

Laser Heating Control with Polarized Light in Isolated Multiwalled Carbon Nanotubes

Mariusz Zdrojek,* Jarosław Judek, and Michał Wąsik

Faculty of Physics, Warsaw University of Technology, Koszykowa 75, 00-662 Warsaw, Poland

(Received 27 January 2012; published 29 May 2012)

We propose a novel method of laser heating control only through change in polarization of the incident light, keeping its power density constant. The idea combines an antenna effect found in isolated multiwalled carbon nanotubes and the possibility of their heating by light illumination. To observe this we used the Raman spectroscopy technique, where the heating manifests itself in a pronounced downshift of the Raman G and $2D$ lines as a function of the polarization angle. Our method can be useful in field electron emission devices or in selective nanotubes heating and destruction. It can also be extended to other one dimensional nano-objects, if only certain conditions are fulfilled.

DOI: 10.1103/PhysRevLett.108.225501

PACS numbers: 61.48.De, 61.80.Ba, 78.20.Ci, 84.40.Ba

In the infrared and visible regime, carbon nanotubes can act as optical antennae. Particularly, they can exhibit the polarization antenna effect that suppresses their optical response when the electric field of the incoming radiation is polarized perpendicular to the nanotube axis. Usually, the response stands for the elastic [1,2] or inelastic [3–5] light scattering; however, it can be generalized to other effects caused by the light illumination, as we demonstrate in this Letter.

Our experiment starts from the identification of the antenna effect in isolated multiwalled carbon nanotubes (MWCNTs) using polarized Raman spectroscopy. We do this by analyzing the G and $2D$ Raman feature intensities as a function of the angle φ between the nanotube axis and the polarization of incident light. Comparison to bundles [6,7] and a reference to the resonance conditions [8] suggest that signal intensity suppression can originate in a depolarization effect [9,10]. Further analysis of the G and $2D$ modes shows that neither their shapes nor positions are affected by the change in the φ angle for low laser power densities (LPDs). For higher LPDs, mode positions begin to shift toward lower frequencies because laser light absorption induces thermal expansion of the sp^2 lattice [11–16]. However, these shifts for fixed power densities are φ dependent. This means that we are able to observe a rise in the local temperature that depends only on the polarization of the incident light due to the anisotropic optical absorption. To our knowledge, this is the first experimental demonstration of polarization dependent heating effect in carbon nanotubes probed by Raman spectroscopy or by any other technique. Potential application includes the control of a nanotube work function in field electron emission devices by a change in the local temperature [17]. By tuning the laser power density, it is also possible to destroy MWCNTs ordered in a selected direction keeping the remaining tubes unaffected.

Carbon nanotubes with a diameter $d = 15\text{--}30$ nm were grown using the chemical vapor deposition technique and dispersed on the Si/SiO₂ substrate. In our experiment, we

used a Dilor XY-800 Raman spectrometer equipped with a microscope in order to acquire a spectra from individual carbon nanotubes [see Fig. 1(a)]. A polarizer, an analyzer and a $\lambda/2$ retardation plate were used to control the direction of the electric field vector of the incoming and outgoing light. All experiments were performed in the backscattering geometry under ambient conditions. For the excitation, a 514 nm line from an Ar⁺ laser was used. Calibrated laser power on the sample was changed from 0.7 up to 20 mW. In order to verify any variation of laser power on the sample as a function of light polarization, silicon oriented substrates (100 and 111) for polarization calibration have been used. The laser spot focused with the $100\times$ objective had a diameter of ~ 2 μm ; thus, the power density varied from ~ 20 to ~ 600 kW/cm^2 .

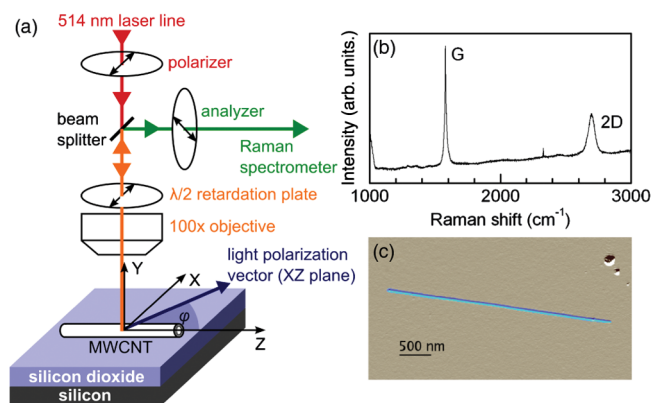


FIG. 1 (color online). (a) Schematic of the experimental setup used for exploring the dependence of the inelastic scattering amplitude and phonon energy on the angle φ between the carbon nanotube axis and the direction of the electric field vector of the incident e_i and scattered e_s light. For the VV configuration $e_i \parallel e_s$, for the VH configuration $e_i \perp e_s$. (b) Raman spectrum from an isolated multiwalled nanotube. (c) Atomic force microscopy image of isolated MWCNT ($d_{\text{isolated}} = 30$ nm) on the SiO₂/Si substrate.

A typical MWCNT Raman spectrum is presented in Fig. 1(b), whereas a typical atomic force microscopy image in Fig. 1(c).

Figure 2 shows the G and $2D$ band intensity dependencies on the angle φ between the carbon nanotube axis and the direction of the electric field vector of the incident e_i and scattered e_s light for VV ($e_i \parallel e_s$) and VH ($e_i \perp e_s$) configurations. Experimental data in Figs. 2(a) and 2(c) (VV configuration) are well described by the $\cos^4\varphi$ function. Data in Figs. 2(b) and 2(d) (VH configuration) are described by $\sin^2 2\varphi$. Band intensities on all four graphs reach their minimum when the incident or scattered light polarization vector is perpendicular to the nanotube axis ($\varphi_{\min} \sim 90^\circ$ for both configurations, for VH also $\varphi_{\min} \sim 0^\circ$). It means that inelastic photon scattering occurs mainly for the light polarized along the tube axis, exactly as in a dipolar antenna. The detailed comparison between our results for the isolated multiwalled nanotube and bundles of MWCNTs [6] reveals two significant differences. First, the Raman bands are described only by one Lorentzian function and the changes in their shape are never observed. It means that the inner diameter of our tube is too large for splitting of the G mode. Second, the minimum of the G band intensity has shifted from

$\varphi_{\min} \sim 55^\circ$ for bundles to $\varphi_{\min} \sim 90^\circ$ in our case. This shift can be explained by selective suppression of Raman scattering. To show this, we express the line intensity I using the concept of the Raman tensor \mathbf{R}

$$I \propto |\mathbf{e}_i \cdot \mathbf{R} \cdot \mathbf{e}_s|^2. \quad (1)$$

Literature [6,7] provides the following form of the Raman tensor for a G mode

$$\mathbf{R} \propto \begin{pmatrix} -1/2 & 0 & 0 \\ 0 & -1/2 & 0 \\ 0 & 0 & 1 \end{pmatrix}. \quad (2)$$

Such a tensor well describes the experimental data for the MWCNT bundles [6], despite the fact that it is derived from nonresonant Raman scattering [7]. Now, we introduce the positive attenuation ratio $c < 1$ for light polarized perpendicular to the nanotube axis that implies

$$\mathbf{R} \propto \begin{pmatrix} -c/2 & 0 & 0 \\ 0 & -c/2 & 0 \\ 0 & 0 & 1 \end{pmatrix}. \quad (3)$$

Next, we can write the equation for a normalized band intensity for the VV configuration

$$\frac{I^{VV}(\varphi)}{I^{VV}(0^\circ)} = \left(\cos^2(\varphi) - \frac{c}{2} \sin^2(\varphi) \right)^2. \quad (4)$$

When there is no attenuation, $I^{VV}(90^\circ)/I^{VV}(0^\circ) = 1/4$ and $\varphi_{\min} = \arccos(1/\sqrt{3}) \sim 55^\circ$ [6]. In the case of strong attenuation, $I^{VV}(90^\circ)/I^{VV}(0^\circ) \rightarrow 0$ and $\varphi_{\min} \rightarrow 90^\circ$. Experimental data show that we are in the second regime. Cautious estimation of the attenuation ratio c yields values below 0.25 and $I^{VV}(90^\circ)/I^{VV}(0^\circ)$ is below 0.04. The exact value of c is difficult to obtain due to the very low signal intensity for $\varphi \sim 90^\circ$ and not precise enough polarization control.

Raman scattering from multiwalled carbon nanotubes in the visible range is resonant; therefore, we attribute attenuation of the Raman scattering in the perpendicular direction to the suppression of the absorption. Discrepancies between our results and those obtained for the bundles can be explained by the depolarization effect due to stronger localization of electronic states in the perpendicular direction ($d_{\text{isolated}} \sim 30 \text{ nm} \ll d_{\text{bundle}} \sim 1 \mu\text{m}$) [10] and staying within the electrostatic limit [18]. According to Marinopoulos [10], the suppression of an optical absorption occurs only in the completely isolated nano-objects because when the tubes are interacting with each other, the electronic states start to delocalize, depolarization weakens, and the system is similar to a bulk metal. Another possible explanation was presented by Zhang [18], who found that strong anisotropy due to the depolarization observed for nanowires disappears when their diameters approach the value of the incident light wavelength. In our case, the electrostatic limit was not reached ($d_{\text{isolated}} \sim 30 \text{ nm} \ll \lambda \sim 500 \text{ nm}$), whereas, for instance, the limit

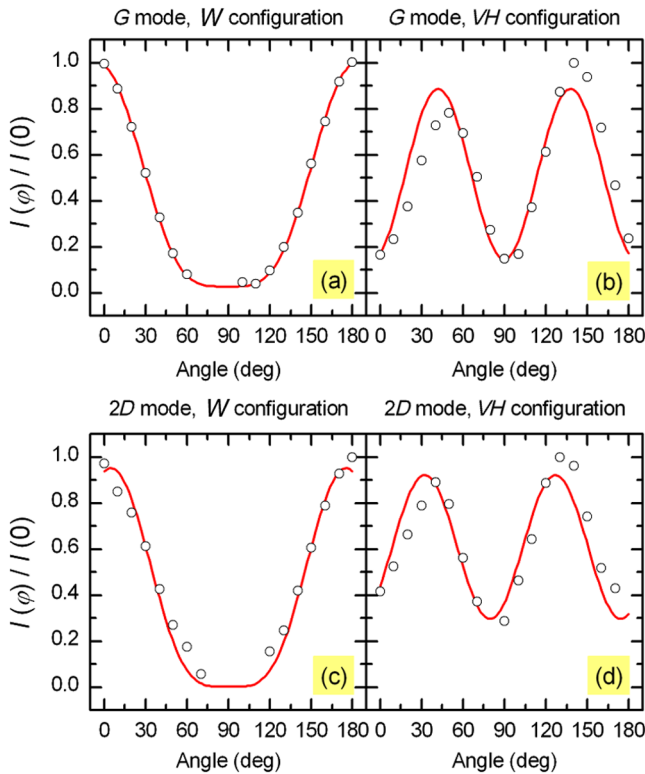


FIG. 2 (color online). Normalized integrated intensity for VV and VH configuration for G (a), (b) and $2D$ (c), (d) bands. Experimental data (open symbols) follow theoretical predictions (lines). Unequal maxima and weaker extinction of intensity for VH configuration are attributed to the light polarization misalignment [19].

for the multiwalled bundle was greatly exceeded ($d_{\text{bundle}} \sim 1000 \text{ nm} \gg \lambda \sim 500 \text{ nm}$). The polarization effect was also observed in the single-walled nanotubes [3–5]. However, here, the antenna behavior seems to be an effect of the specific resonance conditions that are different for parallel and perpendicular directions and change with the photon energy [8]. Generally, in single-walled carbon nanotubes, the intensity ratio, $I^{VV}(90^\circ)/I^{VV}(0^\circ)$, can assume values larger or smaller than 1, depending on the type of resonance. The disagreement with the expected depolarization effect is explained as a characteristic resonance feature in the single-walled nanotubes. In our experiment, there are no specific resonance conditions because the electronic structure of a nanotube with such a large diameter resembles the electronic structure of a rolled graphite.

Angular dependencies of the G and $2D$ band positions for both VV and VH configuration are presented in Fig. 3. The Raman shift $\hbar\omega$ depends on the φ angle between the electric field vector and the nanotube axis and is well described by the cosine squared function:

$$\hbar\omega(\varphi) = \hbar\omega(90^\circ) - A\cos^2(\varphi). \quad (5)$$

For the G band $\hbar\omega(90^\circ) = 1582 \text{ cm}^{-1}$, for the $2D$ band $\hbar\omega(90^\circ) = 2700 \text{ cm}^{-1}$, A stands for maximal change in

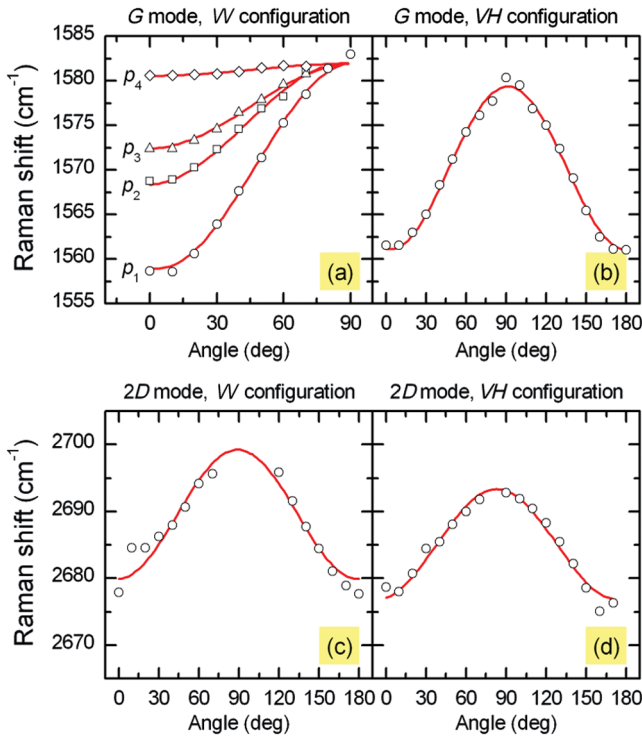


FIG. 3 (color online). Angular evolution of G and $2D$ band positions for both polarization configurations. Four data series in a plot (a) acquired for the different laser power densities ($p_1 > p_2 > p_3 > p_4$) prove the thermal origin of the Raman shift change. Experimental data (open symbols) were fitted with the $\cos^2(\varphi)$ function (lines).

Raman shift obtained for $\varphi \sim 0^\circ$ and is different for the G and $2D$ band. Figure 3(a) depicts the influence of a laser power density p on the value of A . Higher LPD implies larger changes in peak positions. For the lowest power p_4 , only slight peak shifts are observed. Moreover, the shapes of the G and $2D$ bands can be described by single Lorentzian functions showing no effects on the band symmetry and only their centers are φ dependent. Therefore, it is neither a symmetry-related evolution nor an artifact. In Figs. 3(b)–3(d) only results for one laser power density were shown for sake of clarity. We note that in the existing literature on the polarization dependence of the Raman lines for carbon nanotubes there is no indication of frequency shifts.

Interpretation of the observed phenomena is based on the fact that the position of the G band depends on the local sample temperature [11–16]. Local temperature depends on the efficiency of the laser heating that is the function of absorption and thermal conductivity. Because the latter is constant (we illuminate the same nanotube all the time), the experimentally obtained dependence of the G band position reflects the relation between the MWCNT absorption rate and the polarization angle. This means that we harness the strong anisotropy of optical absorption to control locally the sample temperature which is observed as the angular evolution of the position of the G and $2D$ modes. The results obtained in a VH configuration resemble those for the VV because heating by light absorption depends only on the polarization of the incident light and not on the scattered.

We now turn our attention to the evaluation of the optical anisotropy by conducting two experiments. In the first one, the φ angle was changed gradually and at each step the laser power density was increased in order to keep the G mode intensity constant [Fig. 4(a)]. Thus, normalized laser power density $p(\varphi)/p(0^\circ)$ can be expressed as an inverse

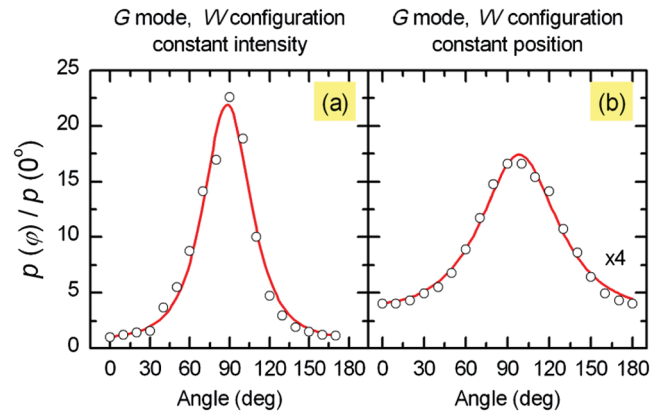


FIG. 4 (color online). Normalized laser power density required to keep the G mode intensity (a) and position (b) constant. Experimental points (open symbols) are assisted by theoretical curves (lines). In the right picture mode position 1570 cm^{-1} was set.

Eq. (4). Maximal power (reaching 600 kW/cm^2) was used for $\varphi \sim 90^\circ$, and we got $p(90^\circ)/p(0^\circ) = 22.6$. This value corresponds with the data shown in Fig. 2(a). In the second experiment, light polarization was set along the nanotube axis and the laser power density was adjusted to value $p(0^\circ)$ for which the G mode position was 1570 cm^{-1} . Afterwards, φ was increased and at each point we compensated p to keep the G mode position constant [Fig. 4(b)]. Normalized LPD in this case should follow:

$$\frac{p(\varphi)}{p(0^\circ)} = [\cos^2(\varphi) + c_{\text{pos}}\sin^2(\varphi)]^{-1}, \quad (6)$$

where c_{pos} stands for attenuation of absorption for the direction perpendicular to the nanotube axis. The fit of Eq. (6) to experimental points [Fig. 4(b)] yields $c_{\text{pos}} = 0.23$. Whereas value of $p(\varphi)/p(0^\circ)$ ratio, while keeping the G mode intensity constant [Fig. 4(a)], is equivalent to the typical polarized Raman study presented in Fig. 2, the evaluation of the LPD increase, while keeping the G mode position constant [Fig. 4(b)], allows us to observe directly the anisotropy of optical absorption. In details, Raman scattering is a third order process including the photon absorption and emission. In order to keep the line intensity constant, the laser power should be adjusted due to change both in the photon absorption and emission. To keep the G mode position constant, LPD had to be adapted only to the decrease in the absorption coefficient. In addition, it should be noted that the c_{pos} coefficient can be underestimated because for such high laser power densities, the vicinity of the MWCNT can also be heated thus lowering heat outflow.

Next, we estimated the local temperature arising in the nanotube upon polarized light illumination. According to the literature data for multiwalled nanotubes [11], the temperature coefficient for G mode is $\chi_G = -0.028 \text{ cm}^{-1}/\text{K}$. In our experiment, the maximal change in Raman shift equaled 24 cm^{-1} . This corresponds to an 860°C temperature rise in the material. Such temperatures should be treated with care because the tubes have different χ depending on the growth technique and amount of defects. Despite that, we observed that several nanotubes were damaged for LPD above 200 kW/cm^2 . For the tubes that have been damaged, we observed a substantial decrease in the Raman bands intensity (G , $2D$) and the appearance of the defect induced D peak.

In summary, we observed the antenna effect in the polarized Raman spectra on the isolated multiwalled nanotubes. The discrepancies between the isolated and bundled MWCNTs are attributed to the depolarization effect. Our study reveals that the change in the position of the Raman modes upon the rotation of the light polarization is not symmetry but heating-related. We ascribe this behavior to the optical absorption anisotropy. We expect that the effect presented here can be found in other high aspect ratio nano-objects, if only localization of the electronic states

is high enough [10] and/or they stay within the electrostatic limit [18].

The authors acknowledge support from the Foundation for Polish Science (Homing Plus). The authors also thank A. Bachtold and M. J. Esplandiu (Barcelona) for the help in the fabrication of the samples, and L. Wirtz and R. Bacewicz for a critical reading of the manuscript.

*zdrojek@if.pw.edu.pl

- [1] Y. Wang, K. Kempa, B. Kimball, J.B. Carlson, G. Benham, W.Z. Li, T. Kempa, J. Rybczynski, A. Herczynski, and Z.F. Ren, *Appl. Phys. Lett.* **85**, 2607 (2004).
- [2] K. Kempa, J. Rybczynski, Z. Huang, K. Gregorczyk, A. Vidan, B. Kimball, J. Carlson, G. Benham, Y. Wang, A. Herczynski, and Z. Ren, *Adv. Mater.* **19**, 421 (2007).
- [3] J. Hwang, H.H. Gommans, A. Ugawa, H. Tashiro, R. Haggemueller, K.I. Winey, J.E. Fischer, D.B. Tanner, and A.G. Rinzler, *Phys. Rev. B* **62**, R13310 (2000).
- [4] G.S. Duesberg, I. Loa, M. Burghard, K. Syassen, and S. Roth, *Phys. Rev. Lett.* **85**, 5436 (2000).
- [5] A. Jorio, A.G. Souza Filho, V.W. Brar, A.K. Swan, M.S. Unlu, B.B. Goldberg, A. Righi, J.H. Hafner, C.M. Lieber, R. Saito, G. Dresselhaus, and M.S. Dresselhaus, *Phys. Rev. B* **65**, 121402R (2002).
- [6] A.M. Rao, A. Jorio, M.A. Pimenta, M.S.S. Dantas, R. Saito, G. Dresselhaus, and M.S. Dresselhaus, *Phys. Rev. Lett.* **84**, 1820 (2000).
- [7] R. Saito, T. Takeya, T. Kimura, G. Dresselhaus, and M.S. Dresselhaus, *Phys. Rev. B* **57**, 4145 (1998).
- [8] A. Jorio, M.A. Pimenta, A.G. Souza Filho, G.G. Samsonidze, A.K. Swan, M.S. Unlu, B.B. Goldberg, R. Saito, G. Dresselhaus, and M.S. Dresselhaus, *Phys. Rev. Lett.* **90**, 107403 (2003).
- [9] H. Ajiki and T. Ando, *Physica B (Amsterdam)* **201**, 349 (1994).
- [10] A.G. Marinopoulos, L. Reining, A. Rubio, and N. Vast, *Phys. Rev. Lett.* **91**, 046402 (2003).
- [11] F. Huang, K. Yue, P. Tan, S. Zhang, Z. Shi, X. Zhou, and Z. Gu, *J. Appl. Phys.* **84**, 4022 (1998).
- [12] H.D. Li, K.T. Yue, Z.L. Lian, Y. Zhan, L.X. Zhou, S.L. Zhang, Z.J. Shi, Z.N. Gu, B.B. Liu, R.S. Yang, H.B. Yang, G.T. Zou, Y. Zhang, and S. Iijima, *Appl. Phys. Lett.* **76**, 2053 (2000).
- [13] N.R. Raravikar, P. Keblinski, A.M. Rao, M.S. Dresselhaus, L.S. Schadler, and P.M. Ajayan, *Phys. Rev. B* **66**, 235424 (2002).
- [14] L. Ci, Z. Zhou, L. Song, X. Yan, D. Liu, H. Yuan, Y. Gao, J. Wang, L. Liu, W. Zhou, G. Wang, and S. Xie, *Appl. Phys. Lett.* **82**, 3098 (2003).
- [15] M.Z. Atashbar and S. Singamaneni, *Appl. Phys. Lett.* **86**, 123112 (2005).
- [16] I.-K. Hsu, R. Kumar, A. Bushmaker, S.B. Cronin, M.T. Pettes, L. Shi, T. Brintlinger, M.S. Fuhrer, and J. Cumings, *Appl. Phys. Lett.* **92**, 063119 (2008).
- [17] C.M. Tan, J.J. Jia, and W.B. Yu, *Appl. Phys. Lett.* **86**, 263104 (2005).

- [18] J. Zhang, A.A. Lutich, J. Rodriguez-Fernandez, A.S. Susha, A.L. Rogach, F. Jackel, and J. Feldmann, *Phys. Rev. B* **82**, 155301 (2010).
- [19] In real optical setups, direction of the electric field vector of laser light can be diffused around some value.

Therefore, the function describing angular evolution of the Raman feature intensities should be convoluted with, e.g., the Gaussian distribution. Particularly, data in Figs. 2(b) and 2(d) do not have zero values for $\varphi \sim 0^\circ$ and $\sim 90^\circ$ due to the polarization misalignment.