

Ambient Pressure Structural Quantum Critical Point in the Phase Diagram of $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$

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The quasiskutterudite superconductor $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ features a pronounced anomaly in electrical resistivity at $T^* \sim 138$ K. We show that the anomaly is caused by a second-order structural transition, which can be tuned to 0 K by applying physical pressure and chemical pressure via the substitution of Ca for Sr. A broad superconducting dome is centered around the structural quantum critical point. Detailed analysis of the tuning parameter dependence of T^* as well as insights from lattice dynamics calculations strongly support the existence of a structural quantum critical point at ambient pressure when the fraction of Ca is 0.9 (i.e., $x_c = 0.9$). This establishes the $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ series as an important system for exploring the physics of structural quantum criticality without the need of applying high pressures.

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Quantum criticality associated with a second-order phase transition at zero temperature has been a fruitful concept in the search for superconductivity. Particularly clear examples include the heavy fermion [1,2] and iron-based systems [3–6], where superconductivity is stabilized in the vicinity of an antiferromagnetic quantum critical point (QCP). In addition to magnetic instabilities, structural instabilities also have an intricate influence on the superconductivity. However, clean systems with a continuous structural transition tunable by external parameters are relatively rare, and for the few cases studied in recent years, the structural order is found in the vicinity of magnetic order [7], making the notion of structural quantum criticality and its influence on the superconductivity less advanced than the magnetic counterpart.

Superconducting stannides [8,9], including $A_3T_4\text{Sn}_{13}$ systems, where $A = \text{La}, \text{Sr}, \text{Ca}$ and $T = \text{Ir}, \text{Rh}$, have recently been reexamined using a wide range of probes [10–22]. $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$, a member of this stoichiometry family, is a strong-coupling superconductor with a critical temperature $T_c \approx 4.7$ K. The superconducting gap function Δ is nodeless, with $2\Delta/k_B T_c = 3.87$ [11]. In addition to superconductivity, the normal state of $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ features another instability, which sets in at $T^* \approx 138$ K. In $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$, which is also a superconductor with $T_c = 7.8$ K, no signature of the anomaly associated with T^* was observed. The substitution series $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ is readily available, allowing us to track the evolution of T^* and its interplay with superconductivity when Ca is partially replaced by Sr, which simulates a negative chemical pressure, as well as

when positive physical pressure is applied directly on the crystals.

A recent investigation of the related series $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Ir}_4\text{Sn}_{13}$ revealed the existence of a structural quantum phase transition at 18 kbar in $\text{Ca}_3\text{Ir}_4\text{Sn}_{13}$ [13]. Since $\text{Ca}_3\text{Ir}_4\text{Sn}_{13}$ sits at the limit of the chemical pressure in this series, physical pressure has to be applied in order to access and study the physics of the structural quantum phase transition, thus restricting the accessibility of many experimental probes. In this Letter, we argue that the $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ series is an excellent alternative for the investigation of structural criticality. Here, we show that T^* is a second-order structural transition temperature that can be suppressed with a suitable combination of physical and chemical pressures. Importantly, T^* can be driven to 0 K for $(\text{Ca}_{0.9}\text{Sr}_{0.1})_3\text{Rh}_4\text{Sn}_{13}$ (i.e., $x = 0.9$) without applying physical pressure. In the vicinity of $x = 0.9$, the structural QCP, a broad superconducting dome is observed, giving rise to a phase diagram reminiscent of cases featuring an antiferromagnetic QCP.

Single crystals of $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ were synthesized by the Sn flux method using similar parameters to those described in Ref. [10]. The high pressure electrical resistivity $\rho(T)$ was measured using a four-wire configuration in a piston-cylinder cell with Daphne 7373 as the pressure transmitting medium. The pressure achieved was determined using the superconducting transition of lead. The heat capacity $C_p(T)$ was measured using a standard pulse relaxation method. Both $C_p(T)$ and $\rho(T)$ were measured in a Physical Property Measurement System (Quantum Design). Single crystal XRD measurements

were performed, using an Oxford Diffraction Supernova A utilizing Mo $K\alpha$ radiation and equipped with an Oxford Cryostream Plus. Data integration and analytical absorption corrections were performed with CRYSTALIS PRO [23]. Structures were solved using direct methods and refined against $|F|^2$ using SHELX-97 [24] via the OLEX-2 [25] interface. The electronic structure has been calculated using the local density approximation. The VASP [26] code was employed using projector-augmented wave potentials. A plane wave cutoff of 300 eV was used and a minimum $4 \times 4 \times 4$ k -point grid was set up, which was scaled appropriately for supercells. The phonon spectrum was calculated by using PHONOPY [27] in conjunction with VASP. All calculations were nonspin polarized.

The normal state of $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ features a phase transition at $T^* \approx 138$ K, which is detectable in a wide range of physical quantities including the heat capacity [Fig. 1(a)] and electrical resistivity [Fig. 1(b)]. The minimum in $d\rho/dT$ is used to define T^* , as indicated in Fig. 1(b). The measurements of both $C_p(T)$ and $\rho(T)$ have been carried out on cooling down and warming up, at the same sweep rates. The cooling and warming curves overlap perfectly, showing the absence of hysteresis. In addition, $C_p(T)$ near T^* exhibits a lambdalike jump. These observations indicate that T^* corresponds to a transition temperature of a second-order or only very weakly first-order phase transition.

Single crystal x-ray diffraction measurements at 300 K confirmed that $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ adopts a $Pm\bar{3}n$ cubic structure, the I phase, with $a = 9.80610(18)$ Å. Below T^* , additional weak reflections are observed that are consistent with a phase transition to a body-centered cubic structure, the I' phase, with a lattice parameter twice that of the high temperature phase (Sec. C of Ref. [28]). An acceptable refinement could only be achieved in space group $I\bar{4}3d$ with $a = 19.5714(3)$ Å at 100 K ($R_{1\text{obs}}$, $wR_{2\text{obs}}$, and χ^2 of 3.1%, 8.9%, and 1.1%). The structural distortion is similar to that observed in $\text{Sr}_3\text{Ir}_4\text{Sn}_{13}$ at 147 K [13,18], with the bond distances in the $\text{Sn}(1)\text{Sn}(2)_{12}$ icosahedra distorting into four groups of three identical bond distances [see Fig. 1(c)]. This occurs in concert with tilting of three quarters of the $\text{RhSn}(2)_6$ trigonal prisms with only those trigonal prisms whose axes are along the (111) direction remaining untilted. The transition temperature remains the same on heating and cooling, consistent with the expected second-order transition between the $Pm\bar{3}n$ and $I\bar{4}3d$ space groups, which have a group-subgroup relationship.

Mode decomposition of the low temperature phase was carried out using the ISODISTORT software suite [29] by comparison with the parent high symmetry phase. It revealed that the active \mathbf{q} vector is (0.5, 0.5, 0) and its symmetry equivalents. In particular, the structural distortion appears to be primarily driven by the modes belonging to the M_2 -irreducible representation, primarily causing displacement of the Sn(2) sites along with small

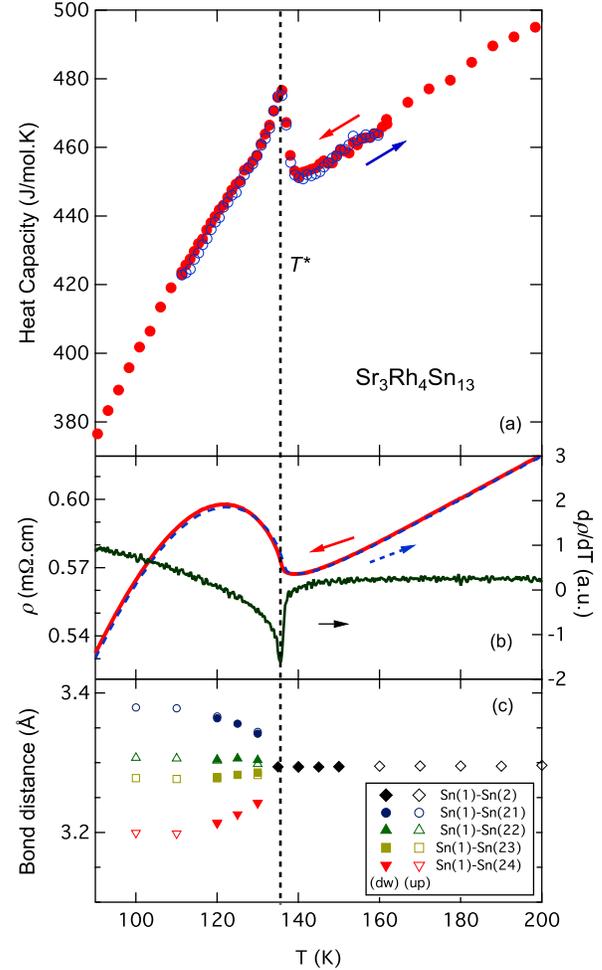


FIG. 1 (color online). The temperature dependence of the (a) heat capacity, (b) electrical resistivity, and (c) Sn(1)–Sn(2) bond distances for $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ near T^* . In (a) and (c), the closed (open) symbols denote the data collected on cooling (warming). In (b), the solid (broken) line is the cooling (warming) curve. All these point to the absence of hysteresis at T^* . The temperature derivative of the electrical resistivity in (b) provides a means to define T^* .

movements of the strontium cations. It has previously been shown that the transition between the primitive and body-centered structures is linked to charge transfer between Sn(2) and Sn(1) atoms, which increases the ionicity of the bond between these atoms [30,31].

Having established the origin of the anomaly at T^* , we now examine its pressure dependence. We show in Fig. 2(a) the high pressure $\rho(T)$ of $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ ($x = 0$). T^* decreases rapidly when hydrostatic pressure is applied. Concurrently, the superconducting transition temperature T_c , defined using the resistive midpoint of the transition, changes only slightly under pressure. In $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ ($x = 1$), no signature of T^* was detected for all pressures studied, while T_c is higher than that in $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ and shows a gentle variation with pressures (Figs. S5 and S6 of Ref. [28]).

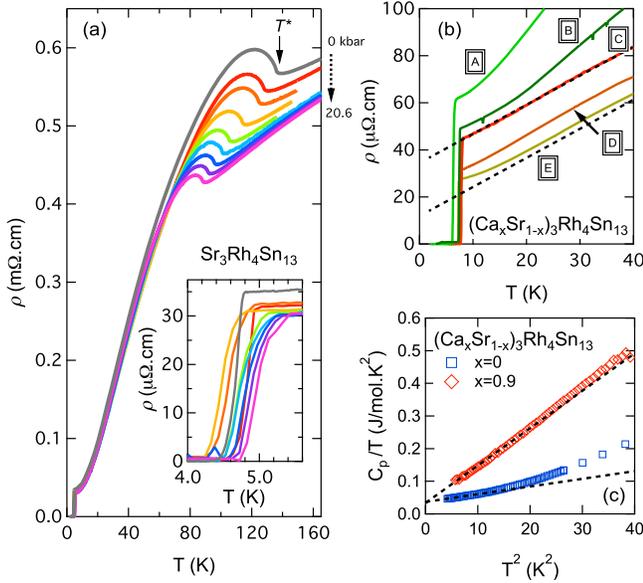


FIG. 2 (color online). (a) High pressure $\rho(T)$ in $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ for the determination of T^* and T_c (inset). The pressure values increase sequentially as indicated by the arrow, and are given by 0, 4.0, 6.0, 12.7, 14.0, 15.0, 17.3, 19.1, and 20.6 kbar. (b) Representative $\rho(T)$ for the system situated at five different locations of the universal phase diagram (see arrows in Fig. 3), spanning a range that straddles the QCP. The dashed straight lines are guides for the eyes [32]. (c) The normal state C_p/T is plotted against T^2 for $x = x_c = 0.9$ (diamond), right at the quantum critical point, and at $x = 0$ (square).

Both $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ and $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ adopt the $Pm\bar{3}n$ structure at room temperature; the lattice constant of $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ shrinks linearly and uniformly with increasing calcium content x , obeying Vegard's law (Fig. S1 of Ref. [28]). Therefore, the substitution of Ca for Sr simulates the effect of hydrostatic pressure, and $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ can be regarded as the high pressure analogue of $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$. Detailed analysis of all T^* and T_c data suggests that $\Delta x = 1$ corresponds to $\Delta p = 68$ kbar (see Sec. A of Ref. [28]). Placing $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ at -68 kbar and $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ at the origin of the pressure scale, we construct the universal phase diagram (Fig. 3), which illustrates the interplay between the structural order and superconductivity driven by both the physical pressure and the chemical pressure. The universal phase diagram is reminiscent of the cases where superconductivity is stabilized on the border of magnetism (e.g., Refs. [1–6]). The crucial difference here is that it is a structural transition that is suppressed by the nonthermal tuning parameter. T^* extrapolates to 0 K at $p_c = -6.8$ kbar relative to $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$, or, equivalently, at $x_c = 0.9$. If the structural transition remains second order, or weakly first order, a structural QCP at x_c or p_c is expected. Thus, the phase diagram of $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ features a QCP that is accessible at ambient pressure by fine tuning the Ca content.

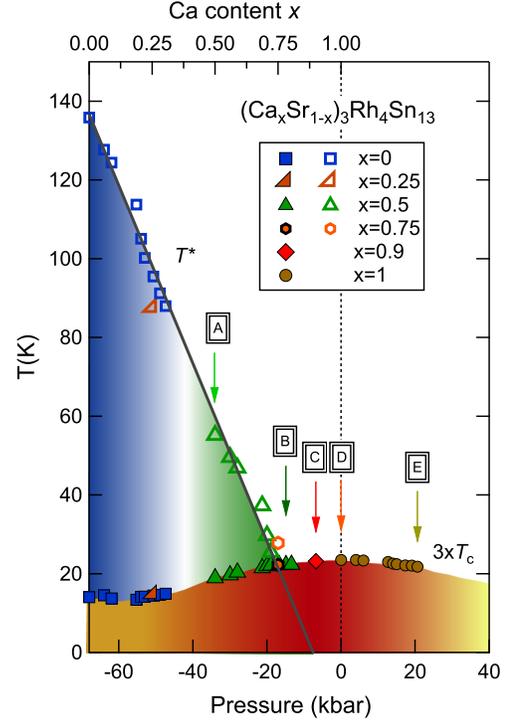


FIG. 3 (color online). Universal phase diagram showing the pressure dependence of T^* and T_c for $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$. The origin of the pressure scale is chosen to coincide with $x = 1$, as indicated by the dashed vertical line. In this representation, the part of the phase diagram to the left of the dashed line is accessible by varying the calcium content, whereas the part to the right is accessible only by physical pressures. The arrows indicate the positions in the parameter space where $\rho(T)$ curves are presented in Fig. 2(b).

When a crystalline system undergoes a second-order displacive structural transition, the relevant phonon mode must go soft at the transition temperature T^* . The existence of a phonon mode at $Q \neq 0$ such that $\omega_Q \rightarrow 0$ implies the emergence of a new periodicity. When T^* is tuned to 0 K, the associated softening of the phonon mode occurs at 0 K. This gives rise to additional low-lying phonon modes, which can be excited at low temperatures. Using the generalized Bloch-Grüneisen expression [33,34], the contribution to the electrical resistivity due to phonon scattering can be written as

$$\Delta\rho_{\text{ph}}(T) \propto \sum_{\mathbf{q}} \alpha_{(\text{tr})\mathbf{q}}^2 \frac{x_{\mathbf{q}} e^{x_{\mathbf{q}}}}{(e^{x_{\mathbf{q}}} - 1)^2} \approx \sum_{\mathbf{q}} \alpha_{(\text{tr})\mathbf{q}}^2 \left(\frac{k_B}{\hbar\omega_{\mathbf{q}}} \right) T,$$

where $\alpha_{(\text{tr})\mathbf{q}}^2$ is a \mathbf{q} -dependent Fermi surface average of the electron-phonon interaction, $\omega_{\mathbf{q}}$ is the phonon frequency, and $x_{\mathbf{q}} = \hbar\omega_{\mathbf{q}}/k_B T$. The approximation in the last part of the expression is applicable when $\hbar\omega_{\mathbf{q}} \ll k_B T$, i.e., the softening of the phonon frequency relative to the thermal energy. Therefore, the phononic contribution to the electrical resistivity is linear in T when $\hbar\omega_{\mathbf{q}} \ll k_B T$; the degree of the contribution is enhanced with softer $\omega_{\mathbf{q}}$.

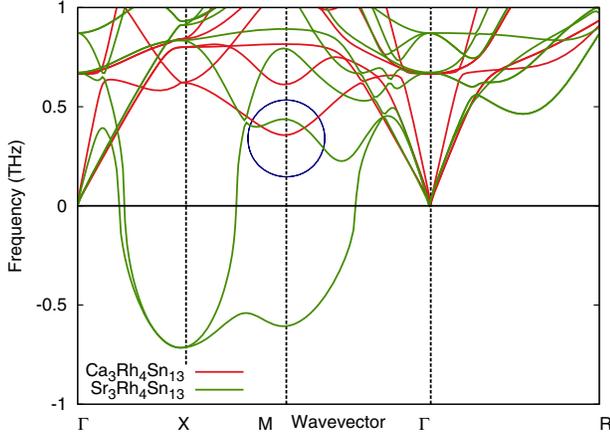


FIG. 4 (color online). Dispersion relation $\omega(q)$ for $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ and $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ at $T = 0$. The circle highlights the presence of an anomalously soft phonon mode in $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ near M . The negative part of the frequency axis denotes imaginary phonon mode frequencies.

Figure 2(b) shows five representative traces of $\rho(T)$ whose composition and pressure values are chosen to straddle $x_c = 0.9$ ($p_c = -6.8$ kbar), the location where $T^* \rightarrow 0$ K, and their positions in the universal phase diagram are indicated by the arrows in Fig. 3. Far below x_c , $\rho(T)$ exhibits a large curvature just above T_c . However, at x_c , a distinct T -linear resistivity is observed over a wide temperature range (curve C). According to our treatment in the preceding paragraph, this observation at low temperatures suggests the existence of a very soft phonon mode, so as to satisfy $\hbar\omega_q \ll k_B T$. This is consistent with the scenario of a second-order structural quantum phase transition at x_c . Moving away from x_c , the curvature gradually builds up again, as evidenced in the $\rho(T)$ traces labeled D and E [see also Fig. S7 of Ref. [28] for intermediate $\rho(T)$ between D and E].

We can gain further insight by calculating the phonon dispersion relation for $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$, which is below x_c , and $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$, which according to our experimental results is above x_c . In Fig. 4 the low energy modes of the phonon spectrum are shown for $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ and $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$. The depicted points along the high symmetry lines lie at coordinates $X(0.5, 0, 0)$, $M(0.5, 0.5, 0)$, and $R(0.5, 0.5, 0.5)$ in the Brillouin zone corresponding to the simple cubic lattice. While $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ with its larger lattice parameter possesses imaginary mode frequencies, indicating a lattice instability, for $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ these modes have hardened to yield real frequencies. This is consistent with the experimentally observed phase diagram, in which $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ undergoes a structural transition at ~ 138 K while for $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ the transition is absent. However, as highlighted by the circle in the figure, $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ retains a soft mode of approximately 0.35 THz at M , the wave vector associated with the structural transition in $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$. This suggests that $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ is near to the critical end point

for the structural phase transition, and the calculations lend further support to our conclusion that a structural QCP is reachable in the $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ series by tuning the Ca content.

When a QCP located at x_c is approached, the order parameter correlation length ξ and correlation time ξ_τ diverge according to $\xi \sim |x - x_c|^{-\nu}$ and $\xi_\tau \sim \xi^z$ [6,35]. The dynamical exponent z can be determined from the dispersion relation at x_c : $\omega \propto q'^z$. We can analyze our system by approaching the critical concentration x_c from the quantum disordered side. The dispersion relation for $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$ ($x > x_c$) around the minimum at M (see Fig. 4) can be Taylor expanded as $\omega \propto \sqrt{c^2 q'^2 + \Delta^2}$, where q' is the wave vector measured from M and Δ is the gap between the soft mode frequency at M and $\omega = 0$. With the alloying of Sr, Δ decreases and it eventually vanishes at x_c , triggering a structural quantum phase transition. At x_c , the dispersion relation around M becomes $\omega \propto q'$, analogous to an acoustic branch with slope c . Consequently, z is unity and the effective dimension for quantum criticality in our system $d_{\text{eff}} = d + z = 4$, where d is the spatial dimension [6,35].

The additional phonon branch with a linear dispersion relation at the QCP strongly affects the low temperature heat capacity. Analogous to Debye's treatment, the contribution of this additional branch to the heat capacity is proportional to T^3 . In Fig. 2(c), we plot the normal state C_p/T versus T^2 for $x = 0$ and $x = x_c = 0.9$. The low temperature part of the data can be described by $C_p/T = \gamma + \beta T^2$ (dashed lines), where γ is the coefficient governing the electronic contribution to the heat capacity and the slope β is related to the phonon contribution due to linear $\omega(q)$. While γ is more or less the same in both compositions, β at the QCP is ~ 4.6 times larger than that at $x = 0$, which is far away from the QCP. This additional T^3 contribution at the QCP is consistent with the model we proposed in the preceding paragraph.

In summary, we have established the T - p phase diagram of $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$. We show that a second-order structural transition temperature T^* can be tuned to 0 K by a suitable combination of p and x . When $T^* \rightarrow 0$, a linear-in- T electrical resistivity is observed, a significant T^3 contribution to the heat capacity is recorded, and the superconducting transition temperature peaks near this pressure or composition. Lattice dynamics calculations reveal the existence of imaginary phonon frequencies in $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$, indicative of structural instabilities. These imaginary phonon mode frequencies completely disappear in $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$. Our combined experimental and computational effort highlights a particularly important feature of this series, in that $T^* \rightarrow 0$ can be achieved for $x = 0.9$ without the need of applying pressure, which opens up a new avenue for detailed investigation of the physics of structural quantum criticality.

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- [28] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevLett.114.097002> for (a) lattice constant a versus x for $(\text{Ca}_x\text{Sr}_{1-x})_3\text{Rh}_4\text{Sn}_{13}$ (b) further high pressure $\rho(T)$ data of $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$, $(\text{Ca}_{0.5}\text{Sr}_{0.5})_3\text{Rh}_4\text{Sn}_{13}$, and $\text{Ca}_3\text{Rh}_4\text{Sn}_{13}$, (c) the resulting zone images of selected low index planes from diffraction in $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ at 100 and 300 K, and (d) crystal structure information files (.cif) and reflection intensities files (.hkl) on $\text{Sr}_3\text{Rh}_4\text{Sn}_{13}$ from single crystal x-ray diffraction at 100 and 300 K.
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