Reactivity of lithium and platinum at elevated densities

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Through a series of x-ray diffraction experiments, combined with first-principles calculations, we show that the nobility of platinum can be overcome, reacting with lithium under compression at room temperature. Pressures as low as 2.3 GPa lead to the synthesis of $Li_{11}Pt_2$, exhibiting the highest lithium content of any known intermetallic compound. Above 16 GPa, $Li_{11}Pt_2$ expels Li and transforms to Li_2Pt . This change is accompanied by a transformation in bonding characterized by the formation of one-dimensional Pt-Pt bonds with increasingly covalent character at higher densities.

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Being the simplest and lightest metallic element, dense lithium (Li) is of fundamental importance due to the strong influence of quantum effects on its behavior [1-8]. As other alkali metals, Li shows a pronounced minimum in the melting temperature with pressure, as low as 190 K, leading to counterintuitive cold melting above 40 GPa at room temperature [7,9,10]. When compressed at low temperatures, Li shows remarkable structural complexity, with at least five high-pressure configurations including low-symmetry semiconducting phases containing up to 88 atoms in the unit cell [6,7,11-13]. Platinum (Pt) is a highly unreactive transition metal with widespread industrial uses, such as in catalytic converters. Due to its stability and resistance to corrosion, it is widely used as a material for analytical standards such as pressure calibration, the standard hydrogen electrode in electrochemistry [14], and the now defunct International Prototype Kilogram and Meter which defined the base units of the metric system. Organometallic platinum compounds such as cisplatin and carboplatin play a role in cancer chemotherapy and owe their discovery to the use of an "inert" Pt electrode in a bacterial growth chamber [15,16].

Compounds of the alkali and noble metals have been comprehensively studied since the early 20th century [17–19]. In these compounds, electron density is transferred from the electropositive alkali metal to the electronegative noble metal. Increasing differences in electronegativity lead to increasingly ionic behavior, reaching a logical conclusion in cesium auride (CsAu) which is a yellowish transparent semiconductor with a band gap of 2.6 eV [20]. After gold, platinum (Pt) has the highest electron affinity of any metal [21]. Compounds of platinum with group 1 or 2 metals

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compounds contain negatively charged Pt substructures such as Pt^{-2} anions, dimers ([Pt-Pt]⁶⁻), and chains such as those found in Li₂Pt [22–25]. Four Li-Pt compounds are known at ambient pressures: LiPt₇, LiPt₂, LiPt, Li₂Pt all synthesized at high temperatures [25–27]. Unlike the Au-Li and Ag-Li systems, there are no known higher Li-content intermetallic compounds than the dilithide, Li₂Pt [19,28]. High-pressure reaction conditions have been an effective route to the formation of new compounds of the alkali metals [29,30] and could promote the formation of new high Li-content alloys for a range of applications such as energy storage [31,32].

exhibit significant charge transfer and can be considered as "platinides." While CsAu has a simple cubic structure, these

Here, we show that these two archetypal metals, lithium and platinum, become highly reactive with the application of moderate pressures (<3 GPa), resulting in the formation of an intermetallic compound Li₁₁Pt₂. This compound exhibits the highest Li content for any known intermetallic alloy which, combined with its low-pressure room-temperature synthesis conditions, suggests that these compounds may be overlooked as potential storage materials. With increasing pressure, Li₁₁Pt₂ transforms to the known compound Li₂Pt, driven by the formation of negatively charged one-dimensional Pt-Pt chains as incompressible as pure Pt. The arrangement of these chains has been suggested to give rise to Dirac cones [33]. However, our electronic structure calculations show that these cones are "gapped out" at ambient conditions and increasing hydrostatic pressure exacerbates this breakdown of semimetallic behavior.

All experiments were carried out with excess Li concentrations (estimated 1%–2% Pt by volume). A detailed description of the sample preparation, experimental data collection, and theoretical procedures are given in the Supplemental Material (SM) [34]. Under these conditions pressures as low as 2.3 GPa lead to a complete reaction of Pt. X-ray diffraction data collected at this pressure [Fig. 1(a)] indicate the formation of a compound crystallizing in cubic space group $I\bar{4}3m$, a = 9.2056(1) Å. Structure solution by charge flipping [35]

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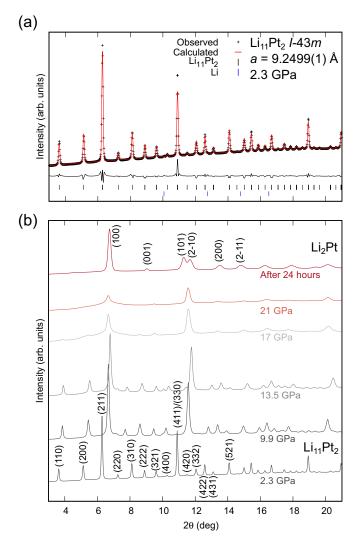


FIG. 1. High-pressure x-ray diffraction data on $Li_{11}Pt_2$. (a) Rietveld profile refinement ($\lambda = 0.4141$ Å) for $Li_{11}Pt_2$ ($wR_p = 3.84\%$, $R_{expt} = 3.66\%$): Experimental data shown with crosses, calculated profile shown in red, and the difference is shown as the lower black trace (raw x-ray diffraction image is shown in Fig. S1). (b) Diffraction patterns under increasing quasihydrostatic pressure. Peaks due to $Li_{11}Pt_2$ (bottom, black) broaden and rapidly lose intensity.

indicates eight Pt atoms per unit cell on the 8*c* site. Three possible Li-atom sites were identified in Fourier difference maps giving a total of 44 Li atoms in the unit cell. Density functional theory (DFT) calculations confirm the stoichiometry of this phase, $Li_{11}Pt_2$ (experimental and optimized atomic positions are given in Tables S1 and S2). Geometry optimization performed at increasing pressures show excellent agreement with the experimentally determined unit-cell length [Fig. 2(a)].

Li₁₁Pt₂ adopts a γ -brass-type structure which has been observed in a number of transition-metal alloys (e.g., Zn₁₁Ir₂, Zn₁₁Pt₂, etc. [36,37]) but has not been observed with an alkali metal. Li₁₁Pt₂ also displays the highest Li content of any Licontaining intermetallic [38–41]. The unit cell of Li₁₁Pt₂ contains two 26-atom clusters arrayed on a body-centered lattice. Each cluster is constructed from four edge- and face-sharing

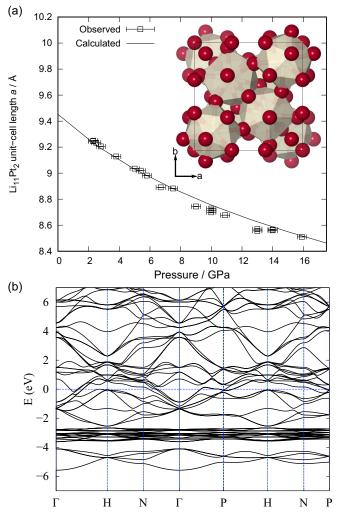


FIG. 2. Crystal and electronic structure of $Li_{11}Pt_2$. (a) Change in unit-cell lengths with pressure for $Li_{11}Pt_2$: The inset shows the $Li_{11}Pt_2$ crystal structure, Li atoms are red, and Pt atoms beige. (b) Band structure of $Li_{11}Pt_2$ at 10 GPa.

irregular PtLi₁₂ icosahedra [inset of Fig. 2(a)]. As a result of its extremely high Li content, $Li_{11}Pt_2$ is highly compressible and its unit-cell volume decreases by 20.9% (-163.7 Å³) between 2.8 and 15.9 GPa. Volume versus pressure data were fitted with a third-order Birch-Murnaghan equation of state giving a bulk modulus of 24(4) GPa, only twice that of pure Li [10.43(24) GPa] [42], suggesting the mechanical properties are dominated by Li-Li interactions.

The electronic band structure of $Li_{11}Pt_2$ is shown in Fig. 2(b) and can be split into three regions: -5.6 to -4.1 eV corresponds mainly to the Pt 6s band; -3.6 to -2.6 eV corresponds to the Pt 5d band; and above. This remaining region above -2.6 eV is contributed to by all the bands, although the density of states (DOS) at E_F is dominated by contributions from the Pt 5d and Li 2p orbitals. Interestingly, this structure shows similarities to the band structures of $Zn_{11}Ni_2$ and $Zn_{11}Pd_2$. The stabilities of these phases have been explained by Fermi-surface Brillouin-zone interactions involving {330} and {411} lattice planes and, similarly in $Li_{11}Pt_2$, the d states are well immersed below the Fermi level.

The metallic character of $Li_{11}Pt_2$ is also manifested by a flatness index of ≈ 0.4 for all studied pressures. See SM for a full Bader analysis of $Li_{11}Pt_2$.

Within its coordination polyhedra, each Pt atom is surrounded by valence electron localization function (ELF) attractors (pink spheres in Fig. S2). The integration of the electron density within their corresponding chemical basins plus that contained in the electronic core leads to the Pt charge values of Table S4. Most importantly, the valence ELF attractors with the highest ELF value (0.615 at 0 GPa) appear outside the Pt coordination polyhedra and are almost coincident with the non-nuclear maxima (NNM) of the electron density (blue spheres in Fig. S2). Moreover, the localization window connecting them is very low (0.03 at 0 GPa). This generates a three-dimensional network of channels (η_{mesh} in Table S4) with an almost constant ELF. These characteristics are reminiscent of prototypical metallic systems such as bcc Li.

At pressures above 11 GPa, significant broadening of the $Li_{11}Pt_2$ peaks occurs, coinciding with the growth of a broad peak of diffuse scattering centered on the (211) reflection and which increases in intensity with pressure [Fig. 1(b)]. By 21 GPa, the peaks corresponding to $Li_{11}Pt_2$ are replaced by those from a different high-pressure phase. From this pressure, samples were decompressed to approximately 10 GPa to avoid possible damage to the diamond anvils and left at room temperature for 24 h. The sample was then compressed or decompressed to discern the stability range of this compound. X-ray diffraction surveys indicated some regions to be composed entirely of Li₂Pt, showing well-defined peak shapes and no diffuse scattering. All peaks from this highpressure phase are indexed to a hexagonal cell a = 3.8573(5), c = 2.6034(6) Å (at 21 GPa), and subsequent Rietveld refinement confirms this phase to be Li₂Pt crystallizing in space group P6/mmm (CaHg₂ type), where Li atoms lie on the 2d sites $(\frac{1}{3}, \frac{2}{3}, \frac{1}{2})$ with Pt atoms lying on the 1*a* site (0, 0, 0) [Fig. 3(a)]. Li₂Pt is known to exist at ambient pressures and is formed by the reaction of LiH and Pt at temperatures of 870 K [25], however, here pressure is used to drive its formation at room temperature. The structure shows no variable positional parameters, therefore all structural parameters (i.e., bond distances, angles etc.) are fully determined by the values of the lattice constants. The structure of Li₂Pt is characterized by linear chains of Pt atoms along the c axis which pass through hexagonal, graphenelike sheets of Li atoms [Fig. 3(b)]. Both phases are stable to 3 GPa, the lowest pressure reached on decompression.

Li₂Pt exhibits highly anisotropic compressibility; between 3 and 22 GPa the *a* axis reduces by 0.3339(7) Å [-8.0%, $K_a = 167(14)$ GPa] and by contrast the *c* axis is relatively incompressible, reducing by only 0.0466(8) Å [-1.8%, $K_c =$ 319(14) GPa] over the same pressure range. This behavior can be easily understood by considering changes in Pt-Pt distances and their relative compressibilities, as measured by linear moduli—one-dimensional equivalents to bulk moduli—with pressures as shown in Fig. S3(a). Pt-Pt distances between chains in Li₂Pt are highly compressible, reducing by 8% up to 22 GPa, with a linear modulus of 47(9) GPa which is comparable to those in Li₁₁Pt₂ [72(12) GPa]. By contrast, Pt-Pt distances along *c* are shorter than those found in elemental

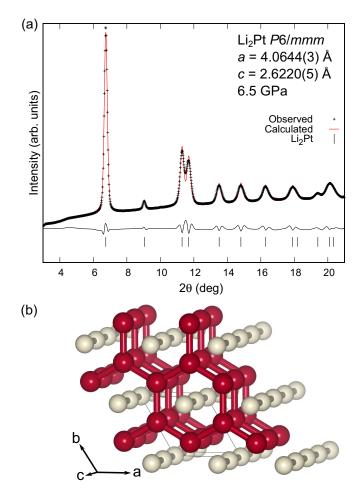


FIG. 3. High-pressure x-ray diffraction and crystal structure of Li₂Pt. (a) Rietveld profile refinement ($\lambda = 0.4141$ Å, raw x-ray diffraction image is shown in Fig. S1) for Li₂Pt ($wR_p = 5.77\%$, $R_{expt} = 3.35\%$). (b) 2 × 2 supercell of Li₂Pt: Chains of Pt atoms lie along the *c* axis, each coordinated by 12 Li atoms. Li atoms from graphenelike hexagonal sheets in the *ab* plane.

Pt and are consequentially highly incompressible, reducing by just 2% over the same pressure range, with a linear modulus of 816(141) GPa which is comparable to 795 GPa for pure Pt [43].

Li₂Pt has been proposed as a three-dimensional (3D) Dirac semimetal with a Dirac point at *K* in analogy with graphene due to the linear chains of Pt anions ordered hexagonally in 2D [33]. Figures 4(a) and 4(b) show the band structures at 0 and 80 GPa with spin-orbit coupling included. Although a Dirac point at *K* appears at 0 GPa when spin-orbit coupling is not considered, both pressure and spin-orbit coupling gap it out. The inclusion of spin-orbit coupling also leads to greater dispersion of the valence bands, particularly the lowest one along the reciprocal lines parallel to c^* (Γ -*A*, *H*-*K*, *L*-*M*), associated with a strong interaction of the Pt atoms along the *c* axis [Fig. 4(c)]. A partial ionic character is manifested by the flatness index as low as ≈ 0.13 , typical of ionic solids. See SI for a full Bader analysis of Li₂Pt.

The ELF topological analysis shows a hexagonal distribution of the ELF valence maxima around and between the

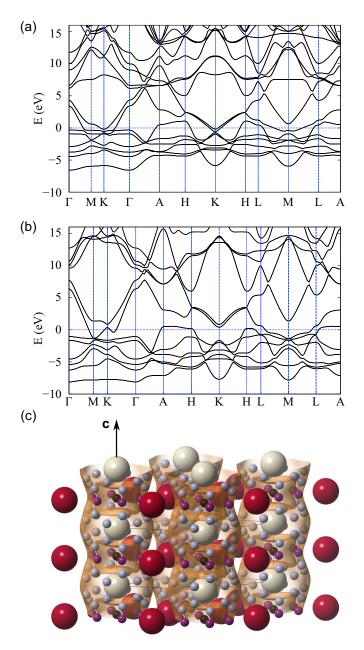


FIG. 4. Electronic structure of Li_2Pt at (a) 0 and (b) 80 GPa. Pt 5*s*, 5*p*, 6*s*, 5*d* and Li 1*s*, 2*s* electrons are treated as valence. Spin-orbit coupling is included. (c) ELF = 0.34 isosurface for Li_2Pt at 0 GPa. Valence ELF maxima are represented by blue balls. The valence ELF maxima located at 6*k* and 1*b* sites are indicated by purple and black balls, respectively.

Pt atoms [Fig. 4(c)]. This distribution generates channels of low ELF with a hexagonal prism shape along the Pt-Pt chains on the *c* axis. We analyze the ELF value and integrated electron density of the valence ELF maxima that appear at the middle of the shortest Pt-Pt distance on the *c* axis. Apart from the middle point (1*b* site, black spheres), we have also considered the hexagon of ELF maxima (6*k* sites, purple spheres) surrounding this 1*b* site and conjugated to the Li₆ hexagon coordinated to the Pt atoms at that *z* coordinate. With increasing pressure these maxima show increased ELF value and electronic charge up to 40 GPa. A further detailed discussion of the ELF analysis is given in SM.

Regarding the implications of these results, conductivity measurements using Pt as the electrodes have been used to demonstrate wide ranging phenomena of Li at high pressure. These observations include superconductivity, high-pressure melting, and the influence of isotopic effects on T_c [2,8,44,45]. This work has shown that Li-Pt compounds are readily formed in Diamond-anvil cells (DACs) at 300 K, and undergo structural transitions at modest pressures. It is highly likely that electronic measurements of Li are compromised at high pressure through sample contamination by the Pt (or indeed any noble metal) electrodes. Indeed, the possible contamination of Li at the contact point with Pt electrodes was postulated in early superconductivity measurements [2]. Simultaneous secondary diagnostics such as x-ray diffraction can be applied in order to definitively rule out pressure-induced chemical reactions [45], although measurements should be made sufficiently close to the electrode-sample interface to detect any reaction. Our observations may help explain the significant discrepancies that exist in the literature regarding the Li melting curve [7,44] and the reportedly large isotopic effect on T_c between ⁶Li and ⁷Li [8]. It should be noted that a subsequent theoretical study has been unable to reproduce the reported effect [46].

Our study has uncovered an alternative route to the synthesis of intermetallic compounds by compressing to moderate pressures at room temperature. At low pressures, we observe the formation of a compound Li₁₁Pt₂ which has the highest Li content for any known intermetallic compound. The low pressures required for synthesis suggest that this material could be used to store extensive quantities of Li for energy storage applications. Under further compression Li₁₁Pt₂ transforms to Li₂Pt, characterized by a chain of Pt atoms bonding in a quasi-one-dimensional manner resulting in highly anisotropic mechanical properties. Furthermore, our observation of strong pressure-driven reactivity between Li and Pt raises questions for a number of experiments measuring the electrical properties of Li at pressure using Pt electrodes [2,8,44,45]. In the absence of diagnostic x-ray diffraction confirming the purity of the Li samples studied, the formation of Li-Pt compounds in these experiments cannot be ruled out and it remains an open question what impact this will have on our current understanding of this fundamentally important system at high densities.

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