# Neutron charge radius deduced from Bragg reflection technique

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The possibility of the determination of the neutron mean square charge radius from high-precision thermalneutron measurements of the nuclear scattering length and of the scattering amplitudes of Bragg reflections is considered. Making use of the same Pendellösung technique as Shull [Phys. Rev. Lett. **21**, 1585 (1968)], the scattering amplitudes of about eight higher-order Bragg reflections in silicon could be measured without single-scattering contamination. This would provide a value of the neutron charge radius as precise as the disagreeing Argonne-Garching and Dubna values, as well as a Debye-Waller factor of silicon ten times more precise than presently available.

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## I. INTRODUCTION

The so-called mean square charge radius of the neutron  $\langle r_n^2 \rangle$  is an important structure constant which reflects the internal charge structure of the neutron. In terms of the generally used form factors,  $\langle r_n^2 \rangle$  is related to the derivative of the Sachs form factor  $G_E^n$  at vanishing transferred momentum Q,

$$\langle r_{\rm n}^2 \rangle = -\frac{1}{6} \left. \frac{dG_{\rm E}^{\rm n}}{dQ^2} \right|_{Q=0}.$$
 (1)

Low-energy neutron-atom scattering experiments have been found useful [1,2] to determine  $\langle r_n^2 \rangle$  via high-precision measurements of the neutron-electron scattering length  $b_{ne}$ ,

$$b_{\rm ne} = \frac{1}{3} \frac{\alpha m_{\rm n} c^2}{\hbar c} \langle r_{\rm n}^2 \rangle, \qquad (2)$$

where  $m_n$  is the mass of the neutron. Because of the fundamental importance of these quantities for our understanding of the nucleon many ambitious experiments have been performed in the last decades (see, e.g., Ref. [3], and references therein).

Despite these efforts the determination of  $b_{\rm ne}$  is still unsatisfactory because there exist two sets of results which differ more than three standard deviations from each other, as explained in Sec. II. In this paper we propose an independent method based on high-precision measurements of neutron Bragg reflections on silicon. The principle of the experimental technique, proposed by Shull in 1968 but only applied to the (111) reflection at that time [4], is summarized in Sec. III. In Sec. IV, we show that, by assuming for all higher-order reflections a precision similar to that achieved by Shull, one can solve the discrepancy in the values of  $b_{\rm ne}$ . Concluding remarks are given in Sec. V.

## II. DISCUSSION OF THE NEUTRON-ELECTRON–SCATTERING-LENGTH VALUES

In a first theoretical estimate,  $b_{ne}$  is expected to be given by the Foldy scattering length [5],

$$b_{\rm ne}^{\rm theory} = -1.467\,971(4) \times 10^{-3}$$
 fm, (3)

corresponding to the well-known value of the anomalous magnetic moment of the neutron [6]. The actual  $b_{ne}$  value can be deduced from low-energy neutron-atom scattering, for which the "scattering length" reads [1]

$$b(Q) = b_{\text{nuclear}} \underbrace{-b_{\text{ne}}Z[1-f(Q)]}_{\text{electrostatic}}$$
$$= b_{\text{nucleus}} \underbrace{+b_{\text{ne}}Zf(Q)}_{\text{electrons}}, \qquad (4)$$

where  $b_{\text{nuclear}}$  is the nuclear-interaction scattering length,  $b_{\text{nucleus}}$  is the neutron-nucleus scattering length (nuclear and electrostatic interactions), *Z* is the atomic number, and f(Q) is the atomic form factor normalized to the forward direction: f(0)=1. In Eq. (4), the *Q*-dependent electrostatic term is typically three orders of magnitude smaller than the constant nuclear term, hence it is reasonable to speak of "scattering length" despite the *Q* dependence. To get a sufficiently high sensitivity to this term scattering processes at  $Q \neq 0$ [hence 0 < f(Q) < 1] and high *Z* should be chosen.

The best experimental values of  $b_{\rm ne}$  (see, e.g., Table I in Ref. [7]) can be grouped into two sets. Those deduced from thermal-neutron angular scattering on noble gases [8] and from thermal- and epithermal-neutron transmission on liquid lead and bismuth [3,9] agree on the value  $b_{\rm ne} = -1.31(3) \times 10^{-3}$  fm (Argonne-Garching). The second set of  $b_{\rm ne}$  values obtained from another transmission experiment on Bi [10] and a measurement of thermal-neutron Bragg reflections on tungsten monocrystals [11] is centered around  $b_{\rm ne} = -1.59(4) \times 10^{-3}$  fm (Dubna). There is a clear discrepancy because the  $b_{\rm ne}$  values of the two sets differ more than three standard deviations from each other. In addition they differ about 10% from  $b_{\rm ne}^{\rm theory}$ , one being smaller than  $b_{\rm ne}^{\rm theory}$ , the other larger. While the difference in the  $b_{\rm ne}$  values ob-

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tained from the Bi-transmission experiments [3,10] is well understood [7,9,12], the Bragg-reflection method of Ref. [11] has never been revisited. In the latter experiment, the value of  $b_{ne}$  is extracted from the slope of b(Q) as a function of 1-f(Q) [see Eq. (4)]. The fitted data are the forward value  $b_{nuclear}$ , obtained from a Christiansen-filter experiment and the values for eight Bragg reflections  $b(Q_{hkl})$ , where  $Q_{hkl}$  is the transferred momentum for the Bragg reflection with Miller indices hkl.

For two different crystals, two very different values of  $b_{ne}$ were obtained in Ref. [11], i.e.,  $b_{ne} = -1.06$  and -2.2 $\times 10^{-3}$  fm, which led the authors to postulate the existence of an additional scattering process. We rather consider that this discrepancy reveals the systematic uncertainty of the experiment, where  $b(Q_{hkl})$  is deduced from an integralintensity measurement of the Bragg peak. This systematic uncertainty has probably been underestimated in Ref. [11] for different reasons: (i) extinction is neglected, and (ii) the uncertainty on the temperature factor (which is shown below to be of crucial importance) is neglected. Consequently, the accuracies obtained for  $b(Q_{hkl})$  (typically 0.01 fm) seem unrealistic as compared with accuracies generally obtained by this kind of measurement (typically 0.1 fm) [13]. A hint to our conjecture is also given by the strong disagreement between other results [14] obtained inter alia with an integralintensity measurement on monocrystals, and the Christiansen-filter results of Ref. [15] for W isotopes.

#### **III. THE BRAGG-REFLECTION TECHNIQUE**

For the hkl Bragg reflection, the transferred momentum reads

$$Q_{hkl}/4\pi = \sqrt{h^2 + k^2 + l^2}/2a_0 = \sin\theta_{hkl}/\lambda,$$
 (5)

where  $a_0$  is the side of the conventional cubic unit cell (we only consider cubic crystals here),  $\theta_{hkl}$  is the Bragg angle, and  $\lambda$  is the wavelength. This equation shows that  $\lambda$  must be of the same order of magnitude as  $a_0$ . For neutrons, this corresponds to thermal energies.

Let us now consider the method proposed by Shull in Ref. [4], which aims at measuring  $b(Q_{hkl})$  for Bragg reflections on monocrystals. We refer to Ref. [4] for details about the experimental setup. A collimated full-spectrum incident beam is Bragg reflected on a monocrystal blade (typical thickness t=1 cm) in Laue transmission geometry. The wavelength selection is made through the Bragg condition (5), the crystal and the detector being moved simultaneously to form angles  $\theta$  and  $2\theta$ , respectively, with the incidentbeam direction. As shown in the dynamical theory of Bragg reflection for neutrons [6,13], the reflected beam inside the crystal consists of two coherent waves, the so-called *Pendellösung* (pendulum solutions), interfering with each other and generating Pendellösung oscillations. An entrance and scanning gadolinium slits (width 0.13 mm) are placed on both faces of the crystal to extract these fringes. Making use of the symmetry of the interference pattern a very precise alignment of the scanning slit with respect to the entrance slit is possible. The intensity at the center of the interference pattern is then measured as a function of the wavelength and compared with the theoretical expression. This expression may be obtained either by the Kato spherical-wave dynamical theory [16] or by solving the Takagi-Taupin equations [6]. For a perfectly flat crystal and with infinitely-narrow slits, this intensity reads

$$I_{hkl}(\lambda) \propto I_0(\lambda) \lambda^2 |F_{hkl}|^2 J_0^2 \left[ \frac{t |F_{hkl}|}{a_0^3 \cos \theta_{hkl}(\lambda)} \lambda \right], \qquad (6)$$

where  $I_0$  is the incident intensity,  $J_0$  is a Bessel function, and  $|F_{hkl}|$  is the unit-cell structure factor of the crystal. This expression has to be slightly modified to take into account the finite curvature of the crystal, which adds the curvature radius R (typically 20 km) as a parameter to be fitted to the data, and to take into account the finite opening of the slits [16]. Typically, 40 fringes are observed for neutrons ranging the full spectrum of a thermal reactor. The structure factor can then be deduced with high precision from the period of these fringes. Shull's method is thus in essence an interferometric method [17], which explains why it provides very high accuracies (contrary to the integral-intensity method).

For a diamond-structure crystal such as silicon, the unit cell consists of four elementary cells with two atoms. The corresponding structure factor reads [13]

$$F_{hkl} = 4 \times (1 + i^{h+k+l}) \times b_{\text{meas}}(Q_{hkl}), \tag{7}$$

where  $b_{\text{meas}}(Q)$  is related to b(Q) through the Debye-Waller temperature factor [13]

$$b_{\text{meas}}(Q) = b(Q) \times \exp[-B(Q/4\pi)^2].$$
 (8)

To get a precise  $b(Q_{hkl})$  value, as needed to estimate  $b_{ne}$ , one needs precise values of both the structure factor  $F_{hkl}$  and the temperature parameter *B*. For instance, for the (111) reflection on Si (Z=14,  $a_0=5.43072$  Å), Shull's method provides [16]  $b_{meas}(Q_{111})=4.1053(8)$  fm. Using the Debye-Waller factor obtained by measuring x-ray Pendellösung fringes at room temperature [18],

$$B = 0.4613(27) \text{ Å}^2, \tag{9}$$

one gets

$$b(Q_{111}) = 4.1538(11)$$
 fm, (10)

where three additional units in the error are due to the error on *B*. Equation (8) shows that this additional uncertainty increases exponentially with  $Q^2$  (see Fig. 1).

Following the approach of Ref. [16], we can deduce  $b_{\text{nuclear}}$  from value (10) with Eq. (4), using the Argonne-Garching  $b_{\text{ne}}$  and the atomic form factor [19]:  $f(Q_{111}) = 0.7526$  (a precise value of f(Q) is actually not required [1]). This yields  $b_{\text{nuclear}} = 4.1495(11)$  fm (which slightly differs from the value of Ref. [16] because of the more recent temperature factor), in agreement with the more precise value obtained recently by non-dispersive-sweep neutron interferometry [20],

$$b_{\text{nuclear}} = 4.1507(2) \text{ fm.}$$
 (11)



FIG. 1. Experimental  $b_{\text{nuclear}}$  of Eq. (11) and  $b(Q_{111})$  of Eq. (10) compared with the theoretical curves (4) for  $b_{\text{nuclear}}$ =4.1507 fm,  $b_{\text{ne}}$ =-1.31 (Argonne-Garching), and -1.59×10<sup>-3</sup> fm (Dubna). The simulated points correspond to the reflections of Table I; they are calculated with the Argonne-Garching  $b_{\text{ne}}$  value, with error bars only due to the uncertainty on the temperature factor (9).

Conversely,  $b_{ne}$  can be deduced from values (10) and (11) with Eq. (4), which provides (see Fig. 1)

$$b_{\rm ne}^{\rm Si} = -0.89(32) \times 10^{-3}$$
 fm. (12)

This value agrees with the Argonne-Garching value but does not exclude the Dubna value (see Fig. 2). The same method (diamond structure with  $a_0$ applied to germanium = 5.6575 Å)  $b_{\text{nuclear}} = 8.1929(17) \text{ fm}$ using [21],  $b_{\text{meas}}(Q_{111}) = 8.0829(15) \text{ fm} [22], B = 0.57(1) \text{ Å}^2 [23], \text{ and}$  $f(Q_{111}) = 0.8542$  [19] leads to the value  $b_{ne}^{Ge} = 0.28(83)$  $\times 10^{-3}$  fm. Although measurements on Ge (Z=32) could, in principle, lead to a value of  $b_{\rm ne}$  two-times more precise than on Si (Z=14), this advantage is compensated by the two-times-larger scattering length, which implies a twotimes-larger absolute error on  $b(Q_{hkl})$ , the relative accuracy of Shull's method being 0.02% in both cases. Moreover, no  $b_{\text{nuclear}}$  value for Ge of comparable precision to Eq. (11) is available at present. Hence we only consider Si in the following.



FIG. 2. Comparison of the theoretical  $b_{\rm ne}$  value (3) with the Argonne-Garching and Dubna experimental values and with the result (12). The simulated point has the Argonne-Garching value with an error bar of  $0.06 \times 10^{-3}$  fm, as estimated in the text.

### **IV. PROPOSED EXPERIMENT**

In Fig. 1, values (10) and (11) are represented as a function of 1-f(Q). It is seen that the (111) reflection is not the optimal choice to calculate the slope of the line since it is close to the origin. Higher-order reflections are more appropriate, as shown by the simulated data. The error bars of these simulated points are calculated with the error  $\sigma_B$ = 0.0027 Å<sup>2</sup> on the temperature factor (9) only, assuming an ideal experiment with an infinite accuracy on  $b_{\text{meas}}$ . With such large uncertainties, there is no hope to reach a high precision on  $b_{\text{ne}}$ , in particular to distinguish between the Argonne-Garching (solid line in Fig. 1) and Dubna (broken line) values. However, if enough (at least two) reflections are measured, *both* B and  $b_{\text{ne}}$  could be deduced from the data. The precision of the extracted B and  $b_{\text{ne}}$  can be improved by increasing the number of reflections.

Let us now determine the possibly observable reflections. A diamond-structure crystal has a face-centered cubic lattice, which implies that all Miller indices have to be either even or odd [13]. Moreover, for a diamond structure, the structure factor (7) leads to three types of reflections: (i) forbidden when h+k+l=2+4n, where *n* is natural; (ii) weak when h+k+l=3 and (iii) strong when h+k+l=4+4n. A thermal-neutron beam has typically a maximum flux for  $\lambda = 1.2$  Å and the accessible angular range of the setup is assumed to be  $0 \le 2\theta \le 110^\circ$ . These conditions combined with Eq. (5) imply that sixteen reflections between (111) and (642) could be measured with reasonable intensities for Si.

However, if the incident beam has a typical full spectrum  $0.8 \leq \lambda \leq 2.5$  Å, contamination has to be taken into account. Let us first consider contamination by single reflections. For instance, while the (111) reflection is pure for  $15^{\circ} \leq 2\theta$  $\leq 45^{\circ}$  thanks to the absence of the (222) reflection which is forbidden (hence the result of Refs. [4,16]), (333) and (444) would be measured on  $45^{\circ} \le 2\theta \le 110^{\circ}$  and  $61^{\circ} \le 2\theta$  $\leq 110^{\circ}$ , respectively. This mixing implies that the reflected intensity would not be described by Eq. (6) and that no structure factor could easily be extracted. Among the 16 considered reflections, seven could be contaminated by single reflections. The remaining nine reflections are listed in Table I. Besides this single-reflection contamination, multiple scattering should also be taken into account. Since the theoretical description of this effect is rather involved [24] its impact must be evaluated for each specific experimental situation. In general, however, the impact of this effect on the measurement can be minimized or even avoided experimentally by a proper orientation of the crystal with respect to the Ewald sphere. We suppose that because of this possibility no consideration of this contamination was made in Refs. [4,16].

For the scope of the present paper it seems therefore well justified to assume that Eq. (6) is valid for the nine reflections of Table I. However, this requires experimental verification for each considered reflection in the actual measurement. Three crystal blades would suffice to measure them all [for instance, a crystal blade cut parallel to the (220) planes could be used to measure the (111), (422), (511), (533), (711), and (551) reflections]. Among these reflections, three are strong (underlined in Table I); the intensities range be-

TABLE I. Weak and strong (underlined) Bragg reflections which could be measured with a thermal-neutron reactor without a contamination problem. hkl are the Miller indices,  $f(Q_{hkl})$  is the atomic form factor (Ref. [19]),  $\lambda$  the neutron-wavelength interval, and 2  $\theta$  the corresponding angles between the incident and reflected beams.  $|F_{hkl}|^2$  is the squared structure factor which provides the reflection intensity through Eq. (7).

hkl	$f(Q_{hkl})$	λ (Å)	2 θ(°)	$ F_{hkl} ^2$ (Å <sup>2</sup> )
(111)	0.7526	0.8-2.5	15-47	540
(422)	0.4788	0.8 - 1.8	42-110	918
(511)	0.4600	0.8 - 1.7	45-110	448
(531)	0.4150	0.8 - 1.5	52-110	421
( <u>620</u> )	0.3902	0.8 - 1.4	56-110	811
(533)	0.3764	0.8 - 1.4	58-110	396
(551)	0.3432	0.8 - 1.2	63-110	372
(711)	0.3432	0.8 - 1.2	63-110	372
( <u>642</u> )	0.3249	0.8 - 1.2	67-112	715

tween 0.7 and 1.7 times the intensity of the already-measured (111) weak reflection. Finally, Eq. (6) allows to estimate the number of fringes which would be measured for each of them. For instance, 44 fringes would be observed for the (711) reflection if t=1 cm.

These considerations show that these higher reflections should not be more complicated to measure than the (111) reflection already measured in Refs. [4,16]. Hence, we can reasonably assume that the precision on  $b_{\text{meas}}$  would be equal to that of Ref. [16]: 0.0008 Å. With this assumption, the attainable precision for *B* and  $b_{\text{ne}}$  can be estimated. For *B*, we apply the approximate linear relation [Eqs. (4) and (8)]

$$\ln b_{\text{meas}}(Q) \approx \ln b_{\text{nuclear}} - B(Q/4\pi)^2, \qquad (13)$$

while for  $b_{\rm ne}$  we use Eq. (4). The uncertainty of a linear-fit slope can be calculated from the abscissas of the experimental points and from their errors [25]. Measuring the three strong reflections would give  $\sigma_B = 0.000 \, 40 \, \text{\AA}^2$  and  $\sigma_{b_{\rm ne}} = 0.11 \times 10^{-3}$  fm, while with the eight new reflections of Table I, one gets  $\sigma_B = 0.000 \, 27 \, \text{\AA}^2$  and  $\sigma_{b_{\rm ne}} = 0.06 \, \times 10^{-3}$  fm. The temperature factor would be ten times more

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precise than the most precise value (9) and the  $b_{ne}$  value would be nearly as precise as the Argonne-Garching and Dubna values (see Fig. 2).

#### **V. CONCLUSIONS**

In the present paper, we have proposed to extend the Pendellösung technique of Refs. [4,16] to higher-order Bragg reflections. Temperature effect and single-reflection contamination have been taken into account rigorously while multiple scattering contamination is assumed to be avoidable through a proper experimental setup. The proposed experiment would provide a precise value of the neutron-electron scattering length and hence of the neutron charge radius. Since the present method is similar to the Dubna method on W but much more precise because of its interferometric nature, such a new result would certainly be of great help to decide the  $b_{ne}$  discrepancy between the Dubna, Argonne-Garching, and Foldy values. Moreover, an interesting byproduct of the proposed measurement would be a precise value of the silicon temperature factor, an important quantity in solid-state physics.

These results show that the measurement of Pendellösung fringes in Bragg reflections is a very powerful technique. This is known for x rays [18,26] but generally ignored for neutrons. By confirming Shull's pioneer conclusion [4]

"... the high sensitivity of the fringe positions ... can be exploited ... [for] example ... in assessing the neutron-electron interaction strength ... "

the present work opens new prospects for the field of *neutron Pendellösung interferometry*.

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