Electronic Excitations of a Magnetic Impurity State in the Diluted Magnetic Semiconductor (Ga,Mn)As

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The electronic structure of doped Mn in (Ga,Mn)As is studied by resonant inelastic x-ray scattering. From configuration-interaction cluster-model calculations, the line shapes of the Mn L_3 resonant inelastic x-ray scattering spectra can be explained by d-d excitations from the Mn ground state dominated by chargetransferred states, in which hole carriers are bound to the Mn impurities, rather than a pure acceptor Mn^{2+} ground state. Unlike archetypical d-d excitation, the peak widths are broader than the experimental energy resolution. We attribute the broadening to a finite lifetime of the d-d excitations, which decay rapidly to electron-hole pairs in the host valence and conduction bands through the hybridization of the Mn 3d orbital with the ligand band.

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A diluted magnetic semiconductor (DMS), in which a host semiconductor is doped with a low concentration of magnetic ions, is a material of interest for spintronics research. In particular, ferromagnetic DMSs have attracted considerable attention because the itinerant carriers are considered to mediate the magnetic interaction between doped ions [1,2]. This type of magnetism is called "carrierinduced ferromagnetism" and enables the possibility of manipulating both electronic charge and spin degrees of freedom. III-V based DMS $Ga_{1-x}Mn_xAs$ is an archetypical ferromagnetic DMS and has been studied both fundamentally and in applications as a ferromagnetic DMS [3]. Spintronic devices using Ga_{1-x}Mn_xAs have been fabricated and their effectiveness has been demonstrated [4,5], although the Curie temperature (T_c) of $Ga_{1-r}Mn_rAs$ is below room temperature ($T_C < 200$ K).

Understanding the mechanism of ferromagnetism in $Ga_{1-x}Mn_xAs$ is strongly desirable to develop DMS for applications in spintronic devices. Several physical models of the ferromagnetism have been proposed, e.g., the Zener p-d exchange model [1,6,7], the Mn 3d impurity band model [8–10], and the magnetic polaron model [11,12]. These models depend on the localization of hole carriers around the Mn ions, the energy position of the Mn 3d states, and the strength of the hybridization between the Mn 3dorbital and the ligand band. To obtain a fundamental understanding of ferromagnetism, it is essential to know the Mn 3d electronic configurations in the ground state. If the valence state of Mn in the ground state is divalent (Mn^{2+}) , the Mn ion acts as an acceptor supplying a hole to the host GaAs, supporting the Zener p-d exchange model. Thus, the electronic structure of Mn 3d ions in Ga_{1-r}Mn_rAs has been intensively investigated experimentally [10,13–15]. However, the presence of divalent (Mn2+) or trivalent (Mn^{3+}) states has not been conclusively determined, and the degree of the localization of the Mn 3d states remains a matter of dispute.

In this Letter, to address the electronic structure of the doped Mn ions, we report the results of Mn L_3 x-ray absorption spectroscopy (XAS) and resonant inelastic x-ray scattering (RIXS) measurements of $Ga_{1-x}Mn_xAs$ (x = 0.04). RIXS is a powerful tool to investigate electronic excitations in element- and symmetry-specific ways including d-d and charge-transfer (CT) excitations for open shell 3d orbitals [16], and magnetic excitations for spin or charge-ordered systems [17]. These excitations are sensitive to electron correlation, crystalline symmetry, and the strength of hybridization with the ligand band. Moreover, RIXS is less sensitive to the surface structure because of its long probing depth (>100 nm). The RIXS spectra obtained for Ga_{1-r}Mn_rAs are compared with configuration-interaction (CI) cluster-model calculations [18], and the electronic structure parameters are estimated. It is found that the bulk electronic structures of Mn ions in Ga_{1-r}Mn_rAs consist predominantly of chargetransferred states. Based on the experimental findings, we shall discuss the Mn 3*d* electronic state in $Ga_{1-x}Mn_xAs$.

 $Ga_{1-x}Mn_xAs$ (x = 0.04) thin films with a thickness of 20 nm were grown on a GaAs(001) substrate at 270 °C under an ultrahigh vacuum by a molecular beam epitaxy method. To avoid surface oxidation, the samples were covered by GaAs (2 nm) and an As capping layer (1 nm) after the deposition of the $Ga_{1-x}Mn_xAs$ layer to produce a structure of As/GaAs/Ga_{0.96}Mn_{0.04}As/GaAs(buffer)/GaAs(001). The thin film sample after the deposition was cut into two pieces and one half of the sample was annealed in an ultrahigh vacuum of 6.2×10^{-9} Torr at 270 °C for 2 hours. T_C values of the as-grown and annealed samples were ~65 and ~ 100 K, respectively, as determined by the Arrott plot of the magnetic circular dichroism. The RIXS and XAS experiments were performed using the high-resolution soft x-ray emission station HORNET [19] at the long undulator beam line BL07LSU of SPring-8, Japan [20]. The total energy resolution for the RIXS experiments was about 170 meV at the Mn L_3 edge ($E/\Delta E > 3750$). The RIXS spectra were measured with linear horizontal polarization at room temperature under a vacuum of 1.0×10^{-5} Pa. The XAS signals were collected in the total-fluorescence-yield (TFY) mode. The XAS spectra of MnO were measured in the total-electron-yield mode as a reference.

Figure 1(a) shows the Mn L_3 XAS spectra of the $Ga_{1-r}Mn_rAs$ thin films. The XAS spectra show a broad profile compared with the reference MnO, similar to the XAS spectra previously reported for chemically etched $Ga_{1-x}Mn_xAs$ thin films [21,22]. The XAS spectra exhibit the same features as those of the "intrinsic Mn component" [see Fig. 2(a)], which consists mainly of Mn ions substituting Ga sites and partial filling of some interstitial site [23]. In $Ga_{1-r}Mn_rAs$, there are ferromagnetic and paramagnetic Mn components, which come from the intrinsic Mn and extrinsic oxidized Mn ions, respectively. Details for the assignment of Mn species are described in Ref. [23]. It has been reported that low-temperature annealing of Ga_{1-x}Mn_xAs induces interstitial Mn defects to diffuse toward the surface [24,25] creating Mn oxides at the surface [26]. However, because of the bulk sensitivity of the fluorescence yield mode, the XAS spectra do not show the oxide components and reflect the bulk electronic structure.

Figure 1(b) shows the Mn L_3 RIXS spectra of the Ga_{1-x}Mn_xAsthin films taken with various excitation (incident photon) energies $h\nu$ across the L_3 edge in the XAS spectra. The spectra are normalized by area and plotted as a function of energy loss (E_L). The line shape of the RIXS spectra strongly depends on the excitation energy. As a reference, Mn L_3 RIXS spectra of MnO for the corresponding excitation energy are also shown in Fig. 1(b). The RIXS spectra of MnO demonstrate sharp *d*-*d* excitation peaks from $E_L = 2$ to 6 eV and broad features that originate from CT excitation above $E_L = 6$ eV [16]. Notably, the energy resolution is high enough to resolve individual *d*-*d* excitation peaks in MnO. In contrast, the RIXS spectra of Ga_{1-x}Mn_xAs show broad line shapes even at the same high energy resolution, which are similar to those previously reported at the lower energy



FIG. 1 (color online). Mn L_3 resonant inelastic x-ray scattering spectra of the as-grown and annealed $\text{Ga}_{1-x}\text{Mn}_x\text{As}$ (x = 0.04) thin films. (a) Mn L_3 XAS spectra taken in the TFY mode. The vertical dashed lines denote the excitation photon energy for the RIXS measurements. (b) Mn L_3 RIXS spectra. The excitation energies are 639.6, 640.0, and 640.6 eV. The spectra of MnO are also shown as a reference.

resolution [26]. Therefore, the broad features of the Mn L_3 RIXS spectra are expected to accurately reflect the electronic structure of the Mn ions in Ga_{1-x}Mn_xAs. Since the CT excitation features from MnO are absent, the CT excitation energy is likely to be reduced to an energy level comparable to that of the *d*-*d* excitation ($2 < E_L < 6$ eV) and contribute to the broadening of the *d*-*d* transition peaks [27]. There are differences in the RIXS spectra taken at $h\nu = 640$ and 640.6 eV between the as-grown and annealed samples. These changes in the RIXS spectra taken at $h\nu = 640$ and 640.6 eV by annealing may be attributed to the out diffusion of interstitial Mn atoms during annealing.

The intrinsic Mn $L_{2,3}$ XAS and x-ray magnetic circular dichroism (XMCD) spectra are compared with CI clustermodel calculations, as shown in Figs. 2(a) and 2(b), respectively. In the model, tetrahedral $[MnAs_4]^{10-}$ and $[MnAs_4]^{9-}$ clusters, in which Mn 3d and As 4p orbitals are considered, are assumed for nominally divalent (Mn^{2+}) and trivalent (Mn³⁺) ionic states, respectively. Both the calculated spectra for the Mn²⁺ and Mn³⁺ states reproduce the experimental spectra. Common electronic structure parameters are used to fit both the XAS and XMCD: the ligand-to-3d CT energy $\Delta = 1.5$ eV, the d-d Coulomb interaction energy $U_{dd} = 3.5$ eV, the Slater-Koster parameter $(pd\sigma) = -0.9$ eV, and the electrostatic crystal field $10Dq_{crvs} = 0$ eV. The Slater integrals for the 3*d*-3*d* and 3d-2p multipole interactions are reduced to 65% of the atomic Hatree-Fock values. Note that the Mn 3*d*-As 4p



FIG. 2 (color online). RIXS spectra compared with the CI cluster-model calculations. (a),(b) Comparison of the intrinsic Mn $L_{2,3}$ XAS and XMCD spectra with the CI calculations, respectively. The experimental XAS and XMCD spectra are extracted from the magnetic-field dependence of XMCD [23]. (c) Comparison of experimental RIXS spectra with CI calculations for Mn²⁺ and Mn³⁺ states. The CI calculations for RIXS were performed using the parameters common to those of XAS (XMCD). The calculated spectra are broadened by the experimental energy resolution of 170 meV.

hybridization included in our tetrahedral MnAs4 cluster model causes an effective T_d crystal-field splitting between the e_q and t_{2q} levels on the Mn site $10Dq_{hyb} = -0.39$ eV in addition to $10Dq_{crys}$ [28]. This effective T_d crystal-field value is comparable to the value of -0.5 eV adopted in an atomic model in Ref. [29]. Here, the CT energy Δ is defined as the energy difference between the d^5 and d^6L states for Mn^{2+} , and between d^5L and d^6L^2 states for Mn^{3+} , where L denotes a hole in the ligand bands. The values of Δ , U_{dd} , and $(pd\sigma)$ are the same as those estimated from the photoemission experiments [30]. We note that spectra calculated using an electrostatic field $10Dq_{crvs}$ smaller than -0.3 eV could not reproduce the experimental spectra. The small crystal-field splitting is similar to that of the iron pnictide $BaFe_2As_2$ [31] and the chalcopyrite CuFeS₂ [32], as expected from the small electronegativity of the heavy chalcogenide anions compared with that of oxygen.

The CI calculations for the Mn^{2+} and Mn^{3+} states were also performed for the RIXS spectra using the same electronic structure parameters as those for XAS and XMCD. Figure 2(c) shows the Mn L_3 RIXS spectra of the $Ga_{1-x}Mn_xAs$ thin film compared with the CI calculations broadened by a Gaussian of the experimental energy resolution. The calculated spectra for the "Mn³⁺" state reproduce the energy distribution of the experimental spectra except for the linewidths of each *d*-*d* excitation peak, while the calculated spectra for the Mn²⁺ state are different from the experimental RIXS spectra. The RIXS spectra of the $Ga_{1,r}Mn_rAs$ samples show finite spectral features in the low E_L region below 1.5 eV as indicated by the calculation for the Mn³⁺ state, although the experimentally observed intensity of the spectra taken at $h\nu = 640.6$ eV is quite weaker than that of the calculation. This will be more than confirmed by the absence of the spectral weight in this region for both the experimental RIXS spectra of MnO and the calculations for the Mn^{2+} state. The electronic structure of $Ga_{1-r}Mn_rAs$ is clearly different from that of MnO and the Mn²⁺ state. It should be noted that the profiles of the RIXS spectrum for the annealed film are also close to the calculated spectrum for the Mn^{3+} state. The spectral change induced by the annealing may be attributed to the relative decrease of Mn atoms in interstitial sites because annealing decreases the number of Mn in interstitial sites relative to the substitutional site [25,33]. Interstitial Mn defects are expected to act as double donors and have a similar electronic structure to the CI calculations for Mn²⁺. The relative decrease of Mn atoms in interstitial sites induced by annealing may also cause the peak at $E_L = 2$ eV to be more pronounced. The results suggest that the doped Mn atoms in bulk $Ga_{1-x}Mn_xAs$ are predominantly in the formally trivalent Mn³⁺ state. In addition, delocalized charge-transferred states $(3d^5L)$ and $3d^6L^2$) are dominant in the electronic configurations of this Mn³⁺ ground state (the weight of configurations in this state are $3d^4:3d^5\underline{L}:3d^6\underline{L}^2:3d^7\underline{L}^3 = 9\%:68\%:21\%:1\%)$ and this Mn³⁺ ground state is inherently different from the Mn²⁺ ground state with a localized d^5 state hybridized with a d^6L state. The situation is analogous to the creation of the Zhang-Rice singlet state [34] in cuprate superconductors: the spin of a hole on the ligand t_{2q} molecular orbital is antiferromagnetically coupled with that of the Mn²⁺ ion with the ${}^{6}A_{1}$ $(t_{2a}^{3}e_{q}^{2})$ symmetry, forming a Mn³⁺-ion-like ground state with the ${}^{5}T_{2}$ symmetry. On the other hand, to make the $3d^{4}$ configurations dominant, the value of Δ must be at least larger than 6 eV, which fails to explain the XAS and RIXS spectra. From the cluster analysis, we also found that the weak peak at $E_L \sim 0.8$ eV, which is clearly seen in both in the theoretical Mn^{3+} and experimental spectra with $h\nu = 640.6$ eV in Fig. 2(c), originates from a $3d^5L$ state decoupled with the $3d^4$ states, i.e., an excited state, where the spin of the ligand hole is not coupled with that of the Mn^{2+} state with the ${}^{6}A_{1}$ symmetry. This indicates that the Mn³⁺-ion-like hole-bound state is more stable than the Mn^{2+} state and the binding energy of the ligand hole is around 0.8 eV. Thus, the Mn L_3 RIXS spectra are very sensitive to the valence state of Mn and our results conclusively show that the charge-transferred states are dominant in the ground state of Ga_{1-x}Mn_xAs.

Based on our experimental findings, we discuss the detail of the Mn 3*d* electronic states in Ga_{1-x}Mn_xAs. The transport properties of Ga_{1-x}Mn_xAs have been reported to show a peak in the vicinity of T_C accompanied with a metal-insulator transition with metallic behavior below the peak (see Fig. S1 in the Supplemental Material [35]) [36,37]. A theoretical model based on bound magnetic polarons, in which Mn ions weakly bind holes [38], has qualitatively well explained this transport property. Because the occupied Mn 3d levels are located below the As 4p ligand band [30] and the present results suggest that the hybridization between the 3d orbital and the As 4p ligand is strong enough, the Anderson impurity model predicts the formation of a split-off state above the valence-band maximum [14]. The split-off state acts as an acceptor level and leads to the formation of the impurity band [14], in which the Mn ion weakly binds a hole, which may be the origin of the bound magnetic polarons. Our RIXS results show that the charge-transferred states are dominant in the Mn ground state and indicate the presence of a ligand hole bound to the Mn configuration. The present interpretation of the Mn electronic structure described by the Anderson impurity model is consistent with the theoretical model based on the bound magnetic polarons [38]. This arrangement will significantly reduce ionized impurity scattering and support the presence of quantum states in $Ga_{1-x}Mn_xAs$ quantum wells observed by resonant tunneling spectroscopy [10,39].

The RIXS spectra are now well reproduced by the CI cluster-model calculation using a common parameter set with XAS and XMCD. However, the experimental RIXS profile is still broader than the calculated profile with 0.17 eV broadening because of the experimental resolution. To evaluate the additional broadening of the Mn d-d excitation, the CI calculations are further broadened. The calculated RIXS spectra with additional broadening are shown in Fig. 3 and broadened spectra with a full width at half maximum of 0.5 eV reproduce the experimental RIXS spectra relatively well. According to Ref. [27], the RIXS spectra can be broadened because of stronger hybridization between *d*-*d* excitations and particle-hole pairs in the host valence and conduction bands itself. However, this effect should be accompanied by the distortion of the overall RIXS profile [27] and may fail to explain either the experimental RIXS or XAS/XMCD spectra. Another reason could be a double exchange interaction between the d orbitals [40] but this hardly contributes to the broadening of the Mn 3d states because hole carriers have a *p*-type ligand character as discussed above. Alternatively, we can attribute the additional broadening to a lifetime broadening in the RIXS final state where fast decay of the d-d excitation into an electronhole pair in the host valence and conduction bands occurs through the hybridization between the Mn 3d state and the As 4p ligand band. To account for this, we have applied Lorentzian, rather than the Gaussian, broadening to the RIXS spectra as shown in Fig. 3. The additional broadening implies two steps for the RIXS decay process. First, the resonantly excited core-hole state decays into a d-d excitation involving a localized bound hole state (split-off state) because of the strong hybridization between the Mn 3d and As 4p orbitals. Then, through the interaction with the As 4pband, the *d*-*d* excitation decays into the electron-hole pair in



FIG. 3 (color online). Lorentzian broadening of the RIXS spectra calculated by the CI cluster model. The experimental RIXS spectra of the annealed sample are compared with the calculations for the Mn^{3+} configuration convolved by Lorentzian line shapes. The full width at half maximum values of the Lorentzian broadening are 0.25, 0.50, and 0.75 eV. The elastic peaks in the calculated spectra are removed before the broadening. The calculated spectra are plotted with a baseline offset for clarity.

the host valence and conduction bands. With an increasing energy resolution, RIXS may provide information about the time evolution of the decay process, and expand the applications of RIXS in the near future.

In conclusion, we have conducted Mn L_3 RIXS and XAS measurements on $Ga_{1-x}Mn_xAs$ (x = 0.04) thin films to investigate the electronic structure of the doped Mn ions. The XAS spectrum taken in the TFY mode includes a few contributions from extrinsic surface Mn oxides, suggesting that the TFY-XAS spectra largely reflect the bulk electronic structure of $Ga_{1-x}Mn_xAs$. The RIXS spectra, which also reflect bulk sensitive information, show a broad profile even at energy resolutions high enough to distinguish individual *d*-*d* excitation peaks of MnO. An analysis by the CI calculations indicates that the Mn ground states mainly consist of the Mn³⁺ electronic configuration composed of the charge-transferred states $(d^5\underline{L} \text{ and } d^6\underline{L}^2)$, in which the ligand hole is weakly bound to the Mn $3d^5$ state, rather than the pure Mn^{2+} state. In order to reproduce the experimental broadening of the Mn d-d excitation, not only the (Gaussian) broadening by the energy resolution but also the 0.5 eV Lorentzian broadening is required. The additional Lorentzian broadening in the RIXS spectra can be attributed to the lifetime broadening in the final state of the RIXS process where fast decay of the *d*-*d* excitations to an electron-hole pair in the host valence and conduction bands occurs because of the hybridization between the Mn 3d orbital and the ligand band.

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