

Monte Carlo analysis for finite-temperature magnetism of Nd₂Fe₁₄B permanent magnetYuta Toga,¹ Munehisa Matsumoto,¹ Seiji Miyashita,^{2,1} Hisazumi Akai,^{3,1} Shotaro Doi,^{3,1}
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We investigate the effects of magnetic inhomogeneities and thermal fluctuations on the magnetic properties of a rare-earth intermetallic compound, Nd₂Fe₁₄B. The constrained Monte Carlo method is applied to a Nd₂Fe₁₄B bulk system to realize the experimentally observed spin reorientation and magnetic anisotropy constants K_m^A ($m = 1, 2, 4$) at finite temperatures. Subsequently, it is found that the temperature dependence of K_1^A deviates from the Callen-Callen law, $K_1^A(T) \propto M(T)^3$, even above room temperature, $T_R \sim 300$ K, when the Fe (Nd) anisotropy terms are removed to leave only the Nd (Fe) anisotropy terms. This is because the exchange couplings between Nd moments and Fe spins are much smaller than those between Fe spins. It is also found that the exponent n in the external magnetic field H_{ext} response of barrier height $\mathcal{F}_B = \mathcal{F}_B^0(1 - H_{\text{ext}}/H_0)^n$ is less than 2 in the low-temperature region below T_R , whereas n approaches 2 when $T > T_R$, indicating the presence of Stoner-Wohlfarth-type magnetization rotation. This reflects the fact that the magnetic anisotropy is mainly governed by the K_1^A term in the $T > T_R$ region.

DOI: [10.1103/PhysRevB.94.174433](https://doi.org/10.1103/PhysRevB.94.174433)**I. INTRODUCTION**

Rare-earth permanent magnets, particularly Nd-Fe-B, that exhibit strong magnetic performance [1] are attracting considerable attention because of the rapidly growing interest in electric vehicles. The main focus of research involving these materials is to increase the coercive field H_c and improve the temperature dependence [2–6]. Therefore, a number of studies have conducted micromagnetic simulations [7–10] for the magnetization processes using inhomogeneous magnetic parameters to describe the complex structures in sintered magnets. Many of the results predict that the distinctive feature of magnetic anisotropy near the grain boundaries of Nd-Fe-B particles is responsible for the degradation of H_c .

Thus, one of the remaining subjects of theoretical study is to provide quantitative aspects from a microscopic viewpoint or in atomic scale to the m th-order magnetic anisotropy constants K_m^A and their temperature dependence near the grain surfaces or grain boundaries. For K_1^A at the surface of Nd-Fe-B particles, Moriya *et al.* [11] and Tanaka *et al.* [12] calculated the crystal-field parameter A_2^0 using a first-principles technique, and they pointed out that K_1^A (mainly proportional to A_2^0) is negative at the (001) surface when the (001) Nd layer is exposed to a vacuum. However, few theoretical studies have examined the temperature dependence of K_m^A , even for the bulk system, since the qualitative theory was developed by Callen and Callen [13–15]. Recently, Sasaki *et al.* [16] and Miura *et al.* [17] conducted theoretical studies on a quantitative level of the temperature dependence of K_m^A for a Nd₂Fe₁₄B bulk system based on crystal-field theory, and they successfully reproduced various experimental results. However, as these theories relied on the mean-field approach in terms of the exchange coupling between the Nd $4f$ moments and Fe $3d$ spins, the results cannot be directly applied to K_m^A near the surfaces or interfaces of particles. Moreover, because the

crystal-field analysis employed in these works is based on a quantum-mechanical approach, which is typical for $4f$ electronic systems [18], it is effectively impossible to treat finite systems of nm or μm scale using a similar method.

Therefore, in the present work, in anticipation of future work on magnetization reversal in finite-sized particles, we employed a realistic model with a classical Heisenberg Hamiltonian to calculate the magnetic properties of a Nd₂Fe₁₄B bulk system at finite temperatures. The key features of our model are as follows: (i) an appropriate crystalline electric field Hamiltonian [18] is included in the classical manner, (ii) exchange coupling parameters are obtained by first-principles calculations, (iii) K_m^A is directly evaluated from Monte Carlo (MC) simulations without employing the mean-field analysis, and (iv) the constrained Monte Carlo (C-MC) method [19] is adopted to evaluate the temperature dependence of magnetic anisotropy. Note that we can naturally realize the experimentally observed spin reorientation and K_m^A . Reflecting the (inhomogeneous) variation of magnetic parameters in the unit cell composed of 68 atoms (see Fig. 1), K_1^A does not obey the Callen-Callen law [13, 14], which states that $K_1^A(T) \propto M(T)^3$ when considering only the Nd (Fe) anisotropy terms and neglecting the Fe (Nd) anisotropy terms. We also analyze the response of the external magnetic field H_{ext} [20–25] for a barrier height $\mathcal{F}_B(H_{\text{ext}}) = \mathcal{F}_B^0(1 - H_{\text{ext}}/H_0)^n$, and we find that the H_{ext} response deviates from the Stoner-Wohlfarth-type ($n = 2$), especially below room temperature, $T_R \sim 300$ K.

II. MODEL AND METHOD**A. Model**

By treating each atom as having classical spin, we constructed a three-dimensional Heisenberg model including realistic atom locations for Nd₂Fe₁₄B, as shown in Fig. 1. This

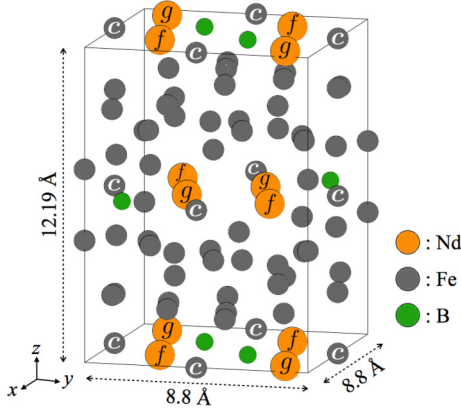


FIG. 1. Unit cell of $\text{Nd}_2\text{Fe}_{14}\text{B}$ including 68 atoms [space group $P4_2/mnm$ (No. 136)] [26]. Only Nd(*f,g*) and Fe(*c*) sites are represented. This figure was plotted using VESTA [27].

model using atomic-scale parameters was defined as follows:

$$\mathcal{H} = -2 \sum_{i<j} S_i J_{ij}^{\text{ex}} S_j \mathbf{e}_i \cdot \mathbf{e}_j - \mu_0 \sum_i m_i \mathbf{e}_i \cdot \mathbf{H}_{\text{ext}} - \sum_{i \in \text{TM}} D_i^A (e_i^z)^2 + \sum_{i \in \text{RE } l=2,4,6} \tilde{\theta}_l^{J_i} A_{l,i}^{m_l} \langle r^l \rangle_i \hat{O}_{l,i}^{m_l}, \quad (1)$$

where $S_i J_{ij}^{\text{ex}} S_j$ is the exchange coupling constant including the spin amplitude between the *i*th and *j*th sites, \mathbf{e}_i is the normalized spin vector at the *i*th site, m_i is the magnetic moment, μ_0 is the magnetic permeability of a vacuum, and \mathbf{H}_{ext} is the external magnetic field. The third and fourth terms include single-ion magnetic anisotropy properties. We consider transition metals (TM) and rare-earth elements (RE) separately. The anisotropy of TM sites is defined using the magnetic anisotropy parameter D_i^A and the *z*-component of \mathbf{e}_i , i.e., e_i^z . The anisotropy of RE sites is based on crystal-field theory [18,28] and uses the Stevens operator $\hat{O}_{l,i}^{m_l}$, the crystal-field parameter $A_{l,i}^{m_l}$, and the Stevens factor $\tilde{\theta}_l^{J_i}$. Here, $\langle r^l \rangle_i$ can be calculated as the spatial average of the 4*f* electron distribution. In the present paper, we consider $m_l = 0$ for simplicity. For reference, note that $\hat{O}_{l,i}^{m_l=0} = \tilde{\theta}_l^{J_i=9/2}$:

$$\begin{aligned} \hat{O}_{2,i}^0 &= 3(J_i^z)^2 - J_i^2, \\ \hat{O}_{4,i}^0 &= 35(J_i^z)^4 - [30J_i^2 - 25](J_i^z)^2 + [3J_i^4 - 6J_i^2], \\ \hat{O}_{6,i}^0 &= 231(J_i^z)^6 - [315J_i^2 - 735](J_i^z)^4 \\ &\quad + [105J_i^4 - 525J_i^2 + 294](J_i^z)^2 \\ &\quad - [5J_i^6 - 40J_i^4 + 60J_i^2], \\ \tilde{\theta}_2^{9/2} &= \frac{-7}{3^2 \times 11^2}, \quad \tilde{\theta}_4^{9/2} = \frac{-2^3 \times 17}{3^3 \times 11^3 \times 13}, \\ \tilde{\theta}_6^{9/2} &= \frac{-5 \times 17 \times 19}{3^3 \times 7 \times 11^3 \times 13^2}, \end{aligned} \quad (2)$$

where $J_i^z = J_i e_i^z$ is the *z*-component of the total angular momentum J_i , which is 9/2 for Nd atoms, and we use J_i^2 instead of $J_i(J_i + 1)$ in the classical manner.

TABLE I. Site occupancies and model parameters of each crystallographically inequivalent atom. The spin magnetic moments, m^s , are calculated from the first-principles calculation code, MACHIKANEYAMA (AKAIKKR) [29]. The anisotropy parameters $D_{i\text{TM}}^A$ and $A_i^{m_l} \langle r^l \rangle$ are taken from previous results [18,30]. We neglected the D^A values of B and Nd, as they are less than 0.1 meV, and we used the $\langle r^l \rangle$ values of Nd, Ref. [31], i.e., $\langle r^2 \rangle = 1.001a_B^2$, $\langle r^4 \rangle = 2.401a_B^4$, and $\langle r^6 \rangle = 12.396a_B^6$, where a_B is the Bohr radius.

Atom	Occ.	m^s (μ_B)	$D_{i\text{TM}}^A$ (meV)	$A_i^{m_l} \langle r^l \rangle$ (K)
B(<i>g</i>)	4	-0.169		
Fe(<i>c</i>)	4	2.531	-2.14	
Fe(<i>e</i>)	4	1.874	-0.03	
Fe(<i>j</i> ₁)	8	2.298	1.07	
Fe(<i>j</i> ₂)	8	2.629	0.58	
Fe(<i>k</i> ₁)	16	2.063	0.55	
Fe(<i>k</i> ₂)	16	2.206	0.38	
			(<i>l, m</i> _l):	(2,0) (4,0) (6,0)
Nd(<i>f</i>)	4	-0.413		295.3 -29.5 -22.8
Nd(<i>g</i>)	4	-0.402		

Table I lists the atomic-scale parameters used in the present study. The 68 atoms in the tetragonal unit cell of $\text{Nd}_2\text{Fe}_{14}\text{B}$ (see Fig. 1) occupy nine crystallographically inequivalent sites, as seen in Table I. These atom locations and lattice constants ($a = b = 8.8$ Å, $c = 12.19$ Å) were set to experimental values [26]. m^s is the spin magnetic moment of valence electrons (excluding 4*f*-electrons). We defined $m_i = m_i^s$ for Fe and B atoms, and $m_i = m_i^s + m_i^{4f}$ for Nd atoms. Here, the magnetic moment of 4*f*-electrons in each Nd atom is $m^{4f} = 8J/11\mu_B \sim 3.273\mu_B$. For the magnetic anisotropy terms, the D^A values were set to previous first-principles calculation results [30] for $\text{Y}_2\text{Fe}_{14}\text{B}$, which has a similar crystal structure to that of $\text{Nd}_2\text{Fe}_{14}\text{B}$. In contrast, we adopted experimental results [18] regarding $A_i^{m_l}$, even though some research for the $A_i^{m_l}$ values of $\text{Nd}_2\text{Fe}_{14}\text{B}$ was performed using first-principles calculations [32,33]. This is because first-principles evaluations of $A_i^{m_l}$ are strongly dependent on the calculation conditions; in particular, the values of the $l = 6$ terms are still open to some debate.

The higher-order crystal-field parameters A_4^0 and A_6^0 of the Nd atoms have a significant effect on the low-temperature properties of $\text{Nd}_2\text{Fe}_{14}\text{B}$. To illustrate these effects, Fig. 2 shows the anisotropy potential for $J = 9/2$ single classical spin:

$$V_l^{m_l}(\theta) = \tilde{\theta}_l^{9/2} A_l^{m_l} \langle r^l \rangle \hat{O}_l^{m_l}(\theta), \quad (3)$$

where θ is the spin angle measured from the *z*-axis (i.e., $e^z = \cos \theta$), and $A_l^{m_l} \langle r^l \rangle$ take the values in Table I. The potential V_2^0 increases monotonically as θ increases, whereas V_4^0 and V_6^0 vary nonmonotonically. Because of this behavior, the total anisotropy potential $V_2^0 + V_4^0 + V_6^0$ attains a minimum at $\theta = 36.7^\circ$ for (a) $J_z = e^z J$. In contrast, for (b) $J_z = 0.8e^z J$, the minimum occurs at $\theta = 0^\circ$. This coefficient (=0.8) of e^z can be regarded as an effect of thermal fluctuations at $T > 0$. The above results indicate that the spin direction is tilted from the *z*-axis at $T = 0$, although this tilting disappears at a certain temperature. This behavior corresponds to the spin reorientation phenomenon. In the case of $\text{Nd}_2\text{Fe}_{14}\text{B}$, the spin

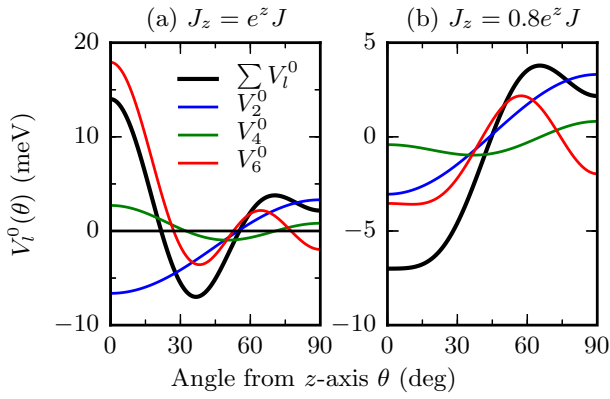


FIG. 2. Anisotropy potentials $V_l^0(\theta)$ (meV) for $J = 9/2$ single spin. Black lines denote total anisotropy potential $V_2^0 + V_4^0 + V_6^0$.

reorientation transition is due to the V_4^0 and V_6^0 values of Nd atoms, and it includes the effects of exchange couplings and the magnetic anisotropy of Fe atoms (for details, see Sec. III A).

Figure 3 shows the exchange coupling constants, $S_i J_{ij}^{\text{ex}} S_j$, which include the spin amplitude as a function of interatomic distance r_{ij} . These constants were calculated with Liechtenstein's formula [34], which has been implemented on the first-principles electronic-structure calculation using the Korringa-Kohn-Rostoker (KKR) Green's-function method, MACHIKANEYAMA (AKAIKKR) [29]. In the calculation, standard muffin-tin-type potentials were assumed, and the local density approximation parametrized by Morruzi, Janak, and Williams [35] was used. Up to d -wave scatterings were taken into account in KKR, and $(8 \times 8 \times 6)$ k -points in the first Brillouin zone were used for the calculation of J_{ij}^{ex} 's. For the Nd $4f$ states, the so called open-core approximation was employed.

From Fig. 3, we can see that the exchange couplings between Fe and Nd have much smaller values than those between Fe atoms. In addition, none of the Nd atoms interacts directly with other Nd atoms. The amplitude relation of the exchange couplings is consistent with experimental results [1] based on a mean-field analysis. Note that all $S_{\text{Fe}} J_{\text{Fe-Nd}}^{\text{ex}} S_{\text{Nd}}$ on $r_{ij} < 4 \text{ \AA}$ have positive values in Fig. 3. As J_{ij}^{ex} is evaluated as the interaction between valence electrons, $S_{\text{Fe(Nd)}}$ can be

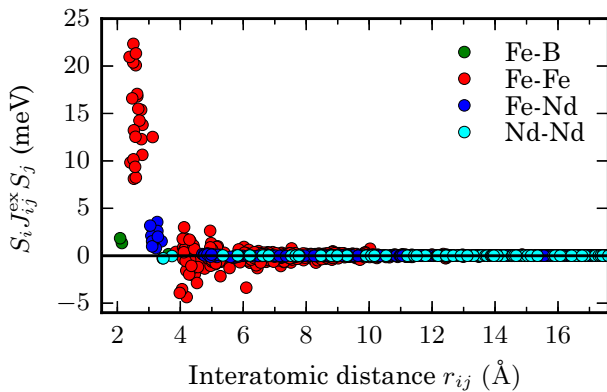


FIG. 3. Exchange coupling constant between each atom as a function of interatomic distance.

regarded as being proportional to $m_{\text{Fe(Nd)}}^z$, i.e., $S_{\text{Nd}} S_{\text{Fe}} < 0$. Hence, the bare exchange couplings $J_{\text{Fe-Nd}}^{\text{ex}}$ have negative values. The couplings between Fe and B, $J_{\text{Fe-B}}^{\text{ex}}$, also take negative values, which can be explained in the same way.

B. Method

To analyze the finite-temperature magnetism of $\text{Nd}_2\text{Fe}_{14}\text{B}$, we applied MC methods based on the METROPOLIS algorithm [36] to the above classical Heisenberg model. Although the magnetic anisotropy is evaluated as the magnetization angle dependence of free energy, this is generally difficult to simulate explicitly using a typical MC approach. Therefore, we also adopted the C-MC method [19] to evaluate the magnetic anisotropy. The C-MC method fixes the direction of total magnetization $\mathbf{M} = (M_x, M_y, M_z) = (1/N_s) \sum_i m_i \mathbf{e}_i$ (N_s is the total number of sites) in any direction for each MC sampling without \mathbf{H}_{ext} , and then calculates the fixed angle θ dependencies of free energy $\Delta\mathcal{F}(\theta)$ and magnetization torque $\vec{T}(\theta)$ as follows [19]:

$$\vec{T}(\theta) = - \left\langle \sum_i \mathbf{e}_i \times \frac{\partial \mathcal{H}}{\partial \mathbf{e}_i} \right\rangle \quad \text{for } \mathbf{M} = \mathbf{M}(\theta), \quad (4)$$

$$\begin{aligned} \Delta\mathcal{F}(\theta) &= \mathcal{F}(\theta) - \mathcal{F}(\theta_0) \\ &= \int_{\theta_0}^{\theta} d\theta' [\mathbf{n}(\theta') \times \vec{T}(\theta')] \cdot \frac{\partial \mathbf{n}(\theta)}{\partial \theta} \Big|_{\theta=\theta'}, \end{aligned} \quad (5)$$

where $\mathbf{n}(\theta) = \mathbf{M}(\theta)/|\mathbf{M}(\theta)|$, and $\mathbf{M}(\theta)$ is the total magnetization in the fixed direction θ .

Note that Asselin *et al.* [19] formulated the C-MC method for systems with homogeneous magnetic moments, i.e., all the magnetic moments have the same value. However, it can easily be extended to systems with inhomogeneous magnetic moments such as $\text{Nd}_2\text{Fe}_{14}\text{B}$. We now briefly explain only the procedure of the extended C-MC method with a fixed \mathbf{M} in the direction of the z -axis:

(a) Select a site i and obtain the new state of i -spin randomly chosen,

$$\mathbf{e}_i \rightarrow \mathbf{e}'_i.$$

(b) Select a site $j (\neq i)$ randomly.

(c) Adjust the new state of the j -spin to preserve \mathbf{M} direction (namely, $M_x = M_y = 0$):

$$\begin{aligned} \mathbf{e}_j &\rightarrow \mathbf{e}'_j, \\ e_j^{x'} &= e_j^x + \frac{m_i}{m_j} (e_i^x - e_i^{x'}), \\ e_j^{y'} &= e_j^y + \frac{m_i}{m_j} (e_i^y - e_i^{y'}), \\ e_j^{z'} &= \text{sign}(e_j^z) \sqrt{1 - (e_j^{x'})^2 - (e_j^{y'})^2}. \end{aligned}$$

If $1 - (e_j^{x'})^2 - (e_j^{y'})^2 < 0$, return to (a).

(d) Calculate the new total magnetization:

$$\mathbf{M}' = \mathbf{M} + \frac{1}{N_s} [m_i (\mathbf{e}'_i - \mathbf{e}_i) + m_j (\mathbf{e}'_j - \mathbf{e}_j)].$$

If $\mathbf{M}' < 0$, return to (a).

(e) Update from the initial spin states ($\mathbf{e}_i, \mathbf{e}_j$) to the new spin states ($\mathbf{e}'_i, \mathbf{e}'_j$) with the probability

$$P = \min \left[1, \left(\frac{M'_z}{M_z} \right)^2 \frac{|e'_j{}^z|}{|e_j{}^z|} \exp(-\beta \Delta E) \right],$$

where β is the inverse temperature and $\Delta E = E(\mathbf{e}'_i, \mathbf{e}'_j) - E(\mathbf{e}_i, \mathbf{e}_j)$ is the energy difference.

(f) Return to (a).

To apply the C-MC method to the $\text{Nd}_2\text{Fe}_{14}\text{B}$ bulk system, we change the procedures (c) and (d) to treat different magnetic moments from those in the original paper [19].

The MC (C-MC) simulations in the present study repeated each calculation for 200 000 (100 000) MC steps, where one MC step is defined as one trial for each spin to be updated. The first 100 000 (30 000) MC steps were used for equilibration, and the following 100 000 (70 000) MC steps were used to measure the physical quantities. We performed simulations for 12 different runs with different initial conditions and random sequences. We then calculated the average results and statistical errors. To check the system-size dependence, we used systems of $N_s = L^3 \times 68$ (unit cell) sites with $L = 3-6$, imposing the periodic boundary conditions.

III. RESULTS AND DISCUSSION

A. Thermodynamic properties

First, we focus on the magnetic transition points to verify the model and parameter values. The results in this subsection are based on typical MC, rather than C-MC.

Figure 4 shows the magnetization curves for each cutoff range r_{cut} . We consider all exchange couplings J_{ij}^{ex} under $r_{ij} \leq r_{\text{cut}}$. Here, $\langle A \rangle$ is defined as the statistical average of A . It can be seen that there are two transition points in Fig. 4.

In the higher-temperature region, $\langle |M_z| \rangle$ approaches 0 at the Curie temperature T_C . The magnetization curves show that T_C is strongly dependent on r_{cut} , even in long-range ($r_{ij} > 3.52 \text{ \AA}$). Thus, T_C was evaluated more accurately using the Binder parameter [36–38] defined as $g_L = 1 - \langle |M| \rangle^4 / 3 \langle |M|^2 \rangle^2$ for system sizes $L = 3-6$. The results are plotted in Fig. 5(a). It is apparent that T_C has quite different values depending on r_{cut} , and the condition of $(8 \times 8 \times 6)$ - k -points

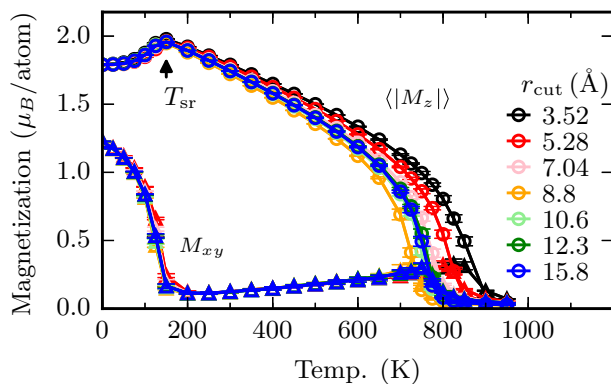


FIG. 4. Magnetizations as a function of temperature for each effective exchange coupling radius r_{cut} . The system size is $L = 6$.

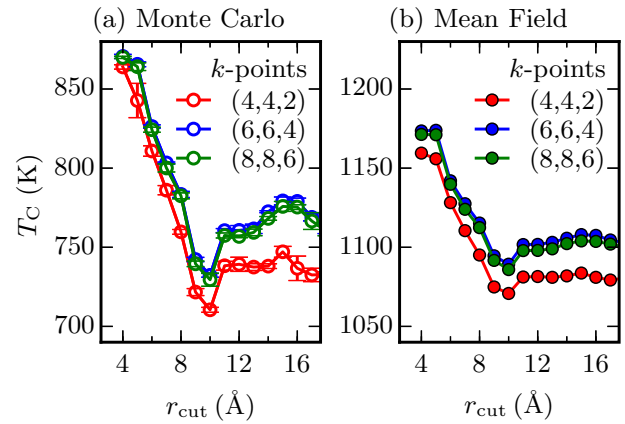


FIG. 5. Curie temperatures with (a) MC and (b) mean field as a function of the effective exchange coupling radius r_{cut} for the number of k -points.

(mean accuracy of $S_i J_{ij}^{\text{ex}} S_j$ in the first-principles calculations) is sufficient for convergence. Similar behavior can be seen in Fig. 5(b), where T_C has been calculated by a 9-sublattice (i.e., 9-inequivalent sites in Table I) mean-field analysis [39,40]. Compared with the MC results, the mean-field results are less sensitive to r_{cut} and tend to overestimate T_C .

To analyze the long-range ($r_{ij} > 3.52 \text{ \AA}$) exchange coupling effect for T_C , Fig. 6 shows the average exchange coupling at the Fe atoms, $\tilde{J}_{\text{Fe}}^{\text{ex}}(r_1, r_2)$, which is defined as follows:

$$\tilde{J}_{\text{Fe}}^{\text{ex}}(r_1, r_2) = \frac{1}{N_{\text{Fe}}} \sum_{i \in \text{Fe}, j} S_i J_{ij}^{\text{ex}} S_j \quad \text{for } r_1 < r_{ij} \leq r_2, \quad (6)$$

where N_{Fe} is the total number of Fe sites. Each bar height in Fig. 6 denotes the sum of $S_i J_{ij}^{\text{ex}} S_j$ per atom in the range of each bar width (here $r_2 - r_1 = 0.2 \text{ \AA}$). Because $\tilde{J}_{\text{Fe}}^{\text{ex}}$ has many exchange bonds that correspond to a spherical surface area ($\propto r_{ij}^2$), it keeps a small but significant value even in the long range. Indeed, the sum of short-range exchange couplings is $\tilde{J}_{\text{Fe}}^{\text{ex}}(0, 3.52 \text{ \AA}) = 154.6 \text{ meV}$ and that over a longer range is $\tilde{J}_{\text{Fe}}^{\text{ex}}(3.52 \text{ \AA}, 17.6 \text{ \AA}) = -10.2 \text{ meV}$. This negative value explains the decreasing trend for T_C shown in Fig. 5. The necessity of long-range exchange coupling has been identified

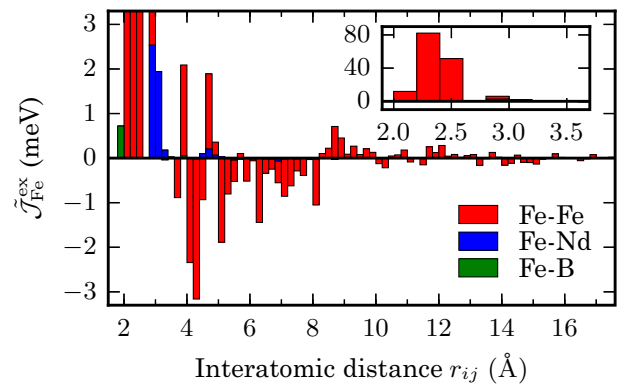


FIG. 6. Average Fe atom exchange coupling $\tilde{J}_{\text{Fe}}^{\text{ex}}$ as a function of r_{ij} . The inset shows a large-area view of $\tilde{J}_{\text{Fe}}^{\text{ex}}$ in $r_{ij} < 3.7 \text{ \AA}$.

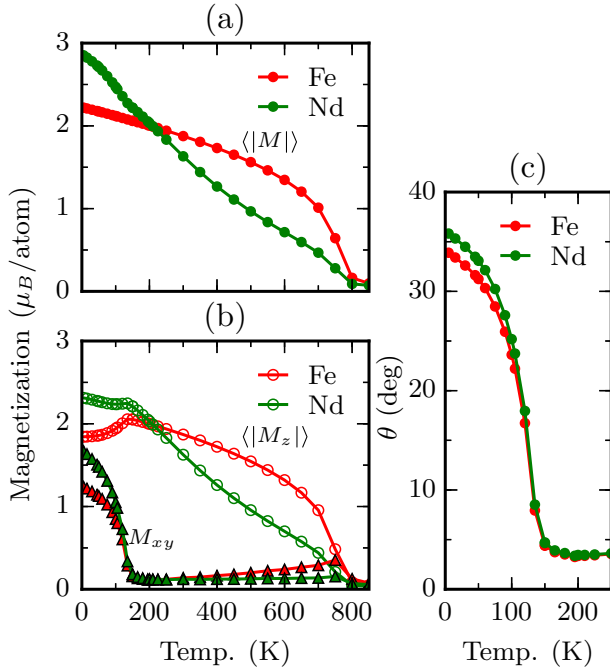


FIG. 7. Temperature dependence of (a) the magnetization amplitude, $\langle |M| \rangle$, and (b) the z -component, $\langle |M_z| \rangle$, and the xy -component, M_{xy} , of Fe and Nd atoms for $r_{\text{cut}} = 10.6 \text{ \AA}$ and $L = 6$. (c) Each magnetization angle measured from the z -axis at the low-temperature region.

for bcc-Fe [41–43] and MnBi [44], and so the dependence of r_{cut} appears to reflect the features of itinerant ferromagnetism. Under the condition that $r_{\text{cut}} = 3.52, 10.6,$ and 17.6 \AA , each atom has approximately 13, 350, and 1660 exchange coupling bonds, respectively. To reduce the computational load, we mainly consider $r_{\text{cut}} = 10.6 \text{ \AA}$.

At the lower temperature point $T_{\text{sr}} (\sim 145 \text{ K})$ in Fig. 4, $\langle |M_z| \rangle$ reaches a maximum and $M_{xy} = \sqrt{\langle M_x^2 \rangle + \langle M_y^2 \rangle}$ approaches 0, which is known to be the spin-reorientation transition of the $\text{Nd}_2\text{Fe}_{14}\text{B}$ magnet. The magnetization direction is tilted 34.4° from the z -axis at $T = 0$ for every r_{cut} . Above T_{sr} , this direction exhibits uniaxial anisotropy along the z -axis. In contrast to T_{C} , T_{sr} has only a weak dependence on r_{cut} . The spin-reorientation transition is mainly driven by the higher-order terms ($l = 4, 6$) of $A_l^{m_l=0}$ on the Nd atoms in Eq. (1). Indeed, in comparison to the tilting angle of the single Nd atom at $T = 0$ [$\theta = 36.7^\circ$ in Fig. 2(a)], we can see that the Fe magnetic anisotropy has little effect on the spin reorientation. The reorientation property of Nd atoms is shared with the whole $\text{Nd}_2\text{Fe}_{14}\text{B}$ through the exchange coupling $J_{\text{Fe-Nd}}^{\text{ex}}$. As shown in Fig. 6, most contributions of $J_{\text{Fe-Nd}}^{\text{ex}}$ are in the range $r_{\text{cut}} \leq 3.4 \text{ \AA}$. Therefore, T_{sr} has only a weak dependence on the long-range parts of J_{ij}^{ex} .

To look into the role for each atom in the above two transitions at T_{C} and T_{sr} , we plot in Fig. 7 the temperature dependence of the magnetizations and the magnetization angle of Nd and Fe atoms. In Fig. 7(a), the reduction of the magnetization amplitude $\langle |M| \rangle$ with the temperature of each atom shows a clear difference. This difference is reflected by the amplitude of exchange couplings: $\tilde{J}_{\text{Fe}}^{\text{ex}}$ ($\tilde{J}_{\text{Nd}}^{\text{ex}}$) for

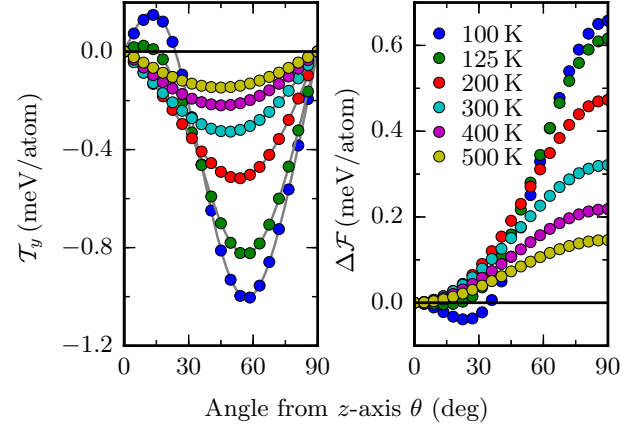


FIG. 8. Angular dependence of y -direction torque T_y (left side) and free-energy $\Delta\mathcal{F}$ (right side) at each temperature for $r_{\text{cut}} = 10.6 \text{ \AA}$ and $L = 4$. The gray lines on the left side show the fit of the torque data to $-\partial\Delta\mathcal{F}/\partial\theta$ in Eq. (8).

$r_1 = 0$ and $r_2 = 10.6 \text{ \AA}$ is 142.9 meV (33.5 meV). Hence, the ferromagnetic order of Fe is responsible for the magnetic order of the magnets. At high temperature, we may have a picture that the magnetization of the Nd atom is maintained by the interaction with the ordered Fe. The rapid decrease of Nd magnetization with temperature corresponds to the poor thermal properties of magnetic anisotropy (see the next section). On the other hand, from each magnetization angle θ in Fig. 7(c), we can verify that T_{sr} is mainly dependent on the magnetic anisotropy of Nd atoms, as was mentioned in the previous paragraph. The magnetization angle θ is calculated by using $\langle |M_z| \rangle$ and M_{xy} in Fig. 7(b) as follows:

$$\theta = \arctan \left(\frac{\langle |M_z| \rangle}{M_{xy}} \right). \quad (7)$$

In Fig. 7(c), the angle of Nd magnetization always has a larger value than the angle of Fe magnetization below T_{sr} . This behavior implies that the spin reorientation occurs because the tilted Nd magnetization attracts Fe magnetization.

It is necessary to keep in mind that the model parameters do not include the thermal variations of the lattice parameters and the electronic states. However, despite using many parameters from first-principles calculations, the above thermodynamic results ($T_{\text{C}} \sim 754 \text{ K}$, $T_{\text{sr}} \sim 145 \text{ K}$ for $r_{\text{cut}} = 10.6 \text{ \AA}$) are basically consistent with experimental values ($T_{\text{C}} \sim 585 \text{ K}$, $T_{\text{sr}} \sim 135 \text{ K}$) [1]. Therefore, the model and the parameter sets are sufficiently reliable for studying the temperature dependence of magnetic anisotropy in $\text{Nd}_2\text{Fe}_{14}\text{B}$.

B. Temperature dependence of magnetic anisotropy

We now discuss the temperature dependence of magnetic anisotropy. Figure 8 shows the y -direction torque T_y and free energy $\Delta\mathcal{F}$ as a function of magnetization angle θ for $L = 4$ as calculated by the C-MC method. In the present paper, the directions of magnetization constrained by the C-MC method are rotated by θ around the y -axis. Therefore, the torque is perpendicular to the x - z plane, i.e., both the x and z components of torque are zero.

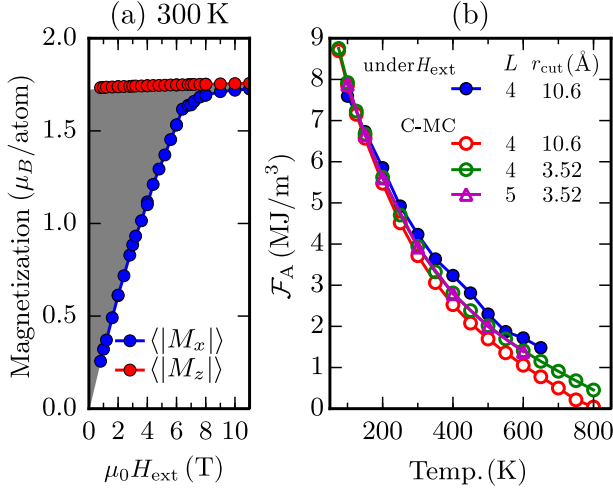


FIG. 9. Left: Magnetization curve $\langle |M_{z(x)}| \rangle$ under the $z(x)$ -direction of the external magnetic field H_{ext} ; the gray area corresponds to magnetic anisotropy energy \mathcal{F}_A in the blue solid lines on the right-hand panel. Right: \mathcal{F}_A for the four calculation conditions.

To verify the C-MC method, we compare the magnetic anisotropy energies \mathcal{F}_A with those given by the typical MC method, \mathcal{F}_A^H . Here, \mathcal{F}_A is defined as $\Delta\mathcal{F}_{\text{max}} - \Delta\mathcal{F}_{\text{min}}$ in Fig. 8, and \mathcal{F}_A^H is derived from the magnetization curves as the gray area on the left of Fig. 9 (see the example at $T = 300$ K), where $\langle |M_{x(z)}| \rangle$ is the magnetization curve under H_{ext} in the $x(z)$ -direction. From the right of Fig. 9, we can confirm that \mathcal{F}_A is in good agreement with \mathcal{F}_A^H , particularly in the low-temperature region, although \mathcal{F}_A^H tends to give an overestimate. This overestimate occurs because, at finite temperatures, the effective magnetic anisotropy of each spin decreases as a result of thermal fluctuations. When evaluating \mathcal{F}_A^H , the thermal fluctuations are suppressed by the external field to saturate the magnetization. This suppression becomes stronger as the temperature increases, causing the overestimation to be significant in the high-temperature region.

We also plot \mathcal{F}_A for other calculation conditions: $(L, r_{\text{cut}}) = (4, 3.52)$ and $(5, 3.52)$ on the right of Fig. 9. These results show that a system size of $L = 4$ is sufficient to obtain convergence in the magnetic anisotropy. Additionally, the length of r_{cut} affects \mathcal{F}_A at high temperatures. As mentioned in terms of spin reorientation, the magnetic anisotropy of Nd is essentially unaffected by differences in r_{cut} . Hence, it is thought that the difference between red and green lines in Fig. 9(b) occurs due to r_{cut} dependence of Fe anisotropy. Therefore, in the high-temperature region where Fe anisotropy becomes larger than the Nd anisotropy (see Fig. 11, $A_l^m = 0$ and $D^A = 0$), the effects on \mathcal{F}_A of differences in r_{cut} are clearly evident.

Returning to Fig. 8, we can see that for 100 and 125 K, the torque (free-energy) curve attains a local maximum (minimum) at $\theta \neq 0$, which reflect the spin reorientation (shown in Fig. 4). In contrast, above $T \geq 200$ K, the local maximum (minimum) disappears and the torque (free-energy) curve approaches $\propto \sin 2\theta$ ($\sin^2 \theta$). This behavior implies that the magnetic anisotropy constant K_1^A becomes dominant as the temperature increases.

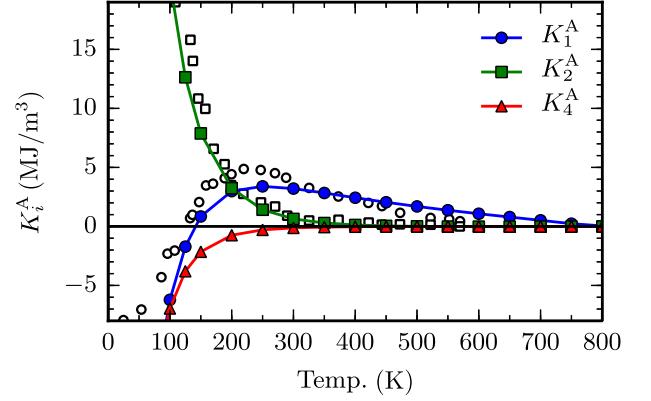


FIG. 10. Anisotropy constants K_m^A as a function of temperature for $r_{\text{cut}} = 10.6 \text{ \AA}$ and $L = 4$. White circles and squares indicate experimental results [45] for K_1^A and K_2^A , respectively.

To clarify the temperature dependence, Fig. 10 shows the magnetic anisotropy constants K_m^A ($m = 1, 2, 4$) that were calculated by fitting \mathcal{T}_y in Fig. 8 to the torque equation:

$$\mathcal{T}_y(\theta, T) = -\frac{\partial}{\partial \theta} \Delta\mathcal{F}(\theta, T),$$

$$\Delta\mathcal{F}(\theta, T) = K_1^A(T) \sin^2 \theta + K_2^A(T) \sin^4 \theta + K_4^A(T) \sin^6 \theta. \quad (8)$$

These constants can only be calculated correctly using the C-MC method. We can confirm that K_2^A and K_4^A tend to zero and K_1^A becomes dominant in the region of $T > 300$ K. Additionally, K_1^A becomes negative in the low-temperature region. This is reflected by the local minimum of $\Delta\mathcal{F}$ in Fig. 8, indicating the spin reorientation transition. The temperature dependence of K_m^A agrees reasonably well with previous experimental results [45–47] and mean-field theory [16,17]. Note that, at $T < 100$ K, all of the $|K_m^A|$ are significantly larger than the experimental values. For classical spin systems, this deviation in K_m^A (and also M) is finite at zero temperature on account of the infinite degrees of freedom of classical spin (for quantum spin systems, the deviations of K_m^A and M at $T = 0$ are zero) [17]. This explains the difference between our results and the experimental results at $T < 100$ K.

To examine the relationship between the exchange coupling and magnetic anisotropy, we considered various input parameter sets. Figure 11 shows the anisotropy energy \mathcal{F}_A for five cases: the same result as shown by the red lines in Fig. 9 (default), a model including only Fe magnetic anisotropy ($A_l^m = 0$), a model including only Nd magnetic anisotropy ($D^A = 0$), a model with all $J_{\text{Fe-Nd}}^{\text{ex}}$ reduced by half ($0.5J_{\text{Fe-Nd}}^{\text{ex}}$), and a model with all $J_{\text{Fe-Nd}}^{\text{ex}}$ increased by half ($1.5J_{\text{Fe-Nd}}^{\text{ex}}$). In the case of $A_l^m = 0$, the anisotropy energy decreases almost linearly with temperature. This behavior is a typical property of the classical Heisenberg models that include only $\sin^2 \theta$ for the anisotropy energy. In contrast, the case of $D^A = 0$ exhibits a rapid decrease, which can be explained by the difference in the exchange coupling $\tilde{J}_{\text{atom}}^{\text{ex}}(r_1, r_2)$ of Nd and Fe atoms [see Eq. (6)]. We have that $\tilde{J}_{\text{Fe}}^{\text{ex}}$ and $\tilde{J}_{\text{Nd}}^{\text{ex}}$ for $r_1 = 0$ and $r_2 = 10.6 \text{ \AA}$ are 142.9 and 33.5 meV, respectively. Here, $\tilde{J}_{\text{Fe(Nd)}}^{\text{ex}}$ is almost given by the Fe-Fe (Nd-Fe) exchange couplings (see Fig. 3).

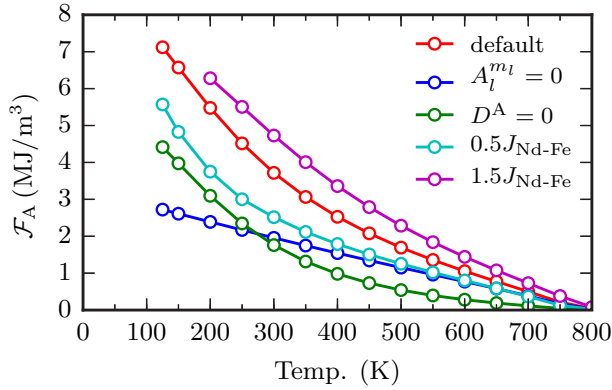


FIG. 11. Comparison of the temperature dependence of magnetic anisotropy energy \mathcal{F}_A in five cases (details in the text) for $r_{\text{cut}} = 10.6 \text{ \AA}$ and $L = 4$.

The Nd atoms, which give the whole $\text{Nd}_2\text{Fe}_{14}\text{B}$ system magnetic anisotropy through $\tilde{\mathcal{J}}_{\text{Nd}}^{\text{ex}}$, are highly susceptible to thermal fluctuations, unlike the Fe atoms, which play a key role in magnetism (such as $|\mathbf{M}|$ and T_C). This difference in thermal susceptibility explains the rapid decrease in \mathcal{F}_A for $D^A = 0$. For the same reason, in the case of $0.5J_{\text{Fe-Nd}}^{\text{ex}}$, which includes both A_l^{mi} and D^A , \mathcal{F}_A decreases rapidly with temperature and approaches $A_l^{mi} = 0$ at approximately 400 K. This means that the effects of Nd magnetic anisotropy are almost wiped out by thermal fluctuations above 400 K. However, for $1.5J_{\text{Fe-Nd}}^{\text{ex}}$, \mathcal{F}_A is almost linear. The above discussion for Fig. 11 allows us to understand that $\tilde{\mathcal{J}}_{\text{Nd}}^{\text{ex}}$ (rather than $\tilde{\mathcal{J}}_{\text{Fe}}^{\text{ex}}$) makes a strong contribution to the magnetic anisotropy of Nd atoms, which supports the results of previous studies [16,48].

To analyze the results shown in Fig. 11 in the context of the Callen-Callen law [13,14], i.e., $K_1^A(T) \propto M(T)^3$ for $K_2^A = K_4^A = 0$, Fig. 12 illustrates the relationship between K_1^A and M above 300 K. It is clear that $1.5J_{\text{Fe-Nd}}^{\text{ex}}$ deviates from this law, because K_2^A is comparable to K_1^A at 300 K. Varying the anisotropy terms A_l^{mi} and D^A affects these relations more

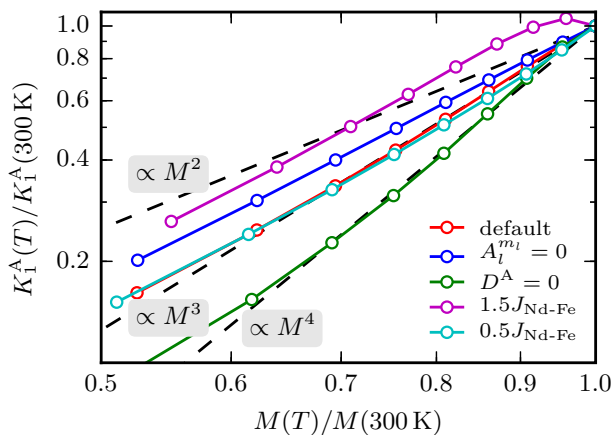


FIG. 12. Relation between $K_1^A(T)$ and $M(T)$ at each temperature for the same parameter sets and calculation conditions in Fig. 11. The natural logarithm is taken for both axes. The Callen-Callen law corresponds to $M(T)^3$, illustrated by a dashed line.

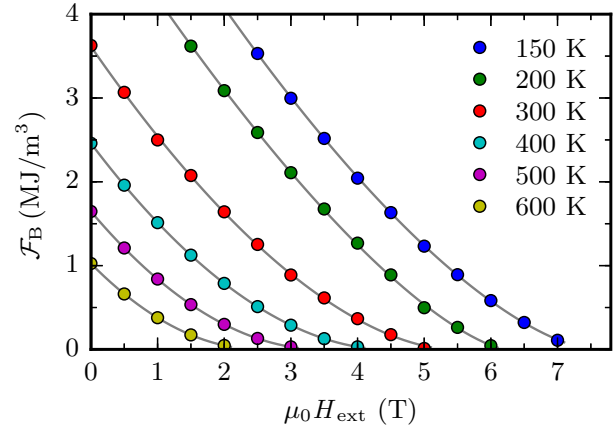


FIG. 13. Height of the energy barrier, \mathcal{F}_B , as a function of the external magnetic field, H_{ext} , at each temperature for $r_{\text{cut}} = 10.6 \text{ \AA}$ and $L = 4$. The gray lines illustrate the fit to Eq. (9).

than $\tilde{\mathcal{J}}_{\text{Nd}}^{\text{ex}}$. For $A_l^{mi} = 0$, Nd magnetization decreases rapidly with temperature, whereas Fe anisotropy decreases gradually. Hence, K_1^A/M tends to increase. Conversely, for $D^A = 0$, Fe magnetization slowly decreases with temperature, whereas Nd anisotropy decreases rapidly, hence K_1^A/M tends to decrease. As the above two effects happen to cancel out, the default case and $0.5J_{\text{Fe-Nd}}^{\text{ex}}$ agree with the Callen-Callen law.

The Callen-Callen law was derived under the assumption of homogeneous ferromagnetic and single-ion anisotropy systems at temperatures far from T_C . Therefore, it is natural that a multi-sublattice model such as $\text{Nd}_2\text{Fe}_{14}\text{B}$ does not follow the Callen-Callen law, which was also pointed out by using a mean-field approach [49]. Additionally, in actual ferromagnetic metals that have two-ion magnetic anisotropy, the temperature dependence of magnetic anisotropy deviates from Callen-Callen law [19,50–53], such as $L1_0\text{-FePt}$, $K_1^A(T) \propto M^{2.1}(T)$ [51]. Therefore, more detailed discussion of the temperature dependence is needed to formulate the theory for itinerant electrons and inhomogeneous systems.

C. Energy barrier

Finally, we discuss the external magnetic field H_{ext} response of the energy barrier (activation energy) [20–24], which governs the probability of magnetization reversal via the thermal fluctuation of spins. If this response can be measured experimentally [25], it would allow the magnetic coercivity mechanism to be predicted at finite temperatures.

Figure 13 shows the height of the energy barrier, \mathcal{F}_B , when H_{ext} is applied opposite to the z -direction of \mathbf{M} . By uniformly rotating the direction of \mathbf{M} using the C-MC method, we evaluated \mathcal{F}_B ; therefore, $\mathcal{F}_B = \mathcal{F}_A$ for $H_{\text{ext}} = 0$. The H_{ext} response of \mathcal{F}_B is generally expressed by [20]

$$\mathcal{F}_B(H_{\text{ext}}) = \mathcal{F}_B^0(1 - H_{\text{ext}}/H_0)^n, \quad (9)$$

where $\mathcal{F}_B^0 = \mathcal{F}_B(H_{\text{ext}} = 0) = \mathcal{F}_A$, and H_0 is equal to the value of H_{ext} at $\mathcal{F}_B = 0$, which corresponds to the upper limit of the coercive field, H_c , under uniform rotation. For finite temperatures, the thermal fluctuation helps the magnetization

TABLE II. Fitting parameters (\mathcal{F}_B^0, H_0, n) at each temperature. The exponent n_K^s was estimated with the single-spin model [Eq. (10)] using the anisotropy constant K_m^A in Fig. 10 instead of κ_m .

Temp. (K)	\mathcal{F}_B^0 (MJ/m ³)	$\mu_0 H_0$ (T)	n	$\frac{\kappa_2^A}{\kappa_1^A}$	$\frac{\kappa_4^A}{\kappa_1^A}$	n_K^s
150	6.53	7.54	1.53	9.35	-2.55	1.56
200	5.37	6.23	1.42	1.08	-0.25	1.44
300	3.61	5.41	1.72	0.2	-0.04	1.72
400	2.46	4.44	1.90	0.05	-0.01	1.91
500	1.65	3.46	1.97	0	0	2.00
600	1.03	2.55	2.00	-0.02	0	2.05

reversal to overcome the energy barrier, and so H_c is much lower than H_0 . The exponent n can take various values, such as $n = 2$ for the Stoner-Wohlfarth model and $n = 1$ for the weak domain-wall pinning mechanism [20].

The parameters \mathcal{F}_B^0 , H_0 , and n were obtained by fitting $\mathcal{F}_B(H_{\text{ext}})$ in Fig. 13, and they are listed in Table II. We can see that n takes values of less than 2 in the low-temperature region (below the room temperature, $T_R \sim 300$ K) and approaches 2 as the temperature increases. This reflects the fact that the magnetic anisotropy is mainly governed by the K_1^A term in the high-temperature region (see Fig. 10). To clarify this, we estimated the exponent n^s by fitting from the anisotropy energy of the single-spin model, which is defined as

$$E_A^s(\theta) = \kappa_1 \sin^2 \theta + \kappa_2 \sin^4 \theta + \kappa_4 \sin^6 \theta + m H_{\text{ext}} \cos \theta. \quad (10)$$

With $\kappa_2 = \kappa_4 = 0$, this corresponds to the Stoner-Wohlfarth model. The dependence of the anisotropy constant on n^s is plotted in Fig. 14. This figure confirms that κ_2 and κ_4 have a significant effect on n^s for (a) $\kappa_1 > 0$, whereas n^s is less sensitive for (b) $\kappa_1 < 0$, which corresponds to the low-temperature region below T_{sr} of Nd₂Fe₁₄B (see Fig. 10).

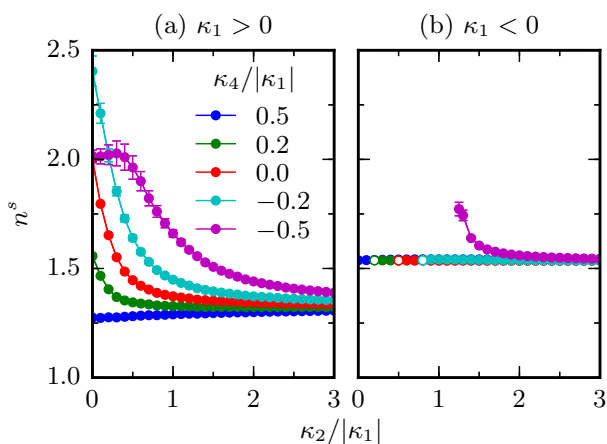


FIG. 14. The exponent n^s in the magnetic-field response for the single-spin model, Eq. (10), as a function of $\kappa_2/|\kappa_1|$ (≥ 0) with fitting error bars (95% confidence) on each $\kappa_4/|\kappa_1|$ for (a) $\kappa_1 > 0$ and (b) $\kappa_1 < 0$. The energy barrier \mathcal{F}_B disappears when $\kappa_2/|\kappa_1|$ is below each white point in (b).

Here, the deviation of n^s given by fitting Eq. (9) becomes large when either $|\kappa_2|$ or $|\kappa_4|$ increases. Therefore, near the points where fitting error bars are large (see Fig. 14), we should pay attention to the n^s values, which are dependent on fitting procedures.

Additionally, we input K_m^A (from Fig. 10) into κ_m in Eq. (10), and we estimated the exponent n_K^s listed in Table II. Despite using the single-spin model, n_K^s is in good agreement with n , where n has been evaluated on an inhomogeneous spin system such as Nd₂Fe₁₄B. This indicates that, in terms of the magnetic-field response of uniform rotation, the anisotropy constants K_m^A are renormalized by the magnetic inhomogeneities and thermal fluctuations. For the Nd₂Fe₁₄B system, in particular, we should bear in mind that the response occurs for $n < 2$ when below room temperature, T_R .

IV. SUMMARY

We have constructed a realistic classical three-dimensional Heisenberg model using parameters from first-principles calculations, and we investigated the magnetic properties of the Nd₂Fe₁₄B bulk system at finite temperatures. Applying the constrained Monte Carlo method to this model, from atomic-scale parameters, we evaluated macroscopic magnetic anisotropies, which include correctly magnetic inhomogeneities and thermal fluctuations. Despite using many parameters from first-principles calculations (except for A_l^m), the model reproduced the experimentally observed spin reorientation and magnetic anisotropy constants K_m^A .

Using this calculation system, we found that, because the exchange couplings between Nd moments and Fe spins are much smaller than those between Fe spins, the magnetic anisotropy of Nd atoms decreases more rapidly than that of Fe atoms. Additionally, due to this magnetic inhomogeneity, the temperature dependence of K_1^A deviates from the Callen-Callen law, even above room temperature ($T_R \sim 300$ K), when the Fe (Nd) anisotropy terms are removed to leave only the Nd (Fe) anisotropy. Furthermore, we also found that the exponent n in the magnetic-field response of barrier height is less than 2 in the low-temperature region below T_R , whereas n approaches 2 when $T > T_R$, indicating Stoner-Wohlfarth-type magnetization rotation. This behavior reflects the fact that the magnetic anisotropy is mainly governed by the K_1^A term in $T > T_R$, which is explained by the single-spin model with a renormalized K_m^A .

We have a plan to extend the constructed framework in the present paper to nonuniform magnetization reversal in finite-size particles, including the effects of the grain surfaces or grain boundaries.

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