Observation of a Continuous Phase Transition in a Shape-Memory Alloy

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Elastic neutron-scattering, inelastic x-ray scattering, specific-heat, and pressure-dependent electrical transport measurements have been made on single crystals of AuZn and $Au_{0.52}Zn_{0.48}$. Elastic neutron scattering detects new commensurate Bragg peaks (modulation) appearing at $Q = (1.33, 0.67, 0)$ at temperatures corresponding to each sample's transition temperature ($T_M = 64$ and 45 K, respectively). Although the new Bragg peaks appear in a discontinuous manner in the $Au_{0.52}Zn_{0.48}$ sample, they appear in a continuous manner in AuZn. Surprising us, the temperature dependence of the AuZn Bragg peak intensity and the specific-heat jump near T_M are in favorable accord with a continuous transition. A fit to the pressure dependence of T_M suggests the presence of a critical end point in the AuZn phase diagram located at $T_M^* = 2.7 \text{ K}$ and $p^* = 3.1 \text{ GPa}$.

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A class of materials exhibiting martensitic (diffusionless) phase transformations yields properties used in a range of technological applications, including implants to increase flow in restricted blood vessels [[1](#page-3-0)], actuators for the treatment of high myopia $[2]$, voltage generators $[3]$ $[3]$, and orthodontic arch wires [\[4](#page-3-0)]. These properties often depend on the history of the material and may allow it to recover its previous shape after deformation, known as the shape-memory effect. It has long been recognized that these transformations are all thermodynamically first order (discontinuous) [\[5–7\]](#page-3-0). Special cases arise when the order parameter is coupled to an external field in a complicated way, leading to a weakly first-order transition [\[8](#page-3-0)], which is believed to result from a complicated coupling between strain and order parameter fluctuations. Because shapememory alloys and other multiferroic materials owe their functionality to complicated cross-field responses between two (or more) pairs of conjugate thermodynamic variables, a description of the free-energy landscape and the ability to predict functionality become one and the same.

Phenomenological descriptions of the martensitic transformation pathway, based on reciprocal space [[9\]](#page-3-0) and real space [\[10\]](#page-3-0) geometries, have established rules to determine relative twin and crystallographic orientations between the austenite (high-temperature) and martensite (lowtemperature) modifications. Knowledge of the symmetry breaking allows for a definition of the energetic driving force in terms of the difference in free energies between phases. In the Ginzburg-Landau (GL) approach, the free energy G is expressed as a sum of symmetry invariants. In its simplest form, G is approximated by a polynomial expansion in even powers of an order parameter, Q, as $G(Q) = aQ^2 + bQ^4 + cQ^6 + \frac{8}{2}|\nabla Q|^2$. For martensitic
transformations Q is taken to be strain or strain counled transformations, Q is taken to be strain or strain coupled
to shuffle (displacements involving quasistatic phonons) to shuffle (displacements involving quasistatic phonons with fractional commensurate wave vectors with uniform shears $[11]$). The coefficients a, b, c, and g are material parameters to be determined experimentally or by firstprinciple simulations.

For the majority of shear-induced transformations, the GL expansion captures the essential physics [\[12,13\]](#page-3-0), such as constitutive response $[14,15]$ $[14,15]$ $[14,15]$ $[14,15]$ $[14,15]$ and the occurrence of antiphase boundaries [\[16\]](#page-3-0). When applied to shape-memory alloys, the presence of intervening (premartensitic) phases [\[17\]](#page-3-0) presents difficulties for both experiment and theory as to a unique characterization of the order parameter. Further issues have arisen in materials exhibiting quantum mechanical effects in the band structure and strong electronphonon coupling [[18](#page-3-0)]. The thermodynamic behavior is best explored when no further transitions occur between T_M and 0 K. This effect was recently reported for Fe-doped NiTi, where all thermodynamic parameters could be correlated self-consistently within mean-field theory [\[19\]](#page-3-0). In order to integrate these cases into GL theory, it is necessary to measure and access each thermodynamic property and infer cross-field couplings of the order parameter. The ability to distinguish subtle differences between first order and weakly first order depends on the ability to resolve the behavior of pertinent physical properties.

In this Letter we examine the AuZn system for thermodynamic properties that contribute to the first-order or weakly first-order nature of the free-energy landscape. We measure elastic neutron scattering, inelastic x-ray scattering, specific heat, and the pressure dependence of the transition by electrical transport. While we observed classical first-order behavior at 45 K for $Au_{0.52}Zn_{0.48}$, we observed a mean-field continuous transformation for AuZn near 64 K in the temperature dependence of the Bragg peak intensity, and a Landau step with a λ anomaly in the excess specific heat.

Single crystals were prepared by fusion of the elements in a Bridgman furnace and were oriented by backreflection Laue diffraction. The neutron experiments were performed on the BT9 triple-axis spectrometer at the NIST research reactor. Measurements of the phonon dispersion were made on X-ray Operation and Research of sector 3 at the Advanced Photon Source, Argonne National Laboratory [\[20,21\]](#page-3-0). Specific-heat measurements were measured using a thermal-relaxation calorimeter by Quantum Design [[22\]](#page-3-0). The pressure dependence of the transition temperature was made by a four-terminal ac-transport method in a mechanical pressure cell designed to reach pressures of 3 GPa.

Previous investigations have shown that AuZn exhibits a martensitic transition as seen with low-temperature electron microscopy [[23](#page-3-0)] and by recoverable transformation strain (shape-memory effect) [[24](#page-3-0)]. Figure 1 shows the temperature dependence of (a) the excess specific heat and (b) the satellite-peak intensity for $Au_{0.52}Zn_{0.48}$ and AuZn. Although we have measured the specific heat previously [\[25\]](#page-3-0), it is plotted along with the elastic neutronscattering data, since it is revealing to compare both sets of measurements that were made on the same samples. We find a latent heat anomaly at 45 K for Au_{0.52}Zn_{0.48}. A λ anomaly was observed for AuZn, but no latent heat was observed. The specific heat near the transition can be influenced by intrinsic defects, including triple-point and antistructure defects, both known to be present in the B2 structure [\[26,27\]](#page-3-0). It was therefore necessary to use a microscopic probe to determine whether the transition could be a broadened first-order transition. Elastic neutron-scattering measurements indicate that in each composition new commensurate Bragg peaks (modulation) appear at $Q =$ $(1.33, 0.67, 0)$ at temperatures corresponding to each sample's T_M [Fig. 1(b)]. Mirroring the specific-heat data, the temperature dependence of the satellite-peak intensity shows a rapid jump at 45 K in $Au_{0.52}Zn_{0.48}$, while a continuous variation occurs for AuZn on warming starting at 64 K. The satellite-peak intensity is proportional to the square of the order parameter, since it leads to the lowtemperature rhombohedral phase. The specific-heat data also indicate thermal hysteresis (not shown) of 6 K at 45 K for $Au_{0.52}Zn_{0.48}$. The satellite-peak intensity for AuZn depicted in Fig. $1(b)$ was measured by neutrons on heating

FIG. 1. (a) The excess specific heat divided by temperature versus temperature in the vicinity of the martensitic transitions for $Au_{0.52}Zn_{0.48}$ and AuZn. (b) Temperature dependence of the satellite-peak intensity along $Q = (1.33, 0.67, 0)$. The satellite-peak intensity is proportional to the square of the order parameter.

and cooling. Unlike the specific-heat data [Fig. $1(a)$], we observe a small discontinuity and 1.5 K difference in transition temperature on cooling.

Figure 2 shows the satellite-peak intensity versus temperature in the vicinity of T_M for AuZn. Near the transition temperature, the satellite peak evolves continuously with increasing temperature, showing a temperature dependence given by $I = A|T - T_M|^2$, where the satellite-
neak intensity is I the critical temperature is T_{tot} and the peak intensity is I, the critical temperature is T_M , and the critical exponent is B. We obtain $T_M = 64.08 \text{ K} + 0.11 \text{ K}$ critical exponent is β . We obtain $T_M = 64.08 \text{ K} \pm 0.11 \text{ K}$

FIG. 2. Temperature dependence of the satellite-peak intensity on heating along $Q = (1.33, 0.67, 0)$ in AuZn, near the transition temperature. A fit to mean-field theory for $T \leq T_c$. The points above 64 K are background.

FIG. 3. Temperature dependence of the phonon dispersion in AuZn along the TA_2 branch. At 300 K one notices considerable phonon softening at $q = (0.33, 0.33, 0)$ (vertical line). The dashed lines between points are provided as guides to the eye.

and $2\beta = 1.02 \pm 0.08$. The value of the critical exponent
obtained from this fit ($\beta = 0.51$) is close to the mean-field obtained from this fit ($\beta = 0.51$) is close to the mean-field
value of $\beta = 0.50$ [28]. One could argue that mean-field value of $\beta = 0.50$ [[28](#page-3-0)]. One could argue that mean-field
exponents were obtained by coincidence given that the exponents were obtained by coincidence given that the critical regime, as defined by the Ginzburg criterion, is restricted to a range of temperatures too small to be resolved by our experiments, or that the correlation length does not diverge, since it is limited by other extrinsic disorder (i.e., martensite twins) or intrinsic fluctuations. The latter possibility seems more plausible to us, because the inverse correlation length, or width of the Bragg peaks (not shown), increases as the temperature is lowered below T_M .

As the temperature is lowered, the unit cell is modulated in the [110] shear direction. A commensurate shuffle of every third unit cell results in a hexagonal primitive unit cell formed from nine primitive cubic cells of the parent phase. This structure can also be described in terms of its conventional rhombohedral unit cell. In Fig. 3 we show the TA_2 phonon-dispersion curves along [110], measured by inelastic x-ray scattering at temperatures of 300 and 200 K. The inflection in the phonon frequency near $\xi = 1/3$ indicates the low shear instability along the $TA_2[110]$ branch. This inflection is comparable to an earlier investigation [\[29\]](#page-3-0) of off-stoichiometric AuZn samples. At 300 K, the phonon energy positions have been fit over the energy range $-4 \text{ meV} \leq E \leq 4 \text{ meV}$ to a double
Lorentzian as shown in Fig. 4. The phonons continuously Lorentzian, as shown in Fig. 4. The phonons continuously soften with decreasing temperature to the point where separation becomes difficult below 200 K.

Figure 5(a) shows the change in resistivity between cooling and warming. The data were collected with warming and cooling rates of 0.2 K min⁻¹. In the region of T_M
there is no hysteresis, to within experimental error there is no hysteresis, to within experimental error. Because of the lack of hysteresis, we elected to measure the pressure dependence of the transition. For a first-order transition, the Clausius-Clapeyron relation, $\ln p/p_0 =$ $\ln p/p_0$ versus $1/T$ with the slope $(-\Delta H/R)$. Here, p
denotes the pressure H the enthalpy R the universal gas $-\Delta H/(RT) + A$, predicts a linear relationship for denotes the pressure, H the enthalpy, R the universal gas

FIG. 4 (color online). Inelastic x-ray scattering data showing the temperature dependence near the soft mode, $q =$ $(0.35, 0.35, 0)$. The 300 and 200 K data are offset by 0.004 and 0.0005, respectively, on the y axis for clarity. The energy positions are fit to a double Lorentzian (solid curve) at 300 K to determine the phonon energy. The intrinsic resolution for inelastic x-ray scattering is 2 meV.

constant, and p_0 is a reference pressure. Figure $5(b)$ shows the plot of lnp versus $1/T$. It appears that there is no observable linear region. Assuming that the entropy does not vary significantly with pressure, the conditions for the applicability of the Clausius-Clapeyron relation are not met. This result provides further evidence that the transition is not first order.

We use a modified GL free-energy expression to describe the quantum saturation in the order parameter expected at sufficiently low temperature [[30](#page-3-0)]. This saturation effect reflects the departure from classical behavior as the absolute temperature approaches zero and is a direct consequence of the third law of thermodynamics. Following Salje et al. [\[30\]](#page-3-0), we fit the pressure dependence of

FIG. 5. (a) The electrical-resistivity data taken on cooling and warming is shown in the vicinity of T_M . (b) Plot of lnp versus $1/T$ to test the applicability Clausius-Clapeyron equation.

FIG. 6 (color online). Pressure dependence of the transition temperature up to pressures of nearly 3 GPa, together with a fit to Salje's model [30] which accounts for quantum saturation effects close to zero absolute temperature. We find that quantum effects become important for temperatures lower than $\theta_s =$ 48.3 \pm 3.7 K, and a critical end point is predicted at $T_M^* =$
2.7 K and $n^* = 3.1$ GPa 2.7 K and $p^* = 3.1$ GPa.

the transition as $\theta_s/T_M(p) = \coth^{-1}\{\coth[\theta_s/T_M(0)] - p/n_s\}$. Here θ is a phenomenological temperature below. p/p_0 . Here θ_s is a phenomenological temperature below which quantum mechanical effects are dominant. We find $p_0 = 5.6 \pm 0.7$ GPa and $\theta_s = 48.3 \pm 3.7$ K. The resulting phase diagram predicts a critical end point at $T_M^* = 2.7$ K
and $n^* = 3.1$ GPa. Figure 6 depicts the experimental presand $p^* = 3.1$ GPa. Figure 6 depicts the experimental pressure dependence of the transition temperature for pressures up to nearly 3 GPa, together with the fit to Salje's classical regime. We find a classical slope of -11.7 ± 0.2 K GPa⁻¹.
The difference between the classical and quantum regimes The difference between the classical and quantum regimes is important for temperatures below θ_s .

In conclusion, we report the results of a detailed experimental study of the martensitic transition in AuZn. While the temperature dependence of the satellite-peak intensity (proportional to the square of the order parameter) shows marked first-order behavior at 45 K for $Au_{0.52}Zn_{0.48}$, we observed a continuous feature well described by a meanfield exponent ($\beta = 0.51$) for AuZn at 64 K. This result is contrary to the established definition of a martensitic trancontrary to the established definition of a martensitic transition. Providing supporting evidence to the continuous nature of the phase transition in AuZn is the lack of thermal hysteresis from electrical-resistivity measurements, the λ anomaly in the specific heat, and the disagreement of pressure data with the Clausius-Clapeyron relation. Using Salje's model to describe the quantum saturation effects close to zero absolute temperature, we predict the presence of a critical end point in the phase diagram located at $T_M^* =$ 2.7 K and $p^* = 3.1$ GPa, with a phenomenological tem-
perature $\theta = 48.3 + 3.7$ K for the onset of quantum perature $\theta_s = 48.3 \pm 3.7$ K for the onset of quantum effects.

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