

## Pressure-dependent intermediate valence behavior in YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub>

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We report a comprehensive structural and valence study of the intermediate valent materials YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub> under pressures up to 60 GPa. YbNiGa<sub>4</sub> undergoes a smooth volume contraction and shows steady increase in Yb valence with pressure, though the Yb valence reaches saturation around 25 GPa. In YbNiIn<sub>4</sub>, a change in pressure dependence of the volume and a peak in Yb valence suggest that a pressure-induced electronic topological transition occurs around 10–14 GPa. In the pressure region where YbNiIn<sub>4</sub> and YbNiGa<sub>4</sub> possess similar Yb-Yb spacings, the Yb valence reveals a precipitous drop. This drop is not captured by density functional theory calculations and implies that both the lattice degrees of freedom and the chemical environment play an important role in establishing the valence of Yb.

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### I. INTRODUCTION

Strongly correlated rare-earth materials have been heavily studied due to the exotic physical properties that they exhibit. Many of the rare earths form compounds in intermediate valence states, which will naturally dictate the magnetic properties of these materials and which can readily be tuned via the application of pressure. While in Ce compounds pressure generally favors a nonmagnetic Ce<sup>4+</sup> state, in Yb compounds pressure favors the magnetic Yb<sup>3+</sup> state [1]. The strong electron correlation in rare-earth bearing materials originates from 4*f* electrons that at normal conditions are localized on the atom. However, if the rare-earth atoms are sufficiently close, due to application of pressure or within a suitable crystal structure, they can interact, displaying behavior that adopts delocalized character. Thus, by choosing appropriate rare-earth compounds, intermediate valence behavior (degree of localization) can be achieved and tuned via the application of pressure. Among Yb-based compounds, one of the only known superconductors, β-YbAlB<sub>4</sub>, shows valence fluctuations at 20 K with an effective valence of  $n = 2.75$ , which has sparked interest in better understanding the relationship between intermediate valence behavior, quantum criticality, and superconductivity [2,3]. The effect of the Yb-valence state on magnetic properties has also been studied in YbNi<sub>3</sub>Ga<sub>9</sub>, which forms a nonmagnetic state at low temperature with an effective valence of  $n = 2.6$  under ambient conditions [4]. With the application of pressure, the valence is increased to 2.9, allowing a magnetic ground state to develop.

Because the rare-earth valence plays such a crucial role in determining magnetic properties, it is imperative to understand both its cause and how to tailor materials to exhibit the desired valence configuration. With this goal in mind, we

have performed density functional theory (DFT) calculations as well as a comprehensive structural and valence study of the orthorhombic YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub> system under pressure at room temperature. Previous work reported a valence of  $n = 2.40$  in YbNiIn<sub>4</sub>,  $n = 2.48$  in YbNiGa<sub>4</sub>, and  $n = 2.66$  in YbNiAl<sub>4</sub>, showing a general trend of increasing Yb valence with decreasing size of the group IIIb ions [5–7]. The Yb valence in YbNiGa<sub>4</sub> was measured up to a pressure of 25 GPa, revealing a steady increase in valence up to a maximum value of  $n = 2.7$  [7].

Our DFT calculations are consistent with the overall trend of the Yb valence at ambient pressure in this system, but suggest a stronger dependence on interatomic spacing. In order to determine if the valence is a simple function of interatomic spacing, we have determined the Yb valence in YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub> under pressures up to 40 GPa. Rather than being a continuous function of atomic volume or lattice spacing, we find that the Yb valence is sensitive to both the lattice degrees of freedom and the chemical environment. Density functional theory does well to reproduce the pressure dependence of the In compound up to a pressure that generates lattice spacings comparable to those of the Ga variant at ambient pressure. However, the substitution of Ga for In results in a precipitous drop in valence at fixed lattice size, an effect not captured by our DFT calculations and one that implies a more prominent role for the 4*f* hybridization with specific *p* states than might be conventionally expected.

### II. EXPERIMENTAL AND THEORETICAL METHODS

Polycrystalline samples with nominal compositions of YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub> were grown via arc melting in argon atmosphere. Due to the low boiling temperature of ytterbium,

a 5% excess of ytterbium was necessary to account for the mass loss during melting. Each sample was melted and flipped six times. Samples were subsequently annealed at 625 °C for ten days and powder x-ray-diffraction (PXRD) measurements were performed both before and after the annealing procedure. There was no detectable mass loss during the annealing process. Powder x-ray-diffraction analysis indicated phase purities greater than 96% for YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub>, with YbGa<sub>2</sub> and YbIn<sub>3</sub> being the main impurity phases, along with less than 1% of Yb<sub>2</sub>O<sub>3</sub>. The Yb compounds order in the *Cmcm* structure, with Yb and Ni occupying the 4*c* Wyckoff position and the Ga or In atoms occupying the 4*a*, 4*c*, and 8*f* Wyckoff positions. The only intermediate composition we successfully synthesized was YbNiGa<sub>3</sub>In, with a phase purity of 93%. Refinement of YbNiGa<sub>3</sub>In suggests that indium has a strong site preference and almost exclusively occupies the 4*a* site [8]. Once this site is fully occupied, the pseudobinary alloy range appears to be truncated, evidenced by nominal compositions of YbNiGaIn<sub>3</sub> and YbNiGa<sub>2</sub>In<sub>2</sub> not yielding a significant phase fraction of the desired 114 phase. Previous reports suggest YbNiAl<sub>4</sub> can be grown via similar methods, but attempts via (1) arc melting, (2) tetra-arc melting, (3) induction melting, and (4) Al-flux growth with a variety of starting compositions and annealing procedures failed to provide specimens of satisfactory quality [9]. In the cases of (1) and (2), Yb<sub>3</sub>Ni<sub>5</sub>Al<sub>19</sub> was the dominant impurity phase, typically 15%–25%, with less than 3% YbAl<sub>3</sub>. Annealing did not improve phase purity and in some instances increased the 3-5-19 phase. Attempts to grow YbNiAl<sub>4</sub> via (3) induction melting as well as attempts to anneal ingots from (1) in Yb atmosphere resulted in a series of new peaks appearing, which are comparable in intensity to the 114 peaks and have not been successfully indexed [8]. In the case of (4) we grew only single crystals of Yb<sub>3</sub>Ni<sub>5</sub>Al<sub>19</sub>, consistent with previous results [10].

Ambient pressure x-ray-diffraction (XRD) patterns were collected with a standard D8 diffractometer and were used to determine the lattice parameters and atomic positions. Refinement of the ambient pressure data suggests all sites are within 2% of being fully occupied. Pressure-dependent XRD studies were performed at sector 16-BMD of the Advanced Photon Source (APS) using a 30-keV x-ray beam. Powdered samples were loaded into a diamond anvil cell (DAC) with a rhenium gasket and pressurized with neon. The pressure was determined via copper powder mixed with the samples and confirmed with ruby spectroscopy at select pressures [11]. The XRD patterns were collected via an area detector and converted to powder patterns using FIT2D [12]. A CeO<sub>2</sub> sample was used as a calibrant to determine the instrument parameters used in the fitting, which was performed using GSAS-I [13,14]. The instrument parameters and atomic positions were then held constant for all subsequent refinements; only lattice parameters, broadening due to strain, and preferred orientation parameters were allowed to vary under pressure. Because of peak broadening that occurs under pressure, some peaks merge and become difficult to index under pressure. If merging peaks prevent an adequate fit, they are removed for the pressure space in which they overlap. In instances where including and excluding the peaks over various pressure ranges cause inconsistencies in the lattice parameters, they are removed for all XRD spectra.

X-ray absorption spectroscopy (XAS) at the ytterbium *L*<sub>III</sub> edge was performed at sector 16-IDD of the APS using partial fluorescence yield (PFY) from the *L*<sub>α</sub> emission line. The incident energy was scanned with a Si (111) fixed-exit, double-crystal monochromator and the *L*<sub>α</sub> emission was recorded using three Si (400) analyzers. Samples were loaded into a DAC with a beryllium gasket for XAS measurements and mineral oil was used as a pressure transmitting medium. Pressure was measured via ruby-fluorescence spectroscopy. For both diffraction and spectroscopy measurements, pressure was increased using a gas-driven membrane. Resonant x-ray-emission spectroscopy (RXES) was fit using the Kramers-Heisenberg formula for photon-atom scattering [15,16]. The Yb volume was found by calculating the Voronoi cell, i.e., the volume closer to one atom than any other. For this calculation, the atomic positions are held constant under pressure and the atoms are weighted by covalent radii.

All calculations are performed within the framework of DFT and similar to a recent study on the rare-earth elemental metals [17]. The necessary assumption for the unknown electron exchange and correlation functional is chosen to be that of generalized gradient approximation. The implementation is done for a full-potential linear muffin-tin orbital (FPLMTO) method [18]. The orbital polarization (OP) is included in the FPLMTO as a parameter-free scheme where an energy term proportional to the square of the orbital moment is added to the total energy functional to account for intra-atomic interactions. It is an approximate method that is analogous to the mean-field approximation for the spin-polarization energy. An  $f_n$  atomic configuration involves intra-atomic interactions such as (vector model)  $\mathbf{s}_i \cdot \mathbf{s}_j$  and  $\mathbf{l}_i \cdot \mathbf{l}_j$  (electron  $i$  spin and angular momenta). Here we replace the energy associated with the angular momenta  $-\frac{1}{2} \sum \mathbf{l}_i \cdot \mathbf{l}_j$  with a mean-field expression  $-\frac{1}{2} (\sum \mathbf{l}_i^z) (\sum \mathbf{l}_j^z)$  ( $z$  component of vector  $\mathbf{l}$ ). This term is proportional to  $L^2$  in analogy to the Stoner energy for spin polarization  $-\frac{1}{2} (\sum \mathbf{s}_i^z) (\sum \mathbf{s}_j^z)$ , which is proportional to  $M^2$ . Here  $L$  and  $M$  are the total orbital and spin moments, respectively. In the OP scheme this then provides for a one-electron eigenvalue shift proportional to  $-Lm_l$  (for each state  $m_l$ ) that enhances the orbital polarization over the spin-orbit coupling only case. The connection between OP and the local-density approximation plus Hubbard-type Coulomb interaction (LDA+U) methodologies was discussed recently [19]. One distinct advantage with the OP scheme over the LDA+U method is that the former does not depend on a parameter whose pressure dependence is unknown.

### III. RESULTS

#### A. Density functional theory

The results of the DFT calculations are shown in Fig. 1 and compared to previously published work. The DFT predicts a continuous increase in Yb valence with decreasing interatomic distance and reproduces the general trend of the previously published experimental data. The DFT calculations, however, suggest a stronger dependence on Yb-Yb spacing than observed, thus predicting a smaller valence for YbNiIn<sub>4</sub>, but predicting a larger than observed valence for YbNiGa<sub>4</sub>,

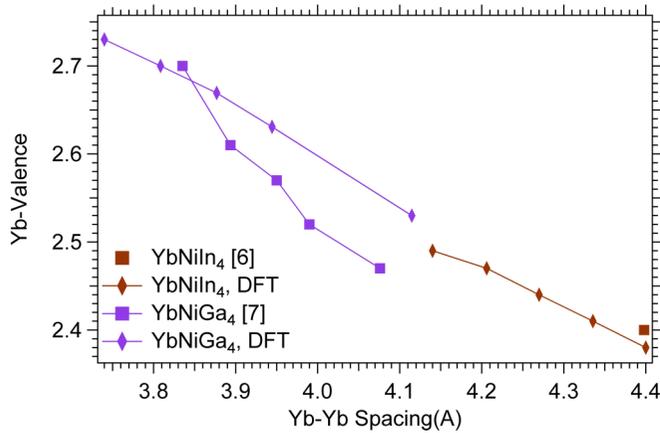


FIG. 1. Summary of DFT calculations and previously published work. The DFT reproduces the general behavior, but implies a stronger dependence on Yb-Yb spacing.

implying that electron correlation effects beyond what is included in the present DFT calculations may be responsible.

**B. Structural studies**

X-ray-diffraction measurements were performed on YbNiGa<sub>4</sub> and YbNiIn<sub>4</sub> up to a pressures of 63 and 45 GPa,

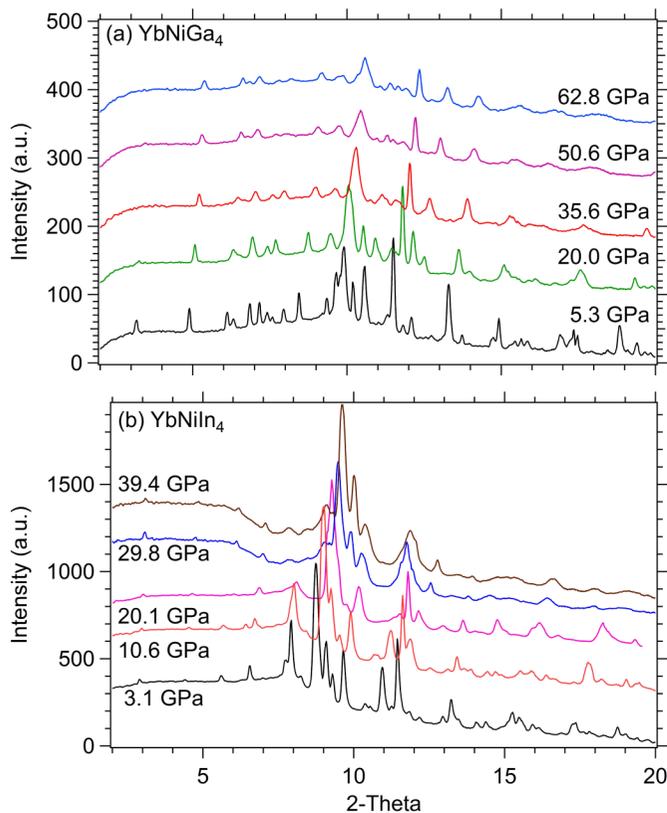


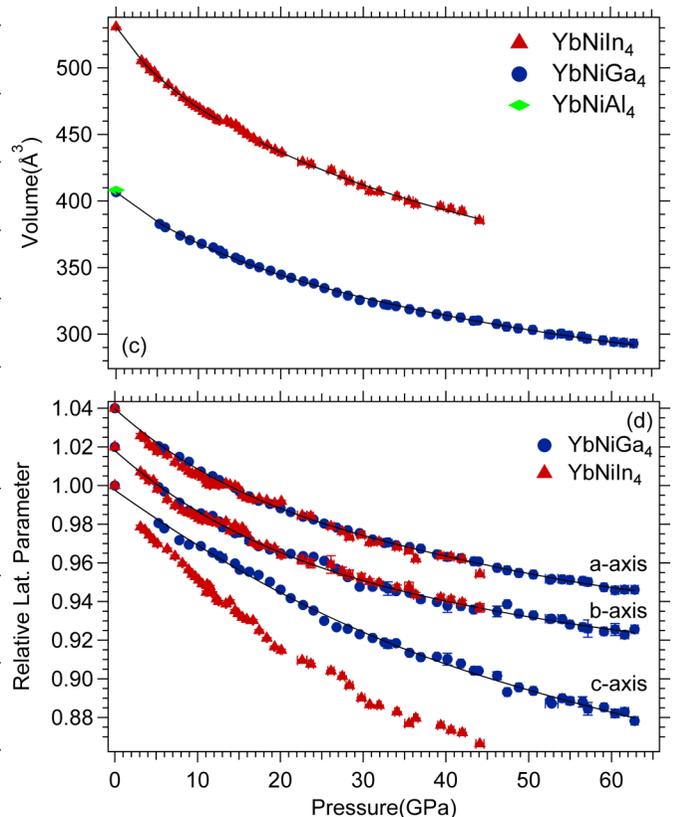
FIG. 2. The PXRD spectra at select pressures for (a) YbNiGa<sub>4</sub> and (b) YbNiIn<sub>4</sub>. In (b) the spectra shown for 29.8 and 39.4 GPa were acquired for a measurement separate from those presented for pressures below, which accounts for some of the difference in intensities of the sample peaks. (c) Volume contraction as a function of pressure. (d) Contraction of lattice parameters as a function of pressure. The *a* and *b* axes appear to contract similarly among the compounds, but the *c* axis displays significantly different behavior. The uncertainties are taken from the uncertainty in GSAS fitting and are generally smaller than the markers. The solid lines in (c) represent a fit to the Birch-Murnaghan equation of state. The solid lines in (d) are guides to the eye, and the *a* and *b* axes are offset by 0.04 and 0.02, respectively.

respectively. Figure 2 shows that YbNiGa<sub>4</sub> contracts without any sign of a structural transition and can be well described by the Birch-Murnaghan (BM) equation of state (EOS) with  $B = 76.7$  GPa and  $B' = 5.5$  [20]. YbNiIn<sub>4</sub> shows a plateau in volume between 12 and 14 GPa, which is caused by plateaus in the *a* and *b* axes in this pressure range [Fig. 2(d)]. Below 12.5 GPa, the BM EOS yields  $B = 54.2$  GPa and  $B' = 7.0$ , and the high-pressure region above 17 GPa yields  $B = 63$  GPa and  $B' = 5.0$ , values that more closely resemble those of YbNiGa<sub>4</sub>. To better determine the origin of the plateau near 13 GPa, we have performed a linearization of the BM EOS as described in Ref. [21] and plot the resulting reduced pressure  $H$  vs the Eulerian strain  $f_E$  in Fig. 3. Plotting the reduced pressure vs Eulerian strain should be linear for any stable compound, while a change in slope may be indicative of an electronic topological transition (ETT). As shown in Fig. 3, YbNiIn<sub>4</sub> shows a sudden spike in the reduced pressure at a Eulerian strain of 0.05 (corresponding to 12.5 GPa), which is accompanied by a change in slope, which may indicate the presence of an ETT.

**C. Valence measurements**

*1. X-ray-absorption spectroscopy*

X-ray-absorption spectroscopy is sensitive to the valence of the Yb ions because the 4*f* states (4*f*<sup>13</sup> and 4*f*<sup>14</sup>) both



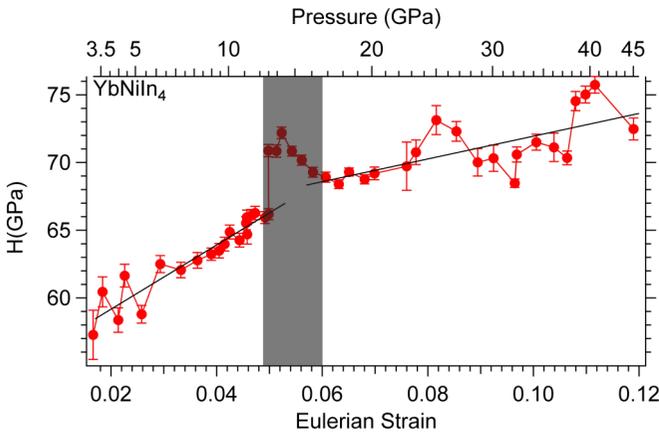


FIG. 3. Reduced pressure  $H$  plotted against Eulerian strain for  $\text{YbNiIn}_4$ . The  $\text{YbNiIn}_4$  displays a sharp spike as well as a change in slope just above a Eulerian strain of 0.05, suggestive of an ETT. The solid black lines represent linear fits below a Eulerian strain of 0.05 and above 0.06. The shaded area represents the transition region.

experience different screening. Each valence state ( $\text{Yb}^{3+}$  and  $\text{Yb}^{2+}$ ) will result in a distinct absorption peak in the XAS spectra, which are separated by approximately 8–12 eV. By calculating the weighted average of the peak intensities, the effective Yb valence can be determined. The XAS spectra can be fit by describing each valence state with a Gaussian and error function. As reported in several papers studying Yb valence in other materials, we observed a splitting of the  $\text{Yb}^{3+}$  peak, which is likely due to the crystal field splitting of the unoccupied  $5d$  states [22–25]. Previous work measuring the valence of  $\text{YbNiGa}_4$  and  $\text{YbNiIn}_4$  was performed in transmission mode and lacked the resolution to fit both  $\text{Yb}^{3+}$  peaks, resulting in a minor difference in the determined valence and pressure dependence thereof compared to the work reported herein [6,7]. Figure 4 shows the details of our fit for the ambient pressure data, as well as several XAS spectra at select pressures.

For  $\text{YbNiGa}_4$  there exists a clear decrease in the intensity of the  $\text{Yb}^{2+}$  peak and an increase in intensity of the  $\text{Yb}^{3+}$  peak up to a pressure of 25 GPa. Above 25 GPa, the ratio of absorption peak to fluorescence decreases, but the valence remains largely unchanged. For  $\text{YbNiIn}_4$  the ambient pressure measurement reveals a larger contribution from the fluorescent region than the subsequent pressure measurements, resulting in an apparent increase of both the  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$  peaks from ambient to 6.8 GPa. Nonetheless, the ratio of amplitudes of these valence peaks results in an increase in valence with pressure, following the trend observed for all the measured pressures. Summarizing, the valence determination via XAS and adding previous valence determinations for  $\text{YbNiGa}_4$  yields Fig. 5 [7].

The Yb valence of  $\text{YbNiGa}_4$  increases up to about 20–25 GPa, at which point the valence saturates at  $n = 2.68$ . The Yb valence in  $\text{YbNiIn}_4$  may be approaching saturation at the highest measured pressures, but there is also a peak in valence close to 10 GPa. This peak is likely another manifestation of the ETT which was observed in the structural measurements and could be the result of the changing electronic density of states near the ETT.

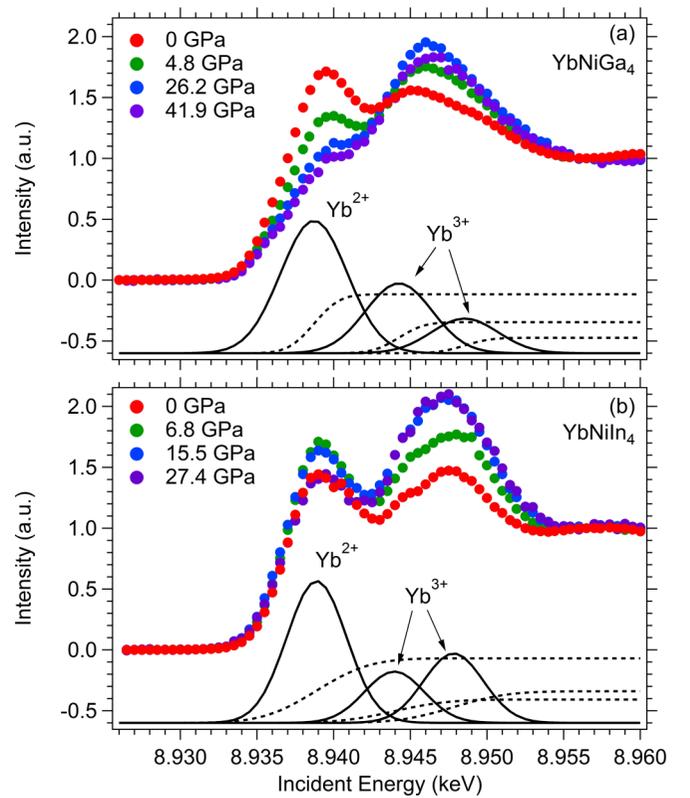


FIG. 4. X-ray-absorption spectra at select pressures and the fit functions for (a)  $\text{YbNiGa}_4$  and (b)  $\text{YbNiIn}_4$ . The spectra are normalized to an edge jump of unity. The solid lines of the fit correspond to the Gaussian functions associated with the valence peaks and the dashed lines are their respective error functions accounting for entering the fluorescent region. The  $\text{Yb}^{2+}$  and  $\text{Yb}^{3+}$  peaks are indicated. With increasing pressure, the  $\text{Yb}^{3+}$  peak gains intensity, while the  $\text{Yb}^{2+}$  peak loses intensity.

## 2. Resonant x-ray-emission spectroscopy

Resonant x-ray-emission spectroscopy is a powerful tool for fully describing the valence state of a given material, which scans the emission energy in addition to the incident energy. Converting the emitted energy to transferred energy and combining this into a single plot results in the RXES spectra shown in Fig. 6 for  $\text{YbNiGa}_4$  and in [8] for the single pressure measured for  $\text{YbNiIn}_4$ . As in the case of PFY measurements, the amplitudes of the absorption peaks allow for determination of the valence.

$\text{YbNiGa}_4$  begins with a rather broad peak due to the overlap of the three distinct contributions of the measured valence peaks, but with increasing pressure the valence state is shifted away from the  $\text{Yb}^{2+}$  and towards the  $\text{Yb}^{3+}$  state. This results in only a weak  $\text{Yb}^{2+}$  structure remaining at 42 GPa. The overall trend of the valence determined from RXES and XAS is the same for  $\text{YbNiGa}_4$ . The valence peaks in  $\text{YbNiIn}_4$  have a larger separation resulting in more distinct peaks and agreeing with the valence determined via XAS.

## IV. DISCUSSION

While both the structural and spectroscopic data of  $\text{YbNiIn}_4$  are suggestive of an ETT, it is important to note

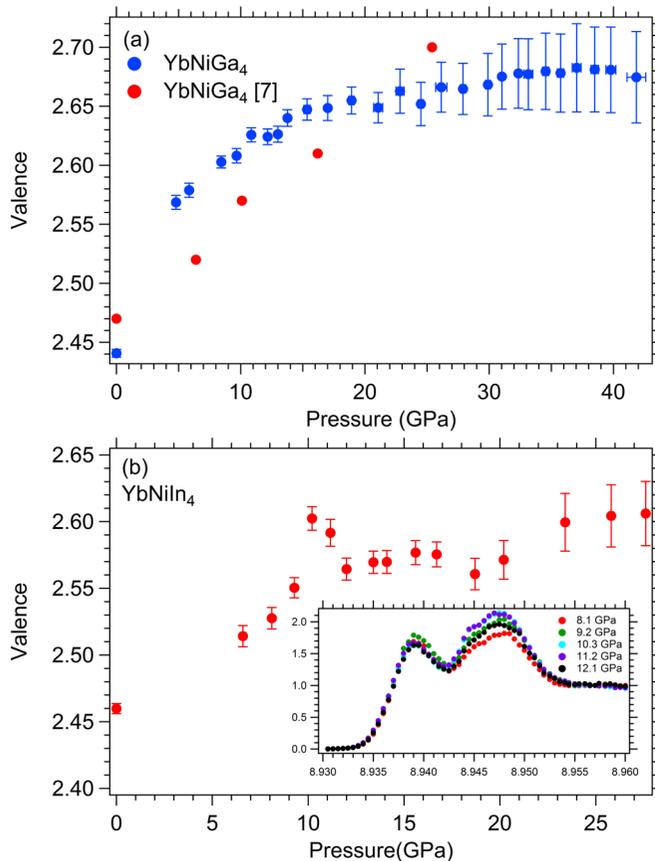


FIG. 5. The Yb valence as a function of pressure determined via XAS for (a)  $\text{YbNiGa}_4$  and (b)  $\text{YbNiIn}_4$ . The inset shows the XAS spectra of  $\text{YbNiIn}_4$  around 10 GPa. The valence for  $\text{YbNiGa}_4$  increases up to about 25 GPa, above which the valence appears to be saturated. For  $\text{YbNiIn}_4$ , the valence reveals a peak near 10 GPa, consistent with the ETT proposed from structural results. Aside from this peak, the valence increases steadily up to the maximum pressure  $P = 27$  GPa, though the highest measured pressure points suggest that the valence may be reaching saturation. The uncertainties were calculated from weighted fitting in IGOR.

that previous work measuring ETTs did not observe a peak in reduced pressure or conversely did not show a plateau in volume near the ETT [21,26,27]. We speculate that this is due to the dual nature of the Yb  $4f$  electrons, which display both local and itinerant character in these intermediate valence states observed in  $\text{YbNiIn}_4$ . Previous pressure-induced ETT has been identified in weakly correlated itinerant systems, implying compressibility changes arising only from the bonding changes driven by subtle changes near the Fermi level. In the case of  $\text{YbNiIn}_4$ , the dual nature of the  $4f$ -electron subsystem yields consequences not only for the electronic structure near the Fermi level, as with the itinerant systems, but also for the local corelike states, which have ramifications for the ionic volume of the Yb atoms independent of the physics at the Fermi level. The physics that drives the dual nature of the  $4f$  electrons in  $\text{YbNiIn}_4$  inherently couples the local part of the wave function to the ETT, which may be expected to yield more pronounced effects on compressibility and volume than typically seen in weakly correlated, itinerant systems.

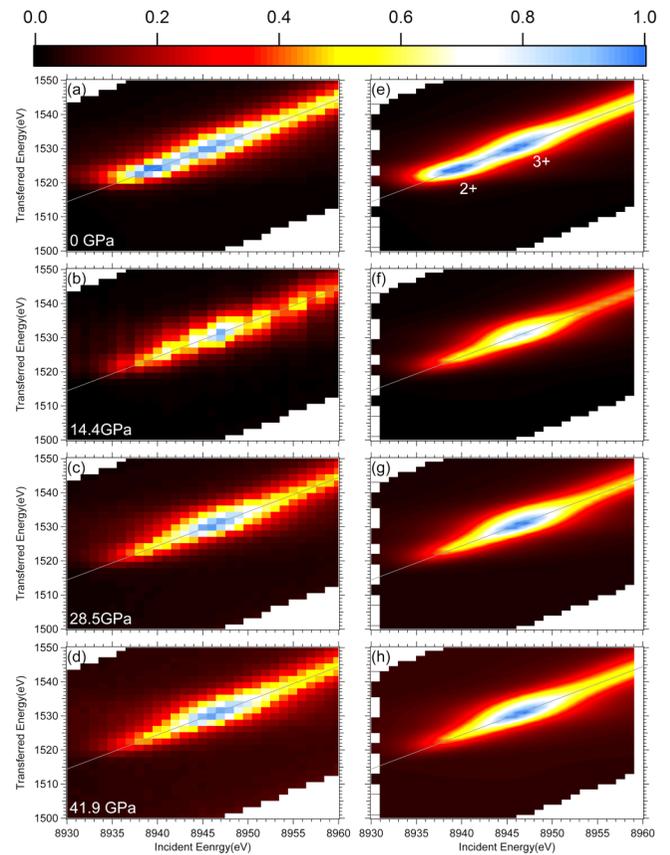


FIG. 6. (a)–(d) The RXES spectra and (e)–(h) the corresponding fit for  $\text{YbNiGa}_4$ . With increasing pressure, the low energy  $2+$  peak decreases in intensity. The gray lines correspond to the XAS PFY scans described above. Intensities are normalized to the maximum intensity of the  $3+$  peak of the experimental data.

Figure 7(a) summarizes the valence determined via each of the described methods and includes the valence determined for  $\text{YbNiAl}_4$  from Ref. [5].  $\text{YbNiGa}_4$  appears to reach saturation at  $n = 2.68$  above  $P = 25$  GPa and surpasses the Yb valence measured in  $\text{YbNiAl}_4$ . Both  $\text{YbNiGa}_4$  and  $\text{YbNiIn}_4$  have comparable Yb valence at ambient pressure, but in  $\text{YbNiGa}_4$  the valence appears to be more sensitive to pressure. As shown in Fig. 7(b), the unit cell volume fails to describe the overlapping region of these materials, though both materials individually reveal the expected trend of increasing valence with decreasing unit cell volume.

Figure 8 shows Yb valence vs Yb-Yb spacing, which shows behavior similar to that for the unit cell volume. Most of the data for  $\text{YbNiIn}_4$ , the high-pressure region of  $\text{YbNiGa}_4$ , and  $\text{YbNiAl}_4$  at ambient pressure appear to follow a smooth valence vs Yb-Yb spacing curve. However, in the region where these compounds have similar Yb-Yb spacing, the Yb valence reveals a precipitous drop, indicating that Yb-Yb spacing does not capture the entirety of the underlying physics. Figure 8 also includes DFT calculations for the Yb valence in the  $\text{YbNiGa}_4$  and  $\text{YbNiIn}_4$  systems. While the zero-pressure value of valence in  $\text{YbNiIn}_4$  predicted by the DFT is lower than the experiments, the pressure-dependent trend of the valence as predicted by the DFT is in good agreement with the experimental observations. For

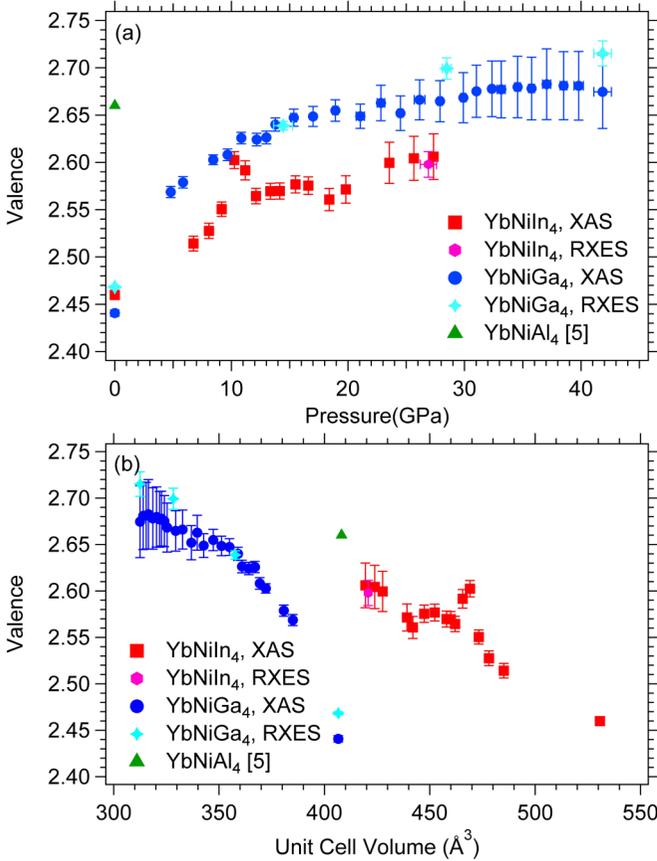


FIG. 7. Summary of valence measurements of this system plotted against (a) pressure and (b) unit cell volume. With increasing pressure, the Yb valence increases, though in YbNiGa<sub>4</sub> the Yb valence appears to reach saturation above 25 GPa. Unit cell volume is insufficient to fully describe the Yb-114 system.

YbNiGa<sub>4</sub> the behavior is also reproduced well for smaller lattice spacings, while the drastic drop at the larger lattice spacings is not predicted by the theory. It is particularly the

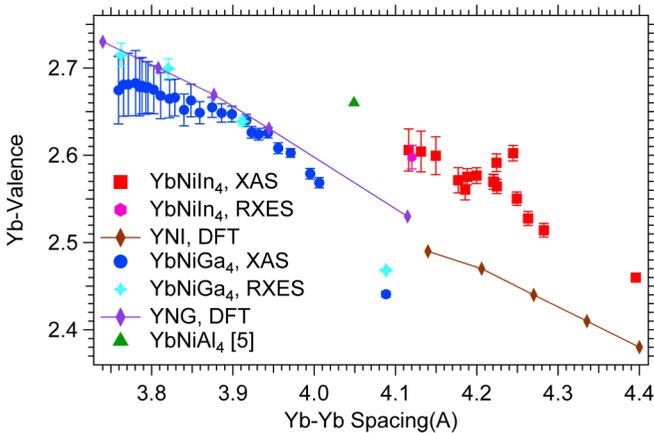


FIG. 8. The Yb valence plotted against the Yb-Yb spacing and DFT calculations for Yb valence vs Yb spacing. The DFT does not reproduce the sharp drop when transitioning from YbNiIn<sub>4</sub> under pressure to YbNiGa<sub>4</sub> under ambient conditions.

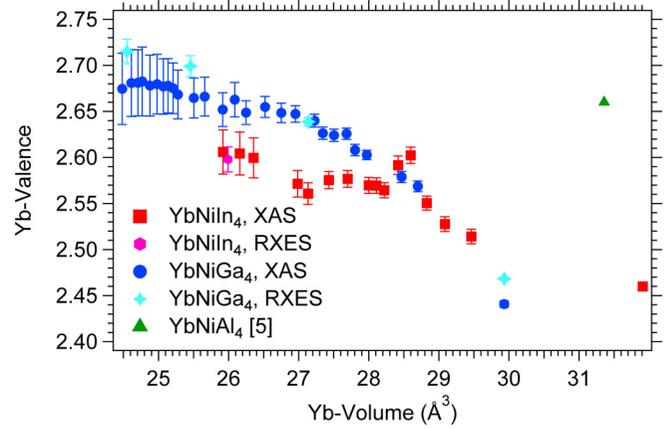


FIG. 9. The Yb valence plotted against the Yb volume. There is no convincing trend and YbNiAl<sub>4</sub> does not fit within this framework.

measured valence at the largest Yb-Yb spacing (4.07 Å) that deviates from theory (2.44 vs 2.54) and the reason is not clear. We speculate that electron correlation effects beyond what is included in the present DFT calculations may be the cause.

As an attempt to account for the effect of substituting In and Ga, we calculated the Yb volume, i.e., the space available to the Yb atoms, for each measured pressure. The results are shown in Fig. 9. Consideration of the Yb volume does not provide a satisfactory result, and YbNiAl<sub>4</sub> does not fit into this scheme. In the case of Yb volume, the In and Ga variants are comparable and exhibit similar slopes, but this still does not fully capture the evolution of the Yb valence. This, combined with the Yb spacing and unit cell volume data, suggests that structural parameters alone are insufficient to fully describe the valence behavior of this system. This implies that the Yb valence is also sensitive to the chemical environment, i.e., the hybridization between the Yb 4*f* and group III*b* *p* states, an effect not captured by DFT calculations.

## V. CONCLUSION

In summary, the Yb valence in the Yb-114 system can be readily modified by pressure, but that valence is not simply described by nearest-neighbor bond distances. By using partial fluorescence yield measurements, we have improved the resolution of the valence determination in YbNiGa<sub>4</sub>, which reveals a steady increase in valence from  $n = 2.44$  up to  $n = 2.68$  near  $P = 25$  GPa, saturating shortly thereafter. The Yb valence of YbNiIn<sub>4</sub> shows similar overall behavior, but we have also observed a sharp valence enhancement in YbNiIn<sub>4</sub> just above 10 GPa. This peak coincides with a plateau in volume, which we speculate is the result of an electronic topological transition. The Yb valence is most closely related to the Yb-Yb spacing in this structure, though this parameter is insufficient to describe the valence across the entirety of the YbNi(Ga, In)<sub>4</sub> system. The hybridization resulting from the Yb-In, Yb-Ga, and Yb-Al bonds appears to be dependent on atomic species and not just the natural bond lengths set by ionic sizes.

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