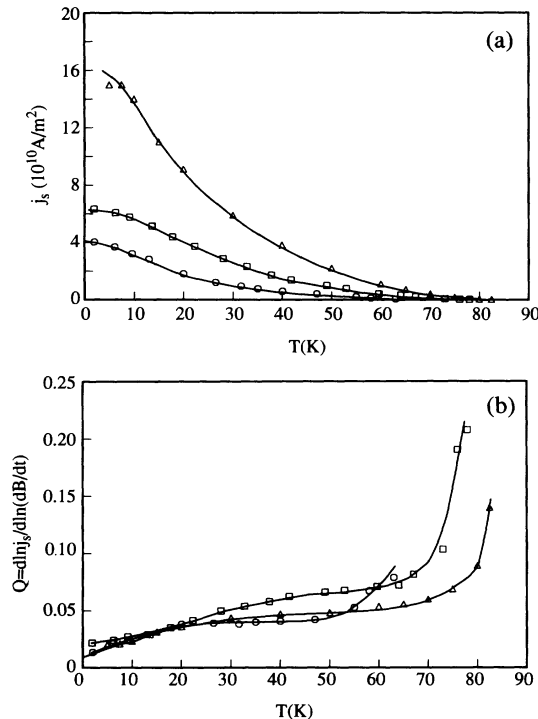


FIG. 1. Experimental data for (a) the superconducting current density j_s and (b) the dynamic relaxation rate $Q \equiv d \ln j_s / d \ln (dB/dt)$ as a function of temperature at $B=0.5$ T. Both j_s and Q are determined at $dB/dt = 4 \times 10^{-2} \text{ T/s}$. The symbols correspond to three different films: (Δ) $\text{YBa}_2\text{Cu}_3\text{O}_7$ magnetron sputtered; (\square) MBE codeposited $\text{YBa}_2\text{Cu}_3\text{O}_7$ with postannealing and (\circ) MBE codeposited $\text{YBa}_2\text{Cu}_4\text{O}_8$ with postannealing. The finite value of Q at $T=0$ is due to quantum creep [16]. The lines are guides to the eye.

In the absence of dissipation $j_c(T,B)$ is equal to $j_s(T,B)$ and can directly be used for a comparison with theory. In the presence of the giant flux creep characteristic of high- T_c superconductors j_s differs, however, markedly from the true critical current j_c (which corresponds to a vanishing activation energy for thermally activated vortex motion) and Q is not simply related to the activation energy $U(j,T,B)$ which governs the hopping probability ($\propto \exp[-U(j,T,B)/kT]$) of a vortex or vortex (super)bundle. As shown recently by Schnack *et al.* [16], this is not a major difficulty since it is possible to determine reliably $j_c(T,B)$ and $U(j,T,B)$ over a large portion of the (j,T) plane (at constant B) by means of the so-called generalized inversion scheme (GIS). The GIS which is described in detail in Ref. [16] is much more general than all the methods of analysis proposed until now [17] because it assumes only that

$$U(j,T,B) = g(T,B) f(j/j_c(T,B), B) \quad (3)$$

and

$$g(T,B) = \left[\frac{j_c(T,B)}{j_c(0,B)} \right]^p G(T). \quad (4)$$

As $j_c = j_c(T,B)$ is defined by the condition $U(j_c, T, B) = 0$, we have necessarily $f(1, B) = 0$. Since only the product $g \times f$ is physically relevant, both g and f are determined up to a multiplicative constant. We set arbitrarily $g(0, B) = 1$. The thermal function $G(T)$ depends on the dimensionality of the superconductor under investigation and on the vortex pinning and creep regime. For a collectively pinned single vortex (SV) one finds $p = 1/2$ and [18]

$$G_{\text{SV}}(T) = (1-t^2)^{-1/4} (1+t^2)^{5/4} \quad (5)$$

as follows directly from Eq. (2) with $B_c \propto (1-t^2)$, $\xi \propto [(1+t^2)/(1-t^2)]^{1/2}$, and $\lambda \propto (1-t^4)^{-1/2}$ with $t = T/T_c$. For the collective creep of small vortex superbundles (SB) in three dimensions Wen *et al.* [7] found $p = -3/2$ and

$$G_{\text{SB}}(t) = (1-t^2)^{3/4} (1+t^2)^{17/4}. \quad (6)$$

Finally for large vortex superbundles (LB) $p = -1/2$ and

$$G_{\text{LB}}(t) = (1-t^2)^{-3/4} (1+t^2)^{7/4}. \quad (7)$$

Both Eqs. (6) and (7) apply to the case of intermediate and large fields when the vortex-vortex interaction plays an essential role. In order to avoid excessive compressional energy increases during a creep process vortex bundles form elongated superbundles containing $(C_{11}/C_{66})^{1/2}$ elementary bundles [19]. This has been taken into account in deriving Eqs. (6) and (7).

We show now that at $B=0.5$ T all the data in Figs. 1(a) and 1(b) are in the single vortex regime over the whole temperature range. To illustrate how this conclusion can be drawn we consider in Fig. 2 the $j_c(T)$ curves obtained by means of the GIS, from the data for the MBE codeposited $\text{YBa}_2\text{Cu}_3\text{O}_7$ (YBCO) film for the three values $p = -3/2, -1/2$, and $1/2$ predicted by theory

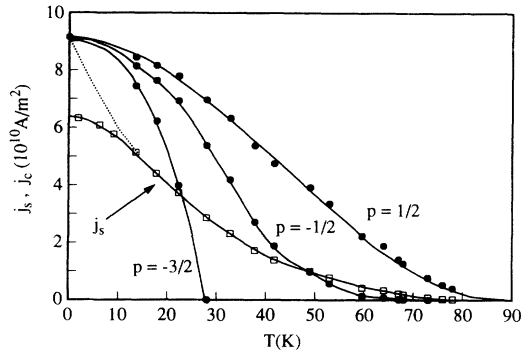


FIG. 2. Critical current density $j_c(T)$ values (●) obtained point by point by means of the generalized inversion scheme (GIS) of Schnack *et al.* [16] from the MBE codeposited $\text{YBa}_2\text{Cu}_3\text{O}_7$ film data in Fig. 1 (□) for three different values of the parameter p in Eq. (4): $p=1/2$ corresponds to the single vortex regime, $p=-1/2$ to the small superbundle regime, and $p=-3/2$ to the large superbundle regime. The curves are guides to the eye. The dotted line indicates how j_s would vary with temperature in the absence of quantum creep [16]. Similar curves are obtained for the other films.

(see above). Both for $p=-3/2$ and $-1/2$ the critical current j_c drops below the measured current j_s above a certain temperature. This would imply that above this temperature the vortex system does not creep but flows. This is in contradiction with the nearly logarithmic dependence of j_s on dB/dt observed at these temperatures. Consequently, $p=1/2$ is the only physically acceptable value [20]. Very similar curves are obtained for the data of the two other types of films (MBE postannealed $\text{YBa}_2\text{Cu}_4\text{O}_8$ and magnetron sputtered $\text{YBa}_2\text{Cu}_3\text{O}_7$). Another evidence in favor of the single vortex regime is the weak field dependence of j_c . Up to $B=7$ T we found for all three types of samples that j_c decreases with increasing B approximately as $B^{-\alpha}$ with $\alpha \approx 0.3$. This is much weaker than the exponential decay $j_c \propto \exp[-2B^{3/2}]$ or $j_c \propto B^{-3}$ predicted by theory in the small bundle and large bundle regimes, respectively [1,19,21]. A further argument is that on theoretical grounds [1] it is expected that the single vortex regime is realized as long as $B < B_{\text{SB}} \cong 5(j_c/j_0)B_{c2}$. With $j_c = 10^{11}$ A/m², $j_0 \cong B_c/\mu_0\lambda = 10^{13}$ A/m², and $B_{c2} = 50$ T one obtains $B_{\text{SB}} \cong 2.5$ T. An additional piece of evidence for $p=1/2$ has also been found from measurements of U_c and j_c in a series of oxygen deficient $\text{YBa}_2\text{Cu}_3\text{O}_x$ films with $6.35 \leq x \leq 7$, which lead to $U_c(x) \propto [j_c(x)]^{0.51}$ [22].

In Fig. 3 we show the critical current as a function of temperature as determined point by point from the data in Figs. 1(a) and 1(b) by means of the GIS with $p=1/2$. To demonstrate the great similarity between the critical currents of the three samples the j_c 's are scaled to their respective values at $T=0$ and the temperature is normalized to T_c . Since the GIS makes it possible to reconstruct U as a function of j and T for a given external field, it is, in particular, possible to determine the functions g and f in Eqs. (3) and (4). For the purpose of the present work

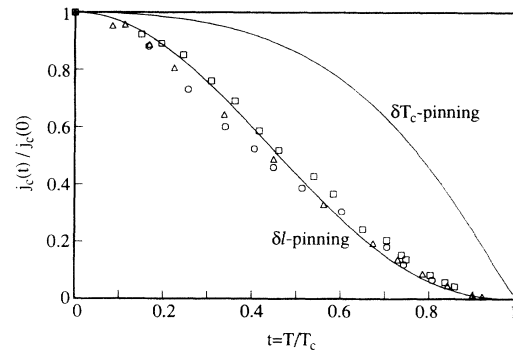


FIG. 3. Normalized temperature dependence of the critical current density for three samples at $B=0.5$ T. The symbols are the same as in Figs. 1(a) and 1(b). The “ δl -pinning” line corresponds to Eq. (8) and the “ δT_c -pinning” line to Eq. (10).

we focus our attention on $g(T, B=0.5 \text{ T})$ which is plotted as a function of the reduced temperature T/T_c in Fig. 4 for the same three samples as in Fig. 3. The fact that the data points in Figs. 3 and 4 merge into a narrow band is rather remarkable since the films are of clearly different types.

At this point it is very important to stress that the data points in Figs. 3 and 4 have been obtained *without any assumptions about the type of pinning*. The temperature dependences in the thermal functions $G(T)$ entering the GIS [Eqs. (5) to (7)] are only related to the T dependence of equilibrium thermodynamic parameters such as B_c , λ , and ξ . The disorder parameter δ does not enter at all the GIS.

Having emphasized this we compare now the data in Figs. 3 and 4 to the prediction of specific theoretical models. As all the data are in the single vortex regime, j_c and U_c can directly be calculated from the relations in Eq. (1) once a theoretical model has been chosen for δ . For δl pinning the disorder parameter is $\delta_m \propto \xi^{-3}$ and one obtains from Eq. (1)

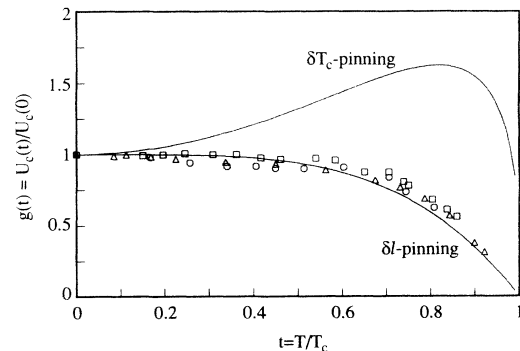


FIG. 4. Normalized temperature dependence of the $g(t)$ function introduced in the generalized inversion scheme [Eqs. (3) and (4)] for three different samples at $B=0.5$ T. This function has the same temperature dependence as the characteristic pinning energy U_c [see Ref. [1]]. The symbols are the same as in Figs. 1(a) and 1(b). The “ δl -pinning” line corresponds to Eq. (9) and the “ δT_c -pinning” line to Eq. (11).

$$j_c(t)/j_c(0) = (1-t^2)^{5/2}(1+t^2)^{-1/2}, \quad (8)$$

$$g(t) = 1 - t^4. \quad (9)$$

For the $g(t)$ function we have used the fact that $g(T, B)$ introduced in the GIS corresponds to $U_c(T, B)/U_c(0, B)$. The lines corresponding to Eqs. (3) and (4) are indicated as “ δl pinning” in Figs. 3 and 4. The central result of this Letter is the remarkably good agreement found between the $j_c(t)/j_c(0)$ and $g(t)$ data and the corresponding “ δl -pinning” lines in these figures. In sharp contrast, the “ δT_c -pinning” lines in Figs. 3 and 4 obtained with $\delta_a \propto \xi$, i.e.,

$$j_c(t)/j_c(0) = (1-t^2)^{7/6}(1+t^2)^{5/6}, \quad (10)$$

$$g(t) = (1-t^2)^{1/3}(1+t^2)^{5/3}, \quad (11)$$

are in total disagreement with the experimental data. This implies that the relevant disorder parameter varies as ξ^{-3} and not as ξ . This conclusion also provides us with a direct explanation of the seemingly large spread in values for L_c reported in literature. From resistivity measurements on YBCO/PrBCO multilayers with increasing YBCO layer thickness Brunner *et al.* [23] concluded that $L_c \cong 40$ nm in bulk $\text{YBa}_2\text{Cu}_3\text{O}_7$ for $B \leq 6$ T at $T/T_c \cong 0.9$. On the other hand, van Dalen *et al.* [24] concluded from their study of quantum creep at low temperatures in similar samples that $L_c \cong 4$ nm. This order of magnitude difference between $L_c(t=0)$ and $L_c(t=0.9)$ is easily understood on the basis of Eq. (1) where for δl pinning $L_c \propto \xi^2 \propto (1+t^2)/(1-t^2)$ and consequently $L_c(t=0.9)/L_c(t=0) = 9.5$. For δT_c pinning $L_c \propto \xi^{2/3}$ and $L_c(t=0.9)/L_c(t=0) = 2.1$ which is much too small.

Finally a rough estimate of the relative importance of δT_c pinning and δl pinning can be made by noting that the reduced temperature at which both disorder parameters δ_a and δ_m are equal is certainly larger than 0.9 as no significant deviations from the “ δl -pinning” line are observed in Fig. 3 below $t=0.9$. Since $\delta_a/\delta_m \propto \xi^4$, $\delta_a/\delta_m = 1$ at $t \geq 0.9$ implies that $\delta_a(t=0)/\delta_m(t=0) \leq 10^{-2}$.

In conclusion, we have found strong evidence for the importance of δl pinning, i.e., pinning related to spatial variations of the charge carrier mean free path in stoichiometric yttrium-based superconductors. Over the temperature range considered in this work δT_c pinning is at least 2 orders of magnitude weaker. This seems to be generally valid since it has been found in MBE codeposited and magnetron sputtered $\text{YBa}_2\text{Cu}_3\text{O}_7$ as well as in $\text{YBa}_2\text{Cu}_4\text{O}_8$ films.

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