Magnon dispersion in $Ca_2Ru_{1-x}Ti_xO_4$: Impact of spin-orbit coupling and oxygen moments

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The magnon dispersion of Ca_2RuO_4 has been studied by polarized and unpolarized neutron scattering experiments on crystals containing 0, 1, and 10% of Ti. Ti is inserted in order to enable the growth of large, partially detwinned crystals. One percent of Ti has a negligible impact on structural and magnetic properties. Also for 10% Ti content magnetic properties still change very little, but the insulating phase is stabilized up to at least 700 K and structural distortions are reduced. The full dispersion of transverse magnons studied for 1% Ti substitution can be well described by a conventional spin-wave model with interaction and anisotropy parameters that agree with density functional theory calculations. Spin-orbit coupling strongly influences the magnetic excitations, as it is most visible in large energies of the magnetic zone-center modes arising from magnetic anisotropy. Additional modes appear at low energy near the antiferromagnetic zone center and can be explained by a sizable magnetic moment of 0.11 Bohr magnetons, which the density functional theory calculations find located on the apical oxygens. The energy and the signal strength of the additional branch are well described by taking into account this oxygen moment with weak ferromagnetic coupling to the Ru moments.

I. INTRODUCTION

Ca_2RuO_4 (CRO) is the Mott insulating [1] end member of the series $Ca_{2-x}Sr_xRuO_4$, which possesses a rich diversity of structural, magnetic, and transport properties [2–5]. Sr_2RuO_4, the other end member, is proposed to be a spin-triplet superconductor with broken time-reversal symmetry [6–9]. The metal-insulator transition in CRO goes along with severe structural distortions [3,10], in particular a flattening of the RuO_6 octahedron, that increase until the onset of antiferromagnetic order [11]. The nowadays widely used picture assumes that an orbital ordering is associated with the structural changes [12–14]. The $4d_{xy}$ orbitals become doubly occupied and the $4d_{xz,yz}$ singly occupied resulting in flattened octahedrons and a $S = 1$ state. In the past the nature of the Mott transition of this multiband system with four d electrons on the Ru site was intensively discussed [12,15–20]. Recently, it was proposed that spin-orbit coupling (SOC) in this $4d$ system is strong enough to change the multiplet structure and couples S and L to j resulting in a nonmagetic $j = 0$ ground state. The occurrence of magnetic order was proposed to be of a singlet-magnetism type (see, e.g., Sec. 5.5. in Ref. [21]), which was called excitonic magnetism in Ref. [22], and a special type of magnon dispersion was predicted in this theory [23]. The main branches of the dispersion obtained by inelastic neutron scattering (INS) experiments, however, could be successfully described with a conventional Heisenberg model [24,25] and disagree with the $j = 0$ calculations [23]. Nevertheless, neutron scattering experiments revealed features such as an upward dispersion beyond the zone boundary

[25] and additional magnetic scattering not describable with a simple square-lattice antiferromagnet. The main in-plane transverse modes exhibit a large anisotropy gap, underlining the impact of broken tetragonal symmetry in combination with SOC. Furthermore, additional signals were detected in the neutron scattering experiments at low and at high energies [24,25] that cannot be explained by the two transverse magnon branches.

The crystal growth of insulating CRO is severely hampered by the metal-insulator transition occurring in pure CRO at $T_{MIT} = 360$ K [3,10]. The space group does not change at this first-order phase transition, but there are sizable jumps in the lattice parameters, in particular for c, with $\Delta c \sim 0.2$ Å [3]. Therefore, the crystals tend to crack upon cooling after the growth process, and only small pieces of mm³ size can be recovered. We circumvented this problem by introducing Ti that seems to broaden the metal-insulator transition.

The paper is arranged as follows. We first show that only for 10% of Ti there are significant changes of physical properties, see Sec. III A. This large substitution considerably stabilizes the insulating phase and suppresses structural distortions, but magnetic properties are nearly identical to those in pure CRO. Therefore, the results on the magnon dispersion discussed in Sec. III B that were obtained from polarized and unpolarized INS experiments on 1% Ti substituted CRO can be taken as representative for the pure material. In addition we performed density functional theory (DFT) calculations that may well reproduce the rather peculiar magnon dispersion. The DFT calculations also reveal a sizable ordered moment located on the apical oxygens, which explains the occurrence of an additional low-energy branch. The magnon dispersion in CRO thus reveals a very strong impact of SOC and of oxygen magnetic moments.

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II. EXPERIMENTAL

Because INS experiments require samples of $\sim$cm^3 size, we substituted a small amount of Ru by Ti in order to cool the crystals through the metal-insulator transition without breaking them into small pieces. Crystals with a mass of 0.5 g to 1 g were obtained with only 1% Ti substitution. We were also able to obtain two larger pieces of pure CRO, one with a mass of 0.45 g and one with 0.3 g, but these crystals possess a bad mosaic spread of 5°, which most likely is the reason why they were not destroyed upon cooling. Nevertheless, these crystals are suitable for some INS studies.

The Ca$_2$Ru$_x$Ti$_{1-x}$O$_4$ crystals with $x = 0, 1$, and 10% used in this neutron scattering study (labeled 0Ti, 1Ti, 10Ti, respectively) were grown by the floating-zone method in a Canon Machinery Inc. SC1-MDH11020-CE furnace equipped with two 2000 W halogen lamps and a cold trap following the procedure described in Refs. [26,27]. CaCO$_3$, RuO$_2$, and TiO$_2$ were mixed in stoichiometric ratios and a Ru excess of 32% was added. The powder was mixed and reacted for 24 h with an intermediate grinding. Then a rod was pressed and sintered at 1350 °C. A growth speed of 17 mm/h and a feed-rod speed of 20 mm/h were used and the atmosphere contained 90% Ar and 10% O at a total pressure of 9 bar. Phase purity was checked by x-ray powder diffraction from crushed parts of the single crystals, which indicates no impurity phases. The lattice constants at room temperature were obtained by LeBail fits of these powder pattern using the FullProf Suite [28]. Single-crystal x-ray diffraction experiments were performed on a Bruker X8 Apex diffractometer equipped with a charge-coupled-device detector. Details of this structure determination and a detailed list of the refined parameters can be found in Appendix A. Magnetization measurements were performed with a superconducting quantum interference device (SQUID) magnetometer from Quantum Design, and the resistivity was measured with a standard four-point method. Spin wave calculations were performed using SpinW [29].

Elastic and inelastic neutron scattering experiments were performed with the cold triple-axis spectrometer (TAS) 4F1 at the LLB, with the thermal TAS IN3 at the ILL and with the polarized thermal TAS IN22 at the ILL. The polarized neutron scattering experiments were performed using Heusler (111) monochromator and analyzer crystals. A PG filter was inserted on the scattered beam before the neutron-spin flipper and the monitor was put on the incoming beam between monochromator and sample. A set of Helmholtz coils was used to produce the guide field and the sample was zero field cooled (less than 2 G) in an orange-type cryostat. A mounting of three crystals with a total mass of 2.5 g containing 1% Ti (1Ti) was used for the polarized neutron scattering experiment. The sample was twinned with nonequal twinning fractions of 2.5:1 determined by scanning the orthorhombic (200) and (020) reflections. The a,b plane [see Fig. 1(a)] was chosen as the scattering plane in order to efficiently integrate the inelastic signal along the vertical direction, where the resolution is poor. We used the standard coordinate system in polarized neutron scattering [30]. **x** is parallel to the scattering vector (**Q**); **y** and **z** are perpendicular to **Q**. While **y** is in the scattering plane, **z** is perpendicular to it. Therefore, **y** lies in the a,b plane of the crystal and **z** parallel to the c axis. In
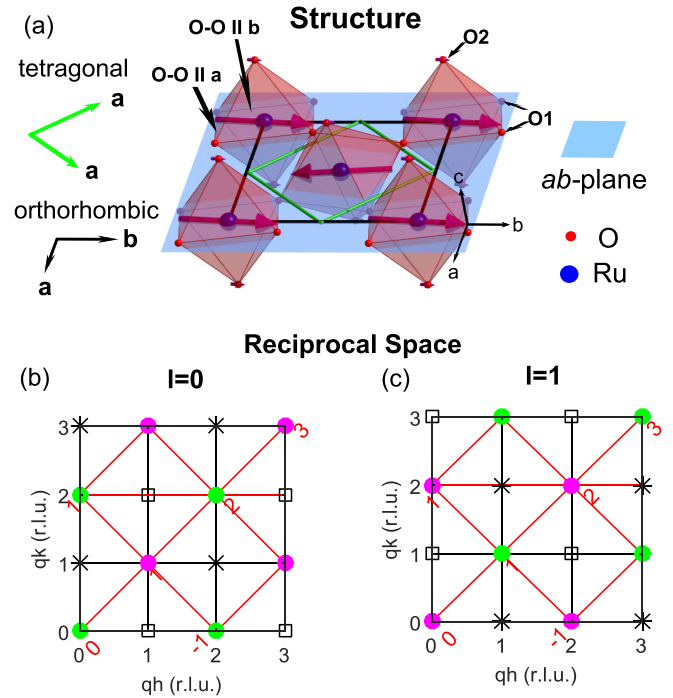

FIG. 1. Crystal and magnetic structure of CRO and the associated reciprocal space. In (a) one layer of the crystal and magnetic structure is shown. The Ru (blue balls) sit in octahedrons of oxygen (red balls). The ab-plane is indicated in light blue. In black and green the orthorhombic and tetragonal cells are shown. The pink arrows denote the magnetic moments on the Ru and O sites. At the tip of the low right Ru moment, the directions of the different polarizations of the magnon modes are indicated by black arrows and some labellings of atoms and atom distances are given. In (b,c) the (hkl) planes in reciprocal space are illustrated for $l = 0$ and $l = 1$, respectively. The orthorhombic and tetragonal cells are drawn in black and red, respectively. Green circles denote tetragonal zone centers and pink circles zone boundaries. Black squares (stars) mark antiferromagnetic zone centers of the A- (B-) centered antiferromagnetic order.

neutron scattering only magnetic moments perpendicular to **Q** intervene. Therefore, the magnetic scattering intensities sense a geometry factor $\sin^2(\alpha)$ with α being the angle between **Q** and the magnetic moment, which corresponds to either the static ordering moment in diffraction or to the oscillating moment in a magnon. With longitudinal neutron polarization six different channels can be analyzed: three spin-flip (SF) and three non-spin-flip (nSF) channels. While phonons always contribute to the nSF channel there is an additional selection rule for magnetic scattering. The magnetic component parallel to the direction of the neutron polarization contributes to the nSF channel, while the components perpendicular to the neutron polarization generate SF scattering. By combining this polarization rule with the geometry factor one can distinguish the different magnetic components. In the geometry we use for the polarized neutron scattering experiment we see the c polarized modes in the SFy channel and they do not loose intensity due to the geometry factor, because the scattering vector is always parallel to the a,b plane and thus perpendicular to c. In contrast the SFz channel contains the in-plane modes, the transverse and longitudinal ones, but weighted with the

geometry factor. For example at $\mathbf{Q} = (2,1,0)$ the geometry factor for the transverse mode (a polarized) is $\sin^2(\alpha) = 0.2$ and that for the longitudinal mode (b polarized) $\sin^2(\alpha) = 0.8$. At $\mathbf{Q} = (1,2,0)$ this ratio is inverted [see Figs. 1(b) and 1(c)]. Respecting the twinning ratio of 2.5:1, we expect that the intensities at $\mathbf{Q} = (2,1,0)$ and $\mathbf{Q} = (1,2,0)$ have a ratio of 0.6 for the transverse mode. For the longitudinal mode this ratio is inverted. Thus, it is possible to distinguish between transverse modes and longitudinal modes by comparing scattering at properly chosen $\mathbf{Q}$. Because the polarization of the neutron beam is not perfect, one has to correct the intensities for the finite flipping ratio (FR) [30,31]. Thereby, the magnetic signals are obtained from the intensities of the different SF channels corrected for the FR:

$$I(M_{y,z}) = \frac{FR+1}{FR-1}[I(SF_x) - I(SF_{y,z})]. \qquad (1)$$

The FR of our experiment on IN22 amounts to 12, which is determined by comparing the signals of rocking scans on the (200) Bragg reflection in the SF and nSF channels.

On the cold TAS 4F1 crystals with different Ti content were used, 0Ti and 10Ti. 0Ti has a mass of 0.45 g and a twinning ratio of 1:1 and 10Ti a mass of 0.93 g and a twinning ratio of 9:1. The scattering plane for both samples was chosen to be the [010]/[001] plane, so the scattering plane for the other twin domains was the [100]/[001] plane. For all 4F1 scans a pyrolytic graphite monochromator and analyzer were used, a cooled Be filter was put on k_f to suppress higher order contaminations. For the cooling of the samples an orange-type cryostat was used and k_f was set to 1.55 Å^{-1} for all scans.

With the IN3 TAS we analyzed the magnetic order of the crystal containing 1% Ti used in the previous neutron scattering study [24] (1TiB). This crystal was found to exhibit a majority twin domain of 95% and the experiment was performed in the [010]/[001] orientation used in the INS experiment.

III. RESULTS AND DISCUSSION

A. Impact of Ti substitution on structural, magnetic, and electronic properties of Ca$_2$Ru$_{1-x}$Ti$_x$O$_4$

1. Crystal structure of Ca$_2$Ru$_{1-x}$Ti$_x$O$_4$

The crystal structure of CRO [11] is similar to the structure of Sr$_2$RuO$_4$ of K$_2$NiF$_4$ type [32], where the Ru atoms sit in oxygen-octahedron cages, which are corner shared in the a,b plane. In CRO the octahedrons are rotated and tilted and they also become severely distorted. As a consequence, the tetragonal symmetry of Sr$_2$RuO$_4$ is reduced to the orthorhombic space group $Pbca$ [11]. One layer of the crystal structure is drawn in Fig. 1, which also depicts the magnetic order with antiferromagnetic moments on Ru pointing along b. The orthorhombic unit cell is rotated by 45° with respect to the tetragonal one and enhanced to: $\mathbf{a}_{\mathrm{orth}} = \mathbf{a}_{\mathrm{tet}} + \mathbf{b}_{\mathrm{tet}}$, $\mathbf{b}_{\mathrm{orth}} = \mathbf{a}_{\mathrm{tet}} - \mathbf{b}_{\mathrm{tet}}$, $a_{\mathrm{orth}} \sim b_{\mathrm{orth}} \sim \sqrt{2}a_{\mathrm{tet}}$. Unless otherwise specified all notations refer to the orthorhombic lattice of the majority twin orientation. Our crystals usually possess two twin orientations, which are obtained by interchanging the a_{orth} and b_{orth} directions.
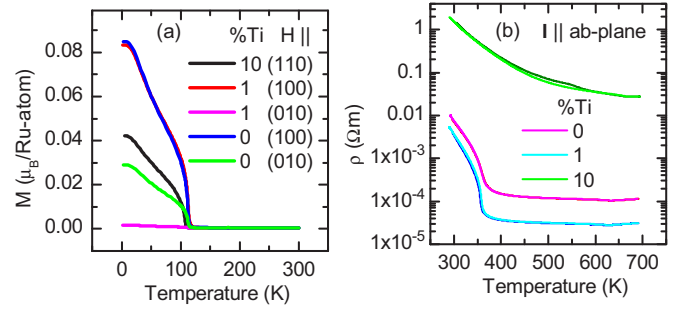

FIG. 2. Temperature dependence of physical properties of CRO with different Ti content. (a) Field-cooled magnetization curves in 0.1 T measured on heating. (b) Resistivity on heating in dark color and on cooling in bright color. Only the 10% Ti curves do not coincide.

All crystals were examined by x-ray diffraction, magnetization, and resistivity measurements in addition to the neutron scattering experiments described below. Figure 2 shows the magnetization and electric resistance data of the three Ti concentrations. The temperature of the metal-insulator transition is 4 K lower for the sample containing 1% Ti than for the pure compound. The sample containing 10% Ti does not show indications for a phase transition upon cooling down to 80 K where the experimental limit of the increasing resistivity is reached. The absolute values of the resistivity curves have a large uncertainty because of the first-order structural phase transition. There the crystals tend to crack, which prohibits the current to flow through the hole sample. This effect has been frequently observed during a single measurement cycle. The resistivity is enhanced by a multiplication factor after passing the structural transition. As the samples have already passed this transition after the crystal growth, there are some cracks inside the sample, which is evident from measuring several pieces of the same crystal without obtaining the same room-temperature specific resistivity.

The magnetic and insulating properties in Ca$_{2-x}$Sr$_x$RuO$_4$ are closely related to the crystal structure [3]. CRO is heavily distorted with respect to Sr$_2$RuO$_4$, which possesses the ideal structure of K$_2$NiF$_4$ type without structural distortions, but which already is close to such a structural instability [33–35]. In the layered ruthenates the distortions can be described as octahedron rotation around the c direction and tilting around an in-plane axis. These distortions and the associated structural phase transitions result from bond-length mismatch, and the distortions in CRO are induced by the chemical substitution of Sr by the isovalent but smaller Ca, which is not able to fill the space between the oxygen octahedrons like Sr does. So the octahedrons start to rotate and tilt in order to reduce the coordination volume around Ca. By chemically substituting the smaller Ti for Ru, one expects a small decrease of these deformations, which is indeed realized. The tilt and rotation angles of 10Ti are significantly smaller than the values determined in the samples containing only 0 and 1% Ti, see Appendix A, Table II.

Most interesting is the impact of the Ti doping on the deformation of the RuO$_6$ octahedron, which is a fingerprint of the lifting of orbital degeneracy [3,11,12]. The structural change at the metal-insulator transition in pure CRO is
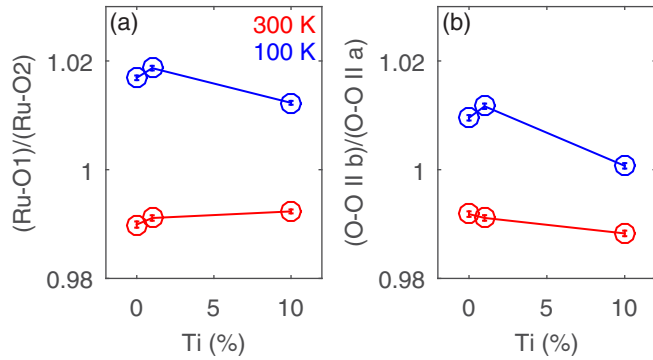
FIG. 3. Temperature and Ti doping dependence of the RuO_6 octahedron distortion obtained by single crystal x-ray diffraction analyses; (a,b) presents the ratios of in-plane to out-of-plane RuO bond distances (flattening) and of the two octahedron edges parallel to b and a, respectively. Both entities are enhanced in the insulating phase at low temperature due to the lifting of orbital degeneracy.

characterized by a jump of the c lattice constant and by a flattening of the octahedron as it is visible in the Ru-O bond distance ratio [3]. Upon cooling into the insulating phase this octahedron flattening continues until it saturates at the antiferromagnetic phase transition. Slightly below room temperature the octahedron shape passes from elongated to flattened [3]. In all samples studied here, this crossover occurs below room temperature, thus at 100 K the octahedron shape is flattened for all Ti substitutions, see Fig. 3. Due to flattening of the octahedron the d_{xy} orbital shifts down in energy compared to the d_{xz}, d_{yz} orbitals. A similarly strong and anomalous temperature dependence is also observed for the ratio of the two O-O octahedron edge lengths parallel a and b which is related to the orthorhombic splitting [$\epsilon = (b-a)/(a+b)$] [3]. At room temperature the octahedron is longer along a, while it is elongated along b at low temperature. All these effects can be attributed to a temperature dependent orbital ordering [3,11,13,24]. The considerable elongation along b at low temperature agrees with SOC and the alignment of the magnetic moment mainly along the b direction [24]. Figure 3 illustrates that these two distortions of the RuO_6 octahedron become suppressed by the 10% Ti substitution. This underlines the orbital ordering character of these distortions, which is obviously suppressed by Ti^{4+} with an empty $3d$ shell.

While the a,b plane (or the average in-plane parameter) increases with high Ti content, the c lattice constant decreases at room temperature. The reduction of the c lattice constant, in hand with an increase of the a,b plane, is the structural signature of the insulating state of CRO. With 1% of Ti substitution there is only a small decrease in T_{MIT} (Fig. 2), but with 10% of Ti substitution there are drastic effects. In the resistivity of 10Ti there are no indications for a sharp metal-insulator transition on cooling from 700 down to 80 K, where the upper experimental limit of the resistance experiment is reached. Note that the absolute values of the resistivity curves are prone to a large uncertainty because of the first order structural phase transition which causes cracks in the crystals. But the orders of magnitude larger resistivity of the 10Ti sample and the absence of the metal-insulator transition is unambiguous. This remarkable stabilization of the insulating

state by only small amounts of Ti substitution is also seen in $Ca_3Ru_2O_7$ [36]. Since the ionic radii of Ti and Ru are very similar, this suppression of the metallic state seems to originate from the very effective suppression of the hopping. The Ti does not contribute states near the Fermi level in metallic ruthenates [36], so that the hopping becomes disrupted. Also in Sr_2RuO_4 Ti substitution has a strong impact: It stabilizes spin-density wave ordering associated with the Fermi-surface nesting of the pure material for only 2.5% Ti [37]. Furthermore, Ti substitution also stabilizes a spin-density wave magnetic instability in $Sr_3Ru_2O_7$ [38]. On the other hand Ti substitution yields little impact on the insulating low-temperature state in CRO as visible in the little changes of magnetic properties induced by 10% Ti concentration.

2. Magnetic structure of $Ca_2Ru_xTi_{1-x}O_4$

The magnetic structure of 1Ti was studied on IN22 with polarization analysis. The analysis of the magnetic structure of 1TiB on IN3 is described in Appendix B. Scans along the (200) and (020) reflections (Fig. 4) reveal that crystal 1Ti exhibits both twin-domain orientations in the ratio 2.5:1. Figures 4(c)–4(f) show the SF intensities in the three channels for scans across the (100), (010), (120), and ($\bar{1}$20) magnetic reflections for crystal 1Ti. The magnetic structure of CRO has been previously determined by neutron powder diffraction [11]. As illustrated in Fig. 1(a) the magnetic moments are essentially aligned along the b direction (corresponding to the tilt axis) with antiferromagnetic alignment between nearest neighbors. The moments are canted along a yielding a net ferromagnetic moment in a single layer. There are two different magnetic structures reported for CRO [3,11,39], which differ by the stacking of the single layer arrangement shown in Fig. 1(a). The magnetic structure of the main antiferromagnetic b component is either A or B centered. In the A-centered phase the two magnetic moments at (0,0,0) and (0,0.5,0.5) are parallel; in the B-centered one the two moments at (0,0,0) and (0.5,0,0.5) are parallel. The magnetic space groups are $Pbca$ (A centered) and $Pbc'a'$ (B centered). While in $Pbca$ the net ferromagnetic canted moments per layer cancel due to an antiferromagnetic stacking, the B-centered $Pbc'a'$ structure results in a total ferromagnetic moment that can be measured with a magnetometer. There is also canting along the c direction in both magnetic structures. This canting corresponds to an antiferromagnetic c component that is B centered in $Pbca$ and A centered in $Pbc'a'$. This moment should, however, be small because the main part of the ordered moment points along the tilt axis, and only microscopic methods can detect such an antiferromagnetic c component.

The analysis of the magnetic Bragg peaks with polarization analysis first confirms that moments point along the b direction and show a dominating B-centered scheme for 1Ti and an almost exclusive B-centered scheme for 1TiB, see Appendix B. In contrast, small pure CRO samples show only the A-centered scheme [39].

In Fig. 2(a) the magnetization upon heating in a magnetic field of 0.1 T applied in the ab plane is shown for different Ti contents. The weak ferromagnetic component dominates the magnetization below T_N. The fact that the magnetization of the lowest temperature is the highest, points to a dominating
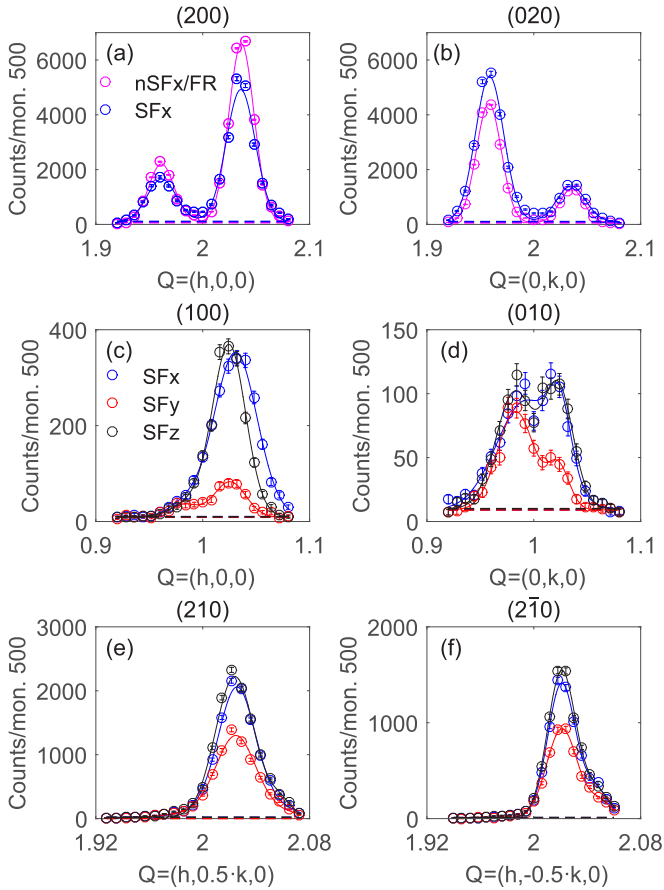
FIG. 4. Elastic neutron scattering scans with polarization analysis. The orthorhombic splitting is clearly visible in these elastic scans. (a),(b) present the values of the nSFx (SFx) channel in magenta (blue) of longitudinal scans across the strong nuclear reflections (200) and (020), respectively. The values of the nSFx channel are divided by the FR. These elastic longitudinal scans show that both twins are well separated. (c)–(f) present data of the SFx (blue), SFy (red), and SFz (black) channels of scans across the magnetic (c) (100), (d) (010), (e) (210), and (f) (2$\bar{1}$0) reflections.

B-centered phase [11] in agreement with the neutron diffraction studies. The small crystal used for the magnetization measurement with 1% Ti content is essentially untwinned, while the sample containing no Ti is partially twinned with nonidentical twin fractions. If the field is applied along [110], the observable ferromagnetic component is reduced by a factor $1/\sqrt{2}$ but both twin domain orientations contribute (sample containing 10% Ti). The highest total ferromagnetic component is observed in the pure sample, and the reduced ferromagnetic component in 1Ti is possibly caused by a slightly reduced *B*-centered phase fraction. We can conclude that CRO exhibits sizable moment canting resulting in an ordered ferromagnetic component along the *a* direction of $\sim 0.08\mu_B$ (canting angle $\alpha = 3.5$ deg). The canting of the magnetic moment arises from a strong Dzyaloshinski-Moriya interaction that in turn stems from the strong SOC. Minimizing the static energy of a pair sensing only the Heisenberg interaction J and the Dzyaloshinski-Moriya interaction, $-\mathbf{D} \cdot \mathbf{S_i} \times \mathbf{S_j} - J\mathbf{S_i} \cdot \mathbf{S_j}$, yields the condition $\tan(2\alpha) = D/J$, and
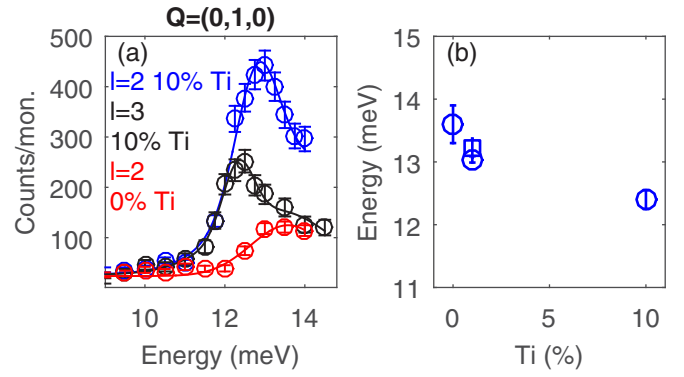

FIG. 5. Influence of Ti substitution on the anisotropy gap. (a) shows energy scans through the anisotropy gap at the antiferromagnetic zone center using the crystal 10Ti, $l = 2$ (blue) and $l = 3$ (black) and 0Ti, $l = 2$ (red). In (b) the energy of the lower zone-center mode of crystals with different Ti content is shown, the data for 1% Ti are from Ref. [24].

thus a rough estimate of the Dzyaloshinski-Moriya interaction in CRO: $D = 0.06J$. The magnetization curves further show a decrease of the Néel temperature with increasing Ti content, see Table II in Appendix A, which can be explained by the dilution of the magnetic lattice by nonmagnetic Ti.

3. Impact of Ti substitution on magnetic excitations

In Fig. 5 the influence of different Ti substitutions on the anisotropy gap in the magnon dispersion is addressed by comparing constant **Q** scans across the in-plane gap at the antiferromagnetic zone center. The modes are slightly split due to finite interlayer interaction, see Sec. III B. Even l corresponds to the higher and odd to the lower modes, respectively. The lower zone-center magnon energy is displayed as function of Ti doping in Fig. 5(b). The anisotropy gap clearly diminishes with increasing Ti content. Since the nonmagnetic Ti dilutes the magnetic lattice, a general softening can be expected, as it is visible in the anisotropy gap, which in first view corresponds to the square root of exchange and anisotropy energies. In addition, Ti also perturbs the lifting of orbital degeneracy, as it is shown in Fig. 3. In consequence also the single-ion anisotropy will be reduced with increasing Ti content. The impact of a small Ti content of the order of one percent on the magnon dispersion can, however, safely be neglected. 10Ti, which is essentially untwinned, shows a l dependence of the anisotropy gap [Fig. 5(a)] consistent with the previous study [24].

B. Magnon dispersion in Ca$_2$Ru$_{0.99}$Ti$_{0.01}$O$_4$

1. Polarization of magnon modes

With the alignment of the antiferromagnetic moment along *b* one expects nondegenerate transverse magnon modes corresponding to polarization along *a* or *c* [31]. With our previous unpolarized INS experiments [24] only the *a* polarized mode could be clearly identified. The gap of the *c* polarized transverse mode could either be hidden in the shoulder of the in-plane signal, which inevitably arises from the folding of the instrument resolution with the steep spin-

wave dispersion, or appear at higher energies where weak signals were detected [24]. With the new polarized experiment the in-plane and *c* polarized transverse magnons can be easily separated in the polarization, and it was the first aim to search for the *c* polarized mode at the antiferromagnetic zone center.

Figures 6(a)–6(c) shows energy scans through the anisotropy gap at the antiferromagnetic zone centers $\mathbf{Q} = (2,1,0)$ and $(1,2,0)$. The nSF channels shown in Fig. 6(a) have a larger background than the SF channels [Figs. 6(b) and 6a(c)] and are thus less informative but confirm the main conclusions. The *c* polarized mode must entirely contribute to the SFy and nSFz channels with a complete geometry factor. Spin-wave calculations show that the two nondegenerate transverse modes exhibit an intrinsic signal strength inversely proportional to their zone-center energy [31]. Therefore, the *c* polarized mode can be excluded in the asymmetric peak of the in-plane transverse mode. These scans confirm the in-plane character of the signal in the range 14 to 20 meV, see Fig. 6(d). A special effort was laid on the analysis of the signal maximum at the in-plane magnon gap, $E = 14$ meV, see Figs. 6(e) and 6(f). With the partially detwinned crystal one expects this signal from the transverse magnon to be reduced by a factor 0.6 for $\mathbf{Q} = (2,1,0)$, which agrees with the measured ratio of the intensities at 14 meV between $\mathbf{Q} = (2,1,0)$ and $\mathbf{Q} = (1,2,0)$ of 0.55(10). We also measured the backfolded mode at the ferromagnetic zone center, which according to the spin-wave calculations, see below, exhibits a *c* polarization. Indeed such a *c* polarized signal can be determined at $Q = (2,0,0)$ and $E = 14$ meV [Fig. 6(g)] by counting for a very long time (45 min for each data point).

In view of the recent observation of longitudinal magnetic excitations in CRO [25] it seemed worthwhile to further search for such modes. Here the term longitudinal mode designates a fluctuation along the sublattice magnetization that in a common system with well-defined local moments is suppressed; longitudinal excitations, however, always arise from excitation and absorption of an even number of magnons as it has been demonstrated for MnF$_2$ [40]. We first studied the two antiferromagnetic zone centers $(2,1,0)$ and $(1,2,0)$, where longitudinal excitations arising from two-magnon processes are expected at the lowest energies [40]. In Fig. 6(i) the fitted background from the in-plane polarized transverse mode from Fig. 6(d) is used to separate a weak signal peaking at 29 meV. The signals for the two different $\mathbf{Q}$ values are assumed to possess the same shape but are scaled with the expected factors for geometry and twinning. Around 30 meV the stronger signal for $\mathbf{Q} = (2,1,0)$ compared to $\mathbf{Q} = (1,2,0)$ thus points to a longitudinal excitation in the sense that it is polarized parallel to the static moment parallel *b*. The energy of this mode corresponds fairly well to the double of the gap of the in-plane transverse branch and the strength is below 10% of that signal. Therefore, it seems most likely that this longitudinal signal stems from the two-magnon excitation, which is expected to appear in the longitudinal polarization channel [40,41]. Note that due to the large magnon gap in CRO, absorption processes do not play a role at low temperature, so that only the two-magnon excitation sensing twice the magnon gap is relevant in our measurements. A similar discussion about an intrinsic longitudinal mode has also been initiated
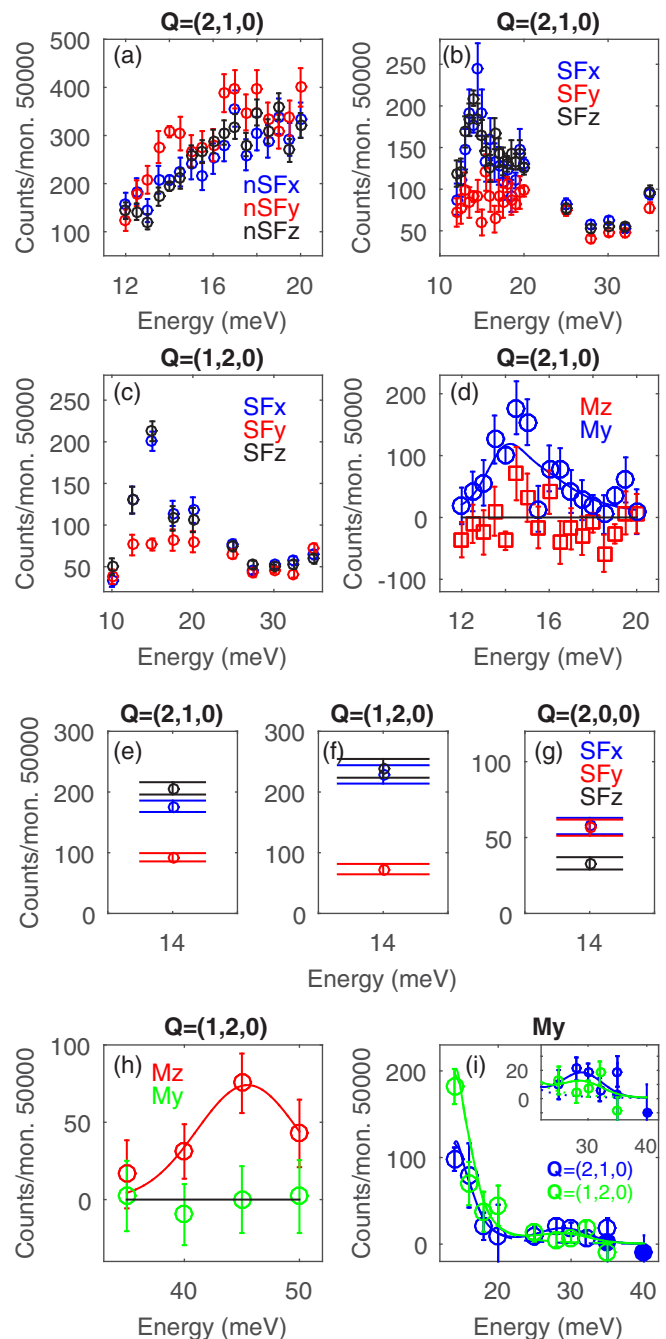

FIG. 6. Energy scans at various $\mathbf{Q}$ values with polarization analysis taken on partially twinned 1Ti. Raw data of different (a) nSF, and (b),(c),(e)–(g) SF channels with the polarization parallel to **x** (**y**),(**z**) in blue (red), (black) at different scattering vectors. (d),(h),(i) show the results of the polarization analysis with Eq. (1), in red M$_z$, and M$_y$ in blue (green) at $\mathbf{Q} = (2,1,0)$ ($\mathbf{Q} = (1,2,0)$). In (d) the blue line depicts the modeled asymmetric shape resulting from the folding of the magnon dispersion with the resolution function, see Ref. [24], which is also used for the modeling in (i), but scaled with the expected correction factors for geometry and twinning fractions. The inset in (i) shows a zoom into the data.

for the parent material of FeAs-based superconductors, for which again the two-magnon explanation seems most likely

[31,42–44]. The 29 meV longitudinal signal observed in CRO at the antiferromagnetic zone center is thus not anomalous but just corresponds to the two-magnon excitation expected to contribute to the longitudinal channel.

The c polarized antiferromagnetic zone-center mode is detected at higher energy, where the unpolarized experiment found some evidence for additional scattering [24] but where the signal strength is expected to be rather small [31]. In order to cover the energies of the order of 40 meV we needed to use a larger value of $k_f = 4.1$ Å^{-1}, which also allows one to avoid the contamination appearing at $E = 44$ meV for the standard value of $k_f = 2.662$ Å^{-1}. The magnetic signals polarized along $\mathbf{z} \simeq c$ and along $\mathbf{y}$ at large energies are shown in the scan at $(1,2,0)$ in Fig. 6(h) and show that the zone-center c-polarized magnon possesses a large energy of 45.5(1.5) meV. Its signal strength is in rough agreement with the signal strength of the in-plane transverse mode, because the energy is enhanced by a factor three. Our conclusion of c polarized modes at the ferromagnetic and antiferromagnetic zone centers agrees with the interpretation of similar polarized neutron scattering experiments performed with twinned crystals in a different scattering geometry [25].

This large c anisotropy is remarkable, as it strongly deforms the magnon dispersion in CRO with respect to a simple isotropic model. The splitting of the two antiferromagnetic zone-center magnons, 14 and 45.5 meV, is larger than the dispersion of the in-plane branch to the zone boundary. As will be discussed below, the entire dispersion of transverse branches is, nevertheless, well described with the $S = 1$ spin-wave model using the Holstein-Primakoff transformation. The in-plane and c-polarized branches exchange their character: While the c-polarized mode is the high-energy mode at the antiferromagnetic zone center, it appears at the lower energy of the in-plane transverse magnon at the ferromagnetic zone center, in accordance to the data shown in Fig. 6(g). The in-plane polarized branch thus starts at 14 meV at the antiferromagnetic zone center, exceeds to the zone boundary at 41 meV and then continues to stiffen till 45.5 meV at the ferromagnetic zone center. The out-of-plane polarized branch just exhibits the opposite dispersion. By comparing the intensities at the ferromagnetic zone centers $\mathbf{Q} = (2,0,0)$ and $\mathbf{Q} = (0,2,0)$ with an energy transfer of 45 meV we may confirm the in-plane transverse character of this high-energy mode appearing at the ferromagnetic zone center. The expected ratio of the signals taking into account the different twinning fractions and geometry factors amounts to 1:2.5 for a transverse and to 2.5:1 for a longitudinal magnon. We obtain a ratio of 0.1(2) indicating the transverse in-plane character. Evidence for an intrinsic longitudinal high-energy branch could not be obtained in our experiment [25,45].

2. Spin-wave calculations of the magnon dispersion

The anisotropy gap of the c-polarized transverse magnon of 45.5 meV leads to the uncommon feature in the magnon dispersion, that the transverse in-plane branch continues to increase in energy between the zone boundary and a ferromagnetic zone center [25]. This peculiarity can, however, be well described with a rather conventional model.
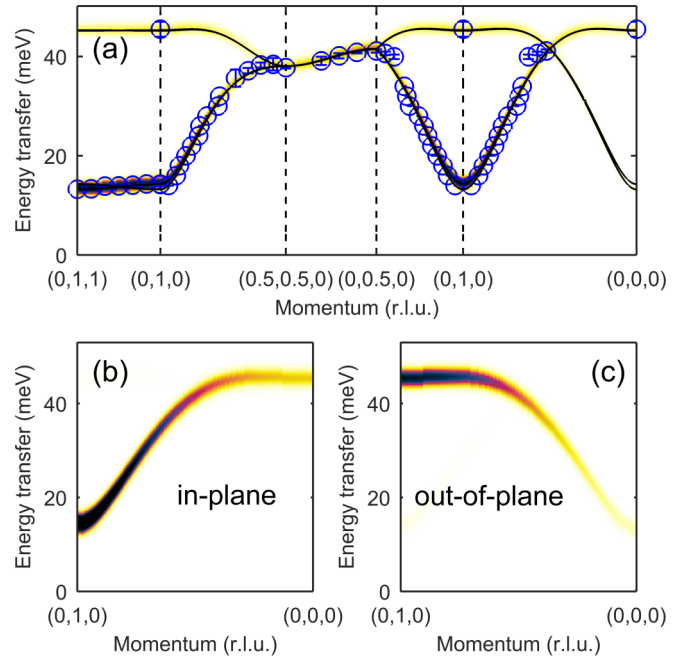

FIG. 7. Magnon dispersion calculated with the SpinW program using the values given in the text. The blue circles denote the fit values of the dispersion obtained in this and previous [24] neutron scattering studies. The black lines are the calculated dispersion and the color code denotes the calculated neutron scattering intensity of the convoluted spectra. In each panel black stands for maximum intensity and white for none. In (a) all transverse magnon modes are shown and in (b),(c) only the in-plane- and out-of-plane polarized transverse modes are shown, respectively, illustrating the opposite dispersion of these branches.

The Hamiltonian, which is used for the description of the magnon dispersion is given by:

$$H = \sum_{\mathbf{i},\mathbf{j}} \mathrm{J}_{i,j} \mathbf{S_i} \cdot \mathbf{S_j} + \gamma \sum_i \left(S_i^x \right)^2 + \epsilon \sum_i \left(S_i^z \right)^2. \quad (2)$$

The sum runs over pairs of magnetic ions, so that each pair or bond appears twice and S is set to 0.67 following the experimental results in Ref. [11]. The spin-wave calculations using the SpinW program and the parameters $J = 5.6$ meV, $J_{na,b} = 0.6$ meV, $J_c = -0.03$ meV, $\gamma = 1.4$ meV, and $\epsilon = 24.5$ meV give a good description of the magnon dispersion obtained in this and our previous [24] neutron scattering studies [Fig. 7(a)], and the main parameters J and ϵ agree with Ref. [25]. Panels Figs. 7(b) and 7(c) show the in-plane and c-polarized branches, respectively, starting at an antiferromagnetic zone center and proceeding to a ferromagnetic one. In this figure the color code denotes maximum neutron scattering intensity of the convoluted spectra with black color and zero intensity with white. The energy resolution is set to 1 meV. The spin-wave calculation thus perfectly describes not only the energy dispersion but also the polarization of the magnon branches. Note that this rather uncommon dispersion is described with strong single-ion parameters arising from SOC. These strong anisotropy terms interfere with the Heisenberg interaction parameters in contrast to models invoking only weak anisotropy. Therefore, the J parameter differs from that obtained by fitting

only the lower part of the dispersion with a small uniaxial anisotropy [24]. Figure 7(a) shows that the nonsinusoidal parts of the dispersion are perfectly described, clearly better than without the strong anisotropy terms. It is, however, important to note that the model is not unique. It is possible to obtain similar fitting by partially reducing the single-ion anisotropy and by inducing an anisotropic nearest-neighbor interaction.

The coupling parameter J_c acting between neighboring layers splits the magnon modes into two by introducing a finite dispersion perpendicular to the planes. Whether the lower or upper mode is seen at $\mathbf{Q} = (0,1,0)$ is determined by the sign of J_c. With the partially detwinned crystals (in particular with the experiment on 1TiB) it is possible to determine the sign of this interaction. Here it is chosen to couple spins at (0,0,0) and (0,0.5,0.5). The dominant b components of these spins are parallel for the A centering and antiparallel for the B centering. If J_c is positive (negative), corresponding to an antiferromagnetic (ferromagnetic) coupling, the upper (lower) mode is seen at $\mathbf{Q} = (0,1,0)$. The experimental data reveal that the upper mode is seen at $\mathbf{Q} = (0,1,0)$, so the coupling is ferromagnetic, stabilizing the A-centered phase. This coupling contradicts the observation of a B-centered structure as the main magnetic scheme, see Sec. III B. CRO thus exhibits the uncommon situation that the minimum magnon energy does not occur at the magnetic Bragg peaks. This observation can be attributed to an anisotropic J_c which differs for the a and b spin components.

3. Density functional calculations of magnetic structure and interaction

The linearized augmented plane wave (LAPW) method as implemented in the WIEN2K package [46] was used for the DFT calculations. The exchange correlation potential for the generalized gradient approximation (GGA) was chosen to be in the form proposed in Ref. [47]. The SOC was treated in a second variational way. The parameter of the plane-wave expansion was chosen to be $R_{MT}K_{\max} = 7$, where R_{MT} is the smallest atomic sphere radii ($R_{MT}^{Ca} = 2.15$ a.u., $R_{MT}^{Ru} = 1.99$ a.u., $R_{MT}^O = 1.71$ a.u.) and $K_{\max}$ the plane-wave cutoff. We used a mesh consisting of 800 k points. The on-site Hubbard repulsion and intra-atomic Hund's exchange were taken to be $U = 3$ eV and $J_H = 0.7$ eV [48,49] in the GGA+U [50] and GGA+U+SOC calculations.

The exchange parameters were recalculated via total energies of three different collinear magnetic configurations (a ferromagnetic and two, in the c direction differently stacked, antiferromagnetic configurations) using the GGA+U approach. We found that $J = 4.9$ meV, which agrees with the results of the spin-wave calculations.

In order to estimate the single-ion anisotropy we add SOC to the calculation scheme and computed the energies of two configurations, where all spins are either directed along c or lie in the a,b plane. The antiferromagnetic order was assumed in these calculations. The lowest total energy corresponds to the configuration, where all spins are directed predominantly along the b axis. The single-ion anisotropy is found to be very large, $\epsilon = 20.1$ meV, again in good agreement with experiment.

Very recently microscopic magnetic parameters were calculated for similar U and J_H values finding a nearest-neighbor
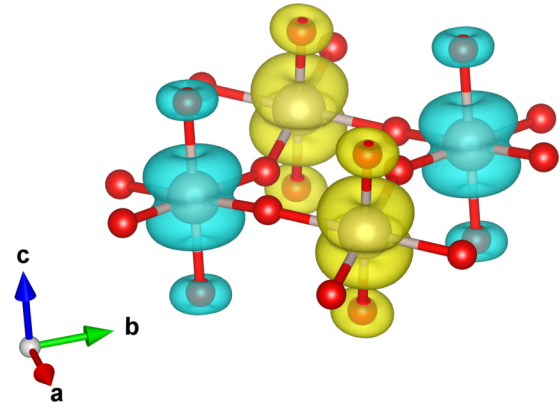

FIG. 8. The spin-density plot, as obtained in the GGA+U+SOC calculations with magnetic moments directed along the b axis. Small red balls are O ions. Different colors of the volumetric data correspond to different spin projections. One may see that there is no spin polarization on the planar oxygens, while apical ones have considerable magnetic moment, which is parallel to the moment of nearest Ru ions.

interaction of 3–6 meV and a large single ion anisotropy in good agreement with our results [51].

The spin moments were estimated by integrating the spin density obtained in the GGA+U+SOC calculation inside each atomic sphere with the radii R_{MT} specified for Ru and O above. The spin moment on the Ru ions was calculated to be $M_s^{Ru} = 1.27\mu_B$ (i.e., $S = 0.63$), while the orbital moment is $M_l^{Ru} = 0.13\mu_B$. We also find a sizable magnetic moment on the apical O $M_s^O = 0.11\mu_B$, while planar ligands stay nonmagnetic. The total spin density is shown in Fig. 8, which illustrates the significant polarization of the apical oxygens as well as the orbital character of the spin density at the Ru position. The total moment, $M_s^{Ru} + 2M_s^O - M_l^{Ru}$, agrees well with the powder neutron diffraction result [11], and the small value of the orbital moment indicates that CRO is not close to a $j = 0$ state driven by SOC as it is also deduced from the calculations in Ref. [51].

C. Additional modes and magnetic polarization of oxygen

The previous unpolarized experiments gave evidence for an additional scattering at the lower energy of 5 meV that could not be explained by the magnon dispersion expected for a simple square-lattice antiferromagnet. The data, which are presented in the mappings shown in Figs. 9(a) and 9(b), were obtained on the thermal TAS IN8 [24]. Panel (c) shows some characteristic scans included in (b). The 5 meV signal is much weaker than the in-plane transverse mode in particular when considering the by a factor of three reduced energy. Furthermore, it is not possible to connect this low-energy feature with a dispersing branch with a slope comparable to that of the transverse branch. Instead the intensity of this feature seems to be rather localized in $\mathbf{Q}$ space and to exhibit a flat dispersion. The signal could also be followed along the vertical direction without any indication for finite dispersion.

In Ref. [24] it was speculated that this extra mode could arise from the orbital disorder induced by the Ti substitution, a scenario which can now be ruled out by the comparison of Ti-containing and Ti-free CRO. Figures 9(d) and 9(e) show
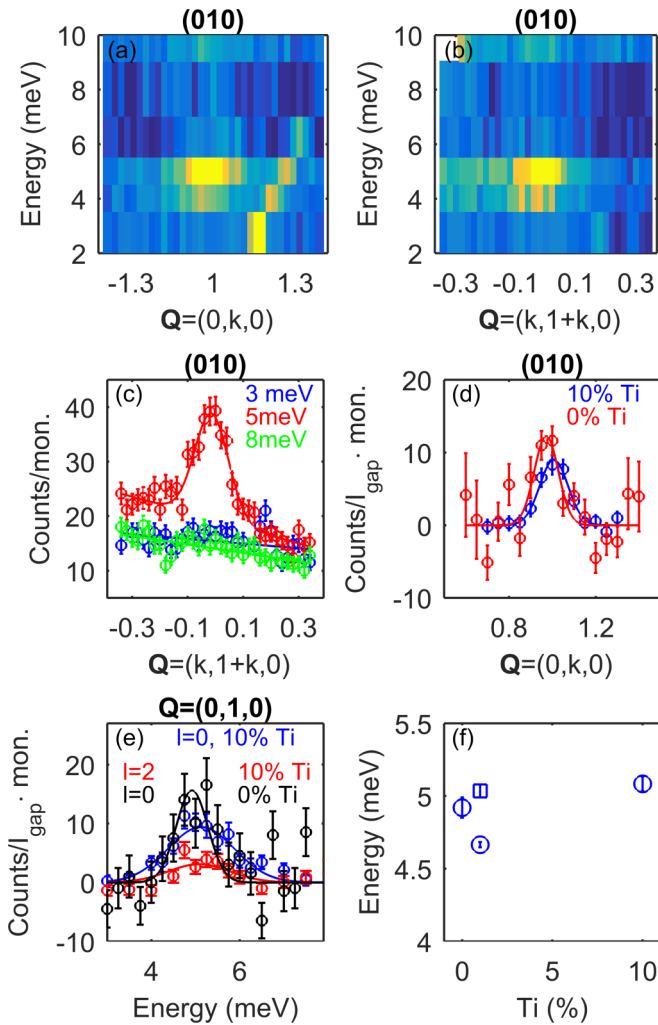
FIG. 9. Additional magnon branch near 5 meV. (a),(b) show scattering maps of energy versus **Q** taken on 1Ti. (c) presents some characteristic scans included in (b) for an energy transfer of 3 meV (5 meV), (8 meV) in blue (red), (green). The data in (a)–(c) are taken from Ref. [24]. (d) shows **Q** scans with an energy transfer of 5 meV for 10Ti (blue) and 0Ti (red). The intensities are scaled with the maximum intensities of the scans in Fig. 5. Scaled in the same way are the energy scans in (e) at **Q** = (1,0,l) using the crystals 10Ti, $l = 0$ (blue), $l = 2$ (red) and 0Ti, $l = 2$ (black). (f) presents the energy of this additional mode at **Q** = (0,1,0) for crystals containing different amounts of Ti.

Q and energy scans across the 5 meV signal on samples with 0 and 10% of Ti. Most interestingly there seems to be no difference in the strength of the low-energy signal when normalized to that of the in-plane gap mode at ~14 meV. The additional signal is thus intrinsic to the magnetic order in CRO, and it is not induced by disorder, which must be significantly enhanced in Ti10. The Ti content furthermore does not change the energy of the additional scattering, see the fitted energy maxima displayed in Fig. 9(f). The minor variation between measurements on the same concentration on different spectrometers can be attributed to the energy calibration.
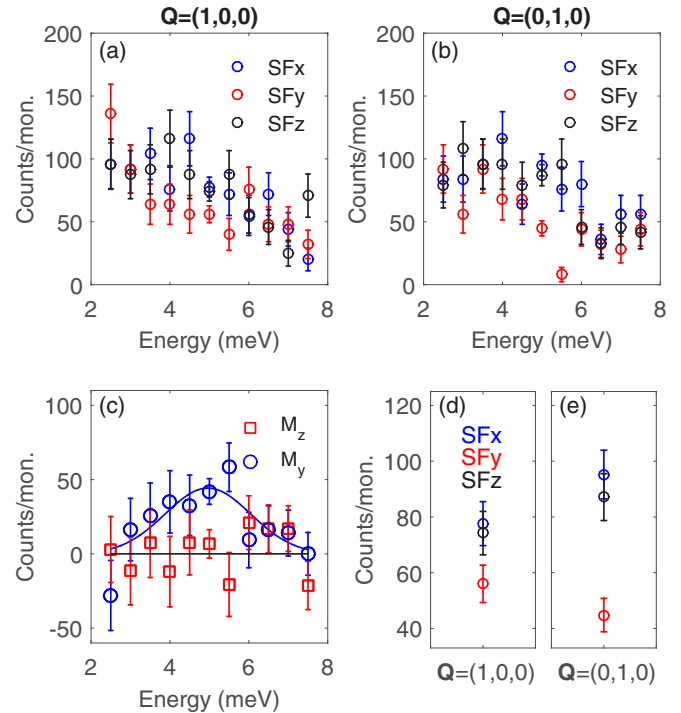

FIG. 10. Polarization analysis of the additional mode. SF intensities of energy scans at (a),(d) **Q** = (1,0,0), and (b),(e) **Q** = (0,1,0) for the SFx (SFy), (SFz) channel in blue (red), (black). (c) shows the results of the polarization analysis with Eq. (1), in red M$_z$ and in blue M$_y$. (d),(e) present only the 5 meV data of the two **Q** positions, which are measured with higher statistics.

The polarization of the additional mode was determined with polarized neutron scattering following the same procedures as those used to determine the character of the dispersion at higher energies. The polarization analysis directly shows that the additional excitation possesses an in-plane character, because it is not observed in the SFy channel (or in M_z), see Fig. 10. Comparing the intensities at **Q** = (1,0,0) and **Q** = (0,1,0) one may furthermore deduce that the additional mode is a polarized, which is transverse with respect to the static moment. The expected ratio of the intensities at 5 meV between **Q** = (0,1,0) and **Q** = (1,0,0) based on the twinning fractions and the geometry factor is 1:2.5 for an in-plane transverse mode and 2.5:1 for an in-plane longitudinal mode, respectively. The measured ratio is 0.25(21) characterizing this mode as an in-plane transverse mode.

The simple antiferromagnetic square lattice exhibits a magnon dispersion consisting of two transverse branches, which were both observed at larger energy in CRO, see above. In order to explain the additional mode the model needs to be extended, and the weakness and localization of the signal point to a smaller moment with smaller coupling.

The GGA+U calculations on CRO reveal a small magnetic moment situated at the two apical oxygens of 0.11 μ_B each. Such a polarization of the oxygen results from the strong hybridization between Ru 4d and O 2p orbitals. DFT calculations on several ferromagnetic or nearly ferromagnetic ruthenates find a sizable amount of magnetic moment on O, which in total sum up to about 30% of the entire magnetization

[52]. Experimentally this sizable magnetization of the O states has been observed by polarized neutron diffraction on $Ca_{1.5}Sr_{0.5}RuO_4$ where indeed 33% of the magnetization stems from O states [53]. CRO, however, exhibits antiferromagnetic order, in which the O bridging two antiparallel Ru moments cannot be polarized, see Fig. 1(a). Therefore, there is no moment on the in-plane O sites, which is also confirmed by the DFT calculations of CRO. However, the apical O is connected only to a single Ru site and can be polarized.

There are two principal contributions to the exchange between Ru and O moments due to hopping between O $2p$ and Ru $4d$ orbitals in the scenario where the Ru d_{xy} orbital is doubly occupied and the d_{xz} and d_{yz} singly occupied, see, e.g., Ref. [21]. One is a hopping from the p_x and p_y to the d_{xz} and d_{yz} orbitals, respectively. There, because of Pauli exclusion principle, only antiparallel spins can hop, leaving parallel spins in the now singly occupied O p orbitals. In total this yields a ferromagnetic coupling:

$$J_{Ru-O}^{t_{2g}-p} \sim \frac{-2t_{pd\pi}^2}{\varepsilon_{t_{2g}} - \varepsilon_p}, \qquad (3)$$

where $t_{pd\pi}$ is the hopping integral between d_{xz} and p_x, and d_{yz} and p_y orbitals of Ru and oxygen. $\varepsilon_{t_{2g}}$ and ε_p are the centers of the Ru t_{2g} and apical O p bands. The factor two in Eq. (3) reflects the fact that there are two processes (virtual hoppings from p_x to d_{zx} and from p_y to d_{yz} orbitals) contributing to this ferromagnetic Ru-O exchange.

The second contribution to the Ru-O exchange comes from the hopping from the p_z orbital of O to the empty $d_{3z^2-r^2}$ orbital of Ru. Because of Hund's rule, predominantly parallel spins hop, leaving anitparallel spins on the O p_z orbital. This yields an antiferromagnetic coupling between Ru and O moments:

$$J_{Ru-O}^{e_g-p} \sim \frac{t_{pd\sigma}^2}{\varepsilon_{e_g} - \varepsilon_p} \frac{J_H}{\varepsilon_{e_g} - \varepsilon_p}. \qquad (4)$$

Here $t_{pd\sigma}$ is the hopping matrix element between a p_z orbital of O and an empty $d_{z^2-r^2}$ orbital of Ru, cf. Ref. [21].

The σ hopping is larger than the π one ($|t_{pd\sigma}| \sim 2.2|t_{pd\pi}|$) [54], but the antiferromagnetic exchange (Eq. 4) occurs through the e_g band, which lies much higher in energy than t_{2g}. Our GGA+U calculations show that $\varepsilon_{t_{2g}} - \varepsilon_p \sim 1$ eV, while $\varepsilon_{e_g} - \varepsilon_p \sim 4.7$ eV. There is only one such antiferromagnetic exchange path, in contrast to two for ferromagnetic exchanges [Eq. (3)] and Hunds exchange on the Ru, $J_H \sim 0.7$ eV, also slightly reduces this exchange [Eq. (4)]. Therefore, one should expect ferromagnetic coupling to dominate, so that the moment of oxygen would be parallel to the moment of neighboring Ru. Our DFT calculations support this conclusion.

We have extended the spin-wave model by a small moment on the apical oxygens, 0.11 μ_B, coupled ferromagnetically to the next Ru moment, see Fig. 1(a). This extension necessitates to reduce the single-ion anisotropy parameter γ in order to keep the anisotropy gap of the in-plane transverse mode at the correct energy. The additional mode cannot be obtained at an energy near 5 meV at the antiferromagnetic zone center without introducing an anisotropy in terms of a single ion anisotropy for the magnetic moment on the O sites or an anisotropic coupling parameter. Because the single ion anisotropy stems from SOC which is small in O, the
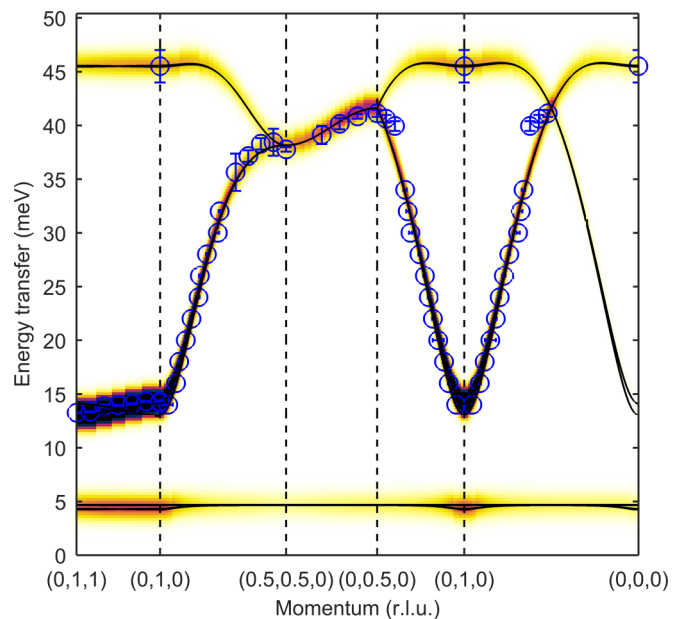

FIG. 11. Magnon dispersion calculated with the SpinW program including the additional mode. The labeling corresponds to the one in Fig. 7.

first scenario seems to be less likely. A better choice is an anisotropic coupling, because the Ru with its larger SOC is also involved in this process. Therefore, we used $J_{RuO} = (-1.5, -3.5, -1.5)$ meV to fit the experimental results. With this anisotropic coupling, γ has to be reduced to 0.5 meV to keep the in-plane transverse mode at the experimental value. With this extension we can perfectly describe the additional feature, see Fig. 11. The flat dispersion of the additional branch results from the lack of coupling between two O moments and perfectly describes the experimental finding. Also the localization of the signal strength at the antiferromagnetic zone center is well reproduced by this model. We can, therefore, conclude that the additional moment, situated on the apical O site and ferromagnetically coupled to the Ru, can explain the additional low-energy magnetic signal in CRO. We also tried to reproduce the data with the O moments being antiferromagnetically coupled to Ru, but such a scenario cannot explain the location of the scattering in Q space.

The significant polarization of the O site is another consequence of the strong hybridization of the Ru $4d$ and O p orbitals resulting in large hopping parameters, which can already be deduced from the closeness of the metallic phase of CRO. It furthermore yields a convincing explanation of the reduced moment, because the in-plane oxygen ion, which should carry a similar amount of charge carriers, cannot be magnetically polarized in the antiferromagnetic structure reducing the ordered moment from the simple $S = 1$ expectation. The ordered moment may thus also oscillate in length by transferring moment between Ru and in-plane O, which can give rise to a longitudinal mode, as an alternative explanation for the observation in Ref. [25].

IV. CONCLUSIONS

10% Ti substitution in CRO results in a stabilization of the insulating phase up to at least 700 K and reduces the

TABLE I. Structural data of the single crystal x-ray diffraction analysis.

Ti content	10%	10%	1%	1%	0%	0%
Temperature	100 K	293 K	100 K	293 K	100 K	293 K
wR(all,F)	0.022	0.0429	0.0521	0.048	0.0549	0.0262
Ru1, Ti1						
x	0	0	0	0	0	0
y	0	0	0	0	0	0
z	0	0	0	0	0	0
U_{11}	0.00119(3)	0.00284(3)	0.00197(5)	0.00348(5)	0.00295(4)	0.00368(4)
U_{22}	0.00164(4)	0.00330(4)	0.00245(5)	0.00368(5)	0.00373(5)	0.00409(5)
U_{33}	0.00151(4)	0.00378(3)	0.00262(4)	0.00390(4)	0.00351(5)	0.00403(4)
U_{12}	−0.00003(3)	−0.000019(15)	0.00000(4)	−0.00001(2)	0.00001(3)	−0.000015(12)
U_{13}	−0.00009(2)	−0.000172(13)	−0.00005(3)	−0.00019(2)	−0.00011(4)	−0.000187(12)
U_{23}	−0.00011(2)	−0.00025(2)	−0.00012(3)	−0.00021(2)	−0.00006(3)	−0.000228(14)
occupancy Ru	0.870(2)	0.866(2)	0.969(3)	0.978(3)	0.9768(18)	0.9859(17)
occupancy Ti	0.130(2)	0.134(2)	0.031(3)	0.022(3)		
Ca1						
x	0.00753(3)	0.01019(2)	0.00479(4)	0.00912(3)	0.00513(4)	0.00909(2)
y	0.05003(4)	0.03986(4)	0.05634(5)	0.04365(5)	0.05536(4)	0.04350(5)
z	0.352152(13)	0.351206(12)	0.352293(17)	0.350910(16)	0.35217(2)	0.350894(13)
U_{11}	0.00420(6)	0.00988(5)	0.00485(7)	0.01022(8)	0.00599(7)	0.01034(6)
U_{22}	0.00526(8)	0.01163(7)	0.00530(9)	0.01104(9)	0.00678(8)	0.01153(8)
U_{33}	0.00261(5)	0.00530(5)	0.00360(7)	0.00523(7)	0.00460(8)	0.00537(6)
U_{12}	−0.00092(5)	−0.00239(4)	−0.00077(7)	−0.00227(5)	−0.00081(6)	−0.00220(3)
U_{13}	−0.00002(4)	0.00020(3)	0.00007(6)	0.00032(4)	0.00004(7)	0.00030(3)
U_{23}	−0.00047(4)	−0.00047(4)	−0.00022(6)	−0.00023(6)	−0.00021(6)	−0.00026(4)
occupancy	1	1	1	1	1	1
O1						
x	0.19908(12)	0.20016(9)	0.19577(15)	0.19758(12)	0.19604(16)	0.19770(9)
y	0.29878(14)	0.29890(10)	0.30051(15)	0.30081(12)	0.30057(16)	0.30061(9)
z	0.02532(6)	0.02228(5)	0.02700(7)	0.02325(6)	0.02681(9)	0.02322(4)
U_{11}	0.0048(2)	0.00732(15)	0.0049(3)	0.0072(2)	0.0057(3)	0.00734(16)
U_{22}	0.0052(3)	0.00788(18)	0.0050(3)	0.0068(2)	0.0059(3)	0.00695(16)
U_{33}	0.0060(2)	0.01175(18)	0.0061(3)	0.0108(2)	0.0074(3)	0.01106(17)
U_{12}	−0.0024(2)	−0.00339(13)	−0.0019(2)	−0.00312(18)	−0.0016(2)	−0.00263(12)
U_{13}	−0.00055(17)	0.00025(14)	0.0006(2)	0.00088(18)	0.0005(3)	0.00107(13)
U_{23}	0.00039(19)	−0.00013(15)	0.0000(2)	−0.0010(2)	−0.0004(3)	−0.00078(14)
occupancy	1.033(4)	1.028(3)	1.016(4)	1.014(4)	1.012(5)	1.009(4)
O2						
x	−0.06456(13)	−0.05697(12)	−0.06762(18)	−0.05808(18)	−0.06701(18)	−0.05813(13)
y	−0.01939(14)	−0.01561(11)	−0.02114(16)	−0.01661(13)	−0.02093(16)	−0.01684(10)
z	0.16450(6)	0.16468(4)	0.16458(7)	0.16488(6)	0.16453(9)	0.16491(4)
U_{11}	0.0064(2)	0.0121(2)	0.0071(3)	0.0128(3)	0.0082(3)	0.0123(3)
U_{22}	0.0072(3)	0.0132(2)	0.0067(3)	0.0125(3)	0.0080(3)	0.0125(2)
U_{33}	0.0034(2)	0.00558(16)	0.0045(3)	0.0055(2)	0.0059(3)	0.00531(16)
U_{12}	0.00075(20)	0.00089(15)	0.0005(2)	0.0009(2)	0.0004(2)	0.00108(15)
U_{13}	0.00013(16)	0.00065(14)	0.0004(2)	0.00065(19)	−0.0001(3)	0.00066(15)
U_{23}	0.00001(18)	0.00023(13)	0.0002(2)	0.0002(2)	0.0002(3)	−0.00008(13)
occupancy	1.046(5)	1.040(5)	1.025(5)	1.029(5)	1.028(6)	1.012(5)

structural distortions, but magnetic properties are very little changed. Crystals with only 1% of Ti show only minor shifts of structural and magnetic transition temperatures and can thus be taken as representative for pure CRO.

Various neutron scattering experiments on partially untwinned crystals of Ca$_2$Ru$_x$Ti$_{1-x}$O$_4$ were performed to determine the magnetic structure and magnon dispersion in the insulating state. Most remarkable is the large magnetic anisotropy, which results in a splitting of the two transverse

zone-center modes that exceeds the full dispersion because of magnetic exchange. The entire dispersion of transverse magnon branches, however, is perfectly described by a conventional spin-wave model, in which the strong impact of the SOC is reflected by large anisotropy parameters. DFT calculations within the GGA+U approximation yield an orbital moment of only $0.13 \, \mu_B$, which indicates that CRO is not close to a $j = 0$ state, which can be seen in the discrepancy of the experimental magnon dispersion with the

$j = 0$ calculation [23]. The calculated magnetic interaction and anisotropy parameters reasonably well agree with those obtained by fitting the dispersion, and also the ordered moment is correctly reproduced in the DFT calculation.

The DFT study finds a sizable amount of magnetic moment on the apical oxygen reminiscent of previous reports on ferromagnetic ruthenates. This additional moment explains an additional signal appearing in the neutron scattering experiments at lower energy and limited to the antiferromagnetic zone center. Extending the spin-wave model to the weakly ferromagnetic coupled oxygen moments describes the flat dispersion and the limited appearance in **Q** space of this signal. The spin-wave dispersion in CRO is thus dominated by the impact of strong SOC and by the presence of magnetic moments on the oxygen sites.

Note added in proof. Recently, a revised version of Ref. [25] appeared as [55].

TABLE II. Structural and physical characteristics of crystals containing different amounts of Ti.

	0% Ti	1% Ti	10% Ti
T_N (K)	112.6(2)	112.3(2)	107.2(3)
T_{MI} (K)	362(1)	358(1)	
300 K			
a (Å)	5.4098(3)	5.4098(3)	5.4247(4)
b (Å)	5.4691(4)	5.4683(4)	5.4585(5)
c (Å)	11.9745(6)	11.9781(9)	11.9536(9)
ϵ	0.00545(14)	0.00537(17)	0.00311(17)
$Ru - O1_{aver.}$ (Å)	1.9816(8)	1.9852(7)	1.9779(5)
$Ru - O2$ (Å)	2.0018(9)	2.003(8)	1.9932(5)
$O - O \| a$ (Å)	2.815(1)	2.817(1)	2.815(1)
$O - O \| b$ (Å)	2.792(1)	2.792(1)	2.782(1)
$\theta - O1$ (deg)	11.253(17)	11.38(2)	10.896(12)
$\theta - O2$ (deg)	9.278(20)	9.423(2)	9.260(9)
ϕ (deg)	11.628(16)	11.666(13)	11.171(11)
100 K			
a (Å)	5.377(11)	5.3957(3)	5.4189(4)
b (Å)	5.5915(12)	5.6023(3)	5.5483(5)
c (Å)	11.789(3)	11.7725(7)	11.7982(10)
ϵ	0.0176(2)	0.0188(4)	0.0118(6)
$Ru - O1_{aver.}$ (Å)	2.0099(6)	2.0120(6)	1.9996(5)
$Ru - O2$ (Å)	1.9765(12)	1.9751(8)	1.9753(7)
$O - O \| a$ (Å)	2.829(1)	2.829(1)	2.827(1)
$O - O \| b$ (Å)	2.856(1)	2.862(1)	2.829(1)
$\theta - O1$ (deg)	12.912(21)	12.986(16)	12.201(14)
$\theta - O2$ (deg)	11.089(19)	11.199(19)	10.680(11)
ϕ (deg)	11.801(18)	11.823(17)	11.287(15)

APPENDIX A: CRYSTAL STRUCTURE DETERMINATION OF $Ca_2Ru_xTi_{1-x}O_4$

Complete crystal structure analyses were performed with an X8-APEX by Bruker AXS single-crystal diffractometer with a goniometer in kappa geometry and x-ray radiation from a molybdenum anode with a wavelength of $\lambda = 0.71073$ Å. The distance between the sample and the detector was set to 50 mm. Structure refinements were carried out using Jana2006 [56]. A type I extinction correction was applied during the refinements and the data were corrected for absorption. The thermal parameters for Ru1 and Ti1 were constrained to be equal, and the total occupancy of this site was fixed to 1. The results of the structural refinements are given in Table I. Table II presents further characteristics of the crystal structure as well as the metal-insulator- and antiferromagnetic transition temperatures. T_N is determined from the magnetization curves presented in Fig. 2(a) by finding the zero point of the second
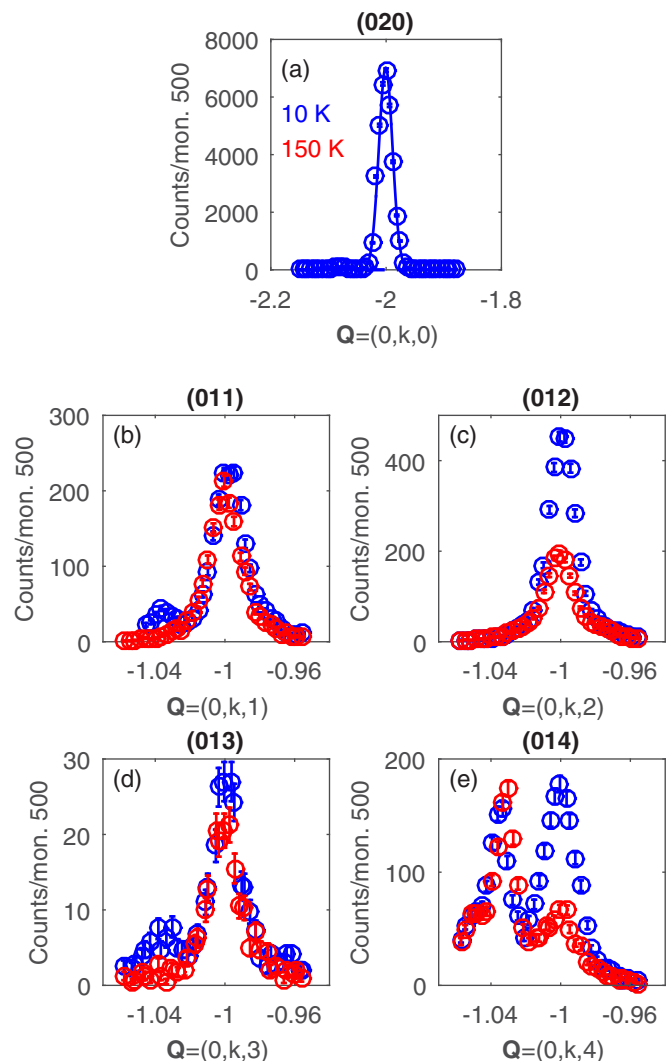

FIG. 12. Elastic neutron scattering scans of 1TiB across (a) **Q** = (0,2,0), (b) **Q** = (0,1,0), (c) **Q** = (0,1,*l*), (d) **Q** = (0,1,2), (e) **Q** = (0,1,2). Blue color denotes scans taken in the magnetic phase at 10 K and red color scans taken in the nonmagnetic phase at 150 K.

derivation. T_{MIT} is determined in the same way from the resistance curves presented in this figure.

APPENDIX B: MAGNETIC STRUCTURE DETERMINATION OF Ca$_2$Ru$_{0.99}$Ti$_{0.01}$O$_4$

The crystal 1TiB used in our previous neutron scattering study [24] was further investigated on the thermal TAS IN3 at the ILL. The crystal was mounted in the [010]/[001] orientation into a ILL orange cryostat, k_f was set to 2.662 Å^{-1} for all scans and a pyrolitic graphite filter was used to suppress higher order wavelength. Rocking scans revealed the mosaic spread to be less than 0.5°. With the good resolution of the IN3 it is easily possible to resolve the Bragg peaks from the two twins present in the crystal. The twinning ratio amounts to 20:1, which is deduced from rocking scans of strong nuclear Bragg reflections at the 2Θ scattering angles of the (020) and

(200) reflections. Figure 12(a) shows a longitudinal scan over the (020) reflection; the smaller twin is barely seen in the tail. Scans across the positions, where magnetic scattering can be expected, are shown in (b)–(e). There several scans of the kind (01l) are compared at 10 K, in the antiferromagnetic phase, and at 150 K, in the nonmagnetic phase. It is clearly seen that the minority twin gains intensity for an odd l and not for an even l. The majority twin gains intensity for even l and much less, but also clearly detectable, for odd l. The (012) and (014) reflections are magnetic Bragg peaks for the B-centered phase but not for the A-centered one. So the strong gain of intensity of these reflections in the antiferromagnetic phase reveals an almost exclusive B-centered phase. Opposite the (011) and (013) peaks are Bragg peaks in the A-centered phase, but not in the B-centered one. So the gain of intensity at low temperatures of those peaks points to the A-centered phase, which is however much weaker than the gain of the Bragg peak intensities of the B-centered phase. So the main stacking scheme of this crystal is the B-centered one.

[1] S. Nakatsuji, S. Ikeda, and Y. Maeno, J. Phys. Soc. Jpn. **66**, 1868 (1997).

[2] S. Nakatsuji and Y. Maeno, Phys. Rev. Lett. **84**, 2666 (2000).

[3] O. Friedt, M. Braden, G. André, P. Adelmann, S. Nakatsuji, and Y. Maeno, Phys. Rev. B **63**, 174432 (2001).

[4] J. P. Carlo, T. Goko, I. M. Gat-Malureanu, P. L. Russo, A. T. Savici, A. A. Aczel, G. J. MacDougall, J. A. Rodriguez, T. J. Williams, G. M. Luke, C. R. Wiebe, Y. Yoshida, S. Nakatsuji, Y. Maeno, T. Taniguchi, and Y. J. Uemura, Nat. Mater. **11**, 323 (2012).

[5] S. Nakatsuji, V. Dobrosavljević, D. Tanasković, M. Minakata, H. Fukazawa, and Y. Maeno, Phys. Rev. Lett. **93**, 146401 (2004).

[6] Y. Maeno, H. Hashimoto, K. Yoshida, S. Nishizaki, T. Fujita, J. G. Bednorz, and F. Lichtenberg, Nature (London) **372**, 532 (1994).

[7] K. Ishida, H. Mukuda, Y. Kitaoka, K. Asayama, Z. Q. Mao, Y. Mori, and Y. Maeno, Nature (London) **396**, 658 (1998).

[8] G. M. Luke, Y. Fudamoto, K. M. Kojima, M. I. Larkin, J. Merrin, B. Nachumi, Y. J. Uemura, Y. Maeno, Z. Q. Mao, Y. Mori, H. Nakamura, and M. Sigrist, Nature (London) **394**, 558 (1998).

[9] Y. Maeno, S. Kittaka, T. Nomura, S. Yonezawa, and K. Ishida, J. Phys. Soc. Jpn. **81**, 011009 (2012).

[10] C. S. Alexander, G. Cao, V. Dobrosavljevic, S. McCall, J. E. Crow, E. Lochner, and R. P. Guertin, Phys. Rev. B **60**, R8422 (1999).

[11] M. Braden, G. André, S. Nakatsuji, and Y. Maeno, Phys. Rev. B **58**, 847 (1998).

[12] T. Mizokawa, L. H. Tjeng, G. A. Sawatzky, G. Ghiringhelli, O. Tjernberg, N. B. Brookes, H. Fukazawa, S. Nakatsuji, and Y. Maeno, Phys. Rev. Lett. **87**, 077202 (2001).

[13] E. Gorelov, M. Karolak, T. O. Wehling, F. Lechermann, A. I. Lichtenstein, and E. Pavarini, Phys. Rev. Lett. **104**, 226401 (2010).

[14] D. Sutter, C. G. Fatuzzo, S. Moser, M. Kim, R. Fittipaldi, A. Vecchione, V. Granata, Y. Sassa, F. Cossalter, G. Gatti, M. Grioni, H. M. Rønnow, N. C. Plumb, C. E. Matt, M. Shi, M. Hoesch, T. K. Kim, T.-R. Chang, H.-T. Jeng, C. Jozwiak, A.

Bostwick, E. Rotenberg, A. Georges, T. Neupert, and J. Chang, Nat. Commun. **8**, 15176 (2017).

[15] J. H. Jung, Z. Fang, J. P. He, Y. Kaneko, Y. Okimoto, and Y. Tokura, Phys. Rev. Lett. **91**, 056403 (2003).

[16] T. Hotta and E. Dagotto, Phys. Rev. Lett. **88**, 017201 (2001).

[17] A. V. Puchkov, M. C. Schabel, D. N. Basov, T. Startseva, G. Cao, T. Timusk, and Z.-X. Shen, Phys. Rev. Lett. **81**, 2747 (1998).

[18] A. Liebsch and H. Ishida, Phys. Rev. Lett. **98**, 216403 (2007).

[19] A. Liebsch, Europhys. Lett. **63**, 97 (2003).

[20] V. I. Anisimov, I. A. Nekrasov, D. E. Kondakov, T. M. Rice, and M. Sigrist, Eur. Phys. J. B **25**, 191 (2002).

[21] D. I. Khomskii, *Transition Metal Compounds* (Cambridge University Press, Cambridge, England, 2014).

[22] G. Khaliullin, Phys. Rev. Lett. **111**, 197201 (2013).

[23] A. Akbari and G. Khaliullin, Phys. Rev. B **90**, 035137 (2014).

[24] S. Kunkemöller, D. Khomskii, P. Steffens, A. Piovano, A. A. Nugroho, and M. Braden, Phys. Rev. Lett. **115**, 247201 (2015).

[25] A. Jain, M. Krautloher, J. Porras, G. H. Ryu, D. P. Chen, D. L. Abernathy, J. T. Park, A. Ivanov, J. Chaloupka, G. Khaliullin, B. Keimer, and B. J. Kim, arXiv:1510.07011.

[26] H. Fukazawa, S. Nakatsuji, and Y. Maeno, Physica B (Amsterdam) **281**, 613 (2000).

[27] S. Kunkemöller, F. Sauer, A. A. Nugroho, and M. Braden, Cryst. Res. Technol. **51**, 299 (2016).

[28] J. Rodríguez-Carvajal, Physica B (Amsterdam) **192**, 55 (1993).

[29] S. Toth and B. Lake, J. Phys. Condens. Matter **27**, 166002 (2015).

[30] In *Neutron Scattering from Magnetic Materials*, edited by T. Chatterji (Elsevier Science, Amsterdam, 2005).

[31] N. Qureshi, P. Steffens, S. Wurmehl, S. Aswartham, B. Büchner, and M. Braden, Phys. Rev. B **86**, 060410 (2012).

[32] J. J. Randall and R. Ward, J. Am. Chem. Soc **81**, 2629 (1959).

[33] M. Braden, W. Reichardt, S. Nishizaki, Y. Mori, and Y. Maeno, Phys. Rev. B **57**, 1236 (1998).

[34] M. Braden, W. Reichardt, Y. Sidis, Z. Mao, and Y. Maeno, Phys. Rev. B **76**, 014505 (2007).

[35] M. Braden, A. H. Moudden, S. Nishizaki, Y. Maeno, and T. Fujita, Physica C (Amsterdam) **273**, 248 (1997).

[36] X. Ke, J. Peng, D. J. Singh, T. Hong, W. Tian, C. R. Dela Cruz, and Z. Q. Mao, Phys. Rev. B **84**, 201102 (2011).

[37] M. Braden, O. Friedt, Y. Sidis, P. Bourges, M. Minakata, and Y. Maeno, Phys. Rev. Lett. **88**, 197002 (2002).

[38] P. Steffens, J. Farrell, S. Price, A. P. Mackenzie, Y. Sidis, K. Schmalzl, and M. Braden, Phys. Rev. B **79**, 054422 (2009).

[39] P. Steffens, O. Friedt, P. Alireza, W. G. Marshall, W. Schmidt, F. Nakamura, S. Nakatsuji, Y. Maeno, R. Lengsdorf, M. M. Abd-Elmeguid, and M. Braden, Phys. Rev. B **72**, 094104 (2005).

[40] W. Schweika, S. V. Maleev, T. Brückel, V. P. Plakhty, and L.-P. Regnault, Europhys. Lett. **60**, 446 (2002).

[41] I. U. Heilmann, J. K. Kjems, Y. Endoh, G. F. Reiter, G. Shirane, and R. J. Birgeneau, Phys. Rev. B **24**, 3939 (1981).

[42] C. Wang, R. Zhang, F. Wang, H. Luo, L. P. Regnault, P. Dai, and Y. Li, Phys. Rev. X **3**, 041036 (2013).

[43] M. Fidrysiak, Eur. Phys. J. B **89**, 41 (2016).

[44] F. Waßer, C. H. Lee, K. Kihou, P. Steffens, K. Schmalzl, N. Qureshi, and M. Braden, arXiv:1609.02027.

[45] Jain *et al.* report evidence for a longitudinal mode appearing at 52 meV at the FM zone center. $\mathbf{Q} = (2,2,0)$ is the smallest scattering vector for which an energy transfer of 52 meV at a FM zone center is not prohibited by kinematics in our setup using $k_f = 4.1\,\text{Å}^{-1}$. A smaller $\mathbf{Q}$ is however required due to the magnetic form factor in order to detect weak magnetic signals. In spite of very long counting we could not pick up a magnetic signal at this position. We also tried to find an almost equivalent longitudinal magnetic signal at $\mathbf{Q} = (2.2,0,0)$, where the longitudinal branch proposed in Ref. [25] should not possess an energy far from 52 meV and exhibit about half of the strength of the transversal branch at 45 meV at $\mathbf{Q} = (0,2,0)$, which we found to have 150(40) Counts/(mon 500000). At $\mathbf{Q} = (2.2,0,0)$ we could not pick up a sizable magnetic signal [13(34) and 16(39) Counts/(mon. 500000) for M_y and M_z, respectively]. So we cannot deliver experimental hints for this longitudinal mode, but we cannot exclude its presence due to the big error bars and the different $\mathbf{Q}$ position. Note that one data point in one spin flip channel is measured for 50 minutes.

[46] P. Blaha, K. Schwarz, G. Madsen, D. Kvasnicka, and J. Luitz, *WIEN2k, An Augmented Plane Wave + Local Orbitals Program for Calculating Crystal Properties* (Techn. Universität Wien, Wien, 2001).

[47] J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. **77**, 3865 (1996).

[48] S. Lee, J.-G. Park, D. T. Adroja, D. Khomskii, S. Streltsov, K. A. McEwen, H. Sakai, K. Yoshimura, V. I. Anisimov, D. Mori, R. Kanno, and R. Ibberson, Nat. Mater. **5**, 471 (2006).

[49] S. V. Streltsov and D. I. Khomskii, Phys. Rev. B **86**, 064429 (2012).

[50] V. I. Anisimov, F. Aryasetiawan, and A. I. Lichtenstein, J. Phys. Condens. Matter **9**, 767 (1997).

[51] G. Zhang and E. Pavarini, Phys. Rev. B **95**, 075145 (2017).

[52] I. I. Mazin and D. J. Singh, Phys. Rev. B **56**, 2556 (1997).

[53] A. Gukasov, M. Braden, R. J. Papoular, S. Nakatsuji, and Y. Maeno, Phys. Rev. Lett. **89**, 087202 (2002).

[54] W. A. Harrison, *Elementary Electronic Structure* (World Scientific, Singapore, 1999), p. 817.

[55] A. Jain, M. Krautloher, J. Porras, G. H. Ryu, D. P. Chen, D. L. Abernathy *et al.*, Nature Physics, doi:10.1038/nphys4077; arXiv:1705.00222.

[56] V. Petříček, M. Dušek, and L. Palatinus, Z. Kristallogr. **229**, 345 (2014).