

Anomalous spin-orbit torque switching in synthetic antiferromagnetsChong Bi,^{1,*} Hamid Almasi,¹ Kyle Price,¹ Ty Newhouse-Illige,¹ Meng Xu,¹ Shane R. Allen,²
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We report that synthetic antiferromagnets (SAFs) can be efficiently switched by spin-orbit torques (SOTs) and the switching scheme does not obey the usual SOT switching rule. We show that both the positive and negative spin Hall angle (SHA)-like switching can be observed in Pt/SAF structures with only positive SHA, depending on the strength of applied in-plane fields. A switching mechanism directly arising from the asymmetric domain expansion is proposed to explain the anomalous switching behaviors. Contrary to the macrospin-based switching model where the SOT switching direction is determined by the sign of SHA, this switching mechanism suggests that the SOT switching direction is dominated by the field-modulated domain wall motion and can be reversed even with the same sign of SHA. The presence of this switching mechanism is further confirmed by the domain wall motion measurements. The anomalous switching behaviors provide important insights for understanding SOT switching mechanisms and also offer novel features for applications.

DOI: [10.1103/PhysRevB.95.104434](https://doi.org/10.1103/PhysRevB.95.104434)**I. INTRODUCTION**

Electrical manipulation of magnetization is a crucial step for encoding data in spintronic memory and logic devices. It is usually achieved through the spin transfer torque (STT) effect [1] generated by a spin-polarized current in a spin-valve or magnetic tunnel junction (MTJ) structure. In recent years, spin-orbit torques (SOTs), a new type of spin torque driven by in-plane currents flowing in heavy metals (HMs) [2–6], topological insulators [7–9], or antiferromagnets [10,11], have emerged as a more efficient way to manipulate magnetization. SOTs have been successfully employed to switch magnetization [2–4,8,10,12–18], drive domain wall (DW) motion [19,20], and excite spin-torque nano-oscillators [21,22]. In many applications such as magnetic random access memory (MRAM), SOTs have advantages over STTs due to their higher efficiency and the ability to switch a MTJ without passing a large current through the tunnel barrier. However, as a fundamental question, the underlying SOT switching mechanism is still under debate. Moreover, the contributions of various interfacial effects, such as the Rashba effects, spin Hall effects (SHEs), and Dzyaloshinskii-Moriya interaction (DMI) to the SOT switching also remain elusive. One widely accepted SOT switching mechanism is based on the macrospin model [4], in which the SOT nucleates initial domains through the macrospin model and switches the entire ferromagnet by subsequent domain expansion [4,12,23–25]. In this model, SHEs dominate the SOT switching, which obeys the rule shown in Figs. 1(a) and 1(b). The final magnetization switches to the direction determined by $\mathbf{H} \times \boldsymbol{\sigma}$. Here, \mathbf{H} is the effective in-plane field that is provided by an external magnetic field [2–4,8,12–15,18], exchange bias, or interlayer coupling [10,17], and $\boldsymbol{\sigma}$ is the spin polarization injected from adjacent materials. The spin Hall angle (SHA) of adjacent materials determines the direction of $\boldsymbol{\sigma}$ and thus the switching

direction for a given \mathbf{H} . Usually the adjacent materials can be classified into two basic types with a positive SHA, such as Pt [4], (Bi, Sb)Te [8], and PtMn [10], and a negative SHA, such as Ta [3] and W [26]. Therefore $\boldsymbol{\sigma}$ and the switching directions in Pt/ferromagnet (Pt/FM) and Ta/FM structures are opposite [2–4,8,10,12–17]. So far all reported SOT switchings are the SOT switchings of a single ferromagnet [2–4,12,14,18,23–25,27], in which reversing either current or in-plane field is necessary to switch magnetization [2–4,12,14,18,23–25,27] and the switching rule shown in Figs. 1(a) and 1(b) is well obeyed.

In high-density MRAMs, synthetic antiferromagnets (SAFs) are widely adopted in MTJs as the reference layer [28–30] and even as the free layer [31,32] to improve the thermal stability of MTJs and reduce the coupling field between the reference and free layers. A SAF with strong interlayer coupling can generally be regarded as a FM layer with the effective magnetization of $M_{\text{eff}} = M_A + M_B$, where M_A and M_B are the magnetization of two coupled FMs (see Supplemental Material [33]). According to the macrospin model, one will expect the SOT switching of HM/SAF to be similar to that of HM/FM, since the SHA and corresponding SOTs are exactly the same in the two systems. Up to date, the SOT switching of SAFs and its efficiency have not been investigated.

Here we demonstrate that the SAFs can also be efficiently switched like a single ferromagnet. Surprisingly, the switching scheme of SAFs does not obey the usual switching rule shown in Figs. 1(a) and 1(b). It is shown that the SOT switching direction of SAFs can be reversed depending on the strength of applied in-plane fields, even with the same sign of SHA. These results indicate that the switching of SAFs can be achieved without any direction changes of the applied in-plane field and current, contrary to the switching of a single ferromagnet [2–4,12,14,18,23–25,27] in which the direction of either current or in-plane field has to be reversed. The observed anomalous magnetization switching (AMS) behaviors invalidate the conventional macrospin model and prompt a new understanding of SOT switching. To explain

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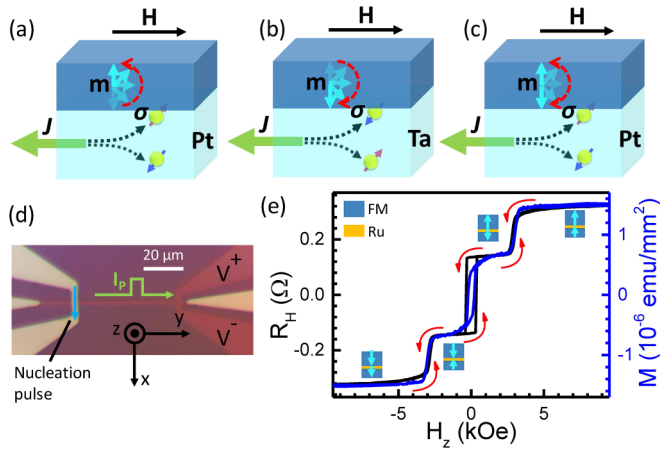


FIG. 1. The SOT switching with a (a) positive or (b) negative SHA. \mathbf{J} is the injected current density, and \mathbf{m} is the unit magnetization vector of adjacent magnets. \mathbf{m} switches to up or down for the positive or negative SHA, respectively. (c) SOT-induced anomalous switching observed in this work, in which \mathbf{m} can be switched to both up and down states under the SOT with the same sign. (d) Top view of a Hall bar structure showing the configurations of electrical measurements and coordinate system. (e) Magnetic properties of the Pt/SAF structure characterized by AHE (black) and VSM (blue). The red arrows show the switching sequence of magnetization. The insets show the magnetization configuration at each field stage.

the AMS behaviors, we then propose a SOT switching mechanism directly arising from the asymmetric domain expansion/contraction due to the field-modulated chiral DW motion. This switching mechanism suggests that the SOT switching direction is only determined by the in-plane field-modulated relative velocity between $\uparrow\downarrow$ and $\downarrow\uparrow$ domains (V_{RD}) [19,20], regardless of the initially nucleated domains through the macrospin model, and thus does not directly depend on the sign of SHA. The current-driven DW motion measurements further confirm this switching mechanism and demonstrate that the AMS arising from the unique chiral DW motion is due to the special magnetization configuration of SAFs. The unique DW motion in SAFs cause the switching directions to deviate from the macrospin model, clearly clarifying the SOT switching mechanism. These results highlight the DMI effects that determine the chiral DW motion during the SOT switching and also provide a guideline for optimizing SOT switching in applications. Furthermore, this switching behavior combined with tunable interlayer coupling [34–37] could also enable many new SOT-related applications.

II. EXPERIMENTAL DETAILS

Sample fabrication. The samples employed in this work have the structure of Si-wafer/SiO₂ (300 nm)/Pt (4 nm)/BML/Ru t_{Ru} /TML/Ru (0.6 nm)/SiO₂ (10 nm), where the BML (bottom magnetic layer) is Co (0.6 nm), TML (top magnetic layer) is Co (0.4 nm)/Pt (1 nm)/Co (0.4 nm)/Pt (1 nm)/Co (0.4 nm), and t_{Ru} is the thickness of the Ru spacer layer in the range of 0–1.5 nm. Here we adopted a symmetric TML to minimize the SOTs from the inside Pt layers. Control samples with a thicker BML were also fabricated. All the

stack structures were deposited on Si/SiO₂ (300 nm) substrates by magnetron sputtering. The deposition rates for each layer are Pt 0.05 nm/s, Co 0.018 nm/s, Ru 0.01 nm/s, and SiO₂ 0.074 nm/s. The base vacuum was better than 1.5×10^{-8} Torr before sputtering. The samples were then patterned into Hall bar structures with a feature width of 2.5 μm , as shown in Fig. 1(d). To monitor DW motion, an orthogonal DW nucleation line with a width of 3 μm was directly deposited on the top of each Hall bar structure. The distance between the nucleation line and the voltage bars is 50 μm . The magnetic properties of fabricated continuous films were measured by vibrating sample magnetometry (VSM) measurements, and the anomalous Hall effect (AHE) measurements were performed in patterned Hall bar structures. The ferromagnetic (FM) or antiferromagnetic (AFM) interlayer coupling for each sample was determined by the combination of AHE and VSM measurements.

Electrical measurements. The current pulses and a 0.3-mA dc current for sensing anomalous Hall resistance (R_H) were applied by the same Keithley 6220 current source. The Hall voltage was monitored by a Keithley 2000 multimeter. For all SOT switching-related measurements (by sweeping current pulses or external fields), a 1-ms current pulse was applied first. After waiting 3 s, the applied external field was then removed and a 0.3-mA dc sense current was applied to detect magnetization states after each current pulse. For DW motion measurements, the magnetization of the Hall bar was first initialized to a uniform up or down state by a positive or negative 6-kOe perpendicular field, respectively. After that, a 1-ms nucleation current pulse was applied by an independent Keithley 2400 sourcemeter to create domain nucleation. The amplitude of the nucleation pulse was 120 mA. The nucleation current was negative for up initial states and positive for down initial states. To increase the probability of domain nucleation, an assisted perpendicular field, a bit smaller (typically 50 Oe) than reversal switching fields, was applied during the nucleation current pulse. After domain nucleation, a current pulse was then applied by the Keithley 6220 current source to drive DW motion. The R_H was measured after the current pulse to determine if the DW had arrived at the voltage bars. The length of the applied current pulse varied between 1 ms and 10 s. We chose the proper injected current densities to make sure that the time for DW motion between the nucleation line and the voltage bars was within 1 ms to 10 s. The detailed measurement process is given in Supplemental Material [33]. Hereafter, we mainly present the experimental results from the sample with $t_{Ru} = 0.66$ nm that shows strong AFM coupling.

III. RESULTS

The schematic process of the SOT-induced anomalous switching in this work is illustrated in Fig. 1(c). In contrast to the conventional switching in Pt/FM [Fig. 1(a)] and Ta/FM [Fig. 1(b)], the switching sign in a Pt/SAF structure can reverse even at the same external field direction. Figure 1(e) presents the AHE and VSM results, both of which show three clear perpendicular switching loops, indicating an AFM interlayer coupling as well as a strong perpendicular magnetic anisotropy (PMA) in the sample. The switching that occurred around ± 3 kOe can be explained in that the applied perpendicular

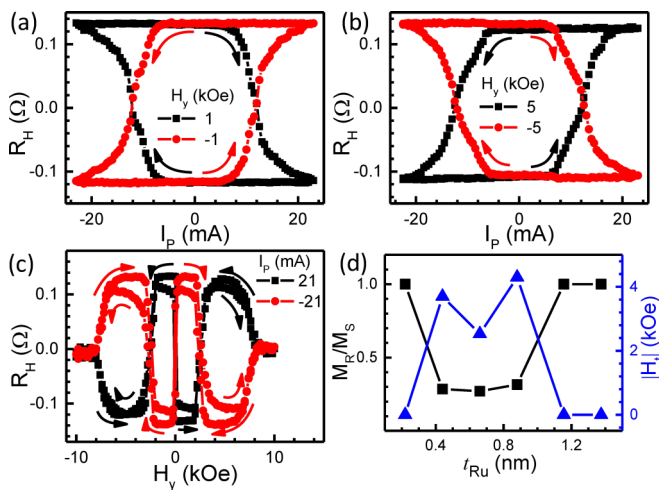


FIG. 2. The current-driven magnetization switching under (a) ± 1 kOe and (b) ± 5 kOe in-plane magnetic fields. (c) The magnetization switching induced by ± 21 -mA current pulses as a function of in-plane field. The red and black arrows indicate the corresponding switching sequence of red and black curves, respectively. (d) M_R/M_S ratio (black) and $|H_t|$ (blue) as a function of t_{Ru} . $|H_t| = 0$ indicates no AMS observed.

field is larger/smaller than the effective field of AFM coupling, which induces the switching of BML. The switching at around 0 Oe arises from the switching of TML, which simultaneously induces the switching of BML again because of the strong AFM coupling. The configurations of magnetization in the TML and BML at each field stage are illustrated in insets of Fig. 1(e). The current-induced magnetization switching is shown in Fig. 2. The measurement setup is similar to that of previously reported SOT-induced switching [2,14]. Under an in-plane external field (H_y), we first applied a 1-ms current pulse (I_p) with gradually varying amplitudes and then applied a 0.3-mA DC current after removal of H_y to detect the magnetization state after each I_p . As shown in Fig. 2(a), the magnetization can be completely switched between two states and the switching orientation depends on the direction of the external fields. For example, when $H_y = +1$ kOe, the switching loop is clockwise, which becomes anticlockwise when $H_y = -1$ kOe. The critical current density for showing switching behaviors is about 4.5×10^7 A/cm², which is comparable with that for switching a single ferromagnet [4,14], even though the total thickness of ferromagnets in the SAF is about 3 times larger than that of the single ferromagnet. These results indicate that the perpendicular SAFs can also be efficiently switched like a single ferromagnet by SOTs.

The surprising switching behavior occurs under a larger in-plane external field, as shown in Fig. 2(b). Now the switching loop changes the sign to be anticlockwise for +5 kOe and clockwise for -5 kOe, which is like the SOT switching with an opposite SHA sign. Figures 2(a) and 2(b) indicate that the magnetization can also be switched without any direction change of field or current. This unusual switching behavior has never been observed before and is quite different from all previous experimental results [2–4,8,10,12–17] and the macrospin-based models [3,4,38–42]. As illustrated in Figs. 1(a) and 1(b) and widely verified in previous studies, the

SOT switching orientation is only determined by the direction of applied in-plane field and the sign of SHA. Because only Pt with a positive SHA is involved in the sample, the switching orientation should stay the same if the direction of the applied magnetic field is not changed. To get a full switching phase of the sample after current-induced switching as a function of in-plane field, as shown in Fig. 2(c). In this measurement, the switching current pulse was maintained at ± 21 mA while sweeping the in-plane field. It can be clearly seen that R_H changes its sign at the transition fields (H_t) of ± 2.5 kOe, in addition to the sign change around zero magnetic field that is expected from the conventional SOT switching. This magnetic field dependence of SOT switching further confirms the AMS in AFM coupled samples. Figure 2(d) shows the t_{Ru} -dependent SOT switching as well as the ratio between remanent magnetization (M_R) and saturation magnetization (M_S), in which $M_R/M_S \approx 0.3$ indicates AFM coupling and $H_t = 0$ Oe indicates no AMS observed. One can see that only the AFM coupled samples show AMS. Moreover, when t_{Ru} approaches the values for FM coupling, the magnetization can only be partially switched at high-field regions. As shown in Figs. 3(a) and 3(b), when $t_{Ru} = 0.44$ nm and 0.88 nm, the magnetization after positive and negative switching currents only shows a slight change at high-field regions. It should be noted that the partial switching is not due to the insufficient applied current. As shown in Figs. 3(a) and 3(b), even for larger applied currents of $I_p = \pm 25$ mA, the switching loops stay the same as those of $I_p = \pm 21$ mA.

To understand the mechanism of this unusual switching behavior, the SOT switching was also studied in a control sample of Pt (4 nm)/Co (0.8 nm)/Ru (0.66 nm)/Co (0.2 nm)/Pt (1 nm)/Co (0.4 nm)/Ru (0.6 nm)/SiO₂ (10 nm) with a thicker BML. Even with the same SOT and t_{Ru} , the AMS is missing in the control sample as shown in Fig. 3(c) and the switching behaviors are consistent with previous reports [2,4]. All of these results indicate that two conditions must be met to exhibit AMS: (1) AFM coupling and (2) the BML is thinner than the TML. The observed AMS cannot be explained by the macrospin model, in which the SOT switching of a strongly coupled SAF is the same as that of a single FM with an effective magnetization M_{eff} (see Supplemental Material [33]). In addition, the partial switching in the high-field regions as shown in Figs. 3(a) and 3(b) also sheds light on the violation of the macrospin model, because the partial switching should gradually evolve to a full switching by increasing the applied current.

IV. DISCUSSION

Recently, it has been experimentally observed that an in-plane field can result in asymmetric domain expansion in HM/FM structures [12,23–25,43–45]. Here we propose a switching model to explain the AMS directly arising from the asymmetric domain expansion. The simple picture of this model (one-dimensional) is given in Figs. 3(d)–3(f). First, without external fields, the applied current induces a demagnetized state and drives all DWs to move with the same velocity (V_D). The demagnetized state is induced by the combination of all spin torques and thermal effects, which has been proven to destabilize a uniform magnetization [46–49]

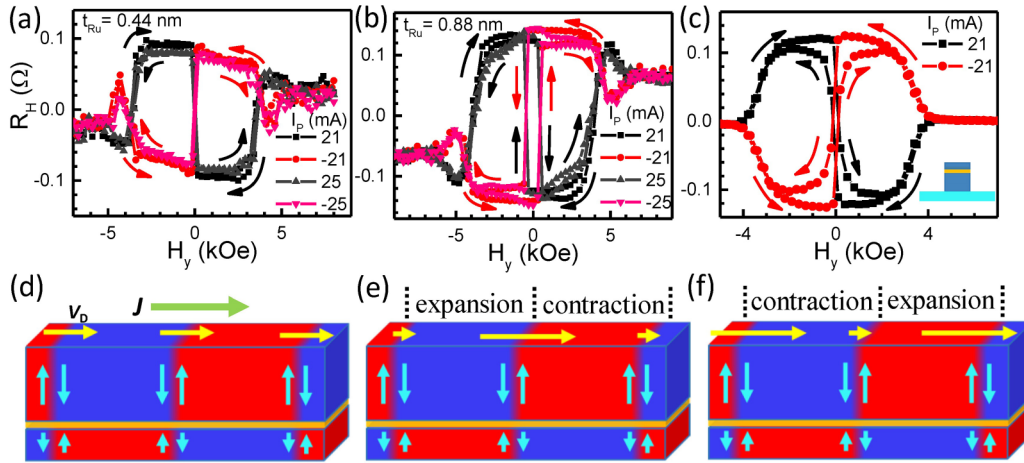


FIG. 3. t_{Ru} -dependent current-induced magnetization switching with (a) $t_{Ru} = 0.44$ nm and (b) 0.88 nm. (c) The current-induced magnetization switching in a control sample with a thicker BML. The arrows indicate the switching sequence. Inset of (c) shows the relative thickness of BML and TML. (d–f) Illustrations of SOT switching based on DW motion. (d) All DWs move with the same velocity, and domains keep the same shape during the current-driven DW motion. (e, f) The different velocities of $\uparrow \downarrow_{BML}$ and $\downarrow \uparrow_{BML}$ DWs induce the expansion or contraction of opposite domains during the current-driven DW motion. Yellow arrows represent DW velocities.

and finally leads to a demagnetized state with equal spin-up and spin-down domains. The applied in-plane field may also assist with the formation of the demagnetized state but mainly modulates the DW motion, inducing the asymmetric expansion/contraction of a domain as shown below. In this case, the domain shape and area stay the same during DW motion and no favored magnetized direction is formed [Fig. 3(d)]. Second, if an applied field can separately modulate the velocities of $\uparrow \downarrow_{BML}$ and $\downarrow \uparrow_{BML}$ DWs by increasing the V_D of one type of DW and decreasing that of another type of DW, the domains will expand or contract during the current-driven domain motion, as shown in Figs. 3(e) and 3(f). Now the magnetization will favor either spin-up or spin-down states, depending on the relative velocity between the two DWs, V_{RD} . Third, to reach a full magnetization switching, V_{RD} has to be large enough to collapse those contracted domains and any possible nucleated reversal domains within the expanded domains. This is because a reversal domain will nucleate again within the expanded domain to keep the demagnetized state if the expanded domain is larger than a critical value. Above the critical value, a domain can still be thought as a uniform magnetization with higher magnetic energy that is unstable under the large current. As demonstrated before, the separate control of V_D for two types of DWs can be realized in HM/FM bilayers and V_{RD} depends on the strength of applied in-plane fields [19,20]; therefore, a large enough field is necessary to realize a full SOT switching [2–4,27] in these structures.

According to this model, the magnetization switching orientation is only determined by the sign of V_{RD} , regardless of the sign of the SHA and the initial domain nucleation directions. This is because, no matter whether SHA is positive or negative, SOTs will lead to the same demagnetized state (first condition) and the second and third conditions are only determined by V_{RD} . This is a distinct difference between this model and previous models [3,4,38–42]. Although previous models also suggested that the switching process can be incoherent, the sign of the SHA still determines the nucleation direction of the first domain according to the macrospin

switching model and thus decides the final switching orientations [12,23–25], because the subsequent switching process is based on the expansion of the initially nucleated domains. One can see that the role of in-plane fields in this model are also very different from previous models where the external field was used to stabilize magnetization or break symmetry [2–4,16].

To verify this model, we measured V_D in our sample. Figures 4(a) and 4(b) present V_D as a function of applied current when $H_y = \pm 1$ kOe and ± 4.5 kOe, which shows that

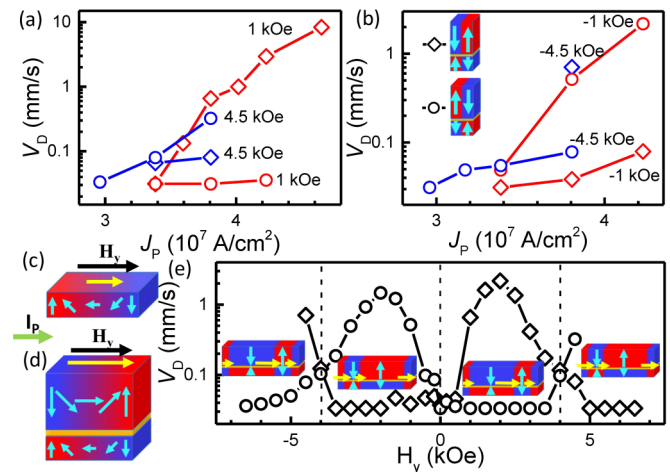


FIG. 4. The current-driven DW motion as a function of current density under (a) positive and (b) negative in-plane fields. Red and blue represent ± 1 kOe and ± 4.5 kOe external fields, respectively. (c, d) The schematic of in-plane field-modulated DW motion for (c) a single magnetic layer and (d) AFM coupled bilayer. (e) V_D as a function of in-plane field driving by the pulses with a current density of 3.97×10^7 A/cm². The inserted illustrations illustrate the domain expansion/contraction at each in-plane field region, consistent with the four SOT switching regions shown in Fig. 2(c). For all figures, diamonds and circles represent $\uparrow \downarrow_{BML}$ and $\downarrow \uparrow_{BML}$ DWs, respectively. For clarification, the error bars are omitted (see Supplemental Material [33] for the determination of error).

V_D increases with applied current density for both types of DWs. When $H_y = +1$ kOe, V_{RD} between $\uparrow \downarrow_{\text{BML}}$ and $\downarrow \uparrow_{\text{BML}}$ DWs is positive, and thus the spin-up state of BML is favored for $+I_P$ according to the illustrations of Fig. 3(e). This is completely consistent with the switching orientation shown in Figs. 2(a) and 2(c). On the contrary, when $H_y = +4.5$ kOe, V_{RD} becomes negative, and consequently, the spin-down state of BML is favored for $+I_P$, also consistent with the switching in Fig. 2(c). When either H_y or I_P changes sign, the switching orientations in Fig. 2(c) can also be understood through the sign change of V_{RD} . Figure 4(e) gives the measured V_D for two types of DWs at different in-plane fields. The four regions with positive or negative V_{RD} are consistent with the four switching regions of Fig. 2(c), confirming our explanations. It should be noted that V_{RD} changes sign at around ± 4 kOe, which is larger than H_t shown in Fig. 2(c). This is because the DW velocity was measured at $I_p = \pm 9$ mA and the resultant thermal effect is much smaller than that of $I_p = \pm 21$ mA used for SOT switching, resulting in a larger H_t (see Supplemental Material [33]).

As shown in Fig. 4(e), the modulation effects of H_y on V_D are quite different from those in HM/FM structures [19,20]. In HM/FM structures, V_{RD} changes sign only once at around $H_y = 0$ Oe [19,20]. Therefore, according to our model, the switching orientation only reverses once around 0 field, consistent with the experimental results [2–4]. In our samples, V_{RD} changes sign three times, corresponding to the four contrasting switching regions as shown in Fig. 2(c). Furthermore, in HM/FM structures, the switching orientation due to V_{RD} is the same as that predicted by the macrospin model [4], while in our samples, V_{RD} changes sign even for the same field direction and the switching orientation contradicts the macrospin model, clearly clarifying the AMS as arising from field-modulated V_{RD} . The partial switching shown in Figs. 3(a) and 3(b) may be due to a small V_{RD} , which is not large enough to collapse all nucleated reversal domains within the expanded domains.

The unique field modulation effects in our structures can be attributed to the strong AFM coupling. It is shown that the field modulation of V_D is determined by the parallel/antiparallel configuration between H_y and the internal magnetization of DWs [19,20]. As shown in Fig. 4(c), in a single FM (also the same for two FM coupled magnetic layers, or two AFM coupled layers but with a thicker BML, see Supplemental Material), the transition between the parallel and antiparallel configurations only occurs once when $H_y \approx -H_{DMI}$, where H_{DMI} is the DMI effective field [19,20]. However, for a SAF with a thicker TML [Fig. 4(d)], M_{eff} is determined by the thicker TML layer. Correspondingly, the transition between the parallel and antiparallel configurations occurs twice when $H_y \approx -(H_{DMI} + H_{\text{exc}})$ (see Supplemental Material [33]), resulting in the unique H_y -modulated V_{RD} . Here, H_{exc} is the effective AFM coupling field in BML, and we ignore the SOTs and H_{DMI} in TML. In our structures, the evaluated $H_{DMI} = -0.8$ kOe and $|H_{\text{exc}}| = 4.3$ kOe for $\uparrow \downarrow_{\text{BML}}$ DWs, and $H_{DMI} = 1.0$ kOe and $|H_{\text{exc}}| = 4.5$ kOe for $\downarrow \uparrow_{\text{BML}}$ DWs (see Supplemental Material [33]).

The understanding of AMS based on the domain nucleation and DW motion indicates that the DW energy dominates the magnetization dynamics in such structures. In multilayers with PMA, magnetic domains or skyrmions can be formed

with very high density (indicating high DW energy) during SOT-induced magnetization dynamics because of the narrow size of DWs [50], and thus the involvement of DW energy is required to improve previous switching models [3,4,38–42] to explain AMS. As demonstrated here, the DW energy may dominate the magnetization dynamics compared with other contributions (such as the sign of SOT). The recently reported memristive behaviors in antiferromagnet-ferromagnet bilayers [10] are probably also due to a small V_{RD} like the partial switching in Figs. 3(a) and 3(b). Because of the narrow DWs (typically several nanometers [50]), the reversal mechanism demonstrated here will still dominate the switching process for the feature size of tens of nanometers in future SOT-MRAMs.

V. SUMMARY

In summary, we have demonstrated the SOT switching of SAFs, which shows an anomalous switching behavior compared with a single ferromagnet. These results offer new possibilities to explore SOT-related magnetization dynamics in magnetically coupled multilayers and also clarify the SOT switching mechanism. Although plenty of physical phenomena such as the Rashba effects, SHEs, and DMI effects have been observed in the HM/FM interfaces after the discovery of SOT switching, how these effects directly contribute to the SOT switching is not clear. In the macrospin model, only the dampinglike torques from SHEs was included. The demonstrated switching model based on the chiral DW motion clearly indicates that all those interfacial effects contribute to the SOT switching indirectly by driving and modulating DW motion. The fieldlike torques that can also originate from the Rashba effects and dramatically modulate the DW motion [51] may also determine the SOT switching. This switching model also highlights the essential role of DMI effects, which are the origin of DW chirality [19,20], in the SOT switching.

In application, the demonstrated SAF switching will benefit the high-density SOT-MRAMs by addressing the emerging challenges in nanosized MTJs with high thermal stability and efficient switching. In addition, it has been shown theoretically and experimentally that the interlayer coupling can be changed between AFM and FM states by voltage [35–37]. Together with the sign control of SOT switching demonstrated here and a very large DW velocity realized as the magnetization of BML and TML approach each other [52], a voltage-tunable high-speed, low-energy manipulation of magnetization could possibly be realized in the HM/SAF structures.

Note added. Recently, we noticed an interesting work by Shi *et al.* reporting the switching of similar synthetic antiferromagnets by the spin-orbit torques generated by the Ta interlayer [53]. Also, a DW motion model [54] developed by Risinggård and Linder can qualitatively reproduce our DW-motion-related experimental results.

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