

Structure and magnetism in the bond-frustrated spinel ZnCr_2Se_4 P. Zajdel,¹ W.-Y. Li,² W. van Beek,³ A. Lappas,⁴ A. Ziolkowska,⁵ S. Jaskiewicz,⁵ C. Stock,⁵ and M. A. Green^{6,*}¹*Institute of Physics, University of Silesia, ulica Uniwersytecka 4, 40007 Katowice, Poland*²*Department of Chemistry, University College London, Gordon Street, London WC1H 0AJ, United Kingdom*³*Swiss Norwegian Beam lines, European Synchrotron Radiation Facility, Polygone Scientifique Louis Neel, 6 Rue Jules Horowitz, 38000 Grenoble, France*⁴*Institute of Electronic Structure and Laser, Foundation for Research and Technology – Hellas, 71 110 Heraklion, Greece*⁵*School of Physics and Astronomy, University of Edinburgh, Edinburgh, EH9 3JZ, United Kingdom*⁶*School of Physical Sciences, Ingram Building, University of Kent, Canterbury, Kent CT2 7NH, United Kingdom*

(Received 19 September 2016; published 3 April 2017)

The crystal and magnetic structures of stoichiometric ZnCr_2Se_4 have been investigated using synchrotron x-ray and neutron powder diffraction, muon spin relaxation (μSR), and inelastic neutron scattering. Synchrotron x-ray diffraction shows a spin-lattice distortion from the cubic $Fd\bar{3}m$ spinel to a tetragonal $I4_1/amd$ lattice below $T_N = 21$ K, where powder neutron diffraction confirms the formation of a helical magnetic structure with magnetic moment of $3.04(3)\mu_B$ at 1.5 K, close to that expected for high-spin Cr^{3+} . μSR measurements show prominent local spin correlations that are established at temperatures considerably higher (< 100 K) than the onset of long-range magnetic order. The stretched exponential nature of the relaxation in the local spin-correlation regime suggests a wide distribution of depolarizing fields. Below T_N , unusually fast ($> 100\mu\text{s}^{-1}$) muon relaxation rates are suggestive of rapid site hopping of the muons in static field. Inelastic neutron scattering measurements show a gapless mode at an incommensurate propagation vector of $k = [000.4648(2)]$ in the low-temperature magnetic ordered phase that extends to 0.8 meV. The dispersion is modeled by a two-parameter Hamiltonian, containing ferromagnetic nearest-neighbor and antiferromagnetic next-nearest-neighbor interactions with a $J_{nnn}/J_{nn} = -0.337$.

DOI: [10.1103/PhysRevB.95.134401](https://doi.org/10.1103/PhysRevB.95.134401)**I. INTRODUCTION**

ZnCr_2Se_4 has been the subject of numerous studies in the past few decades since being reported as a canonical incommensurate simple helical magnet [1]. Its p -type semiconductor ($E_g = 0.3$ eV) [2] band structure contrasts with the more common itinerant ferromagnetic helical structures, such as that found in elemental Ho [3] or MnSi [4]. Such complex magnetic systems have recently attracted renewed attention due to their potential route to colossal magnetoresistance [5] and multiferroic properties [6,7]. More recently, the importance of incommensurate magnetic spiral structures has been highlighted in the field of iron-based superconductors and related systems [8–14]. ZnCr_2Se_4 displays both significant spin-phonon coupling [15] and magnetostriction [16], demonstrating the importance of fully understanding its structure and magnetism.

ZnCr_2Se_4 has a large positive Curie-Weiss temperature of $\theta_{CW} = 110$ K [17] as a result of strong near- 90° Cr-Se-Cr ferromagnetic superexchange, so it does not show the geometric frustration present in many spinels. However, bond frustration reduces the magnetic ordering temperature to 21 K [1], which is significantly below θ_{CW} , and is antiferromagnetic as a result of the collective strength of next-nearest-neighbor interactions [2,18]. Theoretical work has demonstrated that the stability of the helical structure in ZnCr_2Se_4 relies on the interplay of five magnetic exchange pathways [19], including the two large nearest-neighbor ferromagnetic Cr-Se-Cr superexchange terms ($\angle \sim 90^\circ$, $J \sim 25.4$ K), and next-nearest-neighbor Cr-Se-Cr-Se-Cr and Cr-Se-Zn-Se-Cr ($J' \sim 6.6$ K) antiferromagnetic exchange, which has been experimentally

corroborated [20]. The extent of the bond frustration in ZnCr_2Se_4 ($\theta_{CW}/T_N \sim 5.2$) is high even compared with other spinel structures. Spin fluctuations in ZnCr_2Se_4 were observed by diffuse magnetic scattering above T_N , suggesting a magnetic microdomain phase below 45 K [21,22]. Further studies conducted between the Curie-Weiss and Néel temperatures revealed the presence of negative thermal expansion below 100 K [16,23], shifting of the magnetic resonance line [20], anomalous behavior of eigenfrequencies of phonon modes [16], and deviation from Curie-Weiss behavior [1], making further investigation with muon spin-relaxation measurements attractive. The presence of local magnetic correlations is well established in the analogous oxide, ZnCr_2O_4 [24–26], but it possesses, in contrast to selenide, a large negative Curie-Weiss constant. Therefore, the suppression of long-range magnetic order in ZnCr_2O_4 ($\theta_{CW}/T_N = 32$) is the result of inherent geometric frustration and not only from bond frustration. The ability to probe local field distribution in time and space makes muon spin relaxation [27] an ideal technique to investigate the local magnetism in ZnCr_2Se_4 . For example, it was successfully employed in the investigation of the onset of local correlations above T_N in $\text{Li}_2\text{Mn}_2\text{O}_4$ [28].

Spinel compounds, such as ACr_2O_4 ($A = \text{Zn}$ or Cd) [26,29,30], often adopt lower-symmetry structures to relieve magnetic frustration. For ZnCr_2Se_4 the case is not clear as neutron diffraction results suggested an $Fddd$ orthorhombic structure [31], whereas synchrotron diffraction did not reveal a distinct structural transition, but showed unusual trends in Zn-Se bond lengths that were indicative of some structural modifications [32]. Previous neutron studies have reported a significantly reduced moment on the Cr ions in the helical arrangement, suggesting substantial magnetic disorder [32,33], possibly coming from a frustrated lattice configuration. Here

*m.green@kent.ac.uk

we show that ZnCr_2Se_4 adopts a tetragonal $I4_1/amd$ nuclear structure below T_N and the helical magnetic structure contains Cr moments of $3.04(4) \mu_B$; close to the theoretical spin-only values for high-spin Cr^{3+} . These synchrotron and neutron scattering studies are complemented with local muon spin-relaxation (μSR) magnetic studies and magnetic fluctuations through inelastic neutron scattering.

II. EXPERIMENTAL METHODS

ZnCr_2Se_4 was synthesized by combination of the elements in an evacuated ($<10^{-5}$ torr) quartz tube and heated at 900°C for several days with intermittent regrindings. Neutron scattering measurements were performed at the NIST Center for Neutron Research, National Institute of Standards and Technology, Gaithersburg, MD, USA. High-resolution neutron powder diffraction was performed on the BT-1 diffractometer between 1.2 and 300 K. A Ge (311) monochromator at a 75° take-off angle with $\lambda = 2.0782 \text{ \AA}$ was used to collect data over a $1.3\text{--}166.3^\circ$ angular range with a step size of 0.05° . The magnetic excitations in ZnCr_2Se_4 were studied using the Disk Chopper Spectrometer (DCS) with $E_i = 3.3$ and 1.7 meV. Long-wavelength neutron powder diffraction data were extracted between -0.2 and 0.2 meV from the elastic contribution of the DCS data at $\lambda = 5.0 \text{ \AA}$ at 1.5 K. Chemical analysis was performed using Prompt Gamma Activation Analysis [34] on the NG-7 instrument [35] at NIST Center for Neutron Research. High-resolution powder x-ray diffraction experiments were performed at both the Swiss-Norwegian and ID31 beamlines at the European Synchrotron Radiation Facility, Grenoble, France from 5 to 500 K, with wavelengths of $\lambda = 0.72323(1)$ and $\lambda = 0.40016(7) \text{ \AA}$, respectively. Both powder neutron and x-ray diffraction datasets were refined using the FULLPROF program [36]. Muon spin-relaxation measurements were performed on the MuSR instrument at ISIS facility, Rutherford Appleton Laboratory, United Kingdom, and the GPS/S μ S beamline of the Paul Scherrer Institut, Villigen, Switzerland. Data were fitted using the WIMDA package (ISIS) and MUFIT program (PSI).

III. RESULTS AND DISCUSSION

A. Synthesis

The synthesis temperature and heating times were found to be critical to the resultant composition of the structure, which also varied greatly with the target mass of the final product. Deviation of the stoichiometry is well established for these systems, as well as extensive changes to the resultant electronic properties [2]. In order to monitor the process and ensure the complete reaction of the constituent elements, a variety of characterization techniques were employed. Samples that were heated significantly below 900°C failed to react completely and showed poor crystallinity, while those heated significantly above this temperature proved not to achieve exact stoichiometry. The procedure that proved to be the most successful was to heat samples at 900°C in an evacuated tube for 24-h periods. Initial monitoring of the reaction was performed using laboratory x-ray powder diffraction. If unreacted starting materials were present, the samples were reground and returned to the furnace. Once the amount of unreacted elements dropped

below the level that could be detected by x-ray diffraction, Rietveld refinement of powder neutron diffraction data using the BT1 diffractometer at NIST was performed after each heating procedure until exact composition was achieved. The final stoichiometry of the sample used in the present study was confirmed to be $\text{Zn}_{1.003(6)}\text{Cr}_{2.01(1)}\text{Se}_{3.998(7)}$ by a combination of powder neutron diffraction and Prompt Gamma Activation Analysis. Additional confirmation was obtained from Rietveld refinement of the synchrotron powder diffraction data at room temperature using the ID31 diffractometer, which confirmed a composition of $\text{Zn}_{1.000(8)}\text{Cr}_{1.99(2)}\text{Se}_{4.01(3)}$. It was concluded that within the errors of these experiments the compound was exactly stoichiometric.

B. Variable-temperature powder synchrotron x-ray diffraction

To perform the symmetry analysis on the neutron diffraction data and determine the precise magnetic structure, it is essential to determine the low-temperature space group of the crystal structure. It is well established that strong spin-lattice effects lower the symmetry from the ideal $Fd\bar{3}m$ space group in many spinels. To evaluate the true symmetry below T_N , we performed high-resolution synchrotron x-ray diffraction measurements on ID31 at the European Synchrotron Radiation Facility (ESRF). The possible maximal subgroups of $Fd\bar{3}m$ are

- 141 $I4_1/amd$ tetragonal
- 166 $R\bar{3}m$ rhombohedral
- 203 $Fd\bar{3}$ cubic
- 210 $F4_132$ cubic
- 216 $F\bar{4}3m$ cubic

The data collected at 5 K, indexed to a tetragonal $I4_1/amd$ cell and full Rietveld refinement using the FULLPROF package, which are shown in Fig. 1(a), confirmed a distortion to a subgroup of $Fd\bar{3}m$. Attempts to refine the data in an orthorhombic symmetry including the recently proposed $Fddd$ symmetry [37,38] or other tetragonal symmetries led to poorer fits. A summary of the key tetragonal models is provided in Table I. Final refined parameters at 50 K were lattice parameters of $a = 10.486154(5) \text{ \AA}$, giving a cell volume of $1153.051(1) \text{ \AA}^3$ and atomic coordinates of Zn (0.125 0.125 0.125), Cr (0.5 0.5 0.5), and Se [0.25960(1) 0.25960(1) 0.25960(1)]. The refinement at 5 K for the $I4_1/amd$ symmetry gave lattice parameters of $a = 7.41947(1) \text{ \AA}$ and $c = 10.48653(1) \text{ \AA}$, which gives a cell volume of $577.267(1) \text{ \AA}^3$ and atomic coordinates of Zn (0 0.25 0.375), Cr (0 0 0), and Se [0 0.48066(3) 0.25944(2)]. A comparison of a section of the Rietveld refinement $\sim 13^\circ$ demonstrates the splitting of the [622] reflection in the cubic space group at 50 K, shown in Fig. 1(b), to the [422] and [206] reflections in the tetragonal symmetry, shown in Fig. 1(c).

At 50 K in the cubic $Fd\bar{3}m$ space group, the Zn-Se is in a perfect tetrahedron possessing $\bar{4}3m$ site symmetry. Any value of the structural parameter u in the Se ($u u u$) position above the ideal spinel value of 0.25 represents a movement of the Se anion down the [111] direction and a homogeneous expansion of the tetrahedron. In the case of ZnCr_2Se_4 at 50 K, the value

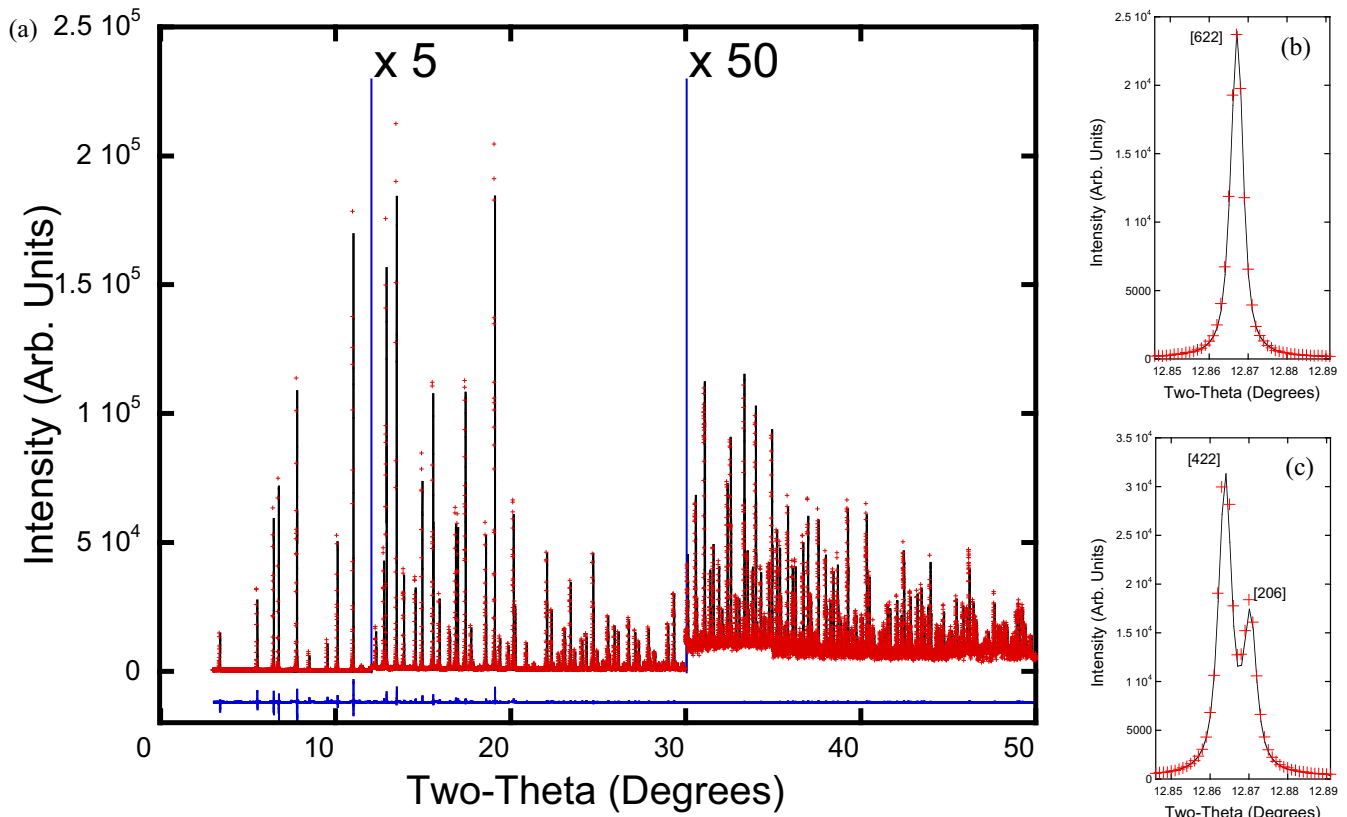


FIG. 1. Synchrotron x-ray powder diffraction of ZnCr_2Se_4 showing (a) the observed, calculated, and difference plot of the Rietveld refinement at 5 K and a comparison of a section of the pattern at (b) 50 K and (c) 5 K highlighting the tetragonal distortion to an $I4_1/amd$ space group at low temperature.

TABLE I. Comparisons of the structural parameters obtained from Rietveld refinements of three tetragonal models of ZnCr_2Se_4 at 5 K using the ID31 diffractometer, showing $I4_1/amd$ as the true space group below the magnetic ordering temperature.

	$I4_1/amd$ (Model 1)	$I\bar{4}m2$ (Model 2)	$Fddd$ (Model 3)
a (Å)	7.41947(1)	7.41946(1)	10.49279(1)
b (Å)			10.49234(1)
c (Å)	10.48653(1)	10.48653(1)	10.48650(1)
V (Å ³)	577.267(1)	577.267(1)	1154.501(2)
R_{exp} (%)	11.84	11.84	11.84
R_{wp} (%)	20.0	20.1	20.4
R_p (%)	17.3	17.4	17.8
χ^2	2.86	2.89	2.97
R_B	4.72	4.83	5.64
R_F	4.28	4.38	5.72
$U_{\text{eq}} \text{Zn}$ (Å ²)	0.0062(2)	0.0058(3)	0.0067(7)
$U_{\text{eq}} \text{Cr}$ (Å ²)	0.0059(2)	0.0061(2)	0.0067(7)
$U_{\text{eq}} \text{Se}$ (Å ²)	0.0054(2)	0.0053(5)	0.0060(6)
x Cr		0.2488(8)	
z Cr		0.3752(5)	
x Se1	0	0.7695(3)	0.25951(8)
y Se1	0.48066(3)	0	0.2599(1)
z Se1	0.25944(2)	0.6157(2)	0.25944(3)
x Se2		0.2694(3)	
z Se2		0.1350(2)	

of 0.259 60(1) represents an expansion of the tetrahedron by ~ 0.175 Å so that all Zn-Se bond distances are 2.4447(3) Å. There is a consequential reduction in the Cr-Se bonds by ~ 0.1 Å, which reduces the site symmetry from octahedral $m\bar{3}m$ to a $\bar{3}m$ trigonal antiprism symmetry. Although all six Cr-Se bonds are equidistant at 2.5249(1) Å, there is a distortion to six Se-Cr-Se bond angles to 85.343(7) Å and six Se-Cr-Se bond angles to 94.657(7) Å, which would be 90° at $u = 0.25$.

Calculating the equivalent cubic lattice parameters of $\sqrt{2} \times 7.419 47(1)$ for ZnCr_2Se_4 at 5 K gives a c/a ratio of 0.999 41. Figure 2(a) shows a comparison of the Zn-Se and Cr-Se bond lengths, along with the Se-Zn-Se bond angle, as a function of temperature from 5 to 50 K. Despite the significant magnetoelastic transition at T_N that reduces the symmetry to the $I4_1/amd$ space group, there is no difference within experimental error to either the four Zn-Se or the six Cr-Se bond distances. Within the transition the change in the c/a ratio is entirely accommodated by a modification to the CrSe_6 trigonal antiprism and consequently the ZnSe_4 units. This is best demonstrated by inspection of the Zn-Se bond angles as a function of temperature, which is shown in Fig. 2(a), where the ideal bond angle of 109.47° is distorted to two different values; at 5 K two are at 109.57(4) Å while the remaining four are at 109.42(2) Å. It is important to emphasize that despite this distortion and reduction in symmetry so that the two Cr-Se bonds along the c direction are inequivalent to the four Cr-Se bonds in the a - b plane, they remain, within the resolution of

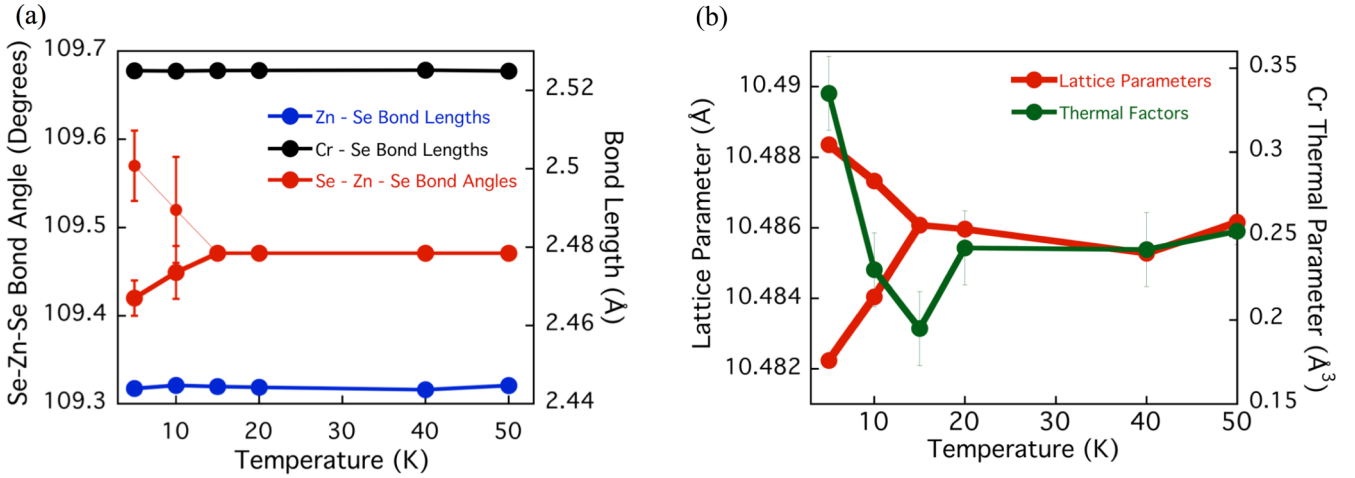


FIG. 2. Structural parameters obtained from fitting of the synchrotron powder x-ray diffraction data showing that (a) both the Zn-Se and Cr-Se bond lengths are insensitive to changes with temperature, whereas there is a significant modification of their bond angles through the magnetoelastic transition at $T_N = 21$ K and (b) the changes in the lattice parameters at the $Fd\bar{3}m \rightarrow I4_1/amd$ transition are accompanied by an increase in the Cr thermal factors.

the experiment, the same length; at 5 K two are at $2.5253(8)$ Å compared with four at $2.5250(5)$ Å. The reduction of the Cr site symmetry to $2/m$ produces four different Se-Cr-Se bond angles, $2 \times 85.33(2)^\circ$, $4 \times 85.36(2)^\circ$, $2 \times 94.67(2)^\circ$, and $4 \times 94.64(2)^\circ$. Further analysis shows that the greatest change between the two structures above and below T_N is in the Cr thermal factor, which is shown in Fig. 2(b) along with the change in lattice parameters as a function of temperature. The large increase in the Cr thermal parameter is likely to be the result of a distribution of local static displacements, which are being averaged by the diffraction experiment, as a result of the variation of magnetic exchange each of the Cr^{3+} ions experiences through the incommensurate nature of the magnetism.

C. Magnetic structure

With the knowledge of the correct space group, powder neutron diffraction was used to further evaluate the structure and magnetism at low temperature. Although previous works all agree on a simple helical magnetic structure, one of the key unresolved issues is the magnitude of the moment, which has been reported to be 1.7 [33,39] and $1.9 \mu_B$ [32], all of which are well below that expected for Cr^{3+} . Rietveld refinement of the low-temperature magnetic structure was performed using a magnetic cone in real space [40]. The agreement factors R_{Bragg} for nuclear and R_{mag} for magnetic structures were 1.9 and 9.6%, respectively. The refined value of the ordered magnetic moment on chromium was found to be $3.04(3)$ with a propagation vector of $k = [000.4648(2)]$. The final fits to the neutron diffraction data are shown in Fig. 3(a), along with a graphical representation of the helical magnetic model in Fig. 3(b). The magnitude of the ordered moment is that expected for an $S = 3/2$ system, contrary to previous reports that have reported reduced moment [32,33,39].

D. Local magnetism from muon spin relaxation

Data from muon spin-relaxation measurements, performed at both the ISIS facility, United Kingdom, and Paul Scherrer

Institute, Switzerland, were fitted with simple phenomenological models at temperatures below and above T_N . In the magnetically ordered phase below T_N for zero-field (ZF) measurements, we applied a combination of two functions: an exponentially damped oscillation that describes the initial dip in the muon asymmetry with a nonzero slope of initial relaxation coupled with a slow exponentially relaxing background, which overall takes on the form written in Eq. (1) below:

$$G_{\text{below}}(t) = A_1 \exp(-\lambda_1 t) \cos(2\pi \nu t + \varphi) + A_2 \exp(-\lambda_2 t). \quad (1)$$

A similar approach was previously employed in the frustrated magnet $\alpha\text{-Zr}_{1-x}\text{Fe}_x$ [41] and the significant nonzero slope of the relaxation at early times makes $\exp(-\lambda t) \cos(2\pi \nu t + \varphi)$ more suitable than the $\exp(-\sigma^2 t^2) J_0(\omega t)$ factor with 0 derivative at $t \rightarrow 0$ used for the double-spiral system CeAl_2 [42].

In the temperature regime above T_N , we used a single stretched exponential decay in the form of Eq. (2) below [27,42–45]:

$$G_{\text{above}}(t) = A_0 \exp[(-t\lambda_3)^k]. \quad (2)$$

In this case, the exponent k is preferred instead of the usual term β to avoid confusion with the critical exponent.

The justification for the selected phenomenological models is presented below. There are a limited number of examples in the literature of μSR on true incommensurate helical magnetic structures, except for the double-helix model in CeAl_2 [42] and some numerical simulations for $\text{Cr}_{1-x}\text{Mn}_x$ [46]. Much more effort has been placed on incommensurate spin-density-wave (SDW) materials [47], where the initial relaxation was successfully modeled using a spherical Bessel J_0 function or the itinerant MnSi [48,49], which shows a Kubo-Toyabe type of relaxation. For numerous reasons, our ZnCr_2Se_4 data could not be fitted below the T_N to any of these previously reported models. First, the use of Bessel J_0 functions implies a zero relaxation at the $t \rightarrow 0$ limit, while our data show nonzero values (see Fig. 4), although the presence of such

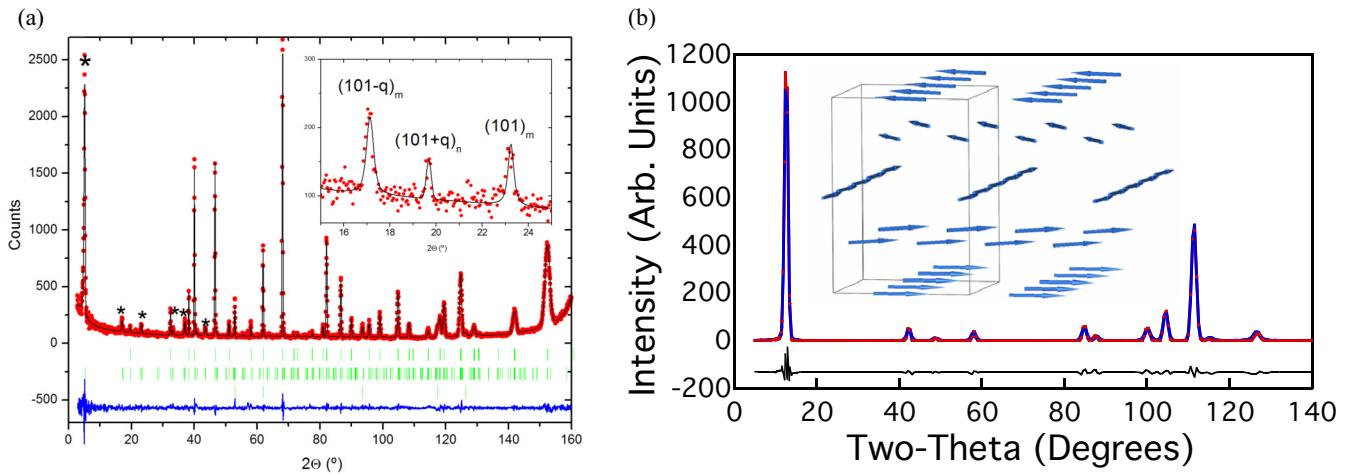


FIG. 3. (a) Observed, calculated, and difference plot of the powder neutron diffraction pattern of ZnCr_2Se_4 at 5 K at $\lambda = 2.078 \text{ \AA}$ using the BT1 diffractometer. The main magnetic reflections are marked with an asterisk. Inset highlights the low angle region, indexed in the tetragonal cell. (b) Observed, calculated, and difference plot of the powder neutron diffraction pattern of ZnCr_2Se_4 at 1.5 K using -0.2 to 0.2 -meV elastic contribution of the DCS at $\lambda = 5.0 \text{ \AA}$. Inset to (b) shows a graphical representation of the Cr moments within the spinel structure showing two-dimensional planes of ferromagnetic order that propagate in a helical fashion down the c axis.

fast relaxation might prohibit the measurement of a zero-slope region. Second, there is a sharp dip in the asymmetry [see 5-K data in Fig. 4(a)] at an early time period, which is impossible to fit using a Bessel function alone. Third, we lack the presence of a substantial nonrelaxing tail below T_N .

The theoretical models for local field distribution in the case of a spin-density wave can be approximated by the Overhauser distribution, which is a continuous distribution of fields between zero and a singularity at B_{max} [50,51]. In a helical incommensurate magnetic structure, a low-field cutoff is additionally present [42,46], so the local field is never equal to zero. The dominant value of B_{max} plays a role of the coherent local field, and generates the oscillatory-like feature, “the dip,” in the depolarization rate, which is modeled using a cosine function in the first component in Eq. (1). The maximum value of the precession frequency is below

40 MHz, which is equivalent to the internal field (B_{max}) of about 0.3 T. Despite being incommensurate, the presence of ferromagnetic Cr planes creates a local magnetic field that is mostly two-dimensional in character. The inset in Fig. 5 shows that the onset of the oscillation frequency at T_N caused by the creation of long-ranged helical magnetic order is associated with the appearance of a fast depolarizing component, which damps the oscillations. This fast relaxation coupled to the oscillatory part is taken into account in the first component of Eq. (1) using the exponential factor $\exp(-\lambda_1 t)$. The relaxation at the long times is modeled using a second exponent, $\exp(-\lambda_2 t)$.

In order to explain the lack of the nonrelaxing tail below T_N in the ZF data, let us note that in both the case of $\alpha\text{-Zr}_{1-x}\text{Fe}_x$ [41] and CeAl_2 [42], the importance of additional spatial field averaging is highlighted, which is caused by the fast muon

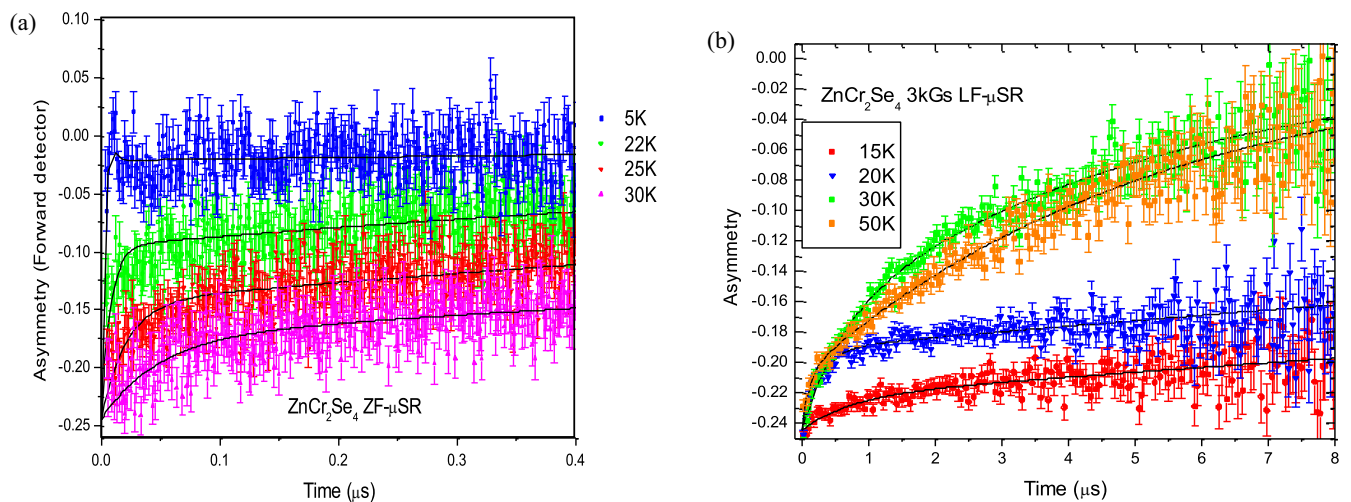


FIG. 4. (a) Early relaxation times for ZF- μ SR of ZnCr_2Se_4 showing very fast relaxation and small initial dip, and (b) example data of 0.3-T LF- μ SR.

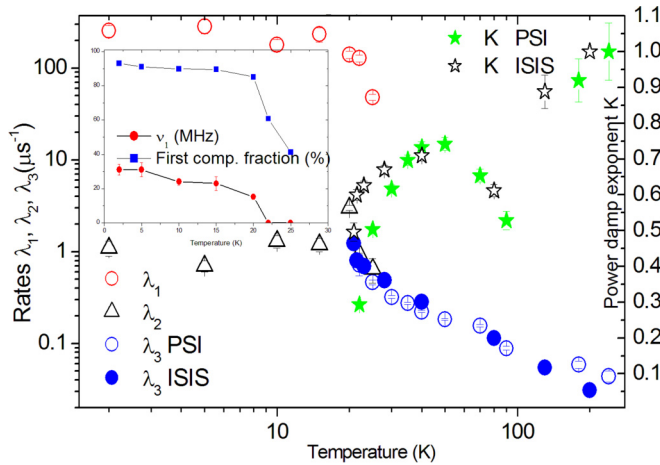


FIG. 5. Zero-field measurement results of relaxation rates of the two components below T_N and the value of the relaxation rate and power factor of the stretched exponent above T_N . Inset shows the temperature dependence of the fast relaxing component and the oscillation frequency.

hopping between symmetrically equivalent sites resulting in a distribution of muon frequencies. The application of a 0.3-T longitudinal field (LF) with the strength on the order of B_{\max} [see Figs. 4(b) and 6] removes the initial dip and significantly reduces, but not fully removes, the depolarization from 150 to $20 \mu\text{s}^{-1}$ below T_N , which is consistent with the static ordered moment being aligned in the direction of the field.

The stretched exponent form of relaxation [Eq. (2)] used above the T_N was found to model the data better than the sum of two exponents in Eq. (1) with $\nu = 0$ Hz. The difference between a stretched exponent and a double-exponent model has been previously evaluated in the case of manganates, where a simple procedure was suggested to differentiate between the two [52]. In such an evaluation, a comparison is made

between the resulting curves obtained by plotting $-t/\ln(G(t))$ against $\ln t$, which, in both cases, are linear with a gradient of $1 - k$. The measurements for ZnCr_2Se_4 at 0.3-T LF and 25 K indicate a power type of decay with exponent equal to 0.6, which was also the best fit obtained for the zero-field data above the point where λ_2 reaches its maximum. Therefore, the high-temperature regime was subsequently fitted with a stretched exponential with variable exponent, instead of two exponentially damped components given in Eq. (1). The results from the final fits are shown in Figs. 5 and 6.

One of the most significant findings of the ZF data above the magnetic transition is the presence of a dramatic change in the relaxation rate around 100 K, which is shown in Fig. 5. The relaxation rate λ_3 increases monotonically when approaching the T_N from higher temperature, which is expected behavior due to the critical slowing down of spin fluctuations, whereas the exponent k shows a domelike behavior with temperature, possessing a value of ~ 0.5 at both T_N and 100 K, while showing a maximum at a midtemperature point at ~ 50 K.

Since the stretched exponential form of relaxation can be explained in terms of multiple exponential curves, but with variable rates [44], it is reasonable to assume that ZnCr_2Se_4 starts to develop areas of correlated ferromagnetic spins at temperatures above 100 K. The development of a long-ranged order at its Curie-Weiss value of 110 K [17] is hindered by the bond frustration due to multiple competing interactions present in the lattice [19,20], leaving uncorrelated two-dimensional ferromagnetic planes. The signature of this is found in the temperature range 50-100 K, where the value of the stretched exponent approaches 1, which indicates a unification of relaxation rates. As the application of a 0.3-T LF is not sufficient to fully align the Cr moments, the muons are still depolarized above T_N , but below 100 K, and the power factor k still retains slight anomalous temperature dependence, as shown in Fig. 6.

E. Inelastic neutron scattering

At low temperatures of 2 K, shown in Fig. 7(a), in the magnetically ordered phase, a gapless mode originating from the magnetic ordering wave vector of $[0\ 0\ 0.4684(4)]$ is seen with a higher-energy mode extending up to at least 0.8 meV. At higher temperatures of 30 K, shown in Fig. 7(b), the low-temperature structured spectrum is replaced by momentum and energy broadened fluctuations, while Fig. 7(c) illustrates a high-temperature scan showing the background and the clear absence of any magnetic fluctuations.

Given the lack of an orbital degree of freedom in Cr^{3+} , we consider the origin of the spiral order as resulting from competing symmetric exchange terms in the Hamiltonian, and compare this against scattering and thermodynamic data. Such a description based on excitations and diffraction has been successfully applied to other insulating magnets to understand the origin of the incommensurate ordering wave vector [53,54]. Previous theoretical studies of spinels [19] have considered this symmetric exchange-only approach and noted a number of exchange constants (at least five in the case of Ref. [19]) need to be considered given the local bonding in the ideal (undistorted) spinel framework.

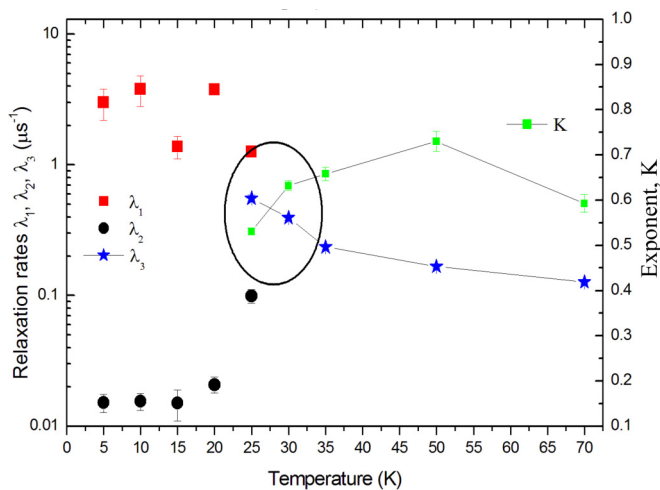


FIG. 6. Fit results for the 0.3-T (3-kG) longitudinal field measurement. At all temperatures two fractions are present with amplitudes λ_1 (fast) and λ_2 (slow). Above T_N (from the region marked by the circle) they can be replaced by one stretched exponential form of relaxation.

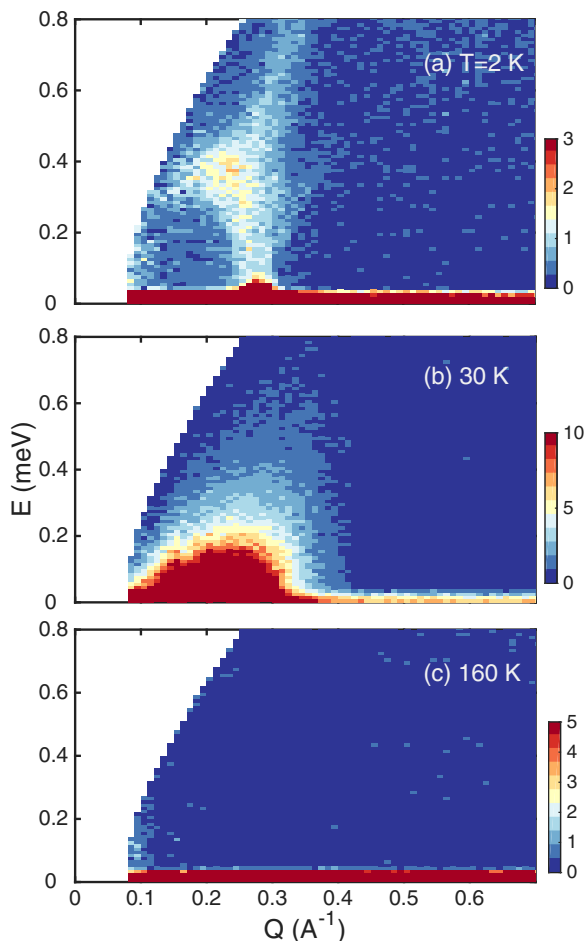


FIG. 7. Temperature dependence of the powder-averaged magnetic fluctuations taken with $E_i = 1.7$ meV on DCS. Panel (a) illustrates the spectrum in the magnetic ordered state at 2 K, with (b) showing that these are replaced by momentum and energy-broadened excitations above T_N . (c) High-temperature scan illustrating the background.

To simplify this description and to provide a tractable means of parametrizing our powder data, we consider a heuristic model based upon coupled chains. We therefore write a Hamiltonian with a ferromagnetic nearest-neighbor exchange term (J_{nn}) for coupling within a given Cr^{3+} chain, as well as the coupling to nearest-neighbor chains, which originates from the 90° Cr-Se-Cr bond. For simplicity, we approximate both the exchange constants within and between chains (inter- and intrachain) to be equal given the bonding close to 90° . We then consider a second exchange coupling Cr^{3+} spins in one chain to the next-nearest-neighbor chain, which we denote here as J_{nnn} . Given that this interaction is more complex, we consider this exchange term to be antiferromagnetic to stabilize spiral magnetic order. This approach is broadly consistent with the analysis of Ref. [19], where nearest-neighbor exchange parameters were ferromagnetic, whereas further exchange was antiferromagnetic to stabilize the observed magnetic structure. This simplified heuristic model provides us a means of parametrizing the data in terms of only two exchange constants describing coupling between Cr^{3+} chains rather than a more complex series, which is difficult to uniquely determine for

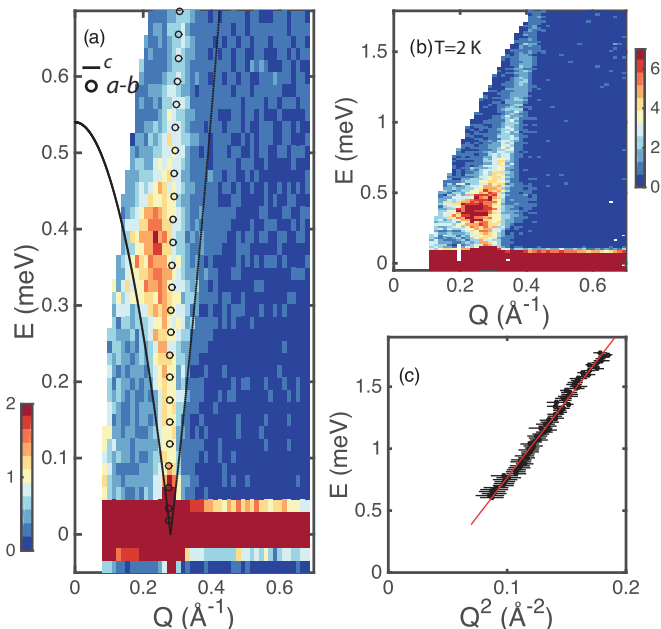


FIG. 8. (a) $E_i = 1.33$ meV data from DCS with the solid line and open points being the predicted single-crystal dispersion along c and within the a - b plane. (b) Lower-resolution $E_i = 3.3$ meV data illustrating a high-energy branch extending up to 1.5 meV. (c) Plot of the peak position as a function of Q^2 illustrating a fit to a ferromagnetic dispersion curve ($\propto Q^2$).

powder data. As noted in Ref. [19] (and in particular Table II of that work), there are multiple possibilities of the five expected exchange constants that could result in the required spiral structure. The Hamiltonian in this simplified heuristic model takes the form

$$H = -J_{nn} \sum_j \vec{S}_j \cdot \vec{S}_{j+ab} - \sum_i (J_{nn} \vec{S}_i \cdot \vec{S}_{i+1} + J_{nnn} \vec{S}_i \cdot \vec{S}_{i+2}). \quad (3)$$

We first consider the ordering wave vector of $[0 \ 0 \ 0.4684(4)]$. The incommensurate ordering wave vector in our symmetric-only simplified magnetic Hamiltonian considering connectivity along c (in terms of L) can be obtained by minimizing the classical exchange energy derived from the above Hamiltonian,

$$E(L) = J_{nn} \cos(2\pi L/4) + J_{nnn} \cos(2\pi L/2). \quad (4)$$

By minimizing this classic energy, we find that $E(L)$ is minimized at $L = 0.468$ when $J_{nnn}/J_{nn} = -0.337$. The ordering wave vector only gives the ratio of the nearest- and next-nearest-neighbor exchange constants and not the absolute values. To obtain an estimate of the exchange constants we now investigate the spin excitations in a powder of ZnCr_2Se_4 .

The powder-averaged spin excitations are illustrated in Figs. 7 and 8. At low temperatures the data show a low-energy branch extending up to 0.5 meV, and a higher-energy branch to around 0.8 meV. The momentum range spanned by the experiment covers the first Brillouin zone of ZnCr_2Se_4 , and therefore we compare the data against single-crystal

dispersion relations. Within the symmetric-only model above, the spin-wave dispersion is obtained using the following equation [55]:

$$(\hbar\omega)^2 = \langle S \rangle^2 (J_{Q_0} - J_q) \left(J_{Q_0} - \frac{1}{2} (J_{Q_0+q} - J_{Q_0-q}) \right), \quad (5)$$

where $J_q = \sum_{\vec{r}_n} J_n e^{-i\vec{q}\cdot\vec{r}}$ and Q_0 is the magnetic ordering wave vector. Since the powder-averaged data presented in Fig. 7 are within the first Brillouin zone, we can compare the dispersion curve directly. The low-temperature powdered average magnetic spectral weight is presented in Fig. 8. The high resolution ($E_i = 1.33$ meV setting) is shown in Fig. 8(a) and the solid line is a plot of the c -axis dispersion with J_{nn} set to 1.0 meV. The open circles represent the dispersion along the tetragonal a direction. The solid lines provide a good description of the lower bound of the magnetic intensity with the open circles tracking the momentum dependence of the high-energy branch which extends beyond the dynamic range of the spectrometer. A lower resolution ($E_i = 3.3$ meV setting) is shown in Fig. 8(b), illustrating a high-energy branch that extends up at least 1.5 meV. To obtain an estimate of the nearest-neighbor exchange constant, we plot the peak position of this branch in energy as a function of Q^2 in Fig. 8(c). From the fit to a dispersion relation which is $\propto q^2$, we obtain an estimate of $J_{nn} = 1.0$ meV. The fact that the curve does not intercept at $Q = 0$ implies ordering a finite wave vector despite the dominant nearest-neighbor ferromagnetic exchange constant, consistent with neutron diffraction data. Our simplified 2- J model based on coupled chains provides a reasonable description of both the spin-wave dispersion (albeit in a powder sample), and also the magnetic ordering wave vector.

Based on the values for the exchange constants used to parametrize the spin excitations and the incommensurate wave vector, we can obtain an estimate for the Curie-Weiss

temperature (θ_{CW}) using the formula

$$k_B\theta_{CW} = \frac{1}{3}s(s+1) \sum_n J_n \quad (6)$$

$$= \frac{1}{3}s(s+1)(6 \times 1.0 \text{ meV} + 4 \times -0.337 \text{ meV}) \quad (7)$$

$$\sim 67 \text{ K}, \quad (8)$$

which is to be compared with an experimental value of 110 K [17]. The discrepancy may originate from the presence of a structural transition at T_N which may alter the exchange constants. More accurate description of the magnetic Hamiltonian would be provided by a more complex exchange framework than the $J_{nn} - J_{nnn}$ model considered here. However, this simple model in terms of two competing exchange constants does provide a means of understanding the spin fluctuations, the ordering wave vector, and an estimate of the Curie-Weiss constant.

These elastic and inelastic neutron scattering results discussed above show that the nature of the magnetic phase transition, which was previously considered to be first order [1,40], is more complex than previously established. Figure 9(a) shows the temperature dependence of $M(T)/M_0$ obtained from Rietveld fits of the neutron powder diffraction, and provides a value for the critical exponent β of 0.14(1), which is similar to the values reported for $\text{Li}_2\text{Mn}_2\text{O}_4$ [28] and $\text{Pb}_2\text{Sr}_2\text{TbCu}_3\text{O}_8$ [56]. Both of these systems undergo similar transitions from two-dimensional short-range order to long-ranged three-dimensional order, but where the structural anisotropy from the layered systems creates Warren-type diffuse magnetic scattering [57]. ZnCr_2Se_4 has high cubic symmetry, so the formation of short-ranged planes is not preferred within one particular plane, and so does not lead to substantial diffuse Warren scattering. As the magnetic transition does not evolve from a paramagnetic state, the magnetization increases faster on cooling than in a classical three-dimensional Heisenberg-type magnet ($\beta \sim 0.36$).

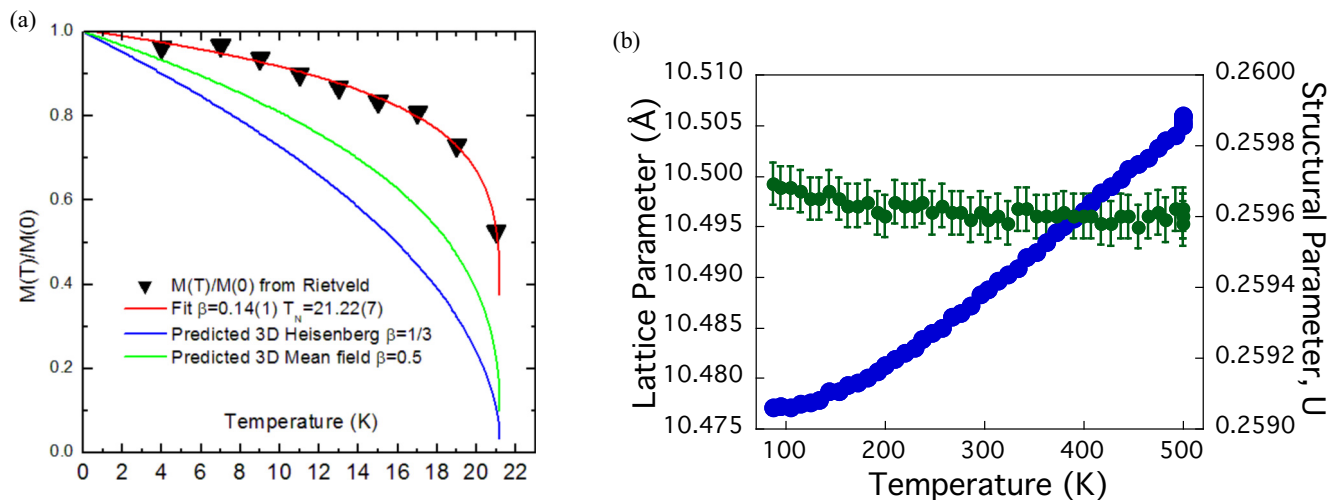


FIG. 9. (a) Fit with numerous models to the evolution of magnetic moment in ZnCr_2Se_4 as a function of temperature, showing an unusual exponent of $\beta = 0.14(1)$, where the development of magnetic ordering occurs much quicker than expected within the Heisenberg model as a result of the short-ranged ordering occurring below 100 K. (b) Evolution of the lattice and structural parameters in ZnCr_2Se_4 as a function of temperature.

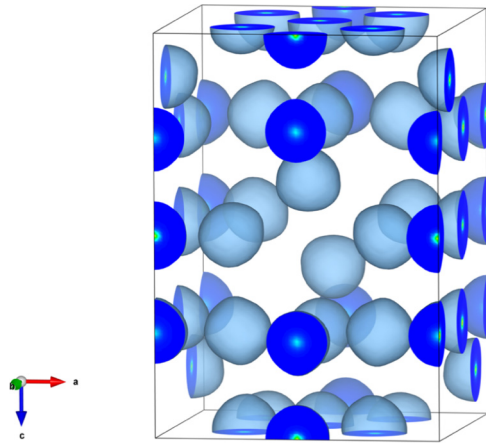


FIG. 10. Reconstruction of the electron density of ZnCr_2Se_4 from ID31 synchrotron powder x-ray diffraction data at 4 K using the maximum entropy method. No assumptions of symmetry are made during this analysis and the isotropic densities of all of the atoms present are indicative of little significant site disorder. Rietveld refinement of the neutron and synchrotron data shows the materials were stoichiometric as made.

The lack of ferromagnetic ordering at Curie-Weiss temperature of 110 K in ZnCr_2Se_4 is a result of the bond frustration and competing interaction. To evaluate the structural changes as a function of temperature, synchrotron x-ray diffraction experiments were performed between 80 and 500 K; the derived lattice parameters and structural parameter u are shown in Fig. 9(b). It can be seen that there is a gradual reduction of the rate of lattice contraction, such that below 100 K there is almost zero expansion. Indeed, previous studies have shown that ZnCr_2Se_4 enters a regime of negative thermal expansion between 75 and 21 K [16,23]. The structural parameter u , which controls the bond angles within the ZnSe_4 tetrahedral unit and Cr_4Se_4 cube, shows little change, except for a slight possible increase. The consequence of this is overall there is an increase in the Se-Cr-Se bond angle and subsequent weakening of the nearest-neighbor ferromagnetic exchange and an increase to the components promoting antiferromagnetic exchange. Within this temperature regime there is a corresponding increase in the short-range two-dimensional ferromagnetic order, which is overcome by the antiferromagnetic interactions to form a long-ranged helical magnetic structure.

The manifestation of the spin-lattice coupling is very different in the case of ZnCr_2Se_4 compared with ZnCr_2O_4 . The latter shows very strong geometric frustration with the large and negative Curie-Weiss constant, implying strong antiferromagnetic exchange that is frustrated on a tetrahedral lattice. ZnCr_2O_4 shows a lowering of symmetry to $I4m2$ at 12.5 K where long-ranged antiferromagnetic order is established, but the commensurate magnetic structure is associated with specific and identifiable distortions to the Cr_4O_4 cubes [26]. In the case of ZnCr_2Se_4 , the bond frustration from competing interaction produces an incommensurate helical magnetic structure. The complex interactions between the magnetic ions will then be a function of the propagation vector along c . Thus each a - b plane of Cr^{3+} ions will experience a different combination of magnetic exchanges, which prohibits a specific and well-defined rearrangement of the Cr_4Se_4 cubes, and local static spatial distribution of Cr ions is stabilized. Further details of the distribution of Cr ions around its (0 0 0) position will require local structure techniques, as diffraction averages these positions and only a threefold increase of the thermal parameters is measured. However, maximum entropy reconstruction of the electron density map from synchrotron x-ray diffraction at 4 K, shown in Fig. 10, does not indicate any significant static disorder.

IV. CONCLUSIONS

We have synthesized a stoichiometric sample of the bond-frustrated spinel ZnCr_2Se_4 and shown that despite the presence of such frustration, the Cr^{3+} ions order with a full moment. The magnetic ordering is accompanied by a distortion from the ideal cubic spinel to a tetragonal $I4_1/amd$ lattice. Previous reports of reduced ordered moments and more complex symmetry lowering are the result of nonstoichiometry that is prevalent in these systems. The frustration does, however, result in considerable local spin correlations at temperature around five times the long-ranged ordering temperature, as measured by μSR spectroscopy.

ACKNOWLEDGMENTS

Funding for this project was provided by the Royal Society, EPSRC, and the Carnegie Trust for the Universities of Scotland. We acknowledge the support of the NIST, ESRF, ISIS, and PSI for providing access to national research facilities used in this work.

- [1] F. K. Lotgering, in *Proceeding of the International Conference on Magnetism* (Institute of Physics and the Physical Society, Nottingham, London, 1964), p. 533.
- [2] T. Watanabe, *J. Phys. Soc. Jpn.* **37**, 140 (1974).
- [3] W. C. Koehler, J. W. Cable, M. K. Wilkinson, and E. O. Wollan, *Phys. Rev.* **151**, 414 (1966).
- [4] Y. Ishikawa, K. Tajima, D. Bloch, and M. Roth, *Solid State Commun.* **19**, 525 (1976).
- [5] D. R. Parker, M. A. Green, S. T. Bramwell, A. S. Wills, J. S. Gardner, and D. A. Neumann, *J. Am. Chem. Soc.* **126**, 2710 (2004).
- [6] D. I. Khomskii, *J. Magn. Magn. Mater.* **306**, 1 (2006).
- [7] Y. Tokura and S. Seki, *Adv. Mater.* **22**, 1554 (2010).
- [8] E. E. Rodriguez, D. A. Sokolov, C. Stock, M. A. Green, O. Sobolev, J. A. Rodriguez-Rivera, H. Cao, and A. Daoud-Aladine, *Phys. Rev. B* **88**, 165110 (2013).
- [9] E. E. Rodriguez, C. Stock, P.-Y. Hsieh, N. P. Butch, J. Paglione, and M. A. Green, *Chem. Sci.* **2**, 1782 (2011).
- [10] E. E. Rodriguez, C. Stock, K. L. Krycka, C. F. Majkrzak, P. Zajdel, K. Kirshenbaum, N. P. Butch, S. R. Saha, J. Paglione, and M. A. Green, *Phys. Rev. B* **83**, 134438 (2011).

- [11] E. E. Rodriguez, C. Stock, P. Zajdel, K. L. Krycka, C. F. Majkrzak, P. Zavalij, and M. A. Green, *Phys. Rev. B* **84**, 064403 (2011).
- [12] C. Stock, E. E. Rodriguez, and M. A. Green, *Phys. Rev. B* **85**, 094507 (2012).
- [13] C. Stock, E. E. Rodriguez, M. A. Green, P. Zavalij, and J. A. Rodriguez-Rivera, *Phys. Rev. B* **84**, 045124 (2011).
- [14] C. Stock, E. E. Rodriguez, O. Sobolev, J. A. Rodriguez-Rivera, R. A. Ewings, J. W. Taylor, A. D. Christianson, and M. A. Green, *Phys. Rev. B* **90**, 121113 (2014).
- [15] T. Rudolf, Ch. Kant, F. Mayr, J. Hemberger, V. Tsurkan, and A. Loidl, *Phys. Rev. B* **75**, 052410 (2007).
- [16] J. Hemberger, H.-A. Krug von Nidda, V. Tsurkan, and A. Loidl, *Phys. Rev. Lett.* **98**, 147203 (2007).
- [17] P. K. Baltzer, H. W. Lehmann, and M. Robbins, *Phys. Rev. Lett.* **15**, 493 (1965).
- [18] R. Plumier, *J. Phys.* **27**, 213 (1966).
- [19] K. Dwight and N. Menyuk, *Phys. Rev.* **163**, 435 (1967).
- [20] K. Siratori, *J. Phys. Soc. Jpn.* **30**, 709 (1971).
- [21] R. J. Plumier, M. Lecomte, A. Miedan-Gros, and M. Sougi, *Phys. Lett. A* **55**, 239 (1975).
- [22] A. S. Cameron, Y. V. Tymoshenko, P. Y. Portnichenko, J. Gavilano, V. Tsurkan, V. Felea, A. Loidl, S. Zherlitsyn, J. Wosnitzer, and D. S. Inosov, *J. Phys.: Condens. Matter* **28**, 146001 (2016).
- [23] R. Kleinberger and R. de Kouchkovsky, *C. R. Acad. Sci. Paris.* **262**, 628 (1966).
- [24] S.-H. Lee, C. Broholm, W. Ratcliff, G. Gasparovic, Q. Huang, T. H. Kim, and S.-W. Cheong, *Nature* **418**, 856 (2002).
- [25] S.-H. Lee, C. Broholm, T. H. Kim, W. Ratcliff, and S.-W. Cheong, *Phys. Rev. Lett.* **84**, 3718 (2000).
- [26] S. Ji, S.-H. Lee, C. Broholm, T. Y. Koo, W. Ratcliff, S.-W. Cheong, and P. Zschack, *Phys. Rev. Lett.* **103**, 037201 (2009).
- [27] S. L. Lee, S. H. Kilcoyne, and R. Cywinski, *Muon Science: Muons in Physics, Chemistry and Materials* (IOP, London, 1999).
- [28] C. R. Wiebe, P. L. Russo, A. T. Savici, Y. J. Uemura, G. J. MacDougall, G. M. Luke, S. Kuchta, and J. E. Greedan, *J. Phys.: Condens. Matter* **17**, 6469 (2005).
- [29] S.-H. Lee, H. Takagi, D. Louca, M. Matsuda, S. Ji, H. Ueda, Y. Ueda, T. Katsufuji, J.-H. Chung, S. Park, S.-W. Cheong, and C. Broholm, *J. Phys. Soc. Jpn.* **79**, 011004 (2010).
- [30] S. H. Lee, G. Gasparovic, C. Broholm, M. Matsuda, J. H. Chung, Y. J. Kim, H. Ueda, G. Xu, P. Zschack, K. Kakurai, H. Takagi, W. Ratcliff, T. H. Kim, and S. W. Cheong, *J. Phys.: Condens. Matter* **19**, 145259 (2007).
- [31] M. Hidaka, N. Tokiwa, M. Fujii, S. Watanabe, and J. Akimitsu, *Phys. Status Solidi B* **236**, 9 (2003).
- [32] F. Yokaichiya, A. Krimmel, V. Tsurkan, I. Margiolaki, P. Thompson, H. N. Bordallo, A. Buchsteiner, N. Stüßler, D. N. Argyriou, and A. Loidl, *Phys. Rev. B* **79**, 064423 (2009).
- [33] J. Akimitsu, K. Siratori, G. Shirane, M. Iizumi, and T. Watanabe, *J. Phys. Soc. Jpn.* **44**, 172 (1978).
- [34] G. L. Molnar, *Handbook of Prompt Gamma Activation Analysis with Neutron Beams* (Kluwer Academic, Dordrecht, The Netherlands, 2004).
- [35] R. L. Paul and R. M. Lindstrom, *J. Radioanal. Nucl. Chem.* **243**, 181 (2000).
- [36] J. Rodriguez-Carvajal, *Physica B* **192**, 55 (1993).
- [37] M. Hidaka, M. Yoshimura, S. Takahashi, S. Watanabe, and J. Akimitsu, *Phys. Status Solidi B* **236**, 209 (2003).
- [38] M. Hidaka, M. Yoshimura, N. Tokiwa, J. Akimitsu, Y. J. Park, J. H. Park, S. D. Ji, and K. B. Lee, *Phys. Status Solidi B* **236**, 570 (2003).
- [39] R. J. Plumier, *J. Appl. Phys.* **37**, 964 (1966).
- [40] J. M. Hastings and L. M. Corliss, *Phys. Rev.* **126**, 556 (1962).
- [41] D. H. Ryan, J. van Lierop, and J. M. Cadogan, *J. Phys.: Condens. Matter* **16**, S4619 (2004).
- [42] A. Schenck, D. Andreica, F. N. Gygax, and H. R. Ott, *Phys. Rev. B* **65**, 024444 (2001).
- [43] Y. J. Uemura, T. Yamazaki, D. R. Harshman, M. Senba, and E. J. Ansaldo, *Phys. Rev. B* **31**, 546 (1985).
- [44] I. A. Campbell, A. Amato, F. N. Gygax, D. Herlach, A. Schenck, R. Cywinski, and S. H. Kilcoyne, *Phys. Rev. Lett.* **72**, 1291 (1994).
- [45] A. Lappas, K. Prassides, F. Gygax, and A. Schenck, *J. Solid State Chem.* **145**, 587 (1999).
- [46] P. Peretto, R. Venegas, and G. N. Rao, *Phys. Rev. B* **23**, 6544 (1981).
- [47] A. Lappas, K. Prassides, F. N. Gygax, and A. Schenck, *J. Phys.: Condens. Matter* **12**, 3401 (2000).
- [48] I. M. Gat-Malureanu, A. Fukaya, M. I. Larkin, A. J. Millis, P. L. Russo, A. T. Savici, Y. J. Uemura, P. P. Kyriakou, G. M. Luke, C. R. Wiebe, Y. V. Sushko, R. H. Heffner, D. E. MacLaughlin, D. Andreica, and G. M. Kalvius, *Phys. Rev. Lett.* **90**, 157201 (2003).
- [49] Y. J. Uemura *et al.*, *Nat. Phys.* **3**, 29 (2007).
- [50] A. Amato, R. Feyerherm, F. N. Gygax, A. Schenck, H. Vonlohneysen, and H. G. Schlager, *Phys. Rev. B* **52**, 54 (1995).
- [51] A. T. Savici, Y. Fudamoto, I. M. Gat, T. Ito, M. I. Larkin, Y. J. Uemura, G. M. Luke, K. M. Kojima, Y. S. Lee, M. A. Kastner, R. J. Birgeneau, and K. Yamada, *Phys. Rev. B* **66**, 014524 (2002).
- [52] R. H. Heffner, D. E. MacLaughlin, G. J. Nieuwenhuys, and J. E. Sonier, *J. Phys.: Condens. Matter* **16**, S4541 (2004).
- [53] C. Stock, L. C. Chapon, A. Schneidewind, Y. Su, P. G. Radaelli, D. F. McMorrow, A. Bombardi, N. Lee, and S.-W. Cheong, *Phys. Rev. B* **83**, 104426 (2011).
- [54] C. Stock, S. Jonas, C. Broholm, S. Nakatsuji, Y. Nambu, K. Onuma, Y. Maeno, and J. H. Chung, *Phys. Rev. Lett.* **105**, 037402 (2010).
- [55] J. Jensen and A. R. Mackintosh, *Rare Earth Magnetism, Structures and Excitations* (Clarendon, Oxford, 1991).
- [56] S. Y. Wu, W.-H. Li, K. C. Lee, J. W. Lynn, T. H. Meen, and H. D. Yang, *Phys. Rev. B* **54**, 10019 (1996).
- [57] B. E. Warren, *Phys. Rev.* **59**, 693 (1941).