Theory of Thermal Relaxation of Electrons in Semiconductors

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We compute the transient dynamics of phonons in contact with high energy "hot" charge carriers in 12 polar and nonpolar semiconductors, using a first-principles Boltzmann transport framework. For most materials, we find that the decay in electronic temperature departs significantly from a single-exponential model at times ranging from 1 to 15 ps after electronic excitation, a phenomenon concomitant with the appearance of nonthermal vibrational modes. We demonstrate that these effects result from slow thermalization within the phonon subsystem, caused by the large heterogeneity in the time scales of electron-phonon and phononphonon interactions in these materials. We propose a generalized two-temperature model accounting for phonon thermalization as a limiting step of electron-phonon thermalization, which captures the full thermal relaxation of hot electrons and holes in semiconductors. A direct consequence of our findings is that, for semiconductors, information about the spectral distribution of electron-phonon and phononphonon coupling can be extracted from the multiexponential behavior of the electronic temperature.

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Following the seminal works of Kaganov *et al.* [1] and Allen [2], the thermalization of a system of highly energetic charge carriers with a lattice is frequently understood as an electron-phonon mediated, temperature equilibration process with a single characteristic time scale τ_{el-ph} . Such a description, referred to as the two-temperature (2T) model, relies on the central assumption that both electrons and phonons remain in distinct thermal equilibria and can therefore be described by time-dependent temperatures $T_{\rm el}(t)$ and $T_{\rm ph}(t)$ during the thermal equilibration process. In metals, due to the relative homogeneity of the electronphonon interactions and the rates of thermalization within the electronic and phononic subsystems, the hypothesis of subsystem-wide thermal equilibrium is generally accurate, and the 2T model has been successful in modeling ultrafast laser heating [3–5], despite some notable deviations from the 2T predictions in graphene and aluminum [6–8]. In semiconductors, the highly heterogeneous electron-phonon interactions (e.g., in polar semiconductors with Fröhlich interactions [9]) and, in some cases, the higher lattice thermal conductivity in comparison to metals weaken the hypothesis of a thermalized phononic subsystem [10,11], hence calling for the reexamination of the 2T physical picture in semiconductors.

In this context, the advent of first-principles techniques able to predict the mode- and energy-resolved electronphonon [12–14] and phonon-phonon interactions [15,16] provides an important opportunity: In their modern implementations [13,16,17], these methods have been able to predict lattice thermal conductivities [18–21], the temperature and pressure dependence of the electronic band gap [22–28], electrical conductivities [29,30], and hot carrier dynamics [31,32]. However, to the best of our knowledge and despite these early successes, these approaches have yet to be applied to the computation of electron-induced, nonequilibrium phonon distributions and their effects on thermal relaxation of electrons.

In this work, we combine first-principles calculations of electron-phonon and third-order phonon-phonon interactions within the semiclassical Boltzmann transport equation (BTE) for predicting the joint time evolution of electron and phonon populations after hot carrier excitation. For 12 polar and nonpolar cubic semiconductors, we show that the resulting phonon and electron dynamics departs qualitatively from the 2T physical picture over time scales of 1–15 ps after excitation. We demonstrate that this disagreement stems from the breakdown of the hypothesis of thermal equilibrium within the lattice subsystem, caused by the wide range of time scales associated with electronphonon and phonon-phonon interactions in these systems. We generalize the 2T model of Allen to account for slow phonon thermalization as a limiting step of electron-phonon thermalization, show that our generalized 2T model captures the transient dynamics for all compounds, and discuss its implication for time-resolved spectroscopy experiments. We anticipate our findings to apply to any material with broad spectral distributions of electron-phonon interactions (including, but not limited to polar materials) and weak phononphonon interactions (in comparison to bulk metals).

We compute the mode- and time-resolved phonon occupation function $n_{\mathbf{q},\nu}(t)$ in the presence of an electron occupation function $f_{n\mathbf{k}}(t)$ solving a coupled system of equations parametrized using density functional theory (DFT)-based approaches. The time evolution of the phonon occupation function is obtained by solving the BTE: $dn_{\mathbf{q}\nu}(t)/dt = [\partial n_{\mathbf{q}\nu}(t)/\partial t]_{ep}[n_{\mathbf{q}\nu}(t), f_{n\mathbf{k}}(t)] + [\partial n_{\mathbf{q}\nu}(t)/\partial t]_{pp}[n_{\mathbf{q}\nu}(t)]$, where the drift term has been neglected due to the lack of spatial temperature gradient,

and [...] indicates the functional dependence. The two terms on the right denote the time-dependent scattering potentials due to electron-phonon (EPI) and phononphonon interactions (PPI), both computed using firstprinciples methods, as detailed below. Importantly, we make the assumption that charge carriers are in thermal equilibrium and that $f_{n\mathbf{k}}(t)$ can be approximated by a timedependent Fermi-Dirac function centered near the top of the valence band for holes and near the bottom of the conduction band for electrons at the temperature $T_{\rm el}(t)$. Depending on the material and the nature of charge carriers, the time scale of the phonon-mediated carrier thermalization to the band edges was found to range from 0.1 to 1 ps [31,33], which also corresponds to limits of validity of the semiclassical description. Hence, we expect our simulation method and the approximation of $f_{n\mathbf{k}}(t)$ to be quantitative at subsequent times.

Specifically, we define the EPI scattering potential as an explicit functional of the phonon and electron occupation functions at time t, and compute it using Fermi's golden rule: $[\partial n_{\mathbf{q}\nu}(t)/\partial t]_{ep} = (4\pi/\hbar)\sum_{\mathbf{k},m,n} |g_{\mathbf{q}\nu}(m\mathbf{k}+\mathbf{q}, t)|^2$ $(n\mathbf{k})|^2 \mathcal{M}_{mn\nu\mathbf{k}\mathbf{q}}(t)$, in which $|g_{\mathbf{q}\nu}(m\mathbf{k}+\mathbf{q},n\mathbf{k})|$ is the timeindependent electron-phonon matrix element involving electronic states $|n\mathbf{k}\rangle$ and $|m\mathbf{k} + \mathbf{q}\rangle$ and vibrational state $|\mathbf{q}\nu\rangle$ evaluated using Wannier interpolation with the EPW code [13]. $\mathcal{M}_{mn\nu \mathbf{k}\mathbf{q}}(t)$ is the time-dependent joint density of states computed from $n_{\mathbf{q}\nu}(t)$, $f_{n\mathbf{k}}(t)$, $f_{m\mathbf{k}+\mathbf{q}}(t)$, and the electron and phonon spectral densities (detailed formulas are given in Supplemental Material [34]). Similarly, we evaluate the scattering caused by PPI $[\partial n_{\mathbf{q}\nu}(t)/\partial t]_{pp}$ from Fermi's golden rule, using the time-independent 3-phonon scattering matrix elements $|\Psi_{\mathbf{qq'q\pm q'+G}}^{\nu\nu'\nu''}|^2$ computed with DFT [15] and the time-dependent density of final states computed from $n_{\mathbf{q},\nu}(t)$, $n_{\mathbf{q}',\nu'}(t)$, $n_{\mathbf{q}\pm\mathbf{q}',\nu''}(t)$. At each time step, the net energy transfer Q_{ep} between electrons and phonons is computed and a new electronic temperature is derived as $T_{\rm el}(t + \Delta t) = T_{\rm el}(t) - Q_{ep}(t)/C_{\rm el}(T_{\rm el})$ where $C_{\rm el}(T_{\rm el})$ is the instantaneous electronic heat capacity at temperature $T_{\rm el}$. The BTE is solved for 48000 phonon modes using an explicit time-stepping scheme with a time step of 0.5 fs and a total simulation time of 25 ps for 12 cubic semiconducting compounds (BN, BP, BAs, BSb, AlP, AlAs, AlSb, GaN, GaP, GaAs, diamond, and Si). All the simulations discussed below were initialized with a Bose-Einstein distribution of phonons at 300 K and a Fermi-Dirac distribution of electrons at 3000 K with the Fermi level set at 0.3 eV below the valence band maximum (other choices of initial temperatures and Fermi energies are shown to lead to similar conclusions in Supplemental Material [34]).

The electronic structure was computed with DFT in the local density approximation, using norm-conserving pseudopotentials, a $10 \times 10 \times 10$ k grid and the Quantum Espresso package [38]. The phonon dispersion was

computed using density functional perturbation theory [39] and a $5 \times 5 \times 5$ **q** grid. Third-order force constants were computed in real space using finite differences on a $6 \times 6 \times 6$ supercell [40] and Fourier transformed to obtain phonon-phonon interaction matrix elements $|\Psi_{qq'q\pm q'+G}^{\nu\nu'\nu''}|^2$ on a $20 \times 20 \times 20$ **q** grid [41]. The present approach neglects the temperature dependence of the third-order force constants [42,43], as this simplification has been shown to accurately predict the temperature-dependent lattice thermal conductivity for cubic semiconductors [19,41,44,45]. Electron-phonon interactions were evaluated on $20 \times 20 \times 20$ and $40 \times 40 \times 40$ grids for phonons and electrons, respectively. Convergence studies are provided in Supplemental Material [34].

In Fig. 1, we show the time-dependent phonon occupations along high symmetry directions of the Brillouin zone for BN and BAs (snapshots for all materials can be found in Supplemental Material [34]). At short times t < 1 ps, the electronic energy is transferred to long-wavelength optical phonons, an effect originating from the larger electronphonon scattering phase space associated with lowmomentum phonons near the top (bottom) of the valence (conduction) bands in all 12 compounds, and further magnified by the 1/q divergence in the Fröhlich coupling in polar semiconductors [9,46]. Accordingly, we observe that more energy is transferred to the LO and TO modes of BN than to the modes of BAs, as expected from the Born effective charges (1.86 for BN vs 0.56 for BAs) and polarity, which also lead to a larger electron-phonon coupling and LO-TO splitting. Surprisingly, these hot phonon modes are found for all compounds to achieve near-thermal equilibrium with the electrons rather than with the rest of the phonons, a strong departure from the hypothesis of thermal equilibrium among phonons. At longer times, 1 < t < 10 ps, long-wavelength LO and TO phonons in BN remain in near-thermal equilibrium with electrons, while transferring their energy to acoustic modes via 3-phonon processes, through Klemens (decay to two acoustic phonons) [47] and Ridley (decay to one optical and one acoustic phonons) mechanisms [48]. The "hot phonon" cooling in BAs is slower in comparison to BN as the large acoustic-optical phonon band gap [19,49] (originating from the mass mismatch) truncates the Klemens scattering phase space, while the Ridley decay is reduced by the small LO-TO splitting. Near thermalization within the phonon subsystem (and, concomitantly, between electrons and phonons) is achieved in BN at $t \simeq 25$ ps, with an electronic temperature 50 K away from the average lattice temperature $T_{\rm ph} = 380$ K. In stark contrast to BN, for BAs both electrons (602 K) and hot phonons remain in near equilibrium with each other, but far from the average lattice temperature $T_{\rm ph} = 344$ K. Importantly, the same two trends are observed for all simulated materials: (1) electrons first achieve near thermal equilibrium with a small number of high energy phonon



FIG. 1. Temperature maps of phonon modes in cubic boron nitride (BN, top row) and boron arsenide (BAs, bottom row) along the $L-\Gamma-X$ directions as a function of time starting from a hot equilibrium electron distribution at 3000 K. In each panel, the phonon modes with the highest temperature are observed to be nearly in equilibrium with electrons (temperature color bar is saturated for T > 1000 K).

modes; (2) full electron-lattice thermalization and intraphonon thermalization are always achieved simultaneously; both trends imply that electron cooling is limited by thermalization within the phonon subsystem.

Further illustrating the nonequilibrium among phonon modes, we see in Fig. 2 that the agreement between the BTE simulation and a 2*T* model parametrized from first principles (see Supplemental Material [34] for details) is good at times t < 0.05 ps and t < 0.2 ps for BN and BAs, respectively, but quickly deteriorates afterwards. For all compounds, the 2*T* model predicts a thermalization process that is at least an order of magnitude faster than that observed in the full BTE simulation, proving that electronic cooling becomes limited by another mechanism, not accounted for in the 2*T* model [50].

To test our hypothesis of a phonon thermalization limited process, we perform a constrained simulation of electron cooling in which the phonons are partitioned into multiple subspaces defined by the strength $\eta_{q\nu}$ of their interactions with electrons and phonons,

$$\eta_{\mathbf{q}\nu} = \hbar \omega_{\mathbf{q}\nu} \left(\sum_{\mathbf{k},m,n} |g_{\mathbf{q}\nu}(m\mathbf{k} + \mathbf{q}, n\mathbf{k})|^2 \delta_{\mathbf{q}\nu,mn\mathbf{k}} + \sum_{\mathbf{q}'\nu',\mathbf{q}''\nu''}^{\eta_{\mathbf{q}\nu'} > \eta_{\mathbf{q}\nu}} |\Psi_{\mathbf{q}\mathbf{q}'\mathbf{q}\pm\mathbf{q}'+\mathbf{G}}^{\nu\nu'\nu''}|^2 \delta_{\mathbf{q}\nu,\mathbf{q}'\nu',\mathbf{q}''\nu''} \right),$$
(1)

where the terms on the right approximate the scattering due to EPI and PPI for each mode $|\mathbf{q}\nu\rangle$ (the PPI term only includes modes with a larger interaction strength and is computed self-consistently). $\delta_{\mathbf{q}\nu,mn\mathbf{k}}$ and $\delta_{\mathbf{q}\nu,\mathbf{q}'\nu',\mathbf{q}''\nu'}$ are energy conservation delta functions for electron-phonon and phonon-phonon scattering, respectively. The phonon modes in the largest $\eta_{\mathbf{q}\nu}$ subspace [51] are a small subset [see Figs. 2(c) and 2(d)] of the total number of phonons and primarily consist of long-wavelength optical phonons with strong electron-phonon interaction [34]. At time t = 0, only the modes belonging to the subspace with the largest $\eta_{q\nu}$ are allowed to interact with electrons until thermalization. Upon thermalization of the first subspace, the next



FIG. 2. Electronic and lattice temperatures in BN (a) and BAs (b) obtained from the 2*T* model, BTE, and a constrained "successive thermalization" (ST) simulation using the 2*T* model on a subset of phonons (see the main text). The red squares indicate the times (and corresponding equilibration temperatures) at which subspace thermalization is achieved and a new set of modes is introduced in the ST simulation. Histograms of the distribution of interaction strength $\eta_{q\nu}$ [Eq. (1)] for BN (c) and BAs (d) show the partition scheme and the phonons included in the ST simulation at t = 1 ps. Phonon modes I are the phonons thermalization, and III are the phonons noninteracting with electrons.

subspace is introduced in the simulation along with the thermalized system of electrons and the first subspace. This constrained ST process continues until all modes are included [52]. As shown in Fig. 2 and Supplemental Material [34], this constrained ST simulation achieves quantitative agreement at all times for all materials considered, validating our central finding: Electron cooling in semiconductors is limited by intraphonon thermalization, a direct consequence of the orders-of-magnitude heterogeneities in the mode-dependent electron-phonon interactions and slow phonon thermalization.

We conclude this work by proposing a simple generalization of the 2T model based on these findings, and discuss its consequences in interpreting material properties measured by the time-resolved decay of the electronic temperature [53,54]. We start by partitioning the system in an analogous way to our ST simulation, i.e., with three subsystems well described by a subsystem-wide temperature: (I) a system containing electrons and phonons fully thermalized with each other [by definition, $T_{I}(t) = T_{el}(t)$], (II) phonon modes in contact with electrons and/or phonons of system (I) (in the process of being thermalized) at temperature $T_{\rm ph}(t=0) < T_{\rm II}(t) < T_{\rm el}(t)$, and (III) "cold" phonons not in contact with (I) $[T_{III}(t) = T_{ph}(t=0)]$. In this generalized partition scheme, the 2T model is strictly recovered by setting $(I) = \{el\}, (II) = \{ph\}, and$ $(III) = \emptyset$. As more modes become thermalized with electrons as a function of time, the long-time electronphonon thermalization can be understood as system (I) absorbing systems (II) and (III). Hence, the heat capacity of (I) becomes time dependent with $C_{I}(t)$ increasing from $C_{\rm I}(t=0) = C_{\rm el}$ to $C_{\rm I}(t \to \infty) = C_{\rm el} + C_{\rm ph}$ (temperature dependences were omitted for simplicity of notation). Such time-dependent heat capacity $C_{I}(t)$ can be understood as an accumulation function of the phonons over the time scales of their interactions. $C_{I}(t)$ can be computed heuristically by defining an effective mode-dependent thermalization time $t_{\mathbf{q}\nu}$ (that we set to the relaxation time) and $C_{\mathrm{I}}(t) =$ $\sum_{\mathbf{q}\nu} C_{\mathbf{q}\nu} \Theta(t - t_{\mathbf{q}\nu})$ where $\Theta(t)$ is the Heaviside function. Similarly, the electron-phonon coupling accumulation can be defined as $G_{ep}(t) = \sum_{\mathbf{q}\nu} G_{ep,\mathbf{q}\nu} \Theta(t - t_{\mathbf{q}\nu})$ (see Supplemental Material [34] for definitions of $t_{q\nu}$, $C_{q\nu}$, $G_{ep,\mathbf{q}\nu}$ and their values for all compounds). Noteworthily, for materials with large heterogeneities in their modedependent electron-phonon coupling $G_{ep,q\nu}$, these two accumulation functions have very different time dependences: as seen in Figs. 3(b) and 3(d), $G_{ep}(t)$ reaches 50% of its total value 1–1000 times faster than $C_{I}(t)$.

At short times [comparable to the time of accumulation of $G_{ep}(t)$], the observed electronic temperature decay rate given by this generalized 2T model can be approximated by $G_{I-II}/C_I \simeq G_{ep}(t \to \infty)/C_{el}$, i.e., the decay rate predicted by a "standard" 2T model. Correspondingly, in Fig. 3(c), we observe an excellent correlation between the initial



FIG. 3. (a) Electronic temperature decay in BAs along with the decay rates at short (α) and long (β) time instants. (b) Accumulation of lattice heat capacity $C_{\rm I}(t)$ at $T_{\rm ph} = 300$ K and electron-phonon coupling $G_{ep}(t)$ at $T_{\rm el} = 3000$ K and $T_{\rm ph} = 300$ K, as a function of phonon thermalization time scale in BAs. (c) Comparison of electron-phonon coupling coefficient obtained from the decay rate of electronic temperature at $t \rightarrow 0$ and directly from DFT for all 12 semiconductors considered in this work. (d) Comparison of the ratio between initial and long-time decay rates with the ratio of time scales of heat capacity and electron-phonon coupling accumulation for all compounds considered in this work.

decay rate and the electron-phonon coupling strength predicted directly from first principles for all compounds considered in this work—indicating that, at short time, the determination of the single-exponential decay of the electronic temperature yields the total electron-phonon coupling. At longer times, the decay rate of the electronic temperature G_{I-II}/C_I is reduced by the accumulation of heat capacity in I, as $C_{\rm I} \gg C_{\rm el}$. As shown in Fig. 3(d), the reduction of the decay rate for all compounds shows a good correlation with the disparity of time scales between $C_{I}(t)$ and $G_{ep}(t)$, suggesting that a measurement of the electronic temperature decay across time scales in semiconductors would yield both the total electron-phonon coupling coefficient and information about the distribution of phonon interaction strength (and its heterogeneity) in a given material. Interestingly, as the phonon interaction strength involves both EPI and PPI [see Eq. (1)], the time dependence of the decay rates is particularly important for materials with very heterogeneous EPI (diamond, BN, and GaN), and large phonon band gaps (BAs and BSb), and vanishes for nearly homogeneous EPI (e.g., GaAs [32]).

In conclusion, we have demonstrated that electron cooling in semiconductors is limited by intraphonon thermalization at time scales on the order of 1-20 ps. We have proposed a generalized two-temperature model accounting for this effect, and shown that such a model can be used to extract information from the measurement of the electronic temperature about both the total electron-phonon

coupling and the distribution of electron-phonon and phonon-phonon interactions. More generally, we expect the phonon-limited thermalization identified in this work to have consequences for both heat and electron transport, fields in which long-lasting nonequilibrium phonon distributions have been shown to impact spectroscopic measurements [6,7], current-voltage characteristics [55], and hot electron lifetimes [56,57]. Specifically, our work offers a direct estimate of the time scales at which equilibrium models become quantitative in the presence of hot electrons (and their relationship to materials properties), and, via the tunability of the phonon interaction strength, new pathways to control the time scales of electronic energy dissipation.

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