

Magnetism in Sr₂CrMoO₆: A combined *ab initio* and model studyPrabuddha Sanyal,¹ Anita Halder,² Liang Si,³ Markus Wallerberger,³ Karsten Held,³ and Tanusri Saha-Dasgupta²¹*Department of Physics, Indian Institute of Technology, Roorkee 247667, India*²*Department of Condensed Matter Physics and Materials Science, S. N. Bose National Center for Basic Sciences, Kolkata, India*³*Institute for Solid State Physics, TU Wien, 1040 Wien, Austria*

(Received 22 March 2016; revised manuscript received 15 June 2016; published 14 July 2016)

Using a combination of first-principles density functional theory (DFT) calculations and exact diagonalization studies of a first-principles derived model, we carry out a microscopic analysis of the magnetic properties of the half-metallic double perovskite compound Sr₂CrMoO₆, a sister compound of the much discussed material Sr₂FeMoO₆. The electronic structure of Sr₂CrMoO₆, though appearing similar to Sr₂FeMoO₆ at first glance, shows nontrivial differences with that of Sr₂FeMoO₆ on closer examination. In this context, our study highlights the importance of charge transfer energy between the two transition metal sites. The change in charge transfer energy due to a shift of Cr *d* states in Sr₂CrMoO₆ compared to Fe *d* in Sr₂FeMoO₆ suppresses the hybridization between Cr *t*_{2g} and Mo *t*_{2g}. This strongly weakens the hybridization-driven mechanism of magnetism discussed for Sr₂FeMoO₆. Our study reveals that, nonetheless, the magnetic transition temperature of Sr₂CrMoO₆ remains high since an additional superexchange contribution to magnetism arises with a finite intrinsic moment developed at the Mo site. We further discuss the situation in comparison to another related double perovskite compound, Sr₂CrWO₆. We also examine the effect of correlation beyond DFT, using dynamical mean field theory.

DOI: [10.1103/PhysRevB.94.035132](https://doi.org/10.1103/PhysRevB.94.035132)**I. INTRODUCTION**

In recent years, double perovskites with a general formula $A_2BB'O_6$ (*A*: rare-earth/alkaline-earth cation; *B*: 3*d* transition metal; *B'*: 4*d*/5*d* transition metal) have been the focus of discussion due to their attractive properties. This includes multiferroicity [1,2], magnetodielectric and magnetocapacitive properties [3,4], complex spin behavior [5], magnetostructural coupling [6], etc. Many of these compounds show half-metallic behavior, which is of importance for spintronics and related technological applications [7–14]. The high magnetic transition temperatures (T_c) exhibited by compounds such as Sr₂FeMoO₆ open up the possibility of room-temperature applications [15,16]. The pioneering work reporting [15] room-temperature magnetoresistivity of Sr₂FeMoO₆ (SFMO) was met with excitement and was followed by a surge of activity exploring the dependence of 3*d* transition metal (TM) ions on the properties of these compounds.

Following this motivation, Cr-based double perovskites have been synthesized and studied [17], in particular, the sister compound of SFMO, Sr₂CrMoO₆ (SCMO) [18]. Unlike in SFMO, there can be no valence compensation between Cr and the Mo in SCMO: Cr can only be in the 3+ state, making Cr³⁺/Mo⁵⁺ the only possible combination, while for SFMO, both Fe³⁺/Mo⁵⁺ and Fe²⁺/Mo⁶⁺ combinations are possible. Thus, SCMO was expected to be an even better candidate for room-temperature spintronics. However, although the measured transition temperature of SCMO is high [18,19], the observed moment [20] and the magnitude of the tunneling magnetoresistance turned out to be disappointingly low. The presence of large antisite disorder, together with an oxygen vacancy, was responsible for this [19]. The antisite disorder present in the samples was estimated to be as high as 43%–50% [18,21]. To appreciate the role of 3*d* TM in the properties of double perovskites, it is thus highly desirable to understand the electronic and magnetic properties of pure SCMO.

Theoretical studies on SCMO have been carried out within the framework of density functional theory (DFT) [22,23]. However, no microscopic analysis has been carried out, and the transition temperature has yet to be calculated, except for a recent mean field analysis [24] based on a classical Ising model. The latter neglects the itinerant electronic character of Mo 4*d* electrons, a crucial component in understanding the behavior of these 3*d*-4*d* double perovskites.

Furthermore, though the hybridization-driven mechanism of magnetism was identified [10] a while ago as the driving force in setting up the high Curie temperature in compounds such as SFMO, the role of charge transfer between the *B* and *B'* sites in the perspective of properties of double perovskite compounds in general has not been studied in detail. In particular, no systematic study exists. This is, however, an important issue in understanding the comparative magnetic properties of double perovskite compounds.

In the present study, we aim to fill this gap by combining state-of-art DFT calculations and exact diagonalization of the DFT derived model Hamiltonian. We consider the case of SCMO, in comparison to SFMO as well as Sr₂CrWO₆ (SCWO) [17], which is another double perovskite from the Cr family but with 5*d* W instead of 4*d* Mo. Our microscopic study shows a considerable suppression of the Cr *t*_{2g}-Mo *t*_{2g} hybridization in SCMO compared to that in SFMO or SCWO, driven by the change in charge transfer energy between *B* and *B'* sites in different compounds. The suppressed hybridization in SCMO makes the Mo *t*_{2g} electrons more localized compared to SFMO or SCWO. This, in turn, opens up an additional superexchange contribution to magnetism. The calculated magnetic transition temperature T_c without a superexchange contribution gives about 79% of the measured T_c . The consideration of an additional superexchange contribution makes the calculated T_c comparable to the measured one. Our work reveals the importance of the superexchange in SCMO in addition to the hybridization-driven mechanism operative in SFMO or SCWO, and underlines the crucial role played

by the charge transfer energy in activating the additional superexchange channel in SCMO [15].

We further study the effect of correlation on the half-metallic property of SCMO by including local correlations on top of DFT within the framework of dynamical mean field theory (DMFT). The DMFT results confirm the half-metallic ground state below the magnetic transition temperature, implying that the qualitative description remains unaffected by correlation effects.

II. COMPUTATIONAL DETAILS

The first-principles DFT calculations have been carried out using the plane-wave pseudopotential method implemented within the Vienna *Ab initio* simulation package (VASP) [25]. The exchange-correlation functional was considered within the generalized gradient approximation (GGA) in the framework of Perdew and Wang (PW91) [26]. Additionally, calculations have been carried with an exchange-correlation functional of the local density approximation (LDA) [27] as well as GGA within the framework of Perdew-Burke-Ernzerhof (PBE) [28], in order to check any possible influence of the exchange-correlation functional on the calculated electronic structure. The projector augmented-wave (PAW) potentials [29] were used and the wave functions were expanded in the plane-wave basis with a kinetic-energy cutoff of 500 eV. Reciprocal-space integrations were carried out with a k -space mesh of $8 \times 8 \times 8$.

In order to extract a few-band tight-binding (TB) Hamiltonian out of the full DFT calculation which can be used as an input to the multiorbital, low-energy model Hamiltonian in exact diagonalization calculations, we have carried out N th-order muffin-tin orbital (NMTO) calculations [30]. A prominent feature of this method is the downfolding scheme. Starting from a full DFT calculation, it defines a few-orbital Hamiltonian in an energy-selected, effective Wannier function basis, by integrating out the degrees of freedom that are not of interest. The NMTO technique relies on the self-consistent potential parameters obtained out of linear muffin-tin orbital (LMTO) [31] calculations. In order to cross check the TB parameters generated out of NMTO-downfolding calculations, further calculations were carried out using WIEN2WANNIER [32]. This generates maximally localized Wannier functions [33] from WIEN2K [34] which employs a full potential linear augmented plane-wave (FLAPW) basis. For self-consistent DFT calculations in the FLAPW basis, the number of k points in the irreducible Brillouin zone was chosen to be 64. The commonly used criterion relating the plane-wave and angular momentum cutoff, $l_{\max} = R_{\text{MT}} \times K_{\max}$, was chosen to be 7.0, where R_{MT} is the smallest MT sphere radius and K_{\max} is the plane-wave cutoff for the basis. The chosen atomic radii for Sr, Cr, Mo, and O were 1.43, 1.01, 1.01, and 0.87 Å, respectively.

The exact diagonalization of the *ab initio* derived low-energy Hamiltonian, defined in the Wannier function basis, has been carried out for finite-size lattices of dimensions $4 \times 4 \times 4$, $6 \times 6 \times 6$, and $8 \times 8 \times 8$. The results presented in the following are for the $8 \times 8 \times 8$ lattice.

The DFT+DMFT calculations [35,36] have been carried out using the WIEN2WANNIER [32] derived maximally localized

Wannier functions of WIEN2K as a starting point. We restricted the DMFT to the low-energy degrees of freedom, i.e., to the “ t_{2g} ” orbitals of Mo and the “ t_{2g} ” and “ e_g ” orbitals of Cr [these low-energy orbitals are actually a mixture of predominately transition metal t_{2g} (e_g) character with some admixture of oxygen p character]. We supplemented the DFT low-energy Hamiltonian in the Wannier basis by a local Coulomb interaction in Kanamori parametrization. For details, see Ref. [37]. For the interaction values we chose the typical values for $3d$ and $4d$ transition metal oxides. The choices were as follows: interorbital Coulomb repulsion of $U' = 4$ eV (2.4 eV) and a Hund’s coupling $J_H = 0.7$ eV (0.3 eV) for Cr (Mo) as estimated for neighboring vanadium [38] (ruthenium [39]) perovskites. The intraorbital (Hubbard) repulsion follows from orbital symmetry as $U = U' + 2J_H$, and the pair-hopping term is of equal strength as J_H . As a DMFT impurity solver, continuous-time quantum Monte Carlo simulations [40] in the w2DYNAMICS [37] implementation was used, which includes the full SU(2) symmetry. The effect of electronic correlations beyond GGA, within the framework of static theory, was also checked by performing GGA+ U calculations with a choice of the same U parameters, as in DMFT calculations. We have also checked the validity of our results by varying the U value by ± 1 eV at the Cr site, and by ± 0.5 eV at the Mo site. The trend in the results was found to remain unchanged.

III. RESULTS

A. Basic DFT electronic structure

We first revisit the basic DFT electronic structure of SCMO, which has been calculated before using a variety of basis sets, including plane wave [41], LAPW [22], and LMTO [23]. SCMO crystallizes in the cubic $Fm\bar{3}m$ space group (225), with a lattice parameter of 7.84 Å [19].

The upper panel of Fig. 1 shows the spin-polarized density of states (DOS) calculated in a plane-wave basis. The basic features of the DOS, which agree very well with previous studies [22,23,41], are the following: The states close to the Fermi level E_F (set as zero in the figure) are dominated by Cr and Mo d states hybridized with O p states. The states of dominant O p character are positioned further down in energy, separated by a small gap from Cr and Mo d states. Empty Sr states, not shown in the energy scale of the figure, remain far above E_F . Due to the large octahedral crystal field at the Mo sites, the empty Mo e_g states lie far above E_F . The Cr t_{2g} states are occupied in the up spin channel and are empty in the down spin channel. Cr e_g states are empty in both spin channels, in agreement with the nominal Cr³⁺ (d^3) valence. The empty and highly peaked Mo t_{2g} states in the up spin channel appear in between the crystal-field-split Cr t_{2g} and Cr e_g states for the same spin. In contrast for the down spin channel, the Mo t_{2g} hybridize more strongly with Cr t_{2g} , which explains their much larger bandwidth.

The above described basic features for B and B' states are rather similar for SCMO and SFMO [42], and in that respect also for SCWO [43]. Nevertheless, the DOS of SCMO differs from both SFMO and SCWO in important details. The hybridization between B t_{2g} and B' t_{2g} in the down spin channel is found to be significantly lower compared

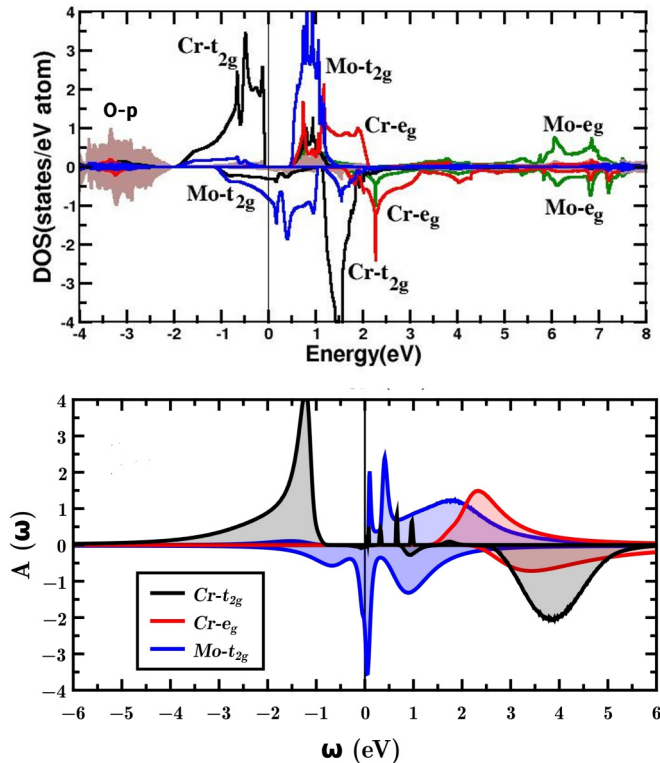


FIG. 1. Upper panel: GGA-PW91 spin-polarized DOS projected onto Cr t_{2g} (black solid line), Cr e_g (red solid line), Mo t_{2g} (blue solid line), Mo e_g (green solid line), and O p (shaded area) in the FLAPW basis. The zero of the energy is set to the Fermi energy. The positive (negative) axis of DOS corresponds to DOS in the up (down) spin channel. Lower panel: DMFT spectral density, calculated at 200 K within the Mo t_{2g} and Cr $t_{2g} + e_g$ Wannier basis [45].

to that of SFMO or SCWO [42,43]. A measure of this is the Cr contribution in the down spin bands crossing E_F of predominant Mo character. We estimate this contribution or admixture to be 35% for SCMO, while the corresponding estimates for SCWO and SFMO are much larger, 66% and 72%, respectively.

The top rows in Table I show the calculated magnetic moments at Cr and Mo sites, as well as the total moment in three choices of exchange-correlation functionals, GGA-PW91, GGA-PBE, as well as LDA. The half-metallic nature of the ground state is found to be robust in all different choices of exchange-correlation functionals with a total magnetic

TABLE I. Calculated magnetic moments (in μ_B) within LDA, GGA-PW91, GGA-PBE, GGA+ U , and DMFT. Note that within GGA there is also a moment on the oxygen sites which is accounted for in the Cr and Mo moment in DMFT as the predominately metal d Wannier functions also have some oxygen admixture.

	Sr	Cr	Mo	total
LDA	0.00	-2.18	0.32	-2.00
GGA-PW91	0.00	-2.29	0.43	-2.00
GGA-PBE	0.00	-2.36	0.49	-2.00
GGA+ U	0.00	-2.54	0.58	-2.00
DMFT	0.00	-2.84	0.84	-2.00

moment of $2.0\mu_B/\text{f.u.}$ and finite moments residing at the O sites. As is seen, while the individual moments on Cr and Mo are found to vary depending on the nature of the approximation, the magnetic moments on the Cr and Mo site being aligned in an antiparallel manner, the enhancements/reductions of individual moments cancel, thereby retaining the half metallicity and net moment of $2\mu_B/\text{f.u.}$ Our calculated GGA magnetic moments at Cr and Mo sites are in good agreement with those reported by Liet *et al.* [41]. The calculated site-specific moment reported by Wu [23] is, however, much smaller than that of ours as well as that of Liet *et al.* [41]. This is presumably due to different choices of the muffin-tin radii as well as the exchange-correlation functional.

The measured magnetic moments are significantly smaller than the theoretical values. Experimentally, the total moment is only $0.5\mu_B$ [20] and the moment on the Cr sites is $0.8\mu_B$ [44]. This discrepancy was hitherto argued to be due to the large disorder present in the sample [19]. It is interesting to note that the value of the calculated GGA-PW91 magnetic moment at the Mo sites ($0.43\mu_B$) is larger than the calculated B' -site moment for SFMO ($0.23\mu_B$) and SCWO ($0.30\mu_B$). This indicates a weaker itinerancy of the Mo electrons in SCMO, compared to SFMO or to that of W in SCWO. This in turn suggests that a small, but finite, intrinsic moment develops at the Mo site as a consequence of the weaker hybridization between Cr and Mo. The suppression of the hybridization and the reduced itinerancy of the Mo t_{2g} electrons in SCMO has been also pointed out in the study by Wu [23], though no detailed understanding of the mechanism of magnetism was provided. In the following we will elaborate on the microscopic mechanism of magnetism in SCMO.

B. DMFT spectral density

In order to study the effect of electronic correlations, especially the dynamical correlation which may be important for SCMO due to the metallic nature of the ground state, we further carry out DMFT calculations in the Wannier function basis. The lower panel in Fig. 1 shows the DMFT spectral density calculated at a temperature of 200 K.

The spectrum shows qualitatively similar behavior to that obtained in DFT calculations, though the down spin spectrum shows the correlation physics with a feature resembling the lower and upper Hubbard bands and a well-defined quasiparticle peak at Fermi energy. This demonstrates the dual nature of the Mo down spin with localized electrons (Hubbard bands) and itinerant electrons (quasiparticle band) at the same time. The fully spin-polarized conducting electrons are Fermi-liquid-like with a linear frequency dependence of the self-energy (not shown). Hence, the quasiparticle peak will lead to a Drude peak in the optical conductivity.

The DMFT magnetic moments are shown in the last row of Table I. We note the DMFT calculated magnetic moments are in the basis of Cr and Mo t_{2g} Wannier functions, which includes the effect of oxygen. This makes a direct comparison between DMFT and GGA values somewhat more difficult. However, since the difference between GGA and DMFT magnetic moments is larger than the GGA oxygen contribution, we can conclude that electronic correlations somewhat enhance the magnetic moment on both Mo and Cr sites. This is

further supported by the magnetic moments calculated within the static theory of GGA+ U , with a choice of the same U parameters as in DMFT calculations, shown in the second row of Table I, which shows an enhancement of the moment both at Cr and Mo sites, compared to that of GGA.

C. Few-orbital, low-energy Hamiltonian

In order to understand the driving mechanism of magnetism in SCMO in a more quantitative manner, we carry out a NMTO downfolding in order to estimate the positions of exchange-split Mo t_{2g} energy levels, before and after switching on the hybridization between Mo t_{2g} and Cr t_{2g} . The former provides the estimate of intrinsic spin splitting at the Mo site, while the latter provides the information of the spin splitting at the Mo site renormalized by the hybridization effect from Cr t_{2g} . As a first step of this procedure, we downfold O p , Sr, as well as Cr and Mo e_g degrees of freedom. This defines a few-orbital Hamiltonian consisting of Cr t_{2g} and Mo t_{2g} states. In the second step, we perform a further downfolding, keeping only the Mo t_{2g} degrees of freedom, i.e., downfolding everything else including the Cr t_{2g} degrees of freedom. On-site matrix elements of the few-orbital Hamiltonian in a real-space representation defined in the Cr t_{2g} -Mo t_{2g} basis and the massively downfolded basis give the energy level positions before and after switching of the hybridization, respectively. The obtained result is presented in Fig. 2. First of all, we notice that the Mo t_{2g} states are energetically situated in between the exchange-split energy levels of Cr t_{2g} 's. Thus switching of the hybridization between Cr and Mo pushes the Mo up spin states down because these are below the Cr states of the same spin. In contrast, the Mo down spins are above their Cr counterpart and hence the hybridization shifts them upwards. Thus the hybridization induces a renormalization of the spin splitting at the Mo site, with a renormalized value of about 0.70 eV, and being oppositely oriented (negative) with respect to that at the Cr site. Note the intrinsic (in the absence of hybridization) spin splitting at Mo site is small, having a value of 0.15 eV.

In this respect, the situation is very similar to SFMO or SCWO, for which also a negative splitting is induced at the itinerant B' sites because of the hybridization with the large spin at the B sites [10,15,43,46]. This supports the notion that

the hybridization-driven mechanism is operative in SCMO as well. While in all three cases of SFMO, SCWO, and SCMO, the B' t_{2g} states appear in between the strong exchange-split energy levels of the B site, which is an essential ingredient for the hybridization-driven mechanism to be operative, we notice the relative energy position of B' t_{2g} states with respect to the exchange-split B states is different in the case of SCMO, as compared to SFMO or SCWO. For SFMO or SCWO, the down spin B' t_{2g} states appear very close to the B -site down spin states, while for SCMO, they are shifted down [15,43]. This hints towards a significantly different charge transfer energy in the case of SCMO as compared to SFMO or SCWO. This will be elaborated on below. We further notice that the intrinsic splitting at the Mo sites (0.15 eV) is somewhat larger than for SCWO (0.05 eV) [43]. This further confirms the conclusion drawn from the calculated magnetic moment at the Mo site that is in SCMO, that, unlike SFMO, Mo has a finite intrinsic moment. The magnetism in SCMO, as mentioned already, thus has an additional contribution, originating from the superexchange between the large moment at the Cr site and the intrinsic moment at the Mo site, on top of the hybridization-driven mechanism, as operative in SFMO. Note that the superexchange is antiferromagnetic, aligning the Cr and Mo moments antiparallely, i.e., in the same way as for the hybridization-driven mechanism.

In this situation, the low-energy model Hamiltonian for SCMO in the Cr t_{2g} and Mo t_{2g} Wannier basis, describing the hybridization and superexchange mechanism, is given by [43,47]

$$\begin{aligned}
 H = & \epsilon_{\text{Cr}} \sum_{i \in B} f_{i\sigma\alpha}^\dagger f_{i\sigma\alpha} + \epsilon_{\text{Mo}} \sum_{i \in B'} m_{i\sigma\alpha}^\dagger m_{i\sigma\alpha} \\
 & - t_{\text{Cr-Mo}} \sum_{(ij)\sigma,\alpha} f_{i\sigma,\alpha}^\dagger m_{j\sigma,\alpha} \\
 & - t_{\text{Mo-Mo}} \sum_{(ij)\sigma,\alpha} m_{i\sigma,\alpha}^\dagger m_{j\sigma,\alpha} \\
 & - t_{\text{Cr-Cr}} \sum_{(ij)\sigma,\alpha} f_{i\sigma,\alpha}^\dagger f_{j\sigma,\alpha} + J \sum_{i \in \text{Cr}} \mathbf{S}_i \cdot f_{i\alpha}^\dagger \vec{\sigma}_{\alpha\beta} f_{i\beta} \\
 & + J' \sum_{i \in \text{Cr}, j \in \text{Mo}} \mathbf{S}_i \cdot \mathbf{s}_j. \tag{1}
 \end{aligned}$$

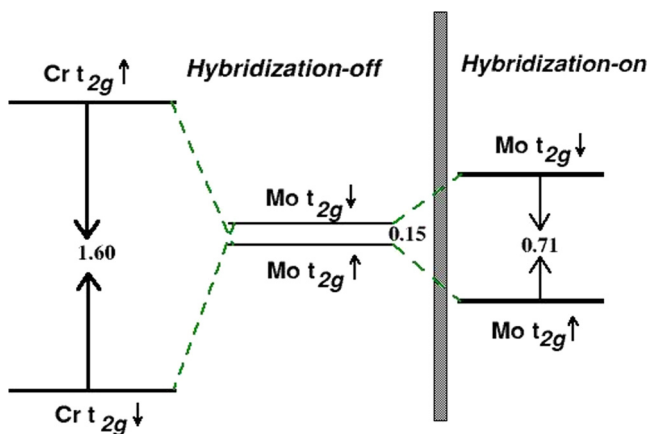


FIG. 2. The energy level diagram for SCMO, in the absence and presence of Cr-Mo hybridization. The energies are in units of eV.

Here, the f 's and m 's are the second quantization operators for the Cr t_{2g} and Mo t_{2g} degrees of freedom; σ is the spin index and α is the orbital index that spans the t_{2g} manifold; $t_{\text{Cr-Mo}}$, $t_{\text{Mo-Mo}}$, and $t_{\text{Cr-Cr}}$ represent the nearest-neighbor Cr-Mo, the second-nearest-neighbor Mo-Mo, and the Cr-Cr hopping, respectively. The on-site energy difference between Cr t_{2g} and Mo t_{2g} levels is $\Delta = \epsilon_{\text{Cr}} - \epsilon_{\text{Mo}}$. To take into account the dual nature of the electrons that are both itinerant and localized, we include a large core spin \mathbf{S}_i at the Cr site that couples with the itinerant electron delocalized over the Cr-Mo network, via a double-exchange-like mechanism. The last term represents the superexchange mechanism in terms of coupling between the Cr spin (\mathbf{S}_i) and the intrinsic moment on the Mo site (\mathbf{s}_j).

All TB parameters of the model Hamiltonian, i.e., Δ , $t_{\text{Cr-Mo}}$, $t_{\text{Mo-Mo}}$, $t_{\text{Cr-Cr}}$, are extracted from non-spin-polarized DFT calculations through two independent means: (a) through the NMTO downfolding technique, and (b) through the

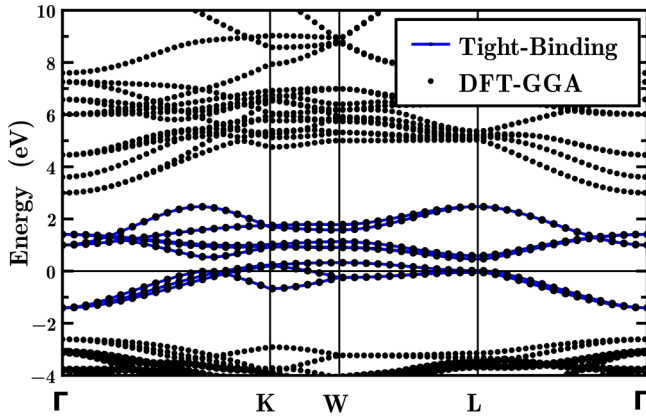


FIG. 3. Comparison of the full paramagnetic DFT band structure (solid line) and the few-orbital TB band structure (+) plotted along a high symmetry path through the Brillouin zone [Γ : (000); K : (3/8, 3/4, 3/8); W : (1/4, 3/4, 1/2); L : (1/2, 1/2, 1/2); and X : (0, 1/2, 1/2)].

construction of maximally localized Wannier functions in the basis of the effective Cr and Mo t_{2g} degrees of freedom. The latter scheme has been also employed in the DMFT study, presented before. In both cases we integrate out all other degrees of freedom except for the Cr and Mo t_{2g} 's states. The comparison of the full DFT non-spin-polarized band structure and the few-orbital TB bands in maximally localized Wannier function basis is shown in Fig. 3. The agreement between the two is found to be as good as possible, proving the effectiveness of the Wannier function projection.

The DFT estimates of Δ , $t_{\text{Cr-Mo}}$, $t_{\text{Cr-Cr}}$ and $t_{\text{Mo-Mo}}$, obtained by two different schemes of calculations, are shown in Table II. The agreement of the values in two independent schemes of calculations is remarkable and provides confidence regarding the results. The TB parameters for SCWO [43] and SFMO [42] are also listed in the table for comparison.

While the estimated hopping parameters of SCMO are rather similar to those reported for SCWO [43], we find the charge transfer energy Δ (−0.4 eV) to be quite different from that estimated for SFMO (−1.0 eV) [42] and for SCWO (−0.7 eV) [43]. This forms the most crucial observation of our study, which dictates the differences in the electronic structure of SCMO, and SFMO and SCWO. Related discussions involving the importance of Δ and t in hybridization-driven mechanisms were mentioned by another group for comparisons between SFMO and A -site B -site randomly doped Ti oxides, although it was not a systematic one [48].

TABLE II. Tight-binding parameters (in eV) of the few-orbital Hamiltonian for SCMO, SCWO [43], and SFMO [42] in a Wannier function basis, extracted out of DFT calculations.

	Δ	$t_{\text{Cr-Mo}}$	$t_{\text{Mo-Mo}}$	$t_{\text{Cr-Cr}}$
SCMO (NMTO)	−0.35	0.33	0.14	0.08
SCMO (Wannier90)	−0.42	0.30	0.12	0.06
SCWO	−0.66	0.35	0.12	0.08
SFMO	−1.04	0.26	0.11	0.04

The change in the bare, non-spin-polarized charge transfer energy (Δ), in turn, modifies the renormalized charge transfer energy Δ_R between Cr and Mo in the down spin channel in the spin-polarized phase. It is much larger in SCMO (0.81 eV) compared to that in SCWO (0.51 eV) [43], due to moving down the column of the periodic table from the $4d$ element Mo in SCMO to the $5d$ element W in SCWO. As a consequence of the larger charge transfer splitting Δ_R , the effect of the hybridization which is governed by $t_{\text{C-Mo}}^2/\Delta_R$ is significantly suppressed in SCMO, by about 63% compared to the corresponding value for SCWO.

The parameter involving J in (1) corresponds to the spin splitting at the Cr site, while J' is obtained from the enhanced intrinsic spin splitting observed at the Mo site in the case of SCMO as compared to that for the SCWO compound at the W site. As discussed below, frustration effects make the convergence of total energy calculations in different spin configurations difficult, prohibiting direct extraction of J' in a single system. Thus, J' needs to be determined by a comparison of the independently obtained intrinsic spin splittings (obtained by turning off the B - B' hybridization) at the B' sites of the related compounds. This approach was used earlier [43], where three compounds Sr₂CrWO₆, Sr₂CrReO₆, and Sr₂CrOsO₆ were considered. It was pointed out [43] that the intrinsic splitting at the B' site due to a purely hybridization-driven mechanism should increase proportionally to the electron filling at the B' site (the filling increases from 1 to 2 to 3 moving from W to Re to Os); the additional increase, if any, is due to the presence of the localized moment at the B' site giving rise to superexchange J' between the B and B' sites. In the present case, all the three compounds considered, namely, SFMO, SCMO, and SCWO, have the same electron filling, namely, 1. Hence, considering the compound SCWO as the benchmark, the difference in intrinsic splitting between the Mo site of SCMO and the W site of SCWO is used to obtain the J' value for SCMO [49].

D. Exact diagonalization of model Hamiltonian

As the magnetism of SCMO turns out to be driven by both a hybridization-driven mechanism and superexchange mechanism, the calculation of the magnetic transition temperature in a first-principles way is challenging. For example, an antiferromagnetic configuration of Cr spins leads to a frustration of the intrinsic Mo moment. On the other hand, the hybridization-driven mechanism disfavors the stabilization of magnetic configurations with Cr spins aligned in parallel to the spins at Mo sites. This thus leads to a frustration effect which makes the convergence of different spin configurations to extract the magnetic exchanges a challenge within the framework of a full blown DFT scheme.

Hence, we consider the model Hamiltonian approach in the following. To this end, we solve the *ab initio* constructed model Hamiltonian (1) using exact diagonalization on a finite lattice. In particular, we consider the stability of the ferromagnetic arrangements of Cr spins measured as the energy difference between the paramagnetic (PM) spin configuration and the ferromagnetic (FM) spin configuration. The PM phase was simulated as disordered local moment calculations, where the calculations were carried out for several (≈ 50) disordered

configurations of Cr spin and were averaged to get the energy corresponding to the PM phase. This energy difference provides an estimate of the magnetic T_c .

In order to check the influence of the intrinsic moment at the Mo site on the magnetic transition, we first carried out an exact diagonalization calculation considering the Mo site to be totally nonmagnetic, i.e., setting the last term of Hamiltonian (1) to zero. This boils down to a hybridization-only driven mechanism of magnetism as suitable for SFMO or SCWO [43,50]. The energy difference between the PM and FM in this calculation turned out to be 0.067 eV/f.u., which is of the same order as, but less than, that obtained using the TB parameters of SCWO (0.085 eV/f.u.) [43,51]. As mentioned already, the hopping parameters between SCMO and SCWO are very similar, thus the difference is caused by the different charge transfer energy in the two compounds. Mapping this energy difference to the mean field transition temperature, one would get T_c in SCMO to be 79% smaller compared to that of SCWO [17]. The calculated PM and FM energy difference, considering the full model Hamiltonian, describing both the hybridization-driven and superexchange-driven mechanism, is 0.080 eV/f.u., which makes the calculated T_c of SCMO similar to that of SCWO. Mapping the PM and FM energy difference to the mean field T_c , one obtains the values 822 K for SCMO and 870 K for SCWO, which are an overestimation of the experimental values [17–19]. This is presumably due to nonlocal fluctuations beyond the mean field. However, the trend is very well reproduced with $T_c^{\text{SCMO}}/T_c^{\text{SCWO}} = 0.95$.

IV. SUMMARY

Starting from a DFT description, we provide a microscopic analysis of the magnetic behavior of SCMO, a sister compound of SFMO. DFT calculations on pure, defect-free SCMO show that, as SFMO and SCMO, it is a half-metallic magnet. The calculated DMFT results confirm the robustness of the half metallicity of SCMO upon inclusion of the dynamical correlation effect. The DMFT results show a splitting of the Mo down spin band into Hubbard bands and a quasiparticle peak. This indicates the dual nature of the Mo electrons having both a local spin moment and itinerant behavior.

The origin of magnetism in SCMO turns out to be somewhat different than in SFMO. The charge transfer energy between Cr and Mo in the down spin channel of SCMO is found to be larger than that between Fe and Mo in SFMO. This suppresses the effect of hybridization between the B and B' sites in SCMO compared to SFMO and has two consequences: (i) The hybridization-driven mechanism for magnetism is reduced and (ii) a small but finite intrinsic moment develops at the Mo sites. The latter gives rise to a partial localized character of the Mo electrons, which was absent in SFMO. This in turn opens up a superexchange contribution to magnetism in SCMO, which was absent for SFMO. We compare our results on SCMO to another Cr-based double perovskite, SCWO. The magnetism in the latter, as in SFMO, is well described by a hybridization-only picture. The computed T_c obtained through exact diagonalization of the *ab initio* derived model Hamiltonian shows the T_c of SCMO to be similarly high as in SCWO, only upon inclusion of both superexchange and hybridization-driven contributions in the case of SCMO. Thus, while magnetism in both SCWO and SFMO is governed by hybridization, the story in SCMO appears with a twist. It needs to be described by a combination of hybridization and superexchange mechanisms.

We conclude that, in general, the magnetism in the double perovskite family has to be understood as an interplay between the hybridization and superexchange between the B and B' sites. The relative contribution of one mechanism versus the other one is dictated by the charge transfer between the $3d$ transition metal at the B site and the $4d$ or $5d$ transition metal at the B' site.

ACKNOWLEDGMENTS

P.S. acknowledges the SRIC Project No. PHY/FIG/100625. A.H. and T.S.D. acknowledge support from DST through thematic Unit of Excellence on Computational Materials Science, and L.S. and K.H. from the European Research Council under the European Union's Seventh Framework Program (FP/2007-2013)/ERC through Grant Agreement No. 306447 and the Austrian Science Fund (FWF) through the Doctoral School W1243 Solids4Fun (Building Solids for Function). We are grateful to Tulika Moitra, P. Gunacker, and D. D. Sarma for valuable discussions.

-
- [1] M. Azuma, K. Takata, T. Saito, S. Ishiwata, Y. Shimakawa, and M. Takano, *J. Am. Chem. Soc.* **127**, 8889 (2005).
 - [2] V. S. Zapf, B. G. Ueland, M. Laver, M. Lonsky, M. Pohlitz, J. Müller, T. Lancaster, J. S. Möller, S. J. Blundell, J. Singleton, J. Mira, S. Yañez-Vilar, and M. A. Señaris-Rodríguez, *Phys. Rev. B* **93**, 134431 (2016).
 - [3] H. Das, U. V. Waghmare, T. Saha-Dasgupta, and D. D. Sarma, *Phys. Rev. Lett.* **100**, 186402 (2008).
 - [4] S. Baidya and T. Saha-Dasgupta, *Phys. Rev. B* **84**, 035131 (2011).
 - [5] S. Kanungo, B. Yan, C. Felser, and M. Jansen, *Phys. Rev. B* **93**, 161116(R) (2016); M. P. Ghimire, L.-H. Wu, and X. Hu, *ibid.* **93**, 134421 (2016); S. M. Disseler, J. W. Lynn, R. F. Jardim, M. S. Torikachvili, and E. Granado, *ibid.* **93**, 140407(R) (2016); T. Koga, N. Kurita, M. Avdeev, S. Danilkin, T. J. Sato, and H. Tanaka, *ibid.* **93**, 054426 (2016).
 - [6] D. Yang, R. J. Harrison, J. A. Schiemer, G. I. Lampronti, X. Liu, F. Zhang, H. Ding, Y. Liu, and M. A. Carpenter, *Phys. Rev. B* **93**, 024101 (2016).
 - [7] T. Saitoh, M. Nakatake, A. Kakizaki, H. Nakajima, O. Morimoto, Sh. Xu, Y. Moritomo, N. Hamada, and Y. Aiura, *Phys. Rev. B* **66**, 035112 (2002).
 - [8] T. Saitoh, M. Nakatake, H. Nakajima, O. Morimoto, A. Kakizaki, S. Xu, Y. Moritomo, N. Hamada, and Y. Aiura, *J. Electron Spectrosc. Relat. Phenom.* **144-147**, 601 (2005).
 - [9] Y. Tomioka, T. Okuda, Y. Okimoto, R. Kumai, K.-I. Kobayashi, and Y. Tokura, *Phys. Rev. B* **61**, 422 (2000).
 - [10] D. D. Sarma, P. Mahadevan, T. Saha-Dasgupta, S. Ray, and A. Kumar, *Phys. Rev. Lett.* **85**, 2549 (2000).

- [11] B. García-Landa, C. Ritter, M. R. Ibarra, J. Blasco, P. A. Algarabel, R. Mahendiran, and J. García, *Solid State Commun.* **110**, 435 (1999).
- [12] B. Martínez, J. Navarro, L. Balcells, and J. Fontcuberta, *J. Phys.: Condens. Matter* **12**, 10515 (2000).
- [13] D. D. Sarma, S. Ray, K. Tanaka, M. Kobayashi, A. Fujimori, P. Sanyal, H. R. Krishnamurthy, and C. Dasgupta, *Phys. Rev. Lett.* **98**, 157205 (2007).
- [14] T. Saitoh, M. Nakatake, H. Nakajima, O. Morimoto, A. Kakizaki, S. Xu, Y. Moritomo, N. Hamada, and Y. Aiura, *Phys. Rev. B* **72**, 045107 (2005).
- [15] K.-I. Kobayashi, T. Kimura, H. Sawada, K. Terakura, and Y. Tokura, *Nature (London)* **395**, 677 (1998).
- [16] D. D. Sarma, *Curr. Opin. Solid State Mater. Sci.* **5**, 261 (2001).
- [17] J. B. Philipp, P. Majewski, L. Alff, A. Erb, R. Gross, T. Graf, M. S. Brandt, J. Simon, T. Walther, W. Mader, D. Topwal, and D. D. Sarma, *Phys. Rev. B* **68**, 144431 (2003); H. Kato, T. Okuda, Y. Okimoto, Y. Tomioka, Y. Takenoya, A. Ohkubo, M. Kawasaki, and Y. Tokura, *Appl. Phys. Lett.* **81**, 328 (2002); Y. Krockenberger, K. Mogare, M. Reehuis, M. Tovar, M. Jansen, G. Vaitheeswaran, V. Kanchana, F. Bultmark, A. Delin, F. Wilhelm, A. Rogalev, A. Winkler, and L. Alff, *Phys. Rev. B* **75**, 020404 (2007).
- [18] F. K. Patterson, C. W. Moeller, and R. Ward, *Inorg. Chem.* **2**, 196 (1963).
- [19] A. Arulraj, K. Ramesha, J. Gopalakrishnan, and C. N. R. Rao, *J. Solid State Chem.* **155**, 233 (2000).
- [20] Y. Moritomo, S. Xu, A. Machida, T. Akimoto, E. Nishibori, M. Takata, and M. Sakata, *Phys. Rev. B* **61**, R7827 (2000).
- [21] T. S. Chan, R. S. Liu, G. Y. Guo, S. F. Hu, J. G. Lin, J.-F. Lee, L.-Y. Jang, C.-R. Chang, and C. Y. Huang, *Solid State Commun.* **131**, 531 (2004).
- [22] C. M. Bonilla, D. A. Landínez Téllez, J. Arbey Rodríguez, E. Vera López, and J. Roa-Rojas, *Physica B* **398**, 208 (2007).
- [23] H. Wu, *Phys. Rev. B* **64**, 125126 (2001).
- [24] G. D. Ngantso, A. Benyoussef, A. El Kenz, and S. Naji, *J. Supercond. Novel Magn.* **28**, 2589 (2015).
- [25] G. Kresse and J. Hafner, *Phys. Rev. B* **47**, 558(R) (1993); G. Kresse and J. Furthmüller, *ibid.* **54**, 11169 (1996).
- [26] J. P. Perdew, J. A. Chevary, S. H. Vosko, K. A. Jackson, M. R. Pederson, D. J. Singh, and C. Fiolhais, *Phys. Rev. B* **46**, 6671 (1992).
- [27] W. Kohn and L. J. Sham, *Phys. Rev.* **140**, A1133 (1965); R. G. Parr and W. Yang, *Density Functional Theory of Atoms and Molecules* (Oxford University Press, Oxford, UK, 1989).
- [28] J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- [29] P. E. Blöchl, *Phys. Rev. B* **50**, 17953 (1994).
- [30] O. K. Andersen and T. Saha-Dasgupta, *Phys. Rev. B* **62**, R16219 (2000).
- [31] O. K. Andersen, *Phys. Rev. B* **12**, 3060 (1975).
- [32] J. Kunes, R. Arita, P. Wissgott, A. Toschi, H. Ikeda, and K. Held, *Comput. Phys. Commun.* **181**, 1888 (2010).
- [33] A. A. Mostofi, J. R. Yates, Y.-S. Lee, I. Souza, D. Vanderbilt, and N. Marzari, *Comput. Phys. Commun.* **178**, 685 (2008).
- [34] S. P. Blaha, K. Schwartz, G. K. H. Madsen, D. Kvasnicka, and J. Luitz, *WIEN2K, An Augmented Plane Wave+ Local Orbitals Program for Calculating Crystal Properties*, edited by K. Schwarz (Technische Universität Wien, Austria, 2001).
- [35] G. Kotliar, S. Y. Savrasov, K. Haule, V. S. Oudovenko, O. Parcollet, and C. A. Marianetti, *Rev. Mod. Phys.* **78**, 865 (2006).
- [36] K. Held, *Adv. Phys.* **56**, 829 (2007).
- [37] N. Parragh, A. Toschi, K. Held, and G. Sangiovanni, *Phys. Rev. B* **86**, 155158 (2012); M. Wallerberger (unpublished).
- [38] C. Taranto, M. Kaltak, N. Parragh, G. Sangiovanni, G. Kresse, A. Toschi, and K. Held, *Phys. Rev. B* **88**, 165119 (2013).
- [39] L. Si, Z. Zhong, J. M. Tomczak, and K. Held, *Phys. Rev. B* **92**, 041108(R) (2015).
- [40] E. Gull, A. J. Mills, A. I. Lichtenstein, A. N. Rubtsov, M. Troyer, and P. Werner, *Rev. Mod. Phys.* **83**, 349 (2011).
- [41] Q. F. Li, X. F. Zhu, and L. F. Chen, *J. Phys.: Condens. Matter* **20**, 255230 (2008).
- [42] P. Sanyal, H. Das, and T. Saha-Dasgupta, *Phys. Rev. B* **80**, 224412 (2009).
- [43] H. Das, P. Sanyal, T. Saha-Dasgupta, and D. D. Sarma, *Phys. Rev. B* **83**, 104418 (2011).
- [44] J. Blasco, C. Ritter, L. Morellón, P. A. Algarabel, J. M. De Teresa, D. Serrate, J. García, and R. M. Ibarra, *Solid State Sci.* **4**, 651 (2002).
- [45] The sharp spikes in the Cr t_{2g} up spin channel above the Fermi level might well be an artifact of the maximum entropy method employed for the analytical continuation.
- [46] J. Kanamori and K. Terakura, *J. Phys. Soc. Jpn.* **70**, 1433 (2001).
- [47] K. Samanta, P. Sanyal, and T. Saha-Dasgupta, *Sci. Rep.* **5**, 15010 (2015).
- [48] H. Iwasawa, K. Yamakawa, T. Saitoh, J. Inaba, T. Katsufuji, M. Higashiguchi, K. Shimada, H. Namatame, and M. Taniguchi, *Phys. Rev. Lett.* **96**, 067203 (2006); H. Iwasawa, S. Kaneyoshi, K. Kurahashi, T. Saitoh, I. Hase, T. Katsufuji, K. Shimada, H. Namatame, and M. Taniguchi, *Phys. Rev. B* **80**, 125122 (2009).
- [49] We have also repeated the NMTO calculation to obtain the energy level diagram and intrinsic spin splitting at the B' site, by carrying out calculations of SCWO with the same crystal structure as SCMO, constructed by replacing Mo by W in the SCMO lattice. A comparison of the intrinsic spin splitting observed at the Mo site in the case of SCMO with that at the W site of this hypothetical SCWO compound captures solely the chemical effect, eliminating any possible lattice effect. The obtained result shows the intrinsic spin splitting at the W site of the hypothetical SCWO compound to be 0.05 eV, in good agreement with the value of 0.06 eV obtained for the SCWO [43] in its own crystal structure. This proves the dominance of the chemical effect (W vs Mo) compared to the lattice effect.
- [50] O. Navarro, E. Carvajal, B. Aguilar, and M. Avignon, *Physica B* **384**, 110 (2006); L. Brey, M. J. Calderón, S. Das Sarma, and F. Guinea, *Phys. Rev. B* **74**, 094429 (2006); J. L. Alonso, L. A. Fernandez, F. Guinea, F. Lesmes, and V. Martin-Mayor, *ibid.* **67**, 214423 (2003).
- [51] The energy differences between the PM and FM phases for different lattice sizes are found to scale with size, although the ratio between the values for SCMO and SCWO is found to remain almost the same in calculations on different lattice sizes. These values are found to be 0.050 eV/f.u. for SCMO, 0.063 eV/f.u. for SCWO with a ratio of 79.2% for a $4 \times 4 \times 4$ lattice, 0.066 eV/f.u. for SCMO, 0.084 eV/f.u. for SCWO with a ratio of 78.5% for a $6 \times 6 \times 6$ lattice, and 0.067 eV/f.u. for SCMO, 0.085 eV/f.u. for SCWO with a ratio of 78.8% for an $8 \times 8 \times 8$ lattice. The values presented in the text are for $8 \times 8 \times 8$.