Universal tensor-network algorithm for any infinite lattice

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We present a general graph-based projected entangled-pair state (gPEPS) algorithm to approximate ground states of nearest-neighbor local Hamiltonians on any lattice or graph of infinite size. By introducing the structural matrix, which codifies the details of tensor networks on any graphs in any dimension *d*, we are able to produce a code that can be essentially launched to simulate any lattice. We further introduce an optimized algorithm to compute simple tensor updates as well as expectation values and correlators with a mean-field-like effective environments. Though not being variational, this strategy allows to cope with PEPS of very large bond dimension (e.g., *D* = 100) and produces remarkably accurate results in the thermodynamic limit in many situations, and specially when the correlation length is small and the connectivity of the lattice is large. We prove the validity of our approach by benchmarking the algorithm against known results for several models, i.e., the antiferromagnetic Heisenberg model on a chain, star and cubic lattices, the hardcore Bose-Hubbard model on square lattice, the ferromagnetic Heisenberg model in a field on the pyrochlore lattice, as well as the three-state quantum Potts model in field on the kagome lattice and the spin-1 bilinear-biquadratic Heisenberg model on the triangular lattice. We further demonstrate the performance of gPEPS by studying the quantum phase transition of the 2*d* quantum Ising model in transverse magnetic field on the square lattice, and the phase diagram of the Kitaev-Heisenberg model on the hyperhoneycomb lattice. Our results are in excellent agreement with previous studies.

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I. INTRODUCTION

In recent years, tensor-network (TN) states and methods [\[1–3\]](#page-13-0) have been recognized as powerful tools in different areas of physics such as quantum information theory, condensed matter physics and, recently, even quantum gravity. From the perspective of condensed matter, TN methods are widely used to understand quantum many-body systems [\[4,5\]](#page-13-0), both theoretically and numerically. In one spatial dimension, matrix product states (MPS) [\[6,7\]](#page-13-0) provide an efficient representation for the ground-state of 1*d* gapped local Hamiltonians based on their entanglement structure. MPS is also the variational wave function generated by the density matrix renormalization group (DMRG) [\[8,9\]](#page-13-0) and the time evolution block decimation method (TEBD) [\[10,11\]](#page-13-0). Projected entangled-pair states (PEPS) [\[12,13\]](#page-13-0) are a generalization of MPS, and provides an ansatz for the ground-state of quantum many-body systems in higher dimensions. The infinite-size version of PEPS (iPEPS) [\[14,15\]](#page-13-0) has also been put forward for studying the groundstate properties of 2*d* systems in the thermodynamic limit, and has been successfully applied to many different models $[16–23]$.

Despite its many virtues, a problem with the iPEPS algorithm is that it needs to be mostly re-programmed every time that one considers a new lattice. Long story short, the idea of iPEPS is generic, but the details of the implementation

are lattice-dependent. Because of this, a common strategy is to map complex 2*d* lattices to a square lattice of tensors (e.g., via some coarse-graining), in such a way that one can recycle the square-lattice code. Dealing with the square lattice [\[14,15,24,25\]](#page-13-0) indeed facilitates tensor updates and effective-environment calculations via, say, boundary MPS [\[14\]](#page-13-0), tensor renormalization group (TRG) $[26,27]$, and corner transfer matrix renormalization group (CTMRG) [\[15,24,28\]](#page-13-0). The calculation of such effective environments is however costly, and in practice is done up to PEPS bond dimension $D \sim 10-20$ in the best-case scenario. Thus, although recent development in TN techniques have extended the application of iPEPS to more complicated 2*d* structures such as triangle [\[29,30\]](#page-13-0), honeycomb $[21,31]$, kagome $[17,32]$, star $[23]$ and cubic [\[33,34\]](#page-13-0) lattices, many different structures are still left behind, including important 3*d* lattices such as pyrochlore, hyperhoneycomb, and diamond lattices, to name a few.

In this paper, by introducing a new and efficient standard for storing the connectivity information of a TN corresponding to a given lattice structure i.e., the *structure* matrix, we present a generic tensornetwork algorithm for the simulation of nearest-neighbor local Hamiltonians on any infinite lattice. More specifically, we develop a graph-based projected entangled-pair state (gPEPS) method for any infinite lattice structure or graph in any dimension *d*, assuming translation invariance. In our implementation, we use a simple update (SU) algorithm to simulate imaginary-time evolution (ITE) in order to approximate the ground state (GS) of the system on lattices with coordination number *z*, using rank- $(z + 1)$

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tensors. On top of being generic, our approach can accurately handle large PEPS bond dimension (such as $D = 100$) in the thermodynamic limit. In our approach, expectation values are estimated using a mean-field-like environment, which provides a remarkably good approximation in many cases, specially if the correlation length is small and the coordination number *z* is large. As benchmarks, we apply our gPEPS technique to several 2*d* and 3*d* models, i.e., the antiferromagnetic Heisenberg (AFH) model on a chain, star and cubic lattices, the hardcore Bose-Hubbard (HBH) model on square lattice, the spin-1 bilinear-biquadratic (BLBQ) Heisenberg model on the triangular lattice, the three-state quantum Potts (3SQP) model in field on the kagome lattice, and the ferromagnetic Heisenberg model in field (FHF) on the pyrochlore lattice. We further challenge our technique by studying the quantum phase transition (QPT) of the transverse-field Ising model (ITF) on the square lattice and the full phase diagram of the Kitaev-Heisenberg model on the hyperhoneycomb lattice.

The paper is organized as follows. In Sec. II A, we introduce the concept of structure matrix to store the connectivity information of any TN graph and on top of that, we develop the gPEPS machinery and an efficient simpleupdate algorithm for approximating the ground-stet of local Hamiltonians. Further discussions regarding the calculation of expectation values with both simple and full environment, as well as relation to the Bethe and Husimi trees are provided in this section. We present our energy benchmark results for different models in Sec. [III](#page-4-0) and demonstrate the performance of gPEPS technique for studying the QPT in Sec. [IV.](#page-7-0) Finally, Sec. [V](#page-8-0) is devoted to conclusion and further discussions on the advantages and drawbacks of the method.

II. METHOD

In this section, we first review the basic ideas of iPEPS and how the ground state of local Hamiltonians ar represented and stored in TN language.

A. gPEPS basics

Consider a generic infinite lattice composed of a periodically repeating unit cell in arbitrary dimension *d*. To each vertex *i* of the lattice, we associate a rank- $(z + 1)$ iPEPS tensor $T_{l_1,\dots,l_s}^{s_i}$, where *s* is the physical index taking up to *p* values for the local basis $\{|\mathcal{C}\rangle_{\mathcal{C}=1,\dots,p}\}$, and l_1,\dots,l_z are virtual indices taking up to *D* values. We also associate diagonal bond matrices λ_k to edges E_k of the lattice. In 1*d* with open boundary conditions, these λ matrices contain the Schmidt coefficients (singular values) obtained when considering the bipartiion of one half of the system versus the other half. In two and higher dimensions, they are an approximation to the relevant degrees of freedom describing the physical system for the environment connected by the bond index. By gluing these tensors along theirs virtual legs, we end up with a *d*-dimensional PEPS with the same structure as the original lattice.

In order to approximate the GS wave function of a quantum lattice model with nearest-neighbor Hamiltonian terms $H_{i,j}$, we apply the imaginary-time evolution operator $U_{i,j}$ = exp(−δτ*Hi*,*^j*) on each edge *k* shared between two neighboring

FIG. 1. (a) The 2*d* star lattice. The blue region hughlights a sixsite unit cell. (b) The iPEPS TN corresponding to the star lattice unit cell.

tensors T_i and T_j of the PEPS, and subsequently update the λ_k matrix as well as the T_i and T_j tensors. To make this as general and systematic as possible, we need extra information about the connections between neighboring tensors in the TN. More precisely, considering each local iPEPS tensor as a multidimensional array $T(p, D_1, \ldots, D_z)$, we have to know *a priori* which dimensions of the T_i , T_j arrays are connected along the edge E_k of the lattice so that we could update the tensors along their corresponding shared edges,each time the imaginary-time evolution operator acts on the lattice. Current state-of-the-art iPEPS algorithm typically takes care of this technical issue by mapping the 2*d* lattices to coarse-grained square structure. However, extending this strategy to any structure particularly, the 3*d* lattices, is not possible. In the next section, we present a generic method to resolve this problem.

B. Structure matrix

Here we present an efficient method for storing the connectivity information of a TN corresponding to a given lattice structure. We illustrate our strategy for the example of the star lattice in 2*d* [Fig. 1(a)]. The generalization to other lattices and dimensions is straightforward (see Appendixes). Figure 1(b) illustrates the translationally invariant six-site unit cell TN of an infinite star lattice. Considering this TN as a graph in which the tensors T_i correspond to graph nodes and edges E_k (tensor legs) correspond to graph links, the connectivity information of the star TN is given by the so called *incidence* matrix [\[35\]](#page-13-0):

The rows (columns) of matrix (1) correspond to tensors (edges), and the two nonzero entries in each column distinguish the two connected tensors along that edge. Although the incidence matrix already contains important data about the underlying network, crucial information regarding the corresponding bond dimensions of connected virtual indices is still missing. To fill this gap, we introduce another matrix, i.e., the *structure* matrix (SM), which is obtained from the incidence matrix by replacing its nonzero elements at each row by the corresponding label of the index in the tensor array:

This matrix now contains detailed information about the PEPS for the star lattice of Fig. $1(b)$ and the connectivity information of two neighboring tensor along their shared edges are stored in the columns of the SM. For example, according to the second column of SM (2) , the edge E_2 connects the bond matrix λ_2 and the dimensions 3 and 2 of tensors T_1 and T_3 , respectively. Thanks to this information, the algorithm can automatically recognize the links and the tensors where twobody gates are applied, and implement a simple update. This is done by looping over the columns of the SM and systematically updating the iPEPS tensors along their corresponding edges, which can now be done automatically and regardless of the underlying lattice.

The choice of unit cell for different lattice structures is generically both problem and lattice dependent. One should therefore choose the unit cell in such a way that it is commensurate with the expected ordering and symmetries of the ground-state wave function. Beside, in some of the TN optimization process such as ITE, one breaks the single-site transnational invariance of the lattice due to the two-body nature of the ITE operator which acts on a unit cell with twodifferent local sites $[10,11]$. However, this intentional braking of symmetry is usually restored in the thermodynamic limit [\[25\]](#page-13-0). Beside, choice of unit cell will influence the overall computational cost of the algorithm since, unit cells with larger number of sites and bonds will need more computational time in order to converge to the ground state. See also Appendix [A.](#page-9-0) On the other hand, choosing a too small unit cell might come with the risk of missing some phases which span beyond the unit cell. One should therefore run the simulations on unit cells with different sizes and symmetries to make sure the correct results is achieved.

Let us further remark that the SM formalism that we just introduced can also be used for simulation of systems with global symmetries, such as $U(1)$ and $SU(2)$ [\[36–38\]](#page-13-0). In this setting, edges in the graph may be directed which can be easily handled by adding a sign: outgoing (incoming) links can be distinguished in the SM with positive (negative) nonzero elements.

Last but not least, the nonzero elements of the SM (2) at each row start from 2 which is due to the fact that the first dimension of tensors T_i in our notation corresponds to the physical bonds and play no role in the connectivity of the underlying TN. One can therefore use other desired convention for labeling the virtual dimensions or use composite numbers to encode extra information in each row and column of the SM.

C. Simple update for gPEPS

In our scheme, we approximate the ground state of a system by means of imaginary-time evolution and the simple update [\[39\]](#page-13-0) generalized for arbitrary graphs. This method is particularly suitable for our needs, since it does not rely on an effective environment approximation (such as the full and fast-full updates $[25]$), and is therefore implemented very similarly regardless of the lattice.

Let us now review the basics of the simple update. The ground state of a given Hamiltonian *H*, can be obtained by evolving an initial state $|\Psi_0\rangle$ in imaginary time τ as described by

$$
|\Psi_{GS}\rangle = \lim_{\tau \to \infty} \frac{e^{-\tau H} |\Psi_0\rangle}{||e^{-\tau H} |\Psi_0\rangle||}.
$$
 (3)

When the Hamiltonian is a translationally invariant sum of nearest-neighbour terms, $H = \sum_{\langle i,j \rangle} H_{i,j}$, one can approximate the ITE operator for infinitesimal time steps $\delta \tau$ by applying a *Suzuki-Trotter* decomposition, i.e.,

$$
e^{-\delta \tau H} \approx \prod_{\langle i,j \rangle} U_{i,j} = \prod_{\langle i,j \rangle} e^{-\delta \tau H_{i,j}}.
$$
 (4)

The GS of the system is then evaluated by iteratively applying $U_{i,j}$ on every shared link of the two neighboring tensors T_i , T_j and updating the tensors along the corresponding links. In this scheme, the update changes only the tensors along the link where a given gate is acting. Therefore one can update lower-rank subtensors related to them and substantially reduce the computational cost of the algorithm $[25]$, thus allowing to achieve larger bond dimension *D*.

Let us briefly revisit how the SU proceeds for the subtensors, in the context of gPEPS. Given a tensor network and its corresponding structure matrix, the SU consists of the following iterative main steps.

(1) Do for all edges E_k , $k \in [1, N_{\text{Edge}}]$ (columns of SM matrix).

(a) Find tensors T_i , T_j and their corresponding dimensions connected along edge *Ek* .

(b) Absorb bond matrices λ_m to all virtual legs $m \neq k$ of T_i , T_j tensors.

(c) Group all virtual legs $m \neq k$ to form P_l , P_r MPS tensors.

(d) QR/LQ decompose P_l , P_r to obtain Q_1 , R and L , Q_2 subtensors, respectively [\[25\]](#page-13-0).

(e) Contract the ITE gate $U_{i,j}$, with R , L and λ_k to form Θ tensor.

(f) Obtain \tilde{R} , \tilde{L} , $\tilde{\lambda}_k$ tensors by applying an SVD to Θ and truncating the tensors by keeping the *D* largest singular values (similar to 1*d* infinite TEBD [\[40,41\]](#page-13-0)).

(g) Glue back the \tilde{R} , \tilde{L} , subtensors to Q_1 , Q_2 , respectively, to form updated tensors P'_l , P'_r .

(h) Reshape back the P'_l , P'_r to the original rank-(*z* + 1) tensors T_i' , T_j' .

(i) Remove bond matrices λ_m from virtual legs $m \neq k$ to obtain the updated tensors \tilde{T}_i and \tilde{T}_j .

Figure $2(a)$ shows all these steps graphically. This process is then iterated until a convergence criteria is met.

FIG. 2. (a) Graphical representation of the SU optimization, used in the gPEPS algorithm. (b) One-site and (c) two-site expectation values, as computed with a mean-field environment, in the gPEPS scheme.

In order to have an efficient and universal algorithm applicable to any infinite lattice, the following remarks are in order. (i) In steps (b), (c), (g), and (h) one can locate the lambda matrices corresponding to each leg of a tensor from rows of the SM. For example, according to row three of the SM [\(2\)](#page-2-0), λ_2 , λ_4 , and λ_6 are connected to dimensions (legs) two, three and four of tensor T_3 , respectively. One can therefore design clever functions for absorbing (removing) λ matrices to (from) each tensor legs as well as for grouping (ungrouping) the nonupdating tensor legs by using the information stored in each row of the SM. (ii) In our SU optimization, we perform the ITE iteration starting from $\delta \tau = 10^{-1}$ and gradually decrease it to 10⁻⁵ after iterating 4000 times for each $\delta \tau$. We further check the convergence of the algorithm in each (or every 100) step by calculating the energy and comparing it to a tolerance of the order $\epsilon =$ 10[−]16. (iii) Furthermore, one can increase the stability of the SU algorithm by applying the gauge-fixing introduced in Appendix [B.](#page-11-0)

Let us further note that the computational cost of the SU scales as $O(pD^z)$, and evidently depends on the coordination number of the underlying lattice. Henceforth, the maximum achievable bond dimension *D* is lattice dependent and is larger for structures with less coordination number, though structures with large *z* usually need low *D* because of entanglement monogamy. For example, in the case of star lattice with $z = 3$, we managed to reach convergence for $D = 100$ on a core*i*7 PC (with four threads) in 16 hours. This time is quickly decreased on HPC clusters, where also larger bond dimension could be reached.

D. Expectation values and correlators

Once the tensors approximating a GS are found, they can be used to estimate expectation values of local operators such as local order parameters and two-point correlators. The usual procedure in iPEPS is to evaluate the effective environment surrounding some local tensors, which can be done by methods such as TRG, CTMRG, etc. These methods, however, are not easily adapted to arbitrary lattices in a systematic way. Because of this, in gPEPS we consider a simpler approach which is applicable to any graph. In this approach, we use the bond matrices λ [\[42\]](#page-13-0) (calculated during the SU optimization) in the same spirit as in one-dimensional systems [\[40,41\]](#page-13-0), i.e., we close the bond indices with the λ matrices, which is exact in one dimension, and corresponds to a mean-field approximation of the effective environment in higher dimensions. A diagrammatic representation of oneand two-site expectation values in this scheme is shown in Figs. $2(b)$ and $2(c)$. Similar approach has also been used in Refs. [\[32,42](#page-13-0)[–48\]](#page-14-0). Extension to other multisite operators and correlation functions is straightforward.

Some remarks are in order. First, due to larger bond dimension *D* which is handled in the gPEPS algorithm, λ matrices provide a better approximation to the environment of local tensors compared to conventional SU algorithms. Second, this scheme can be applied systematically, regardless of the underlying lattice. Third, we expect this scheme to work well in higher dimensions whenever the correlation length is small and the connectivity is large. And fourth, for 1*d* graphs, the gPEPS algorithm is exactly equivalent to the iTEBD algorithm and bond matrices satisfy the canonical

TABLE I. gPEPS benchmark results for the GS energy per site of several lattice models. See the main text for simulation details.

Model	Lattice	gPEPS	Previous studies
AFH	Chain	-0.44304	-0.44315 [8.49]
AFH	Star	-0.37523	-0.37523 [23]
AFH	Cubic	-0.89253	-0.904 [47]
HBH	Square	-0.30258	-0.30232 [50]
FHF	Pyrochlore	-0.80000	-0.80000
3SOP	Kagome	-4.00074	
BLBO	Triangular	2.95252	2.95254 [29]

forms [\[40,41\]](#page-13-0), whereas in higher dimensions it provides an approximation to expectation values which, though not being variational, may be remarkably accurate.

III. ENERGY BENCHMARK RESULTS

We benchmarked the gPEPS algorithm for several quantum lattice models, namely, the spin-1/2 AFH model on chain, star and cubic lattices, the HBH model on square lattice, spin-1/2 FHF model on pyrochlore lattice, as well as the 3SQP model in field on kagome and the spin-1 BLBQ Heisenberg model on the triangular lattices. Our results for the GS energy per site of these models are summarized and benchmarked against previous studies (when it was available) in Table I, where one can clearly see the excellent agreement between our results and previous findings. Detailed discussion about each model is presented in the following.

A. Antiferromagnetic Heisenberg model on 1*d* **chain**

As the first example of a lattice model, we calculate the GS energy of a 1*d* model, i.e., the spin-1/2 antiferromagnetic Heisenberg model on a chain. The Hamiltonian of the AFH model is given by

$$
H_{\text{AFH}} = J \sum_{\langle ij \rangle} \mathbf{S}_i \cdot \mathbf{S}_j,\tag{5}
$$

where the sum runs over the nearest-neighbor sites *i*, *j* of the lattice and S_i is the ordinary spin operator at site *i*. Here we consider the antiferromagnetic Heisenberg coupling $J = 1$. In order to evaluate the GS of the AFH model on a chain, we consider an infinite chain with a transitionally invariant two-site unit cell [Fig. $10(a)$] and associate a rank-3 tensor to each vertices of the chain. Figure $10(b)$ illustrates the labeling on tensors which, corresponds to graph nodes, in the unit cell. The corresponding SM of the chain is further given in Appendix [A 1.](#page-9-0)

Using this SM along with the simple update introduced in previous section, we evaluated the GS energy per site, ε_0 , of the AFH model on chain for different values of bond dimension *D*. Figure 3 demonstrates the scaling of energy versus inverse bond dimension *D* for the AFH model on 1*d* chain up to $D_{\text{Max}} = 60$. As one can see, there is a very good convergence for energies, particularly fore large *D*s (see also the inset of the figure). The lowest energy we obtained from gPEPS method is $\varepsilon_0 = -0.44304$ which is in excellent agreement with the exact analytical results $\varepsilon_0^{\text{exact}} = 1/4 - \ln(2) =$

FIG. 3. Scaling of the gPEPS ground state energy per site, ε_0 , with respect to inverse bond dimension *D* for the AFH model on 1*d* chain up to $D_{\text{Max}} = 60$. The inset shows the zooming for large bond dimensions.

−0.44314 [\[49\]](#page-14-0) and previous density matrix renormalization group (DMRG) result, $\varepsilon_0^{\text{DMRG}} = -0.44315$, of Ref. [\[8\]](#page-13-0).

As we pointed out previously, the gPEPS in 1*d* is fully equivalent to the infinite time-evolution block decimation (iTEBD) method and therefore one should obtain the exact same energy from a standard iTEBD algorithm.

B. Antiferromagnetic Heisenberg model on 2*d* **star lattice**

As the second benchmark, we use the gPEPS method to calculate the GS energy of the AFH model on the star lattice. The Hamiltonian of the AFH model on the star lattice reads [\[23](#page-13-0)[,51–54\]](#page-14-0)

$$
H_{\text{AFHS}} = J_e \sum_{\langle ij \rangle \in e} \mathbf{S}_i \cdot \mathbf{S}_j + J_t \sum_{\langle ij \rangle \in t} \mathbf{S}_i \cdot \mathbf{S}_j,\tag{6}
$$

where the first sum runs over the nearest-neighbour sites on the expanding links connecting the triangles of the lattice and the second sum runs over nearest-neighbour sites on the triangles. The SM of the star lattice for a six-site unit cell is already provided in Eq. [\(2\)](#page-2-0).

Using [\(2\)](#page-2-0), we calculated the ε_0 for the AFH model on the star lattice for $J_e = 1$, $J_t = 0.05$ up to $D_{\text{Max}} = 100$. Figure [4](#page-5-0) depict the scaling of GS energy per site for inverse of different bond dimensions. The very good convergence of energies, as well as the unprecedented large bond dimension $D_{\text{Max}} = 100$, definitely confirms the efficiency and power of the gPEPS technique for simulation of strongly correlated quantum many-body Hamiltonians.

Let us further note the our gPEPS energy, $\varepsilon_0 = -0.37523$, is in exact agreement with previous iPEPS study of the AFH model on the star lattice [\[22\]](#page-13-0).

C. Antiferromagnetic Heisenberg model on 3*d* **cubic lattice**

In order to challenge the power of gPEPS technique for 3*d* lattices, we apply it to the AFH model on the simple cubic lattice. Figure [14](#page-11-0) depicts an eight-site unit cell of the

FIG. 4. Scaling of the gPEPS ground-state energy per site, ε_0 , with respect to inverse bond dimension *D* for the AFH model on 2*d* star lattice for $J_e = 1$ and $J_t = 0.05$ up to $D_{\text{Max}} = 100$. Details of setups A, B are provided in Ref. [\[22\]](#page-13-0). The inset further shows the zooming for large bond dimensions.

cubic lattice and the corresponding labeling of vertices. The corresponding SM matrix is further given in Appendix [A 5.](#page-10-0)

Using Hamiltonian (5) and structure matrix $(A5)$, we calculated the GS energy of the AFH model on the simple cubic lattice for different bond dimensions. Figure 5 shows the scaling of energy versus inverse bond dimension up to $D_{\text{Max}} = 14$ on the cubic lattice. The results show a very good convergence of the gPEPS energies to $\varepsilon_0 = -0.89253$ which is in close agreement with the results of Ref. [\[47\]](#page-13-0) with $\varepsilon_0 = -0.904$. Our findings once again confirms how the idea of SM can simplify the implementation of TN methods to 3*d* lattice models.

D. Hardcore Bose-Hubbard model on 2*d* **square lattice**

In this section, we test our gPEPS algorithm for another lattice model, i.e., the hardcore Bose-Hubbard model on the square lattice. Figures $11(a)$ and $11(b)$ demonstrate the square lattice and the four-site unit cell that we used for our simulation. Hamiltonian of the HBH model further reads

$$
H_{\rm HBH} = -J \sum_{\langle ij \rangle} (a_i^{\dagger} a_j + a_j^{\dagger} a_i) - \mu \sum_i \hat{n}_i, \tag{7}
$$

where the first hopping term is on the nearest-neighbor vertices of the square lattice and the second sum is an on-site chemical potential. Here we set $J = 1$. *a* and a^{\dagger} are bosonic annihilation and creation operators. The SM of the square lattice which is required for the gPEPS simulation is further provided in Appendix [A 2.](#page-9-0)

Figure [6](#page-6-0) demonstrate our findings for the GS energy of the HBH model for $\mu = -2$ for different bond dimensions up to $D_{\text{Max}} = 14$. The convergence at large *Ds* are quite good and the GS energy per site of the system for $D = 14$ is $\varepsilon_0 = -0.30258$ which is even lower than previous iPEPS results of Ref. [\[50\]](#page-14-0) with $\varepsilon_0^{\text{IPEPS}} = -0.30232$.

FIG. 5. Scaling of the gPEPS ground-state energy per site, ε_0 , with respect to inverse bond dimension *D* for the AFH model on 3*d* cubic lattice up to $D_{\text{Max}} = 14$. The inset shows the zooming for large bond dimensions.

E. Spin-1 bilinear-biquadratic Heisenberg model on 2*d* **triangular lattice**

As another example for benchmarking the gPEPS method, we studied the spin-1 bilinear-biquadratic Heisenberg model on 2*d* triangular lattice [Fig. [12\(a\)\]](#page-11-0). This model has already been studied in detail in Ref. [\[29\]](#page-13-0) with iPEPS method and the full phase diagram of the system has already been investigated. The iPEPS machinery for triangular lattice is performed by mapping it to square lattice with both nearest and nextnearest-neighbor interactions.

Here instead, we study the model by means of gPEPS technique on an infinite triangular lattice with nine-site unit cell [see Fig. $12(b)$]. In the gPEPS framework, all of the interactions are between nearest-neighbour vertices and simulation for larger bond dimensions is also possible.

Hamiltonian of the spin-1 BLBQ model according to the convention of Ref. [\[29\]](#page-13-0) reads

$$
H_{\text{BLBQ}} = \cos(\theta) \sum_{\langle ij \rangle} \mathbf{S}_i \cdot \mathbf{S}_j + \sin(\theta) \sum_{\langle ij \rangle} (\mathbf{S}_i \cdot \mathbf{S}_j)^2, \qquad (8)
$$

where both sums run on nearest neighbours. The first sum, however, is the bilinear term which is nothing but the standard Heisenberg model and the second term is the biquadratic term.

In order to benchmark the gPEPS results with previous studies, we calculate the GS of the system for $\theta = 1.5865$. This point is very close to $\theta = \frac{\pi}{2}$. However, since $\theta = \frac{\pi}{2}$ is a phase boundary in the phase diagram of the BLBQ model on the triangular lattice [\[29\]](#page-13-0), we chose a slightly different point to evaluate the GS of the system to show how the gPEPS can converge to the true GS of the system.

Using Hamiltonian (8) and the SM of the triangular lattice presented in Appendix [A 3,](#page-9-0) we were able to reproduce the results of Ref. [\[29\]](#page-13-0) with very high accuracy. Figure [7](#page-6-0) depicts the scaling of the gPEPS GS energy per site, ε_0 , with respect to inverse bond dimension *D* for the BLBQ model for $\theta = 1.5865$. As one can clearly see, the convergence of the algorithm is quite notable even at small bond dimensions

FIG. 6. Scaling of the gPEPS ground-state energy per site, ε_0 , with respect to inverse bond dimension *D* for the HBH model on 2*d* square lattice for $\mu = -2$ up to $D_{\text{Max}} = 14$. The inset shows the zooming for large bond dimensions.

and our gPEPS energy $\varepsilon_0 = 2.95252$ is almost the same as $\varepsilon_0^{\text{iPEPS}} = 2.95253$ of the Ref. [\[29\]](#page-13-0).

F. Three-state quantum Potts model in field on 2*d* **kagome lattice**

Here we present our gPEPS results for the three-state Potts model in field on the kagome lattice which, to the best of our knowledge, is the first TN implementation of this model. Generic Hamiltonian of the *q*-state Potts model, also known as vector Potts model, in the presence of field reads [\[55\]](#page-14-0)

$$
H_{\text{Potts}} = -J \sum_{\langle ij \rangle} U_i U_j^{\dagger} - \Gamma \sum_i V_i + \text{H.c.}, \tag{9}
$$

where

$$
U = \text{diag}(1, \omega, \omega^2, \dots, \omega^{q-1}), \quad \omega = e^{\frac{2\pi i}{q}}, \qquad (10)
$$

and

$$
V = \begin{pmatrix} 0 & I_{q-1} \\ 1 & 0 \end{pmatrix} , \qquad (11)
$$

where I_{q-1} is a $(q-1)\times(q-1)$ identity matrix. By setting $q = 3$ in the above relations, Hamiltonian of the 3SQP is obtained. We then apply Hamiltonian (9) to a kagome lattice with a twelve-site unit cell (Fig. [13\)](#page-11-0). The corresponding SM of the kagome unit cell is given in Appendix \overline{A} 4.

We have calculated the ε_0 for the 3SQP model in field on the kagome lattice with the gPEPS method up to $D_{\text{Max}} = 30$. The GS energy of the system at this point is exact and for all bond dimensions *D* for finite field value $\Gamma = 0.1$ is equal to $\varepsilon_0 = -4.00074.$

G. Ferromagnetic Heisenberg model in magnetic field on 3*d* **pyrochlore lattice**

In order to challenge the gPEPS algorithm with a nontrivial 3*d* lattice, we applied it to one of the most complicated structures, i.e., the pyrochlore lattice and studied the FHF model on

FIG. 7. Scaling of the gPEPS ground-state energy per site, ε_0 , with respect to inverse bond dimension *D* for the BLBQ model on 2*d* triangular lattice for $\theta = 1.5865$ up to $D_{\text{Max}} = 14$. The inset shows the zooming for large bond dimensions.

this lattice. We stress that, to the best of our knowledge, this is the first application of TN methods to the pyrochlore lattice.

Hamiltonian of the FHF model is given by

$$
H_{\text{FHF}} = -J \sum_{\langle ij \rangle} \mathbf{S}_i \cdot \mathbf{S}_j - h \sum_i \mathbf{S}_z,\tag{12}
$$

where the first sum is on nearest-neighbor sites and the second sum runs over all of the vertices of the lattice. Here we set $J = 1$. We apply Hamiltonian (12) to the pyrochlore lattice [Fig. $15(a)$] with an eight-site unit cell [Fig. $15(b)$]. The corresponding SM of the pyrochlore lattice is given in Appendix [A 6.](#page-10-0)

The FHF model on pyrochlore lattice has an exact meanfield ground state with energy

$$
\varepsilon_0^{\text{exact}} = \frac{1}{N_s} \left(-\frac{N_s h}{2} - \frac{N_b J}{2} \right),\tag{13}
$$

which is a state with $D = 1$ and thus no correlations. It is simply the state with all spins aligned in *z* direction. In the above relation, N_s is the number of lattice sites and N_b is the number of nearest-neighbor bonds of pyrochlore lattice. For a translationally invariant unit cell of the pyrochlore lattice with 8 sites and 24 bonds, such as the one depicted in Fig. [15\(b\),](#page-11-0) GS energy per site of the system for $h = 0.1$ is $\varepsilon_0^{\text{exact}} = -0.8$.

Our gPEPS results for the GS energy of the FHF model for $h = 0.1$ is in exact agreement with the mean-field results and we obtained $\varepsilon_0 = -0.80000$ for different bond dimensions *D* up to $D_{\text{Max}} = 14$. Although the ground state is a $D = 1$ product state, we pushed the simulation to large *D* to show the algorithm is stable up to large bond dimensions even for states with zero entanglement. This once again certifies that the gPEPS technique is a powerful universal TN method for simulation of lattice Hamiltonians on the exotic lattice structures.

FIG. 8. (a) The GS energy per site and magnetization m_z of the ITF model with respect to field strength *h* for the gPEPS method (*D* = 6), compared with the iPEPS+CTMRG ($D = 6$, $\chi = 80$) on a 2×2 unit cell. The inset in the energy plot demonstrates the gPEPS relative error with respect to the iPEPS energies.

IV. QUANTUM PHASE TRANSITION WITH gPEPS

Next, we challenged the gPEPS technique for systematic study of QPT in quantum lattice models on different structures. In what follows, we present our results for phase diagram of the ITF model on square lattice which is an standard benchmark model for any new algorithm and then we investigate the nontrivial phase diagram of the Kitaev-Heisenberg model on the hyperhoneycomb lattice which is one of the most complicated structures for any numerical technique.

A. Quantum Ising model in transverse magnetic field on 2*d* **square lattice**

Using the gPEPS algorithm, we studied the zerotemperature phase diagram of the ITF model on a square lattice. In particular, we studied the following Hamiltonian:

$$
H_{\text{ITF}} = -J \sum_{\langle ij \rangle} \sigma_i^z \sigma_j^z - h \sum_i \sigma_i^x, \tag{14}
$$

where the first sum runs over nearest-neighbor sites and the second one runs over the vertices of the square lattice. By measuring the GS energy and magnetization along *x* and *z* directions, we pinpointed the QPT point at $h^c \approx 3.04$ which is in perfect agreement with previous studies [\[15,25,27](#page-13-0)[,50\]](#page-14-0). Figures $8(a)$ and $8(b)$ show the GS energy per site as well as the magnetization of the ITF model, respectively. The QPT is best captured by discontinuities in the magnetization and energy plots.

The gPEPS relative error with respect to the iPEPS energies in the inset of Fig. $8(a)$ is of the order 10^{-16} everywhere except at the vicinity of the transition point which is increased to 10^{-3} . This is best explained by the fact that at the critical point, the correlation length diverges and the mean-field environment does not necessarily provide the best approximation of the iPEPS environment. Nonetheless, the gPEPS still captures the QPT with very high accuracy in most cases. The same order of errors holds between the meanfield and CTMRG magnetization curves of Fig. 8(b) as well (not shown in the figure).

B. Kitaev-Heisenberg model on 3*d* **hyperhoneycomb lattice**

Spin-orbit entangled Mott insulators in iridates [\[56,57\]](#page-14-0) can realize instances of 2*d* [\[58,59\]](#page-14-0) and 3*d* [\[60,61\]](#page-14-0) arrangements of tricoordinated lattices with Kitaev interactions. In particular, it has been shown that the polymorph β -Li₂IrO₃ realize threedimensional arrangements of the spin-orbit tangled moments which retain the hyperhoneycomb lattice [\[61\]](#page-14-0). In this material, the Ir^{4+} ions arrange in a hyperhoneycomb structure and the combined effect of spin-orbit coupling, Coulomb interaction, and exchange geometry generates $J_{\text{eff}} = 1/2$ moments subject to a combination of anisotropic Kitaev and Heisenberg interactions [\[56,57\]](#page-14-0). In 2*d*, it has been shown that the resulting Kitaev-Heisenberg (KH) model host various phases ranging from quantum spin-liquid (QSL) to magnetically ordered phases such as ferromagnetic (FM), antiferomagnetic (AFM), stripy, and zigzag on the honeycomb lattice [\[58,59\]](#page-14-0).

Recent studies based on mean-field theory [\[62\]](#page-14-0) and TN on Bethe lattice [\[63\]](#page-14-0) have also predicted similar phases for the KH model on the hyperhoneycomb lattice. However, a systematic study of the full phase digram of the model on the original hyperhoneycomb lattice in the thermodynamic limit is still missing. Thanks to the gPEPS technique, we were able to apply, for the first time, the TN method, directly to a translationally invariant unit cell of the hyperhoneycomb lattice (see Fig. [16\)](#page-12-0) and map out the phase diagram of the KH model on the full parameter space. More specifically, we applied the gPEPS to the following KH Hamiltonian

$$
H_{\rm KH} = 2\cos(\theta) \sum_{\alpha - \text{link}} S_i^{\alpha} S_j^{\alpha} + \sin(\theta) \sum_{\langle ij \rangle} \mathbf{S}_i \cdot \mathbf{S}_j, \qquad (15)
$$

FIG. 9. (Left) Full phase diagram of the KH model on the hyperhoneycomb lattice. (Right) GS energy per site, ε_0 , and magnetic order parameter *M* for $\theta = [0, 2\pi]$. Details of the phase diagram can be found in the main text.

where the first sum is the Kitaev term with $(\alpha = x, y, z)$ and the second term is the Heisenberg interaction acting on the nearest-neighbor sites of the hyperhoneycomb lattice.

In order to capture the phase boundaries and characterize the nature of underlying phases, we calculated the GS energy, entanglement entropy, magnetization, ground-state fidelity and two-site spin-spin correlators in the full parameter space $\theta = [0, 2\pi]$. Figure 9(left) demonstrates the phase diagram of the KH model on the hyperhoneycomb lattice. The phase diagram is composed of four magnetically ordered phases i.e., FM, AFM, zigzag and stripy and two QSL phase at the vicinity of the FM and AFM Kitaev couplings. Orientation of spins in each magnetic phase has also been shown in the figure. detailed discussion regarding the QSL phase at the pure Kitaev points can be found in Ref. [\[64\]](#page-14-0). Figure 9(right) further show the GS energy per site as well as the magnetic order parameter, $M = \sqrt{\langle S^x \rangle^2 + \langle S^y \rangle^2 + \langle S^z \rangle^2}$. One can clearly see that *M* precisely detect the phase boundaries and distinguishes magnetic phases from QSL phases with no local order parameter.

We refer the interested reader to Ref. [\[65\]](#page-14-0) for further details regarding this model and the phase diagram of KH model on other 3*d* tricoordinated lattices [\[64\]](#page-14-0). Let us further stress that our findings are in excellent agreement with previous studies [\[62,63\]](#page-14-0)

V. CONCLUSIONS AND DISCUSSION

In this paper, we introduced the concept of structure matrix which encodes the connectivity information of a given tensor network and developed a generic graph-based projected entangled-pair state algorithm for local Hamiltonians of quantum lattice models that can be applied to any lattice in any dimension in the thermodynamic limit. Our approach relies on the simple update algorithm for imaginary-time evolution and a mean-field-like approximation to effective environments. Though not being variational, the scheme produces accurate

results in most situations and is capable of handling large bond dimensions such as $D \sim 100$.

We benchmarked our method with several quantum lattice models on different structures in one, two and three dimensional lattices. Our method facilitates the applicability of iPEPS algorothms to complex lattices in 2*d* and 3*d*. Most importantly, it also opens the possibility to simulate quantum materials on complex crystallographic structures via tensornetwork methods. The gPEPS method can further be extended to deal with fermionic systems and symmetric tensor networks, as well as finite temperature.

Let us further remark that the gPEPS ground-state tensors of all infinite 2*d* systems can additionally be contracted by using TRG [\[26,27\]](#page-13-0), boundary MPS [\[14\]](#page-13-0), or CTMRG [\[15,24,28\]](#page-13-0) both directly or rather by grouping several adjacent tensors into a coarse-grained square lattice of block-sites in order to obtain variational energies. Unfortunately extension of these ideas to generic 3*d* structures is not straightforward. For example, the CTMRG has only been extended to simple cubic lattice and other 3*d* lattices are left behind. A new generic technique for contracting infinite lattices both in 2*d* and 3*d* is currently under development by our group [\[66\]](#page-14-0) which can be used as a supplement to gPEPS method for doing variational optimization with TN on any infinite graph. With this new approach we will be able to do full update within the gPEPS framework.

It is worth noting that extension of TN methods to generic lattices can alternatively be done by using Husimi lattices [\[48\]](#page-14-0) which are obtained from a Bethe lattice in which every vertex is replaced by a *p* polygon [\[67,68\]](#page-14-0). Nevertheless, one must note that the physics obtained on the Husimi lattice might be different from the one on the original lattice. This is mainly due to the slight differences between a lattice and its Husimi counterpart. For example, a Husimi lattice might not create the same closed loop structure as the original lattice. This is important particularly for those models in which closed loops of the lattice play key roles in the physics of the system. For example, it is already known that the closed loops of the Kitaev model act as integrals of motion which carries zero fluxes in the system [\[69\]](#page-14-0). This once again shows the significance of gPEPS in studying generic infinite lattice with TN methods.

As last remark, let us point out that although gPEPS technique produces reliable and accurate results for many quantum lattice models in different dimensions, applications of the method to frustrated system should be handled with care. Due to the longer range of correlations which might exist in the GS of frustrated systems such as some spin-liquid states, the role of environment around local GS tensors becomes very important, and the bond matrices λ which are used in gPEPS method as mean-field environment for calculation of the expectation values might not provide the best approximation to the environment. One might therefore obtain higher or unexpectedly lower values for the GS energies of the system and expectation values. For example the gPEPS method fails in producing accurate results for the AFH model on the kagome lattice and the best TN results so far, belongs to the projected entangled-simplex state [\[70,71\]](#page-14-0). It is therefore advised that the gPEPS energies for frustrated system be benchmarked against other methods to make sure the correct results are obtained.

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FIG. 10. (a) The infinite 1*d* spin chain with a two-site unit cell (blue region). (b) Labeling of vertices (graph nodes) in the unit cell.

APPENDIX A: STRUCTURE MATRIX FOR VARIOUS LATTICE STRUCTURES

In this Appendix, we present structure matrix of various widely used 2*d* and 3*d* lattices. The main strategy to construct the SM corresponding to a given infinite lattice with translational invariance is to first define a unit cell of the lattice with desired number of vertices and periodic boundary condition and then constructing the incidence matrix (IM) of the unit cell. The SM can then be obtained straightforwardly from the IM of the lattice. The IM of arbitrary graphs can be obtained by using efficient graph libraries of MATLAB, PYTHON, MATHEMATICA, or other desired languages.

1. 1*d* **chain**

Equation (A1) corresponds to the SM of an infinite 1*d* spin chain with a two-site unit cell (see Fig. 10).

$$
SM_{\text{chain}} = \begin{pmatrix} E_1 & E_2 \\ \overline{T_1} & 2 & 3 \\ \overline{T_2} & 2 & 3 \end{pmatrix} . \tag{A1}
$$

2. 2*d* **square lattice**

Equation (A2) corresponds to the SM of an infinite 2*d* square lattice with a four-site unit cell (see Fig. [11\)](#page-11-0).

3. 2*d* **triangular lattice**

Equation (A3) corresponds to the SM of an infinite 2*d* triangular lattice with a nine-site unit cell (see Fig. [12\)](#page-11-0).

4. 2*d* **kagome lattice**

Equation (A4) corresponds to the SM of an infinite 2*d* kagome lattice with a 12-site unit cell (see Fig. [13\)](#page-11-0).

		E_1	E_{2}	E ₃	E_4	E_5	E_6	E_7	E_8	E_9	E_{10}				E_{11} E_{12} E_{13} E_{14} E_{15}		E_{16} E_{17}		E_{18}	E_{19}	E_{20}	E_{21}	E_{22}	E_{23}	E_{24}
	T_1	2	3	4						$_{0}$	Ω	0	θ	0	Ω	0	$_{0}$	0	0	Ω	0	Ω	Ω	Ω	0
	T_{2}	Ω	Ω	0	0	າ	3	4		0	θ	$\mathbf{0}$	$\overline{0}$	θ	θ	Ω	θ	0	Ω	$\overline{0}$	0	$\mathbf{0}$	θ	θ	0
	T_3	0	θ	0	θ		0	0	θ	3	4	5	θ	θ	θ	$\mathbf{0}$	$\mathbf{0}$	$\mathbf{0}$	$\mathbf{0}$	$\mathbf{0}$	$\mathbf{0}$	$\mathbf{0}$	θ	θ	Ω
	T_{4}	2	$\mathbf{0}$	0	0	$\mathbf{0}$	Ω	0	0	3	θ	$\mathbf{0}$	4	5	θ	$\mathbf{0}$	$\overline{0}$	$\mathbf{0}$	$\mathbf{0}$	$\mathbf{0}$	θ	$\overline{0}$	$\mathbf{0}$	$\overline{0}$	Ω
	T_{5}	0	2	θ	Ω	Ω	Ω	0	0	0	$\mathbf{0}$	$\mathbf{0}$	3	θ	4	5	Ω	Ω	Ω	$\mathbf{0}$	$\mathbf{0}$	$\mathbf{0}$	θ	θ	Ω
$SM_{\text{kagome}} =$	T_6	Ω	Ω	0		Ω			0	0	3	Ω	$\overline{0}$	θ	4	$\mathbf{0}$	5	$\mathbf{0}$	Ω	$\mathbf{0}$	Ω	$\mathbf{0}$	Ω	θ	Ω
	T ₇	0	$\overline{0}$	0	0	Ω	Ω	Ω	Ω	$\mathbf{0}$	θ	2	$\overline{0}$	3	θ	$\overline{0}$	$\mathbf{0}$	4	5	$\mathbf{0}$	θ	$\overline{0}$	$\mathbf{0}$	$\overline{0}$	Ω
	T_8	Ω	Ω	Ω	0	Ω	0	0	0	Ω	θ	$\mathbf{0}$	$\overline{0}$	θ	Ω	2	3	Ω	Ω	4	5	θ	θ	$\overline{0}$	θ
	$T_{\rm o}$	0	θ		0	Ω	θ	0	0	Ω	θ	Ω	$\overline{0}$	θ	Ω	Ω	Ω	$\mathbf{0}$	$\mathbf{0}$	3	Ω	4	5	θ	0
	T_{10}	Ω	Ω	0		Ω	0	0	0	Ω	$\mathbf{0}$	$\mathbf{0}$	$\overline{0}$	θ	θ	$\overline{0}$	Ω	3	θ	$\mathbf{0}$	Ω	4	Ω	5	Ω
	T_{11}	Ω	$\overline{0}$	0	0	Ω	0			Ω	$\overline{0}$	Ω	$\overline{0}$	θ	θ	Ω	Ω	Ω	3	$\overline{0}$	Ω	Ω	$\mathbf{0}$	4	
	$\sqrt{T_{12}}$	$\mathbf{0}$	$\overline{0}$	0	Ω	θ	Ω	0	2	Ω	$\overline{0}$	$\mathbf{0}$	$\boldsymbol{0}$	θ	θ	$\overline{0}$	Ω	$\mathbf{0}$	Ω	$\mathbf{0}$	3	$\mathbf{0}$	4	θ	5.
																									(A4)

5. 3*d* **cubic lattice**

Equation (A5) corresponds to the SM of an 3*d* cubic lattice with a eight-site unit cell. (See Fig. [14.](#page-11-0))

6. 3*d* **pyrochlore lattice**

Equation (A6) corresponds to the SM of an 3*d* pyrochlore lattice with a eight-site unit cell. (See Fig. [15.](#page-11-0))

7. 3*d* **hyperhoneycomb lattice**

Equation (A7) corresponds to the SM of an 3*d* hyperhoneycomb lattice with a four-site unit cell. (See Fig. [16.](#page-12-0))

$$
SM_{\text{hyperhoney}} = \begin{pmatrix} E_1 & E_2 & E_3 & E_4 & E_5 & E_6 \\ \hline T_1 & 2 & 3 & 4 & 0 & 0 & 0 \\ T_2 & 2 & 0 & 0 & 3 & 4 & 0 \\ T_3 & 0 & 0 & 0 & 2 & 3 & 4 \\ T_4 & 0 & 2 & 3 & 0 & 0 & 4 \end{pmatrix} . \tag{A7}
$$

FIG. 11. (a) The infinite 2*d* square lattice with a four-site unit cell (blue region). (b) Labeling of vertices (graph nodes) in the unit cell.

APPENDIX B: GAUGE-FIXING FOR GPEPS

In this section, we show how to locally fix the gauge degrees of freedom on the virtual bonds of the gPEPS TN. This can substantially improve the algorithm by stabilizing the ITE optimization and results in faster convergence of the ITE iteration and more accurate estimation of expectation values and correlators. To this end, we first introduce the boundary matrices for each link of the TN: consider a virtual bond of a TN shared between tensor *A*, *B* and their corresponding λ matrices, such as the one shown in Fig. [17\(a\),](#page-12-0) the left and right boundary matrices are defined as [see also Fig. [17\(b\)\]](#page-12-0)

$$
(M_L)^i_{i'} = \sum_{p,j,k,p',j',k'} A^p_{i,j,k} \overline{A}^{p'}_{i',j',k'} \lambda^2_{j,j'} \lambda^2_{kk'},
$$

$$
(M_R)^j_{j'} = \sum_{p,i,k,p',i',k'} B^p_{i,j,k} \overline{B}^{p'}_{i',j',k'} \lambda^2_{ii'} \lambda^2_{kk'}.
$$
 (B1)

We choose the gauge degrees of freedom such that a Schmidt form is imposed on all virtual degrees of freedom on the TN network. this involves choosing the gauge such that (i) the M_L and M_R boundary matrices represent an orthonormal basis, i.e., $M_{i'}^i = \lambda_i^2 \delta_{i,i'}$, and (ii) the bond matrices λ are diagonal, normalized and positive, $\lambda_{i,j} = \delta_{i,j} s_i$ with s_i the Schmidt coefficients, which are ordered $s_i \geq s_{i+1}$. A *canonical form* for the tensor network is defined by requiring that every virtual bond is in Schmidt form [\[43,](#page-13-0)[72\]](#page-14-0).

We now present a method to fix the gauge degrees of freedom on any virtual link of a given network. Note that under change of the gauge all local tensors associated to a link are altered i.e., $A \rightarrow \tilde{A}$, $B \rightarrow \tilde{B}$, and the shared lambda matrix $\lambda \rightarrow \tilde{\lambda}$. Correspondingly, after applying the gauge-fixing to all virtual bonds of the TN the iPEPS wave function is altered as well [see Fig. $18(a)$].

In order to identifying the gauge change matrices *x* and *y* (and their inverses) we first calculate the boundary matrices M_L and M_R and then diagonalize them such that

$$
M_L = u_L d_L u_L^{\dagger},
$$

$$
M_R = u_R d_R u_R^{\dagger},
$$
 (B2)

see Fig. $17(b)$, with unitary matrices u_L , u_R and real diagonal matrices d_L , d_R . Notice that, due to the positivity of the boundary matrices M_L and M_R , it follows that d_L and d_R are

FIG. 12. (a) The infinite 2*d* triangular lattice with a nine-site unit cell (blue region). (b) Labelling of vertices (graph nodes) in the unit cell.

FIG. 13. (a) The infinite 2*d* kagome lattice with a 12-site unit cell (blue region). (b) Labeling of vertices (graph nodes) in the unit cell.

FIG. 14. The infinite 3*d* cubic lattice with a eight-site unit cell. The numbers at vertices label the graph nodes in the unit cell.

FIG. 15. (a) The infinite 3*d* pyrochlore lattice composed of up and down tetrahedrons. (b) The eight-site unit cell of the pyrochlore lattice. The numbers represent the labeling of vertices (graph nodes) in the unit cell.

FIG. 16. (a) The infinite 3*d* hyperhoneycomb lattice. (b) The four-site unit cell of the hyperhoneycomb lattice. The numbers represent the labeling of vertices (graph nodes) in the unit cell.

non-negative, thus possess real roots $\sqrt{d_L}$ and $\sqrt{d_R}$. We now use these to transform the bond matrix λ ,

for unitary w_L , w_R and positive diagonal $\tilde{\lambda}$. The gauge change matrices *x* and *y* are now defined as
 $x \equiv w_L^{\dagger} \sqrt{d_L} u_L^{\dagger}$

$$
x \equiv w_L^{\dagger} \sqrt{d_L} u_L^{\dagger},
$$

$$
y \equiv u_R \sqrt{d_R} w_R.
$$
 (B5)

$$
\lambda' \equiv \sqrt{d_L} u_L^{\dagger} \lambda u_R \sqrt{d_R}, \tag{B3}
$$

and take the singular value decomposition to obtain

λ

$$
\lambda' = w_L \tilde{\lambda} w_R^{\dagger} \tag{B4}
$$

This process is further depicted in Figs. $17(c)$ and $17(d)$. One should note that $M_L = x^{\dagger} x$ and $M_R = yy^{\dagger}$. Under this choice of gauge the new bond matrix is simply the $\tilde{\lambda}$ from Eq. (B4) or equivalently

$$
\tilde{\lambda} = x\lambda y,\tag{B6}
$$

FIG. 17. (a) Local state $|\psi\rangle$ composed of tensors *A*, *B* and their relevant λ matrices as effective mean-field environment. (b) Boundary matrices M_L and M_R and their eigendecompositions. (c) Definition of modified bond matrix λ' , which is then decomposed via the SVD. (d) Definition of the gauge change matrices x and y that transform the initial state $|\psi\rangle$ to its gauge related state $|\tilde{\psi}\rangle$.

FIG. 18. (a) Gauge related initial and final states. (b) A change of gauge, which leaves the state $|\psi\rangle$ invariant, is enacted on the index between *A* and *B* via matrices *x* and *y* together with their inverses.

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which is positive and diagonal by construction. Furthermore the new left and right tensors read

$$
\tilde{A} = Ax^{-1},
$$

\n
$$
\tilde{B} = y^{-1}B,
$$
\n(B7)

see also Fig. 18(b). Once the gauge is fixed on all virtual legs of the *A*, *B* tensors the Schmidt form or orthonormality are satisfied when the eigenvalues in d_L , d_R are uniformly distributed, i.e., all the diagonal elements are equals to 1 and $M^i_{i'} - \lambda^2_i \delta_{i,i'} = 0$ for both left and right boundary matrices.

In Ref. [43]. this process of gauge-fixing is alternatively dubbed as superorthogonality and is also equivalent as doing high-order SVD on local tensors. In order to bring all of the tensors in a TN into a superorthogonal form, one can iteratively do the above process or rather incorporate it into the simple-update optimization and fix the gauge on all tensors before every step of ITE. We refer the interested reader for detailed discussion on this subject to Refs. [43[,72\]](#page-14-0).

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