Photoemission study of the electronic structure of the Kondo lattices $Yb_2Pt_6X_{15}$ (X=Al, Ga)

Awabaikeli Rousuli,^{1,*} Shogo Nakamura,² Hitoshi Sato,^{3,†} Takuya Ueda,⁴ Yuji Matsumoto,^{4,‡} Shigeo Ohara,⁴ Eike F. Schwier,³

Toshiki Nagasaki,¹ Kojiro Mimura,⁵ Hiroaki Anzai,⁵ Katsuya Ichiki,⁵ Shigenori Ueda,^{6,7} Kenya Shimada,³

Hirofumi Namatame,³ and Masaki Taniguchi³

¹Graduate School of Science, Hiroshima University, Higashi-Hiroshima 739-8526, Japan

²Department of Materials Science and Engineering, Tokyo Institute of Technology, Yokohama 226-8502, Japan

³Hiroshima Synchrotron Radiation Center, Hiroshima University, Higashi-Hiroshima 739-0046, Japan

⁴Graduate School of Engineering, Nagoya Institute of Technology, Nagoya 466-8555, Japan

⁵Graduate School of Engineering, Osaka Prefecture University, Sakai 599-8531, Japan

⁶Synchrotron X-ray Station at SPring-8, National Institute for Materials Science, Hyogo 679-5148, Japan

 7Q uantum Beam Unit, National Institute for Materials Science, Tsukuba 305-0047, Japan

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The electronic structure of Yb-based Kondo lattices Yb₂Pt₆X₁₅ (X=Al, Ga) has been investigated by means of photoemission spectroscopy using hard x-ray (hv = 5.95 keV) and vacuum ultraviolet (hv = 182 eV) synchrotron radiation. The Yb 3*d* spectra of Yb₂Pt₆X₁₅ showed both Yb²⁺- and Yb³⁺-derived structures, directly indicating valence fluctuation. The Yb valences of Yb₂Pt₆Al₁₅ were estimated to be 2.89 ± 0.01 at 250 K and ~2.83 at 20 K, while those of Yb₂Pt₆Ga₁₅ were ~2.34 at 300 K with almost no temperature dependence. With changing X ions from Al to Ga, the Pt 5*d* and Pt 4*f* peaks were shifted to shallower binding energies, and the Yb³⁺ 4*f* peaks were shifted to the deeper binding energies. The X dependences of the Yb valence and the Kondo temperature of Yb₂Pt₆X₁₅ are discussed based on the Yb³⁺ 4*f* hole level relative to the Fermi level E_F and Pt-derived density of states at E_F .

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I. INTRODUCTION

Yb-based Kondo lattices show a wide variety of intriguing physical properties such as heavy-fermion behavior, valence fluctuation, and superconductivity. The ground state in such systems is believed to be qualitatively determined by the competition between the intersite Ruderman-Kittel-Kasuya-Yosida (RKKY) interaction and the on-site Kondo effect, both arising from the hybridization between the localized Yb 4 f electrons and itinerant conduction electrons (c-f)hybridization). When the RKKY interaction is dominant, the 4 f electrons interact indirectly with each other mediated by the conduction electrons, and a magnetic ordered state is realized at low temperature. On the other side, when the Kondo effect is dominant, the 4f moments are screened by the conduction electrons, leading to a nonmagnetic Fermi-liquid ground state. This situation is summarized in the so-called Doniach phase diagram [1]. The boundary point at zero temperature separating the magnetic and nonmagnetic ground-state regions defines the quantum critical point (QCP).

The Kondo lattices Yb₂Pt₆X₁₅ (X=Al, Ga) have a quasitwo-dimensional Sc_{1.2}Fe₄Si_{9.8}-type crystal structure with space group $P6_3/mmc$ [2] or Cmcm [3]. Several transport measurements were carried out for Yb₂Pt₆Al₁₅ [2]. The magnetic susceptibility at high temperature shows the Curie-Weiss behavior with an effective magnetic moment of $\mu_{eff} \sim$ $4.1\mu_B$, close to the value of the free Yb³⁺ ion. After reaching its maximum at $T_{max} \sim 30$ K, the susceptibility drops and becomes almost temperature independent at lower temperature with a paramagnetic ground state. The experimental data are well fitted with the Coqblin-Schrieffer model for J = 7/2 of Yb³⁺ with a Kondo temperature of $T_K = 64$ K [2].

The physical properties of Yb₂Pt₆Ga₁₅ have not been reported except for the lattice constant and heat-capacity measurements on a series of R_2 Pt₆Ga₁₅ (R = rare earth) [4]. The lattice constants of Yb₂Pt₆Ga₁₅ deviate to larger values from the monotonic shrink curve from R = La to Lu expected from the lanthanide contraction, suggesting a nearly divalent Yb in this compound. From the magnetic susceptibility measurements (see Fig. 3 below), T_K is suggested to be higher than 1000 K.

Utsumi et al. investigated the electronic structure of the Kondo lattices $YbNi_3X_9$ by means of hard x-ray photoemission spectroscopy (HAXPES) [5]. Its crystal structure is $\operatorname{ErNi}_{3}X_{9}$ type (space group R32), similar to that of $\operatorname{Yb}_{2}\operatorname{Pt}_{6}X_{15}$, as described later. In spite of the same crystal structure and similar nominal electron configurations (Al $3s^23p^1$ and Ga $4s^24p^1$), YbNi₃Al₉ exhibits a chiral helical order with a Néel temperature of $T_N = 3.4$ K [6], while YbNi₃Ga₉ exhibits valence fluctuation within the nonmagnetic ground state [7,8]. The T_K values of YbNi₃X₉ change from 2.7 K for YbNi₃Al₉ to 570 K for YbNi₃Ga₉. Thus, depending on X, YbNi₃X₉ is located in different phases separated by the QCP in the Doniach phase diagram. It should be noted, however, that YbNi₃Ga₉ shows pressure-induced antiferromagnetic order at 9 GPa [9], demonstrating its state being close to the QCP. Based on the HAXPES results, Utsumi et al. interpreted that the increased Ni 3d density of states (DOS) at E_F and the 4f hole level closer to E_F for YbNi₃Ga₉ lead to an increase in T_K and induce the strong valence fluctuation [5].

Although $Yb_2Pt_6X_{15}$ compounds are located in the nonmagnetic region of the phase diagram, the X dependence

^{*}d144409@hiroshima-u.ac.jp

[†]jinjin@hiroshima-u.ac.jp

[‡]Present address: Graduate School of Science and Engineering, University of Toyama, Toyama 930-8555, Japan.



FIG. 1. (a) Yb_2X_3 layer and (b) local coordination around the Yb ion of $Yb_2Pt_6X_{15}$ and $YbNi_3X_9$.

of T_K similar to that for YbNi₃X₉ indicates that Yb₂Pt₆X₁₅ compounds occupy different positions: close to the QCP for Yb₂Pt₆Al₁₅ but farther away from the QCP for Yb₂Pt₆Ga₁₅. In addition to the similar X dependences of T_K , the two systems have similar crystal structures in that the Yb₂X₃ layer shown in Fig. 1(a) is sandwiched between two X-Pt(Ni)-X layers as X-Pt(Ni)-X-Yb₂X₃-X-Pt(Ni)-X, where the X and Pt(Ni) layers form triangular lattices. This unit of layers stacks along the *c* axis in Yb₂Pt₆X₁₅, and the X layer is located between the two units of layers in YbNi₃X₉. The local coordination around the Yb ion is similar between the two systems, as shown in Fig. 1(b).

Pt and Ni have 5d and 3d electrons, respectively, and belong to the same family in the periodic table. The physical properties of $Yb_2Pt_6X_{15}$ and $YbNi_3X_9$ are expected to be interpreted mainly from the characters of the 5d and 3d bands. $Yb_2Pt_6X_{15}$ is thus a suitable system to extend the studies on YbNi₃ X_9 and provides an opportunity to systematically investigate the relation between the electronic structure and physical properties of Yb compounds. In this study, we carry out HAXPES with hv = 5.95 keV and vacuum ultraviolet photoemission spectroscopy (VUVPES) with hv = 182 eV on $Yb_2Pt_6X_{15}$ and investigate the X dependence of the electronic structure in comparison with YbNi₃ X_9 [5]. The HAXPES and VUVPES results reveal the similarity in the X dependences of the electronic structure between $Yb_2Pt_6X_{15}$ and $YbNi_3X_9$. The Yb 3d spectra show valence fluctuation behavior of $Yb_2Pt_6X_{15}$. The Yb valence of $Yb_2Pt_6Al_{15}$ decreases from 2.89 ± 0.01 (T = 250 K) to 2.83 (T = 20 K), while that of $Yb_2Pt_6Ga_{15}$ is ~2.34, with no significant temperature dependence. The Pt 5d and Pt 4f peaks of $Yb_2Pt_6Ga_{15}$ are shifted to the shallower binding energy E_B , while the Yb³⁺ 4f peaks are found at deeper E_B compared to those of Yb₂Pt₆Al₁₅. We explain the differences in the Yb valence and T_K in $Yb_2Pt_6X_{15}$ via the $Yb^{3+} 4f$ hole level above the Fermi level E_F and the Pt-derived DOS at E_F by making comparisons to previous results of YbNi₃ X_9 [5]. From the combined HAXPES results of Yb₂Pt₆ X_{15} and YbNi₃ X_9 [5], we find correlations among T_K , the Yb valence, and Yb³⁺ 4*f* peak energy. We discuss the higher T_K for Yb₂Pt₆ X_{15} compared to YbNi₃ X_9 based on the DOS at E_F derived from the broad Pt 5d band.

II. EXPERIMENTS

Single crystals of $Yb_2Pt_6X_{15}$ (X = Al, Ga) were synthesized by means of the self-flux method. The starting materials were mixed with a ratio of Yb:Pt:X = 1:3:30 and put in an alumina crucible sealed in an evacuated quartz tube. For

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FIG. 2. XRD patterns (thick lines) of $Yb_2Pt_6X_{15}$ in comparison to calculated ones derived from the fits with the $Sc_{1.2}Fe_4Si_{9.8}$ -type crystal structure (thin lines).

Yb₂Pt₆Al₁₅, the ampoule was heated to 1150 °C and kept at that temperature for 5 h. Then it was cooled to 850 °C over 1 h, again heated to 900 °C over 1 h, and cooled to 750 °C over 30 h. The excess Al was removed in a centrifuge and further washed out in a NaOH solution. The growth method of Yb₂Pt₆Ga₁₅ is described in Ref. [4]. The single phase of the grown sample was verified by x-ray powder diffraction (XRD). No XRD peak due to impurity phases was detected, as shown in Fig. 2.

Figure 3 shows the magnetic susceptibilities of $Yb_2Pt_6Ga_{15}$ under magnetic fields of 10 kOe measured using a superconducting quantum interference device magnetometer (Quantum Design). The two results under magnetic fields applied parallel to the *a* and *c* axes exhibit similar temperature dependences with weak anisotropy. The susceptibility continuously de-



FIG. 3. Magnetic susceptibilities of $Yb_2Pt_6Ga_{15}$ under magnetic fields of 10 kOe parallel to the *a* and *c* axes.

creases on cooling from 330 K and becomes almost constant below 200 K. The ground state is paramagnetic, as is the case in Yb₂Pt₆Al₁₅. Based on these measurements and by scaling the susceptibility for Yb₂Pt₆Al₁₅, T_{max} in Yb₂Pt₆Ga₁₅ has to be located above 350 K, which in turn gives a lower limit for T_K of at least 1000 K from $3T_{max}$. Below 30 K, the magnetic susceptibilities rapidly increase, probably caused by a small amount of magnetic impurity, such as Yb₂O₃. However, the impurities were not identified within the XRD patterns in Fig. 2.

The HAXPES experiments were performed at undulator beamline BL15XU [10] of SPring-8. The excitation energy was hv = 5.95 keV monochromatized using a Si 111 double crystal and Si 333 channel-cut monochromators [11]. A high-energy-resolution hemispherical photoelectron analyzer (VG Scienta R4000) was used to collect the angle-integrated HAXPES spectra. The total energy resolution was set to 240 meV. The angle-integrated VUVPES spectra were measured with hv = 182 eV at bending beamline BL-7 of the Hiroshima Synchrotron Radiation Center (HSRC) equipped with a Dragon-type monochromator [12]. A hemispherical photoelectron analyzer (VG Scienta SES2002) was used, and the total energy resolution was set to 150 meV. E_B of the HAXPES and VUVPES spectra was calibrated by measuring the Fermi edge of polycrystalline Au. Clean surfaces were obtained by fracturing in situ.

III. RESULTS AND DISCUSSION

It is well known that line shape analysis of the Yb 3d HAXPES spectra provides the mean Yb valence with high accuracy [13]. Temperature-dependent Yb 3d spectra from Yb₂Pt₆Al₁₅ and Yb₂Pt₆Ga₁₅ are displayed in Figs. 4(a) and 4(b), respectively. The Yb 3d spectra are composed of the $3d_{5/2}$ part at 1515–1540 eV and $3d_{3/2}$ part at 1560–1590 eV [see Fig. 4(b)]. In the case of Yb₂Pt₆Al₁₅, an intense Al 1*s* peak is visible at 1559.5 eV, and the tail strongly disturbs the spectral feature in the $3d_{3/2}$ region. Therefore, we measured only the $3d_{5/2}$ spectra for Yb₂Pt₆Al₁₅, as shown in Fig. 4(a).

The Yb $3d_{5/2}$ spectra of Yb₂Pt₆Al₁₅ exhibit clear temperature dependence between 250 and 20 K. The spectral feature is similar to that of other Yb compounds [5,13–15]. With decreasing temperature, the Yb²⁺-derived single peak at 1520 eV is gradually enhanced, while the Yb³⁺-derived multiplet structures due to the $3d^94f^{13}$ final states at 1524–1536 eV are reduced. The temperature dependence of the spectra indicates that the Yb valence is shifted to the divalent side at low temperature, which is clear evidence of valence fluctuation in Yb₂Pt₆Al₁₅.

In contrast, the temperature dependence of the Yb 3d spectra of Yb₂Pt₆Ga₁₅ in Fig. 4(b) is much weaker. The $3d_{3/2}$ region is also divided into the Yb²⁺ peak at 1568 eV and Yb³⁺ multiplet structures at 1574–1582 eV. The intensities of the Yb³⁺ structures are significantly weak compared to the Yb²⁺ peaks, indicating the Yb ion in Yb₂Pt₆Ga₁₅ is closer to divalent. The weak structures shown by the vertical bars at 1537 and 1584 eV come from the Yb²⁺-derived plasmon satellites [5].

Since the Yb²⁺ and Yb³⁺ components are clearly resolved in the Yb 3*d* spectrum, the Yb valence can be precisely deduced from their integral intensity ratio. In order to estimate



FIG. 4. (a) Temperature dependences of Yb $3d_{5/2}$ HAXPES spectra of Yb₂Pt₆Al₁₅ measured between 250 and 20 K. (b) Temperature dependences of Yb 3*d* HAXPES spectra of Yb₂Pt₆Ga₁₅ measured between 300 and 20 K. Vertical bars indicate the plasmon satellite structures.

the Yb valence, we fitted the Yb 3d spectra with the procedure described in Ref. [13]. The fit of the spectrum of $Yb_2Pt_6Ga_{15}$ measured at 20 K is shown in Fig. 5(a) as an example. In the decomposed spectrum in the lower part, the broad curves around 1544 and 1597 eV correspond to the Yb³⁺-derived plasmon satellites. The evaluated Yb valences of $Yb_2Pt_6X_{15}$ are summarized in Fig. 5(b). The Yb valence of $Yb_2Pt_6Al_{15}$ is 2.89 ± 0.01 at 250 K and gradually decreases on cooling to 2.83 at 20 K. On the other hand, the valence of Yb₂Pt₆Ga₁₅ is \sim 2.34 in whole temperature range. The different temperature dependences of the valence in the two compounds can qualitatively be interpreted from their different T_K . According to a theoretical calculation using the noncrossing approximation, the Yb valence changes significantly around T_K [16]. Since T_K in Yb₂Pt₆Al₁₅ is ~60K [2], its valence changes within the measured temperature region, while that of Yb₂Pt₆Ga₁₅ with $T_K > 1000$ K is almost constant below 300 K. The constant Yb valence (\sim 2.2) below 300 K is also reported for YbAl₂ with $T_K = 2000-2600$ K [17].

In order to examine the electronic states near E_F including the Yb 4*f* states, we measured the valence-band HAXPES spectra of Yb₂Pt₆X₁₅, as shown in Fig. 6(a). A dashed line on the spectrum of Yb₂Pt₆Al₁₅ represents that of Yb₂Pt₆Ga₁₅



FIG. 5. (a) Fit of the Yb 3*d* HAXPES spectrum of Yb₂Pt₆Ga₁₅ measured at 20 K. (b) Yb valences of Yb₂Pt₆ X_{15} estimated from the fits of the spectra as a function of temperature.

normalized with the intense peak at 4-5 eV for comparison. Taking into account the photoionization cross sections at $h\nu =$ 6 keV [18], the spectra are mainly derived from the Pt 5d, Yb 4f, Ga 4s, and Al 3s states with a ratio of 14.5:2.6:3.3:1. Note that the delocalized s states are distributed in a wide E_B range, while the localized d and f states are in a narrow E_B range. The most prominent features at 3-6 eV are ascribed to the Pt 5dstates, which is confirmed by the VUVPES spectra measured at $h\nu = 182$ eV [see Fig. 6(b)]. The Yb²⁺ 4f_{7/2} and 4f_{5/2} states are observed in the vicinity of E_F with a spin-orbit splitting of about 1.3 eV. On the other hand, the Yb³⁺ 4f multiplet structures due to the $4f^{12}$ final states are buried in the intense Pt 5d peak above 5 eV. Some $Yb^{3+} 4f$ peaks are observed in the Yb₂Pt₆Al₁₅ spectrum, as indicated by vertical bars. Almost no structure is found in the Yb₂Pt₆Ga₁₅ spectrum in the corresponding region, reflecting that the Yb valence is close to divalent. Only one peak is very weakly detected at 11.3 eV, as we see in the enlarged spectrum in Fig. 6(a). This peak is considered to correspond to the peak at 10.7 eV in the Yb₂Pt₆Al₁₅ spectrum.

In order to clearly detect the Yb³⁺ 4*f* multiplet structures, in particular, for Yb₂Pt₆Ga₁₅, we measured valence-band VUVPES spectra at $h\nu = 182$ eV, where the Yb³⁺ 4*f* structures are enhanced by the Yb 4*d*-4*f* resonance [19]. The VUVPES spectra of Yb₂Pt₆X₁₅ are compared in Fig. 6(b). Due to the resonant enhancement, we clearly see the Yb³⁺ 4*f* multiplet structures at 5–13 eV, in particular, in the Yb₂Pt₆Al₁₅



FIG. 6. (a) Valence-band HAXPES spectra of $Yb_2Pt_6X_{15}$ measured at 20 K. The dashed line on the spectrum of $Yb_2Pt_6Al_{15}$ represents that of $Yb_2Pt_6Ga_{15}$ for comparison. Vertical bars indicate the Yb^{3+} 4*f*-derived peaks. (b) Valence-band VUVPES spectra of $Yb_2Pt_6X_{15}$ measured at 20 K. The bulk and surface Yb^{2+} 4*f* components are indicated with "B" and "S", respectively. Two peaks of the enlarged spectrum of $Yb_2Pt_6Ga_{15}$ shown by vertical bars correspond to those of the HAXPES spectrum.

spectrum. The overall feature including the relative intensity is consistent with that observed for other Yb compounds [20]. The shallowest peak at 5.8 eV, which is completely overlapped by the Pt 5*d* peak in the HAXPES spectrum in Fig. 6(a), is ascribed to the ³*H*₆ multiplet. Although the Yb³⁺ 4*f* structures are still weak in the Yb₂Pt₆Ga₁₅ spectrum, they are clearly observed in the enlarged spectrum. A peak at 6.5 eV is attributed to the ³*H*₆ multiplet, and a peak at 11.3 eV corresponds to the 10.7 eV peak in the Yb₂Pt₆Al₁₅ spectrum.

The Yb²⁺ 4*f* region between E_F and 3.5 eV in the VUVPES spectra is composed of the surface components as well as the bulk components because of the surface sensitivity of VUVPES. The peak just below E_F is due to bulk Yb 4 $f_{7/2}$ states, as observed in the bulk-sensitive HAXPES spectra in Fig. 6(a). In contrast, the peak around 2.5–3.0 eV is due to surface $4f_{5/2}$ states since they are not detected in the HAXPES spectra. The bulk $4f_{5/2}$ and surface $4f_{7/2}$ states contribute to the structure around 1.3 eV and are resolved

in the case of Yb₂Pt₆Ga₁₅; the shallower peak at 1.3 eV is derived from the surface $4f_{7/2}$ states, and the shoulder at 1.45 eV is derived from the bulk $4f_{5/2}$ states. We see that the bulk- and surface-originated electronic states almost equally contribute to the VUVPES spectra from their intensity ratio. Both components should also contribute to the Yb³⁺ 4*f* spectra, although they are not resolved. The Yb³⁺ 4*f* peaks of YbRh₂Si₂ measured at $h\nu = 115$ -800 eV are clearly split into the bulk and surface components, and the bulk-derived peak is located at 0.11 eV shallower E_B than the surface-derived peak [21].

There is almost no structure around 5 eV in the VUVPES spectra of both compounds. As indicated by vertical bars in the enlarged spectrum of Yb₂Pt₆Ga₁₅, the two peaks are detected at corresponding energies of the Pt 5*d*-attributed peaks in the HAXPES spectrum. Already in the off-resonant case, the contribution by the Pt 5*d*, Ga 4*s*, and Al 3*s* states to the spectra is less than 5% compared to Yb 4*f* at hv = 180 eV [22]. The Yb 4*f* signal is further increased by the resonant condition, making Yb 4*f* the dominant contribution. The VUVPES therefore clarifies that the most noticeable peak at 3–6 eV in the HAXPES spectra originates from the Pt 5*d* states.

The Yb²⁺ 4 f peaks of Yb₂Pt₆Ga₁₅ are located away from E_F compared to Yb₂Pt₆Al₁₅. Since the Yb²⁺ 4f_{7/2} peak is assigned to the Kondo peak [17], the deeper E_B reflects the higher T_K in Yb₂Pt₆Ga₁₅; the peak position of 0.18 eV corresponds to $T_K \sim 2100$ K. The Yb³⁺ 4 f multiplet structures for Yb₂Pt₆Ga₁₅ are also shifted to deeper E_B by $\Delta E_B = 0.6$ eV. Its energy position is related to the Yb³⁺ 4f hole level ($\varepsilon_f > 0$) above E_F , as discussed in Ref. [5]. In the multielectron picture, ε_f corresponds to the energy for the $f^{13} \rightarrow f^{14}$ transition, and the Yb³⁺ 4f peaks in the photoemission spectra ($f^{13} \rightarrow f^{12}$ transition) are roughly located at $-\varepsilon_f + U$, with U being the Coulomb repulsion energy between the 4 f holes in the f^{12} final states. Since U is expected to be almost intra-atomic and unchanged between Yb₂Pt₆Al₁₅ and Yb₂Pt₆Ga₁₅, the deeper E_B of the Yb³⁺ 4f peaks for Yb₂Pt₆Ga₁₅ suggests smaller ε_f , that is, the Yb³⁺ 4f hole level being closer to E_F . This observation is consistent with the nearly divalent Yb in Yb2Pt6Ga15 because of the lower energy required for the transfer of the conduction electrons to the Yb³⁺ 4 f hole. In the Pt 5d peak a shift to shallower E_B by $\Delta E_B = 0.3$ eV is detected for Yb₂Pt₆Ga₁₅.

In order to investigate the X dependence of the conduction states from the peak shift of the core levels, we measured the Pt 4f HAXPES spectra of Yb₂Pt₆X₁₅ at 20 K, as shown in Fig. 7. The Pt 4 $f_{7/2}$ and 4 $f_{5/2}$ peaks of Yb₂Pt₆Ga₁₅ are located at 72.0 and 75.4 eV, respectively, with shallower E_B than those of Yb₂Pt₆Al₁₅ by $\Delta E_B = 0.3$ eV. The direction of the energy shift is the same as observed for the Pt 5d peaks in Fig. 6(a), and their amounts are comparable, which indicates that the E_F position of the conduction-band DOS moves to the lower-energy side on going from Yb₂Pt₆Al₁₅ to Yb₂Pt₆Ga₁₅.

The HAXPES results of Yb₂Pt₆ X_{15} are comparable to those previously reported for YbNi₃ X_9 [5] with a similar X dependence of T_K (2.7 K for YbNi₃Al₉ and 570 K for YbNi₃Ga₉ [7,8]). Note again that Yb₂Pt₆ X_{15} and YbNi₃ X_9 possess similar crystal structures and local coordinates around the Yb ion. Yb 3*d* HAXPES revealed that YbNi₃Ga₉ exhibits



FIG. 7. Pt 4 f HAXPES spectra of Yb₂Pt₆ X_{15} measured at 20 K.

strong valence fluctuation behavior with Yb valences of 2.59 at 300 K and 2.43 at 22 K. The valence of YbNi₃Al₉ is nearly 3 (\sim 2.97), with little temperature dependence. In both systems, the valence is shifted to the divalent side with exchanging Al with Ga.

Similar X dependences are observed in the core level and valence-band HAXPES spectra as well as the Yb valence for Yb₂Pt₆X₁₅ and YbNi₃X₉. Figures 8(a) and 8(b) exhibit the Ni $2p_{3/2}$ and valence-band spectra of YbNi₃X₉, respectively [5], and Figs. 8(c) and 8(d) show the Pt $4f_{7/2}$ and valence-band spectra of Yb₂Pt₆X₁₅ for comparison. The Ni $2p_{3/2}$ peak of YbNi₃Ga₉ is shifted to shallower E_B by $\Delta E_B = 0.3$ eV, and



FIG. 8. Comparison of HAXPES spectra measured at 20 K between Yb₂Pt₆ X_{15} and YbNi₃ X_9 . (a) Ni $2p_{3/2}$ and (b) valence-band spectra of YbNi₃ X_9 , and (c) Pt $4f_{7/2}$ and (d) valence-band spectra of Yb₂Pt₆ X_{15} . Thin lines in (d) depict VUVPES spectra in the Yb³⁺ 4f region.



FIG. 9. Yb valence plotted for (a) the ${}^{3}H_{6}$ peak position and (b) T_{K} for Yb₂Pt₆ X_{15} together with those of YbNi₃ X_{9} [5]. Dashed lines are guides for eye.

so is the Ni 3*d* peak within the valence bands. By contrast, the Yb³⁺ 4*f* multiplet structures exhibit an opposite energy shift by $\Delta E_B = 0.4$ eV. Based on the HAXPES results, Utsumi *et al.* described the *X* dependence of physical properties of YbNi₃X₉ by the differences of the Ni 3*d* DOS at E_F and the Yb³⁺ 4*f* hole level relative to E_F [5]. The increased Ni 3*d* DOS at E_F and the 4*f* hole level closer to E_F for YbNi₃Ga₉ enhance the *c*-*f* hybridization, that is, increase T_K and induce the strong valence fluctuation.

The X dependences of the Yb valence and the energy shifts in the Pt 4 f, Pt 5d, and Yb³⁺ 4 f peaks found for Yb₂Pt₆ X_{15} are similar to those for $YbNi_3X_9$ [5]. For both cases, the substitution of an X ion from Al to Ga decreases the Yb valence and shifts the Pt 4 f and Pt 5d (Ni 2 $p_{3/2}$ and Ni 3d) peaks toward shallower E_B and the Yb³⁺ 4f peaks toward deeper E_B . In line with the argument for YbNi₃X₉ [5], the increased Pt 5d DOS at E_F and Yb³⁺ 4f hole level closer to E_F for Yb₂Pt₆Ga₁₅ increase the *c*-*f* hybridization and T_K and decrease the Yb valence. Although the Pt 5d states mainly distribute at 3–6 eV in Fig. 6(a), we see that the Pt 5d DOS has a long tail toward E_F . Assuming a similar tail feature for both compounds, Yb₂Pt₆Ga₁₅ with its shallower Pt 5d peak is expected to have a larger Pt 5d DOS at E_F , which might be reflected in the increased spectral intensity around 1 eV in $Yb_2Pt_6Ga_{15}$. The similarities between the X-dependent spectra of YbNi₃ X_9 and Yb₂Pt₆ X_{15} suggest some systematic changes in the electronic structure when compounds with the same crystal structure and similar conduction electron states move from nonmagnetic to magnetic regions in the Doniach phase diagram. The same trend is also found for antiferromagnetic YbNiSi₃ and nonmagnetic YbNiGe₃ [23].

Finally, we compare the electronic structures of Yb₂Pt₆ X_{15} and YbNi₃ X_9 [5]. Here, recall T_K in these materials is 2.7 K for YbNi₃Al₉, 64 K for Yb₂Pt₆Al₁₅, 570 K for YbNi₃Ga₉, and higher than 1000 K for Yb₂Pt₆Ga₁₅, ~2100K derived from the Yb²⁺ 4 $f_{7/2}$ peak position. The difference in T_K is reflected well by the systematic changes in the Yb valence and the energy position of the ³ H_6 peak. The valence at 300 K decreases in the order of YbNi₃Al₉ (2.97), Yb₂Pt₆Al₁₅ (2.89), YbNi₃Ga₉ (2.59), and Yb₂Pt₆Ga₁₅ (2.34), and the ³ H_6 peak energies are 5.5, 5.8, 6.0, and 6.6 eV, respectively [24]. Figure 9(a) clearly indicates a correlation between the Yb valence and ³ H_6 peak energy for Yb₂Pt₆ X_{15} and YbNi₃ X_9 . This trend can qualitatively be understood assuming that the deeper ${}^{3}H_{6}$ peak reflects the Yb³⁺ 4*f* hole level closer to E_{F} , as mentioned above. The conduction electrons are more readily transferred into the Yb³⁺ 4*f* hole, and the Yb valence is shifted to the divalent side. The Yb valence is also plotted for T_{K} in Fig. 9(b), and we notice that the valence is systematically shifted to the divalent side with increasing T_{K} . In order to quantitatively explain the correlation among T_{K} , the Yb valence, and ${}^{3}H_{6}$ peak energy, theoretical analyses of the spectra based on the single-impurity Anderson model with parameters such as the Yb 4*f* level, 4*f*-4*f* Coulomb interaction energy, and *c*-*f* hybridization energy are required.

The higher T_K in Yb₂Pt₆ X_{15} indicates that the *c*-*f* hybridization is larger than that for $YbNi_3X_9$. The larger c-f hybridization is reflected in Yb₂Pt₆Al₁₅ with no magnetic order, while YbNi₃Al₉ exhibits chiral helical order. Based on our observations, the increased Pt 5d and Ni 3d DOSs at E_F are important for describing the higher T_K in the Ga compounds. The Pt 5d DOS at E_F in Yb₂Pt₆ X_{15} is expected to be larger than the Ni 3d DOS at E_F in YbNi₃X₉. However, the Pt 5d peak is located at $E_B = 3-6 \,\text{eV}$, and the Ni 3d peak is located at $E_B \sim 2 \,\text{eV}$. If we assume that both DOS features have a similar tail toward E_F , the Pt 5d DOS at E_F should be lower than the Ni 3d DOS, which contradicts our expectation. Here, note that the Pt 5d DOS has a broader bandwidth than the Ni 3d DOS. In general, the Pt 5d states are relatively delocalized, while the Ni 3d states are relatively localized. In a photoemission study on Ni-Pt alloys, it was reported that the Pt 5d bands spread over the top 8 eV region in the valence bands, while the Ni 3d bands spread over the top 4 eV region [25]. Although the crystal structure is completely different between the Ni-Pt alloy and $Yb_2Pt_6X_{15}$ and $YbNi_3X_9$, the results are a measure of the Pt 5d and Ni 3d bandwidths. Thus, the broad bandwidth of the Pt 5d states may qualitatively explain the larger c-fhybridization and higher T_K in Yb₂Pt₆ X_{15} . In order to clarify this point, a band-structure calculation based on the precise structural analysis is necessary in a future study.

IV. SUMMARY

The electronic structure of the Kondo lattices $Yb_2Pt_6X_{15}$ has been investigated by means of HAXPES with $h\nu =$ 5.95 keV and VUVPES with $h\nu = 182$ eV. Both Yb²⁺- and Yb³⁺-derived structures were clearly observed in the Yb 3dspectra. For Yb₂Pt₆Al₁₅, the intensity of the Yb²⁺ (Yb³⁺) structures gradually increases (decreases) with decreasing temperature. The estimated Yb valence of 2.89 at 250 K gradually decreases to 2.83 at 20 K. On the other hand, the Yb valence of $Yb_2Pt_6Ga_{15}$ is 2.34, with almost no temperature dependence. The Pt 4f and Pt 5d peaks are shifted to shallower E_B with changing the X ion from Al to Ga. At the same time, an energy shift toward deeper E_B in the $Yb^{3+} 4f$ multiplet structures is observed. We described the enhanced c-f hybridization and T_K in Yb₂Pt₆Ga₁₅ based on the Pt-derived DOS at E_F and the Yb³⁺ 4 f hole level relative to E_F together with results from YbNi₃X₉ [5]. Systematic changes with the same order of T_K were observed in the Yb valence and ${}^{3}H_{6}$ peak energy. We believe that the higher T_{K} for $Yb_2Pt_6X_{15}$ compared to $YbNi_3X_9$ is derived from the large Pt 5d DOS at E_F due to the broad bandwidth.

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