Fermi-level Dirac crossings in 4d and 5d cubic metal oxides: NaPd₃O₄ and NaPt₃O₄

Samuel M. L. Teicher,^{1,*} Leo K. Lamontagne,¹ Leslie M. Schoop,^{2,†} and Ram Seshadri¹

¹Materials Department and Materials Research Laboratory, University of California, Santa Barbara, California 93106, USA ²Department of Chemistry, Princeton University, Princeton, New Jersey 08540, USA

(Received 14 February 2019; revised manuscript received 4 May 2019; published 28 May 2019)

The cubic oxide metals NaM_3O_4 (M = Pd or Pt) crystallize in the nonsymmorphic $Pm\bar{3}n$ space group. First-principles calculations are employed here to understand the role of the MO_4 square planes and M-M interactions in the development of the electronic structure. The compounds host numerous Dirac crossings near the Fermi level which, in the absence of spin-orbit coupling, appear to form a cubic nodal state. Spin-orbit coupling fragments this nodal state into smaller regions with Dirac-like character, with the fragmenting being more pronounced in the M = Pt compound.

DOI: 10.1103/PhysRevB.99.195148

I. INTRODUCTION

Dirac semimetals (DSMs) have become an active area of research due to their exceptional transport properties, arising as a consequence of a linear band crossing, or Dirac point, which results in massless charge carriers known as Dirac fermions. In order for the transport to be dominated by Dirac fermions, Dirac points in the electronic structure should be proximal to the Fermi level with few or no other bands at the same energy. DSMs could be considered three-dimensional (3D) analogs of graphene, which is a two-dimensional (2D) material with purely linear band crossings at the Fermi level [1]. There is an additional constraint on DSMs, however, that graphene does not satisfy; namely, that the crossing must be symmetry-protected such that it does not gap out due to spin-orbit coupling (SOC) and/or an applied magnetic field [2]. In recent years, a push to develop design principles for 3D DSMs that possess these protected linear band crossings near the Fermi level has yielded the experimental verification compounds Na₃Bi and Cd₃As₂ [3-6]. A closely related class of materials to DSMs have degenerate Dirac crossings along a line in the Brillouin zone, rather than points, and these are called nodal-line semimetals [7-9]. SOC is often unfavorable for the formation of these features in the electronic structure since symmetry protections tend to exist only along high-symmetry lines in k space, and thus do not usually fully stabilize the nodal lines, which are frequently circular or oval [10]. Nodal-line semimetals with certain symmetry protections exhibit drumhead-shaped surface states that are proposed to host novel correlated electron physics [9]. Recently, it has been proposed that, rather than displaying nodal lines, there could exist compounds in which the degenerate band crossings trace a connected 3D surface-a spherical shell-in the Brillouin zone [11], these nodal-sphere semimetals could exhibit similar correlated surface states.

Analyzing and understanding crystallographic and compositional motifs that lead to linear band crossings proximal to the Fermi level is important in identifying new compounds and advancing potential technological applications. It is also of particular interest to examine oxide compounds which are somewhat under-represented in the space of Dirac and related materials. To this end, we present electronic structure calculations of the complex platinum group metal oxides, Na M_3O_4 (M = Pd, Pt). Since the strength of SOC scales approximately as Z^4/n^3 (Z is the atomic number and n is the principal quantum number), comparing the Pd and Pt compounds allows the role of SOC in the development of the electronic structure to be scrutinized.

From the perspective of current applications, these complex Pd and Pt oxides are known for their their ability to be p doped. Unlike most oxides which have valence bands dominated by localized O p orbitals and are most conveniently electron doped, these Pd and Pt compounds can be easily hole doped by using alkali metals [12,13]. The semiconducting d^8 oxides (Ca/Sr)(Pd/Pt)₃O₄ can be hole doped by Na substitution for Ca or Sr. Substitution on this site does not significantly change the crystal structure or the qualitative nature of bands, allowing fine tuning of the Fermi level. Sodium doping of CaM_3O_4 and SrM_3O_4 has been shown to convert these semiconductors into metals [14-17]. Na M_3O_4 can be viewed as the 100%-doped end member in which Ca or Sr has been completely replaced by Na. Lamontagne et al. recently proposed that Na doping in CaPd₃O₄ and SrPd₃O₄ may proceed via a nonuniform percolative transition, with significantly increased local disorder in SrPd₃O₄ impeding percolation relative to CaPd₃O₄ [18].

Platinum metal has also long been ubiquitous as a chemical catalyst, and Pd oxides are well known for their use in automotive catalytic converters. In the 1970s, $Na_xPt_3O_4$ was proposed to be one of the main active components of the Adams catalyst, a compound useful for catalyzing organic hydrogenation reactions that was first described by Adams and Voorhees in 1922, whose properties proved difficult to reproduce over a half-century of subsequent syntheses [19,20]. In recent years,

^{*}steicher@ucsb.edu

[†]lschoop@princeton.edu

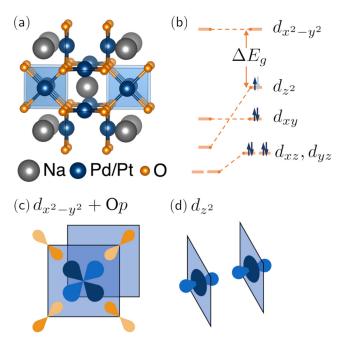


FIG. 1. (a) Crystal structure of Na(Pd/Pt)₃O₄ in the nonsymmorphic space group $Pm\bar{3}n$ (#223). (b) Crystal-field splitting for the general square planar case, modified here through the interaction between the partially filled d_{z^2} levels across the faces of the square planes. The energy gap for the corresponding d^8 compounds would be between the d_{z^2} and $d_{x^2-y^2}$ levels. For Sr M_3O_4 and Ca M_3O_4 , the d_{z^2} states are filled, resulting in a band insulator. For Na M_3O_4 , however, the d_{z^2} states are partially filled, resulting in a metal. (c) The $d_{x^2-y^2} + O p$ interaction for a single square plane. (d) The stacked MO_4 square planes in the unit cell yield a network of interplanar $M_{d_z^2}$ bonding.

there has been significant general interest in the enhanced catalytic action of ionic noble-metal species and noble-metal oxides [21–24]. It is interesting to ask whether topological surface states and/or potentially enhanced charge transport due to linear dispersion at Dirac and Weyl band crossings could play a role in the catalytic activity of these compounds. Indeed, a growing number of catalysts including the pure noble metals Pd and Pt have been suggested as hosting nontrivial topological states and linear band crossings in their electronic structure [25,26].

NaPd₃O₄ and NaPt₃O₄ crystallize in the cubic $Pm\bar{3}n$ space group (#223) shown in Fig. 1(a). The structure consists of corner-connected MO_4 square planes. Closely spaced, infinite parallel stacks of MO_4 square planes also run along all three cubic axes. The space group has an *n* glide that results in nonsymmorphic symmetry. Recently, there has been significant interest in 3D nonsymmorphic crystals, especially due to the fact that these compounds can host protected multiply degenerate band crossings in which more than four bands overlap at high-symmetry points in the Brillouin zone. At these overlap points, even more exotic analogs of Dirac fermions could potentially be realized [27–29]. Group #223 can host band crossings that are up to eight-fold degenerate [29].

The crystal-field splitting of square planar M as presented in Fig. 1(b) has been described by using tight-binding models as popularized by Hoffmann [30], and an important feature is that interaction of the d_{z^2} orbitals raises this level over the d_{xy} level, as shown on the right. The $d_{x^2-y^2}$ orbitals form a 3D-connected bonding network with the O p orbitals at the corner of each square planar group [Fig. 1(c)]. The d_{7^2} orbitals form independent bonding chains with two square planar groups per unit cell along the cubic axes [Fig. 1(d)]. 3D bonding increases the energy dispersion of the $d_{x^2-y^2}$ bands (which would not be expected to have particularly significant dispersion in the isolated chain model), allowing them to cross the d_{z^2} bands near the center of the Brillouin zone, and, as will be shown, resulting in symmetry-protected Dirac crossings. Doublet, Canadell, and Whangbo have developed a semi-empirical tight-binding model for NaM_3O_4 and provided band-folding explanations for the electronic structure of the $d_{x^2-y^2}$ and d_z^2 states near the Fermi-level [31]. The Pd and Pt ions in Na M_3O_4 have a nominal 2.33+ charge, making these compounds metallic with partially filled d_z^2 orbitals. This is in contrast to the isostructural, band semiconducting compounds with Ca or Sr which have $d^8 Pd^{2+}$ or Pt^{2+} and a fully filled d_z^2 level. The compounds Ca M_3O_4 and Sr M_3O_4 have previously been suggested to be topological semimetals, but these compounds are actually insulating both at appropriate levels of electronic structure theory as well as in experimental studies [18,32].

We present here a detailed electronic structure description of the two compounds $NaPd_3O_4$ and $NaPt_3O_4$ using a combination of plane-wave pseudopotential and plane-wave local orbital density-functional theory calculations in tandem with tight-binding models to identify Dirac crossings near the Fermi level that appear to form a nodal cube, not unlike the nodal-sphere semimetallic state proposed previously by Wang *et al.* [11]. Spin-orbit coupling fragments this nodal cubic state into 14 protected Dirac crossings in both the Pd and Pt compounds. This work points to the potential of complex oxides to enrich the domain of Dirac and related quantum materials and supports the view that there may be a role for previously ignored features in the electronic structure playing a role in the catalytic properties of compounds of the platinum group metals.

II. COMPUTATIONAL METHODS

First-principles electronic structure calculations were performed by using the Vienna ab initio Simulation Package VASP [33-35] and WIEN2K codes [36], with and without SOC. Computation performed using VASP utilized projectoraugmented waves [37,38], while computations involving WIEN2K employed linear augmented plane waves and local orbitals [39]. The Perdew, Burke, Ernzerhof (PBE) generalized gradient approximation [40] was used for the exchange energy while Vosko, Wilk, and Nusair interpolation was used for the correlation energy [41]. While computationally expensive hybrid functionals such as HSE06 [42] are known to improve band-gap estimations in a wide range of semiconductors, including in simple and complex Pd oxides [43], previous empirical comparisons of transition-metal and metallic transition-metal oxide systems suggest that the additional electron screening in hybrid functionals provides little improvement over PBE and can reduce accuracy in estimations of key material parameters [44,45]. We employed

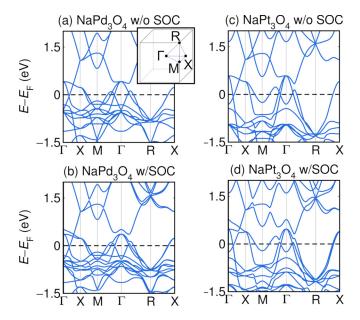


FIG. 2. Band structures of NaPd₃O₄ (a) without and (b) with SOC and NaPt₃O₄ (c) without and (d) with SOC calculated by using the PBE functional in VASP. A linear band crossing near the Fermi level is evident in the Γ -X direction with large energy dispersion (\approx 0.5 eV). Smaller linear band crossings occur along *M*- Γ and Γ -*R*.

a $10 \times 10 \times 10$ Monkhorst–Pack [46] k-point grid. Structural relaxations and static self-consistent calculations used the default smearing algorithms for VASP and WIEN2K, first-order Methfessel–Paxton [47] and tetrahedral smearing with Blöchl corrections [48], respectively. The plane-wave energy cutoff for VASP was set to 520 eV and the plane-wave expansion parameter, RKMAX, for WIEN2K was set to 8.5. The density of states calculations provided in the Supplemental Material utilized tetrahedral smearing and band structures were calculated by using Gaussian smearing [49]. Structures were initially relaxed in VASP (to lattice parameters of 5.728 and 5.765 Å for the Pd and Pt compounds, respectively, within expected error from experimental values of 5.650 Å [50] and 5.689 Å [51]) using the conjugate gradient descent algorithm with a force cutoff of 0.01 eV/Å. Self-consistent static calculations and electronic structure calculations were subsequently performed using VASP and WIEN2k with energy convergence better than 10^{-5} eV.

Brillouin zone energy gap calculations on the $k_z = 0$ plane were calculated in VASP using a $30 \times 30 \times 1$ *k*-mesh and Gaussian smearing. Orbital-projected band structures were calculated using WIEN2k without SOC but were checked against results from VASP and a spin-polarized WIEN2k calculation with SOC. The charge density of valence electrons contributing to energy states within 0.1 eV of the Fermi level was computed using WIEN2k. Irreducible representations of electronic bands were determined using the IRREP subprogram in WIEN2k. The Brillouin zone Dirac crossing calculations were performed by projecting our VASP calculations onto maximally localized Wannier functions using Wannier90,[52] starting from initial projectors corresponding to valence orbitals (Na s and p; Pd/Pt s,p, and d; O s and p), constructing a tight-binding model from these

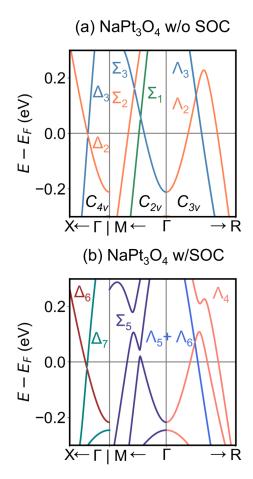


FIG. 3. Close-up views of the Dirac cones along the Γ -*X*, *M*- Γ , and Γ -*R* paths of the band structures of NaPt₃O₄ (a) without and (b) with SOC calculated in WIEN2K. The Dirac cones along *M*- Γ are gapped by SOC whereas the other Dirac cones are protected in both compounds. Colors of the plotted bands represent calculated irreducible representations (irreps).

localized Wannier functions, and subsequently using the GAPPLANE and GAPCUBE functions in the Wannier Tools package [53] with 400×400 and $200 \times 200 \times 200$ sampling meshes, respectively. The tight-binding models are very slightly asymmetrical with respect to the cubic crystalline structure; this is a result of Wannier90's Wannier function localization algorithm rather than inherent asymmetry in the original DFT simulation. Structures were visualized with VESTA [54].

III. RESULTS AND DISCUSSION

The electronic band structures of NaPt₃O₄ and NaPd₃O₄ with and without the inclusion of SOC are given in Fig. 2. We focus our attention on the bands near the Fermi level from the Γ to X high-symmetry points. These bands cross just below the Fermi level with a large linear energy dispersion that is as great as 0.5 eV in NaPt₃O₄. Viewing along the *M*- Γ and Γ -*R* branches, additional smaller linear band crossings can be seen. A close-up view of the band crossings is provided in Fig. 3. The Dirac crossings along Γ -*X* and Γ -*R* are symmetry protected, while the *M*- Γ crossing is gapped out when SOC is included. Calculations in

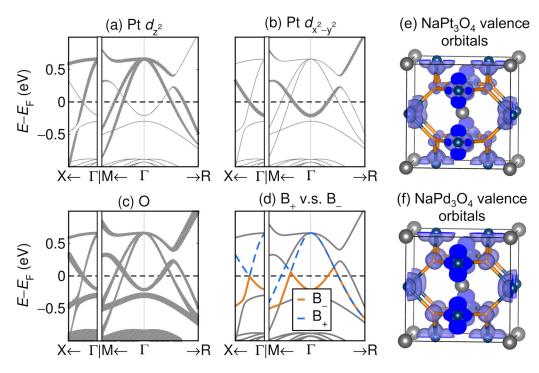


FIG. 4. Determination of orbital character. (a)–(c) Orbital projected band structures of NaPt₃O₄ calculated in WIEN2K. Dirac cones are generated by the crossing of $M d_{z^2}$ and $M d_{x^2-y^2}$ bands. (d) The same plot with an overlay of the simulation-defined bands B_+ and B_- that form the tops and bottoms of the Dirac crossings, respectively. These bands will be further visualized in Figs. 5 and 6. (e), (f) Charge density of valence states contributing to bands within 0.1 eV of the Fermi level in NaPt₃O₄ and NaPd₃O₄, respectively, suggesting primary contributions from Pt $d_{x^2-y^2}$ and d_{z^2} orbitals in combination with O p orbitals. The charge density is visualized at a constant contour of $0.0021e^{-1}$ Å⁻³.

VASP and WIEN2K show good agreement for states above the Fermi level and in the Fermi-level region where the Dirac crossings lie. Predicted energies for states below the Fermi level differ slightly for the two methods. This is unsurprising given the local orbital approach in WIEN2K. Although WIEN2K calculations are not shown in Fig. 2 for the sake of visual clarity, a direct comparison of the band structures calculated in both codes is available in the Supplemental Material [49].

Nonsymmorphic symmetry protects multiply degenerate band crossings in both materials. Eightfold crossings can be seen at energies near 1.5 eV and between -0.6 and -1.5 eV at the *R* point, the energy of the latter sub-Fermi level degeneracy varying significantly with *M* site species and simulation program. While the eight-fold crossings are very far from the Fermi level in these compounds, they are relevant to the electronic structure of closely related compounds. It has been shown that the Fermi level of the isostructural compound LaPd₃S₄ lies near the upper eightfold degeneracy at *R* and that Pt₃O₄ appears to have a Fermi level near the lower eightfold degeneracy [29].

To explore the origins of the Fermi level Dirac crossings, we consider symmetry and orbital character. Formally, symmetry protects band crossings from gapping if the two crossing bands have different irreducible representations (irreps) with respect to the point group of the *k* vector [10]. When SOC is significant, the two-band irreps must be different with respect to the double group, which accounts for spin. The gapping of the Dirac crossing along M- Γ when SOC is introduced can be explained as a consequence of the cubic space group. The Γ -X, M- Γ , and Γ -R lines in k space obey C_{4v} , C_{2v} , and C_{3v} point-group symmetries, respectively. Without SOC, the crossing bands have different irreps in each of these point groups and all three crossings are allowed (see the irrep-colored bands of Fig. 3). When SOC is introduced, there is only one spinor irrep available in the double group of C_{2v} and the Dirac crossing along M- Γ therefore gaps out. Explicitly, we calculate the irreps of the crossing bands along the three high-symmetry Brillouin-zone lines in the system with SOC: Γ -*X* : { Δ_6 , Δ_7 } \rightarrow { Δ_7 , Δ_6 }; *M*- Γ : { Σ_5 , Σ_5 } \rightarrow $\{\Sigma_5, \Sigma_5\}; \ \Gamma - R : \{\Lambda_4, \Lambda_5 + \Lambda_6\} \rightarrow \{\Lambda_5 + \Lambda_6, \Lambda_4\}$ [55]. The irreps are identical for both the Pd and Pt compounds. There is an exchange of irreps at the Γ -X and Γ -R crossings, and these crossings are therefore protected, while no such exchange can occur along M- Γ . More specifically, the Dirac crossings along Γ -X and Γ -R are protected by C_4 and C_3 rotational symmetry, respectively, as the irreps of the crossing bands at these points are distinguished by their character with respect to these symmetry operations. Overall, we can state that the symmetry protections in the system are completely general properties of the cubic Brillouin zone and can be extended to cubic compounds in other space groups.

Figure 4 presents orbital-projected band structures and a visualization of the valence electron charge distribution contributing to energy states near the Fermi level in Na M_3 O₄. The difference in symmetry between the bands crossing at the Dirac cones can be seen to reflect a difference in orbital origin. The dominant contributions are from $M d_{x^2-y^2}$ and d_{z^2} orbitals in tandem with O (mainly *p*) states. Density of

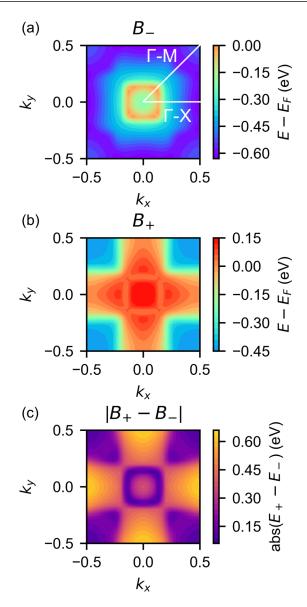


FIG. 5. Heat maps of band energies on the $k_z = 0$ plane. Energies of the B_- and B_+ bands are shown for NaPt₃O₄ with SOC. The absolute value of the energy separation between the B_- and B_+ bands is shown in panel (c). The Fermi-level Dirac crossings between the B_- and B_+ comprise an approximately square energy degeneracy loop in the region $\{-0.2 \le k_x \le 0.2; -0.2 \le k_y \le 0.2\}$.

states calculations supporting this conclusion are also offered in the Supplemental Material [49]. The Dirac crossings are created by the overlap of partially filled d_{z^2} bands with largedispersion $d_{x^2-y^2}$ bands that dip down in energy at Γ . Oxygen states overlap significantly with both orbitals but interact more strongly with the $d_{x^2-y^2}$ band. It is important to note here that we are using a local bonding basis to describe the states, rather than the crystal coordinates, such that $d_{x^2-y^2}$ refers to an *M* orbital with lobes pointing along the square planar coordinated bonds towards oxygen. Although the orbital decomposition generally follows the energy-level diagram of Fig. 1(c), the non-negligible interaction of oxygen states with the d_{z^2} might be somewhat unexpected. Figures 4(e) and 4(f) show that the O *p* orbitals are oriented along [111] axes, facing the out-of-square coordination plane defined by *M*-O bonding such that they can interact with additional orbitals along the *z* axis. The Pd compound has more disperse orbitals than the Pt compound and has less directional *d* orbital preference along the Pd-O bonds (the Pd *d* orbitals have more d_{xy} character in addition to the primary $d_{x^2-y^2}$, d_{z^2} character).

From the perspective of simulation, all three Dirac crossings of interest occur between the same two bands. Since the system is metallic, there is no good division between "valence" and "conduction" bands, we refer to the band with greater energy at the Γ point, which descends in energy towards the Dirac crossings along Γ -X, Γ -M, and Γ -R, as the " B_+ " band, while the band with lower energy at Γ will be referred to as " B_- ." These simulation-defined bands are visualized in Fig. 4(d). While this scheme is sufficient for our purposes, note that the Γ -R Dirac cone is actually sixfold degenerate (composed of three separate sets of spin-degenerate bands) in the non-SOC calculations, although this degeneracy is gapped by SOC.

Figure 5 shows the energy landscape of the B_- and B_+ bands along a constant $k_z = 0$ slice in the (cubic) Brillouin zone for NaPt₃O₄. As discussed by Doublet *et al.*, the roughly square shape of the Fermi surface of the d_{z^2} band can be viewed as a consequence of the relative independence of the MO_4 bonding chains along x and y [31]. Despite the 3D connectivity of the $M d_{x^2-y^2} + O p$ bonding network, the corresponding Fermi surface is roughly square as well. The net result is that both the B_- and B_+ bands have cubic symmetry. There are two regions where these bands come close together: along a square loop at the center of the Brillouin zone, corresponding to the Dirac crossings, and, far below the Fermi level, along the Γ -M line from the center of the diagram to the corners.

To achieve a higher k-space sampling resolution to more precisely visualize the locations of the Dirac crossings, we implemented Wannier-interpolated tight-binding models as discussed in Sec. II. Figure 6 shows calculations of the energy gaps in the area of the Brillouin zone near the central square intersection region of Fig. 5. Figures 6(a) and 6(b)demonstrate that there is a full square nodal line in the $k_z = 0$ plane that is interrupted by gapping along Γ -M when SOC is introduced. Figure 6(c) and 6(d) show that the nodal square in the Pd compound without SOC is part of a larger smallenergy-splitting region with roughly cubic shape, but small holes in the nodal cube, which SOC gaps into eight protected degeneracies along Γ -R and six protected degeneracies along Γ -X for a total of 14 Dirac points. The Γ -R crossings have sharp dispersion whereas the Γ -X Dirac crossings have very small dispersion along the faces of the cube and appear flat and plate-like even after SOC is included. The nodal cube in NaPt₃O₄ in the absence of SOC is found to be a complete cube [Fig. 6(e)], similar to a nodal sphere semimetal. More generally, the cubic degeneracy region in the Pt compound is noticeably more square than the similar region in the Pd compound and the stronger SOC in this compound more fully removes the nodal cube degeneracy, although small, plate-like dispersion along Γ -X remains (see Supplemental Material for $k_7 = 0$ plane nodal lines in NaPt₃O₄) [49]. At the smallest energy splittings available in our tight-binding models, the intersection region in the Pd compound is reduced to a number of nodal lines. The Pt compound, by contrast, has a full,

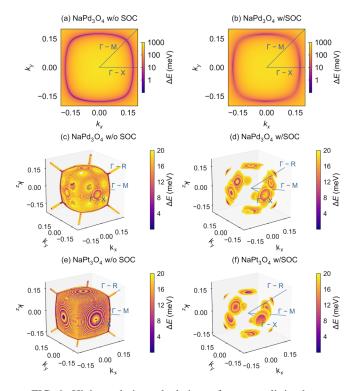


FIG. 6. High-resolution calculations of energy-splitting between $NaM_3O_4 B_+$ and B_- bands in k space, performed with the tightbinding model: 2D plots in panels (a) and (b) show all energies on a logarithmic scale, while for 3D plots in panels (c)–(f) the maximum energy cutoff was 20 meV.

nodal-cubic Dirac degeneracy in the absence of SOC down to the lowest reasonable simulation energies (see Supplemental Material) [49]. These qualitative findings were found to be robust to reparametrization of the tight-binding model by using a number of alternative orbital projections, including a model including only the $M d_{x^2-y^2}$, d_{z^2} , and O p orbitals previously discussed.

IV. CONCLUSIONS

We have shown that the high-symmetry structures of Na M_3O_4 result in the formation of 14 Fermi level Dirac crossings that are protected against spin-orbit coupling, with a particularly large linear energy dispersion region about Γ -X. These compounds can potentially be doped between interesting electronic states including the nonsymmorphic-symmetry-enabled degenerate R point bands of M_3O_4 , the Dirac cones of Na M_3O_4 discussed, and the insulating compounds Ca M_3O_4 and Sr M_3O_4 .

The $Pm\bar{3}n$ cubic space group appears promising for the generation of a stable nodal cube state with surface states that could host exciting correlated electron physics [11]. While the compounds discussed here have significant SOC that prevents the realization of a true nodal cube, this state might be achieved in related systems by lowering SOC using lighter elements. The presence of a full nodal cube in the Pt compound without SOC also gives us some hints as to how we might go about creating such a state in a real material. The cubic Dirac degeneracy of these compounds is a direct consequence of the

quasi-independence of the MO_4 square-planar bonding chains in the unit cell. The d_{z^2} bonding network is nearly independent along the cubic axes resulting in a B_+ band that creates the nodal cubic state [Fig. 5(a)]. Dirac character along the Γ -Xdirection is directed towards the cubic faces of the (001) single crystal surface with metallic or mixed Dirac-metallic character along other high-symmetry directions. Interaction with the ligand oxygens increases the 3D connectivity of the $d_{x^2-y^2}$ bonding network and reduces the how cubic the Dirac crossings are. In the absence of SOC, increased Pd-O bonding in the Pd compound compared with the Pt-O bonding of the Pt compound results in a less perfect nodal cube degeneracy. A possible way to reduce the M-O bonding would be to increase the lattice parameter though chemical strain; for example, though substituting K⁺ or Rb⁺ for Na⁺.

Effects of these predicted Fermi-level Dirac crossings and large *k*-space regions of linear dispersion may be detectable in magnetotransport measurements. A number of compounds in this family, including Na_xPt₃O₄, have been previously synthesized as single crystals suitable for transport studies [56–59]. Nodal-line semimetals frequently show signs of anomalous transport such as large magnetoresistance and low-field quantum oscillations [60,61].

The band inversion in this system results from bonding chains, rather than the relativistic contraction of *s* orbitals on a single atomic site (as in the now-canonical DSM examples Na_3Bi and Cd_3As_2), so this compound can be classified with other new "molecular" semimetals such as ZrSiS that feature large linear energy dispersion [62].

The significance of the transport contribution from linear bands to catalysis has not yet been fully deciphered. Nonetheless, Dirac transport in NaM_3O_4 may further our understanding of the Adams catalyst on the centennial of its discovery. While topological semimetals have been previously examined as catalysts for the hydrogen evolution reaction, hydrogenation catalysts may also prove a fruitful avenue of inquiry.

ACKNOWLEDGMENTS

We thank Claudia Felser for encouraging this work, and Douglas Fabini and Xie Zhang for helpful discussions. This work was supported by the National Science Foundation through DMR 1710638. Initial support to S.M.L.T. from a New Partnerships Challenge Grant program of the California Nanosystems Institute (CNSI) is gratefully acknowledged. L.M.S. was supported by the Princeton Center for Complex Materials, a Materials Research Science and Engineering Center (MRSEC) DMR-1420541, and by the Princeton Catalysis Initiative (PCI). We gratefully acknowledge the use of the computing facilities of the Center for Scientific Computing at UC Santa Barbara, supported by NSF CNS-1725797 and by the NSF MRSEC Program (NSF DMR-1720256). S.M.L.T. has been supported by the National Science Foundation Graduate Research Fellowship Program under Grant No. DGE-1650114. Any opinions, findings, and conclusions or recommendations expressed in this material are those of the authors and do not necessarily reflect the views of the National Science Foundation.

- [2] N. P. Armitage, E. J. Mele, and A. Vishwanath, Rev. Mod. Phys. 90, 015001 (2018).
- [3] Q. D. Gibson, L. M. Schoop, L. Muechler, L. S. Xie, M. Hirschberger, N. P. Ong, R. Car, and R. J. Cava, Phys. Rev. B 91, 205128 (2015).
- [4] B.-J. Yang and N. Nagaosa, Nat. Commun. 5, 4898 (2014).
- [5] Z. K. Liu, B. Zhou, Y. Zhang, Z. J. Wang, H. M. Weng, D. Prabhakaran, S.-K. Mo, Z. X. Shen, Z. Fang, X. Dai, Z. Hussain, and Y. L. Chen, Science 343, 864 (2014).
- [6] Z. K. Liu, J. Jiang, B. Zhou, Z. J. Wang, Y. Zhang, H. M. Weng, D. Prabhakaran, S.-K. Mo, H. Peng, P. Dudin, T. Kim, M. Hoesch, Z. Fang, X. Dai, Z. X. Shen, D. L. Feng, Z. Hussain, and Y. L. Chen, Nat. Mater. 13, 677 (2014).
- [7] C. Fang, Y. Chen, H.-Y. Kee, and L. Fu, Phys. Rev. B 92, 081201(R) (2015).
- [8] L. S. Xie, L. M. Schoop, E. M. Seibel, Q. D. Gibson, W. Xie, and R. J. Cava, APL Mater. 3, 083602 (2015).
- [9] Y.-H. Chan, C.-K. Chiu, M. Y. Chou, and A. P. Schnyder, Phys. Rev. B 93, 205132 (2016).
- [10] L. M. Schoop, F. Pielnhofer, and B. V. Lotsch, Chem. Mater. 30, 3155 (2018).
- [11] J. Wang, Y. Liu, K.-H. Jin, X. Sui, L. Zhang, W. Duan, F. Liu, and B. Huang, Phys. Rev. B 98, 201112(R) (2018).
- [12] G. Hautier, A. Miglio, G. Ceder, G.-M. Rignanese, and X. Gonze, Nat. Commun. 4, 2292 (2013).
- [13] S. Sheng, G. Fang, C. Li, S. Xu, and X. Zhao, Phys. Status Solidi A 203, 1891 (2006).
- [14] S. Ichikawa and I. Terasaki, Phys. Rev. B 68, 233101 (2003).
- [15] K. Itoh and N. Tsuda, Solid State Commun. 109, 715 (1999).
- [16] T. Taniguchi, Y. Nagata, T. C. Ozawa, M. Sato, Y. Noro, T. Uchida, and H. Samata, J. Alloys Compd. 373, 67 (2004).
- [17] K. Itoh, Y. Yano, and N. Tsuda, J. Phys. Soc. Jpn. 68, 3022 (1999).
- [18] L. K. Lamontagne, G. Laurita, M. Knight, H. Yusuf, J. Hu, R. Seshadri, and K. Page, Inorg. Chem. (Washington, DC, U. S.) 56, 5158 (2017).
- [19] V. Voorhees and R. Adams, J. Am. Chem. Soc. 44, 1397 (1922).
- [20] D. Cahen and J. A. Ibers, J. Catal. 31, 369 (1973).
- [21] M. S. Hegde, G. Madras, and K. C. Patil, Acc. Chem. Res. 42, 704 (2009).
- [22] J. A. Kurzman, L. M. Misch, and R. Seshadri, Dalton Trans. 42, 14653 (2013).
- [23] Y. Nishihata, J. Mizuki, T. Akao, H. Tanaka, M. Uenishi, M. Kimura, T. Okamoto, and N. Hamada, Nature (London) 418, 164 (2002).
- [24] J. Li, U. G. Singh, T. D. Schladt, J. K. Stalick, S. L. Scott, and R. Seshadri, Chem. Mater. 20, 6567 (2008).
- [25] C. R. Rajamathi, U. Gupta, N. Kumar, H. Yang, Y. Sun, V. Süß, C. Shekhar, M. Schmidt, H. Blumtritt, P. Werner, B. Yan, S. Parkin, C. Felser, and C. N. R. Rao, Adv. Mater. 29, 1606202 (2017).
- [26] B. Yan, B. Stadtmüller, N. Haag, S. Jakobs, J. Seidel, D. Jungkenn, S. Mathias, M. Cinchetti, M. Aeschlimann, and C. Felser, Nat. Commun. 6, 10167 (2015).
- [27] B. J. Wieder, Y. Kim, A. M. Rappe, and C. L. Kane, Phys. Rev. Lett. 116, 186402 (2016).

- [28] L. M. Schoop, M. N. Ali, C. Straßer, A. Topp, A. Varykhalov, D. Marchenko, V. Duppel, S. S. P. Parkin, B. V. Lotsch, and C. R. Ast, Nat. Commun. 7, 11696 (2016).
- [29] B. Bradlyn, J. Cano, Z. Wang, M. G. Vergniory, C. Felser, R. J. Cava, and B. A. Bernevig, Science 353, aaf5037 (2016).
- [30] R. Hoffmann, Angew. Chem., Int. Ed. Engl. 26, 846 (1987).
- [31] M. L. Doublet, E. Canadell, and M. H. Whangbo, J. Am. Chem. Soc. 116, 2115 (1994).
- [32] G. Li, B. Yan, Z. Wang, and K. Held, Phys. Rev. B 95, 035102 (2017).
- [33] G. Kresse and J. Hafner, Phys. Rev. B 49, 14251 (1994).
- [34] G. Kresse and J. Furthmüller, Phys. Rev. B 54, 11169 (1996).
- [35] G. Kresse and J. Furthmüller, Comput. Mater. Sci. 6, 15 (1996).
- [36] P. Blaha, K. Schwarz, G. Madsen, D. Kvasnicka, and J. Luitz, WIEN2K: An augmented plane wave and local orbitals program for calculating crystal properties, WIEN2k 18.2, Vienna University of Technology, Vienna, Austria (Karlheinz Schwarz, 2001).
- [37] P. E. Blöchl, Phys. Rev. B 50, 17953 (1994).
- [38] G. Kresse and D. Joubert, Phys. Rev. B 59, 1758 (1999).
- [39] K. Schwarz, P. Blaha, and G. K. H. Madsen, Comput. Phys. Commun. 147, 71 (2002).
- [40] J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. 77, 3865 (1996).
- [41] S. H. Vosko, L. Wilk, and M. Nusair, Can. J. Phys. 58, 1200 (1980).
- [42] A. V. Krukau, O. A. Vydrov, A. F. Izmaylov, and G. E. Scuseria, J. Chem. Phys. **125**, 224106 (2006).
- [43] J. A. Kurzman, M.-S. Miao, and R. Seshadri, J. Phys.: Condens. Matter 23, 465501 (2011).
- [44] P. Janthon, S. A. Luo, S. M. Kozlov, F. Viñes, J. Limtrakul, D. G. Truhlar, and F. Illas, J. Chem. Theory Comput. 10, 3832 (2014).
- [45] F. Tran, D. Koller, and P. Blaha, Phys. Rev. B 86, 134406 (2012).
- [46] H. J. Monkhorst and J. D. Pack, Phys. Rev. B 13, 5188 (1976).
- [47] M. Methfessel and A. T. Paxton, Phys. Rev. B 40, 3616 (1989).
- [48] P. E. Blöchl, O. Jepsen, and O. K. Andersen, Phys. Rev. B 49, 16223 (1994).
- [49] See Supplemental Material at http://link.aps.org/supplemental/ 10.1103/PhysRevB.99.195148 for comparisons between electronic structure in VASP, WIEN2K, and our tight-binding model, density of states calculations, and additional comparisons between the B_+ and B_- bands described in the text for both the platinum and palladium compound, with and without SOC.
- [50] R. V. Panin, N. R. Khasanova, A. M. Abakumov, E. V. Antipov, G. V. Tendeloo, and W. Schnelle, J. Solid State Chem. 180, 1566 (2007).
- [51] J. Waser and E. D. McClanahan, Jr., Experientia 6, 379 (1950).
- [52] A. A. Mostofi, J. R. Yates, G. Pizzi, Y.-S. Lee, I. Souza, D. Vanderbilt, and N. Marzari, Comput. Phys. Commun. 185, 2309 (2014).
- [53] Q. Wu, S. Zhang, H.-F. Song, M. Troyer, and A. A. Soluyanov, Comput. Phys. Commun. 224, 405 (2018).
- [54] K. Momma and F. Izumi, J. Appl. Crystallogr. 44, 1272 (2011).
- [55] G. F. Koster, J. O. Dimmock, R. G. Wheeler, and H. Statz, *Properties of the Thirty Two Point Groups*, Dover Books on Chemistry (MIT Press, Cambridge, 1963).
- [56] D. Cahen, J. A. Ibers, and J. B. Wagner, Inorg. Chem. (Washington, DC, US) 13, 1377 (1974).

- [57] W. H. Weber, G. W. Graham, A. E. Chen, K. C. Hass, and B. L. Chamberland, Solid State Commun. 106, 95 (1998).
- [58] P. L. Smallwood, M. D. Smith, and H.-C. zur Loye, J. Cryst. Growth 216, 299 (2000).
- [59] H. Samata, S. Tanaka, S. Mizusaki, Y. Nagata, T. C. Ozawa, A. Sato, and K. Kosuda, J. Cryst. Process Technol. 2, 16 (2012).
- [60] S. Pezzini, M. R. van Delft, L. M. Schoop, B. V. Lotsch, A. Carrington, M. I. Katsnelson, N. E. Hussey, and S. Wiedmann, Nat. Phys. 14, 178 (2018).
- [61] J. Hu, Z. Tang, J. Liu, X. Liu, Y. Zhu, D. Graf, K. Myhro, S. Tran, C. N. Lau, J. Wei, and Z. Mao, Phys. Rev. Lett. 117, 016602 (2016).
- [62] S. Klemenz, S. Lei, and L. M. Schoop, Annu. Rev. Mater. Res.
 49, (2019), doi: 10.1146/annurev-matsci-070218-010114.