Anisotropic-Defect Production in Compound Semiconductors by Electron Irradiation

In a recent Letter, Pons and Bourgoin¹ concluded that the main defects observed by deep-level transient spectroscopy (DLTS) in electron-irradiated GaAs $(E1, E2, \text{ and } E3)^2$ are caused by Asatom displacements. This is in direct conflict with the interpretation of Lang, Logan, and Kimerling,³ who had previously concluded that the displacements were of Ga atoms. Both used arguments based upon the anisotropy of defect production by high-energy electron irradiation.

Lang, Logan, and Kimerling had found that the damage production rate at 1 MeV was greater along the [111] Ga direction than the [111] As direction and concluded therefore that Ga displacements were involved. (The [111] Ga direction is the As to Ga molecule direction with an empty lattice space directly behind the Ga atom.) Pons and Bourgoin confirmed their observations at ~1 MeV but reported the surprising result that at lower electron energies, nearer damage threshold, the anisotropy reverses and that [111] As damage dominates, thus suggesting As displacements.

To my knowledge, the only detailed *microscopic* information available on anisotropy of displacement damage in a compound semiconductor is from earlier EPR studies of Frenkel pair formation in ZnSe.⁴ There it was *known* that metalatom displacement was involved and clear evidence of easy damage production in the [111] Zn direction even at 1.5 MeV was demonstrated. Lang, Logan, and Kimerling used this result in support of their original interpretation.³ I have therefore reexamined my original data to see if it contains information that can help clear up this apparent discrepancy.

In Fig. 1, I show previously unpublished data where 1.5-MeV irradiation was alternated between the [111] Se and [111] Zn directions on a single ZnSe sample. The concentration of zinc displacement Frenkel pairs aligned along the [111] beam direction, $V'(\overline{111})$, and along the [111] direction, $V'(\overline{111})$, were monitored directly by EPR. The strong anisotropy of damage rates for the differently oriented pairs clearly reflects the easy [111] Zn displacement.

To compare to the DLTS studies of Pons and Bourgoin¹ and Lang, Logan, and Kimerling,³ however, we must estimate the *total* demage.

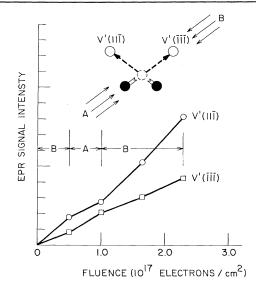


FIG. 1. Production of zinc-vacancy-zinc-interstitial Frenkel close pairs in ZnSe vs 1.5-MeV electron fluence at 20.4 K for two different beam directions: A, [111] Zn; B, [111] Se.

This can be obtained from the figure as $V'(\overline{111}) + 3V'(\overline{111})$, the factor of 3 arising since there are also $V'(\overline{111})$ and $V'(\overline{111})$, which are equivalent to $V'(\overline{111})$ but are not being monitored. We find that the total damage rate (η) is indeed greater in the [111] Se direction

$$\eta_{[111] \text{ Se}} \cong 1.5 \eta_{[111] \text{ Zn}}$$
.

These results therefore confirm the interpretation of Pons and Bourgoin that at these high energies the total damage rate is greater in the "hard-displacement" direction. The *individual* atom displacements, as monitored directly by EPR, still reflect the easy-displacement directions, however.

George D. Watkins

Department of Physics, Lehigh University Bethlehem, Pennsylvania 18015

Received 16 December 1981 PACS numbers: 61.80.Fe, 61.70.Bv, 71.55.Fr

¹D. Pons and J. Bourgoin, Phys. Rev. Lett. <u>47</u>, 1293 (1981).

²D. V. Lang and L. C. Kimerling, in *Lattice Defects in Semiconductors 1974*, IOP Conference Proceedings No. 23 (Institute of Physics, London, 1975), p. 581.

³D. V. Lang, R. A. Logan, and L. C. Kimerling, Phys. Rev. B 15, 4874 (1977).

⁴G. D. Watkins, Phys. Rev. Lett. 33, 223 (1974).