Dramatic slowing of compositional relaxations in the approach to the glass transition for a bimodal colloidal suspension

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Molecular dynamics simulation was used to study a model colloidal suspension with two species of slightly different sized colloidal particles in an explicit solvent. In this work we calculated the four interdiffusion coefficients for the ternary system, which were then used to calculate the decay coefficients D_{\pm} of the two independent diffusive modes. We found that the slower *D*[−] decay mode, which is associated with the system's ability to undergo compositional changes, was responsible for the long-time decay in the intermediate scattering function. We also found that a decrease in *D*_− to negligible values at a packing fraction of $\Phi_g = 0.592$ resulted in an extreme slow-down in the long-time decay of the intermediate scattering function often associated with the glass transition. Above Φ_g , the system formed a long-lived metastable state that did not relax to its equilibrium crystal state within the simulation time window. We concluded that the inhibition of crystallization was caused by the inability of the quenched fluid to undergo the compositional changes needed for the formation of the equilibrium crystal.

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I. INTRODUCTION

The nature of the glass transition remains a matter of enduring interest [\[1\]](#page-11-0). Glasses share similarities with crystalline solids, since they are both mechanically rigid, but also with liquids because they both have similar disordered structures at the molecular level [\[2\]](#page-11-0). A glass can be obtained by cooling a liquid to below its glass transition temperature or for a colloidal suspension by condensing it to a packing fraction above the glass transition. For the system to form a glass, the quench must be fast enough that the first-order phase transition toward a crystalline structure is avoided. Because the glass transition depends on the quench rate, it differs from thermodynamic phase transitions such as the solid-liquid transition.

Colloidal systems are ideal for studying vitrification (glass formation) as their size and diffusive dynamics ensure that their relaxation times are experimentally accessible. They are also one of the simplest experimental systems known to have a glass transition $[3]$. In many cases, colloidal particles can be considered to be simple hard spheres. This was first experimentally demonstrated in the mid-1980s by Pusey and van Megen, who essentially replicated the theoretical hardsphere phase diagram using colloidal samples [\[3\]](#page-11-0). A single component hard-sphere (HS) system (where all spheres are the same size) exists as a liquid up to a packing fraction of $\Phi \leqslant$ 0.494 and coexists as liquid and solid between $0.494 < \Phi \leq$ 0*.*545, above which it is a solid [\[4,5\]](#page-11-0). Interestingly, a small amount of polydispersity (particles with slightly different sizes) efficiently inhibits crystallization $[6–10]$. This is the case for experimental suspensions, which can be compressed above the freezing transition without crystal growth. At a packing fraction $\Phi_g \approx 0.57{\text -}0.59$ the relaxation times become sufficiently large compared to experimental time scales that the system does not relax, and it forms a glass $[3,11]$.

We found that at finite wave vectors the intermediate scattering function could be modelled as a double exponential decay with effective short- and long-time diffusion coefficients [\[13\]](#page-11-0). This is in line with experimental analysis [\[17\]](#page-11-0), though our definition of the short-time diffusion coefficient differs from the usual convention. Usually it is defined as the zero time limit of the time-dependent diffusion coefficient; however, this can be ambiguous and difficult to determine, so instead we isolated the two exponential decay modes to determine their individual amplitudes and decay coefficients and related the short-time diffusion coefficient to the decay rate of the faster decaying mode. The conclusions drawn in Ref. [\[13\]](#page-11-0) are unchanged if the zero time limit definition is used. However, the model studied did not include polydispersity, which is key to inhibiting crystallization and allowing a glass transition to occur. Therefore, one of the aims of this work is to add polydispersity into the existing model and study the change in dynamics leading up to the glass transition rather than the freezing point.

In previous work, we studied a unimodal model colloidal suspension with an explicit solvent $[12,13]$. In this system the colloidal particles were modeled using a Weeks-Chandler-Anderson (WCA) potential that was modified to include a hard core, while the solvent was modeled using a simple WCA potential. The explicit solvent was included in the model in an attempt to match the dynamics of a real colloidal suspension by having the larger particles diffuse through a solvent having viscosity and inertia rather than moving ballistically through a vacuum. However, our attempt to include hydrodynamic interactions was only partially successful as the trend in the self-diffusion coefficients with packing fraction is qualitatively similar to that observed in Brownian dynamics simulations [\[13\]](#page-11-0). We calculated the intermediate scattering function (which is a key quantity measured in dynamic light-scattering experiments $[7,8,11,14-16]$ over a large range of packing fractions and wave vectors in order to systematically study the change in dynamics on the approach to the freezing point.

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The simplest way to include polydispersity is by introducing a second colloidal species with a slightly different size. Binary HS systems show extremely rich phase behavior which depends on the size ratio α and composition of the two species [\[18\]](#page-11-0). Kranendonk and Frenkel [\[19\]](#page-11-0) studied the freezing and melting points of binary HS systems in the range $0.85 \le \alpha \le 1.00$. They found that for an equimolar composition the freezing and melting points were slightly shifted from the single component HS system. Binary HS mixtures have also been found to form a glass [\[20–23\]](#page-11-0), which makes them an ideal system to study vitrification through calculation of the intermediate scattering function.

The expected form for the intermediate scattering function in the macroscopic diffusive limit (infinite wavelength) for a ternary colloidal suspension has been derived by Pusey *et al.* [\[24\]](#page-11-0). They predicted the existence of two diffusive modes with decay rates related to the four bulk interdiffusion coefficients. These two modes have been observed in colloidal suspensions with polydispersity $[25,26]$ and in ternary polymer solutions [\[27\]](#page-11-0), but their link to the glass transition has never previously been explored.

It is possible that the slower *D*[−] mode, which is associated with the system's ability to make compositional changes [\[24\]](#page-11-0), may be connected to the inhibition of crystallization at large packing fractions. This follows from the observation made in both simulations [\[22,28\]](#page-11-0) and experiments [\[7–9,16,29,30\]](#page-11-0) that compositional fluctuations control crystallization in polydisperse HS systems. It is this possibility that will be explored in this work through quantitative calculations of the D_{+} coefficients of the two independent modes.

The outline of this paper is as follows: First, we give a summary of the computational model used and describe how we calculated the interdiffusion coefficients in equilibrium molecular dynamics (MD) using time correlation functions. Then we discuss results for the four bulk interdiffusion coefficients, as well as the decay rates of the two independent modes, over a range of packing fractions from a moderate packing fraction up to a supercompressed metastable state. We then use a multiexponential analysis of the colloidal particle intermediate scattering function over the same range of packing fractions to show that the slow diffusive mode governs its long-time decay.

Last, we show results for the intermediate scattering function of a system above the glass transition packing fraction. We demonstrate that the extremely long relaxation time of the fluid, which manifests as an extremely slow decay and ultimate arrest of the intermediate scattering function, is linked to the reduction in the decay coefficient *D*[−] of the slow mode to a negligible value. Once this link is made, it allows us to show that the growth in relaxation time observed at the glass transition is just a continuation of the trend established at lower packing fractions. It also allows a physical interpretation of the cause of the glass transition and goes some way to explaining why polydispersity is essential to the formation of a glass in multicomponent HS systems.

II. THEORY

A. Intermediate scattering function

The intermediate scattering function $F_{\alpha\beta}(k,\tau)$ is defined as the normalized autocorrelation function of a Fourier component of the number density:

$$
F_{\alpha\beta}(k,\tau) = \frac{1}{N} \frac{\langle n_{\alpha}(\mathbf{k},\tau)n_{\beta}^{*}(\mathbf{k},0)\rangle}{S_{\alpha\beta}(k)},
$$
(1)

where $n_{\alpha}(\mathbf{k},t)$ is given as

$$
n_{\alpha}(\mathbf{k},t) = \sum_{j=1}^{N_{\alpha}} \exp(-i\mathbf{k}\cdot\mathbf{r}_j(t))
$$
 (2)

and *N* is the total number of particles, N_{α} is the number of particles of species α , and $\mathbf{r}_i(t)$ denotes the position of particle *j* at time *t*. The static structure factor $S_{\alpha\beta}(k)$ is defined as

$$
S_{\alpha\beta}(k) = \frac{1}{N} \langle n_{\alpha}(\mathbf{k}, 0) n_{\beta}^*(\mathbf{k}, 0) \rangle, \tag{3}
$$

where $*$ signifies the complex conjugate. In MD simulation, the wave vector **k** being studied must be consistent with the periodic boundary conditions of the simulation box:

$$
\mathbf{k} = \frac{2\pi}{L}(a_1, a_2, a_3),\tag{4}
$$

where a_i is an integer and L is the length of the simulation box (in this work the box is cubic so $L_x = L_y = L_z$).

From Eq. (4) we see that the lowest nonzero *k* value that can be studied in an MD simulation has a magnitude of $|\mathbf{k}_{\text{min}}| = 2\pi/L$. As the fluid is isotropic the correlation functions $F_{\alpha\beta}(k,\tau)$ and $S_{\alpha\beta}(k)$ only depend on the magnitude $k = |\mathbf{k}|$, so an average is done over all **k** of equal magnitude.

In this work we have two species of colloidal particles with a size (diameter) ratio of 0.925. This is done in order to introduce a 3.9% polydispersity into our model. In order to make comparisons with experimental work, the intermediate scattering functions that were calculated included all colloidal particles. Therefore we calculated the property:

$$
F_c(k,\tau) = \frac{1}{N} \frac{\langle n_c(\mathbf{k},\tau) n_c^*(\mathbf{k},0) \rangle}{S_c(k)},
$$
\n(5)

where the subscript *c* indicates that both colloidal species are included in the calculation. But, in order to simplify the notation, the subscript *c* will be dropped, and we will simply denote $F(k, \tau) = F_c(k, \tau)$.

B. Macroscopic diffusive limit

In the macroscopic diffusive limit $(k \to 0)$, the decay rate of the colloidal particle intermediate scattering function $F(k, \tau)$ can be related to the diffusion coefficients of the fluid. This is because small fluctuations about equilibrium in the colloidal particle number densities δn_1 and δn_2 of long wavelength can be described by the coupled linear diffusion equations [\[24\]](#page-11-0):

$$
\frac{\partial}{\partial t}\delta n_1(\mathbf{r},t) = D_{11}\nabla^2 \delta n_1(\mathbf{r},t) + D_{12}\nabla^2 \delta n_2(\mathbf{r},t),
$$

\n
$$
\frac{\partial}{\partial t}\delta n_2(\mathbf{r},t) = D_{21}\nabla^2 \delta n_1(\mathbf{r},t) + D_{22}\nabla^2 \delta n_2(\mathbf{r},t),
$$
\n(6)

where the D_{ij} are the four interdiffusion coefficients of the ternary system and convective terms have been neglected. The matrix of diffusion coefficients can be diagonalized to provide two independent diffusional modes which are linear combinations of δn_1 and δn_2 if we define:

$$
\delta n_{+} = \alpha_{+} \delta n_{1} + \delta n_{2},
$$

\n
$$
\delta n_{-} = \alpha_{-} \delta n_{1} + \delta n_{2},
$$
\n(7)

which relax according to

$$
\frac{\partial}{\partial t}\delta n_{+}(\mathbf{r},t) = D_{+}\nabla^{2}\delta n_{+}(\mathbf{r},t),
$$

$$
\frac{\partial}{\partial t}\delta n_{-}(\mathbf{r},t) = D_{-}\nabla^{2}\delta n_{+}(\mathbf{r},t).
$$
 (8)

Here

$$
\alpha_{\pm} = \frac{(D_{11} - D_{22}) \pm [(D_{11} - D_{22})^2 + 4D_{12}D_{21}]^{1/2}}{2D_{12}} \tag{9}
$$

and

$$
D_{\pm} = \frac{1}{2}(D_{11} + D_{22}) \pm \frac{1}{2}[(D_{11} - D_{22})^2 + 4D_{12}D_{21}]^{1/2},
$$
\n(10)

where D_{\pm} are the effective diffusion coefficients of the two independent diffusive modes. The fluctuations in the total number density of colloidal particles $δn_c$ can therefore be written as

$$
\delta n_c(\mathbf{r},t) = \delta n_1(\mathbf{r},t) + \delta n_2(\mathbf{r},t)
$$

=
$$
\frac{1 - \alpha_-}{\alpha_+ - \alpha_-} \delta n_+(\mathbf{r},t) + \frac{1 - \alpha_+}{\alpha_+ - \alpha_-} \delta n_-(\mathbf{r},t).
$$
 (11)

Thus the space-time correlation function is

$$
\langle \delta n_c(0,0)\delta n_c(\mathbf{r},t)\rangle = \left[\frac{1-\alpha_-}{\alpha_+-\alpha_-}\right]^2 \langle \delta n_+(0,0)\delta n_+(\mathbf{r},t)\rangle + \left[\frac{1-\alpha_+}{\alpha_+-\alpha_-}\right]^2 \langle \delta n_-(0,0)\delta n_-(\mathbf{r},t)\rangle.
$$
\n(12)

The colloidal particle intermediate scattering function is simply the Fourier transform of this correlation function [\[24\]](#page-11-0):

$$
\lim_{k \to 0} F(k,t) = \left[\frac{1 - \alpha_-}{\alpha_+ - \alpha_-} \right]^2 \langle |\delta n_+(k,0)|^2 \rangle \exp(-k^2 D_+ t)
$$

$$
+ \left[\frac{1 - \alpha_+}{\alpha_+ - \alpha_-} \right]^2 \langle |\delta n_-(k,0)|^2 \rangle \exp(-k^2 D_- t).
$$
\n(13)

Therefore, Eq. (13) predicts that in the $k \to 0$ limit the decay of the colloidal particle intermediate scattering function will be the sum of two exponentials with decay rates proportional to diffusion coefficients D_{+} that are calculated from a combination of the four bulk interdiffusion coefficients using Eq. (10).

A physical interpretation of the two independent diffusive modes was given by Pusey $[24]$. The $+$ mode describes a collective compression-dilation motion of the particle mixture in which the relative compositions remain unchanged (i.e., $\delta n_1/n_1 = \delta n_2/n_2$). The – mode describes compositionfluctuation dynamics in which species 1 and 2 are exchanged under the preservation of a constant total number density (i.e., $\delta n_1 + \delta n_2 = 0$).

The $+$ mode (compression-dilation) is expected to relax on a much quicker time scale than the − mode (compositional fluctuations). This is because relaxations in the composition of the fluid occur on a much longer time scale than relaxations in the total density. Compositional relaxations are also expected to slow drastically as the packing fraction of the colloidal particles increases. Therefore, we expect a divergence of the two time scales on the approach to the glass transition.

In this work we focus on studying the decay of $F(k, \tau)$ in the $k \to 0$ limit (or as close to it as is computationally feasible). In this hydrodynamic limit, other theories such as mode coupling theory (MCT) [\[31\]](#page-11-0) and multicomponent self-consistent generalized Langevin equation theory [\[32\]](#page-11-0) should agree with hydrodynamic theory and will therefore be consistent with our analysis in terms of *D*⁺ and *D*−. Outside the hydrodynamic limit, the additional predictions [such as the nonexponential stretching in $F(k, \tau)$] should appear.

C. Calculation of interdiffusion coefficients

The four interdiffusion coefficients given in Eq. [\(6\)](#page-1-0) can be calculated from equilibrium MD simulations using a combination of the Green-Kubo [\[33,34\]](#page-11-0) and Kirkwood-Buff theories [\[35\]](#page-11-0). The calculation of the diffusion coefficients follows the same method used in our previous work for a binary system [\[13\]](#page-11-0), though here we generalize the method for a ternary system. The diffusion coefficients for a threecomponent isothermal fluid appearing in Eq. [\(6\)](#page-1-0) are defined by the linear flux-force relations [\[36\]](#page-11-0):

$$
\mathbf{J}_1^0 = -D_{11}\nabla n_1 - D_{12}\nabla n_2,
$$

\n
$$
\mathbf{J}_2^0 = -D_{21}\nabla n_1 - D_{22}\nabla n_2,
$$
\n(14)

where J_{α}^0 is the diffusive molecular flux defined as

$$
\mathbf{J}_{\alpha}^{0} = n_{\alpha}(\mathbf{v}_{\alpha} - \mathbf{v}^{0}). \tag{15}
$$

Here \mathbf{v}_{α} is the velocity of species α and \mathbf{v}^0 is the volume average streaming velocity given by

$$
\mathbf{v}^0 = \sum_{\alpha=1}^3 n_\alpha v_\alpha \mathbf{v}_\alpha, \qquad (16)
$$

where v_α is the partial molecular volume of species α .

The diffusion coefficients appearing in Eq. [\(6\)](#page-1-0) are defined in terms of the flux measured relative to the volume average streaming velocity, but it is more convenient for computation to calculate the phenomenological coefficients defined by flux laws expressed in terms of the barycentric (mass average) streaming velocity. The relationship between the phenomenological coefficients with respect to one reference velocity and the diffusion coefficients measured relative to another can be written as [\[36\]](#page-11-0)

$$
\mathbf{D} = T^{-1}\mathbf{B} \cdot \mathbf{L} \cdot \mathbf{A} \cdot \mathbf{\Gamma},\tag{17}
$$

where T is the temperature. Γ is the two-dimensional matrix with elements

$$
\Gamma_{ik} = \left(\frac{\partial \mu_i}{\partial n_k}\right)_{p,T,n_{j\neq k}},\tag{18}
$$

where μ_i is the chemical potential of species *i* and *p* is the pressure. The **L** matrix is also two-dimensional and contains the phenomenological coefficients defined in the relationship

$$
\mathbf{J}_i = \sum_{k=1}^2 L_{ik} \mathbf{X}_k, \tag{19}
$$

where X_k are the thermodynamic forces [\[36\]](#page-11-0). The **A** matrix comes about when eliminating a flux using the Gibbs-Duhem equation. It relates the independent forces to gradients in the chemical potential

$$
\mathbf{X}_{i} = -\sum_{k=1}^{2} A_{ik} \frac{(\nabla \mu_{k})_{p,T}}{T}
$$
 (20)

and for fluxes measured relative to the barycentric reference velocity takes the form of

$$
A_{ik} = \delta_{ik} + \frac{c_i}{c_3} \frac{x_k}{x_i},\tag{21}
$$

where x_i is the number fraction of species *i* and c_i is its mass fraction. Matrix **B** is required to convert between the diffusive flux relative to the barycentric velocity, and the flux relative to the volume averaged reference velocity. The elements of matrix **B** are given by [\[36\]](#page-11-0)

$$
B_{ik} = \delta_{ik} + \left(n_3 \nu_3 \frac{c_k}{c_3} - n_k \nu_k \right) \frac{c_i}{c_k}.
$$
 (22)

Therefore, to determine the diffusion coefficients, we need to determine all thermodynamic factors for Γ , the partial volumes *νi*, and phenomenological coefficients for **L**.

The phenomenological coefficients are calculated from Green-Kubo theory [\[33,34\]](#page-11-0), which states that

$$
L_{\alpha\beta} = \frac{V}{3k_B} \int_0^\infty \langle \mathbf{J}_\alpha(\tau) \cdot \mathbf{J}_\beta(0) \rangle d\tau,\tag{23}
$$

where due to time-reversal symmetry, $L_{\alpha\beta} = L_{\beta\alpha}$. The thermodynamic factors and partial volumes can be calculated from Kirkwood-Buff theory [\[35\]](#page-11-0). The well-known expression found by Kirkwood and Buff states that for a fluid with *m* species the thermodynamic factors are calculated from

$$
\frac{1}{k_B T} \Gamma_{\alpha\beta} = \frac{C^{\alpha\beta} \sum_{i,j=1}^{m} n_i n_j C^{ij} - \sum_{i,j=1}^{m} n_i n_j C^{i\alpha} C^{j\beta}}{|\mathbf{C}| \sum_{ij=1}^{m} n_i n_j C^{ij}}
$$
(24)

and the partial volumes from

$$
\nu_{\alpha} = \frac{\sum_{j=1}^{m} n_j C^{\alpha j}}{\sum_{i,j=1}^{m} n_i n_j C^{ij}}.
$$
\n(25)

The matrix **C** is constructed from the elements

$$
C_{\alpha\beta} = n_{\alpha} n_{\beta} G_{\alpha\beta} + n_{\alpha} \delta_{\alpha\beta} \tag{26}
$$

and the quantity $C^{\alpha\beta}$ denotes the cofactor of the element $C_{\alpha\beta}$ in the determinant |**C**|. The volume integrals of the radial distribution functions $G_{\alpha\beta}$ are calculated from

$$
G_{\alpha\beta} = \int (g_{\alpha\beta}(r) - 1) dr = 4\pi \int r^2 (g_{\alpha\beta}(r) - 1) dr, \quad (27)
$$

where $g_{\alpha\beta}(r)$ is the radial distribution function of species α and β . These integrals can be difficult to calculate as statistical error in $g_{\alpha\beta}(r)$ at large *r* is magnified by the factor of r^2 , so the numerical integrals may not converge. As was shown in the previous work $[13]$, a much simpler way to calculate this quantity is through the partial structure factors:

$$
G_{\alpha\beta} = \frac{1}{x_{\alpha}x_{\beta}n} \left[\lim_{k \to 0} S_{\alpha\beta}(k) - x_{\alpha} \delta_{\alpha\beta} \right],\tag{28}
$$

where *n* is the total number density of all species. By calculating the low-*k* values of the partial structure factors *S_{αβ}*(*k*), and extrapolating $k \to 0$, the values of $G_{\alpha\beta}$ can be calculated in a much simpler way.

III. SIMULATION METHODS

The pair potential and parameters for our model are similar to those described in our previous work $[13]$, but here we expand the model to include a second colloidal species. We modeled the colloidal particles and solvent using a WCA potential (a shifted and truncated Lennard-Jones potential) which is modified to include a hard core. The potential takes the following form:

$$
\phi(r_{ij}) = \begin{cases}\n\infty & \text{if } r_{ij} \leq c_{\alpha\beta} \\
4\epsilon \left[\left(\frac{\sigma}{r_{ij} - c_{\alpha\beta}} \right)^{12} - \left(\frac{\sigma}{r_{ij} - c_{\alpha\beta}} \right)^6 \right] + \epsilon & \text{if } c_{\alpha\beta} < r_{ij} < c_{\alpha\beta} + 2^{1/6}, \\
0 & \text{otherwise}\n\end{cases}
$$
\n(29)

where r_{ij} is the center-to-center distance between the particles *i* and *j*, ϵ is the depth of the potential well, and σ is the nominal length scale of the potential (in this work all quantities are expressed in reduced units where $\epsilon = \sigma = 1$). The $c_{\alpha\beta}$ parameter introduces a hard core to the potential where *α* and β are the two interacting species. This creates an excluded region that increases the size of the colloidal particles relative to the solvent and also gives different sizes to the colloid species. A diagram of the potential is shown in Fig. [1.](#page-4-0) The hard-core parameters we used in this work are given in Table [I.](#page-4-0)

In this model, the hard-core parameters $c_{\alpha\beta}$ in all colloidsolvent interactions are set to zero in order to remove the large excluded volume around the colloidal particles that led to strong depletion effects $[12,13]$. This means that the hydrodynamic interactions are still present, and momentum transfer can occur through the solvent, but these interactions may be weak compared to those found in experimental systems.

FIG. 1. Diagram of the pair potential between colloidal particles with $\epsilon = \sigma = 1.00$ and $c_{\alpha\beta} = 3.034$.

The colloid packing fraction Φ was calculated from

$$
\Phi = \frac{\pi}{6V} \left(N_1 d_1^3 + N_2 d_2^3 \right),\tag{30}
$$

where d_i is the diameter of the colloidal particles. In the usual hard-sphere simulations the diameter of the particle is clear, but in this work we are using a hard core plus a WCA repulsive potential. The WCA repulsive potential adds an extra contribution to the diameter so that the effective hard sphere diameter is not uniquely defined.

Hess *et al.* [\[37\]](#page-11-0) have determined expressions for the effective hard-sphere diameter of WCA particles as a function of temperature. These authors defined the effective diameter *d* to be the interparticle separation where the interaction potential is equal to Boltzmann's constant times the temperature $\phi(d)$ = $k_B T$. At the reduced temperature of 1.0 used in this work, this gives an extra diameter of 1.0 to the particles due to the WCA repulsion. This gives the colloidal species an effective diameter of 4.03 and 3.73 times the diameter of the solvent particles. Using the size ratio of the two colloidal species as $\gamma = 3.73/4.03 = 0.925$ the polydispersity *s* is given by [\[22\]](#page-11-0):

$$
s = \left(1 - \frac{2\gamma}{1 + \gamma}\right)100\%,\tag{31}
$$

which for the present system gives $s = 3.9\%$.

All simulations were done with a total of 108 000 particles. The calculated packing fraction Φ , number of particles and the volumes for the systems studied are shown in Table II.

The masses of the colloidal particles were set with the goal of making them approximately neutrally buoyant in the solvent. The mass needed to do this was calculated in the same way as done by McPhie [\[38\]](#page-11-0), which for a size ratio d_1/d_2 of 4.03 and 3.73 gave a mass ratio m_1/m_2 of 50 and 39.62, respectively. Therefore, we used a mass of 1.0 for the solvent

TABLE I. Hard-core parameter $c_{\alpha\beta}$ for each species interaction type.

Pair type	$c_{\alpha\beta}$
$C1-C1$	3.034
$C1-C2$	2.882
$C2-C2$	2.733
$C1-S$	0.000
$C2-S$	0.000
$S-S$	0.000

TABLE II. Number of solvent particles N_s , colloidal particles N_c , average volume $\langle V \rangle$, and packing fraction Φ for the systems studied.

System no.	$N_{\rm s}$	N_c	$\langle V \rangle$	Φ
	106910	690	128 670	0.261
2	106510	1090	128778	0.356
3	106 116	1490	129 131	0.449
4	105928	1884	129404	0.493
5	105728	2072	129 905	0.539
6	105628	2372	129812	0.563
7	105 528	2472	130764	0.582
8	105480	2520	131 162	0.592
9	105428	2572	131791	0.601
10	105 328	2672	133434	0.617
11	105728	2976	141788	0.635

particles and a mass of 50.0 and 39.62 for the two colloidal species. This size and mass ratio is significantly smaller than in an experimental colloidal suspension, but it has been shown to be large enough for the larger particles to behave as Brownian particles in a solvent [\[39\]](#page-11-0).

All simulations were run using the MD package LAMMPS [\[40\]](#page-11-0) and results were post-processed using in-house code. Simulations at each packing fraction were done under NPT conditions at a reduced temperature of 1.00 and reduced pressure of 7.85. The time integration scheme used follows the time-reversible measure-preserving Verlet integrator derived by Tuckerman *et al.* [\[41\]](#page-11-0) with a time step of 0.005. The temperature is held fixed using a Nosé-Hoover thermostat while the pressure is held fixed using a Nosé-Hoover type barostat, both with a damping parameter of 10. The simulations were done at constant temperature and pressure in order to better replicate the experimental conditions of a real colloidal suspension.

IV. RESULTS

A. Interdiffusion coefficients

In order to calculate accurate values for the interdiffusion coefficients, accurate values of the phenomenological coefficients $L_{\alpha\beta}$ and thermodynamics factors $\Gamma_{\alpha\beta}$ are needed. The phenomenological coefficients $L_{\alpha\beta}$ were calculated from the integral of the colloidal particle mass-flux correlation functions defined in Eq. (23) . The mass-flux correlation functions were calculated out to a maximum delay time of 25 000 time steps. Numerical integration with the trapezoid rule was performed on the correlation functions and the integrals were found to converge. The resulting values of the phenomenological coefficients are shown in Fig. [2.](#page-5-0)

For Φ < 0.36, all three coefficients are observed to change very little with Φ . But as the freezing point Φ_f is approached, L_{11} (circles) and L_{22} (diamonds) were both found to decrease while L_{12} (squares) increased. This trend continues into the metastable region $\Phi_f < \Phi < \Phi_m$ with no obvious change in behavior as the freezing point is crossed. This is quite different behavior to that observed in previous work for the unimodal system [\[13\]](#page-11-0). For the unimodal system, the single

FIG. 2. Plot of the phenomenological coefficients *L*¹¹ (circles), L_{22} (diamonds), and L_{12} (squares) calculated from Eq. [\(23\)](#page-3-0). Error bars have not been shown as they are smaller than the symbols.

phenomenological coefficient *L*¹¹ was found to increase, and then plateau, at the freezing point.

To determine the thermodynamic factors *αβ*, values for $G_{\alpha\beta}$ are needed. The integrals of the radial distribution functions $G_{\alpha\beta}$ were calculated from the zero-*k* values of the partial static structure factors $S_{\alpha\beta}$ using Eq. [\(28\)](#page-3-0). As an example of how the $S_{\alpha\beta}(k \to 0)$ were calculated, we have shown data for all $S_{\alpha\beta}(k)$ at $\Phi = 0.36$ in Fig. 3. These are plotted against k^2 , as $S_{\alpha\beta}(k)$ is even in *k*. Although it is possible that $S(k)$ could be a nonanalytic function of k , and could therefore also depend on odd or fractional powers of $|k|$, we saw no evidence of this in our data. Therefore to determine $S_{\alpha\beta}(k \to 0)$, a fifth-order polynomial in k^2 was fitted to the data and extrapolated back to $k = 0$. This was done for all packing fractions studied.

All values calculated for $S_{\alpha\beta}(k \to 0)$ were used in Eq. [\(28\)](#page-3-0) to calculate $G_{\alpha\beta}$. The values for $G_{\alpha\beta}$ were used in Eq. [\(24\)](#page-3-0) to calculate the thermodynamic factors $\Gamma_{\alpha\beta}$. Using the values of the thermodynamic factors $\Gamma_{\alpha\beta}$, along with the values of the phenomenological coefficients $L_{\alpha\beta}$ shown in Fig. 2, the four interdiffusion coefficients were calculated using Eq. [\(17\)](#page-2-0). These are shown in Fig. 4.

The main-term diffusion coefficients D_{11} and D_{22} relate the flux of each component to its own concentration gradient, while D_{12} and D_{21} are the cross-term diffusion coefficients relating the flux of each component to the concentration gradient of the other. At each given packing fraction the four interdiffusion coefficients are of the same order of magnitude, because of the comparable size of the two colloidal species. All four coefficients increase as Φ increases, similar to the behavior of the single interdiffusion coefficient of the unimodal colloidal suspension in our previous work [\[13\]](#page-11-0), which showed an almost exponential increase on the approach to the freezing point.

Although there are four interdiffusion coefficients in a ternary system, there are only two independent diffusion modes. The effective diffusion coefficients of these modes are designated by D_{\pm} and were calculated using Eq. [\(10\)](#page-2-0). These are shown in Fig. [5.](#page-6-0)

FIG. 3. Plot of the low-*k* values of the static structure factors of a system at a packing fraction of 0.36. A fifth-order polynomial line of best fit was used to obtain the $S_{\alpha\beta}(k \to 0)$ values.

As discussed earlier, the $+$ mode corresponds to compression-dilation at fixed composition, while the − mode corresponds to compositional changes at fixed total concentration. As expected, D_{+} is found to have a much larger

FIG. 4. Plot of the four interdiffusion coefficients D_{11} (circles), D_{12} (squares), D_{21} (diamonds), and D_{22} (triangles) against packing fraction Φ .

FIG. 5. Plot of the two independent diffusion coefficients D_{+} for the (a) + mode and (b) – mode. Circles show the values calculated from Eq. [\(10\)](#page-2-0) and crosses show the values calculated from fit to intermediate scattering function (ISF).

magnitude than $D_$ (roughly $100 \times$ larger), indicating that the total colloidal particle concentration relaxes on a much quicker time scale than the time scale on which compositional changes occur. D_+ also increases as Φ increases, showing that compression fluctuations relax faster with increased Φ . This is also seen in the single interdiffusion coefficient in the unimodal system [\[13\]](#page-11-0) as that mode also corresponds to relaxations in the total density of the colloidal particles.

However, *D*_− is observed to decrease as Φ increases, indicating that the composition is relaxing at a slower rate. This is consistent with the observations of Williams and coauthors [\[22,28\]](#page-11-0) who observed that compositional relaxation slowed down dramatically with increased Φ , which in turn inhibited crystallization. In this work we relate the relaxation directly to *D*−, and so are able to calculate the relaxation rate quantitatively. To determine the complete behavior of this coefficient at larger Φ , the values for *D*− were determined independently from fits to $F(k, \tau)$. These are also shown in Fig. 5 (crosses) for the stable and metastable fluids, as well as for a glassy state. Independent values for D_+ could not be calculated from the fits to $F(k, \tau)$ due to strong wave-vector dependence at low *k*.

The last intermediate scattering function data point in Fig. 5 corresponds to a packing fraction well above the melting point Φ_m , where the equilibrium state of the system is a crystal [\[42\]](#page-11-0). But, on the total simulation time scale, it was not observed to relax to its equilibrium crystal state. At this value of Φ , we estimate the magnitude of *^D*[−] to be *<*10−9, indicating that compositional changes can occur but will do so on an extremely long time scale. This appears to be obstructing the formation and growth of the colloidal crystal.

How $D_$ is determined from $F(k, \tau)$ will be shown in the next section. We will also show that the − mode corresponds

FIG. 6. Plot of $F(k, \tau)$ data (symbols) for the packing fractions at (a) $\Phi = 0.356$, (b) $\Phi = 0.449$, and (c) $\Phi = 0.493$ together with the multiexponential fit described in the text (red solid line through the data points). Data shows the low wave-vector $(kd = 0.55)$ decay. The various components of the fitting function given in Eq. [\(32\)](#page-7-0) are also displayed separately according to the legend. The damped cosine mode has also not been shown as its contribution is negligible on this scale. For graphical clarity, not all available data points have been displayed.

to the long-time decay of $F(k, \tau)$ and therefore the decrease of *D*_− to an insignificant value at Φ = 0.592 corresponds to an extremely slow decay in $F(k, \tau)$ that is often associated with structural arrest.

B. Intermediate scattering function

In this section, calculations of the colloidal particle intermediate scattering function $F(k, \tau)$ will be shown for the high-density stable fluid phase. The wave vectors studied were those consistent with the periodic boundaries of the simulation box [given in Eq. [\(4\)](#page-1-0) up to $n_1 = n_2 = n_3 = 15$]. We will report wave vectors in the dimensionless form *kd* where *d* is the average diameter of the colloidal particles $(d = 3.73)$. This allows direct comparison with corresponding *kd* values measured in light-scattering experiments.

 $F(k, \tau)$ calculated for this system is shown in Fig. 6 for three high-density fluid states at packing fractions Φ approaching the freezing point. A multiexponential analysis was applied to the $F(k, \tau)$ data in order to isolate the individual contributions. We expect there to be at least two decay modes that correspond to the \pm modes predicted from Eq. [\(13\)](#page-2-0), but additional modes appear away from the $k \to 0$ limit. These additional kinetic modes were observed in the unimodal system [\[13\]](#page-11-0) and can also be attributed to memory effects. This multiexponential analysis technique is similar to our previous work for the unimodal system [\[13\]](#page-11-0) and is ultimately based on the arguments of Barocchi and coauthors [\[43–45\]](#page-11-0). The fitting function takes the form:

$$
F(k,t) \approx A_+ \exp(-k^2 D_+ t) + A_- \exp(-k^2 D_- t)
$$

+ $A_K \exp(-k^2 D_K t)$
+ $A_d \exp(-\alpha t) \cos(-\beta t + \phi)$, (32)

where all coefficients are allowed to be wave-vector dependent. The first two terms in Eq. (32) are the two thermodynamic modes predicted from macroscopic nonequilibrium thermodynamics. The third term is a kinetic mode, which only appears outside the macroscopic diffusive limit (*k >* 0). The last term is a damped cosine which takes into account the very short-time nondiffusive decay of $F(k,t)$. This term is present in our system as the dynamics of the colloidal particles are not truly Brownian at short times, but its contribution is almost negligible. The total fit functions from Eq. (32), along with the individual contributions, are also shown in Fig. [6.](#page-6-0)

The $F(k, \tau)$ data shown in Fig. [6](#page-6-0) is at $kd = 0.55$. This is one of the lowest wave vectors that we could study that was still consistent with the periodic boundary conditions of the simulation box. In this low wave-vector limit, $F(k, \tau)$ is expected to approach a double exponential decay as the wavelengths being probed approach the macroscopic diffusive limit (infinite wavelength). This behavior is observed for the packing fraction of $\Phi = 0.356$ [Fig. $6(a)$] where the dominant contributions come from the + and − modes, and the other modes have negligible amplitudes.

For the higher packing fractions in the low-*k* region [Figs. $6(b)$ and $6(c)$], the + and − modes still dominate, but the kinetic mode is nonzero and has a greater contribution at short delay times. The kinetic mode makes a larger contribution because, at these packing fractions, the wave vector being studied is not low enough to be in the macroscopic diffusive limit. As Φ increases, the maximum wave vector where macroscopic diffusive behavior is seen decreases, just as was observed in the unimodal system [\[13\]](#page-11-0). But we will show later that $A_k \to 0$ in the $k \to 0$ limit for all Φ , as expected.

We have identified two of the modes in Fig. [6](#page-6-0) as the $+$ (blue dashed line) and − (purple dashed-dotted line) thermodynamic decay modes, which can be justified in the following way; First, these two modes have nonzero amplitudes in the $k \to 0$ limit identifying them as thermodynamic decay modes. Second, as we will show, their diffusion coefficients extrapolate in the $k \to 0$ limit to the D_{\pm} coefficients calculated independently from Green-Kubo and Kirkwood-Buff theory.

As seen in Fig. [6,](#page-6-0) the short-time decay is governed by the $+$ and kinetic modes, while at long times the decay is dominated by the − mode. Therefore, this long-time decay is due to extremely slow long-range compositional relaxations of the colloidal particles. As Φ increases, the amplitude of the − mode increases, while the diffusion coefficient decreases.

FIG. 7. Plot of the amplitudes of the $(a) +$, (b) kinetic, and (c) − exponential decay modes for packing fractions of 0.356 (circles), 0.449 (squares), and 0.493 (diamonds).

This indicates that compositional relaxations are becoming more prominent, while at the same time relaxing more slowly. But at these packing fractions, the magnitude of *D*−(*k*) is still nonzero, so $F(k, \tau)$ is still able to decay to zero within an accessible time scale. As was shown in Fig. [5,](#page-6-0) the decrease in *D*_− to negligible values at larger Φ results in a metastable fluid that cannot relax on any accessible time scale.

To display in more detail the complete behavior of the mode amplitudes and their dependence on packing fraction and wave vector, Fig. 7 shows the amplitudes of the three diffusive decay modes at three packing fractions. As previously observed for the unimodal system [\[13\]](#page-11-0), at low wave vectors the amplitude of the kinetic mode A_k approaches zero. This indicates that the decay of $F(k, \tau)$ is approaching a double exponential decay (as expected in the macroscopic diffusive limit). We should also note that in the limit in which the asymmetry of the two colloidal species vanishes, *A*[−] would also vanish [\[24\]](#page-11-0). Therefore, the extremely slow decay in $F(k, \tau)$ is not observed, and we also do not observe a glass transition [\[13\]](#page-11-0).

The relationship between A_+ and A_k is similar to that seen between corresponding modes A_L and A_s in the unimodal system [\[13\]](#page-11-0), but the amplitude of the new *A*[−] mode shows interesting wave-vector dependence. It has a maximum in the $k \rightarrow 0$ limit and then decreases towards zero at some finite wave vector. The gradual decrease in the contribution from this mode at larger wave vectors can be explained by the fact that, in the stable liquid phase, large wave-vector (or short wavelength) density fluctuations do not require compositional rearrangements in order to decay. But, as we will discuss in the next section, this is not true for systems above the glass transition packing fraction.

FIG. 8. Plot of low-*k* values of the effective diffusion coefficients for the (a) D_+ and (b) D_- modes. Arrows on y axis indicate the value of the diffusion coefficients calculated for the corresponding packing fractions. Data for D_+ has been shifted up by multiples of 0.5 for clarity.

Since the $k \to 0$ behavior of $F(k, \tau)$ should be given by the solution to the hydrodynamic description of Eq. [\(13\)](#page-2-0), the decay should be the superposition of two exponentials with diffusion coefficients equal to the two independent macroscopic diffusion coefficients defined in Eq. [\(10\)](#page-2-0). To check this, in Fig. 8 we have plotted the low-*k* values of $D_-(k)$ and $D_+(k)$ calculated from the fits (symbols) along with $D_$ and D_+ calculated from equilibrium MD using the Green-Kubo and Kirkwood-Buff theory (arrows).

The wave-vector dependence of $D_+(k)$ is similar to that observed for the long-time diffusion coefficient in the unimodal system [\[13\]](#page-11-0). In that work, we found that the wave-vector dependence could be fitted well with a Lorentzian type function of the form:

$$
D_{+}(k) = \frac{D_{+}}{1 + \alpha |k|^{\beta}},
$$
\n(33)

where the coefficients α and β are free fitting parameters that are not wave-vector dependent. Figure 8 shows that this functional form also fits the data for D_+ quite well over the range of wave vectors investigated at each packing fraction. Though we have not shown the values here, the α parameter was found to increase exponentially with packing fraction, while β remained relatively constant with an approximate value of 2. This is identical to the observed behavior of the corresponding mode in the unimodal system [\[13\]](#page-11-0).

Interestingly, over the range of wave vectors studied, *D*−(*k*) does not have any observable wave-vector dependence. The values calculated from the fit to $F(k, \tau)$ are also in good agreement with the bulk diffusion coefficient *D*[−] (as shown in Fig. [5\)](#page-6-0). Because *D*−(*k*) does not show any clear wave-vector dependence, the values used in Fig. [5](#page-6-0) are the average of *D*−(*k*)

over the range shown in Fig. 8. We can also conclude from the independence of $D_-(k)$ on wave vector that the time scale on which compositional fluctuations decay scales as k^2 for a given Φ .

In the stable liquid phase, the compositional fluctuations decay at a much slower rate than the total density fluctuations $(D_+ \gg D_-)$. But at these packing fractions the fluid is still able to relax to equilibrium within a reasonable experimental time scale after a quench. As we will see in the next section, *D*_− decreases to negligible values in densely packed fluids $(\Phi > 0.58)$. This means that the compositional changes, which are necessary for the formation of the equilibrium crystal [\[22,28\]](#page-11-0), cannot occur within a reasonable experimental time scale, resulting in an extremely long-lived metastable state.

C. Glass transition

Above the melting packing fraction of the solid, the thermodynamic force driving the fluid towards crystallization increases to the point where crystallization can occur within an experimental time frame. However, the development of an equilibrium crystal structure is hindered by the rate at which compositional changes can occur in regions of the fluid [\[7,8\]](#page-11-0). To study the effects of these two counteracting influences, extremely low density homogeneous fluids were quickly compressed using the barostat to a range of packing fractions above the melting point. The systems were then allowed to equilibrate to determine at which Φ crystallization would occur. To determine whether crystallization has occurred, the radial distribution function $g(r)$ was calculated over intervals of 500 000 time steps until no noticeable changes in the structure of the system could be observed. The final $g(r)$ calculated are shown in Fig. 9.

The systems in the range of packing fractions $0.563 \leqslant \Phi \leqslant$ 0.582 were all found to have completely crystallized. The $g(r)$ in these systems show a crystalline structure with the lattice planes clearly seen as sharp peaks. These peaks extend out to interparticle separations well beyond the first major peak, showing long-range order characteristic of a crystal. This phase

FIG. 9. Plot of the radial distribution function $g(r)$ for packing fractions above the melting point. Data for each Φ have been shifted up by multiples of 2 for clarity.

FIG. 10. Plot of the MSD for the larger species of colloidal particles at the packing fractions indicated in the legend. For clarity, not all data points have been shown.

behavior is consistent with the results from previous binary HS systems [\[19\]](#page-11-0).

For all $\Phi \geqslant 0.592$ the thermodynamic driving force towards a crystalline state has increased, but there is no clear onset of crystallization. In these systems, *g*(*r*) does not show clear lattice planes but instead has a structure that resembles a dense liquid. These systems have been allowed to equilibrate well beyond the time it took for the lower Φ systems to completely crystallize, indicating that for all packing fractions greater than $\Phi_g = 0.587 \pm 0.005$ there is a mechanism inhibiting crystallization and causing a long-lived metastable state to form. This value of Φ_{g} is close to the glass transition packing fraction observed in experimental systems with similar polydispersity [\[46\]](#page-11-0) as well as binary HS simulations [\[22\]](#page-11-0), but it remains to be seen whether the state points shown in Fig. [9](#page-8-0) have actually formed a glass or just a long-lived metastable liquid.

To determine whether glassy behavior is observed for $\Phi > \Phi_g$, the mean-squared displacement (MSD) was calculated for the larger species of colloidal particles at packing fractions ranging from a moderately dense stable liquid at $\Phi = 0.261$ to $\Phi = 0.635$ and is presented in the form of a double-logarithmic plot in Fig. 10. Diffusive motion can be identified by those regions where the data follow a straight line, which can be observed to occur at short times for all Φ . This short-time behavior corresponds to motion of the tagged colloidal particle through the solvent inside its local cage.

Linear behavior is also observed at long times for the systems with $\Phi < \Phi_g$, with the intermediate region showing the transition from short-range diffusion through the solvent to long-range diffusion through the solution. The intermediate region is caused by the interaction of the tagged colloidal particle with its neighboring colloidal particles. This intermediate time grows as Φ increases until the particles become trapped by their nearest neighbors, leading to the extreme slow-down and approach to a plateau of the MSD seen at at $\Phi \geqslant \Phi_g$. This slow-down has been observed in many experimental works on colloidal suspensions [\[46,47\]](#page-11-0) and indicates that although small-scale movement of the tagged particle inside its local cage is possible, large-scale diffusion is extremely unlikely within the experimental time window studied.

FIG. 11. Plot of $F(k, \tau)$ against τ for (a) at low wave vector $(kd = 1.62)$ and (b) at the structure factor peak, for the packing fractions shown in the legend. Data for each Φ has been shifted along the horizontal axis and for clarity not all data points have been shown.

The MSD reflects the single-particle dynamics, and so highlights the confinement of the tagged particle by its surrounding particles. It is also interesting to look at the behavior of the collective dynamics as seen through $F(k, \tau)$. The dramatic decrease in the *D*_− coefficient seen in Fig. [5](#page-6-0) manifests in $F(k, \tau)$ as an extremely slow decay at long times. In the stable fluid ($\Phi \le 0.493$) this is only observed at small *k*, as seen in Fig. [7.](#page-7-0) This was also true for the metastable fluid below the melting point ($\Phi = 0.539$) but it may not be true for metastable systems when $\Phi > \Phi_g$.

To determine whether this is the case, $F(k, \tau)$ was calculated for systems with $\Phi > \Phi_g$ and is shown in Fig. 11 on a logarithmic scale for decays at a low wave vector [Fig. $11(a)$] and at the structure factor peak [Fig. $11(b)$]. Also shown is the $F(k, \tau)$ for the stable liquid ($\Phi = 0.449$ and 0.493) and metastable liquid ($\Phi = 0.539$) below the melting point. Simulations were prepared by first compressing the systems to the given packing fraction. After compression, multiple consecutive simulations were run, with each lasting $10⁶$ time steps (5000 reduced time units). $F(k, \tau)$ was calculated for each run and comparisons were made to determine when the system had reached a steady state where changes in the decay curve between runs could not be readily observed. After this point, $F(k, \tau)$ was accumulated and averaged over approximately 50 (or more) individual simulation runs each lasting $10⁶$ time steps. However, we do note that since the systems with packing fractions above the glass transition are in a nonequilibrium long-lived metastable state, slow ageing of these systems may still occur [\[48\]](#page-11-0).

The low- k data in Fig. [11\(a\)](#page-9-0) for the Φ dependence of $F(k, \tau)$ shows a definite trend. The slow long-time mode that we associate with compositional relaxations is observed as a secondary step in the decay. It is present at all Φ and is seen to decay slower as Φ increases, corresponding to the decrease in *D*_− observed in Fig. [5.](#page-6-0) At $\Phi > 0.592$, $F(k, \tau)$ is still decaying at long times, but its decay rate is almost negligible. We estimate that at $\Phi > 0.592$ the value of *D*_− has decreased to *<*10−9, and at this rate of decay the time scale on which compositional relaxations would completely decay is well beyond the accessible simulation time.

Light-scattering experiments are limited in the lowest wave vector that can be achieved (for an exception, see Martinez *et al.* [\[14\]](#page-11-0)). Usually, the main interest is at the wave vector which corresponds to the structure factor peak. Therefore, we have displayed the decay at these wave vectors in Fig. $11(b)$. At these larger wave vectors (or, conversely, small wavelengths), the large-scale compositional relaxation mode is not observed at low Φ . We can explain this by noting that at small length scales, density fluctuations do not require large-scale motion of the colloidal particles in order to relax.

At $\Phi > 0.592$ the decay of $F(k, \tau)$ undergoes a dramatic change in behavior at large wave vectors. We see a slower decay mode emerge, shown by the secondary step in the decay. This two-step relaxation is found in experimental systems [\[15\]](#page-11-0) and is predicted by MCT with the faster β decay usually associated with relaxation of particles inside their local cage and the slower α decay associated with breaking of particles from their local cage [\[31\]](#page-11-0).

Our analysis shows that we can interpret the secondary relaxation as the emergence of the compositional relaxation mode at this wave vector. It shows that in order for the smaller wavelength density fluctuations to decay at these packing fractions, large-scale compositional rearrangements are needed. This is completely consistent with the idea of strongly caged colloidal particles where small-scale motion within a cage may be easy, but large-scale excursions require the cooperative motion of a larger number of particles [\[15\]](#page-11-0). However, there is a subtle but important difference in our interpretation. In our previous work using this model without polydispersity, we did not observe any glassy behavior, so there must be a fundamental difference between the monodisperse and polydisperse cases to explain this behavior. Some authors studying deeply quenched monodisperse HS systems have been able to observe glassy behavior at short times [\[49\]](#page-11-0), but these systems were able to readily crystallize via short-range "shuffling" which allows amorphous regions to gradually transform into crystallites [\[50\]](#page-11-0). These authors note that additional fractionation must occur if polydisperse systems are to crystallize. Thus for the polydisperse case shown here, the key to the long-lived glass is that caging prevents the composition from relaxing and never allows the system to reach its equilibrium state.

The two different wave vectors displayed in Fig. [11](#page-9-0) give very different pictures of the nature of the − mode. At low *k*, the − mode is always present while progressively its amplitude increases and decay coefficient decreases. At the larger *k* the − mode appears to emerge as the glass transition is approached. This apparent emergence may simply be the result of an effect that is usually only observed at large length scales being driven to smaller length scales as particles become more caged. In order for a colloidal particle to break out of its neighboring cage, it requires cooperative motion of a larger number of other colloidal particles.

We note that since *D*[−] is small but nonzero, we do not observe complete structural arrest often considered as the signature of a glassy state. However, the persistent downturn of $F(k, \tau)$ does not forbid us from identifying $\Phi_g \approx 0.592$ as the glass transition packing fraction as this downturn is also observed in some experimental [\[15,46,51,52\]](#page-11-0) and simulation [\[53,54\]](#page-11-0) results above the glass transition. We have identified this as Φ_{g} because, above this packing fraction, the relaxation time of the system has grown to the point where the time it would take the system to relax to its equilibrium crystal is well beyond the accessible simulation time.

It may be possible to see further arrest by increasing the polydispersity to levels seen in experimental colloidal suspensions ($\geq 5\%$) or by using a continuous distribution of particle sizes rather than two discrete sizes. A later study on the dependence of *D*[−] on the total polydispersity and distribution shape may be of interest in order to find conditions that maximise the glass forming ability of the system.

V. CONCLUSION

Molecular dynamics simulations were conducted on a model colloidal suspension with explicit solvent. In this study, we extended previous work by including polydispersity into the model by adding a second smaller species of colloidal particles. The introduction of polydispersity had the effect of inhibiting crystallization at large packing fractions, resulting in glassy behavior. By calculating the intermediate scattering function and relating its decay to multicomponent interdiffusion coefficients, we attempted to establish the cause of this crystallization inhibition.

We found that the inhibition of crystallization was caused by the inability of the quenched fluid to undergo compositional changes needed for the formation of crystals. This link was determined by studying the − diffusive mode predicted for the ternary system from nonequilibrium thermodynamic theory. The effective diffusion coefficient *D*[−] of this mode was calculated independently from the Green-Kubo and Kirkwood-Buff theories. It was found to decrease as the packing fraction increased, showing that compositional relaxations become increasingly slow at large colloidal particle densities.

The long-time decay of the colloidal particle intermediate scattering function at low wave vector was dominated by the − decay mode. In the stable liquid phase, *D*−(*k*) was found to be wave-vector independent, while the amplitudes had a maximum in the $k \to 0$ limit and then decreased to zero at a finite wave vector. This showed that in this phase, compositional relaxations only contribute to small wave vector (large wavelength) density relaxations.

A decrease of *D*_− to negligible values at $\Phi \approx 0.592$ resulted in an extreme slow-down in the long-time decay of the intermediate scattering function. This was identified as the glass transition packing fraction because above this density, the system did not relax to its equilibrium crystal state within

the simulation time window. Unlike in the stable fluid phase, the amplitude of the − mode was observed to be nonzero for large wave vectors above the glass transition packing fraction. We interpret this emergence of the compositional relaxation mode by stating that in order for the smaller wavelength density fluctuations to decay at these packing fractions, large-scale compositional rearrangements are now needed. However, the compositional relaxation time has grown to the point where the time it would take the system to relax to its equilibrium

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crystal is well beyond the accessible simulation time, thus crystallization is inhibited.

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