First-principles quantum corrections for carrier correlations in double-layer two-dimensional heterostructures

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We present systematic *ab initio* calculations of the charge carrier correlations between adjacent layers of two-dimensional materials in the presence of both charged impurity and strain disorder potentials using the examples of monolayer and bilayer graphene. Our analysis yields unambiguous first-principles quantum corrections to the Thomas-Fermi densities for interacting two-dimensional systems described by orbital-free density functional theory. Specifically, using density-potential functional theory, we find that quantum corrections to the quasiclassical Thomas-Fermi approximation have to be taken into account even for heterostructures of mesoscopic size. In order for the disorder-induced puddles of electrons and holes to be anticorrelated at zero average carrier density for both layers, the strength of the strain potential has to exceed that of the impurity potential by at least a factor of ten, with this number increasing for smaller impurity densities. Furthermore, our results show that quantum corrections have a larger impact on puddle correlations than exchange does, and they are necessary for properly predicting the experimentally observed Gaussian energy distribution at charge neutrality.

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I. INTRODUCTION

The simulation of two-dimensional (2D) materials and prediction of their properties has become a mainstay of materials science over the past decade, with the promise and realization of valuable applications in both industrial technology and fundamental research [1]. Theoretical and computational methods for 2D materials have been advanced into a sophisticated machinery that enables researchers to deal with ever more realistic settings [2]. The widely used Kohn-Sham density functional theory (KS-DFT) [3,4] presents one particularly popular ab initio approach with the capability of accurately handling hundreds of interacting particles (and up to thousands of atoms in cases where linear-scaling methods apply [5–10]). The development of functionals for 2D systems has been lagging behind that of their 3D counterparts for various reasons: Some of the most heavily used 3D KS-DFT functionals are not bounded from below [11,12] when the 2D limit is approached. This stems from the improper scaling behavior of the density [11]. Furthermore, even consistent first-order gradient corrections of the kinetic energy density (used in developing metageneralized gradient approximations for KS-DFT) were unknown for 2D fermion systems until recently [13–15]. Nonetheless, considerable progress has been made and alternative derivations of some of those 2D functionals have been obtained since the turn of the millennium [16-25].

However, a systematic ab initio methodology that is universally applicable and scales favorably with particle number, thereby enabling high-throughput computations of mesoscopic systems, is not yet available-with orbital-free density functional theory (OF-DFT) being the suspected saviour for almost a century [26-32]. While KS-DFT scales cubically with particle number in generic settings, OF-DFT scales linearly, and sublinear scaling can be achieved in special cases. Functional development, in particular concerning the kinetic energy functional [33-35], and implementations of OF-DFT have gained momentum in recent years [30,32,36-41], also in conjunction with other techniques like ab initio molecular dynamics [42]. Chemical accuracy is approached in selected cases [43-45]. If quantum effects play a minor role or if the considered system is largely homogeneous, OF-DFT can also be used in its most basic form, the Thomas-Fermi (TF) approximation. For instance, the effect of exchange on large disordered systems with long-range interactions was studied in Ref. [46] using OF-DFT in TF approximation. The TF model is not only of historical significance, but presents, as an exact constraint for homogeneous systems and in the limit of infinite nuclear charges [47], an important base line for benchmarking proposed systematic density functional improvements. To what extent then do corrections to the TF approach play a crucial role or dominate over exchange effects in 2D materials? (They do, indeed, for a number of relevant fermionic systems, ranging from atomic Fermi gases to molecules and single atoms.)

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The most severe obstacle for OF-DFT in taking over as the workhorse of theoretical chemistry and materials science is the lack of accurate, reliable, systematic, and preferably universal quantum corrections to the quasiclassical TF approximation, in particular for the kinetic energy of low-dimensional systems [13–15,48,49]. While ad-hoc corrections to the quasiclassical limit and heuristic approximations are available for kinetic energy and particle density of low-dimensional systems [50,51], successful derivations of systematic and consistent corrections are scarce [15,49,52-54]. One promising route towards systematic orbital-free quantum corrections is provided by density-potential functional theory (DPFT) [15,49,54–57], a more flexible reformulation of the original Hohenberg-Kohn DFT [4,58], which circumvents the need for an explicit kinetic-energy density functional and provides natural ways for systematic semiclassical expansions.

In this paper we explore the applicability of DPFT for 2D materials by assessing quantum corrections to the TF approximation for double-layer heterostructures of mono- and bilayer graphene. Of particular interest to us are situations that are not easily tackled with orbital-based techniques, for example 2D material sheets of mesoscopic size that are subjected to aperiodic disorder potentials. Such situations are for example of current interest in studies on Coulomb drag [59] where there exists an unsettled controversy as to whether the behavior of drag measured in experiment [60] is due to correlation [61] or anticorrelation [62] between the density fluctuations of the layers.

Our work contributes in several ways to answering some of the questions raised above. Sections II and III provide the computational framework for obtaining quantum-corrected carrier densities of 2D materials using DPFT. The expressions for the semiclassical particle densities developed here enable us to decide whether or not the quasiclassical TF approximation is sufficient for describing at least conglomerate properties like average interlayer correlations of heterostructures. Section IV introduces the generic double-layer system, with both layers subjected to one layer of charged impurities, while only one of the layers is strained. Charge and strain disorder potentials are expected to compete in creating correlated (from charged impurities) and anticorrelated (from strain) carrier densities in the two layers. Our model setup is designed to extract the strain strengths required for switching between correlation and anticorrelation. In Sec. V we apply our approach to double-monolayer graphene and doublebilayer graphene. We discuss whether or not the electron-hole puddles of both layers, interacting electrostatically, require a self-consistent interlayer treatment. Finally, we analyze the effects of quantum corrections and exchange energy on the correlations with the aid of phase diagrams that chart the correlation measures as functions of impurity density, carrier density, and ratios of strain and charge disorder. The Appendix gathers background information on the units, system parameters, correlation measures, and numerical procedures employed here.

II. DENSITY-POTENTIAL FUNCTIONAL THEORY

Instead of resorting to the computationally demanding orbital-based Kohn-Sham DFT, we make use of orbital-free density-potential functional theory (DPFT) [55,56]. It is formally equivalent to the Hohenberg-Kohn formulation but makes systematic improvements upon the TF approximation technically feasible—in particular for low-dimensional systems.

Specifically, by Legendre-transforming $E_{kin}[n]$, the kinetic energy functional of the particle density $n(\mathbf{r})$, w.r.t. the new variable $V(\mathbf{r}) = \mu - \frac{\delta E_{kin}[n]}{\delta n(\mathbf{r})}$, we recast the total energy of an interacting quantum system with interaction energy $E_{int}[n]$,

$$E[n, \mu] = E_{\text{kin}}[n] + E_{\text{ext}}[n] + E_{\text{int}}[n] + \mu \left(N - \int (d\mathbf{r}) n(\mathbf{r}) \right), \quad (1)$$

as the density-potential functional

$$E[V, n, \mu] = E_1[V - \mu] - \int (d\mathbf{r}) n(\mathbf{r}) \left(V(\mathbf{r}) - V_{\text{ext}}(\mathbf{r}) \right)$$
$$+ E_{\text{int}}[n] + \mu N.$$
(2)

Here, the external potential $V_{\text{ext}}(\mathbf{r})$ yields the external energy $E_{\text{ext}}[n]$, and the particle number N is enforced via the Lagrange multiplier μ , viz. the chemical potential. From Eq. (2) we obtain the ground-state solutions of the three variables V, n, and μ by self-consistently solving

$$n(\mathbf{r}) = \frac{\delta E_1[V - \mu]}{\delta V(\mathbf{r})},\tag{3}$$

$$V(\mathbf{r}) = V_{\text{ext}}(\mathbf{r}) + \frac{\delta E_{\text{int}}[n]}{\delta n(\mathbf{r})},$$
(4)

$$N = \int (d\mathbf{r})n(\mathbf{r}). \tag{5}$$

Equation (5) is obtained by combining $\partial E[V, n, \mu]/\partial \mu$ with Eq. (3) and reveals the particle number constraint in Eq. (1).

Equations (3)–(5) are exact and reminiscent of the KS scheme but without the need of orbitals. However, the noninteracting case aside, we have to approximate the unknown potential functional $E_1[V - \mu]$; here the subscript indicates that E_1 can be written in terms of a single-particle trace over a function of the single-particle Hamilton operator [56]. We also have to provide the equally important interaction energy $E_{int}[n]$ as an explicit functional of the particle density n. Approximate particle densities follow directly from approximations of $E_1[V - \mu]$ (or, rather, its functional derivative) for any given potential V. As is evident from Eq. (4), V constitutes an effective single-particle potential with interaction effects effectively included for any given density n.

Following Refs. [15,49,54,56,57,63] we approximate E_1 by its noninteracting version as the single-particle trace

$$E_1[V - \mu] = \operatorname{tr}\{(H_1 - \mu) \,\eta(\mu - H_1)\},\tag{6}$$

where $H_1 = H_1(\mathbf{R}, \mathbf{P}) = T(\mathbf{P}) + V(\mathbf{R})$ is a single-particle Hamiltonian with dispersion relation *T* and potential energy *V*, while the trace includes the degeneracy factor *g*. For example, g = 4 accounts for the spin and valley multiplicity of unpolarized charge carriers in the cases of mono- and bilayer graphene. **R** and **P** are the position and momentum operators, respectively, and $\eta()$ denotes the step function.

The explicit expression for E_1 in Eq. (6) results in an explicit expression for the particle density in terms of

TABLE I. Thomas-Fermi density $n_{\text{TF}}(\mathbf{r})$ and quantum-corrected density $n_3(\mathbf{r})$ for linear and quadratic dispersion in 2D from evaluating Eq. (7), with degeneracy factor g, Bessel functions $J_{\alpha}(), v(z) = [\mu - V(z)]_+, K(z) = v(z)/(\hbar v_{\text{F}}), \sigma(z) = 2m v(z)/\hbar^2$, and $[x]_+$ denoting $x \eta(x)$. We recover the TF densities from their quantum-corrected successors upon replacing $K(\mathbf{r} + \mathbf{s})$ and $\sigma(\mathbf{r} + \mathbf{s})$ by their local versions $K(\mathbf{r})$ and $\sigma(\mathbf{r})$, respectively. The case of quadratic dispersion is dealt with in Ref. [54]; see Appendix F for the derivations in the case of linear dispersion.

$T(\boldsymbol{p})$	$n_{\mathrm{TF}}(\boldsymbol{r})$	$n_3(\mathbf{r})$		
$v_{\rm F} \boldsymbol{p} $ $\frac{\boldsymbol{p}^2}{2m}$	$\frac{\frac{g}{4\pi}K(\boldsymbol{r})^2}{\frac{g}{4\pi}\sigma(\boldsymbol{r})}$	$\frac{\frac{g}{4\pi^2} \int (ds) \frac{K(r+s)^3}{ s } J_1(2 s K(r+s))}{\frac{g}{4\pi^2} \int (ds) \frac{\sigma(r+s)}{ s ^2} J_2(2 s \sqrt{\sigma(r+s)})$		

arbitrary functions $V(\mathbf{r})$ via Eq. (3). The approximate nature of Eq. (6) aside, the exact particle density including all quantum corrections is thereby obtained for any specified interaction energy $E_{int}[n]$ and without reference to orbitals. Specifically, Eqs. (3) and (6), together with the Fourier transform of the step function, yield the particle density [64]

$$n(\mathbf{r}) = g \int_{\neg \forall i} \frac{dt}{2\pi i t} e^{\frac{it}{\hbar}\mu} \langle \mathbf{r} | e^{-\frac{it}{\hbar}H_1} | \mathbf{r} \rangle, \tag{7}$$

see Refs. [54,65-68].

We seek to approximate the time evolution operator $U = e^{-\frac{u}{\hbar}H_1}$ systematically via split-operator methods, for example of the Suzuki-Trotter type [54,69]. The quasiclassical approximation $U \approx U_2 = e^{-\frac{u}{\hbar}T(P)}e^{-\frac{u}{\hbar}V(R)}$ recovers the quasiclassical TF density n_{TF} , while $U_3 = e^{-\frac{u}{\hbar}T(P)}e^{-\frac{u}{\hbar}V(R)}e^{-\frac{u}{\hbar}T(P)}$ produces the first quantum-corrected density n_3 in a series of expressions that utilize higher-order factorizations [54] of U. We give the corresponding 2D densities for linear and quadratic dispersion in Table I and outline the derivation of n_3 for the case of linear dispersion in Appendix F. In contrast to $n_{\text{TF}}(r)$, which is restricted to classically allowed regions of the potential and only depends on the local value V(r), $n_3(r)$ samples V in an extended region and exhibits evanescent tails beyond the quantum-classical border.

III. SELF-CONSISTENT SIMULATION OF DISORDERED 2D MATERIALS

We target 2D systems with chemical potential in the vicinity of the Dirac point (the point where valence and conduction bands touch) for graphene (effective bilayer graphene). The usual tight-binding approach absorbs the lattice structure in an effective Hamiltonian and yields noninteracting quasiparticles in a homogeneous $[V_{ext}(\mathbf{r}) = 0]$ environment. The electronic structures, viz. atoms, of the materials are thus not modeled explicitly here. The single-particle energies, viz. band structure, associated with such quasiparticles is the dispersion relation $T(\mathbf{p})$ whose operator version appears in Eq. (6). $T(\mathbf{p})$ can be an arbitrary function, but for the purpose of this work we shall restrict ourselves to the analytically more tractable cases of linear and quadratic dispersion [70].

In the following we outline the procedures involved for arriving at the ground state solutions of Eqs. (3)–(5); further details are provided in Appendix E. Upon adding external

potentials $V_{\rm C}(\mathbf{r})$ and $V_{\rm S}(\mathbf{r})$ that model charged impurities and strain, respectively, we initiate the self-consistent loop of Eqs. (3)–(5) by evaluating the density [denoted $n_{-}(\mathbf{r})$ for the quasiparticles that follow the dispersion of the conduction band] with the external potential

$$V_{\text{ext}}(\boldsymbol{r}) = V_{\text{C}}(\boldsymbol{r}) + V_{\text{S}}(\boldsymbol{r})$$
(8)

for these conduction quasiparticles, see Appendix C for details. Since no interactions are included at this stage, the effective potential is $V_{-}(\mathbf{r}) = V_{\text{ext}}(\mathbf{r})$. The density $n_{+}(\mathbf{r})$ of valence quasiparticles, which follow the inverted dispersion, e.g., $T(\mathbf{p}) = -v_{\text{F}}|\mathbf{p}|$ in the case of graphene, is built from the same density expression as $n_{-}(\mathbf{r})$ but takes as an input the inverted potential $2(\mu + \Delta) - V_{\text{ext}}(\mathbf{r})$, with an optional band gap Δ (in the case of mono- and bilayer graphene, we have $\Delta = 0$). The effective potential for the valence quasiparticles reads

$$V_{+}(\mathbf{r}) = 2(\mu + \Delta) - V_{-}(\mathbf{r})$$
(9)

and equals the external potential for the valence quasiparticles if interactions are omitted. The such obtained carrier density $n(\mathbf{r}) = n_{-}(\mathbf{r}) - n_{+}(\mathbf{r})$ updates the effective potential via the interaction contribution in Eq. (4).

As an approximate interaction energy $E_{int}[n]$ for the quasiparticles, we employ the regularized Hartree term for the Coulomb energy,

$$E_{\rm H}[n] = \frac{W}{2} \int (d\mathbf{r}) (d\mathbf{r}') \frac{n(\mathbf{r}) n(\mathbf{r}')}{\max(|\mathbf{r} - \mathbf{r}'|, b)}, \qquad (10)$$

where *b* is half the lattice constant of the numerical implementation and $W = \hbar v_F r_s$, with the ratio r_s of Coulomb potential energy and kinetic energy for graphene. Upon functional differentiation, Eq. (10) leads to the Hartree potential [71]

$$V_{\rm H}(\boldsymbol{r}) = W \int (d\boldsymbol{r}) \frac{n(\boldsymbol{r}')}{\max(|\boldsymbol{r} - \boldsymbol{r}'|, b)}$$
(11)

as an approximate interaction contribution in Eq. (4). Eyeing means of comparison and higher accuracy, we may supplement $E_{\rm H}$ with an exchange energy [72], leading to the exchange potential $V_{\rm X}$; see Appendix B for details.

The updated effective potential V then determines a new quasiparticle density $n_{-}(\mathbf{r})$ via Eq. (3), thereby closing the self-consistent loop. This process is repeated until a predefined relative precision is reached (we find 10^{-6} to be sufficient) when comparing the local densities of subsequent loop iterations. The chemical potential is adjusted in each iteration to enforce a given particle number, viz. average carrier density. Figure 1 highlights the differences in the converged quasiparticle densities $n_{\rm TF}$ and n_3 of a single graphene layer with and without exchange. Both exchange and quantum corrections tend to decrease the peaks of the density landscape. This effect is well known in the case of exchange [46], and a smoothening of the carrier density in an external potential $V_{\rm C}$ is to be expected when tunneling starts to play a role with the inclusion of quantum corrections. In fact, when comparing the corresponding densities in Fig. 1 we find that quantum corrections considered in this work can dominate over exchange effects.

The analysis of disorder averages reveals a striking instance of this observation. As observed in Fig. 1 for a single



FIG. 1. The spatial distributions of quasiparticle densities $n_{\rm TF}$ and n_3 of monolayer graphene visibly depend on exchange and quantum corrections. The upper left panel shows the Thomas-Fermi density $n_{\rm TF}$. The other three panels display the differences between $n_{\rm TF}$ and the quantum-corrected density n_3 (upper right panel), the TF density with the exchange potential of Eq. (B1) included (lower left panel) and n_3 including exchange (lower right panel). The color bar refers to densities measured in units of $0.001/l^2$. The horizontal and vertical axes indicate spatial position, in the units of l as labeled explicitly in the lower right panel. The same disorder realization, with average impurity density $\bar{n}_{\rm imp} = 10^{12}/{\rm cm}^2$, is used in all four panels.

disorder realization, n_3 and n_{TF} differ in their density distribution function. For n_3 the integration over a finite region in the disorder landscape, see Table I, tends to result in smoother densities compared with n_{TF} . This effect can be

quantified by density histograms compiled from many disorder realizations. The density histograms in Fig. 2, calculated for a graphene monolayer on SiO₂ with $\bar{n}_{imp} = 10^{12}/\text{cm}^2$, corroborate the snapshot of one disorder realization in Fig. 1. Figures 1 and 2 highlight the (a priori) importance of not only addressing exchange but also quantum corrections beyond the TF approximation for quantitatively viable investigations of 2D materials via orbital-free DFT: Both the inclusion of exchange and of quantum corrections indicate less pronounced peaks of the carrier-density landscape. Compared to the TF approximation, the quantum corrections exhibit an averaging effect with broader density distributions that have relatively more weight on intermediate values of the density rather than a strong distribution maximum at zero density; cf. Fig. 2 (left). Although exchange also shows visibly less pronounced densities in Fig. 1, this effect stems from a global reduction in density variance rather than a redistribution of densities from very small towards intermediate values. The distributions of the quantum-corrected densities (including exchange) are captured by their Gaussian ('G') or Lorentzian ('L') fits (with offset) more accurately than the distributions resulting from the TF approximation. This observation is in line with experimental results for graphene that point towards Gaussian distributions of density and energy in the presence of charged disorder [73,74]. In Fig. 2 we compare our calculations ' $n_{TF}(+X)$ ' and ' $n_3(+X)$,' with their Gaussian fits 'G(TF)' and 'G,' directly with the experimental data 'exp.' from scanning tunneling spectroscopy; cf. Ref. [74]. We find our quantum-corrected approach to predict the experimental data much better than what can be obtained from the TF approximation-in both qualitative and quantitative terms. In view of this stark improvement over the TF approximation, we want to stress again that our quantum-corrected density expressions are based on first principles without adjustable parameters or fits and rely solely on controlled approximations to quantum mechanics.



FIG. 2. Quantum corrections are crucial for obtaining the experimentally observed Gaussian energy distribution function for monolayer graphene on SiO₂ at charge neutrality. Left: Density histograms for n_{TF} and n_3 , with and without exchange (X), for 500 disorder realizations with $\langle n_j \rangle = 0$. We choose the bin widths for the densities such that 100 counts per bin are obtained on average. In contrast to the TF approximation, the quantum-corrected density distribution n_3 (including exchange) is captured reasonably well by a Lorentzian and even better by a Gaussian fit. Right: Translating local densities into energies $E_D = \text{sgn}(n(r)) \hbar v_F \sqrt{\pi |n(r)|}$, with the signum function sgn(), we find that the local quantum-corrected energies E_D follow a Gaussian distribution that resembles the Gaussian fit to the experimental data 'exp.' (extracted from Ref. [74]) remarkably well. This is in stark contrast to the results of the TF approximation. We convert counts/bin into frequencies by renormalizing the histograms with the maxima of the Gaussian fits 'G' and 'G(TF),' respectively.



FIG. 3. Schematics of the double-layer setup investigated here. Two finite-size monolayers are embedded in h-BN at a layer separation of 5 nm, while the SiO₂ substrate provides the charged impurities at an effective distance [75-77] of 1 nm from the first layer L1. For the purpose of revealing the impact of strain and charge disorder on the electron-hole-puddle correlations between L1 and L2, we model L2 as an unstrained clean layer. We use the same setup for analyzing both monolayer and bilayer graphene heterostructures.

IV. DOUBLE LAYER SETUP

The treatment of monolayers in the previous section forms our basis for the description of more complicated heterostructures. Figure 3 illustrates a two-layer system, where both layers L1 and L2 are sandwiched between h-BN and subjected to a charged impurity layer from the SiO₂ substrate. We expose L2 to the same charge disorder that affects L1, though at a larger separation, but refrain from adding disorder on L2 in order to avoid confusing interlayer correlation effects with effects from independent disorder on L2. For the same reason we model the strain of L1 and the charge disorder by the identical type of disorder, albeit in different realizations. The layer separation of 5 nm suffices to justify a merely classical electrostatic interaction between L1 and L2, i.e., interlayer tunneling of charge carriers can be neglected-in contrast to intralayer tunneling through the disorder potential landscape. The latter is missed by densities in TF approximation but captured (in part) via the higher-order Suzuki-Trotter factorizations.

In what follows we address the puddle correlations between L1 and L2 as a function of the ratio R between the disorder strength of the strain and that of the charged impurities. Further details are provided in Appendix C. The sum of $V_{\rm C}$ and $V_{\rm S}$ results in electron-hole puddles within the first layer L1, whose electrostatic potential adds to the external potential $V_{\rm C}$ for the charge carriers in the second layer [78]. We expect maximal puddle correlation if R = 0, that is, when no strain can obscure the then dominating effect of the charged impurities on both layers: Owing to the intralayer Coulomb interaction, the puddles in L1 are much reduced in weight compared with the case of noninteracting carriers. That is, the tendency of a puddle in L1 to electrostatically induce a puddle of opposite charge in L2 is overcompensated by the charge disorder, which exhibits the tendency to induce a puddle of the same charge [79]. Following the same line of reasoning, we expect maximal anticorrelation if $R = \infty$, with the transition from correlation to anticorrelation occuring at some value R > 1. In the following section we substantiate these claims with quantitative predictions for graphene and bilayer graphene.

V. DENSITY CORRELATIONS IN DOUBLE LAYERS OF MONO- AND BILAYER GRAPHENE

We quantify the interlayer correlations of electron-holepuddles of the double-layer system described in Sec. IV by solving Eqs. (3) and (4) self-consistently [80] and by comparing the converged carrier densities $n_1 = n(L1)$ and $n_2 = n(L2)$ of layers L1 and L2 locally. To that end we calculate the two correlation measures $\xi[n_1, n_2]$ and $\xi_c[n_1, n_2]$, which yield a value of one for perfectly correlated electronhole puddles (i.e., if the density distribution n_2 is proportional to n_1 and their values have the same sign at each position \mathbf{r}), minus one for perfect anticorrelation (i.e., if n_2 is proportional to $-n_1$), and are designed for tracking the transition between these two extremes; see Appendix D for details.

Figure 4 depicts potentials and densities for graphene, viz. linear dispersion, calculated for mean carrier densities $\langle n_j \rangle = 0$ and equal strengths of strain and charge disorder (ratio R = 1). Due to the screening effects of the Coulomb interaction within L1, the effective potential for L1 exhibits less variability than the total external potential, i.e., the sum of the disorder potentials V_C and V_S . The quasiparticle density n_1 of L1, which can be viewed as resulting from this effective potential, induces an external electrostatic potential for the carrier density n_2 of L2. For R = 1 the charge disorder potential for n_2 exceeds the electrostatic potential caused by n_1 by a factor of more than 50. For the setting that leads to Fig. 4, the magnitudes of the total external potential for L2 are smaller than those of L1 by a factor of 3–5.

As a result, the density fluctuations of n_2 are diminished compared with those of n_1 by a factor of 5–10. It is therefore well justified to refrain from a self-consistent treatment of the electrostatically induced potentials V_{ind} of both layers and to consider only V_{ind} stemming from n_1 . As is evident from the bottom row of Fig. 4, the spatial distributions of n_1 and n_2 are correlated rather than anticorrelated for R = 1.

Repeating the calculation which yields the results illustrated in Fig. 4 for different values of *R*, we find the critical value R_0 at the transition from interlayer correlation to anticorrelation. The correlation measure ξ signifies interlayer puddle-correlation (anticorrelation) by taking on positive (negative) values. The corresponding diagram in Fig. 5, calculated with charged impurities density $\bar{n}_{imp} = 10^{12} \text{ cm}^{-2}$, exhibits $R_0 \approx 10 \pm 2$ when n_3 is used, and a similar value in the case of n_{TF} . Here, we take into account exchange effects and report a rough error estimate simply based on our numerical findings from five disorder realizations [81]. Evidently, the disparity in electron-hole puddle landscapes between n_3 and n_{TF} as seen in Fig. 1 does not translate into an appreciable difference between $\xi^{(3)+X}$ (for n_3) and $\xi^{\text{TF}+X}$ (for n_{TF}).

Although it is not surprising per se that integrated quantities like ξ are less sensitive to local differences between n_3 and n_{TF} , it cannot be assumed *a priori*. Our quantitative analysis shows that a quasiclassical approach to interlayer correlations in monolayer-graphene heterostructures is justified in the case of rather large impurity densities like of 10^{12} cm⁻² used for Fig. 5. Commonly, however, the TF approximation is less reliable for smaller particle numbers, and quantum corrections can be expected to play a more



FIG. 4. Visibly correlated carrier densities n_3 (including exchange) for layers L1 and L2 (bottom row) from impurity and strain potentials of equal strength R = 1 (top row): This figure illustrates the intermediate potentials relevant in the work flow towards carrier distributions for a double-monolayer graphene system as depicted in Fig. 3. The color codes are in units of u for energy and l^{-2} for density, respectively, and apply to both graphics in each row. Top row: Charged impurity potential $V_{\rm C}$ (left) and strain potential $V_{\rm S}$ (right) for L1. Second row: Total external potential Vext (left) and converged effective potential V (right) for L1. Third row: Charged impurity potential $V_{\rm C}$ for L2 (left) and electrostatically induced potential V_{ind} on L2 from the charge distribution n_1 of L1 (right, scaled up by a factor of 50). Fourth row: Total external potential V_{ext} (left) and converged effective potential V (right) for L2. Bottom row: Converged carrier densities n_1 (left, scaled by a factor of 1000) and n_2 (right, scaled by a factor of 5000).

dominant role as the carrier density is reduced. The disorder potential for smaller \bar{n}_{imp} is less pronounced and gives rise to puddles that exhibit smaller carrier densities on average. Indeed, with $\bar{n}_{imp} = 10^{11}/\text{cm}^2$ employed for Fig. 6,



FIG. 5. Both the TF- and quantum-corrected approach predict a transition from correlated to anticorrelated electron-hole puddles at $R = R_0^{(3)+X} \approx R_0^{\text{TF+X}} \approx 10 \pm 2$, where *R* is the ratio of the strength of strain to charged impurity disorder. The correlation measure ξ is plotted as a function of *R* for five disorder realizations of double monolayer graphene, extracted from both TF- and quantum-corrected carrier densities with exchange incorporated. The average density of charged impurities is $\bar{n}_{\text{imp}} = 10^{12}/\text{cm}^2$. The error bars demarcate minimal and maximal values of ξ found within the set of disorder realizations; the shaded areas and the dotted line segments guide the eye; the horizontal line at $\xi = 0$ separates the correlation phase (above) from the phase of anticorrelation (below). We use a logarithmic horizontal axis in order to more clearly showcase the transition point and the contrast of ξ over a large range of *R*.

the quantum-corrected $\xi^{(3)+X}$ can be clearly distinguished from the quasiclassical $\xi^{\text{TF}+X}$. Our data shown in Fig. 6 point to the critical value $R_0^{\text{TF}+X} \approx 15 \pm 2$ with an error estimate similar to that in Fig. 5. An increased R_0 at lower \bar{n}_{imp} can be understood from the following simplified picture. With typical values $v_{\text{C}}|_d$ and v_{S} of the charge and strain potentials for L1 and neglecting interactions, we have typical values $n_{\text{TF}}(\text{L1}) \propto (\mu - v_{\text{ext}}(\text{L1}))^2 \approx (v_{\text{C}}|_d + v_{\text{S}})^2$ for $\mu \approx 0$, i.e., approximately $n_{\text{TF}}(\text{L1}) \propto v_{\text{S}}^2$ at $R \approx 10$. Then, the typical



FIG. 6. At lower values of \bar{n}_{imp} , quantum corrections lead to stark differences in the crossover behavior. Here we repeat the plot in Fig. 5, but for $\bar{n}_{imp} = 10^{11}/\text{cm}^2$, and an increased sheet size to ensure enough disorder statistics (640 impurities on $(800 \text{ nm})^2$). The extended crossover of $\xi^{(3)+X}$ from correlation to anticorrelation contrasts with the pattern observed in Fig. 5 and heralds the emergence of quantum effects (i.e., major deviations from $\xi^{\text{TF}+X}$) with decreasing particle numbers.



FIG. 7. Like Fig. 5, but for double bilayer graphene. Using n_3 , we extract $R_0^{(3)} \approx 10 \pm 3$ for the transition from correlated to anticorrelated carrier densities but a somewhat smaller value $R_0^{\text{TF}} \approx 8 \pm 1$ in the case of the TF approximation. Similar to our observations for double-monolayer graphene, we find that quantum corrections have little effect in the case of double-bilayer graphene for $\bar{n}_{\text{imp}} = 10^{12} \text{ cm}^{-2}$.

TF density in L2 is determined by $v_{\text{ext}}(\text{L2}) = v_{\text{C}}|_{\tilde{d}} + v_{\text{ind}}$, where the typical values $v_{\text{C}}|_{\tilde{d}} \approx \frac{1}{2}v_{\text{C}}|_{d}$ at $\tilde{d} - d \approx 5$ nm and $v_{\rm ind} \propto v_{\rm S}^2$ are roughly equal if $\bar{n}_{\rm imp} = 10^{12}/{\rm cm}^2$ and R = 10. For a scaled $\bar{n}_{\rm imp} = \lambda \times 10^{12}/{\rm cm}^2$ and the same R = 10, $v_{\text{ext}}(L1)$ scales with λ as well, but the typical values of $n_{\rm TF}(L1)$ then scale like $\lambda^2 v_{\rm S}^2$, such that $v_{\text{ext}}(\text{L2}) = \lambda v_{\text{C}}|_{\tilde{d}} + \lambda^2 v_{\text{ind}}$. For the case of $\lambda = 1/10$, as represented by Fig. 6, the strength of v_{ind} , relative to $v_C|_{\tilde{d}}$, is diminished by a factor of ten. As strain feeds into v_{ind} , not into $v_{\rm C}|_{\tilde{d}}$, (relatively) more strain is required to counteract the effect of $v_{\rm C}|_{\tilde{d}}$, implying a larger critical R_0 at lower impurity densities. Disregarding the uncertainties for the critical R, one could estimate $R_0^{(3)+X} \approx 30$ for the quantum-corrected corre-lations. However, $\xi^{(3)+X}$ rather exhibits an extended crossover regime, where the magnitude of strain can be varied substantially with little effect on the average correlation $\xi^{(3)+X}$. Our main result is thus that corrections to the TF approximation can become important even for integrated or averaged observables of 2D materials-given the proper conditions, for example, small carrier densities.

We briefly turn to the impact of exchange effects in interlayer correlations. Since quantum corrections modify the TF carrier distributions of the individual layers more profoundly than exchange does, cf. Fig. 1, we expect that exchange plays a minor role in determining carrier correlations of doublelayer systems. This is confirmed with Table II in Appendix B, where ξ is determined in TF approximation for a single disorder realization, with and without exchange. In view of the magnitude of uncertainties displayed in Figs. 5 and 6 we find negligible quantitative differences when omitting or including exchange.

Our results on double bilayer graphene in Fig. 7 show a slight quantitative difference between the transitions of $\xi^{(3)}$ and ξ^{TF} for the here employed density of charged impurities $\bar{n}_{\text{imp}} = 10^{12} \text{ cm}^{-2}$. However, the overall trend for both approximations is still very similar and the spread of ξ , originating in multiple disorder realizations like in Figs. 5 and 6, still implies $R_0^{(3)} \approx R_0^{\text{TF}}$ within the depicted



FIG. 8. Determination of the critical ratio $R_0^{\text{TF}} \approx 8$ for double bilayer graphene (found in Fig. 7) as a function of $\langle n_2 \rangle$, viz $\mu(\text{L2})$. The findings for ξ^{TF} in Fig. 7 correspond to $\langle n_2 \rangle = 0$ here. All data points are obtained for a single fixed disorder realization.

uncertainties. Extrapolating from the double-monolayer case the double-bilayer graphene can be expected to exhibit more pronounced differences between $\xi^{(3)}$ and ξ^{TF} for smaller density of charged impurities.

Figure 8 provides another angle on the transition from correlation to anticorrelation for double bilayer graphene: For fixed $\langle n_1 \rangle = 0$ we chart ξ^{TF} as a function of $\langle n_2 \rangle$ and find a qualitative change in the shape of the curves $\xi(\langle n_2 \rangle)$ as *R* transits across $R_0^{\text{TF}} \approx 8$, consistent with our findings in Fig. 7.

VI. CONCLUSIONS AND PERSPECTIVES

The main result of this work is thus that both exchange and quantum corrections beyond the Thomas-Fermi approximation are essential for quantitatively reliable density distributions in the context of 2D materials. While exchange plays a minor role for some integrated quantities like interlayer correlations of double-layer systems, quantum corrections become important for smaller average carrier densities, viz. cleaner systems. In other words, systems like graphene on boron nitride samples with $\bar{n}_{\rm imp} \approx 10^{10} \ {\rm cm}^{-2}$ call for a more sophisticated description than the Thomas-Fermi approximation can provide. We showed that the regime of high carrier densities may still be described quasiclassically in the computationally highly efficient Thomas-Fermi approximation. For specific systems, however, quantum corrections have to be taken into account at the quantitative level. In this work we provided one such example when analyzing the crossover from correlation to anticorrelation in bilayer heterostructures of 2D materials. We focused on describing double layers of mono- and bilayer graphene via recently developed techniques of calculating quantum corrections, but our approach can be easily adapted for other heterostructures.

Turning to the issue of Coulomb drag [60], our analysis reveals that the strain potential has to be at least ten times stronger than the impurity potential in order for anticorrelation between the layers to occur in both monolayer and bilayer graphene. Any strain potential weaker than this will result in correlation. The DPFT framework presented here thus provides the following path towards an experimental verification of the nature of correlations in graphene heterostructures, a nontrivial task since it is impossible to directly measure the local densities of encapsulated 2D layers via local scanning tunneling microscopy. The strength of the strain potential [82] may first be extracted from local height fluctuation data obtained via atomic force microscopy [83], followed by the impurity strength from conductivity measurements [84]. The resulting ratio R may then be compared against Figs. 5 to 7 to determine the nature of correlation between the layers. This would be especially useful for clarifying the source of the unexpected sign changes [85-87] in Coulomb drag experiments that have been explained by earlier works simply asserting correlation [61,88] or anticorrelation [60,62] without proof. In the case of bilayer graphene, Ref. [89] theoretically demonstrated that a multiband mechanism based on the thermal smearing of Fermi energy in both layers explains the unexpected sign changes at high densities where puddles may be ignored. A complete analysis at the double neutrality where both puddles and multiband effects are important is however still lacking. In the future, this may be addressed by generalizing the DPFT framework presented here to incorporate multiband effects [89] in the determination of puddle-induced quantities such as the interlayer correlation and density fluctuations.

Finally, the findings of this work are useful in the study of van der Waals heterostructures [1] consisting of a pair of two-dimensional electronic layers separated by thin dielectric spacers. These structures serve as ideal platforms for the study of a range of interesting physical effects such as exciton condensation [90–93] and strong light-matter interaction [94–96] and are thus an area of intense research activity. It is likely that the interlayer puddle correlations calculated within this work play a role in the above physics.

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APPENDIX A: UNITS AND SYSTEM PARAMETERS

Throughout this work we use cgs units, measure lengths in $l = \sqrt{3}a \approx 2.46$ Å, where *a* is the lattice constant of graphene, energies in $u = \hbar v_{\rm F}/l \approx 2.678$ eV (for monolayer graphene, with the Fermi velocity $v_{\rm F} = 10^6$ m/s) and $u' = \hbar^2/(m l^2) \approx 36.71$ eV (for bilayer graphene with quasiparticle mass [97] $m \approx 0.39$ eV/(2 $v_{\rm F}^2$)). With the elementary charges in cgs and SI units connected via $e_{\rm SI} = \sqrt{4\pi} e_{\rm cgs}$ and the dielectric constants related via $\epsilon = \kappa \epsilon_0$, the Coulomb coupling strength in Eqs. (10) and (B1) reads

$$W = \frac{e_{\rm cgs}^2}{\kappa\epsilon_0} = \hbar v_{\rm F} r_{\rm s} \approx \frac{2.187}{\kappa} u l \tag{A1}$$

for graphene and

$$W \approx \frac{0.1596}{\kappa} u' l \tag{A2}$$

for bilayer graphene. The static limit [98] for h-BN amounts to setting $\kappa = 5.09$.

We restrict the numerical evaluations to squareshaped sheets of 2D materials with edge length of L = 200 nm = 406.5 l, unless stated otherwise. A graphene sheet of $(200 \text{ nm})^2$ consists of approximately 1.5 million carbon atoms, while the number of charge carriers, derived from the n_3 -type realizations of $n_{\pm}(\mathbf{r})$, typically ranges between 100 and 10 000 for layer L1—depending on the employed disorder strengths. We confirmed the convergence of the correlation measure ξ by increasing the resolution—here a modest grid size of 105^2 grid points proved enough for converging ξ sufficiently.

The main contribution to the quantum-corrected density $n_3(\mathbf{r})$ stems from the effective potential V in the vicinity of \mathbf{r} , see Table I, but the spatial integral for n_3 is *a priori* indefinite. We restrict this spatial integral to the square-shaped sheet of area L^2 . Hence, the bulk of the sheet is modeled adequately, but the region close to the edges requires further consideration. Two scenarios suggest themselves. First, the Coulomb tails of the charged impurities could be taken into account beyond the here employed sheet area L^2 (which would increase the computational cost substantially, with little effect on ξ). For this case of a truly finite sheet also the linear (quadratic) dispersion for graphene (bilayer graphene) would require a modification due to the nonperiodic boundary conditions. Second, we could view the sheet as a representative part of a larger sample whose disorder potential is unknown beyond the employed sheet area [99]. That is, both scenarios render the n_3 densities less precise near the edges (and particularly near the corners) of the sheet when omitting the spatial regions beyond the employed area L^2 (which amounts to setting $V(z) = \mu$ for $z \notin L^2$). We deem this inaccuracy acceptable since the majority of the sheet is adequately taken into account, and the effect on ξ is likely irrelevant given the observed variation due to different disorder realizations.

APPENDIX B: EXCHANGE

Exchange potentials for graphene are available, for example in Ref. [46],

$$V_{\rm X}(\mathbf{r}) \approx W \sqrt{\pi |n(\mathbf{r})|} \operatorname{sgn}(n(\mathbf{r})) \\ \times \left[\frac{1}{4} \log \left(\frac{4\sqrt{3}}{\sqrt{\pi |n(\mathbf{r})|}} \right) - 0.5757 \right]$$
(B1)

TABLE II. The correlation measure ξ , given in Eq. (D1), as a function of the ratio *R* between strain and disorder strength [cf. Eq. (C2)] for one disorder realization of double monolayer graphene in TF approximation, with and without exchange (X). In absolute numbers on the relevant scale of [-1, 1] for the correlation measures and regarding the spread of ξ due to different disorder realizations, cf. Fig. 5, we find exchange effects to be insignificant for investigating correlations between the two layers L1 and L2. We made the same observation for different disorder realizations.

R	0.1	1	10	20	50
$\xi^{\text{TF+X}}$	0.492	0.347	-0.013	-0.051	-0.122
ξIF	0.480	0.341	-0.015	-0.058	-0.135



FIG. 9. Quantum corrections to the TF approximation are visibly significant also at the level of effective potentials. Here, we depict the external potential V_{ext} for a graphene sheet of size L^2 , corresponding to one realization of charged impurity disorder (top left), the self-consistently converged effective potential V in TF approximation (denoted V_{TF} here) with the Hartree term only (top right), the differences between V_{TF} and $V_{\text{TF}(+X)}$, where the Hartree term plus the exchange of Eq. (B1) constitutes the interaction potential (bottom left), and the comparison of the quantum-corrected effective potential (including exchange) with V_{TF} (bottom right). The color code presents energy in units of u. The bottom panels are scaled by a factor of 5.

in the units used here. Figure 9 illustrates the exchange effects on the converged effective potential V, and Table II demonstrates that exchange is negligible as far as interlayer correlations are concerned.

An approximate exchange potential for other materials, viz. other dispersion relations, can for instance be obtained by calculating the according Dirac exchange energy for 2D systems and adding correlation in the spirit of Ref. [100]. It would also be interesting to see if exchange-correlation functionals developed for the 2D electron gas like in Ref. [23] can be used for systems with quadratic dispersion differing in the particle mass, e.g., for effective bilayer graphene. This topic is, however, beyond the scope of this work.

APPENDIX C: DISORDER POTENTIALS

We separate the sheet (with optional strain) of the first layer L1 from the plane that holds the charged impurities by d = 1 nm, cf. Fig. 3. The impurity-induced Coulomb potential in a sheet at distance d reads

$$V_{\rm C}(\mathbf{r}) = W \int (d\mathbf{r}') \frac{C(\mathbf{r}')}{\sqrt{|\mathbf{r} - \mathbf{r}'|^2 + d^2}},$$
 (C1)

where *C* takes on values $c_i \in \{-1, 0, 1\}$ randomly on the grid points $i \in \{1, ..., \Lambda\}$. We employ an average density of charged impurities n_{imp} (e.g., 10^{12} cm⁻², corresponding to 400 impurities on L^2), with half of the impurities positively



FIG. 10. Comparison between the correlation measures ξ and ξ_c for the TF densities, including exchange. As expected, ξ_c provides more contrast than ξ , but its increased variance renders it less useful for our purposes.

charged, such that the net charge in the impurity plane is zero. We model the strain as

$$V_{\rm S}(\mathbf{r}) = R W \int (d\mathbf{r}') \frac{\tilde{C}(\mathbf{r}')}{\sqrt{|\mathbf{r} - \mathbf{r}'|^2 + d^2}},$$
 (C2)

with *R* controlling the relative impact of strain and charged impurities, while \tilde{C} and *C* represent different disorder realizations of the same type. The sum of Eqs. (C1) and (C2) is the external potential V_{ext} , cf. Eq. (8), for a monolayer setup as well as for the first sheet of the double-layer structure. The second sheet of the double-layer structure is not affected by V_{S} , but only by the charged impurities at a distance of $\tilde{d} = 6$ nm, cf. Fig. 3, and by the electrostatic potential from the converged carrier distribution n_1 on the first sheet:

$$V_{\text{ext}}(\text{L2})(\mathbf{r}) = V_{\text{C}}(\mathbf{r})|_{d=\tilde{d}} + \int (d\mathbf{r}') \frac{W n_1(\mathbf{r} + \mathbf{r}')}{\sqrt{|\mathbf{r}'|^2 + (\tilde{d} - d)^2}}.$$
 (C3)

APPENDIX D: CORRELATION MEASURES

In Sec. V we use a normalized version of the interlayer correlation measure [2]

$$\xi[n_1, n_2] = \frac{\langle (n_1 - \langle n_1 \rangle)(n_2 - \langle n_2 \rangle) \rangle}{n_{1,\text{rms}} n_{2,\text{rms}}} \tag{D1}$$

to quantify the degree of (anti)correlation between the density distributions n_1 and n_2 . Here, $\langle n_j \rangle = \frac{1}{\Lambda} \sum_{i=1}^{\Lambda} n_j(i)$ is the mean of the discretized density of layer *j* on the grid points $i \in \{1, ..., \Lambda\}$, and the root mean square of the density fluctuations in layer *j* is $n_{j,\text{rms}} = \sqrt{\langle (n_j - \langle n_j \rangle)^2 \rangle}$. To check for spurious artifacts of the correlation measure ξ in Eq. (D1) and provide an error estimate for the (anti)correlations, we also consider the alternative measure

$$\xi_c[n_1, n_2] = \frac{\langle (n_1 - \langle n_1 \rangle_c)(n_2 - \langle n_2 \rangle_c) \rangle_c}{n_{1, \text{rms}, c} n_{2, \text{rms}, c}}, \qquad (D2)$$

which yields stronger contrast between situations of correlation and anticorrelation—at the expense of larger variance; cf. Fig. 10. Here we define the average

$$\langle A \rangle_c = \frac{\sum_{i=1}^{\Lambda} \chi_c(i) A(i)}{\sum_{i=1}^{\Lambda} \chi_c(i)}$$
(D3)



FIG. 11. Schematic of the workflow for implementing the selfconsistent loop of Eqs. (3)-(5). Steps (I) and (II) represent the initialization. The self-consistent loop is carried out in steps (III)-(VI).

of a quantity A and the corresponding root mean square $n_{j,\mathrm{rms},c} = \sqrt{\langle (n_j - \langle n_j \rangle_c)^2 \rangle_c}$. The characteristic function $\chi_c(i)$ at grid point *i* equals one if $n_i(i) \notin I_i$ (for both j = 1, 2) and zero otherwise [101]—with the cutoff interval $I_i =$ $[\min_i \{n_i(i)\} + c \Delta_i, \max_i \{n_i(i)\} - c \Delta_i]$ and the density spread $\Delta_i = \max_i \{n_i(i)\} - \min_i \{n_i(i)\}$. We choose c = 0.4to dismiss the fluctuations of correlation at small local carrier density. Owing to its increased variance, we refrain from depicting ξ_c in Sec. V.

APPENDIX E: SELF-CONSISTENT GROUND-STATE VARIABLES

In this Appendix we outline the numerical procedures for obtaining the ground-state variables of DPFT, viz. stationary solutions, n, V, and μ of Eqs. (3)–(5), adapted to the quasiparticle system described in Sec. III. Figure 11 illustrates the main steps for implementing the self-consistent loop, with a detailed description below.

Initialization (iteration i = 0): (I) Calculate $n_{\pm}^{(0)} = n_{\pm}^{(i=0)} = n_{\nu}[V_{\pm}^{(0)} - \mu^{(0)}]$ using $V_{-}^{(0)} = V_{\text{ext}}$ from Eq. (8) and $V_{\pm}^{(0)}$ from Eq. (9). The initial choice of the chemical potential is arbitrary, e.g., $\mu^{(0)} = 0$. We omit r dependences in this Appendix for the sake of notational simplicity. $n_{\nu}[V - \mu](\mathbf{r})$ are functionals of $V - \mu$ with parametric dependence on r, for example, the density expressions given in Table I.

(II) Adapt $\mu^{(0)}$ until $N = \int (d\mathbf{r}) n^{(0)} = \int (d\mathbf{r}) (n_{-}^{(0)} - n_{+}^{(0)}).$ For instance, we demand N = 0 to be reached within an absolute accuracy of 10^{-4} .

Self-consistent loop:

(III) Update effective potential

$$V_{-}^{(i+1)} = V_{\text{ext}} + \left. \frac{\delta E_{\text{int}}[n]}{\delta n} \right|_{n_{-}=n_{-}^{(i)}}$$

and determine effective potential $V_{+}^{(i+1)}$ via Eq. (9).

(IV) Update densities $\tilde{n}_{\pm}^{(i+1)} := n_{\nu} [V_{\pm}^{(i+1)} - \mu^{(i+1)}],$ starting with $\mu^{(i+1)} = \mu^{(i)}$, and adjusting $\mu^{(i+1)}$ until $N = \int (dr) \, \tilde{n}_{\pm}^{(i+1)} = \int (dr) \, (\tilde{n}_{\pm}^{(i+1)} - \tilde{n}_{\pm}^{(i+1)}).$

(V) Mix old and new densities with mixing parameter $\theta \in (0, 1)$: $n^{(i+1)} = (1 - \theta) n^{(i)} + \theta \tilde{n}^{(i+1)}$. More sophisticated density mixing like Pulay or Broyden mixing can be expected to improve the convergence behavior.

(VI) Check convergence with an appropriate norm criterion $||n^{(i+1)} - n^{(i)}|| < \epsilon$. We use $\epsilon = 10^{-6}$ and

$$||n^{(i+1)} - n^{(i)}|| = \frac{\sum_{r_j} \chi_{ij} |n^{(i+1)}(r_j) - n^{(i)}(r_j)|}{s^{(i+1)} \sum_{r_j} \chi_{ij}}, \quad (E1)$$

with grid positions r_i ,

$$\chi_{ij} = \eta \left(t^{(i+1)}(\boldsymbol{r}_j) - \frac{s^{(i+1)}}{1000} \right),$$
(E2)

$$t^{(i+1)}(\mathbf{r}_j) = \min\{|n^{(i+1)}(\mathbf{r}_j)|, |n^{(i)}(\mathbf{r}_j)|\},$$
 (E3)

$$s^{(i+1)} = \frac{1}{2} \big(\mathrm{MAX}_i - \mathrm{MIN}_i \big), \tag{E4}$$

$$MIN_{i} = \min_{\boldsymbol{r}} \{ \min\{|n^{(i+1)}(\boldsymbol{r})|, |n^{(i)}(\boldsymbol{r})|\} \}.$$
 (E6)

If densities have converged to within ϵ , the stationary solutions n, V, and μ of Eqs. (3)–(5) are found; otherwise return to step (III) for the next iteration.

The number of self-consistent iterations *i* required for convergence is largely determined by the density-mixing parameter θ , which is highly system dependent. Here, we can choose $\theta = 0.5$ for weak disorder but need to approach the equilibrium very gently in case of strong disorder ($\theta \leq 0.01$). In our implementation *i* usually reaches values on the order of $\sim 10/\theta$.

APPENDIX F: THE QUANTUM-CORRECTED DENSITY n₃ FOR LINEAR DISPERSION

In this Appendix, we outline the derivation of the density expressions for linear dispersion given in Table I. Equations (3) and (6) yield Eq. (7) with the aid of the Fourier transform of the step function $\eta(\cdot)$:

$$n(\mathbf{r}) = g\langle \mathbf{r} | \eta(\mu - H_1) | \mathbf{r} \rangle = g \int_{\neg \lor \lor} \frac{dt}{2\pi i t} e^{\frac{it}{\hbar} \mu} \langle \mathbf{r} | e^{-\frac{it}{\hbar} H_1} | \mathbf{r} \rangle,$$
(F1)

where the degeneracy factor g is part of the trace in Eq. (6). Generally, a Suzuki-Trotter factorization U_{ν} of the timeevolution operator with

$$\langle \boldsymbol{r} | U_{\nu} | \boldsymbol{r} \rangle = \langle \boldsymbol{r} | \prod_{i=1}^{\nu/2} e^{\alpha_i T(\boldsymbol{P})} e^{\beta_i V(\boldsymbol{R})} | \boldsymbol{r} \rangle \tag{F2}$$

$$=:\prod_{i=1}^{\nu/2}\int (d\boldsymbol{r}_i)(d\boldsymbol{p}_i)\,c_i\,e^{-\frac{it}{\hbar}A}\tag{F3}$$

and real-valued functions c_i and A fixed by the here inserted completeness relations in position and momentum space, implies

$$\langle \boldsymbol{r} | \eta(\mu - H_1) | \boldsymbol{r} \rangle = \prod_{i=1}^{\nu/2} \int (d\boldsymbol{r}_i) (d\boldsymbol{p}_i) c_i \eta(\mu - A).$$
 (F4)

For example, the case U_3 in Sec. II is formally obtained by setting $\alpha_1 = \alpha_2 = -\frac{it}{2\hbar}$, $\beta_1 = -\frac{it}{\hbar}$, and $\beta_2 = 0$ in U_4 . Note that U_3 is denoted as $U_{3'}$ in Ref. [54].

- [1] A. K. Geim and I. V. Grigorieva, Nature (London) 499, 419 (2013).
- [2] M. Rodriguez-Vega, J. Fischer, S. Das Sarma, and E. Rossi, Phys. Rev. B 90, 035406 (2014).
- [3] W. Kohn and L. J. Sham, Phys. Rev. 140, A1133 (1965).
- [4] R. M. Dreizler and E. K. U. Gross, *Density Functional Theory* (Springer, Berlin, Heidelberg, 1990).
- [5] S. Goedecker, Rev. Mod. Phys. 71, 1085 (1999).
- [6] J. M. Soler, E. Artacho, J. D. Gale, A. García, J. Junquera, P. Ordejón, and D. Sánchez-Portal, J. Phys. Condens. Matter 14, 2745 (2002).
- [7] D. Bowler and T. Miyazaki, Rep. Prog. Phys. 75, 036503 (2012).
- [8] S. J. Fox, J. Dziedzic, T. Fox, C. S. Tautermann, and C.-K. Skylaris, Proteins: Struct., Funct., Bioinf. 82, 3335 (2014).
- [9] J. Aarons, M. Sarwar, D. Thompsett, and C.-K. Skylaris, J. Chem. Phys. 145, 220901 (2016).
- [10] D. J. Cole and N. D. M. Hine, J. Phys.: Condens. Matter 28, 393001 (2016).
- [11] L. Chiodo, L. A. Constantin, E. Fabiano, and F. Della Sala, Phys. Rev. Lett. 108, 126402 (2012).
- [12] A. D. Kaplan, K. Wagle, and J. P. Perdew, Phys. Rev. B 98, 085147 (2018).
- [13] A. Holas, P. M. Kozlowski, and N. H. March, J. Phys. A: Math. Gen. 24, 4249 (1991).
- [14] M. Brack and R. K. Bhaduri, *Semiclassical Physics*, Frontiers in Physics, Vol. 96 (Addison-Wesley, Reading, MA, 2003).
- [15] M.-I. Trappe, Y. L. Len, H. K. Ng, C. A. Müller, and B.-G. Englert, Phys. Rev. A 93, 042510 (2016).
- [16] L. Pollack and J. P. Perdew, J. Phys: Cond. Mat. 12, 1239 (2000).
- [17] S. M. Reimann and M. Manninen, Rev. Mod. Phys. 74, 1283 (2002).
- [18] S. Pittalis, E. Räsänen, N. Helbig, and E. K. U. Gross, Phys. Rev. B 76, 235314 (2007).

We obtain the TF density by choosing v = 2 and $\alpha_1 = \beta_1 = -\frac{it}{\hbar}$. Specifying linear dispersion, $T(\mathbf{p}) = v_F |\mathbf{p}|$, we find

$$n_{\rm TF}(\mathbf{r}) = 2\pi g \int_0^\infty \frac{dp \, p}{(2\pi \hbar)^2} \eta(\mu - v_{\rm F} \, p - V(\mathbf{r})), \qquad ({\rm F5})$$

whose analytic result is easily obtained and given in Table I. The computation of n_3 requires the insertion of three completeness relations, and the so-introduced two-dimensional momentum integrations are reduced to one-dimensional integrals via appropriate coordinate transformations. The subsequent analytic evaluations of the remaining one-dimensional integrals lead to the expression for n_3 in Table I, leaving one spatial integral to be evaluated numerically.

- [19] S. Pittalis, E. Räsänen, and M. A. L. Marques, Phys. Rev. B 78, 195322 (2008).
- [20] L. A. Constantin, J. P. Perdew, and J. M. Pitarke, Phys. Rev. Lett. 101, 016406 (2008); 101, 269902(E) (2008).
- [21] S. Pittalis, E. Räsänen, and E. K. U. Gross, Phys. Rev. A 80, 032515 (2009).
- [22] S. Pittalis, E. Räsänen, C. R. Proetto, and E. K. U. Gross, Phys. Rev. B 79, 085316 (2009).
- [23] S. Pittalis and E. Räsänen, Phys. Rev. B 82, 165123 (2010).
- [24] E. Räsänen, S. Pittalis, J. G. Vilhena, and M. A. L. Marques, Int. J. Quant. Chem. **110**, 2308 (2010).
- [25] J. G. Vilhena, E. Räsänen, M. A. L. Marques, and S. Pittalis, J. Chem. Theory Comput. 10, 1837 (2014).
- [26] L. H. Thomas, Math. Proc. Cambridge Philos. Soc. 23, 542 (1927).
- [27] E. Fermi, Rend. Lincei 6, 602 (1927).
- [28] A. Cangi, D. Lee, P. Elliott, K. Burke, and E. K. U. Gross, Phys. Rev. Lett. 106, 236404 (2011).
- [29] K. Burke, J. Chem. Phys. 136, 150901 (2012).
- [30] V. V. Karasiev, D. Chakraborty, O. A. Shukruto, and S. B. Trickey, Phys. Rev. B 88, 161108(R) (2013).
- [31] A. Pribram-Jones, D. A. Gross, and K. Burke, Annu. Rev. Phys. Chem. 66, 283 (2015).
- [32] W. C. Witt, B. G. del Rio, J. M. Dieterich, and E. A. Carter, J. Mater. Res. 33, 777 (2018).
- [33] L.-W. Wang and M. P. Teter, Phys. Rev. B 45, 13196 (1992).
- [34] F. Della Sala, E. Fabiano, and L. A. Constantin, Phys. Rev. B 91, 035126 (2015).
- [35] J. Xia and E. A. Carter, Phys. Rev. B 91, 045124 (2015).
- [36] V. V. Karasiev, T. Sjostrom, and S. B. Trickey, Phys. Rev. B 86, 115101 (2012).
- [37] M. Chen, J. Xia, C. Huang, J. M. Dieterich, L. Hung, I. Shin, and E. A. Carter, Comput. Phys. Commun. 190, 228 (2015).
- [38] S. Das, M. Iyer, and V. Gavini, Phys. Rev. B **92**, 014104 (2015).

- [39] M. Chen, X.-W. Jiang, H. Zhuang, L.-W. Wang, and E. A. Carter, J. Chem. Theory Comput. 12, 2950 (2016).
- [40] W. Mi, X. Shao, C. Su, Y. Zhou, S. Zhang, Q. Li, H. Wang, L. Zhang, M. Miao, Y. Wang, and Y. Ma, Comput. Phys. Commun. 200, 87 (2016).
- [41] L. A. Constantin, E. Fabiano, and F. Della Sala, J. Phys. Chem. Lett. 9, 4385 (2018).
- [42] L. E. González and D. J. González, Phys. Rev. B 77, 064202 (2008).
- [43] J. Xia, C. Huang, I. Shin, and E. A. Carter, J. Chem. Phys. 136, 084102 (2012).
- [44] A. Borgoo, J. A. Green, and D. J. Tozer, J. Chem. Theory Comput. 10, 5338 (2014).
- [45] L. A. Espinosa Leal, A. Karpenko, M. A. Caro, and O. Lopez-Acevedo, Phys. Chem. Chem. Phys. 17, 31463 (2015).
- [46] E. Rossi and S. Das Sarma, Phys. Rev. Lett. 101, 166803 (2008).
- [47] E. H. Lieb and B. Siman, Adv. Math. 23, 22 (1977).
- [48] L. Salasnich, J. Phys. A: Math. Theor. 40, 9987 (2007).
- [49] M.-I. Trappe, Y. L. Len, H. K. Ng, and B.-G. Englert, Ann. Phys. 385, 136 (2017).
- [50] B. P. van Zyl, E. Zaremba, and P. Pisarski, Phys. Rev. A 87, 043614 (2013).
- [51] B. P. van Zyl, A. Farrell, E. Zaremba, J. Towers, P. Pisarski, and D. A. W. Hutchinson, Phys. Rev. A 89, 022503 (2014).
- [52] R. F. Ribeiro, D. Lee, A. Cangi, P. Elliott, and K. Burke, Phys. Rev. Lett. **114**, 050401 (2015).
- [53] M.-I. Trappe, D. Delande, and C. A. Müller, J. Phys. A: Math. Theor. 48, 245102 (2015).
- [54] T. T. Chau, J. H. Hue, M.-I. Trappe, and B.-G. Englert, New J. Phys. 20, 073003 (2018).
- [55] B.-G. Englert, Lecture Notes in Physics: Semiclassical Theory of Atoms (Springer, Berlin, Heidelberg, 1988).
- [56] B.-G. Englert, Phys. Rev. A 45, 127 (1992).
- [57] M. Cinal and B.-G. Englert, Phys. Rev. A 48, 1893 (1993).
- [58] P. Hohenberg and W. Kohn, Phys. Rev. 136, B864 (1964).
- [59] B. N. Narozhny and A. Levchenko, Rev. Mod. Phys. 88, 025003 (2016).
- [60] R. V. Gorbachev, A. K. Geim, M. I. Katsnelson, K. S. Novoselov, T. Tudorovskiy, I. V. Grigorieva, A. H. MacDonald, S. V. Morozov, K. Watanabe, T. Taniguchi, and L. A. Ponomarenko, Nat. Phys. 8, 896 (2012).
- [61] J. C. W. Song and L. S. Levitov, Phys. Rev. Lett. 109, 236602 (2012).
- [62] D. Y. H. Ho, I. Yudhistira, B. Y.-K. Hu, and S. Adam, Commun. Phys. 1, 41 (2018).
- [63] B.-G. Englert and J. Schwinger, Phys. Rev. A 29, 2339 (1984).
- [64] The contour integration circumvents the singularity at t = 0 in the lower half plane.
- [65] S. Golden, Phys. Rev. 107, 1283 (1957).
- [66] S. Golden, Rev. Mod. Phys. 32, 322 (1960).
- [67] J. C. Light and J. M. Yuan, J. Chem. Phys. 58, 660 (1973).
- [68] S. Y. Lee and J. C. Light, J. Chem. Phys. 63, 5274 (1975).
- [69] N. Hatano and M. Suzuki, Lect. Notes Phys. 679, 37 (2005).
- [70] Hence, the scope of our investigation also extends to energies further away from the Dirac point (band gap) as long as $T(\mathbf{p})$ can be considered linear (quadratic).
- [71] Equation (11) can be regarded as a simplified version of the smooth models for the Coulomb potential discussed in Ref. [102].

- PHYSICAL REVIEW B 99, 235415 (2019)
- [72] P. A. M. Dirac, Proc. Cambridge Philos. Soc. 26, 376 (1930).
- [73] J. Martin, N. Akerman, G. Ulbricht, T. Lohmann, J. H. Smet, K. von Klitzing, and A. Yacoby, Nat. Phys. 4, 144 (2008).
- [74] J. Xue, J. Sanchez-Yamagishi, D. Bulmash, P. Jacquod, A. Deshpande, K. Watanabe, T. Taniguchi, P. Jarillo-Herrero, and B. J. LeRoy, Nat. Mater. 10, 282 (2011).
- [75] Y.-W. Tan, Y. Zhang, K. Bolotin, Y. Zhao, S. Adam, E. H. Hwang, S. Das Sarma, H. L. Stormer, and P. Kim, Phys. Rev. Lett. 99, 246803 (2007).
- [76] J.-H. Chen, C. Jang, S. Adam, M. S. Fuhrer, E. D. Williams, and M. Ishigami, Nat. Phys. 4, 377 (2008).
- [77] S. Samaddar, I. Yudhistira, S. Adam, H. Courtois, and C. B. Winkelmann, Phys. Rev. Lett. 116, 126804 (2016).
- [78] Strictly speaking, the electrostatic effects of L1 and L2 on each other have to be treated self-consistently, but the layer separation of 5 nm ensures that the backreaction of L2 on L1 is of the order of 10^{-6} of the external potential of L1 and therefore negligible at the level of precision we are aiming for in this work.
- [79] In the (unphysical) limit of vanishing layer separation these tendencies become certainties, but a finite layer separation in concert with the disordered potential landscape allows for local variations of correlation, thereby affecting global correlation measures quantitatively.
- [80] During the self-consistent loop the chemical potential μ is adjusted such that the mean carrier densities $\langle n_j \rangle$ are kept at a fixed value (zero, unless stated otherwise).
- [81] At this point we do not calculate disorder averages but rather showcase the predictions of a few disorder realizations (also bearing in mind the limited number of samples usually investigated in actual experiments).
- [82] M. Gibertini, A. Tomadin, F. Guinea, M. I. Katsnelson, and M. Polini, Phys. Rev. B 85, 201405(R) (2012).
- [83] C. R. Dean, A. F. Young, I. Meric, C. Lee, L. Wang, S. Sorgenfrei, K. Watanabe, T. Taniguchi, P. Kim, K. L. Shepard, and J. Hone, Nat. Nanotechnol. 5, 722 (2010).
- [84] S. Adam, E. H. Hwang, V. M. Galitski, and S. D. Sarma, Proc. Natl. Acad. Sci. 104, 18392 (2007).
- [85] J. I. A. Li, T. Taniguchi, K. Watanabe, J. Hone, A. Levchenko, and C. R. Dean, Phys. Rev. Lett. **117**, 046802 (2016).
- [86] K. Lee, J. Xue, D. C. Dillen, K. Watanabe, T. Taniguchi, and E. Tutuc, Phys. Rev. Lett. **117**, 046803 (2016).
- [87] P. Simonet, S. Hennel, H. Overweg, R. Steinacher, M. Eich, R. Pisoni, Y. Lee, P. Märki, T. Ihn, K. Ensslin, M. Beck, and J. Faist, New J. Phys. **19**, 103042 (2017).
- [88] M. Schütt, P. M. Ostrovsky, M. Titov, I. V. Gornyi, B. N. Narozhny, and A. D. Mirlin, Phys. Rev. Lett. **110**, 026601 (2013).
- [89] M. Zarenia, A. R. Hamilton, F. M. Peeters, and D. Neilson, Phys. Rev. Lett. **121**, 036601 (2018).
- [90] J. I. A. Li, T. Taniguchi, K. Watanabe, J. Hone, and C. R. Dean, Nat. Phys. 13, 751 (2017).
- [91] X. Liu, K. Watanabe, T. Taniguchi, B. I. Halperin, and P. Kim, Nat. Phys. 13, 746 (2017).
- [92] G. W. Burg, N. Prasad, K. Kim, T. Taniguchi, K. Watanabe, A. H. MacDonald, L. F. Register, and E. Tutuc, Phys. Rev. Lett. 120, 177702 (2018).
- [93] P. López Ríos, A. Perali, R. J. Needs, and D. Neilson, Phys. Rev. Lett. **120**, 177701 (2018).

- [94] L. Britnell, R. M. Ribeiro, A. Eckmann, R. Jalil, B. D. Belle, A. Mishchenko, Y.-J. Kim, R. V. Gorbachev, T. Georgiou, S. V. Morozov, A. N. Grigorenko, A. K. Geim, C. Casiraghi, A. H. C. Neto, and K. S. Novoselov, Science 340, 1311 (2013).
- [95] W. J. Yu, Y. Liu, H. Zhou, A. Yin, Z. Li, Y. Huang, and X. Duan, Nat. Nanotechnol. 8, 952 (2013).
- [96] F. Withers, O. Del Pozo-Zamudio, A. Mishchenko, A. P. Rooney, A. Gholinia, K. Watanabe, T. Taniguchi, S. J. Haigh, A. K. Geim, A. I. Tartakovskii, and K. S. Novoselov, Nat. Mater. 14, 301 (2015).
- [97] H. Raza, Graphene Nanoelectronics: Metrology, Synthesis, Properties and Applications (Springer, Heidelberg, New York, 2012).

- [98] R. Geick, C. H. Perry, and G. Rupprecht, Phys. Rev. 146, 543 (1966).
- [99] An extension of the position space support by employing appropriate reflections of the disordered sheet at its edges would counteract this inaccuracy but likely introduce spurious effects due to the artificial periodicity, along with a substantial increase in computational cost.
- [100] S. H. Vosko, L. Wilk, and M. Nusai, Can. J. Phys. 58, 1200 (1980).
- [101] All grid points are taken into account, i.e., $\chi_c(i) = 1$ for all *i*, if c = 1/2. Then, Eqs. (D1) and (D2) coincide. For c = 0 all data are dismissed.
- [102] C. E. González-Espinoza, P. W. Ayers, J. Karwowski, and A. Savin, Theor. Chem. Acc. 135, 256 (2016).