Lorentz microscopy and small-angle electron diffraction study of magnetic textures in $La_{1-x}Sr_xMnO_3$ (0.15 < x < 0.30): The role of magnetic anisotropy

A. Kotani, H. Nakajima, K. Harada, Y. Ishii, and S. Mori^{1,*}

¹Department of Materials Science, Osaka Prefecture University, Sakai, Osaka 599-8531, Japan ²Center for Emergent Material Science, the Institute of Physical and Chemical Research (RIKEN), Hatoyama, Saitama 350-0395, Japan (Received 21 December 2015; revised manuscript received 15 June 2016; published 6 July 2016)

Magnetic textures in the ferromagnetic phases of $La_{1-x}Sr_xMnO_3$ for 0.15 < x < 0.30 have been investigated by Lorentz microscopy combined with small-angle electron diffraction experiments. Various types of magnetic textures characterized by stripe, plate-shaped, and cylindrical (magnetic bubble) domains were found. Two distinct types of magnetic stripe domains appeared in the orthorhombic structure with an inversion symmetry of $La_{0.825}Sr_{0.175}MnO_3$, depending significantly on magnetocrystalline anisotropy. Based on *in situ* observations as functions of temperature and the strength of the external magnetic field, a magnetic field-temperature phase diagram was constructed, showing the stabilization of magnetic bubbles in the ferromagnetic phase of $La_{0.825}Sr_{0.175}MnO_3$.

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I. INTRODUCTION

A number of intricate magnetic textures such as magnetic bubbles and periodic magnetic stripes, which are induced by magnetic anisotropy and the uniform ferromagnetic order due to the long-range magnetic dipole interaction, have been studied in magnetic materials [1–5]. For example, magnetic bubbles were generated by applying a perpendicular magnetic field to periodic magnetic stripe domains in thin films such as hexagonal barium ferrites and cobalt thin films [1,6–8]. In addition, when a magnetic field was applied normal to thin films of some manganese oxides such as La_{0.875}Sr_{0.125}MnO₃ and Ru-doped La_{1.2}Sr_{1.8}Mn₂O₇, periodic magnetic stripe domains in the ferromagnetic phase turned into magnetic bubbles, which were observed by Lorentz microscopy (LM) [9–11].

Manganese oxides with the perovskite structure exhibited a large number of ground states and some anomalous phase transitions such as the magnetic-field-induced insulator-tometal transition [12]. As shown in Fig. 1(a), $La_{1-x}Sr_xMnO_3$ exhibited a variety of ground states in the electronic and magnetic phase diagram, which depended strongly upon the Sr concentration (x) [13,14]. The phase diagram was mainly divided into two distinct ground states: the ferromagnetic insulating phase with a monoclinic structure [space group (SG): $P2_1/c$] in x < 0.15 and the orthorhombic ferromagnetic metallic (FMM) phase (SG: Pbnm) in x > 0.15. Note that $P2_1/c$ and Pbnm have inversion symmetries. Our previous work on LM observation of the magnetic domain structures revealed that the orthorhombic ferromagnetic phase of $La_{1-x}Sr_xMnO_3$ for x = 0.175 was characterized by nanoscale magnetic stripe domains with antiparallel magnetic moments in adjacent stripe domains [15]. Conversely, in the rhombohedral ferromagnetic phase (SG: $R\bar{3}c$), the macroscopic magnetic plate-shaped domains were stabilized. Note that La_{0.825}Sr_{0.175}MnO₃ exhibited a rhombohedral-to-orthorhombic transition at approximately 185 K during cooling, accompanying a drastic change in the orientation of the magnetic easy axis from the [111] to the [001] directions.

Recently we found that magnetic domains in La_{0.825}Sr_{0.175}MnO₃ depend significantly on the relationship between the magnetic easy axis and the direction of the applied magnetic field. Moreover, La_{0.825}Sr_{0.175}MnO₃ with an inversion symmetry showed magnetic cylindrical domains (magnetic bubbles) under the application of the magnetic fields normal to the thin films [16]. Although a large number of magnetic textures such as magnetic skyrmions in magnets without inversion symmetry through the relativistic Dzyaloshinskii-Moriya (DM) interaction have been studied to date [17–27], few studies of magnetic textures in magnets with inversion symmetry have been performed [10]. In this work, we systematically investigate magnetic textures and their changes as functions of temperature and the strength of the external magnetic fields perpendicular to thin films in the FMM phase of $La_{1-x}Sr_xMnO_3$ for 0.15 < x < 0.30with inversion symmetry using LM and small-angle electron diffraction (SmAED) experiments.

II. EXPERIMENTAL METHODS

Single crystals of $La_{1-x}Sr_xMnO_3$ for x = 0.15, 0.175, 0.20, 0.25, and 0.30 were grown by the floating zone method. The crystal structures and orientations in the obtained single crystals were examined by powder x-ray diffraction and backplate Laue-type x-ray diffraction experiments, respectively. Thin films for LM observation were obtained by grinding with alumina powder and, subsequently, by Ar⁺-ion sputtering at room temperature. In situ LM experiments were conducted in order to clarify magnetic textures and their changes by applying magnetic fields perpendicular to the thin films. Note that an external magnetic field was applied to the thin films by exciting the objective lens of the conventional transmission electron microscope (TEM) (JEM-2100F and JEM-2010). The SmAED experiments were performed at an angular resolution of $10^{-4} \sim 10^{-6}$ rad [28–30]. For simplicity, the indices of the crystal planes and electron diffraction spots were represented on the basis of the cubic perovskite structure with

^{*}Corresponding author: mori@mtr.osakafu-u.ac.jp

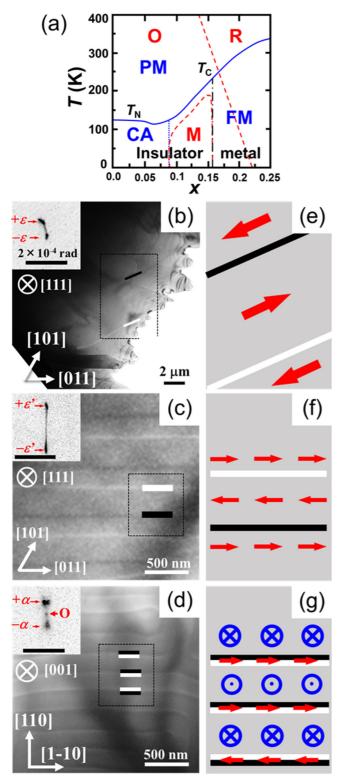


FIG. 1. (a) Electronic phase diagram of $La_{1-x}Sr_xMnO_3$, in which O, R, and M denote the orthorhombic, rhombohedral, and monoclinic crystal systems and PM, CA, and FM denote the paramagnetic, canted antiferromagnetic, and ferromagnetic phases, respectively [13]. (b) Ferromagnetic domain structures in the (111) crystal plane of the rhombohedral structure of $La_{0.8}Sr_{0.2}MnO_3$ at room temperature. (c), (d) Fresnel images showing magnetic stripe domains in (c) the (111) and (d) the (001) planes of the orthorhombic structure of $La_{0.825}Sr_{0.175}MnO_3$ at 100 K. In the insets of (b)–(d), SmAED

 $Pm\bar{3}m$. The in-plane magnetizations of the magnetic textures were analyzed with the aid of a phase retrieval technique based on the transport-of-intensity equation (TIE) [31,32]. Magnetization curves were measured by using a vibrating sample magnetometer (VSM) equipped with Quantum Design Physical Property Measurement System (PPMS).

III. RESULTS

Magnetic textures in the FMM phase of La_{1-x}Sr_xMnO₃ were investigated by obtaining LM (Fresnel) images and SmAED patterns. Note that LM images are obtained by imaging the intensity distribution at a distance Δf below or above the specimen by under- or overfocusing, respectively [33]. Figure 1(b) is an LM image ($\Delta f = -1000 \,\mathrm{nm}$) of La_{0.8}Sr_{0.2}MnO₃ at 297 K in the (111) crystal plane of the rhombohedral $R\bar{3}c$ structure, in which the magnetic easy axis is almost parallel to the (111) direction. A magnetic domain structure characterized as a macroscopic magnetic domain with a width of approximately 5 μ m appears. As understood in the SmAED pattern of the inset of Fig. 1(b), the primary beam splits into two spots with an angular separation of $\pm \varepsilon$, and there is a diffuse streak between the two split spots. Note that the magnitude of ε corresponds to approximately 50 μ rad. This implies that the macroscopic magnetic domain structure in Fig. 1(b) comprises 180° magnetic domains with Bloch-type magnetic domain walls because of large magnetic shape anisotropy in the rhombohedral structure of La_{0.8}Sr_{0.2}MnO₃. Note that one-to-one correspondence between the twin structures due to the rhombohedral distortion and the magnetic domain wall can be seen, as shown in Fig. S1 of the Supplemental Material [34]. The details of the relationship between the twin structures and the magnetic domain wall are described in the Supplemental Material [34]. Conversely, magnetic textures in various crystal planes of the FMM phase with an orthorhombic structure (Pbnm) of La_{0.825}Sr_{0.175}MnO₃ were examined at 100 K. Note that the magnetic easy axis is parallel to the [001] direction in the orthorhombic structure. Figures 1(c) and 1(d) exhibit LM images with a defocused value of $\Delta f = -300$ nm, exhibiting magnetic textures obtained in the (111) and (001) crystal planes at 100 K, respectively. Magnetic stripe domains comprise regions separated by domain walls that appear as an alternating arrangement of straight lines with bright and dark contrast in the (111) crystal plane, as shown in Fig. 1(c). In the SmAED pattern, there are two split spots indicated by $\pm \varepsilon'$, and there is a diffuse streak between them. From the analysis of the SmAED pattern, magnetic stripe domains are characterized as the 180° magnetic domain structure having the Bloch-type magnetic domain walls. Conversely, as shown in Fig. 1(d), magnetic stripe domains were found in the (001)

patterns are shown. (e), (f) Schematic depictions of the magnetic domains with the in-plane antiparallel magnetization in the (111) plane of (e) the rhombohedral and (f) the orthorhombic structures, respectively. (g) Schematic description of the magnetic domains with magnetic moments normal to the (001) plane in the orthorhombic structure. Red arrows represent the directions of magnetization in each magnetic domain. The defocused values in (b), (c), and (d) are $\Delta f = -1000\,\mathrm{nm}, -300\,\mathrm{nm},$ and $-300\,\mathrm{nm},$ respectively.

crystal planes, which are perpendicular to the magnetic easy axis along the [001] direction. In the LM image, alternative bright and dark lines can be seen. Unlike the SmAED patterns of Figs. 1(b) and 1(c), the SmAED pattern in Fig. 1(d) shows the presence of the primary spot in the reciprocal space, as indicated by arrow O, in addition to two split spots ($+\alpha$ and $-\alpha$) due to magnetic deflection by the magnetic moment in the sample. The presence of the primary spot implies that there is a magnetic component perpendicular to the (001) crystal plane. Based on the analysis of both SmAED patterns and LM images, the spatial distributions of magnetization in each domain are illustrated schematically in Figs. 1(e)-1(g). The significant point of difference between the three distinct types of magnetic domain structures in Figs. 1(b)-1(d) is that there exists magnetic moment components perpendicular to thin films only in the magnetic domain structure of the (001) crystal plane of La_{0.825}Sr_{0.175}MnO₃.

Changes in the magnetic domains due to the application of an external magnetic field were examined by in situ LM observation. Figure 2 shows the magnetic field-dependence of a typical 180° ferromagnetic domain structure in the (111) crystal plane of the rhombohedral structure in La_{0.8}Sr_{0.2}MnO₃. When an external magnetic field is applied perpendicular to the (111) plane, magnetization in the magnetic domains is gradually saturated by the Zeeman effect, and magnetic contrasts almost disappear in the field at 80 mT. In contrast, it is found that magnetic stripe domains in the (111) crystal plane of the orthorhombic structure of La_{0.825}Sr_{0.175}MnO₃ changed in a different manner from those of La_{0.8}Sr_{0.2}MnO₃. Changes of magnetic stripe domains in the orthorhombic structure of La_{0.825}Sr_{0.175}MnO₃ as functions of the strengths of external magnetic fields perpendicular to the thin films are shown in Fig. 3. At $B = 34 \,\mathrm{mT}$, the magnetic stripe domains were pinched off, and elliptical isolated magnetic domains appeared at their edges, as shown in the square region of Fig. 3(b). Note that the magnified LM image and the schematic illustration of the elliptical isolated magnetic domain are displayed in Fig. 3(d), and it is found that this domain is similar to the ones observed in the ferromagnetic phase of La_{0.875}Sr_{0.125}MnO₃ [9]. As the strength of the magnetic fields increased, the magnetic stripe and elliptical isolated magnetic domains finally disappeared at $B = 100 \,\mathrm{mT}$ in Fig. 3(c). As the strength of the magnetic field decreased from $B = 100 \,\mathrm{mT}$, the magnetic stripe domains appeared reversibly around B =50 mT with thermal hysteresis.

As shown in Figs. 1(c) and 1(d), two distinct types of magnetic stripe domains appeared in the (111) and (001) planes of the orthorhombic structure of La_{0.825}Sr_{0.175}MnO₃. The first were the 180° magnetic domains with in-plane magnetizations antiparallel to those of adjacent domains, and the second were the magnetic stripe domains with magnetizations perpendicular to the (001) crystal planes. Thus, we applied magnetic fields perpendicular to two distinct types of magnetic stripe domains and investigated their changes as functions of the strength of the external magnetic fields. Unlike the changes observed in the magnetic stripe domains in the (111) plane under the application of an external magnetic field, as shown in Fig. 3, we found the appearances of type-I and -II magnetic bubbles clockwise (CW) and counterclockwise (CCW) spin rotations by applying magnetic

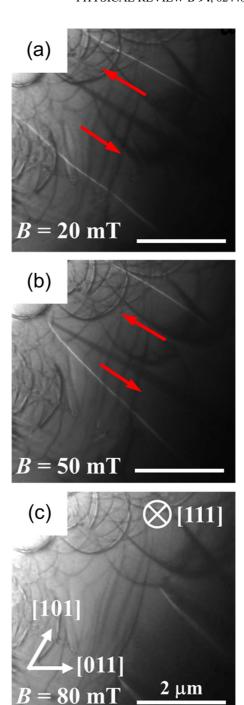


FIG. 2. Changes of magnetic stripe domains in the (111) crystal plane as a function of the strength of magnetic field perpendicular to the thin film of $La_{0.8}Sr_{0.2}MnO_3$. The strengths of the magnetic field were (a) 20 mT, (b) 50 mT, and (c) 80 mT. Red arrows represent the directions of the magnetic moments in each magnetic domain.

fields perpendicular to the magnetic stripe domains in the (001) crystal planes [16]. Note that only the stripe domains in the (001) plane changed into the bubble-type magnetic domains by applying external magenta field perpendicular to the thin film. The transition process from the magnetic stripe domains to type-I and -II magnetic bubbles have been reported in our recent work [16]. Note that the type-I and -II

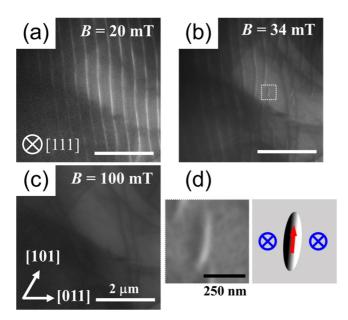


FIG. 3. Changes of magnetic stripe domains in the (111) crystal plane as a function of the strength of the magnetic field perpendicular to the thin film of $La_{0.825}Sr_{0.175}MnO_3$. The strengths of the magnetic field were (a) 20 mT, (b) 34 mT and (c) 100 mT. (d) Magnified ellipsoidal magnetic domain obtained from the square region of (c) and schematic illustration of the ellipsoidal magnetic domain. The red arrow shows the direction of the magnetic moment inside the ellipsoidal magnetic bubble and the magnetic moment is perpendicular to thin film outside.

magnetic bubbles are in accordance with the definition in Ref. [3], and they are illustrated in Fig. 4(a). In order to confirm the spin configurations of type-I and -II magnetic bubbles found in the ferromagnetic phase of La_{0.825}Sr_{0.175}MnO₃, we analyzed the LM images showing type-I and -II magnetic bubbles with the aid of the so-called phase retrieval TIE method. Figure 4 represents the LM images taken at 80 K in the orthorhombic structure of La_{0.825}Sr_{0.175}MnO₃ under a perpendicular magnetic field of approximately 570 mT. The images show three kinds of magnetic domain patterns. Under a zero field, the magnetic stripe domains were observed. As the magnetic field was applied perpendicular to the thin film, the transition to the magnetic bubbles took place in a weak magnetic field of 400 mT. Three kinds of magnetic bubbles appeared at approximately 570 mT, as shown in Fig. 4(b). Note that the LM image in Fig. 4(b) was obtained at a defocused value of $\Delta f = -0.2 \,\mathrm{mm}$. The spatial distribution of type-I magnetic bubbles with dark and bright contrast at the center stemmed from the CW and CCW spin rotations of their domain walls. This randomness of the chirality of the magnetic bubbles contrasts with the single-chirality in DM helimagnets [18,19]. Type-II magnetic bubbles also exhibit random chirality with CW and CCW spin rotations. The LM image in Fig. 4(c) is obtained at the defocused value of $\Delta f = +0.2$ mm. The dark and bright contrast of the magnetic bubbles was completely reversed, as understood by comparing Figs. 4(b) and 4(c). We analyzed the LM image showing the magnetic bubbles [Figs. 4(b) and 4(c)] using the TIE analysis method, which reproduced the orientation and magnitude of the in-plane

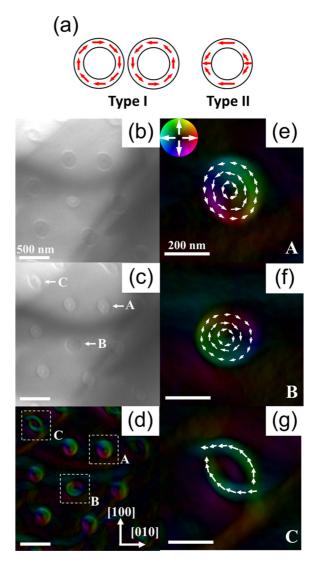


FIG. 4. (a) Schematic illustrations of type-I and -II magnetic bubbles. Red arrows represent the direction of the magnetic moment. (b), (c) Magnetic bubbles generated by applying external magnetic fields perpendicularly to a thin film of La_{0.825}Sr_{0.175}MnO₃ in the (001) crystal plane. The strength of the applied magnetic field is approximately 570 mT. The images were taken at the defocused values of (b) $\Delta f = -0.2$ mm and (c) $\Delta f = +0.2$ mm. (c) Type-I magnetic bubbles with left-handed and right-handed spin rotations and the type-II magnetic bubbles are indicated by capital letters A, B, and C, respectively. (d)–(g) Color representations of the in-plane magnetization calculated with the aid of the TIE analysis method. The directions of the magnetization are (e) type-I with left-handed spin rotation, (f) type-I with right-handed spin rotation, and (g) type-II magnetic bubbles, which were indicated by color mapping and vectorial representations by white arrows.

magnetization [32,33,35–37]. Note that the color mapping in the inset of Fig. 4(e) exhibited the direction of the in-plane magnetization. The directions of the in-plane magnetizations of type-I and -II magnetic bubbles are visualized in Fig. 4(d). The black color at the centers of type-I magnetic bubbles shows that no in-plane magnetization was present, thus magnetization is assumed to be vertical to the thin film with magnetization being parallel to the [001] direction. The magnified images

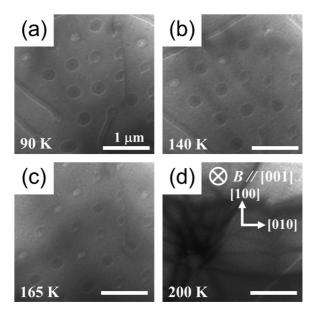


FIG. 5. Temperature variations of the magnetic bubble domains in the (001) crystal plane under a constant magnetic field of 500 mT normal to the thin film of $La_{0.825}Sr_{0.175}MnO_3$. The images were obtained at (a) 90 K, (b) 140 K, (c) 165 K, and (d) 200 K, respectively.

of type-I magnetic bubbles with left- and right-handed spin rotations and those of type-II magnetic bubbles are shown in Figs. 4(e), 4(f), and 4(g), respectively, together with color representations of the in-plane magnetizations calculated using the TIE analysis method.

To clarify the stabilizing region of the magnetic bubbles, we investigated the disappearance of magnetic bubbles as a function of the temperature and the strength of the external magnetic fields. Figure 5 shows a series of changes in the magnetic bubbles on warming from 90 K at a constant magnetic field of 500 mT. The average diameter of the magnetic bubbles was approximately 300 nm at 90 K. Upon increasing the temperature from 90 K, magnetic bubbles shrunk gradually from 300 to 200 nm at 165 K and disappeared above 185 K, as shown in Fig. 5(d). Note that a structural phase transition from rhombohedral to orthorhombic structure occurred at approximately 185 K, during which the magnetic easy axis changed from the [001] direction to the [111] direction. Conversely, when the temperature decreased below 200 K, magnetic bubbles appeared reversibly around 185 K, and the diameters of the magnetic bubbles increased to 300 nm at 90 K. These reversible changes with respect to temperature at a constant strength of $B = 500 \,\mathrm{mT}$ indicate that magnetic bubbles existed as a stable state in the definite range of the *B-T* phase diagram.

The stability of the magnetic bubbles as a function of the strength of external magnetic fields was investigated at constant temperatures of 90, 120, and 165 K in the orthorhombic phase. The average diameter (r) of the magnetic bubbles was approximately 300 nm at $T=90\,\mathrm{K}$ and $B=400\,\mathrm{mT}$. When the strengths of the external magnetic fields increased, the diameters (r) of the magnetic bubbles decreased until the magnetic bubbles disappeared at $B=720\,\mathrm{mT}$. Similar tendencies were exhibited at T=120 and 165 K, as shown in Fig. 6(a). Based on these experimental results, the B-T

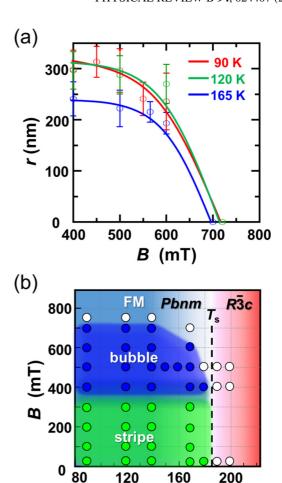


FIG. 6. (a) Variations of the average diameters (*r*) of magnetic bubbles as functions of strength of the applied magnetic fields at 90, 120, and 165 K, respectively. (b) The *B-T* phase diagram constructed on the basis of the experimental results in this study, in which circles indicate experimentally measured points.

T (K)

phase diagram that shows the stability of the magnetic bubbles is presented in Fig. 6(b). Note that the specimen thickness in the LM observation is approximately $100 \sim 150 \,\mathrm{nm}$. The range in which magnetic bubbles exist stably in the *B-T* phase diagram depends strongly upon the specimen thickness, as in the *B-T* phase diagram of the magnetic skyrmion in some helical magnets [18].

IV. DISCUSSION

Here let us briefly discuss the magnetic bubbles found in the FMM phase of $La_{1-x}Sr_xMnO_3$. Formation of magnetic bubbles depends strongly upon magnetic anisotropy. The magnetic easy axis is parallel to the [001] direction in the orthorhombic phase and to the $\langle 111 \rangle$ direction in the rhombohedral phase. The magnetic 180° domain structure with in-plane magnetization is formed in the (111) crystal plane of the rhombohedral phase, as shown in Fig. 2, although the magnetic easy axis is parallel to the incidence direction of the electron beam. This clearly implies that the magnetocrystalline anisotropy in the rhombohedral structure is relatively low and that the shape anisotropy is more dominant. As a result,

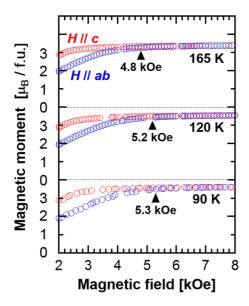


FIG. 7. Magnetization curves of $La_{0.825}Sr_{0.175}MnO_3$ at 90 K, 120 K, and 165 K. Magnetic field (H) was applied parallel (red circles) or perpendicular (blue circles) to the c axis. Each solid triangle shows the anisotropic field.

magnetic bubbles are not formed by the application of external magnetic fields perpendicular to thin films in the rhombohedral structure. Conversely, Nagai *et al.* obtained a magnetic anisotropy constant of $K_{\rm u}=2.1\times10^5\,\rm J\,m^{-3}$ in the ferromagnetic phase of La_{0.875}Sr_{0.125}MnO₃ at 100 K [9], which is larger than $K_{\rm u}\sim1.0\times10^4\,\rm J\,m^{-3}$ for the ferromagnetic phase of La_{0.7}Sr_{0.3}MnO₃ at 300 K [38]. Moreover, it is believed that this difference arises from the fact that magnetic anisotropy in the ferromagnetic phase is enhanced by orbital ordering.

We examined variations of magnetic anisotropy constants $(K_{\rm u})$ as a function of temperature in the ferromagnetic phase with the orthorhombic structure of La_{0.825}Sr_{0.175}MnO₃. Figure 7 shows changes of magnetization curves as a function of

temperature in the ferromagnetic state of La_{0.825}Sr_{0.175}MnO₃ with the orthorhombic structure. Magnetic fields were applied along the c axis and the ab plane perpendicular to the c axis. Values of saturated magnetization at $T = 90 \,\mathrm{K}$, 120 K, and 165 K are approximately 3.7, 3.6, and 3.4 μ_B , respectively. From magnetization curves at $T = 90 \,\mathrm{K}$, 120 K, and 165 K, we obtained the anisotropic magnetic field (H_k) in the ferromagnetic phase with the orthorhombic structure. Note that H_k is defined as the critical value of the magnetic field, above which the difference in magnetization between H//c and H//abis less than 2%. Values of H_k at T = 90 K, 120 K, and 165 K are 5.3, 5.2, and 4.8 kOe, respectively. The $K_{\rm u}$ values obtained based on the anisotropic magnetic field $T = 90 \,\mathrm{K}$, 120 K, and 165 K are, respectively, $1.52 \times 10^5 \, \mathrm{J \, m^{-3}}$, 1.47×10^5 , and $1.30 \times 10^5 \,\mathrm{J}\,\mathrm{m}^{-3}$, which are larger than that of rhombohedral structures such as La_{0.7}Sr_{0.3}MnO₃. Based on these considerations, it is concluded that the magnetic stripe domains in the orthorhombic structure with magnetic moments perpendicular to the thin film are transformed into magnetic bubbles by the application of external magnetic fields perpendicular to the thin film.

V. CONCLUSION

We revealed magnetic textures in the ferromagnetic phases with the inversion symmetry of $La_{1-x}Sr_xMnO_3$ for 0.15 < x < 0.30. These textures were characterized as stripes, plate-shapes, and magnetic bubbles. By applying external magnetic fields perpendicular to the (001) crystal plane of the FMM phase in $La_{0.825}Sr_{0.175}MnO_3$, the magnetic stripe domains changed into type-I and -II magnetic bubbles. Magnetic states with type-I and -II magnetic bubbles in the FMM phase exist stably in the temperature-magnetic-field phase diagram.

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