# Brownian thermal birefringent noise due to the nondiagonal anisotropic photoelastic effect in multilayer coated mirrors

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Thermal noise in the mirror coatings limits the accuracy of today's most optical precision measurement experiments. Unlike the more commonly discussed thermal phase noise, the crystalline coating can generate thermal birefringent noise due to its anisotropic nature. In this study, we propose that the nondiagonal anisotropic photoelastic effect induced by the Brownian motion of mirror coating layers may contribute to this noise. Employing a standard model for the coating surface, we calculate the spectrum of the nondiagonal anisotropic Brownian photoelastic noise to be  $1.2 \times 10^{-11} p_{63} f^{-1/2} / \text{Hz}^{1/2}$ . Further experiments are warranted to validate the influence of this effect and reduce its uncertainty. Our findings highlight that for high-precision experiments involving optical resonant cavities targeting signals imprinted in optical polarizations, this noise could emerge as a limiting factor for experimental sensitivity.

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## I. INTRODUCTION

The investigation of thermal noise in multilayer-coated mirrors is crucial for understanding its impact on the sensitivity of high-precision measurement experiments conducted within optical resonant cavities. Prominent examples include laser interferometer gravitational-wave detectors [1–5] and optical clocks [6–11]. While the thermal noises in these experiments have traditionally centered on the optical phase effect, extensively explored in prior studies [7,12–14], we direct attention to experiments involving signals imprinted on the polarization of light, such as the measurement of vacuum magnetic birefringence (VMB) [15–19], or the search for axionlike particles [20–22], where thermal birefringent noise emerges as a key factor.

Low-loss mirrors are constructed by sequentially depositing two different materials onto the substrate, resulting in thermal noise contributed by the coating, the substrate, and by their interface. Here the coating is the main contributor because that optical field is circulated within the cavity by reflecting from the coating layers. Coating thermal noise has two potential sources: (1) temperature variations due to thermal dissipation and (2) Brownian motion due to mechanical dissipation. Because of their low mechanical losses, crystalline coatings have lower Brownian noise than amorphous coatings [23,24]. However, due to the anisotropic nature of the crystalline coating, its optical properties may exhibit an *anisotropy* which is different from that of the amorphous coatings.

Recent investigations into polarization fluctuations resulting from temperature fluctuations and the anisotropic thermal properties of crystalline coating materials, so-called thermorefringent noise, have been reported [25]. Previous studies [26,27] have considered the photoelastic effect due to Brownian motion in the coating layers, but included only the *isotropic* part which is the block diagonal terms in the photoelastic coefficients matrix. The optical anisotropy produced by the photoelastic effect can be divided into two types. The first one is due to the unequal diagonal elements which produce the birefringence, and this effect has been studied [16] previously. The other one is due to the nondiagonal elements, which leads to the variation in the angle of birefringence (see Sec. II A), which remains unexplored. In this work, we will explore the possible Brownian noise contributed by the nonzero off-diagonal photoelastic coefficients, which could result from the strains induced during the manufacturing process.

Brownian motion of the coating layer, through the nondiagonal anisotropic photoelastic effect, can lead to fluctuations in the polarization state. This noise will appear in the polarization mode that is perpendicular to the incident polarization, termed in this paper as nondiagonal anisotropic Brownian photoelastic (NABP) noise. This noise has the potential to limit the performance of VMB experiments and

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FIG. 1. Structure of multilayer coated mirror. The light colored part represents the coating layer; the dark colored part represents the substrate. The subscript 1 represents GaAs layers and 2 represents AlGaAs layers in our experiments. The thickness of all coating layers is  $\lambda/4$  except the first one, which is  $\lambda/2$ .

other precision optical polarization measurement experiments. Therefore, experimental characterization of the offdiagonal photoelastic coefficient of the coating materials is necessary for an accurate assessment of the NABP noise.

This paper is structured as follows: In Sec. II, we begin by introducing the theoretical model of the photoelastic effect and its impact when light traverses an anisotropic photoelastic medium. Subsequently, we give the expression for the polarization fluctuations of the reflected light employing the standard structure of mirror coating. In Sec. III, the noise level of the thermal birefringent noise contributed by the nondiagonal anisotropic photoelastic effect is calculated and compared with other noise. Finally, Sec. IV summarizes our principal findings and engages in future avenues of research.

#### **II. THEORETICAL MODEL**

#### A. Photoelastic effect

The photoelastic effect, denoting the change in refractive index induced by mechanical strain, is characterized by symmetric tensors representing the strain  $(u_{ij})$  and the change in the optical indicatrix  $(\Delta B_{ij})$ . Utilizing contracted indices  $(i, j \in xx, yy, zz, yz, zx, xy$  rather than x, y, z) for these tensors, the photoelastic effect can be expressed in a vectorial form:

$$\Delta B_i = p_{ij} u_j, \tag{1}$$

where  $\Delta B_i$  is the change of the optical indicatrix,  $p_{ij}$  is photoelastic tensor, and  $u_j$  is strain components (we follow the conventions used in [28] where the six independent matrix elements of  $u_{ij}$  are summarized as one vector  $u_j$ , the same for the relationship between the  $\Delta B_{ij}$  and the  $\Delta B_i$ ). Since fluctuations caused by photoelastic effect in the transverse direction are much smaller than in the longitudinal direction [14], we only consider the variation of the refractive index caused by the change of coating thickness,

$$\Delta B_i = p_{i3} \delta d/d, \tag{2}$$

where  $\delta d/d$  denotes the change in coating thickness. Since light propagates along the z axis (see Fig. 1), and the optical field is confined to the x-y plane, the optical indicatrix **B** is given by

$$\mathbf{B} = \begin{bmatrix} B_{xx} & 0\\ 0 & B_{yy} \end{bmatrix} + \begin{bmatrix} p_{13} & p_{63}\\ p_{63} & p_{23} \end{bmatrix} \frac{\delta d}{d}, \qquad (3)$$

while the dielectric tensor  $\boldsymbol{\varepsilon}$  is inverse of **B**. To the first order of  $\delta d/d$ , the dielectric tensor reads

$$\boldsymbol{\varepsilon} = \mathbf{B}^{-1}$$

$$= \begin{bmatrix} \langle n_x^2 \rangle & 0 \\ 0 & \langle n_y^2 \rangle \end{bmatrix} - \begin{bmatrix} p_{13} \langle n_x^4 \rangle & p_{63} \langle n_x^2 \rangle \langle n_y^2 \rangle \\ p_{63} \langle n_x^2 \rangle \langle n_y^2 \rangle & p_{23} \langle n_y^4 \rangle \end{bmatrix} \frac{\delta d}{d}, \quad (4)$$

where  $\langle n_x \rangle$ ,  $\langle n_x \rangle$  is refractive index in the *x*, *y* direction,  $\langle n_x \rangle = B_{xx}^{-1/2}$ ,  $\langle n_y \rangle = B_{yy}^{-1/2}$ . The strain variation  $\delta d/d$  changes the principle directions of the dielectric tensor, where the refractive index and the new principle directions are given by the diagonalization of  $\boldsymbol{\varepsilon}$ :

$$n_x \approx \langle n_x \rangle - \frac{p_{13} \langle n_x^3 \rangle}{2} \frac{\delta d}{d},$$
 (5)

$$n_y \approx \langle n_y \rangle - \frac{p_{23} \langle n_y^3 \rangle}{2} \frac{\delta d}{d},$$
 (6)

with corresponding eigenvectors

$$\mathbf{v}_{1} \approx \begin{bmatrix} 1\\0 \end{bmatrix} - \frac{p_{63} \langle n_{x}^{2} \rangle \langle n_{y}^{2} \rangle}{\langle n_{x}^{2} \rangle - \langle n_{y}^{2} \rangle} \frac{\delta d}{d} \begin{bmatrix} 0\\1 \end{bmatrix}, \tag{7}$$

$$\mathbf{v}_{2} \approx \begin{bmatrix} 0\\1 \end{bmatrix} + \frac{p_{63} \langle n_{x}^{2} \rangle \langle n_{y}^{2} \rangle}{\langle n_{x}^{2} \rangle - \langle n_{y}^{2} \rangle} \frac{\delta d}{d} \begin{bmatrix} 1\\0 \end{bmatrix}.$$
(8)

This fluctuation in eigenvectors is equivalent to intrinsic birefringence of coating rotated by a small angle,

$$\delta\theta \approx -\frac{p_{63}\langle n_x^2 \rangle \langle n_y^2 \rangle}{\langle n_x^2 \rangle - \langle n_y^2 \rangle} \frac{\delta d}{d}.$$
(9)

Notably, while the influence of  $p_{13}$  and  $p_{23}$  on fluctuations in  $n_x$  and  $n_y$  has been previously studied [26], the impact of the  $p_{63}$  term, leading to fluctuations in the angle of intrinsic birefringence, remains unexplored. Typical optical materials used in the crystalline coating are GaAs/AlGaAs. Theoretically both materials are isotropic and therefore do not have nondiagonal photoelastic coefficients  $p_{63}$ . Here we propose two possible physical scenarios for producing a nonzero  $p_{63}$ . Since the thermal expansion coefficients of the coating  $\alpha_C$  and the substrate  $\alpha_S$  are different, the thermal elasticity will induce a strain in the coating layers when the temperature drops from the manufacturing temperature to the room temperature [29]:

$$\frac{\delta l}{l} = (\alpha_C - \alpha_S) \Delta T = 2 \times 10^{-3}.$$
 (10)

Theoretically, the coating film is stressed uniformly, which means that the lattices on the coating surface are all exposed to the same stresses, and the direction of the stress is along their location to the center of the mirror. For most of the lattices on the coating surface, the direction of the crystal axis and the stress are different. This lowers the symmetry of the crystal system from cubic to monoclinic or triclinic [30], thus producing a nonzero  $p_{63}$ .

Another possible mechanism is that the deformation of the top layer is less than that of the layer closest to the substrate, creating a shear strain at the area where the light spot stays:

$$S = \frac{\delta l_t - \delta l_s w_0}{l},\tag{11}$$

where  $w_0$  is the beam radius, d is the thickness of the full coating layer, and  $\delta l_t/l$ ,  $\delta l_s/l$  are the strains in the top and substrate layers, respectively. This shear strain could lower the symmetry of the crystal system from cubic to triclinic [30], thus producing a nonzero  $p_{63}$ . The value of  $p_{63}$  needs to be measured by further experiments, and in this paper we temporarily use 1% of the photoelastic coefficient  $p_{13}$  to demonstrate the influence of this effect.

#### **B.** Light penetration in mutilayer coating

The conventional structure of a low-loss mirror coating, depicted in Fig. 1, consists of *N* pairs of layers with high and low refractive index. The thickness of most coating layers is  $\lambda/4$ , except for the top layer which has a thickness of  $\lambda/2$ . To compute the light field reflected from the mirror coating, we adopt a method similar to that employed by LIGO in calculating thermorefringent noise [25]. Commencing with transfer matrices for a unit cell, we deduce the matrix representing the fluctuations in the optical field reflected from the mirror coating.

### 1. Transfer matrix of a unit cell

As mentioned above, mirror coating consists of repeating pairs of high and low refractive coating index layers, we denote the propagation matrix of the *i*th layer pair as a unit cell propagation matrix  $\Phi_i$ . A unit cell is composed of two layers and two interfaces between them. A four-dimensional vector is used to represent the light propagating through the coating:

$$\mathbf{E} = \begin{bmatrix} E_{x+} \\ E_{x-} \\ E_{y+} \\ E_{y-} \end{bmatrix}, \qquad (12)$$

where *x*, *y* is the direction of polarization, and  $\pm$  represent the (right/left) direction of light propagation.

The propagation matrix in the bulk medium is

$$\mathbf{T}^{(I)} = \begin{bmatrix} \mathbf{T}_x^{(I)} & \mathbf{0} \\ \mathbf{0} & \mathbf{T}_y^{(I)} \end{bmatrix},\tag{13}$$

where superscript *I* denoted the material (I = 1/2 represents) sents the GaAs and AlGaAs).  $\mathbf{T}_x^{(I)}$  and  $\mathbf{T}_y^{(I)}$  are the propagating matrix of the electric field  $E_{x/y}$  with the coordinate system using basis vector  $\mathbf{v}_1$ ,  $\mathbf{v}_2$  (in [25], these two vectors are defined as "coordinate vectors"), which is the same in Eqs. (7) and (8),  $\mathbf{T}_a^{(I)}$  is

$$\mathbf{T}_{a}^{(I)} = \begin{bmatrix} e^{-in_{a}^{(I)}kd_{i}^{(I)}} & 0\\ 0 & e^{in_{a}^{(I)}kd_{i}^{(I)}} \end{bmatrix},$$
(14)

where  $n_a^{(I)}$  is the refractive index in the material *I* along the *a* direction, *k* is the wave vector of the incident light, and  $d_i^{(I)}$  is the thickness of the material *I* in *i*th layer. Since the coordinate vectors of a layer fluctuate by an angle  $\delta\theta$ , which is obtained by Eq. (9),

$$\delta\theta = \begin{cases} -\frac{p_{63}^{(1)}n_1^4}{n_1^2 - (n_1 + \Delta n_1)^2} \frac{\delta d_i^{(1)}}{d_i^{(1)}}, & \delta d_i^{(I)} = \delta d_i^{(1)} \\ \frac{p_{63}^{(2)}n_2^4}{n_2^2 - (n_2 + \Delta n_2)^2} \frac{\delta d_i^{(2)}}{d_i^{(2)}}, & \delta d_i^{(I)} = \delta d_i^{(2)}. \end{cases}$$
(15)

where the  $\delta d_i^{(I)}$  represent the deformation of the material *I* layer in the *i*th unit cell. Here we denote the optical anisoptropy using the difference of refractive index along x/y directions  $n_y^I = n_x^I + \Delta n_I$  and we have  $\Delta n_I/n_x^I \ll 1$ . The **R**<sup>(IJ)</sup> is the transfer matrix of the interface between material *I* and *J*, which is combined by a rotation matrix and the transfer matrix:

$$\mathbf{R}^{(12)} = \begin{bmatrix} \mathbf{r}_{xx}^{(12)} \cos(\delta\theta) & -\mathbf{r}_{xy}^{(12)} \sin(\delta\theta) \\ \mathbf{r}_{yx}^{(12)} \sin(\delta\theta) & \mathbf{r}_{yy}^{(12)} \cos(\delta\theta) \end{bmatrix}, \quad (16)$$

$$\mathbf{R}^{(21)} = \begin{bmatrix} \mathbf{r}_{xx}^{(21)} \cos(\delta\theta) & \mathbf{r}_{xy}^{(21)} \sin(\delta\theta) \\ -\mathbf{r}_{yx}^{(21)} \sin(\delta\theta) & \mathbf{r}_{yy}^{(21)} \cos(\delta\theta) \end{bmatrix}, \quad (17)$$



FIG. 2. Definition of the unit cell, the incident cell, and the substrate cell when calculating a different material coating layer. The unit cell is marked with shadows and solid lines.

where  $\mathbf{r}_{ab}^{(IJ)}$  is transfer matrix of interface without considering the polarization (both sides in the same polarization direction):

$$\mathbf{r}_{ab}^{(IJ)} = \frac{1}{2} \begin{bmatrix} 1 + \frac{n_b^{(J)}}{n_a^{(J)}} & 1 - \frac{n_b^{(J)}}{n_a^{(J)}} \\ 1 - \frac{n_b^{(J)}}{n_a^{(J)}} & 1 + \frac{n_b^{(J)}}{n_a^{(J)}} \end{bmatrix},$$
(18)

which is described by the boundary conditions derived from Maxwell's equation [25].

As we shall see in the next section [Eqs. (21) and (22)], to the first order of the strain deformation, the contributions of different layers to the propagation matrix are independent of each other, and the strain modification of the full propagation matrix can be obtained by adding the contributions of all layers. Therefore, the key component of our calculation is the strain modification of a single layer. In the upper panel of the schematic diagram of the coating cell (Fig. 2), the strain deformation  $\delta d_i$  of the layer  $d_i^{(1)}$  can influence the interfaces with two neighboring layers marked as a solid line, and they are both included in the unit cell  $\Phi_i$ . However, for the layer  $d_i^{(2)}$ , only one interface is included in the unit cell which means that another definition is needed to calculate the influence of  $\delta d_i^{(2)}$ . Therefore, it is convenient to use two different definitions for calculating  $\delta d_i^{(1)}$  and  $\delta d_i^{(2)}$ :

$$\mathbf{\Phi}_{i} = \begin{cases} \mathbf{R}^{(21)} \mathbf{T}^{(1)} \mathbf{R}^{(12)} \mathbf{T}^{(2)} & \text{when } \delta d_{i}^{(I)} = \delta d_{i}^{(1)} \\ \mathbf{R}^{(12)} \mathbf{T}^{(2)} \mathbf{R}^{(21)} \mathbf{T}^{(1)} & \text{when } \delta d_{i}^{(I)} = \delta d_{i}^{(2)}. \end{cases}$$
(19)

The choice of the definition of  $\Phi_i$  depends on the material *I* of the layer being calculated, which is presented in detail in the next section (Sec. II B 2).

## 2. Transfer matrix of full coating layers

The matrix of the full coating layers can be written as the product of unit cell matrices:

$$\mathbf{M} = \mathbf{\Phi}^{(in)} \left( \prod_{i=2}^{N} \mathbf{\Phi}_{i} \right) \mathbf{\Phi}^{(s)}, \qquad (20)$$

where  $\Phi_i$  represent the matrix of the *i*th unit cell,  $\Phi^{(in,s)}$  is the propagation matrix of the incident cell and the last coating cell near the substrate, and N represents the number of unit cells in the coating. Separating a unit cell matrix into the steady part and the fluctuating part,

$$\mathbf{\Phi}_i = \mathbf{\Phi}_0 + \mathbf{\Phi}' \delta d_i^{(I)}. \tag{21}$$

Since  $\delta d_i^{(I)}$  of different layers are independent of each other [14], their influence on **M** can be calculated separately (except for the first and last layer):

$$\mathbf{M}(\delta d_i^{(I)}) = \mathbf{\Phi}_0^{(in)} \mathbf{\Phi}_0^{i-1} (\mathbf{\Phi}_0 + \mathbf{\Phi}' \delta d_i^{(I)}) \mathbf{\Phi}_0^Q \mathbf{\Phi}_0^{(s)}, \quad (22)$$

where

$$Q = \begin{cases} N - i - 2, & \delta d_i^{(I)} = \delta d_i^{(1)} \\ N - i - 1, & \delta d_i^{(I)} = \delta d_i^{(2)}. \end{cases}$$
(23)

The definition choice of  $\mathbf{\Phi}^{(in)}, \mathbf{\Phi}^{(s)}$ , and  $\mathbf{\Phi}_i$  depends on the material of the fluctuating layer  $\delta d_i^{(I)}$  that needs to be calculated. For the fluctuation of the top layer  $\delta d_1^{(1)}$  and the layer next to the substrate  $\delta d_{N+1}^{(1)}$ , the **M** is

$$\mathbf{M}(\delta d_1^{(1)}) = (\mathbf{\Phi}_0^{(in)} + \mathbf{\Phi}'^{(in)} \delta d_1^{(1)}) \mathbf{\Phi}_0^{N-1} \mathbf{\Phi}_0^{(s)}, \qquad (24)$$

$$\mathbf{M}(\delta d_{N+1}^{(1)}) = \mathbf{\Phi}_0^{(in)} \mathbf{\Phi}_0^{N-1} (\mathbf{\Phi}_0^{(s)} + \mathbf{\Phi}'^{(s)} \delta d_{N+1}^{(1)}).$$
(25)

Using the definition in Eqs. (13)–(19), it turns out that  $\Phi_0$  is block diagonal and  $\Phi'$  is off diagonal (to the first order of  $\delta d_i^{(l)}$ ):

$$\boldsymbol{\Phi}_{i} = \begin{bmatrix} \boldsymbol{\Phi}_{xx} & 0\\ 0 & \boldsymbol{\Phi}_{yy} \end{bmatrix} + \begin{bmatrix} 0 & \boldsymbol{\Phi}_{xy}\\ \boldsymbol{\Phi}_{yx} & 0 \end{bmatrix} \delta d_{i}^{(I)}, \quad (26)$$

where  $\Phi_{xx}$  and  $\Phi_{xy}$  is

$$\begin{split} \mathbf{\Phi}_{xx} &= \mathbf{\Phi}_{yy} \approx \begin{cases} -\frac{1}{2n_1 n_2} \begin{bmatrix} n_1^2 + n_2^2 & n_2^2 - n_1^2 \\ n_2^2 - n_1^2 & n_1^2 + n_2^2 \end{bmatrix}, & \delta d_i^{(I)} = \delta d_i^{(1)} \\ -\frac{1}{2n_1 n_2} \begin{bmatrix} n_1^2 + n_2^2 & n_1^2 - n_2^2 \\ n_1^2 - n_2^2 & n_1^2 + n_2^2 \end{bmatrix}, & \delta d_i^{(I)} = \delta d_i^{(2)}, \end{split}$$

$$\begin{aligned} \mathbf{\Phi}_{xy} &= \mathbf{\Phi}_{yx} \approx \begin{cases} \frac{p_{63}^{(1)} n_1}{4n_2 d_i^{(1)}} \begin{bmatrix} n_2^2 - n_1^2 + i\pi n_1 n_2 & n_1^2 + n_2^2 \\ n_1^2 + n_2^2 & n_2^2 - n_1^2 - i\pi n_1 n_2 \end{bmatrix}, & \delta d_i^{(I)} = \delta d_i^{(1)} \\ \frac{p_{63}^{(2)} n_2}{4n_1 d_i^{(2)}} \begin{bmatrix} n_1^2 - n_2^2 + i\pi n_1 n_2 & n_1^2 + n_2^2 \\ n_1^2 + n_2^2 & n_1^2 - n_2^2 - i\pi n_1 n_2 \end{bmatrix}, & \delta d_i^{(I)} = \delta d_i^{(2)}. \end{aligned}$$

$$\end{aligned}$$

$$\tag{28}$$

The definition of the incident cell  $\Phi^{(in)}$  and the substrate cell  $\Phi^{(s)}$  are also different when calculating different material layers:

$$\mathbf{\Phi}^{(in)} = \begin{cases} \mathbf{R}^{(01)} \mathbf{T}^{(1)} \mathbf{R}^{(12)} \mathbf{T}^{(2)}, & \delta d_i^{(I)} = \delta d_1^{(1)} \\ \mathbf{R}^{(01)} \mathbf{T}^{(1)}, & \delta d_i^{(I)} = \delta d_1^{(2)}, \end{cases}$$
(29)

$$\mathbf{\Phi}^{(s)} = \begin{cases} \mathbf{R}^{(21)} \mathbf{T}^{(1)} \mathbf{R}^{(1s)}, & \delta d_i^{(I)} = \delta d_{N+1}^{(1)} \\ \mathbf{R}^{(1s)}, & \delta d_i^{(I)} = \delta d_{N+1}^{(2)}. \end{cases}$$
(30)

Assuming that incident light comes from the vacuum where  $n_0 = 1$ , the matrix of the incident cell and the substrate cell is

$$\boldsymbol{\Phi}^{(in)} = \begin{bmatrix} \boldsymbol{\Phi}_{xx}^{(in)} & \boldsymbol{\Phi}_{xy}^{(in)} \delta d_1^{(l)} \\ \boldsymbol{\Phi}_{yx}^{(in)} \delta d_1^{(l)} & \boldsymbol{\Phi}_{yy}^{(in)} \end{bmatrix}, \quad (31)$$
$$\boldsymbol{\Phi}^{(s)} = \begin{bmatrix} \boldsymbol{\Phi}_{xx}^{(s)} & \boldsymbol{\Phi}_{xy}^{(s)} \delta d_{N+1}^{(l)} \\ \boldsymbol{\Phi}_{yx}^{(s)} \delta d_{N+1}^{(l)} & \boldsymbol{\Phi}_{yy}^{(s)} \end{bmatrix}, \quad (32)$$

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$$\Phi_{xx}^{(in)} = \Phi_{yy}^{(in)}$$

$$\approx \begin{cases} -\frac{1}{2n_1} \begin{bmatrix} n_1^2 + n_2^2 & n_2^2 - n_1^2 \\ n_2^2 - n_1^2 & n_1^2 + n_2^2 \end{bmatrix}, & \delta d_i^{(I)} = \delta d_i^{(1)} \\ \frac{i}{2} \begin{bmatrix} -1 - n_1 & 1 - n_1 \\ -1 + n_1 & 1 + n_1 \end{bmatrix}, & \delta d_i^{(I)} = \delta d_i^{(2)}, \end{cases}$$
(33)

and

$$\begin{split} \boldsymbol{\Phi}_{xx}^{(s)} &= \boldsymbol{\Phi}_{yy}^{(s)} \\ \approx \begin{cases} \frac{i}{2n_1 n_2} \begin{bmatrix} -n_1^2 - n_2 n_s & -n_1^2 + n_2 n_s \\ n_1^2 - n_2 n_s & n_1^2 + n_2 n_s \end{bmatrix}, & \delta d_i^{(I)} &= \delta d_{N+1}^{(1)}, \\ \frac{1}{2} \begin{bmatrix} 1 + n_s / n_1 & 1 - n_s / n_1 \\ 1 - n_s / n_1 & 1 + n_s / n_1 \end{bmatrix}, & \delta d_i^{(I)} &= \delta d_{N+1}^{(2)}, \end{cases}$$

$$\end{split}$$

$$(34)$$

$$\Phi_{xy}^{(s)} = \Phi_{yx}^{(s)} \approx \begin{cases}
\frac{p_{63}^{(1)}n_1}{8d_{N+1}^{(1)}n_2} \begin{bmatrix}
-2i(n_1^2 - n_2n_s) - n_1(n_2 + n_s)\pi & -2i(n_1^2 + n_2n_s) + n_1(n_s - n_2)\pi \\
2i(n_1^2 + n_2n_s) + n_1(n_s - n_2)\pi & 2i(n_1^2 - n_2n_s) - n_1(n_2 + n_s)\pi
\end{bmatrix}, \quad \delta d_i^{(I)} = \delta d_{N+1}^{(1)} \\
0, \quad \delta d_i^{(I)} = \delta d_{N+1}^{(2)},
\end{cases}$$
(36)

where the  $n_s$  is the refractive index of substrate (usually made by SiO<sub>2</sub>). Note that  $\Delta n_1$  and  $\Delta n_2$  in Eqs. (27)–(36) is neglected since  $\Delta n_I/n_x^I \ll 1$ .

# 3. The reflected light from the coating layers

The entire coating layers can be described as a  $4 \times 4$  matrix **M**, connecting the optical field on the right of the coating to those on the left:

$$\begin{bmatrix} E_{in} \\ E_{rx} \\ 0 \\ E_{ry} \end{bmatrix} = \begin{bmatrix} M_{11} & M_{12} & M_{13} & M_{14} \\ M_{21} & M_{22} & M_{23} & M_{24} \\ M_{31} & M_{32} & M_{33} & M_{34} \\ M_{41} & M_{42} & M_{43} & M_{44} \end{bmatrix} \begin{bmatrix} E_{tx} \\ 0 \\ E_{ty} \\ 0 \end{bmatrix}, \quad (37)$$

where  $E_{in}$  is incident light polarized along the x axis,  $E_{rx}$ ,  $E_{ry}$  is the light field reflected by the coating layers, and  $E_{tx}$ ,  $E_{ty}$  is the transmitted light. Since we are interested in the variation of the polarization state, the only term that needs to be calculated is  $E_{ry}$ :

$$r_{y} = \frac{E_{ry}}{E_{in}} = \frac{M_{41}M_{33} - M_{43}M_{31}}{M_{11}M_{33} - M_{13}M_{31}},$$
 (38)

which is obtained by solving Eq. (37). The real part of  $r_y$  represents the change in polarization angle, and the imaginary part represents the change in ellipticity (i.e., birefringence). It turns out that most of the  $r_y$  is the imaginary part (see Sec. III) so that the real part can be neglected. The transfer matrix **M** describes the response of the light field with respect to the coating surface, therefore **M** depends on the rotation angle of the optical axis  $\delta\theta$ . Furthermore, the  $\delta\theta$  is related to  $\delta d$  by Eq. (9), which leads to a relationship between the ellipticity fluctuations spectrum  $S_{\psi}$  and the deformation spectrum of a single coating layer  $S_{d^{(l)}}$ :

$$S_{\psi}^{1/2} = \sqrt{\sum_{I=1}^{2} \sum_{i=1}^{N} \left(\frac{\partial r_{y}}{\partial d_{i}^{(I)}}\right)^{2} S_{d_{i}^{(I)}}}.$$
 (39)

The fluctuation of the coating layer thickness  $\delta d_i^{(1)}$  is typically a Brownian type thereby showing a Brownian spectrum of  $S_{\psi}$ . Note that the summation is over the influence of different layers. We do not have the cross term like  $S_{d_i^{(I)}d_j^{(I)}}$  is because the deformations caused by Brownian motion in the different layers are independent of each other [14].

#### **III. CALCULATED NOISE LEVEL**

In this section we present the noise level of the nondiagonal anisotropic Brownian photoelastic (NABP) effect using the parameters listed in Table I, based on the theory

TABLE I. Typical parameters used in this paper.

Parameters	Symbol	GaAs	AlGaAs
Refractive index	n	3.37 [33]	2.90 [34]
Poisson's ratio	$\sigma$	0.32 [33]	
Young's modulus (GPa)	Y	100 [23]	
Loss angle	$\phi$	$2.41 \times 10^{-5}$ [35]	
Temperature (K)	Т	293	
Beam radius (µm)	$r_0$	1500	
Photoelastic coefficient	$p_{13}$	-0.14 [36]	
Photoelastic coefficient	$p_{63}$	-0.0014	

established in Sec. II. As mentioned in Sec. II A, the  $p_{63}$  in the crystalline coating materials may arise from strain induced during the deposition process, and there is currently no experimental measurement of  $p_{63}$  in the coating material. On the other hand, the theoretical quantitative estimation of  $p_{63}$  requires a first-principle density-functional calculation [31,32]. In this paper, we focus on demonstrating the influence of the nondiagonal anisotropic Brownian photoelastic effect, and the value of  $p_{63} =$ -0.0014 is assumed to be one percent of  $p_{13}$  and the values of  $p_{63}$  in both coating materials are assumed to be the same. The real value of  $p_{63}$  could be smaller than this assumption, which needs to be determined by further experimental test or more detailed density-functional calculation by the material science theorists.

One can derive a complete expression for the **M** matrix by substituting Eqs. (21)–(36) into Eq. (20). The contribution of the different layers to the ellipticity  $(\partial r_y/\partial d_i)$  is then obtained by Eq. (38), as shown in Fig. 3. We can see



FIG. 3. Influence of different layer thicknesses on the ellipticity for a 19-pair GaAs/AlGaAs coated mirror. As previously mentioned, the imaginary part of  $\partial r_y/\partial d_i$  represents the change in ellipticity of the light reflected from the mirror and the real part represents the change in polarization angle. The real part of  $\partial r_y/\partial d_i$  is 4 orders of magnitude smaller than the imaginary part thereby can be ignored.



FIG. 4. Nondiagonal anisotropic Brownian photoelastic noise (NABP) calculated for a 25-pair GaAs/AlGaAs coated mirror. The comparison with the Brownian noise and thermorefringent noise is also shown.

that the effect of the layer on the ellipticity decreases as the number of layers increases, so that only the first few layers contribute most to the variation of the ellipticity.

For a single coating layer, the thickness fluctuation due to the Brownian motion is obtained using Levin's approach [12,14]:

$$S_d(f) = \frac{8k_B T d_i \phi_i (1 - \sigma_i - 2\sigma_i^2)}{3\pi^2 f Y_i (1 - \sigma_i)^2 r_0^2},$$
(40)

where  $k_B$  is the Boltzmann constant,  $d_i$  is the thickness of this layer,  $\phi_i$  is the loss angle,  $\sigma_i$  is Poisson's ratio,  $Y_i$  is Young's modulus, and  $r_0$  is the radius of the laser beam.

The noise level of the NABP effect is obtained by substituting Eq. (40) into Eq. (39), as shown in Fig. 4. This result demonstrates that, using the sampled parameters for the  $p_{63}$  in Table I, the NABP noise is

$$S_{\psi(\text{NABP})}^{1/2} = 1.2 \times 10^{-11} p_{63} f^{-1/2} / \sqrt{\text{Hz}},$$
 (41)

which is shown in Fig. 4, and the Brownian noise and the thermorefringent noise are also shown for comparison. Here the thermorefringent noise is [25]

$$S_{\psi(TF)} = \left| r_y \frac{\pi}{2} \frac{n_2 \varepsilon_{xy}^{\prime(1)} + n_1 \varepsilon_{xy}^{\prime(2)}}{n_1 n_2 (n_1^2 - n_2^2)} \right|^2 S_{uu}(f), \qquad (42)$$

where  $\epsilon'_{xy} = 10 \text{ ppm/K}$  is the thermorefractive coefficient,  $r_y$  is the reflection amplitude for y polarization, and  $S_{uu}$  is the power spectral density of temperature averaged over the beam spot:

$$S_{uu}(f) = \frac{2k_B T^2}{\pi r_0 c_V D} \operatorname{Re}\left\{\int_0^\infty du \,\frac{u e^{-u^2/2}}{\sqrt{u^2 - ir_0^2 f/D}}\right\}, \quad (43)$$

where  $c_V = 1.6 \times 10^6 \text{ J}(\text{mK})^{-1}$  is heat capacity per unit volume,  $\kappa = 1.38 \text{ W}(\text{mk})^{-1}$  is the thermal conductivity, and  $D = \kappa/c_V$  is thermal diffusivity.

## **IV. CONCLUSIONS**

For experiments such as the measurement of vacuum magnetic birefringence, the measurement principle is to amplify the signal through a resonant cavity, which is then detected by a polarimetry. This polarimetry detects the phase difference between the two polarization directions, so that the phase noise is canceled out. Thus, the sensitivity of these experiments is not limited by the commonly discussed thermal phase noise (Brownian, thermo-optic noise) but by the thermal birefringent noise. Experimental measurements and theoretical investigations of the anisotropy of these coating materials are important.

In this work, we present a comprehensive analysis of a novel kind of birefringent noise induced by the Brownian motion of the mirror coating. In particular, we discussed the possible influence of a novel noise source originating from the nondiagonal anisotropic photoelastic effect, which may have limited the sensitivity of precision polarimetry experiments. The strain deformation caused by Brownian motion in a single coating layer induces a rotation of the intrinsic birefringence due to the nondiagonal anisotropic photoelastic effect. By extending these findings to multilayer surfaces using the matrix method, we elucidate the corresponding fluctuations in the light reflected from the mirror.

For the GaAs/AlGaAs coating, the noise level caused by the nondiagonal anisotropic Brownian photoelastic (NABP) effect is calculated to be  $1.2 \times 10^{-11} p_{63} f^{-1/2}$ /  $Hz^{1/2}$  using the sampled parameters in Table I. The nondiagonal anisotropic Brownian photoelastic noise level is decisively dependent on the value of  $p_{63}$ . Assuming that  $p_{63}$  is one percent of the  $p_{13}$ , the NABP noise level is of the same order of magnitude as the thermorefringent noise. In addition, the anisotropy in the different parts of the coating can introduce loss in reflectivity [29], which also generates noise according to the fluctuation-dissipation theorem, and this noise could be a stronger effect than the NABP noise effect. This anisotropy can be evaluated through measuring the intrinsic birefringence of different parts of the coating surface, hence further compensated probably.

The source of the  $p_{63}$  and other anisotropies of the coating materials could be the strain induced during the manufacturing process; the specific mechanisms and the evaluation of  $p_{63}$  for the coating material require further experimental and theoretical investigation.

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