## Photoemission from Metallic Cesium Oxide Films

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Ultraviolet photoelectron spectra of Cs evaporated onto Ag and oxidized under ultrahigh-vacuum conditions have been measured. Time variation of the spectra indicates the formation of Cs suboxides (e.g.,  $Cs_{7}O$ ,  $Cs_{11}O_3$ ). The high infrared photoelectric yield of Ag/O/Cs cathodes can be explained as due to the excitation of surface plasmons in these compounds.

The infrared sensitivity of Ag/O/Cs photocathodes has been interpreted in terms of different mechanisms, such as defects in the lattice of  $Cs_2O$  formed during the production process of Ag/O/Cs cathodes,<sup>1,2</sup> as well as photoemission from silver.<sup>3,4</sup> All of these interpretations generally link the photoemission in the infrared region to the low work function of one of the cesium oxides.<sup>5</sup> Cathodes with an excess of cesium (compared to  $Cs_2O$ ) show the highest infrared yield. Composition and thickness of the Ag/O/Cs layers as well as the electron emission process are still a matter of dispute. Bulk samples of metalrich cesium oxides (suboxides, e.g., Cs<sub>7</sub>O,  $Cs_{11}O_3$ ) have recently been investigated by ultraviolet photoelectron spectroscopy (UPS).<sup>6</sup> It has been shown that the spectra reflect compositions and crystal structures of the bulk compounds.<sup>6</sup>

We report here UPS measurements of thin films of cesium deposited on a silver substrate and oxidized under ultrahigh-vacuum (UHV) conditions, simulating the active layers in Ag/O/Cs (S-1) photocathodes. The spectral response of films prepared in this way is similar to S-1 cathodes according to Heimann *et al.*<sup>7</sup> The UPS data of these films are almost identical to those of the bulk cesium suboxides.<sup>6</sup> A comparison with bulk data yields information about the composition of the films, the layer thickness, and the low-energy photoelectron emission process.

The measurements were performed at room temperature in a Leybold-Heraeus LHS 10 ESCA system equipped with a preparation chamber and a differentially pumped cold-cathode gas-discharge lamp using the HeI resonance line ( $h\nu$ = 21.2 eV). The residual gas pressure in the analyzer chamber was 2×10<sup>-10</sup> Torr. Evaporated silver served as a substrate and was etched by argon-ion bombardment. The photoelectron energy distribution curve of this sample is shown in Fig. 1(a). The energy scale is relative to the Fermi edge; the resolution is 0.1 eV. The photoelectron spectrum of Cs evaporated for 3 min at 390 K onto the Ag substrate at room temperature is shown in Fig. 1(b). Oxygen was leaked into the analyzer chamber (basic pressure  $p = 2 \times 10^{-10}$ Torr) at a partial pressure of  $5 \times 10^{-7}$  Torr for



FIG. 1. Photoelectron energy distribution curves for silver, a cesium layer on silver, and reaction products with oxygen.



FIG. 2. Photoelectron spectra from Cs and the cesium suboxides  $Cs_7O$  and  $Cs_{11}O_3$  (see also Ref. 11). The inset shows a  $Cs_{11}O_3$  cluster.

20 sec, corresponding to an oxygen exposure of 10 L (1 L = 10<sup>-6</sup> Torr sec). The spectrum recorded immediately after this procedure is given in Fig. 1(c). When the sample was left in the analyzer chamber  $(p_{\text{He}} \approx 1 \times 10^{-7} \text{ Torr during mea-}$ surements; 99.999 + % purity of He), the measured spectrum changed continuously and reached a final form after 50 min [Figs. 1(d), 1(e)].

Structure due to photoemission from the 4*d* bands of the Ag substrate<sup>8,9</sup> (in the energy range 4 to 7.5 eV) is observed in all of the spectra. This fact indicates a film thickness of less than 10 Å, as deduced from the short mean free path of electrons in Cs.<sup>10</sup> In the oxidized samples, photoemission from the 4*d* bands of the substrate increases with time, suggesting either a reduction of the layer thickness, island formation within the oxide layer, or both.

The spectrum 1(b), recorded from the evaporated Cs film, is equivalent to that of pure bulk<sup>6</sup> Cs in Fig. 2.<sup>11</sup> The structures of the two spectra show the filled conduction band (CB) of Cs followed by an energy loss ( $\hbar\omega_{sp}$ ), an Auger transition (O<sub>III</sub> VV) and the spin-orbit split Cs5p levels (B,B'). The addition of oxygen leads to three new peaks (O 2p, A,  $A^{i}$ ). These significant changes correspond definitely to cesium suboxide formation. Peaks A and A' strongly increase with time, accompanied by a decrease of the intensities of B and B'.

Cesium suboxides contain characteristic ion clusters of composition  $Cs_{11}O_3$  shown in the inset of Fig. 2.<sup>12</sup> In the stoichiometric compound  $Cs_{11}O_3$ ,<sup>13</sup> which is stable at room temperature, these clusters are close-packed.<sup>14</sup> The compound  $Cs_7O \equiv [Cs_{11}O_3]Cs_{10}$  consists of  $Cs_{11}O_3$  clusters intercalated with a stoichiometric amount of cesium.<sup>15</sup> If a melt of Cs<sub>7</sub>O is kept in vacuum at room temperature, Cs evaporates, leading to crystallization of  $Cs_{11}O_3$ .<sup>6</sup>  $Cs_7O$  contains Cs atoms in two different bonding states.<sup>12</sup> Those in the clusters are partly ionic; the others are purely metallic. Consequently, two pairs of Cs 5p levels appear in the UPS spectra. Peaks A and A' in Fig. 2 correspond to the 5p levels of Cs incorporated in the  $Cs_{11}O_3$  cluster, where B and B' refer to the intercalated cesium. In the spectrum of the compound  $Cs_{11}O_3$ , only the peaks A and A' are observed. The very sharp structure at 2.7 eV is due to photoemission from the nonbonding O 2p levels.<sup>6</sup> This characteristic peak was already observed by Gregory *et al.*<sup>16</sup> who interpreted it in terms of "oxygen atoms dissolved in the cesium metal" without considering the formation of metallic suboxides.

The appearance of the structures O 2p, A and A' in Fig. 1(c) (Ag/Cs/O) indicates that the oxygen exposure is followed by the formation of some  $Cs_{11}O_3$  clusters in the films. From the relative intensities of peaks A and B it can be concluded that Cs is still present in large excess to the clusters. The photoelectron spectrum taken after 20 min indicates a composition of the film which corresponds approximately to the stoichiometry of  $Cs_7O$  [Fig. 1(d)]. This change is due to the continuous loss of cesium. Further loss of Cs causes peaks B and B' to disappear. The remaining Cs 5p levels A and A' indicate that the film is composed mainly of  $Cs_{11}O_3$  clusters. The filled conduction band (CB) as well as the very sharp O 2p peak, characteristic of all compounds containing the  $Cs_{11}O_3$  clusters, definitely exclude the formation of (semiconducting) Cs<sub>2</sub>O which shows only one broad valence band.<sup>16,17</sup> The observation of the characteristic O 2p peak in the photoelectron spectrum of an Ag-Cs<sub>2</sub>O-Cs photocathode by Neil and Mee<sup>18</sup> is further evidence for suboxide formation in S-1 photocathodes.

All Cs suboxides are metallic. Plasma frequen-





FIG. 3. Quantum yield of Mg and Al, normalized to their surface-plasmon energies, and comparison to the S-1 quantum yield, which is normalized to the surface-plasmon energy of  $Cs_{11}O_3$  ( $\hbar\omega_{sp}=1.55$  eV). The data were taken from Ref. 21 (Mg, Al) and Refs. 22-25 (S-1).

cies can easily be deduced from the energy-loss structures in the UPS spectra. With HeI radiation  $(h\nu = 21.2 \text{ eV})$  as an excitation source, we observe surface-plasmon losses (see  $\hbar\omega_{sp}$  in Fig. 2). With He II radiation  $(h\nu = 40.8 \text{ eV})$ , both surface- and volume-plasmon energy losses could be separated. The decrease of the surface-plasmon energy of the Cs suboxides with respect to bulk Cs is caused by a lower electron concentration as discussed in detail in Ref. 6. The observed plasmon energies are 2.0, 1.75, and 1.55 eV for Cs, Cs<sub>7</sub>O, and Cs<sub>11</sub>O<sub>3</sub>, respectively.

Surface-plasmon decay<sup>19</sup> has been used to explain photoemission from free-electron-like metals, e.g., Li and Na,<sup>20</sup> Mg, and Al, as well as from (Cs/O)-activated Ag.<sup>21</sup> If the photoelectricyield curves<sup>21</sup> of Mg and Al are plotted versus excitation energy normalized to  $\hbar \omega_{sb}$  (7.7 eV for Mg and 10.5 eV for Al), the yield maxima are found near unity in the reduced energy scale (Fig. 3). Similarly, the spectral responses of some commercial S-1 cathodes<sup>22-25</sup> are plotted (curves a-d in Fig. 3). The energy scale is reduced to the  $\hbar \omega_{sp}$  of  $Cs_{11}O_3$  (1.55 eV), since  $Cs_{11}O_3$  clusters have here been proved to be the main constituent in the active layers of such cathodes. From the positions of the maxima in Fig. 3 we conclude that surface-plasmon-induced photoemission from metallic cesium-suboxide layers (especially  $Cs_{11}O_3$ ) is the origin of the infrared sensitivity of S-1 photocathodes.

The present results are confined to Ag/Cs/O layers. Further investigations on all bulk cesium oxides and their chemical reaction with the substrate materials are necessary to understand the completely different UPS spectra of oxidized cesium layers on III-V (e.g., GaAs) substrates.<sup>17,26</sup>

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## ERRATA

SPIN-FLIP RAMAN ECHO IN *n*-TYPE CdS. P. Hu, S. Geschwind, and T. M. Jedju [Phys. Rev. Lett. 37, 1357 (1976)].

The wavelength  $\omega_L$  on Fig. 1(b) should read " $\omega_L \sim 4905.26$  Å." On page 1358, column 1, the seventeenth line from the bottom should read "...two longitudinal modes  $\omega_L$  and  $\omega_R$ , where  $(\omega_L - \omega_R)/2\pi = 32$  GHz."

EXACT SOLUTION FOR THE INFLUENCE OF LASER TEMPORAL FLUCTUATIONS ON RESO-NANCE FLUORESCENCE. G. S. Agarwal [Phys. Rev. Lett. 37, 1383 (1976)].

In the process of printing Ref. 6 was changed. This reference should read as follows: J. H. Eberly, in Proceedings of the Symposium on Resonant Light Scattering, Massachusetts Institute of Technology, Cambridge, Massachusetts, April 1976 (to be published).