

Ferroelectricity of structural origin in the spin-chain compounds $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ J. Shi,¹ J. D. Song,² J. C. Wu,² X. Rao,² H. L. Che,¹ Z. Y. Zhao,^{2,3} H. D. Zhou,^{4,5} J. Ma,^{4,6,7} R. R. Zhang,⁸ L. Zhang,⁸
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We report a systematic study of the structure, electric, and magnetic properties of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ single crystals with $x = 0.72$ and 0.26 . The dc and ac magnetic susceptibilities display anomalies with characteristic of the spin freezing. The crystals show ferroelectric transition at 40 and 35 K (T_{FE}) for $x = 0.72$ and 0.26 , respectively, with a large value of $1400 \mu\text{C}/\text{m}^2$ at 8 K for the electric polarization (P_c) along the spin-chain (c -axis) direction. Interestingly, the electric polarization perpendicular to the chain direction (P_{ab}) can also be detected and has the value of 450 and $500 \mu\text{C}/\text{m}^2$ at 8 K for the $x = 0.72$ and 0.26 samples, respectively. The specific heat and magnetic susceptibility show no anomaly around T_{FE} , which means that the electric polarization of these samples has no direct relationship with the magnetism. The x-ray diffraction and the Raman spectroscopy indicate that these samples may undergo Jahn-Teller distortions that could be the reason of electric polarization.

DOI: [10.1103/PhysRevB.96.064112](https://doi.org/10.1103/PhysRevB.96.064112)**I. INTRODUCTION**

Multiferroic materials have attracted considerable attention in recent years due to their potential applications in magnetoelectronics, spintronics, and magnetic memory technology [1–6]. $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x \approx 1$) was found to be a typical system with magnetism-driven ferroelectricity [6–10]. It is composed of the c -axis spin chains consisting of magnetic ions with alternating face-sharing CoO_6 trigonal prisms and MnO_6 octahedra, as shown in Fig. 1(a). Both the Co^{2+} ions and Mn^{4+} ions are in the high-spin states [7]. Because of the much stronger intrachain interaction than the interchain one and the strong Ising-like anisotropy, it can be characterized by a one-dimensional Ising model [7]. A special up-up-down-down antiferromagnetic structure was found, caused by the frustrated ferromagnetic nearest-neighbor and antiferromagnetic next-nearest-neighbor interactions within the spin chain [6]. The alternating Co^{2+} and Mn^{4+} ionic order breaks the inversion symmetry and induces an electric polarization along the chain direction at 16.5 K [6]. The electric polarization is destroyed when applying magnetic field along the c axis, which was proposed to be due to the magnetic-field-induced transition from the up-up-down-down state to the up-up-up-down spin-solid state [8]. Whereas, a different mechanism for the suppression of electric polarization with magnetic fields near 10 T was proposed to be flopping of the Mn^{4+} spins into the ab plane [9]. Moreover, a slight destruction

of the Co/Mn ionic order could significantly enhance the ferroelectricity. For example, it was found that the polycrystal sample of $\text{Ca}_3\text{Co}_{1.08}\text{Mn}_{0.92}\text{O}_6$ has a ferroelectric transition at $T_{\text{FE}} \sim 30$ K [10]. It was explained that the competition between the ionic order and spin frustration is crucial for improving the ferroelectric performance [10]. It would be interesting to study the electric and magnetic properties of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ with even larger deviation of the composition from $x \approx 1$. Some earlier studies were mainly performed with polycrystal samples and focused on their magnetism. In particular, an amazing memory effect was found in $\text{Ca}_3\text{Co}_{1.62}\text{Mn}_{0.38}\text{O}_6$ single crystal [11–13]. However, the study about their electric properties is still missing.

In this work, we grew single crystals of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ with $x = 0.72$ and 0.26 , and studied the structure, magnetism, thermodynamic property, ferroelectricity, and dielectric constant. It was found that along the spin-chain direction both the magnitude of electric polarization and transition temperature are enhanced several times, compared with those of $\text{Ca}_3\text{CoMnO}_6$ [6,8]. In addition, the electric polarization perpendicular to the chain direction was also detected. This is the first time in the experiment observing a polarization perpendicular to the c direction in the $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ system. The origin of the ferroelectricity in the $x = 0.72$ and 0.26 samples is found to be likely the lattice distortion instead of magnetism.

II. EXPERIMENTS

Single crystals of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 0.72$ and 0.26) were grown using a K_2CO_3 flux [14]. In the first step, single-

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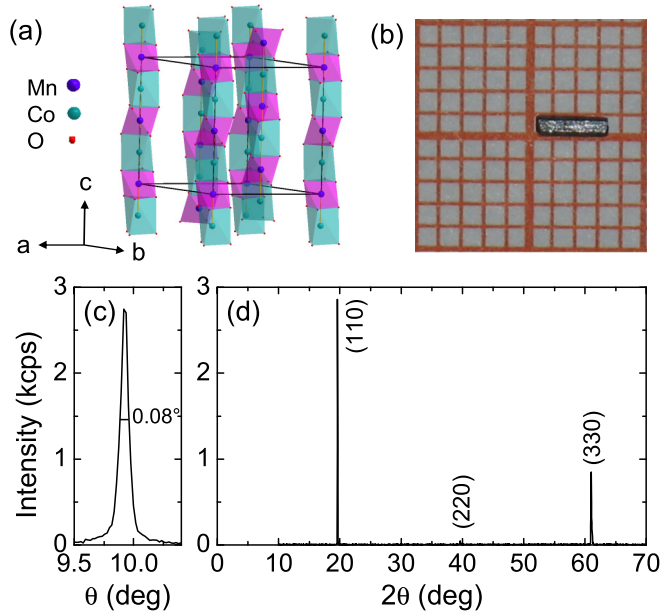


FIG. 1. (a) Crystal structure of $\text{Ca}_3\text{CoMnO}_6$. The Ca ions are omitted for clarity. (b) The photo of a selected $\text{Ca}_3\text{Co}_{1.28}\text{Mn}_{0.72}\text{O}_6$ single crystal, the grid scale is 1 mm. (c) X-ray rocking curve of the (110) reflection, with a FWHM of only 0.08° , for an $x = 0.72$ single crystal. (d) The $(hh0)$ diffraction pattern of this sample.

phase $\text{Ca}_3\text{Co}_{4-2x}\text{Mn}_{2x}\text{O}_9$ powder was obtained by calcining a mixture of CaCO_3 , MnO_2 , and CoO in air at 950°C for 3 days with several intermediate grindings. A mixture of $\text{Ca}_3\text{Co}_{4-2x}\text{Mn}_{2x}\text{O}_9$ and K_2CO_3 in a weight ratio of 1 : 7 was loaded into aluminum crucibles and soaked at 950°C for 6 h and then slowly cooled at a rate of $0.25^\circ\text{C}/\text{h}$ to 880°C , before finally cooled down to room temperature at rate of $100^\circ\text{C}/\text{h}$. This method yielded long-bar-shaped single crystals with the longest dimension along the c axis, as shown in Fig. 1(b). The maximum size of single crystals is about $5\text{ mm} \times 1.5\text{ mm} \times 0.6\text{ mm}$.

The samples were characterized by x-ray diffraction (XRD) at room temperature and the chemical composition was checked via x-ray fluorescence spectrometer (XRF). The low-temperature XRD was also carried out to investigate the structural phase transition. The Raman scattering measurements were performed by using a Horiba Jobin Yvon T64000 Micro-Raman instrument with a Torus laser ($\lambda = 532\text{ nm}$) as an excitation source in a backscattering geometry, and the laser power was kept at $\sim 1\text{ mW}$ to avoid local heating effect. The dc and ac magnetic susceptibilities were measured by using a superconducting quantum interference device-vibrating sample magnetometer (SQUID-VSM, Quantum Design). The specific heat was measured by a relaxation method using a commercial physical property measurement system (PPMS, Quantum Design). For electric polarization and dielectric constant measurements, the crystals were cut into thin-plate shape and polished, and then annealed at 600°C for 6 hours to remove the strain caused by polishing. The samples were poled in an electric field $E \approx 10\text{ kV}/\text{cm}$ from high temperature ($> 50\text{ K}$) to 8 K and then P_c and P_{ab} as a function of temperature were obtained by integrating the pyroelectric current measured

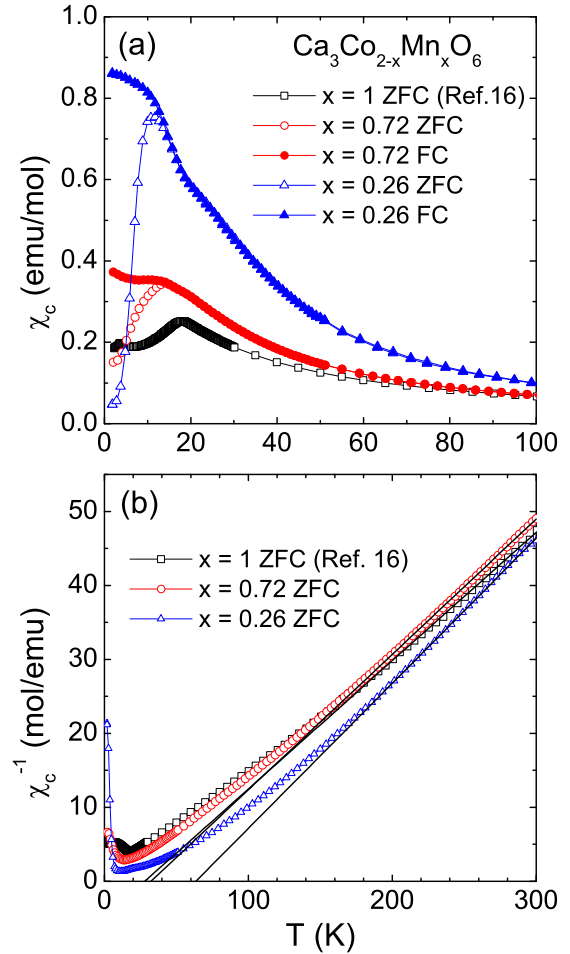


FIG. 2. (a) Magnetic susceptibility χ_c as a function of T for $x = 0.72$ and 0.26 single crystals in 1000 Oe field along the c axis. (b) Temperature dependence of the inverse susceptibility χ_c^{-1} . The solid lines are the Curie-Weiss fitting to the high- T data. For comparison, data of the $x = 1$ sample taken from Ref. [16] are also shown.

by an electrometer (model 6517B, Keithley). The dielectric constant along the c axis was measured by using HP4294 impedance analyzer.

III. RESULTS AND DISCUSSION

The XRD patterns at room temperature indicate that both the $x = 0.72$ and 0.26 samples have a K_4CdCl_6 -type crystal structure (space group $R\bar{3}c$), the same as that of the $x = 1$ sample [15]. Figure 1(c) shows the x-ray rocking curve of the (110) diffraction for an $x = 0.72$ single crystal. The peak is very narrow with the full width at half maximum (FWHM) of 0.08° , indicating the good crystallization of the crystal. The x-ray diffraction pattern of the $(hh0)$ plane is shown in Fig. 1(d). These XRD results indicate that the single crystals are of pure phase and good crystallinity.

Figure 2(a) shows the magnetic susceptibility χ_c as a function of temperature under zero-field cooling (ZFC) and field cooling (FC) conditions for the $x = 0.26$ and 0.72 single crystals, in comparison with χ_c of $x = 1$ single crystal taken from an earlier literature [16]. The applied magnetic

field is $H = 1000$ Oe and along the c axis. As shown in Fig. 2(a), $\chi_c(T)$ of $x = 1$ shows a broad peak around 17 K, which is caused by the onset of up-up-down-down magnetic order [6]. However, the experimental results of ac magnetic susceptibility and neutron diffraction revealed that only finite-size magnetic domains develop at the magnetic transition and these domains exhibit an additional spin freezing at a lower temperature [6]. Thus the magnetic ground state of $x = 1$ is a coexistence of magnetic order and spin freezing. This kind of magnetic state can be formed with smaller x , which, for example, was confirmed in the $x = 0.75$ polycrystal sample by ac magnetic susceptibility and neutron powder diffraction [17,18]. Apparently, our $x = 0.72$ sample exhibits similar magnetic properties to the $x = 1$ and 0.75 cases; the broad peak of $\chi_c(T)$ at $T_{\max} = 14$ K and the separation of the ZFC and FC curves below T_{\max} should be attributed to magnetic ordering and spin freezing, respectively. For the $x = 0.26$ samples, there are two clear anomalies in the $\chi_c(T)$ curves. One is the broad peak at $T_{\max} = 11$ K, below which the FC and ZFC curves separate with each other. Another one is a clear upturn of $\chi_c(T)$ at $T < 18$ K. Both of these two features in $\chi_c(T)$ of $x = 0.26$ agree well with the case of $x = 0$ and 0.20 polycrystal samples [11]. Thus the 18-K upturn should be caused by the intrachain ferromagnetic order, and the ZFC-FC curves separation is attributed to a spin freezing. Therefore the $x = 1, 0.72$, and 0.26 samples all appear a spin freezing, which is likely a common feature in $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$, and is probably due to chemical disorder and competing magnetic interactions [13].

Figure 2(b) shows the inverse magnetic susceptibility χ_c^{-1} . The high-temperature data can be fitted with the Curie-Weiss law ($\chi = \frac{C}{T - \theta_p}$), leading to the Weiss temperatures $\theta_p = 28.0,$

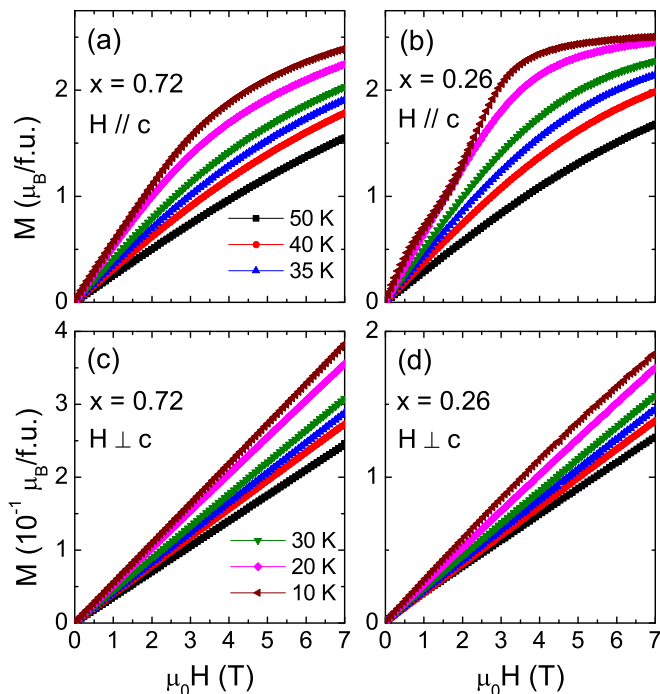


FIG. 3. The magnetization curves with magnetic field parallel and perpendicular to the c direction for $x = 0.72$ and 0.26 single crystals.

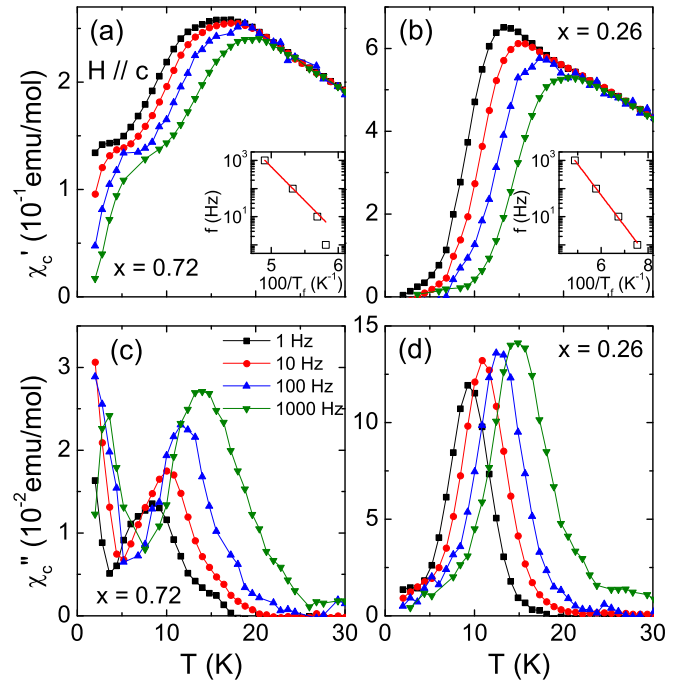


FIG. 4. Real and imaginary parts (χ'_c and χ''_c) of the ac susceptibility of $x = 0.72$ and 0.26 single crystals. The ac magnetic field is 3.8 Oe and the frequencies are 1, 10, 100, and 1000 Hz. The insets show frequency vs inverse temperature of the maximum of χ' . The red lines are fitting results with the Arrhenius behavior.

37.4, and 63.5 K, and the effective moments $\mu_{\text{eff}} = 6.79, 6.50,$ and $6.38 \mu_B/\text{f.u.}$ for the $x = 1, 0.72,$ and 0.26 samples, respectively.

Figure 3 shows the $M(H)$ curves of the $x = 0.72$ and 0.26 single crystals at 50 to 10 K with different field directions. The behaviors are comparable to those of the polycrystal samples and can be understood with the Ising-like Co spins and quasi-isotropic Mn spins [9]. For $H \parallel c$, the $M(H)$ curves at 50 K are nearly linear for both $x = 0.72$ and 0.26 samples. With decreasing temperature, the $M(H)$ curves show clear curvatures, similar to the case of $x = 0.75$ and 0.20 polycrystal samples [11,17]. It is notable that the 10-K $M(H)$ curve of the $x = 0.26$ sample displays a feature of magnetization step, which is similar to the magnetization of $x = 0$ [14]. All these c -axis magnetization behaviors are mainly dominated by the Ising-like Co spins [9]. For $H \perp c$, all the $M(H)$ curves show a linear increase with increasing field and the magnitudes are much smaller than the case of $H \parallel c$. The magnetization for $H \perp c$ should be mainly contributed by the quasi-isotropic Mn spins, and the difference between the $x = 0.72$ and 0.26 samples is supportive for this.

To further confirm the occurrence of spin freezing in $x = 0.72$ and 0.26 samples, the ac magnetic susceptibility were measured at low temperatures with various frequencies. As shown in Fig. 4, large peaks are clearly observed in both the real and imaginary parts of the ac susceptibility. The peaks exhibit strong frequency dependence, shifting to higher temperature with increasing frequency. The Arrhenius behavior, $f = f_0 \exp[-E_a/k_B T_f]$, is used to the quantitative analysis of the shift, where T_f is the temperature of the

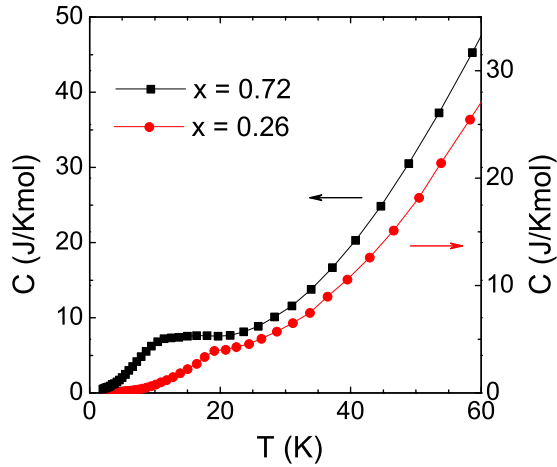


FIG. 5. Specific heat of the $x = 0.72$ and 0.26 single crystals.

maximum χ' [6,19]. The frequency shift of the $x = 0.26$ sample is well fitted with $E_a/k_B = 253$ K and $f_0 = 230$ MHz, as shown in the inset to Fig. 4(b). Another coefficient $K = \Delta T_f/T_f \Delta(\log f)$ can also be used to characterize the spin glass [19]. The K of the $x = 0.26$ data is determined to be 6×10^{-2} , within the range $5 \times 10^{-3} - 8 \times 10^{-2}$ for spin glass [19]. All these indicate that the $x = 0.26$ sample has a spin-glass phase at low temperatures. For the $x = 0.72$ sample, the peaks in ac magnetic susceptibility also exhibit strong frequency dependence. However, the T_f exhibits a discrepancy from the Arrhenius behavior, as shown in the inset to Fig. 4(a). The similar behavior was found in the $x = 0.96$ sample, which is due to the coexistence of magnetic order [6].

Figure 5 shows the specific heat of the $x = 0.72$ and 0.26 single crystals at $T < 60$ K. The specific heat of the $x = 0.72$ sample shows a broad anomaly from 10 to 20 K, similar to that in the $x = 0.75$ polycrystal sample [17], which can be attributed to the magnetic order. For the $x = 0.26$ sample, the specific heat shows a weak anomaly around 18 K, which is associated with the fast increase in the magnetic susceptibility, and can be attributed to the intrachain ferromagnetic order.

Figure 6 shows the temperature-dependent real part of dielectric constant and loss factor of the $x = 0.72$ and 0.26 single crystals. Both of them show a clear frequency dependence. The small value of $\tan \delta$ indicates that these samples have very high insulation, and no space charge contributions below 100 K [20]. Another feature is that both the real part of dielectric constant and loss factor show anomalies around T_{\max} . The real part of dielectric constant presents an abrupt decrease but the loss factor shows a peak. In this regard, the dielectric properties of these samples seem to have some relationship with the spin freezing.

It is known that ferroelectricity appears in $\text{Ca}_3\text{CoMnO}_6$, which was believed to be related to the special up-up-down-down spin structure with the alternating Co^{2+} and Mn^{4+} ionic order [6,8]. The transition temperature, below which electric polarization shows up, is about 16.5 K and the polarization parallel to the c axis (P_c) is about $310 \mu\text{C}/\text{m}^2$ at 2 K [8]. Note that there is no electric polarization perpendicular to the c axis (P_{ab}) [8]. However, our $x = 0.72$ and 0.26 single crystals display very different electric polarization behaviors.

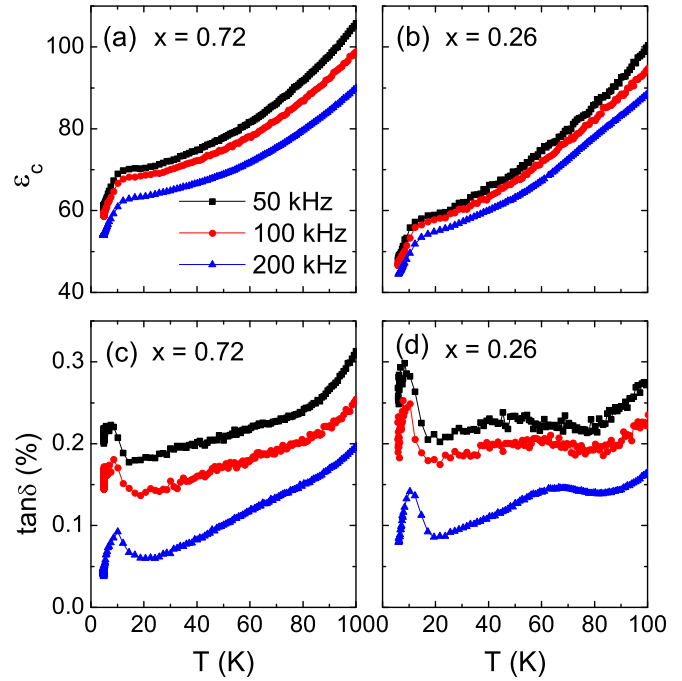


FIG. 6. Real part of dielectric constant and loss factor of the $x = 0.72$ and 0.26 single crystals. The frequencies are 50, 100, and 200 kHz.

First, both P_c and P_{ab} can be detected, as shown in Figs. 7(a) and 7(b). This is the first time to observe the polarization perpendicular to the c direction in $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ system. Second, the electric polarization transition temperatures (T_{FE}) are 40 and 35 K for the $x = 0.72$ and 0.26 single crystals, respectively, much higher than that of $\text{Ca}_3\text{CoMnO}_6$. Third, the magnitude of polarization is much larger in the $x = 0.72$ and 0.26 single crystals. The P_c gradually increases with lowering temperature and reaches $1400 \mu\text{C}/\text{m}^2$ at 8 K for two samples, while the P_{ab} are 450 and $500 \mu\text{C}/\text{m}^2$ at 8 K for $x = 0.72$ and 0.26 , respectively. Since the P_{ab} are nearly one third of the P_c in both samples, the possibility of P_{ab} caused by a small angle deviation from the c direction can be ruled out. In passing, it should be noted that no electric polarization could be observed if the measurements were carried out without poling.

Figures 7(c)–7(f) show the $P_{ab}(T)$ and $P_c(T)$ data for $x = 0.72$ and 0.26 single crystals in different magnetic fields. Here, the magnetic field was applied before cooling, and the measurement with magnetic field applied after cooling gave the same results. Apparently, even high magnetic fields can hardly change either the transition temperature or the magnitude of P . In addition, the dc and ac magnetic susceptibilities, specific heat and dielectric constant do not show any anomaly around T_{FE} . These results indicate that the electric polarization appearing in the $x = 0.72$ and 0.26 samples has no direct relationship with the magnetism, in contrast to the magnetism-driven ferroelectricity in the $x = 1$ material. One may argue that the observed P is caused by some extrinsic effects, in particular, the space charges trapped at the grain boundaries or possible defects as reported elsewhere [21]. However, this possibility can be ruled out because the electric polarizations of these samples can be well reproduced on many different

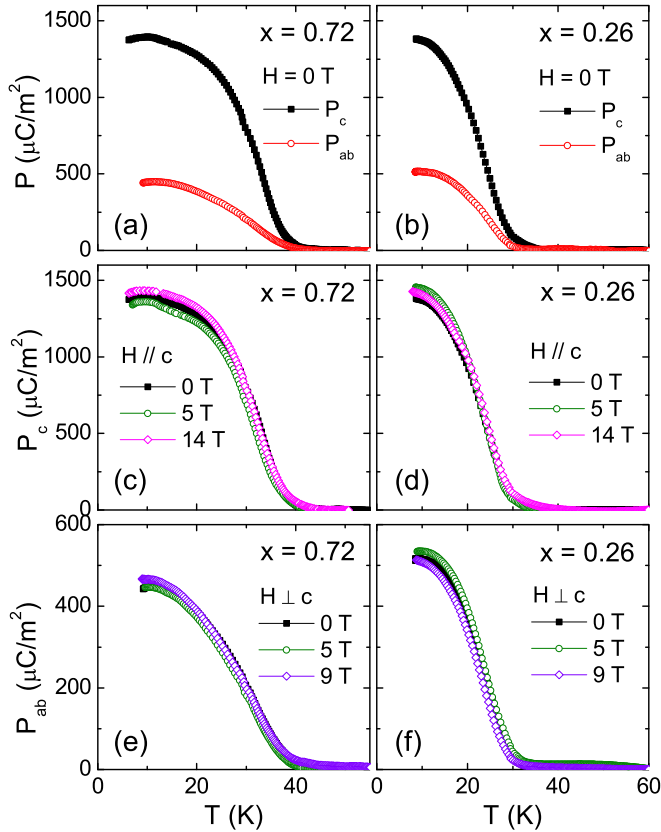


FIG. 7. [(a) and (b)] Electric polarization P_c and P_{ab} as a function of temperature for the $x = 0.72$ and 0.26 single crystals. [(c) and (d)] $P_c(T)$ data in different magnetic fields. [(e) and (f)] $P_{ab}(T)$ data in different magnetic fields.

pieces of single crystals. Therefore the electric polarization appearing at rather high temperatures may originate from the lattice distortion like the case of some ABO_3 perovskites, such as LuCrO_3 and SmCrO_3 with magnetic Cr^{3+} ions at the B site [22,23]. In these materials, the neutron pair distribution function illustrated that the electric polarization could be originated from the Jahn-Teller distortions of Cr^{3+} octahedra, whereas the macroscopic crystal structure was unchanged [24–26]. It is worth of mentioning that an earlier theory has proposed a kind of Jahn-Teller-distortion-induced ferroelectric mechanism in $\text{Ca}_3\text{CoMnO}_6$ [27], which agrees very well with the experimental results of electric polarization for $x = 0.72$ and 0.26 single crystals. In detail, it was reported that in $\text{Ca}_3\text{CoMnO}_6$ the $\text{Co}^{2+}(d^7)$ ions at the trigonal sites are in high-spin state ($S = 3/2$) with unevenly filled degenerate d states and a large orbital magnetic moment of $1.7 \mu_B$, and hence have Jahn-Teller instability. In view of this, if CoO_6 trigonal prisms are theoretically assumed to undergo Jahn-Teller distortion, it will of course affect the electron configuration of Co^{2+} ions because of the change of the crystal-electric-field environment, leading to a redistribution of orbital moment on Co^{2+} . Based on this, two kinds of optimized structures resulting from different types of Jahn-Teller distortions were obtained by $\text{LDA}+U+\text{SOC}$ calculations in terms of the magnitude of the orbital moment on Co^{2+} . Therein, an optimized structure with low orbital moment on Co^{2+} (i.e., $0.56 \mu_B$) removes C_3

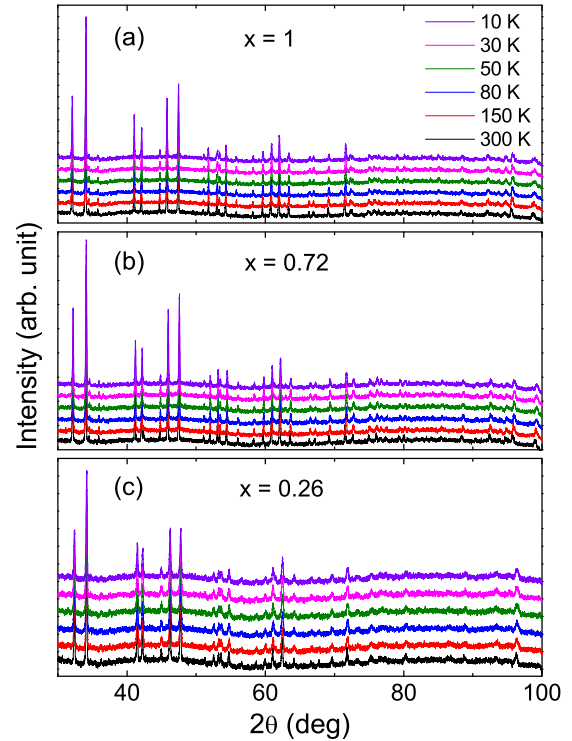


FIG. 8. XRD patterns at different temperatures for $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 1, 0.72,$ and 0.26) polycrystal samples.

rotational symmetry leading to appearance of nonzero electric polarization both along and perpendicular to the c direction [27]. Based on these theoretical results, it is most likely that the electric polarization of $x = 0.72$ and 0.26 are induced by Jahn-Teller distortions. To check this kind of electric-polarization mechanism, we further probed the low-temperature crystal structure of $x = 0.72$ and 0.26 samples by using XRD and Raman spectroscopy techniques.

The XRD of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 1, 0.72,$ and 0.26) polycrystal samples were measured at 10–300 K, as shown in Fig. 8. With decreasing temperature, no extra diffraction peaks appear, indicating no structural phase transition. The structures of the $x = 1, 0.72,$ and 0.26 samples are rhombohedral with the space group $R\bar{3}c$ at all temperatures.

The temperature dependence of lattice constants are shown in Fig. 9. Apparently, anomalies can be observed in all samples. For $x = 1$, the lattice constant a decreases almost linearly with decreasing temperature, and is in good agreement with the principle of thermal expansion. However, the lattice constant c changes in a complicated way with temperature. It decreases with decreasing temperature until 70 K, while suddenly increases below 70 K. It is an abnormal phenomenon, which has never been reported in $\text{Ca}_3\text{CoMnO}_6$. Note that at $T < 70$ K the lattice is elongated only in the c direction and keeps the threefold rotational symmetry. The lattice constants a and c as a function of T for the $x = 0.72$ sample display an anomaly at ~ 40 K. The a and c decrease monotonically with decreasing temperature and become nearly constants at $T \leq 40$ K, indicating that the lattice has some anomalies below 40 K. This temperature is the same as the T_{FE} . The a and c of the $x = 0.26$ sample not only exhibit similar anomaly to

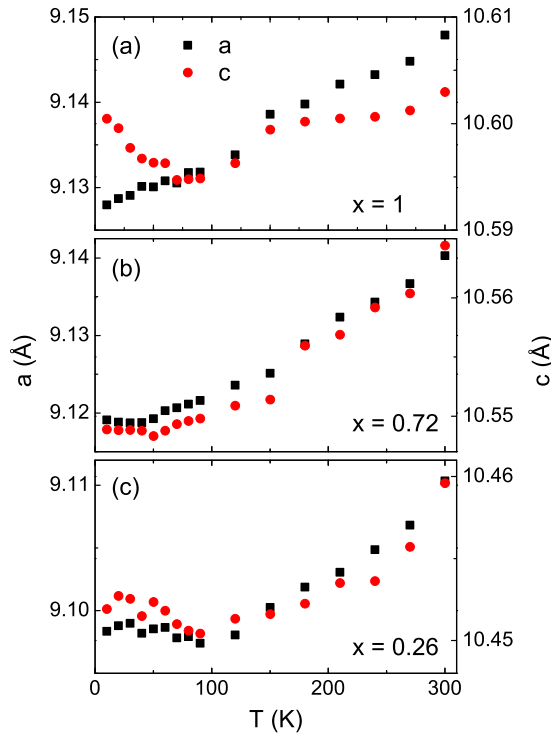


FIG. 9. Lattice constants as a function of temperature for $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 1, 0.72,$ and 0.26) samples.

that of the $x = 0.72$ but also have anomaly at about 90 K, below which the a and c increase with decreasing temperature like the c of $x = 1$ sample. However, this temperature is different from the T_{FE} of $x = 0.26$ sample. The anomalies here may have some relationship with $\text{Ca}_3\text{Co}_2\text{O}_6$ and it will be discussed below. The temperature dependence of the

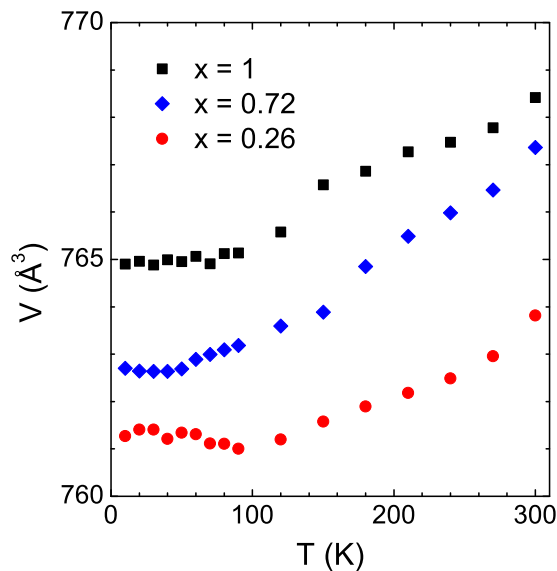


FIG. 10. Temperature dependence of the volume of the unit cell for $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 1, 0.72,$ and 0.26) samples. The data of the $x = 0.72$ and 0.26 samples are shifted upward by 3 and 12 \AA^3 , respectively.

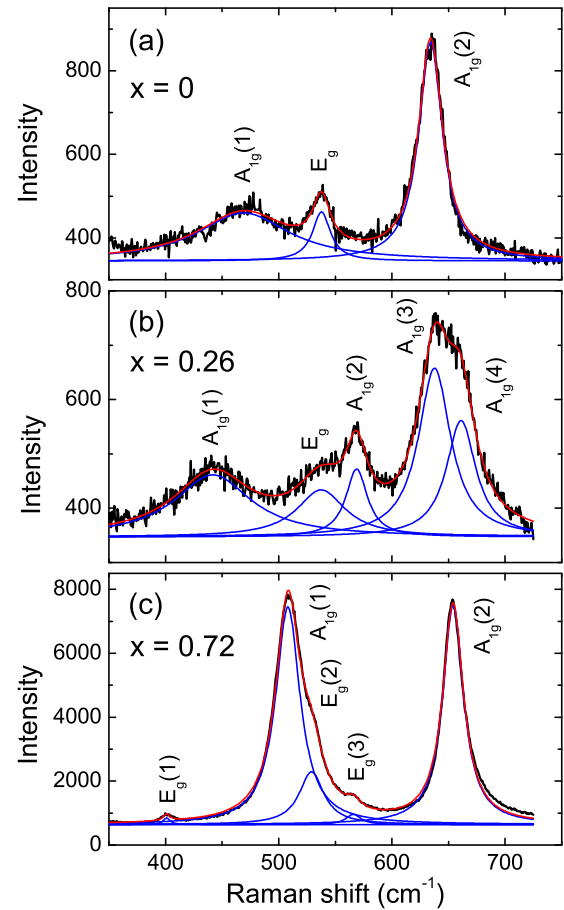


FIG. 11. Room-temperature Raman spectra of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 0, 0.26,$ and 0.72). The red lines are fitting results using the Lorentzian equation and the blue lines are the decomposed fitting curves of the individual Raman-active modes.

volume of the unit cell are displayed in Fig. 10. Again, it is notable that the cell volumes $V(T)$ display anomalies for all samples.

Figure 11 shows the Raman spectra of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ single crystals with $x = 0, 0.26,$ and 0.72 , which were measured at room temperature in the frequency range $350\text{--}725 \text{ cm}^{-1}$. The direction of the polarization of the incident laser is along the c axis. The data are fitted with a Lorentzian function and decomposed into individual components, as shown in Fig. 11. The selection rule predicts 24 Raman-active modes ($4 A_{1g}$ and $20 E_g$), 28 infrared-active modes ($6 A_{2u}$ and $22 E_u$), 11 inactive modes ($5 A_{1u}$ and $6 A_{2g}$), and 3 acoustic translational modes (A_{2u} and $2 E_u$) [28]. In order to determine the Raman modes, polarized Raman spectra of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ single crystals in two different scattering configurations (zz and zx) were also done and shown in Fig. 12. The A_{1g} mode and the E_g mode can be observed in zz and zx scattering configuration, respectively. The E_g modes are related to the vibration of cobalt ions in the octahedra, calcium and oxygen ions. However, the A_{1g} modes are mainly attributed to the vibration of oxygen ions. Base on this result, the individual Lorentzian components in Fig. 11 are characterized. In the case of $x = 0$, three peaks were observed at $468, 537,$ and 633 cm^{-1} , as shown in Fig. 11(a),

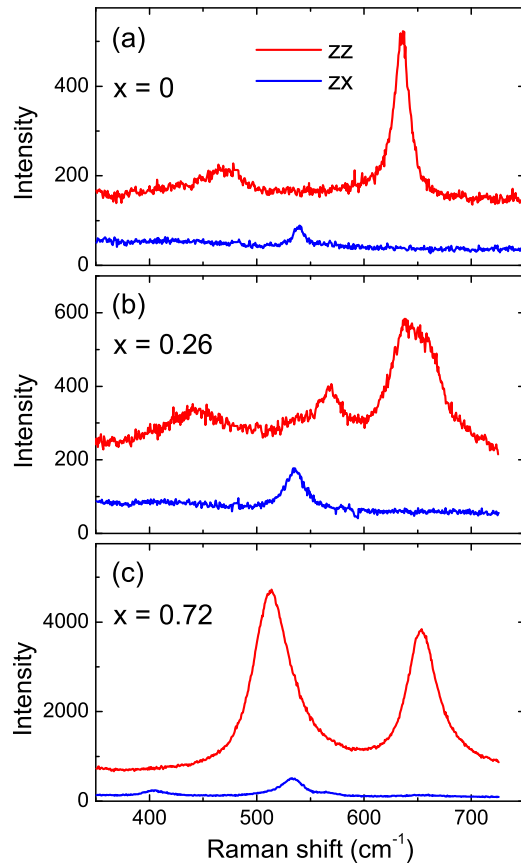


FIG. 12. Polarized Raman spectra of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 0, 0.26,$ and 0.72) with scattering configuration of zz and zx at room temperature.

which are corresponding to the $A_{1g}(1)$, E_g , and $A_{1g}(2)$ modes, respectively. In the case of $x = 0.26$, other two peaks at 568 and 661 cm^{-1} are detected besides the same three modes as the $x = 0$ sample, as shown in Fig. 11(b). These are A_{1g} modes. In the $x = 0.72$ sample, the Raman spectrum changes in such a way that the main features observed in the $x = 0$ sample are no longer detected. Two stronger peaks at 507 and 654 cm^{-1} are corresponding to A_{1g} modes, as shown in Fig. 11(c).

Figure 13 shows the Raman spectra at different temperatures for the $x = 0, 0.26,$ and 0.72 samples. As temperature decreases, the peaks become narrower and better resolved, due to the decrease of disorder arising from thermal motions. No new Raman modes were detected at low temperatures for all samples, indicating no structural phase transition with decreasing temperature. The spectra were also fitted by using Lorentzian function and the temperature dependencies of the $A_{1g}(2)$, $A_{1g}(4)$, and $A_{1g}(2)$ modes are obtained and shown in Fig. 14. Each of them have anomalies with decreasing temperature. There are two sudden changes at 90 and 25 K for the $x = 0$ sample, as shown in Fig. 14(a). The sudden change at 90 K should be related to the anomaly observed in the extended x-ray absorption [29] and Mössbauer measurements [30]. However, the origin of this anomaly is still unclear. The existence of $\text{Co } 3d\text{-O } 2p$ hybridization and incipient magnetic order are presented in x-ray absorption and Mössbauer measurements, respectively. The former is understandable

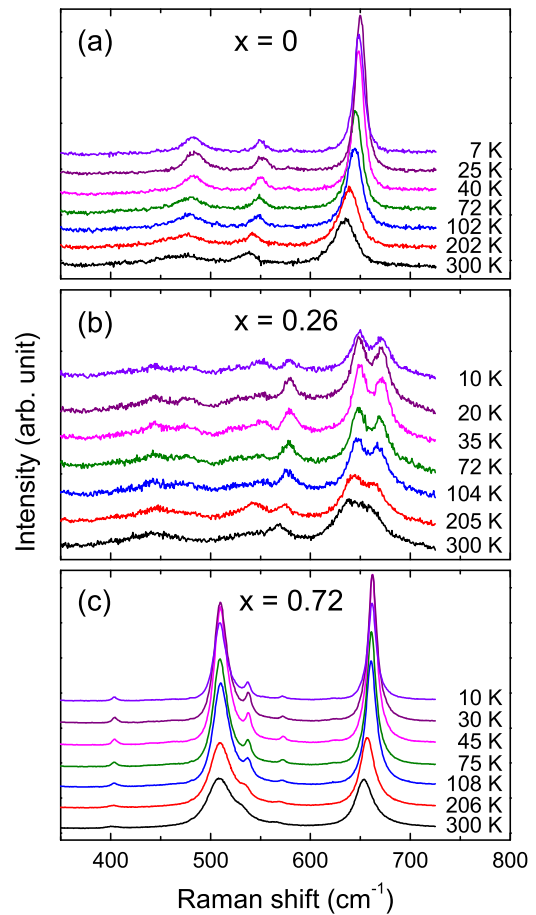


FIG. 13. Raman spectra at different temperatures of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 0, 0.26,$ and 0.72).

because the $A_{1g}(2)$ mode is attributed to the vibration of oxygen ions. The anomaly at 25 K will be discussed later. Figure 14(b) shows the variations of $A_{1g}(4)$ for the $x = 0.26$ sample. Obviously, there are also two anomalies as decreasing temperature. The lower- T one is near 35 K , which is coincided with T_{FE} . The higher- T one is at about 90 K , which is consistent with the temperature of anomaly in XRD data. Since the $x = 0.26$ sample can be treated as $x = 0$ with low doping, it still reserve some properties of $x = 0$ sample and shows a similar anomaly at 90 K to the $x = 0$ sample. The above XRD results indicate that the structural anomaly at 90 K is actually a huge negative expansion effect, which is similar to the case of $x = 0$ reported in some earlier work [29]. It seems that at $T \leq 90 \text{ K}$, there is overall elongation of Co-O bonds, which can be attributed to the $\text{Co } 3d\text{-O } 2p$ hybridization and the holes in the $\text{O } 2p$ bands [29]. We suggest that this structure anomaly does not destroy the spatial inversion symmetry and the threefold rotational symmetry still keeps. As a result, there is probably no electric polarization below 90 K . For the $x = 0.72$ sample, the three Raman modes disappear and two stronger A_{1g} modes are detected. Figure 14(c) shows the variations of $A_{1g}(2)$ for the $x = 0.72$ sample. It just has a sudden change at 40 K , which is in agreement with the anomaly in XRD and electric polarization. Similar anomaly in

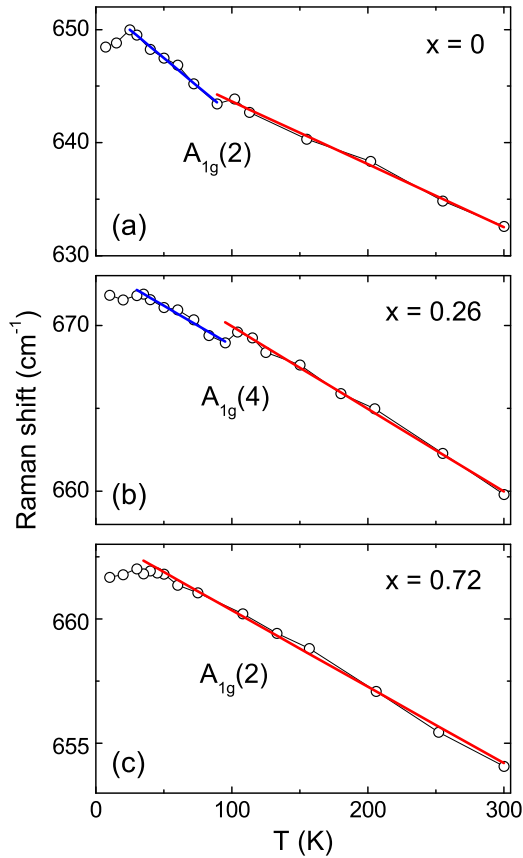


FIG. 14. Variations of Raman peak position as a function of temperature for $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 0, 0.26, \text{ and } 0.72$). The lines are guide to eyes for the nearly linear dependence at some temperature regimes and the anomalies show up at the deviation from these linear behaviors.

Raman modes near the ferroelectric transition were also found in some other materials [31,32].

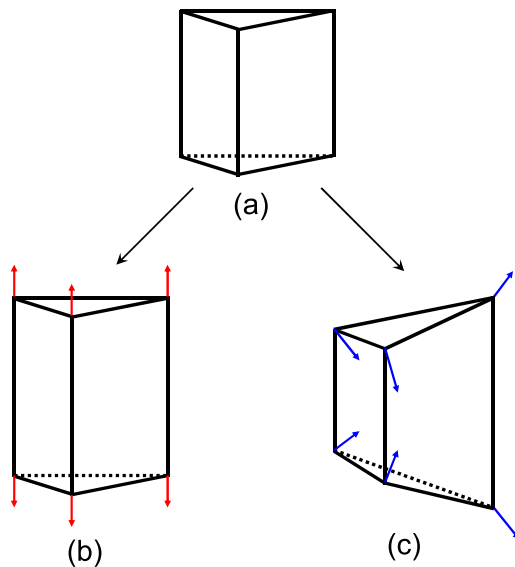


FIG. 15. (a) The sketch of Co^{2+}O_6 trigonal prisms [33]. (b) The trigonal prisms just elongate along c axis and do not break the threefold rotational symmetry. (c) The trigonal prisms produce a cis bond length distortion and break the threefold rotational symmetry.

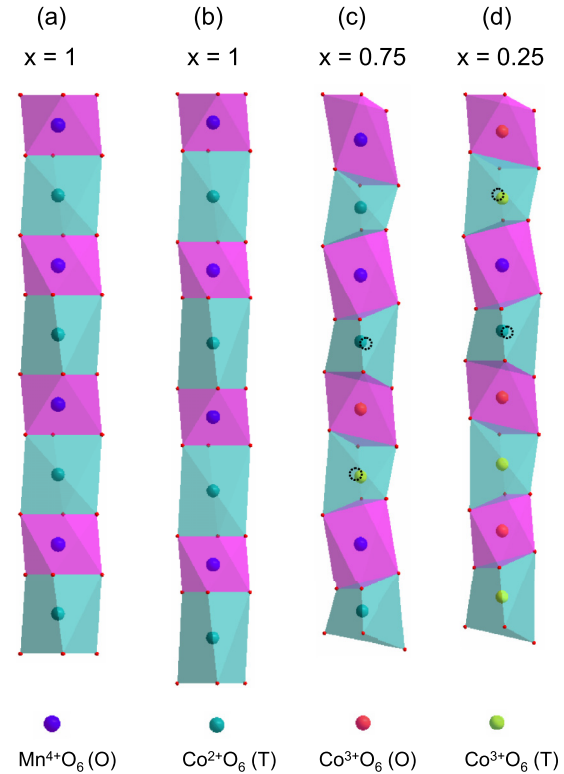


FIG. 16. (a) A periodic arrangement of CoMnO_6 chain of $x = 1$. The Co^{2+}O_6 trigonal prisms and Mn^{4+}O_6 octahedra are alternately arranged. (b)–(d) Possible sketches of CoMnO_6 chains when the Co^{2+}O_6 trigonal prisms undergo a Jahn-Teller distortion. The Co^{2+}O_6 trigonal prisms of $x = 1$ are elongated at the same distance on both sides. The distortions of trigonal prisms in $x = 0.75$ and 0.25 are asymmetric. The black dashed circles are the negative-charge centers of the trigonal prisms, which are not superposed to the positive-charge centers.

Based on the XRD and Raman results, it is clear that the crystal lattices of $\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ ($x = 1, 0.72, \text{ and } 0.26$) have anomalous changes at low temperatures. For the $x = 1$ sample, a clear lattice anomaly appears at 70 K. However, the electric polarization of $\text{Ca}_3\text{CoMnO}_6$ along the c direction is known to be due to the special up-up-down-down spin structure with the alternating Co^{2+} and Mn^{4+} ionic order at 16.5 K [6], instead of this lattice anomaly at 70 K. It is likely that the lattice only elongates in the c direction and keeps the threefold rotational symmetry below 70 K, as shown in Fig. 15(b). In $\text{Ca}_3\text{CoMnO}_6$, the Co^{2+}O_6 trigonal prisms and Mn^{4+}O_6 octahedra are alternately arranged and the Co^{2+}O_6 trigonal prisms are nearly centrosymmetric surrounding the spin chains, as shown in Fig. 16(a). If the Co^{2+}O_6 trigonal prisms are elongated at the same distance on both sides, the positive-charge and negative-charge centers of the Co^{2+}O_6 trigonal prisms are still coincident, as shown in Fig. 16(b). It does not destroy the spatial inversion symmetry in this situation. Therefore the lattice distortion at $T \leq 70$ K in $\text{Ca}_3\text{CoMnO}_6$ does not result in electric polarization. It is also notable that the lattice constants show an abrupt decrease at 150 K for $x = 1$ sample. The earlier works on the $x = 1$ sample did not explore the electric properties around this temperature and therefore it is not clear whether this structural anomaly

can induce electric polarization. In addition, it seems that at 150 K, the lattice constants display quicker shrinking, which however may not necessarily result in electric polarization if the local lattice symmetry is not broken.

For the $x = 0.72$ and 0.26 samples, clear lattice anomaly can also be observed around electric polarization transition temperature (T_{FE}). As discussed above, the electric polarization of $x = 0.72$ and 0.26 samples is most likely caused by a kind of Jahn-Teller distortion of Co^{2+}O_6 trigonal prisms, as shown in Figs. 15(c) and 15(d), which breaks C_3 rotational symmetry leading to nonzero electric polarization both along and perpendicular to the c direction [33]. Figures 16(c) and 16(d) show a periodic arrangement of CoMnO_6 with the particular doping of $x = 0.75$ and 0.25 , which are very close to $x = 0.72$ and 0.26 , respectively. It can be seen that in each unit cell, there is a Co^{2+}O_6 trigonal prism having different neighbors, which will lead to the asymmetric distortion of Co^{2+}O_6 trigonal prism. At these cases, the positive-charge and negative-charge centers of the trigonal prisms would not be coincident. In addition, the trigonal prisms of $x = 0.72$ and 0.26 samples can be occupied by Co ions with different valences. As a result, a net electric polarizations can be induced with both the ab plane and c -axis components. It can be seen that in the unit cell of both $x = 0.75$ and 0.25 there is only a pair of Co^{2+} and Co^{3+} trigonal prisms, of which the electric dipoles can not be canceled with each other. This can be the reason that the electric polarizations of the $x = 0.72$ and 0.26 samples have nearly the same value.

It should be pointed out that there is a similar anomaly of Raman shift at 25 K for the $x = 0$ sample, as shown in Fig. 14(a). This anomaly is likely also caused by the Jahn-Teller distortion due to unevenly filled degenerate $3d$ states of the trigonal prisms Co^{3+} ions [34]. The distortion could be similar to either the $x = 1$ sample or $x = 0.72$ (0.26) sample at low temperatures. If it is similar to the $x = 1$ sample, the positive-charge and negative-charge centers of the Co^{3+}O_6 trigonal prisms are still coincident. It does not

destroy the spatial inversion symmetry in this case. If it is similar to the $x = 0.72$ (0.26) sample, the positive-charge and negative-charge centers of the trigonal prisms would not be coincident. However, all of the Co ions in $x = 0$ sample have the valence of trivalence, there is also no net electric polarizations induced in this case.

IV. CONCLUSION

$\text{Ca}_3\text{Co}_{2-x}\text{Mn}_x\text{O}_6$ single crystals with $x = 0.72$ and 0.26 were studied by various measurements. The dc and ac magnetic susceptibility and specific heat are characteristic of the spin freezing. The electric polarization along the spin-chain direction reaches a large value of $1400 \mu\text{C}/\text{m}^2$ at 8 K and the transition temperature is near 40 and 35 K for the $x = 0.72$ and 0.26 samples, respectively. Interestingly, an electric polarization perpendicular to the c direction was also detected. However, the specific heat and magnetic susceptibility do not show anomaly at T_{FE} . It means that the electric polarization has no direct relationship with the magnetism. The low-temperature x-ray diffraction and the Raman spectroscopy indicate that these samples may undergo a Jahn-Teller distortion, which could be the reason of anomalous electric polarization. Finally, it is worthy of noting that these samples display a special case of co-existence of ferroelectricity and spin glass, which would be interesting for further investigations.

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- [1] S.-W. Cheong and M. Mostovoy, *Nat. Mater.* **6**, 13 (2007).
 - [2] T. Kimura, T. Goto, H. Shintani, K. Ishizaka, T. Arima, and Y. Tokura, *Nature (London)* **426**, 55 (2003).
 - [3] S. Park, Y. J. Choi, C. L. Zhang, and S.-W. Cheong, *Phys. Rev. Lett.* **98**, 057601 (2007).
 - [4] Y. Yamasaki, S. Miyasaka, Y. Kaneko, J.-P. He, T. Arima, and Y. Tokura, *Phys. Rev. Lett.* **96**, 207204 (2006).
 - [5] G. Lawes, A. B. Harris, T. Kimura, N. Rogado, R. J. Cava, A. Aharony, O. Entin-Wohlman, T. Yildirim, M. Kenzelmann, C. Broholm, and A. P. Ramirez, *Phys. Rev. Lett.* **95**, 087205 (2005).
 - [6] Y. J. Choi, H. T. Yi, S. Lee, Q. Huang, V. Kiryukhin, and S.-W. Cheong, *Phys. Rev. Lett.* **100**, 047601 (2008).
 - [7] H. Wu, T. Burnus, Z. Hu, C. Martin, A. Maignan, J. C. Cezar, A. Tanaka, N. B. Brookes, D. I. Khomskii, and L. H. Tjeng, *Phys. Rev. Lett.* **102**, 026404 (2009).
 - [8] Y. J. Jo, S. Lee, E. S. Choi, H. T. Yi, W. Ratcliff, Y. J. Choi, V. Kiryukhin, S.-W. Cheong, and L. Balicas, *Phys. Rev. B* **79**, 012407 (2009).
 - [9] J. W. Kim, Y. Kamiya, E. D. Mun, M. Jaime, N. Harrison, J. D. Thompson, V. Kiryukhin, H. T. Yi, Y. S. Oh, S. W. Cheong, C. D. Batista, and V. S. Zapf, *Phys. Rev. B* **89**, 060404(R) (2014).
 - [10] L. Lin, Y. J. Guo, Y. L. Xie, S. Dong, Z. B. Yan, and J. M. Liu, *J. Appl. Phys.* **111**, 07D901 (2012).
 - [11] Z. W. Ouyang, S. S. Sheng, J. Chen, X. M. Shi, M. Y. Ruan, and Z. C. Xia, *J. Appl. Phys.* **112**, 103923 (2012).
 - [12] M. Y. Ruan, Z. W. Ouyang, Y. M. Guo, Y. C. Sun, J. J. Cheng, Z. C. Xia, and G. H. Rao, *J. Alloys Compd.* **620**, 97 (2015).
 - [13] S. S. Sheng, Z. W. Ouyang, J. Chen, M. Y. Ruan, X. M. Shi, and Z. C. Xia, *J. Alloys Compd.* **556**, 287 (2013).
 - [14] A. Maignan, C. Michel, A. C. Masset, C. Martin, and B. Raveau, *Eur. Phys. J. B* **15**, 657 (2000).
 - [15] V. G. Zubkov, G. V. Bazuev, A. P. Tyutyunnik, and I. F. Berger, *J. Solid State Chem.* **160**, 293 (2001).

- [16] R. Flint, H.-T. Yi, P. Chandra, S.-W. Cheong, and V. Kiryukhin, *Phys. Rev. B* **81**, 092402 (2010).
- [17] S. Rayaprol, K. Sengupta, and E. V. Sampathkumaran, *Solid State Commun.* **128**, 79 (2003).
- [18] V. Kiryukhin, S. Lee, W. Ratcliff, Q. Huang, H. T. Yi, Y. J. Choi, and S.-W. Cheong, *Phys. Rev. Lett.* **102**, 187202 (2009).
- [19] M. Mezard, G. Parisi, and M. A. Virasoro, *Spin-Glass: Theory and Beyond* (World Scientific, Singapore, 1987).
- [20] T. Basu, K. K. Iyer, K. Singh, and E. V. Sampathkumaran, *Sci. Rep.* **3**, 3104 (2013).
- [21] K. Okazaki and H. Maiwa, *Jpn. J. Appl. Phys.* **31**, 3113 (1992).
- [22] K. R. S. Preethi Meher, A. Wahl, A. Maignan, C. Martin, and O. I. Lebedev, *Phys. Rev. B* **89**, 144401 (2014).
- [23] B. Rajeswaran, D. I. Khomskii, A. K. Zvezdin, C. N. R. Rao, and A. Sundaresan, *Phys. Rev. B* **86**, 214409 (2012).
- [24] K. Ramesha, A. Llobet, Th. Proffen, C. R. Serrao, and C. N. R. Rao, *J. Phys.: Condens. Matt.* **19**, 102202 (2007).
- [25] M. El. Amrani, M. Zaghrioui, V. Ta. Phuoc, F. Gervais, and N. E. Massa, *J. Magn. Magn. Mater.* **361**, 1 (2014).
- [26] X. Wang, Y. S. Chai, L. Zhou, H. Cao, C. Cruz, J. Y. Yang, J. H. Dai, Y. Y. Yin, Z. Yuan, S. Zhang, R. Yu, M. Azuma, Y. Shimakawa, H. Zhang, S. Dong, Y. Sun, C. Q. Jin, and Y. W. Long, *Phys. Rev. Lett.* **115**, 087601 (2015).
- [27] Y. Zhang, H. J. Xiang, and M.-H. Whangbo, *Phys. Rev. B* **79**, 054432 (2009).
- [28] S. Gohil, K. K. Iyer, P. Aswathi, S. Ghosh, and E. V. Sampathkumaran, *J. Appl. Phys.* **108**, 103517 (2010).
- [29] R. Bindu, K. Maiti, S. Khalid, and E. V. Sampathkumaran, *Phys. Rev. B* **79**, 094103 (2009).
- [30] P. L. Paulose, N. Mohapatra, and E. V. Sampathkumaran, *Phys. Rev. B* **77**, 172403 (2008).
- [31] L. H. Yin, J. Yang, R. R. Zhang, J. M. Dai, W. H. Song, and Y. P. Sun, *Appl. Phys. Lett.* **104**, 032904 (2014).
- [32] P. Mandal, V. S. Bhadram, Y. Sundarayya, C. Narayana, A. Sundaresan, and C. N. R. Rao, *Phys. Rev. Lett.* **107**, 137202 (2011).
- [33] J. Echeverria, E. Cremades, A. J. Amoroso, and S. Alvarez, *Chem. Commun.* **28**, 4242 (2009).
- [34] Y. Zhang, E. Kan, H. J. Xiang, A. Villesuzanne, and M.-H. Whangbo, *Inorg. Chem.* **50**, 1758 (2011).