

## Pressure-Induced High-Spin to Low-Spin Transition in $\text{CaFeO}_3$

M. Takano,<sup>(1),(a)</sup> S. Nasu,<sup>(1)</sup> T. Abe,<sup>(1)</sup> K. Yamamoto,<sup>(1)</sup> S. Endo,<sup>(1)</sup> Y. Takeda,<sup>(2)</sup>  
and J. B. Goodenough,<sup>(3)</sup>

<sup>(1)</sup>Research Center for Extreme Materials and Department of Materials Physics,  
Faculty of Engineering Science, Osaka University, Toyonaka, Osaka 560, Japan

<sup>(2)</sup>Department of Chemistry, Faculty of Engineering, Mie University, Tsu 514, Japan

<sup>(3)</sup>Center for Materials Science and Engineering, ETC 5.160, University of Texas at Austin, Austin, Texas 78712-1084  
(Received 29 July 1991)

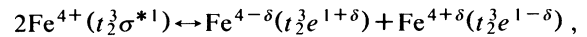
*In situ*  $^{57}\text{Fe}$  Mössbauer spectroscopy and x-ray diffraction on  $\text{CaFeO}_3$  ( $\text{Fe}^{4+}:t_{2g}^3e^1$ ) under pressure have revealed a first-order high-spin ( $S=2$ ) to low-spin ( $S=1$ ) transition near 30 GPa. The spin and structure transition is an alternative manifestation of the electronic instabilities of the  $\sigma$ -bonding Fe-O-Fe orbitals; at atmospheric pressure a second-order disproportionation  $2\text{Fe}^{4+} \leftrightarrow \text{Fe}^{4-\delta} + \text{Fe}^{4+\delta}$ ,  $0 \leq \delta \leq 1$ , occurs at a  $T_i \approx 290$  K. Such instabilities found at the transition from localized to itinerant  $\sigma^*$ -band electrons reflect a sensitivity to interactions with the lattice.

PACS numbers: 62.50.+p, 61.50.Em, 71.70.Ch, 76.80.+y

The recent discovery of high-temperature superconductivity in copper oxides [1] and a new interpretation of photoemission spectroscopy data [2] have revived interest within the condensed-matter physics community in a fuller understanding of the 3d transition-metal oxides. For example, spectroscopic data [3] appear to indicate that the band gaps of insulating late transition-metal oxides like NiO are of a ligand-to-metal charge-transfer type. Perovskite oxides have a relatively simple structure and allow late 3d transition metals to be stabilized in unusually high valence states. An increase in formal valency makes the M-O bond more covalent, thereby increasing the admixture of O:2p and M:3d wave functions in the electronic states near the Fermi energy. Transitions from more ionic to more covalent bonding may enhance the importance of electron-lattice interactions.

The properties of the perovskites  $\text{LaMnO}_3$ ,  $\text{CaFeO}_3$ , and  $\text{SrFeO}_3$  provide an interesting comparison. Each compound contains an octahedral-site transition-metal ion with a high-spin  $d^4=t_{2g}^3e^1$  configuration at atmospheric pressure. In each case the half-filled  $t_{2g}^3$  orbitals are localized, while an evolution in the strength of the e-O:2p $\sigma$ -e interactions imparts quite distinctive properties to each compound. In  $\text{LaMnO}_3$ , the e electrons are sufficiently localized that a classical cooperative Jahn-Teller distortion removes the e-orbital degeneracy [4]. Stronger covalent mixing at an  $\text{Fe}^{4+}$  versus a  $\text{Mn}^{3+}$  ion increases the ca. 180° e-O:2p $\sigma$ -e interactions, and the localized e orbitals become transformed into a narrow  $\sigma^*$  band that is one-quarter filled [5]. Such a band would contribute itinerant-electron ferromagnetic coupling to compete with antiferromagnetic  $t_{2g}^3$ -O:2p $\pi$ - $t_{2g}^3$  superexchange interactions; it would also give metallic conductivity to lowest temperatures. Indeed,  $\text{SrFeO}_3$  is metallic down to 4 K [6] and has a ferromagnetic nearest-neighbor interaction [7]. On the other hand,  $\text{CaFeO}_3$  undergoes a second-order phase transition that is typical of a narrower  $\sigma^*$  band at the transition from localized to itinerant-electronic behavior [8]. The transition is a

disproportionation reaction expressed formally as



where  $\delta=0$  for  $T > 290$  K increases continuously with decreasing temperature below 290 K; typically  $\delta$  approaches unity at low temperatures.

A recent x-ray photoemission and ultraviolet photoemission spectroscopic study of  $\text{SrFeO}_3$  deduced a very small charge-transfer energy of 0–1 eV and also a small charge fluctuation energy,  $\ll 1$  eV, for  $d^4+d^4 \rightarrow d^5+d^4\bar{L}$  ( $\bar{L}$  denotes ligand hole) as the common background of the itinerant character of  $\text{SrFeO}_3$  and the charge disproportionation of  $\text{CaFeO}_3$  [9]. The disproportionation could be interpreted naturally as the formation of a charge-density wave in which a cooperative oxygen displacement introduces ca. 180°  $\text{Fe}^{5+}\text{-O}\cdots\text{Fe}^{3+}$  interactions in all three crystallographic directions. Such a freezing out of a breathing phonon mode has found theoretical support [10]. Where the e-O:2p $\sigma$ -e interactions are stronger, as appears to be the case of  $\text{SrFeO}_3$  in spite of the larger lattice parameter, the quarter-filled  $\sigma^*$  band of itinerant electrons does not disproportionate.

Mössbauer spectroscopic studies [8,11] of the systems  $\text{Ca}_{1-x}\text{Sr}_x\text{FeO}_3$  and  $\text{Sr}_{1-y}\text{La}_y\text{FeO}_3$  have revealed a delicate balance between the relative strengths of the Fe-O-Fe interactions and the electron-phonon interactions for the disproportionation reaction to occur. For example, the disproportionation parameter  $\delta$  was found experimentally to depend monotonically upon the Sr content in each solid solution. It is only pure  $\text{SrFeO}_3$  that retains the single  $\text{Fe}^{4+}$  ( $\delta=0$ ) state down to lowest temperatures.

High pressure provides an alternative physical variable in which to compare these three compounds. For example, we may anticipate a transition from high-spin to low-spin  $\text{Fe}^{4+}$  as an alternative to disproportionation for the creation of empty, itinerant  $\sigma^*$  states and localized electrons—in this case as localized  $t_{2g}^4$  configurations; according to theoretical calculations [12], the gap between the two spin states tends to collapse quickly as the Fe-O

bond is contracted. High pressure also changes the relative strengths of the antiferromagnetic  $t_2^3\text{-O:}2p_{\pi}\text{-}t_2^3$  and either ferromagnetic  $e^1\text{-O:}2p_{\sigma}\text{-}e^0$  superexchange interactions or ferromagnetic correlations within a quarter-filled  $\sigma^{*1}$  band of itinerant-electron states. We have reported previously [13] on the effect of hydrostatic pressure on  $\text{SrFeO}_3$ ; here we report a pressure-induced high-spin to low-spin transition in  $\text{CaFeO}_3$  found by *in situ* Mössbauer spectroscopy. The transition is accompanied by a serious structural change as found *in situ* by x-ray diffraction (XRD).

$\text{Ca}^{57}\text{FeO}_3$  was prepared by oxidation of  $\text{Ca}_2^{57}\text{Fe}_2\text{O}_5$  in a belt high-pressure apparatus using  $\text{KClO}_4$  as an oxygen source under 6 GPa at  $1050^\circ\text{C}$  for 30 min. The product was shown to be single phase by XRD and Mössbauer spectroscopy. At room temperature,  $\text{CaFeO}_3$  has the tetragonal perovskite structure with  $a_t = 5.332 \text{ \AA}$  and  $c_t = 7.550 \text{ \AA}$ , where  $a_t/\sqrt{2} \approx c/2 \approx 3.8 \text{ \AA}$  is a typical cubic-perovskite lattice constant [14]. Mössbauer measurements under pressure up to 51 GPa were performed with a diamond-anvil cell of the Bassett type charged with the sample powder in a rhenium gasket with a 0.3-mm-diam hole [12]. The applied pressure was monitored by the fluorescent shift of ruby powder mixed with the sample. XRD was also performed with a similar diamond-anvil cell. A new sample of  $\text{SrFeO}_3$  was also prepared in the same manner and used for XRD under pressure up to 53 GPa.

The pressure dependence of the room-temperature Mössbauer spectrum is shown in Fig. 1. The ambient-pressure spectrum of a sample containing natural iron, Fig. 1(a), consists of a single narrow line with a CS of  $0.048 \text{ mm/s}$  relative to  $\alpha\text{-Fe}$ . At 21 GPa, Fig. 1(b), a broadened singlet and a weak magnetic component are mixed; the broadening is a saturation effect caused by the packing of the  $^{57}\text{Fe}$ -enriched ( $>90\%$ ) sample in the tiny pressure cell. The magnetic component comes from  $\text{Ca}_2\text{Fe}_2\text{O}_5$ , which is stable under low oxygen pressures and appears to have formed either during a long-term storage of the sample in a desiccator or from heating in the laser beam used for high-pressure calibration. Moreover, the singlet absorption peak from  $\text{CaFeO}_3$  is more strongly broadened by saturation than is the magnetic component due to the  $\text{Ca}_2\text{Fe}_2\text{O}_5$  impurity, which emphasizes the magnetic contaminant.

At 30 GPa, Fig. 1(c), the central singlet peak is beginning to be replaced by a magnetic pattern. At higher pressure, the magnetic splitting becomes clear. Like  $\text{SrFeO}_3$ , but unlike  $\text{CaFeO}_3$  in its disproportionated state,  $\text{CaFeO}_3$  under pressure contains only  $\text{Fe}^{4+}$  ions at room temperature. However, unlike  $\text{SrFeO}_3$ , the CS of  $\text{CaFeO}_3$  is reduced abruptly at 30 GPa by  $0.25 \text{ mm/s}$  and a magnetic hyperfine field of  $H_i = 16 \text{ T}$  appears, see Fig. 1(e). The  $H_i$  value remains constant at higher pressures. The broadened spectrum in the transition region—Fig. 1(c)—can be reproduced by superimposing the paramag-

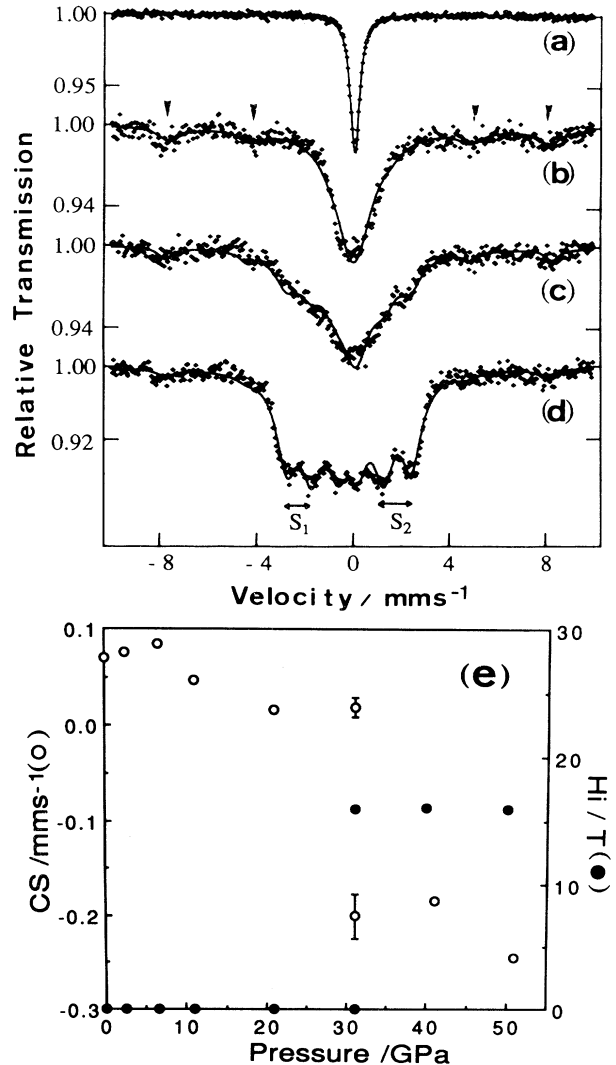


FIG. 1. Pressure-dependent Mössbauer spectra of  $\text{CaFeO}_3$  at room temperature. Applied pressures are (a) 0.1 MPa, (b) 21 GPa, (c) 30 GPa, and (d) 41 GPa ( $S_1 - S_2 = -0.10 \text{ mms}^{-1}$ ). The sample for (a) contains natural iron, while the other spectra were obtained with an  $^{57}\text{Fe}$ -enriched sample. Solid lines are the computer-fitted results. The peaks marked by arrows in (b) come from  $\text{Ca}_2\text{Fe}_2\text{O}_5$  (see text). Pressure-dependent center shift (CS) and magnetic hyperfine field ( $H_i$ ) data for  $\text{CaFeO}_3$  are shown in (e).

netic singlet of Fig. 1(b) and the magnetic pattern of Fig. 1(d). We are thus led to the conclusion that a first-order transition occurs at a  $P_c \approx 30 \text{ GPa}$ .

High-pressure XRD shows a change on crossing the transition from the  $\sqrt{2}a \times \sqrt{2}a \times 2a$  tetragonal structure to a  $2a \times 2a \times 2a$  orthorhombic structure, where  $a$  represents the basic perovskite parameter. Shown in Fig. 2 is the pressure-induced splitting of the (220) peak into the (400) and (040) peaks of the high-pressure phase as well as the pressure dependence of the lattice constants on

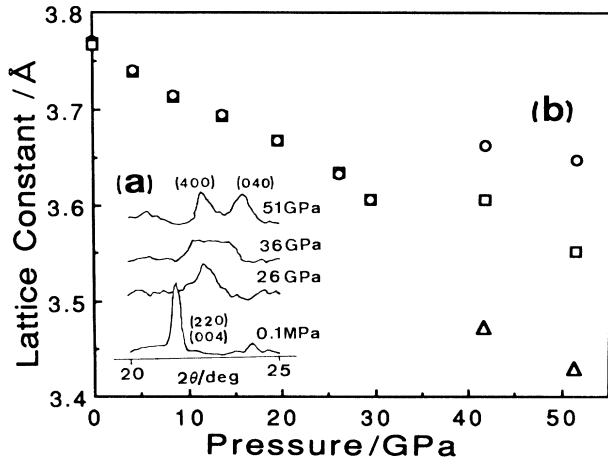


FIG. 2. Pressure-dependent (a) XRD peaks and (b) lattice constants for the hypothetical unimolecular perovskite cell for  $\text{CaFeO}_3$ . In reality  $\text{CaFeO}_3$  has a tetramolecular tetragonal cell of  $a_t \times a_t \times c_t$  for  $P \leq 30$  GPa and an octomolecular orthorhombic one of  $a_o \times b_o \times c_o$ . Plotted here are (○)  $a_t/\sqrt{2}$  and (□)  $c_t/2$  for  $P \leq 30$  GPa, and (○)  $a_o/2$ , (△)  $b_o/2$ , and (□)  $c_o/2$  for  $P > 30$  GPa.

crossing the transition. The orthorhombic distortion of the high-pressure phase is quite remarkable. The appearance of a quadruple interaction in the Mössbauer spectra of the high-pressure phase [Fig. 1(d)] provides evidence of the structural change with a microscopic probe. On release of pressure the original low-pressure phase is recovered as seen by both Mössbauer and XRD measurements.

On the other hand, isotropic compression of  $\text{SrFeO}_3$  up to 53 GPa causes a considerable increase in the Néel temperature  $T_N$  ( $\approx 300$  K at 19 GPa), but it does not induce any disproportion or any other transition [13]. The structure remains cubic over the pressure range studied without any noticeable anomaly at  $T_N$ .

The first-order phase change at a  $P_c \approx 30$  GPa appears to be a high-spin to low-spin transition for the following reasons. First, the drop in CS by  $\sim 0.25$  mm/s is quite remarkable. Such a drop requires a significant electron-density redistribution such that the  $s$ -electron density at an  $^{57}\text{Fe}$  nucleus is enhanced by  $\Delta\rho(0) \sim 0.8$  a.u. [15] in the high-pressure phase. Although we have no quantitative estimates, the observed change in CS is consistent with what has been found at high-spin to low-spin transitions in various iron compounds. Second, the pressure independence of  $H_i$  signals that saturation has been reached, yet its value of about 16 T is significantly smaller than the 23.3 T found for  $\text{SrFeO}_3$  at 53 GPa [13]; it is also smaller than the saturated  $H_i$  values of  $\text{SrFeO}_3$  and  $\text{CaFeO}_3$  at ambient pressure. The  $H_i = 33\text{--}34$  T observed for high-spin  $\text{Fe}^{4+}$  ( $S=2$ ) [8,11] may be expected to be scaled down by a factor of 2 at a low-spin  $\text{Fe}^{4+}$  ( $S=1$ ) ion in high-pressure  $\text{CaFeO}_3$ .

A low-spin state occurs where the cubic-field splitting  $\Delta_c$  exceeds the intra-atomic-exchange splitting  $\Delta_{\text{ex}}$ . It is reasonable to assume that an external pressure increases  $\Delta_c$  through a more enhanced increase in covalency for  $\sigma$  vs  $\pi$  bonding; both  $\text{CaFeO}_3$  and  $\text{SrFeO}_3$  could exhibit a high-spin to low-spin transition. In order to understand why the low-spin state of  $\text{CaFeO}_3$  is stabilized for a unit-cell volume  $V \leq 47 \text{ \AA}^3$  whereas  $\text{SrFeO}_3$  remains cubic and retains the high-spin state even for  $V = 47 \text{ \AA}^3$  consideration of the differences between  $\text{Ca}^{2+}$  and  $\text{Sr}^{2+}$  ions is required.

The  $\text{Ca}^{2+}$  ion is distinguished from the  $\text{Sr}^{2+}$  ion not only by its smaller size (1.35 vs 1.44  $\text{\AA}$  for twelfold coordination), but also by the greater covalent component in the Ca-O bond. The smaller size of the  $\text{Ca}^{2+}$  ion induces a distortion of the structure from cubic to tetragonal symmetry whereas  $\text{SrFeO}_3$  remains cubic to lowest temperatures. The cubic perovskite retains  $180^\circ$  Fe-O-Fe bonding whereas the Fe-O-Fe bonds are bent from  $180^\circ$  in tetragonal  $\text{CaFeO}_3$ . This bending makes the Fe-O bond length greater than half the Fe-Fe distance; it also reduces the strength of the Fe-O-Fe interactions, which is probably why the  $\sigma^*$  band in  $\text{CaFeO}_3$  appears to be narrower than that in  $\text{SrFeO}_3$  despite the smaller lattice parameter of  $\text{CaFeO}_3$ .

The covalent component of the A-O bond of an  $\text{ABO}_3$  perovskite competes directly with the  $t_2$  orbitals of a transition-metal B cation for the  $\text{O}:2p_\pi$  electrons. The difference in covalency between the  $\text{Ca}^{2+}$  and  $\text{Sr}^{2+}$  ions makes the  $\pi$  component of the covalent Fe-O bond smaller in  $\text{CaFeO}_3$  than in  $\text{SrFeO}_3$ . Moreover, bending of the Fe-O-Fe bond from  $180^\circ$  reduces the strength of the Fe-O-Fe interactions, but it does not reduce the  $\sigma$  component of the Fe-O bond covalency. Thus,  $\Delta_c$  in  $\text{CaFeO}_3$  would be increased relative to  $\Delta_c$  in  $\text{SrFeO}_3$ , and it should take a smaller increase in  $\Delta_c$  to make  $\Delta_c/\Delta_{\text{ex}} \approx 1$  in the case of  $\text{CaFeO}_3$ , which would account for a lower  $P_c$  in  $\text{CaFeO}_3$  than in  $\text{SrFeO}_3$ . Another mechanism mediated by a coupling with the lattice system is a strain-induced splitting and crossing of the highest up-spin levels and the lowest down-spin levels. According to theoretical estimations [12], the gap between the two spin states for a regular  $(\text{FeO}_6)^{8-}$  octahedron embedded in a Madelung potential representing the host lattice decreases with decreasing bond distance and finally vanishes at  $d_{\text{Fe-O}} \approx 1.7 \text{ \AA}$  ( $V \approx 40 \text{ \AA}^3$  for a cubic unit cell). However, the gap may be collapsed at a larger average Fe-O distance if the distortion of the octahedron is serious enough to cause large splittings of the relevant levels. A broader  $\sigma^*$  band in  $\text{SrFeO}_3$  may allow a more gradual crossover to the low-spin state without any structural distortion.

As noted above,  $dH_i/dP \approx 0$  in high-pressure  $\text{CaFeO}_3$  indicates a magnetically saturated state. From the Brillouin function for  $S=1$  and a saturation  $H_i = 16$  T, we estimate a magnetic transition temperature  $T_m \geq 800$  K, which is much higher than the  $T_N = 116$  K for the high-

spin state under ambient pressure. The low-spin  $t_2^4$ -O:2p $\pi$ - $t_2^4$  antiferromagnetic superexchange interactions have no ferromagnetic competition from the empty  $\sigma$ -bonding orbitals. By way of contrast, ferromagnetic SrRuO<sub>3</sub> ( $T_c = 160$  K) contains degenerate and delocalized  $t_2$  orbitals forming a narrow  $\pi^*$  band that is two-thirds filled. A narrower  $\pi^*$  band in slightly distorted CaRuO<sub>3</sub> gives a negative Weiss constant in the Curie-Weiss paramagnetic susceptibility and remains paramagnetic down to 4 K [16]. The localized  $t_2^4$ -O- $t_2^4$  configurations in CaFeO<sub>3</sub> give a strong antiferromagnetic superexchange interaction.

In summary, high-pressure Mössbauer spectroscopy and XRD have demonstrated the existence of a first-order phase change at a critical pressure  $P_c \approx 30$  GPa from a high-spin to a low-spin state at the Fe<sup>4+</sup> cations of CaFeO<sub>3</sub>. Since CaFeO<sub>3</sub> exhibits a second-order disproportionation into Fe<sup>(4+ $\delta$ )</sup> and Fe<sup>(4- $\delta$ )</sup> below a transition temperature  $T_t = 290$  K at ambient pressure, the low-spin phase represents a third electronic state in CaFeO<sub>3</sub>. From the hyperfine field in the high-pressure phase, the low-spin CaFeO<sub>3</sub> appears to be antiferromagnetic with a Néel temperature estimated to be  $T_N \gtrsim 800$  K. The different electronic behavior of SrFeO<sub>3</sub> has been rationalized on the basis of the larger size of the Sr<sup>2+</sup> ion and smaller covalent component in the Sr-O bond.

- <sup>(a)</sup>Permanent address: Institute for Chemical Research, Kyoto University, Uji, Kyoto-fu 611, Japan.
- [1] J. G. Bednorz and K. A. Müller, *Z. Phys.* **64**, 189 (1986).
  - [2] J. Zaanen *et al.*, *Phys. Rev. Lett.* **55**, 418 (1985).
  - [3] A. Fujimori *et al.*, *Phys. Rev. B* **42**, 7580 (1990).
  - [4] J. B. Goodenough, *Phys. Rev.* **100**, 564 (1955).
  - [5] C. Gleitzer and J. B. Goodenough, *Struct. Bonding* **61**, 1 (1985).
  - [6] J. B. MacChesney *et al.*, *J. Chem. Phys.* **43**, 1907 (1965).
  - [7] T. Takeda *et al.*, *J. Phys. Soc. Jpn.* **33**, 967 (1972).
  - [8] M. Takano *et al.*, *Mater. Res. Bull.* **12**, 923 (1977).
  - [9] A. E. Bocquet *et al.*, *Phys. Rev. B* (to be published).
  - [10] H. Adachi and M. Takano, *J. Solid State Chem.* **93**, 556 (1991).
  - [11] P. K. Gallagher *et al.*, *J. Chem. Phys.* **41**, 2429 (1964); M. Takano *et al.*, *J. Solid State Chem.* **39**, 75 (1981). See a review, M. Takano and Y. Takeda, *Bull. Inst. Chem. Res., Kyoto Univ.* **61**, 406 (1983).
  - [12] H. Adachi and M. Takano (unpublished).
  - [13] S. Nasu *et al.*, *Hyperfine Interact.* (to be published).
  - [14] Y. Takeda *et al.*, *Mater. Res. Bull.* **13**, 61 (1978).
  - [15] R. Ingalls, A. Van der Woude, and G. A. Sawatzky, in *Mössbauer Isomer Shifts*, edited by G. K. Shenoy and F. E. Wagner (North-Holland, Amsterdam, 1978), Chap. 7.
  - [16] J. B. Goodenough, *Prog. Solid State Chem.* **5**, 145 (1971); T. C. Gibb *et al.*, *J. Solid State Chem.* **11**, 17 (1974).