

Correlation between Fragility and the Arrhenius Crossover Phenomenon in Metallic, Molecular, and Network Liquids

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We report the observation of a distinct correlation between the kinetic fragility index m and the reduced Arrhenius crossover temperature $\theta_A = T_A/T_g$ in various glass-forming liquids, identifying three distinguishable groups. In particular, for 11 glass-forming metallic liquids, we universally observe a crossover in the mean diffusion coefficient from high-temperature Arrhenius to low-temperature super-Arrhenius behavior at approximately $\theta_A \approx 2$ which is in the stable liquid phases. In contrast, for fragile molecular liquids, this crossover occurs at much lower $\theta_A \approx 1.4$ and usually in their supercooled states. The θ_A values for strong network liquids spans a wide range higher than 2. Intriguingly, the high-temperature activation barrier E_∞ is universally found to be $\sim 11k_B T_g$ and uncorrelated with the fragility or the reduced crossover temperature θ_A for metallic and molecular liquids. These observations provide a way to estimate the low-temperature glassy characteristics (T_g and m) from the high-temperature liquid quantities (E_∞ and θ_A).

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The fragility of a glass-forming liquid is a measure of how quickly its dynamics slows down upon cooling. It is usually quantified by the kinetic fragility index m , which is defined as the slope of the Angell plot of transport coefficients in logarithmic scale versus T_g/T evaluated at the glass transition temperature T_g [1]:

$$m = \left. \frac{\partial \log \eta(T)}{\partial (T_g/T)} \right|_{T=T_g}. \quad (1)$$

A liquid that undergoes little change in the slope as a function of temperature is called a kinetically “strong” system. Examples include many of the network liquids such as silica, soda lime glasses, etc. The other end of the spectrum, defined as “fragile,” corresponds to systems that show significant increases in slope with cooling. Examples include many of the van der Waals molecular liquids, polymers, ionic liquids, etc. Notably, many of the glass-forming metallic liquids that are mediated by complex many-body metallic interactions span the intermediate fragility range [2,3]. This fact has inspired us to systematically compare the nature of slow dynamics of these metallic liquids with other glass formers, namely, molecular liquids and network liquids. To date, an agreed quantitative understanding of fragility is still lacking [4,5].

The glass transition temperature T_g and the fragility index m are key parameters quantifying the low-temperature behavior of liquids. The Arrhenius crossover phenomenon occurring well above T_g quantifies the

high-temperature activated behavior of liquids [6–11] and has attracted much attention recently, especially in metallic liquids [12–14]. For instance, in many molecular liquids the Arrhenius crossover or glassy-dynamics-onset temperature T_A marks the deviation of the transport coefficients or the relaxation time from the high-temperature Arrhenius dependence as well as the deviation of the intermediate scattering function from a simple exponential relaxation. This crossover is believed to indicate increasingly dynamically heterogeneous and cooperative motion when temperature is lowered below T_A [15,16]. Above T_A , particles move relatively independently without the need for a collective reorganization of their respective local environment due to the large mobility and phonon localization [12]. However, when the temperature is lowered, collective reorganization of particles (local topological excitations [12], hopping [17], etc.) over increasing length scales is needed to facilitate large amplitude irreversible motions in cold dense media. Such cooperative motion allows the system to overcome large free energy barriers and relax, resulting in highly activated dynamics [10,18–23]. Note that it is T_A , and not T_g , that marks the onset of such cooperativity and dynamic heterogeneity. Similarly, in strong liquids, cooperativity and spatially heterogeneous dynamics have also been observed with emerging slow dynamics. However, the apparent Arrhenius behavior of transport properties remains largely unchanged down to low temperatures since large scale cooperative motions are less relevant due to the

rather uniform topography of the energy landscape and influence of strong covalent bonds [24,25]. There have been limited studies on the nature of the Arrhenius crossover in glass-forming metallic liquids until very recently [13,14]. Consequently, a systematic comparison between metallic liquids and molecular and network liquids regarding the Arrhenius crossover phenomenon and its relation with the fragility is still missing.

In this Letter, we explore the relation between the dynamic fragility and the Arrhenius crossover phenomenon in 11 metallic liquids, 56 molecular liquids, and 12 network liquids. Both T_g and m are low-temperature parameters, while the Arrhenius crossover characterized by E_∞ and T_A is a high-temperature phenomenon. It is not obvious that a connection between these two phenomena exists, nor it is clear how this transpires across the various classes of glass formers. To this end, we have observed a direct correlation between fragility and the Arrhenius crossover in all studied liquids. Three distinct regions in a fragility m versus the reduced Arrhenius crossover temperature θ_A plot have been established. Strong network liquids reveal the highest crossover temperature (relative to T_g) to cooperative dynamics followed by metallic liquids that have intermediate fragilities. Fragile molecular liquids are found to have the lowest θ_A that usually occurs in their supercooled states, unlike metallic liquids. Furthermore, we also find, intriguingly, that the high-temperature effective activation energies for transport E_∞ in metallic and molecular liquids are surprisingly similar, roughly $\sim 11k_B T_g$.

The onset of cooperative dynamics has been previously characterized in fragile liquids by studying the deviation of transport coefficients from their high-temperature Arrhenius behavior [9,10,13,26]. Herein, we compile diffusion coefficient data of 11 metallic liquids [27–38] measured using quasielastic neutron scattering (QENS) that measures the mean relaxation dynamics of multicomponent metallic liquids. Figure 1 shows that all of the 11 glass-forming metallic melts exhibit a clear deviation from their high-temperature Arrhenius behaviors. It should be noted that relaxation in such metallic liquids is slightly stretched even in the very high temperature liquid state due to differences between the mobilities of constituent elements [27]. Nevertheless, previous studies have shown T_A to be the same for all components [26]. In Fig. 1(a), we apply a straightforward scaling of the Arrhenius form of the diffusion coefficient:

$$\frac{D(T)}{D_0} = \exp\left(-\frac{E_\infty}{k_B T}\right) \Rightarrow \frac{\log(D(T)/D_0)}{\log(D(T_A)/D_0)} = \frac{T_A}{T}, \quad (2)$$

where D_0 is the diffusion constant and E_∞ is the high-temperature activation barrier. Both D_0 and E_∞ are obtained from fitting the high temperature data using the Arrhenius equation as shown on the left-hand side of Eq. (2). The diffusion coefficients of all 11 metallic liquids

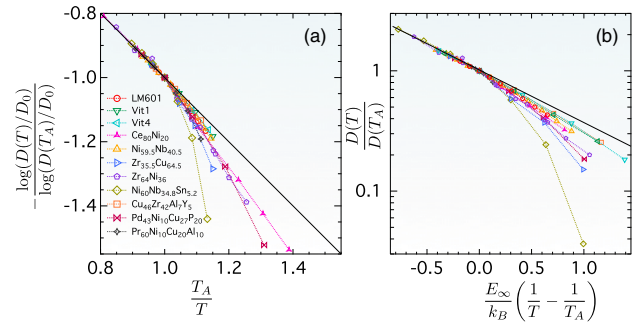


FIG. 1. Arrhenius crossover in 11 metallic liquids. All samples presented here were measured using QENS. (a) and (b) show two different scaling plots of the mean diffusion coefficients. Solid lines represent the Arrhenius law. Both plots clearly show a deviation from the high-temperature Arrhenius law below T_A .

collapse onto a single straight line with a slope of -1 above T_A of the respective sample. Deviations from the straight line are unambiguously observed below T_A for all the metallic liquids. An alternative scaling is also presented in Fig. 1(b) where D_0 is divided out, as shown in Eq. (3).

$$\frac{D(T)}{D(T_A)} = \exp\left[-\frac{E_\infty}{k_B} \left(\frac{1}{T} - \frac{1}{T_A}\right)\right] \quad (3)$$

This scaling also reveals consistent Arrhenius crossover behavior as described above. The time scale associated with this crossover is typically around 10–20 ps in metallic liquids, which is consistent with recent experimental observations [8,9], and a prediction of the elastically collective activated hopping theory, of a crossover time of about 10–100 ps for molecular liquids [10]. For metallic liquids, this time scale is estimated assuming a Fickian behavior $\tau_A \sim d^2/(6D(T_A))$, where d is the average particle diameter. τ_A is much larger than the boson peak time scale in metallic liquids that typically occurs at ~ 5 meV or ~ 0.13 ps. In fact, the Q -dependent relaxation time measured by QENS spans a range of 1–100 ps.

Below the crossover temperature T_A , several analytical forms have been developed to model the super-Arrhenius dependence of transport coefficients. We chose the parabolic formula as a convenient analysis tool with the assumption that the onset temperature of dynamical facilitation [39] is close to T_A and is identified as deviations from the parabolic form, although our analysis does not prove that the “onset temperature” is identical to T_A , neither does it suggest that the low temperature physics is facilitation given that different activated theories predict the similar formula [10]. The parabolic formula when applied to metallic liquids needs to be supplemented by an additional nonvanishing mean-field energy contribution:

$$\frac{D_0}{D(T)} = \exp\left[J^2 \left(\frac{1}{T} - \frac{1}{T_A}\right)^2 + \frac{E_\infty}{k_B} \left(\frac{1}{T} - \frac{1}{T_A}\right)\right], \quad (4)$$

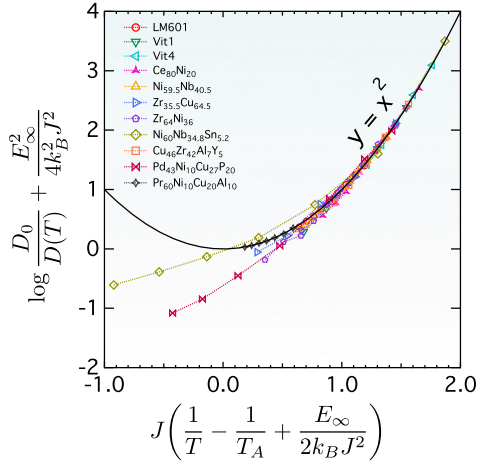


FIG. 2. Scaling of the low-temperature diffusion coefficient of 11 metallic liquids using the parabolic formula of Eq. (3). Collapse of low-temperature diffusion coefficients to a single master parabolic curve is evident. A clear deviation from the parabolic form is observed above T_A for a given metallic liquid.

where the activation energy E_∞ and the crossover temperature T_A are determined from the high-temperature Arrhenius fitting. A simple scaling (details in the Supplemental Material [40]) shows that the inverse diffusion coefficient converges towards the parabolic form at low temperatures, as shown in Fig. 2. The point at which the experimental data deviate from the $y = x^2$ curve (solid black line) is the crossover temperature of a given metallic liquid. Such a low-temperature scaling plot verifies the consistency in determining T_A .

In our recent simulations [26], we found that the Arrhenius crossover is associated with a sudden increase in the size of dynamical clusters of particles, of notably slow to intermediate mobility. This occurs at roughly the same temperature for all constituent elements in metallic liquids. Because of the presence of temporal clusters of varying mobility there is increasing heterogeneous dynamics, which was further validated by quantitative measures such as the non-Gaussian parameter and the four-point correlation functions. Below T_A , the Stokes-Einstein relation begins to break down corresponding to a decoupling of the diffusive and relaxation dynamics for all components. It should be noted that the Arrhenius crossover discussed in this Letter is not identical to other dynamical crossovers, such as the fragile-to-strong crossover, the empirically deduced mode-coupling crossover, the separation of α and the Johari-Goldstein β relaxation, and others, which typically occur at lower temperatures than T_A [70–76].

Once the crossover temperature is identified, we can examine E_∞ and T_A for a variety of liquids with diverse fragilities. The effective activation barrier E_∞ for the high-temperature Arrhenius diffusion process in all metallic liquids is found universally to be $\sim 11k_B T_g$, as shown in Fig. 3(a). The notable outliers are Vitreloy 4 (left facing

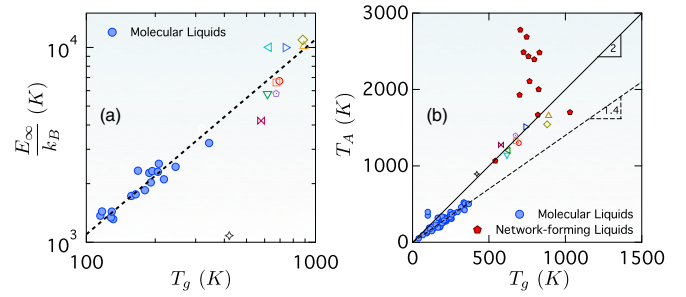


FIG. 3. Correlations of the high-temperature activation energy barrier E_∞ and the Arrhenius crossover temperature T_A with the glass transition temperature T_g . (a) In all metallic and molecular liquids, E_∞ is universally $\sim 11k_B T_g$, independent of their fragilities. While for strong network liquids, $E_\infty \approx E(T_g) = mk_B T_g$. (b) For 11 metallic liquids, $T_A \approx 2T_g$. For 56 molecular liquids, $T_A \approx 1.4T_g$. For 12 network liquids, there is no clear correlation of T_A with T_g , which is expected.

triangle) and the Pr-based glass former (diamond). In both cases, the reason for such variations in E_∞ is likely due to the very limited temperature range of the diffusion coefficient measurements. The activation barrier for the viscosity of Vitreloy 4 is 55 kJ/mol [77], which is very similar to that for other Cu-Zr based systems close to $11k_B T_g$. Interestingly, such an activation barrier of $E_\infty \sim 11k_B T_g$ is surprisingly similar to that of many van der Waals molecular liquids, as recently established experimentally [9,78,79] and also predicted by the microscopic elastically collective activated dynamics theory [10,40]. This behavior is independent of the fragility of these two classes of liquids, as discussed in the Supplemental Material [40]. Note that the hydrogen bonded systems and long chain polymers are characterized by a higher E_∞ [10,80]. For strong network liquids, $E_\infty \approx E(T_g) = mk_B T_g$ because of weak changes in the slope of transport properties in the Angell plot. Consequently, their reduced activation energy $E_\infty/k_B T_g$ is as large as their fragility index m , typically in the range of 20–30. For example, for many silicate and borosilicate based liquids, $E_\infty/k_B T_g$ is indeed very close to their reported fragility index m [81].

Perhaps the more interesting result is obtained by comparing T_A with T_g for the three classes of glass formers as shown in Fig. 3(b). Three distinct behaviors can be identified: (1) For metallic liquids, it is remarkable that all data points follow a straight line with a slope of 2. In our previous QENS experiments, we have made a similar observation of $T_A \approx 2T_g$ for diffusion in LM601 [13,27]. Here we find such a relation is universal in all metallic glass formers examined, which are composed of two to five elements. Furthermore, T_A is found to be higher than the melting temperature T_m , and thus the dynamic crossover occurs in the equilibrium liquid state. These observations of an Arrhenius crossover in the diffusion coefficient are in excellent agreement with recent results from studies of the shear viscosity in 27 glass-forming metallic liquids [13,14].

The T_A identified from deviation of the bulk viscosity from an Arrhenius behavior at high temperatures was also found to be $\sim 2T_g$. It should be noted that the crossover observed in macroscopic viscosity (associated with collective stresses) does not *a priori* imply that an Arrhenius crossover will occur in a microscopic diffusion process at the same place because metallic liquids have been found to violate the Stokes-Einstein relation even above the melting point and in the vicinity of T_A [29,33]. Our new results suggest a physical picture of the dynamic crossover in metallic melts in terms of the single particle self-diffusion coefficient, which is qualitatively consistent with reported results on viscosity. (2) For molecular liquids, the crossover temperatures T_A are typically found [82], and recently predicted theoretically [10], to be $1.4 \pm 0.2T_g$. As seen in Fig. 3(b), almost all the data points fall on the line with slope of 1.4. The two clear outliers represent two different data sets for propanol, a molecule that can form hydrogen bonds. Similarly, other hydrogen bonded systems such as glycerol and sorbitol also display higher relative crossover temperatures of $T_A \sim 1.55\text{--}1.75T_g$ [10]. (3) For network liquids, the crossover temperatures T_A span a very wide range and are typically higher than $2T_g$. This is because many of these systems are kinetically strong and thus do not show a distinct deviation from the high-temperature Arrhenius behavior. Consequently, there are large uncertainties in identifying the crossover temperature.

In Fig. 4, we plot the fragility index m versus the reduced crossover temperature $\theta_A = T_A/T_g$. One can readily see there are three distinct regions associated with the chemically different nature of the glass formers. At a rough qualitative level, m is found to be inversely proportional to θ_A . The highly fragile molecular liquids show a much lower θ_A ; metallic liquids have a higher θ_A , corresponding to their intermediate fragilities, while the network liquids have

very small fragilities but cover a wide range of θ_A . The separation into different groups is likely due to the proximity of the fragilities of liquids in each group. We anticipate other important classes of glass formers such as ionic liquids (m in the range of 50–100 [3], only CaKNO_3 shown here) and chalcogenides (m in the range of 30–70 [83] similar to metallic liquids, not discussed in this work) should likely fill in the gaps in between and around $\theta_A \sim 1.4$ and $\theta_A \sim 2$ in Fig. 4, respectively. In fact, a simple relation between m and θ_A can be established from the parabolic equation [20,39], which connects the onset behavior of glassy dynamics at T_A to the temperature dependence of transport properties. Evaluating Eq. (1) yields the following relation (details in the Supplemental Material [40]):

$$m = \frac{2J^2}{T_g} \left(\frac{1}{T_g} - \frac{1}{T_A} \right) = C_1 \frac{\theta_A}{\theta_A - 1} \quad (5)$$

where the prefactor $C_1 = \log(\tau(T_g)/\tau_0)$. We observe that this equation fits the reduced Arrhenius temperature dependence of the fragility admirably in Fig. 4(a). Since the parabolic form is valid only below the crossover temperature T_A , τ_0 cannot be set not equal to the typical inverse phonon frequency value of 10^{-14} s. From our fittings we obtain $C_1 \approx 18$. The prediction bounds narrow for molecular liquids whose fragility values have been rigorously established giving rise to an almost constant $\theta_A \approx 1.4T_g$. For metallic liquids, the range of fragility values is very narrow hence giving rise to an almost constant θ_A around 2. The large uncertainty associated with θ_A for network liquids yields a wide prediction bound for fragility values.

We speculate that the distinction of θ_A among metallic, molecular, and network liquids can be rationalized, at least partially, in terms of the degree of harmonicity and softness of the pair interaction potential. The interatomic potential of metallic liquids is influenced by Friedel oscillations, and is harmonic and thus soft near its minimum. These features of the potential lead to low fragility [84], similar to tunably soft cross-linked microgel colloids where both experiment [85] and theory [86] find softer repulsions correspond to lower dynamic fragility. For the same reason, the characteristic local atomic structure and connectivity persist at high temperatures, giving rise to a higher reduced crossover temperature θ_A in metallic liquids than typical van der Waals liquids, even higher than some hydrogen bonded molecular liquids such as glycerol and sorbitol. The network liquids are characterized by strong and directional covalent bonds [87]. The latter feature implies relaxation can be achieved by spatially local “bond-breaking” events, with a well-defined activation energy, which results in very low fragilities and very high θ_A . On the other end of the spectrum is molecular liquids which have strongly anharmonic interactions characterized by steep short-range repulsions. Therefore, their packing structures respond more sensitively to changes of density and temperature (a more “fragile structure”), and collective molecular

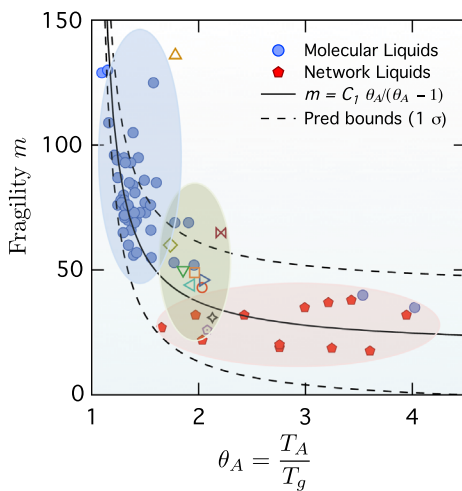


FIG. 4. Correlations of the fragility index m with the reduced Arrhenius crossover temperature θ_A for various glass-formers. An inverse relation is observed between the two quantities. The analytical expression (Eq. (5)) is applied to the data.

rearrangements beyond the first coordination shell only occur at much lower θ_A . Furthermore, the fragility index m has been shown to correlate with the elastic properties of glass-forming liquids such as the Poisson's ratio within a class of metallic or nonmetallic glass formers (but not universally across classes) [81,88–90], composition [91], and elastic constants [92]. Our results extend these correlations to the dynamics of the high-temperature liquid state.

In summary, a clear correlation was found between the dynamic fragility and the Arrhenius crossover phenomenon. The reduced crossover temperature θ_A depends strongly on the liquid fragility, and can be observed either in the supercooled state (molecular glass former) or in the equilibrium liquid state (metallic and network glass formers). The effective activation barrier of the high-temperature Arrhenius behavior takes on a nearly universal value of $11k_B T_g$ for nonpolar molecular and metallic liquids. Such correlations between the low and high-temperature parameters imply that T_g can be estimated from the high-temperature activation barrier E_∞ and the fragility m can be estimated from the reduced crossover temperature $\theta_A = T_A/T_g$. Hence, the low-temperature glassy characteristics can be predicted from the high-temperature Arrhenius crossover in liquids.

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- [1] C. A. Angell, K. L. Ngai, G. B. McKenna, P. F. McMillan, and S. W. Martin, *J. Appl. Phys.* **88**, 3113 (2000).
- [2] M. Chen, *NPG Asia Mater.* **3**, 82 (2011).
- [3] Q. Qin and G. B. McKenna, *J. Non-Cryst. Solids* **352**, 2977 (2006).
- [4] C. A. Angell, *Science* **267**, 1924 (1995).
- [5] P. G. Debenedetti and F. H. Stillinger, *Nature* **410**, 259 (2001).
- [6] J. C. Dyre, *J. Non-Cryst. Solids* **235–237**, 142 (1998).
- [7] G. Tarjus, D. Kivelson, and P. Viot, *J. Phys. Condens. Matter* **12**, 6497 (2000).
- [8] C. M. Roland, *Soft Matter* **4**, 2316 (2008).
- [9] B. Schmidtke, N. Petzold, R. Kahlau, M. Hofmann, and E. A. Rössler, *Phys. Rev. E* **86**, 041507 (2012).
- [10] S. Mirigian and K. S. Schweizer, *J. Chem. Phys.* **140**, 194507 (2014).
- [11] V. Novikov, *Chem. Phys. Lett.* **659**, 133 (2016).
- [12] T. Iwashita, D. M. Nicholson, and T. Egami, *Phys. Rev. Lett.* **110**, 205504 (2013).
- [13] M. E. Blodgett, T. Egami, Z. Nussinov, and K. F. Kelton, *Sci. Rep.* **5**, 13837 (2015).
- [14] N. A. Mauro, M. Blodgett, M. L. Johnson, A. J. Vogt, and K. F. Kelton, *Nat. Commun.* **5**, 1 (2014).
- [15] M. D. Ediger, *Annu. Rev. Phys. Chem.* **51**, 99 (2000).
- [16] R. Richert, *J. Phys. Condens. Matter* **14**, R703 (2002).
- [17] W. Kob and H. C. Andersen, *Phys. Rev. E* **51**, 4626 (1995).
- [18] S. Sastry, P. G. Debenedetti, and F. H. Stillinger, *Nature* **393**, 554 (1998).
- [19] L. Berthier and G. Biroli, *Rev. Mod. Phys.* **83**, 587 (2011).
- [20] D. Chandler and J. P. Garrahan, *Annu. Rev. Phys. Chem.* **61**, 191 (2010).
- [21] J. D. Stevenson, J. Schmalian, and P. G. Wolynes, *Nat. Phys.* **2**, 268 (2006).
- [22] C. Donati, S. Franz, S. C. Glotzer, and G. Parisi, *J. Non-Cryst. Solids* **307–310**, 215 (2002).
- [23] T. B. Schröder, S. Sastry, J. C. Dyre, and S. C. Glotzer, *J. Chem. Phys.* **112**, 9834 (2000).
- [24] M. Vogel and S. C. Glotzer, *Phys. Rev. Lett.* **92**, 255901 (2004).
- [25] D. Coslovich and G. Pastore, *J. Phys. Condens. Matter* **21**, 285107 (2009).
- [26] A. Jaiswal, T. Egami, and Y. Zhang, *Phys. Rev. B* **91**, 134204 (2015).
- [27] A. Jaiswal, S. O'Keeffe, R. Mills, A. Podlesnyak, G. Ehlers, W. Dmowski, K. Lokshin, J. Stevick, T. Egami, and Y. Zhang, *J. Phys. Chem. B* **120**, 1142 (2016).
- [28] A. Meyer, *Phys. Rev. B* **66**, 134205 (2002).
- [29] J. Brillo, A. I. Pommrich, and A. Meyer, *Phys. Rev. Lett.* **107**, 165902 (2011).
- [30] S. M. Chathoth and A. Podlesnyak, *J. Appl. Phys.* **103**, 013509 (2008).
- [31] S. M. Chathoth, B. Damaschke, M. M. Koza, and K. Samwer, *Phys. Rev. Lett.* **101**, 037801 (2008).
- [32] S. M. Chathoth, B. Damaschke, J. P. Embs, and K. Samwer, *Appl. Phys. Lett.* **94**, 201906 (2009).
- [33] S. M. Chathoth and K. Samwer, *Appl. Phys. Lett.* **97**, 221910 (2010).
- [34] S. M. Chathoth, M. M. Koza, and A. Meyer, *Mater. Chem. Phys.* **136**, 296 (2012).
- [35] F. Yang, D. Holland-Moritz, J. Gegner, P. Heintzmann, F. Kargl, C. C. Yuan, G. G. Simeoni, and A. Meyer, *Europhys. Lett.* **107**, 46001 (2014).
- [36] F. Yang, T. Unruh, and A. Meyer, *Europhys. Lett.* **107**, 26001 (2014).
- [37] T. Kordel, D. Holland-Moritz, F. Yang, J. Peters, T. Unruh, T. Hansen, and A. Meyer, *Phys. Rev. B* **83**, 104205 (2011).
- [38] A. Meyer, W. Petry, M. Koza, and M. P. Macht, *Appl. Phys. Lett.* **83**, 3894 (2003).
- [39] A. S. Keys, L. O. Hedges, J. P. Garrahan, S. C. Glotzer, and D. Chandler, *Phys. Rev. X* **1**, 021013 (2011).
- [40] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevLett.117.205701> for relevant data on all glass-formers, derivations presented in main text, which includes Refs. [41–69].
- [41] D. N. Perera, *J. Phys. Condens. Matter* **11**, 3807 (1999).
- [42] G. J. Fan, J. J. Z. Li, W.-K. Rhim, D. C. Qiao, H. Choo, P. K. Liaw, and W. L. Johnson, *Appl. Phys. Lett.* **88**, 221909 (2006).
- [43] A. Meyer, R. Busch, and H. Schober, *Phys. Rev. Lett.* **83**, 5027 (1999).
- [44] E. S. Park, J. H. Na, and D. H. Kim, *Appl. Phys. Lett.* **91**, 031907 (2007).
- [45] S. Mukherjee, Z. Zhou, W. L. Johnson, and W. K. Rhim, *J. Non-Cryst. Solids* **337**, 21 (2004).
- [46] M. Ikeda and M. Aniya, *Intermetallics* **18**, 1796 (2010).

- [47] A. Meyer, J. Wuttke, and W. Petry, *J. Non-Cryst. Solids* **250–252**, 116 (1999).
- [48] L.-M. Wang, Y. Tian, R. Liu, and W. Wang, *Appl. Phys. Lett.* **100**, 261913 (2012).
- [49] J. C. Martinez-Garcia, S. J. Rzoska, A. Drzozd-Rzoska, J. Martinez-Garcia, and J. C. Mauro, *Sci. Rep.* **4**, 5160 (2014).
- [50] H. Matsuoka, *J. Chem. Phys.* **137**, 204506 (2012).
- [51] T. Hecksher, A. I. Nielsen, N. B. Olsen, and J. C. Dyre, *Nat. Phys.* **4**, 737 (2008).
- [52] A. I. Nielsen, T. Christensen, B. Jakobsen, K. Niss, N. B. Olsen, R. Richert, and J. C. Dyre, *J. Chem. Phys.* **130**, 154508 (2009).
- [53] P. Lunkenheimer, L. C. Pardo, M. Köhler, and A. Loidl, *Phys. Rev. E* **77**, 031506 (2008).
- [54] L.-M. Wang, C. A. Angell, and R. Richert, *J. Chem. Phys.* **125**, 074505 (2006).
- [55] K. Niss, C. Dalle-Ferrier, G. Tarjus, and C. Alba-Simionesco, *J. Phys. Condens. Matter* **19**, 076102 (2007).
- [56] C. Klieber, T. Hecksher, T. Pezeril, D. H. Torchinsky, J. C. Dyre, and K. A. Nelson, *J. Chem. Phys.* **138**, 12A544 (2013).
- [57] L.-M. Wang and R. Richert, *J. Phys. Chem. B* **111**, 3201 (2007).
- [58] J. L. Zhang, W. H. Wu, H. W. Zhou, X. Z. Guo, and Y. N. Huang, *Appl. Phys. Lett.* **92**, 131906 (2008).
- [59] M. Paluch, C. M. Roland, J. Gapinski, and A. Patkowski, *J. Chem. Phys.* **118**, 3177 (2003).
- [60] R. Casalini, M. Paluch, and C. M. Roland, *J. Phys. Condens. Matter* **15**, S859 (2003).
- [61] T. Blochowicz, C. Gainaru, P. Medick, C. Tschirwitz, and E. A. Rossler, *J. Chem. Phys.* **124**, 134503 (2006).
- [62] L.-M. Wang, *J. Phys. Chem. B* **113**, 5168 (2009).
- [63] D. R. Neuville, *Chem. Geol.* **229**, 28 (2006).
- [64] M. Liška, P. Šimurka, J. Antalík, and P. Perichta, *Chem. Geol.* **128**, 199 (1996).
- [65] Z. Chen and R. Richert, *J. Chem. Phys.* **135**, 124515 (2011).
- [66] V. A. Popova and N. V. Surovtsev, *J. Chem. Phys.* **135**, 134510 (2011).
- [67] D. Huang, D. M. Colucci, and G. B. McKenna, *J. Chem. Phys.* **116**, 3925 (2002).
- [68] T. Scopigno, J.-B. Suck, R. Angelini, F. Albergamo, and G. Ruocco, *Phys. Rev. Lett.* **96**, 135501 (2006).
- [69] E. Courtens, M. Foret, B. Rufflé, and R. Vacher, *Phys. Rev. Lett.* **98**, 079603 (2007).
- [70] F. Mallamace, C. Branca, C. Corsaro, N. Leone, J. Spooren, S.-H. Chen, and H. E. Stanley, *Proc. Natl. Acad. Sci. U.S.A.* **107**, 22457 (2010).
- [71] C. Zhou, L. Hu, Q. Sun, H. Zheng, C. Zhang, and Y. Yue, *J. Chem. Phys.* **142**, 064508 (2015).
- [72] V. N. Novikov and A. P. Sokolov, *Phys. Rev. E* **67**, 031507 (2003).
- [73] E. Rössler, K.-U. Hess, and V. Novikov, *J. Non. Cryst. Solids* **223**, 207 (1998).
- [74] K. S. Schweizer and E. J. Saltzman, *J. Chem. Phys.* **121**, 1984 (2004).
- [75] H.-B. Yu, W.-H. Wang, and K. Samwer, *Mater. Today* **16**, 183 (2013).
- [76] H. B. Yu, W. H. Wang, H. Y. Bai, and K. Samwer, *Natl. Sci. Rev.* **1**, 429 (2014).
- [77] S. W. Basuki, A. Bartsch, F. Yang, K. Rätzke, A. Meyer, and F. Faupel, *Phys. Rev. Lett.* **113**, 165901 (2014).
- [78] N. Petzold, B. Schmidtke, R. Kahlau, D. Bock, R. Meier, B. Micko, D. Kruk, and E. A. Rössler, *J. Chem. Phys.* **138**, 12A510 (2013).
- [79] D. Kivelson, G. Tarjus, X. Zhao, and S. A. Kivelson, *Phys. Rev. E* **53**, 751 (1996).
- [80] B. Schmidtke, M. Hofmann, A. Lichtinger, and E. A. Rössler, *Macromolecules* **48**, 3005 (2015).
- [81] V. N. Novikov and A. P. Sokolov, *Nature* **431**, 961 (2004).
- [82] Y. S. Elmatad, D. Chandler, and J. P. Garrahan, *J. Phys. Chem. B* **113**, 5563 (2009).
- [83] R. Svoboda and J. Málek, *J. Non-Cryst. Solids* **419**, 39 (2015).
- [84] J. Krausser, K. H. Samwer, and A. Zaccone, *Proc. Natl. Acad. Sci. U.S.A.* **112**, 13762 (2015).
- [85] J. Mattsson, H. M. Wyss, A. Fernandez-Nieves, K. Miyazaki, Z. Hu, D. R. Reichman, and D. A. Weitz, *Nature* **462**, 83 (2009).
- [86] J. Yang and K. S. Schweizer, *J. Chem. Phys.* **134**, 204908 (2011).
- [87] D. L. Sidebottom, *Phys. Rev. E* **92**, 062804 (2015).
- [88] V. N. Novikov, Y. Ding, and A. P. Sokolov, *Phys. Rev. E* **71**, 061501 (2005).
- [89] V. N. Novikov and A. P. Sokolov, *Phys. Rev. B* **74**, 064203 (2006).
- [90] G. P. Johari, *Philos. Mag.* **86**, 1567 (2006).
- [91] Y. Q. Cheng, E. Ma, and H. W. Sheng, *Appl. Phys. Lett.* **93**, 111913 (2008).
- [92] W. H. Wang, *J. Appl. Phys.* **99**, 093506 (2006).