Correlation-Driven Dimerization and Topological Gap Opening in Isotropically Strained Graphene

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The phase diagram of isotropically expanded graphene cannot be correctly predicted by ignoring either electron correlations, or mobile carbons, or the effect of applied stress, as was done so far. We calculate the ground state enthalpy (not just energy) of strained graphene by an accurate off-lattice quantum Monte Carlo correlated ansatz of great variational flexibility. Following undistorted semimetallic graphene at low strain, multideterminant Heitler-London correlations stabilize between $\approx 8.5\%$ and $\approx 15\%$ strain an insulating Kekulé-like dimerized (DIM) state. Closer to a crystallized resonating-valence bond than to a Peierls state, the DIM state prevails over the competing antiferromagnetic insulating state favored by density-functional calculations which we conduct in parallel. The DIM stressed graphene insulator, whose gap is predicted to grow in excess of 1 eV before failure near 15% strain, is topological in nature, implying under certain conditions 1D metallic interface states lying in the bulk energy gap.

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In graphene, which current technology strives to employ in electronics, an insulating state does not naturally occur. Strain engineering has long been considered as providing mechanisms to pry open the symmetry-induced zero gap of the original semimetallic graphene [SEM, see Fig. 1(a)] honeycomb structure [1]. Among them, a nonisotropic three-directional strain was suggested [2] and verified [1,3–6] to introduce a gauge field and a gap.

An insulating state could alternatively be achieved in graphene by simple isotropic expansive strain. Experimentally, indentation experiments suggested that graphene can be isotropically stressed until mechanical failure near 22.5% strain, corresponding to a tensile stress around 50 N/m [7]. No evidence of structural or electronic transition occurring during expansion was provided. Theoretically, idealized rigid-honeycomb Hubbard model, quantum Monte Carlo (QMC) calculations had long suggested [8-12] band narrowing and increased effective electron-electron repulsion could push the correlated π -electron system towards an undistorted honeycomb antiferromagnetic insulator [AFI, see Fig. 1(b)]. Spinpolarized density functional theory (DFT) calculations [1,13] as well as rigid-lattice QMC simulations [14,15] indeed suggest a SEM-AFI crossing of total energies with a semimetal-insulator transition around (8-10)% strain. Alternatively, isotropically stressed graphene could distort to form Peierls or Kekulé-like dimerized [DIM, see Fig. 1(c)] states, discussed by detailed DFT phonon calculations [16,17] and by symmetry considerations [18], with a unitcell size increase from two to six carbons, and an electronic



FIG. 1. (a) SEM honeycomb, semimetallic; (b) AFI honeycomb antiferromagnetic insulator; (c) DIM dimerized Kekulé-like insulator; (d) HEX distorted hexagonal insulator. There are two carbons per unit cell in (a) and (b), six in (c) and (d). Following Ref. [18], t_A , t_B , and t_C schematically denote hopping integrals magnitudes.

gap proportional to the dimerization magnitude. The DIM distortion scenario is nevertheless denied by spin-polarized DFT calculations where the AFI state has lower energy than DIM.

All this work thus leaves the electronic and structural phase diagram of isotropically strained graphene in a state of uncertainty, on two separate accounts. First, the strong band narrowing and increased role of strong electron correlations, improperly treated by DFT, calls for a novel QMC description capable of describing real strained and deformable graphene, a goal never attained so far. Second, the phase diagram under stress must be obtained by comparing enthalpies, therefore including the stress-strain term, rather than just total energies, as was universally done so far. Because the stress-strain equation of state is different for different phases, the correct phase diagram will not be identical to that suggested by minimizing total energy alone. Here, we implement accurate QMC enthalpy calculations, reaching a highly instructive phase diagram for isotropically strained graphene, that is found to differ from that predicted by the best, spin-polarized, DFT.

Main QMC calculations were conducted based on a variational wave function (JAGP), known to be accurate and reliable in the description of strong electron correlations from small molecules [19] to realistic crystalline systems [20],

$$\Psi_{\text{JAGP}} = \mathcal{J}(\mathbf{r}_1, \mathbf{r}_2, ..., \mathbf{r}_N) \Psi_{\text{AGP}}(\mathbf{r}_1 \sigma_1, \mathbf{r}_2 \sigma_2, ..., \mathbf{r}_N \sigma_N),$$
(1)

where \mathbf{r}_i and σ_i , for i = 1, ..., N, are the spatial and the spin coordinates of the electrons. Here, $\mathcal{J} = \prod_{i < i} \exp[u(\mathbf{r}_i, \mathbf{r}_i)]$ is the Jastrow factor, symmetric under all particle permutations, while the determinantal part is the antisymmetrized geminal power (AGP): $\Psi_{AGP} = \mathcal{A}f(\mathbf{r}_1, \mathbf{r}_2)\chi_{\sigma_1, \sigma_2} \cdots f(\mathbf{r}_{N-1}, \mathbf{r}_N)\chi_{\sigma_{N-1}, \sigma_N}$, where \mathcal{A} is the antisymmetrizer and the product $f(\mathbf{r}, \mathbf{r}')\chi(\sigma, \sigma')$ describes a singlet valencebond electron pair with an orbital-symmetric function $f(\mathbf{r}, \mathbf{r}')$ and a spin-antisymmetric one $\chi_{\sigma,\sigma'} = (1/\sqrt{2})$ $(\delta_{\sigma,\uparrow}\delta_{\sigma',\downarrow} - \delta_{\sigma,\downarrow}\delta_{\sigma',\uparrow})$. Ψ_{AGP} reduces to a Slater determinant with a particular choice of the pairing function [21], implying therefore a description of the electron correlation better than those based on any Jastrow-Slater ansatz [22]. The variational freedom contained in the Ψ_{JAGP} ground state naturally permits a quantitative distinction between the spin and charge correlations [23]. Parallel reference DFT calculations were also performed with HSE6 exchange-correlation functional, projector augmentedwave treatment of core levels [24] and a plane-wave basis set [25] as implemented in the Vienna *ab initio* simulation package (VASP) [25,26], with energy cutoff of 600 eV.

All calculations [27] were conducted with $N_{\text{atom}} = 24$ carbon atoms forming four six-atom unit cells of a planar deformable honeycomb lattice whose average interatomic



FIG. 2. (a) Ground state energy *E* relative to the SEM phase E_{SEM} obtained as a function of strain ϵ by DMC in comparison with DFT for the DIM, AFI, HEX phases. (b) Stress (σ)-strain (ϵ) equation of state curve for strained graphene obtained by fitting DMC energies. Dashed lines mark the transition stress values σ_1 and σ_u for SEM-DIM (continuous). (c) Enthalpy *H* of strained graphene relative to that of the SEM phase H_{SEM} for increasing tensile stress σ . The blue-shaded region indicates the error bars on the enthalpies for DIM and AFI phases by DMC. Upper bounds of Eq. (2) for the DIM and AFI enthalpies also shown (DIM UB and AFI UB) greatly reduce the error bars. The corresponding strain ϵ at selected points and phases (indicated by arrows) are also shown.

spacing *a* was successively expanded relative to the zerostress value a_0 . A fully accurate *k*-point average is obtained by boundary-condition twisting.

Figure 2(a) presents the total energy gain of all ordered or distorted states relative to the undistorted, semimetallic, nonmagnetic SEM state, $E - E_{\text{SEM}}$, as a function of isotropic strain $\epsilon = (a - a_0)/a_0$, from both diffusion Monte Carlo (DMC) and DFT calculations. Figure 2(b) shows the DMCcalculated tensile stress, yielding the 2D equations of state of expanded graphene. In DFT, the AFI state [Fig. 1(b)] yields the lowest energy above $\epsilon \approx 7\%$, and represents the ground state until $\epsilon \approx 15\%$. Near 15% strain, DFT energetics predicts a Kekulé DIM state [Fig. 1(c)] to take over very briefly from AFI, just before turning itself unstable and leading to mechanical failure, in agreement with earlier DFT phonon calculations [17].

The more accurate DMC result shows instead that, while both DIM and AFI states appear around $\epsilon \approx 10\%$, DIM has the lowest energy for all increasing strains until failure. Accurate DMC therefore suggests that the charge instability is dominant over the spin, which is just the opposite of what the reference DFT calculation suggested. In line with that, the prevalence of DIM over AFI is reduced in the less accurate variational Monte Carlo calculations [27].

In addition, the lowest energy will not predict the experimental phase diagram, where isotropic strain ϵ is obtained by tensile stress σ . The equilibrium state under stress, rather than energy, minimizes the enthalpy $H(\sigma) = \min_S[E(S) - \sigma S]$, where $\sigma = \partial_S E(S)$ with *S* the mean area. The stress-area term makes in principle all negative-stress states metastable, as an infinitely large enthalpy gain can always be obtained by breaking the lattice apart. A metastable stretched state of graphene is, nonetheless, protected against failure by a large barrier, connected with the positive slope of the area-stress curve—the bulk modulus. A change of sign of that slope signals the vanishing of the barrier, ushering in mechanical failure.

In Fig. 2(b) the maximum strain is $\epsilon_{\text{max}} \sim 15\%$ for the DIM phase, actually close to that obtained in Ref. [17] by arbitrarily ignoring spin. Interestingly, this stability limit of the DIM phase coincides [Fig. 2(a)] with the prevalence within DFT of a HEX phase of Fig. 1(c), an artificial state that foreshadows, as it were, the real mechanical failure in a six-atom cell. The structurally undistorted AFI and SEM phases have higher enthalpies and are ruled out at high stress [Fig. 2(b)] despite their mechanical resilience, until about 20% strain. The QMC-calculated enthalpy of strained graphene, our main result, is shown as a function of isotropic tensile stress in Fig. 2(c). Obtained by evaluating the stress with polynomial interpolation, the result is affected by a large statistical error (shaded region), mostly due to the large uncertainty of the stress obtained by fitting energy-area curves. With luck, however, we reduced this error by means of a rigorous upper bound, which is obeyed by the enthalpy difference of any given phase from the symmetric phase

$$H(\sigma) - H_{\text{SEM}}(\sigma) \le E(S) - E_{\text{SEM}}(S), \tag{2}$$

where *S* is the area corresponding to the stress σ in the symmetric phase. The upper bound is practically coincident with the mean value, totally eliminating the error. The ground-state phase diagram predicted by minimum enthalpy, Fig. 2(c), shows that the SEM state for $\sigma < \sigma_1 = 25.1 \text{ N/m}$ ($\epsilon_1 = 8.5\%$) is followed by a DIM distorted state



FIG. 3. (a) Graphene DIM (left axis) and AFI (right axis) order parameters as a function of strain ϵ . The purple shaded area in the vertical lines indicates a DIM-AFI coexistence region. The Heisenberg model limit is $\simeq 0.54 \ \mu_{Bohr}$ [41]. (b) Correlation energy gain, measured by the energy per atom difference between the single determinant ansatz (Jastrow-Slater determinant wave function) and the corresponding multideterminant JAGP wave function. The largest energy gain occurs in the DIM state, underlining its resonating valence-bond nature, actually increasing for large strain ϵ . Small negative values at small strain are finite-size effects. Inset: finite-size scaling of this correlation energy gain in the DIM state at $\epsilon = 15\%$.

for $\sigma_1 < \sigma < \sigma_u = 30.4$ N/m ($\epsilon_u = 15\%$) where stability of the DIM phase is lost, and mechanical failure ensues. Even though metastable AFI and SEM phases still persist up to 20% strain, their realization should imply an unphysical enthalpy rise. One may therefore speculate that the difference between our calculated mechanical failure point, and that extracted from indentation ($\sigma = 40-50$ N/m, $\epsilon = 22.5\%$) should be attributed to the absence of realistic indentation details in our total uniform idealized description.

We can finally characterize and understand the DIM state, between 8.5% and 15% strain. The dimerizing distortion order parameter of Fig. 3(a), defined as the difference between large and small bond lengths, has the Peierls-Kekulé symmetry of Fig. 1(c) and appears to set in continuously, reaching ~0.17 Å near the DIM stability limit $\epsilon_1 = 15\%$.

The above QMC results for ground state properties of isotropically stressed graphene raise important physical questions. First, how and why do correlations stabilize the DIM phase instead of the AFI preferred by DFT between $\sim 10\%$ and 15% strain? Second, what is the electronic gap

of the insulating DIM phase of the strained graphene? Third, is the DIM insulator topologically trivial or nontrivial and what consequences does the answer entrain?

To the first point, the multideterminantal character of our variational ansatz of Eq. (1), originally a paradigm for the resonating valence bond (RVB) state [42,43], is crucial for the enhanced stability of the correlated DIM state. The Jastrow factor \mathcal{J} partly projects out from the determinantal part Ψ_{AGP} the single C–C molecular orbital (Mulliken) electron pair term, which is largest in unstressed graphene but energetically penalized by electron-electron repulsion under stress. That favors the two-determinant C-C valence bond (Heitler-London) term. All goes qualitatively as in the textbook two-electron problem of strained H₂ molecule. A black bond in Fig. 1(c), with obvious notations, is the entangled combination of the two Slater determinants $c_{A\uparrow}^{\dagger}c_{B\downarrow}^{\dagger}|0\rangle$ and $c_{A\downarrow}^{\dagger}c_{B\uparrow}^{\dagger}|0\rangle$ between A and B, with zero double occupancies. By contrast, the uncorrelated Peierls molecular orbital wave function $(c^{\dagger}_{A\uparrow} + c^{\dagger}_{B\uparrow})(c^{\dagger}_{A\downarrow} + c^{\dagger}_{B\downarrow})|0\rangle$ involves a larger double occupancy for both sites, and a bad electron-electron repulsion. This many-electron entangled wave function of the correlated DIM phase contains a Jastrow factor, acting on an exponentially large number of Slater determinants $2^{N_s/2}$, that appropriately penalizes the atomic configurations, where N_s is the total number of singlet bonds.

To gauge the correlation energy gain permitted by our ansatz, we show in Fig. 3(b) the correlation energy obtained by the multideterminant Ψ_{AGP} over a single determinant, still with the Jastrow factor. This difference is obtained by projecting the pairing function f, for each twist used, to the optimal f_P obtained by restricting to the best single determinant, calculated from the orthogonal eigenfunctions [44] ϕ_i associated to the original pairing function f [i.e., $\int dr'^3 f(\mathbf{r}, \mathbf{r}') \phi_i(\mathbf{r}') = \lambda_i \phi_i(\mathbf{r})$, where λ_i are the corresponding eigenvalues] as $f_P(\mathbf{r}, \mathbf{r}') = \sum_{i=1}^{N/2} \lambda_i \phi_i(\mathbf{r}) \phi_i(\mathbf{r}')$ with the largest $|\lambda_i|$ [45]. Since N electrons exhaust the occupation of the N/2 one particle orbitals ϕ_i , f_P describes the corresponding Slater determinant possessing maximum weight $\prod_i |\lambda_i|$ in the multideterminant expansion of the AGP, as described in Ref. [45]. The small energy excess of this simpler wave function and the full JAGP, computed by correlated sampling, measures the multideterminant "RVB" correlation energy gain. As shown in Fig. 3(b), this correlation energy gain is negligible in both perfect honeycomb structures, i.e., the poorly strained SEM and the largely strained AFI phases. Conversely, it becomes extensive (see inset) and growing with order parameter in the DIM phase, which therefore becomes stabilized, rather than the loser as in DFT. Stabilization of the DIM phase can be attributed to superexchange energy that is poorly treated within DFT.

To the second point, the electronic gap and the difference between charge and spin gaps is not directly obtainable by a



FIG. 4. Tight-binding band structures of systems with zigzag interfaces between (a) DIM and HEX phases and (b) DIM and AFI phases. The vertical axis is the single-particle excitation energy relative to the Fermi level and the horizontal axis the momentum parallel to the interface. The parameters (t_A, t_B, t_C) are assumed to be (0.95t, 0.95t, 1.05t) for DIM, (1.05t, 1.05t)(0.95t) for HEX, and (t, t, t) for AFI. The hopping integrals connecting the two phases are set equal to t. The magnitude of the gap in the AFI phase is set equal to that of the DIM phase, as indicated by the shaded area in (a) and (b). The intragap interface modes are highlighted with thick magenta lines in (a). Schematic figures of the calculated interface between (c) the DIM and HEX phases and (d) the DIM and AFI phases. The dotted-vertical line in (c) and (d) indicates the zigzag interface, and the shaded area the unit cell. The size of the unit cell is determined by the widths W_1 and W_2 of the two phases, indicated in the bottom of (d), and the vertical length d. The calculations in (a) and (b) are for $W_1 = W_2 = 50$. The system as a whole lies on a torus.

QMC ground state calculation, but we get an order of magnitude from DFT, where the DIM electronic Kohn-Sham gap grows from zero at 8.5% strain to about 1.1 eV at 15% [27].

To the third point, we note that adiabatic continuity between the strongly correlated DIM state and the uncorrelated Kekulé state discussed in literature [18,46,47] implies the DIM insulating state of strained graphene is topologically nontrivial, unlike the AFI or HEX states. With reference to Fig. 1 (one-electron tight-binding is sufficient for this purpose), the bond dimerization of the DIM phase corresponds to $|t_C| > |t_A| = |t_B|$, while the HEX phase to $|t_C| < |t_A| = |t_B|$. The nontrivial nature of the DIM phase is protected by the sublattice (chiral) symmetry and the mirror symmetry along a bond [46]. While this fact has no special consequences in infinite perfect 2D graphene strained into the DIM phase, it will, as in other topological insulators [48], show up at interfaces and defects, which can support a topological state energetically placed inside the dimerization gap. As a demonstration of that, we present a model tightbinding DIM-HEX two-phase coexistence with the zigzag interface under periodic-boundary conditions [Fig. 4(c)]. Its electronic structure in Fig. 4(a) shows topological states, with their characteristic gapless modes crossing the Fermi level, localized at the two DIM-HEX interfaces. This is in contrast to a model DIM-AFI interface [Fig. 4(d)] where no

gapless interface states appear [Fig. 4(b)]. This difference is simply understood because the DIM-HEX system preserves the two symmetries described above but the DIM-AFI system does not. Although the bulk single-particle gap increases with the electron correlations, these topological features remain qualitatively the same because of adiabatic continuity.

The impact of increasing electron correlations in isotropically and uniformly strained graphene, calculated by QMC simulations with an accurate variational wave function, is in summary predicted to be nontrivial. The phase diagram dictated by minimizing enthalpy under increasing stress predicts the sequence: SEM-DIMfailure, different from the best spin-polarized densityfunctional predictions. Large electron correlations stabilize the DIM phase, schematized in Fig. 1(c), in the (8.5–15)% tensile strain range corresponding to the (25-31) N/m stress range. Roughly speaking, dimerization freezes Pauling's resonating valence bond, a state which fluctuates in the honeycomb spin-liquid state as described, e.g., by Ref. [9], into a valence-bond solid, realized by a Kekulé-like phase that breaks translation invariance. Remarkably this effect was very recently observed in a lattice model of bilayer graphene [49]. The DIM phase possesses a stress-dependent order parameter and a correspondingly increasing electronic gap. In correspondence with the predicted continuous SEM-DIM transition the mechanical impedance of graphene should exhibit a dissipation singularity. Electronically, the graphene DIM insulator is topological, implying protected intragap states localized around defects with peculiar symmetry properties, including topological 1D Dirac states at grain boundaries and dislocations. Our predicted 15% failure strain is somewhat smaller than the 22.5% reported by experimental indentation studies, possibly due to the role of nonuniformities in indentation mechanics, absent in our so far totally uniform calculations. The onset of the DIM structural deformation and of an electronic gap which DFT estimates in the order of about one eV at failure, as well as of topologically related defect states in this gap could be used in the future to detect spectroscopically this novel state of strained graphene.

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