Large magnetoresistance in compensated semimetals TaAs₂ and NbAs₂

Zhujun Yuan,¹ Hong Lu,¹ Yongjie Liu,² Junfeng Wang,² and Shuang Jia^{1,3,*}

¹International Center for Quantum Materials, School of Physics, Peking University, Haidian District, Beijing 100871, China

²Wuhan National High Magnetic Field Center, Huazhong University of Science and Technology, Wuhan 430074, China

³Collaborative Innovation Center of Quantum Matter, Beijing 100871, China

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We report large magnetoresistance (MR) at low temperatures in single-crystalline nonmagnetic compounds TaAs₂ and NbAs₂. Both compounds exhibit parabolic-field-dependent MR larger than 5×10^3 in a magnetic field of 9 Tesla at 2 K. The MR starts to deviate from parabolic dependence above 10 T and intends to be saturated in 45 T for TaAs₂ at 4.2 K. The Hall resistance measurements and band structure calculations reveal their compensated semimetal characteristics. Their large MR at low temperatures is ascribed to an effect for compensation of electrons and holes with large mobilities. After discussing the MR for different samples of TaAs₂ and other semimetals, we found that the magnitudes of MR are strongly dependent on the samples' quality for different compounds.

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I. INTRODUCTION

Magnetoresistance (MR) is defined as the change of material's resistance in magnetic fields (MR $\equiv \Delta \rho / \rho_{H=0}$). Recent studies on the single-crystalline semimetals including WTe₂ [1,2], Cd₃As₂ [3], NbSb₂ [4], LaSb [5], and MPn (M = Ta and Nb, Pn = P and As) [6–13] have revealed large MR varying from 1×10^3 to 2×10^4 in a magnetic field less than 10 Tesla (T) at low temperatures. Other metallic compounds such as PtSn₄ [14] and PdCoO₂ [15] with much higher carriers' density were reported to have large MR being 5.0×10^3 and 3.5×10^2 in a magnetic field of 14 T around 2 K, respectively. The MR for these compounds was described as "large," because the values are several orders of magnitude larger than that for normal metals such as copper [16]. But compared with the MR for the prototype semimetal element bismuth, which was reported to be as large as 1.6×10^7 at 4.2 K in 5 T [17], the MR for these nonmagnetic compounds is at least two orders of magnitude less.

Several mechanisms have been proposed to explain the large MR in nonmagnetic semimetals [18,19]. A classical two-band model predicts large parabolic-field-dependent MR in a compensated semimetal. A small difference of the electrons and holes densities (*n* and *p*) will cause the MR eventually saturated to a field-independent value [1,18,20]. Most of the compounds with large MR show a power law of field dependence very close to parabolic (MR \propto H²). The compensation of electrons and holes was believed to occur in bismuth and graphite, while their MR was observed to be saturated in an intense magnetic field. On the other hand, the MR for WTe₂ and NbSb₂ is not saturated in a magnetic field as high as 60 and 32 T, respectively.

Some semimetals with large MR show the features that cannot be simply understood as a compensation effect of electrons and holes. The large MR for metallic $PtSn_4$ was reported as the result of ultrahigh mobility despite its large unbalanced electron and hole density [14]. The linear-field-

dependence of the MR for Cd₃As₂ [3], Ag₂Se [21], and the recently reported Weyl semimetal TaAs family [6,8,9,22] is not expected for the compensation effect. The linear-fielddependent MR was understood as a quantum effect near the crossing point of the conduction and valence bands, which have linear energy dispersion when the magnetic field is beyond the quantum limit [23,24]. A classical effect of the spacial mobility fluctuation causes linear MR as well, which is more significant in polycrystalline samples [19]. Based on the studies of the large MR for these nonmagnetic compounds, the mechanisms underneath seem to be different from system to system. Exploring the semimetals which show large MR but crystallize in different structures will help to understand this complexity. Unfortunately, very limited semimetal compounds were reported to enact the forms of high-quality single crystals.

TaAs₂ and NbAs₂ belong to a group of transition metal dipnictides MPn₂ (M = V, Nb, Ta, Cr, Mo, and W, Pn = P, As, and Sb), which crystallize in an OsGe₂-type structure [25,26]. This structure has one crystallographic M site and two Pn sites in one unit cell [Fig. 1(a)]. Two Pn sites are quite distinct in their coordinations: one site is isolated from each other while the other site forms separated Pn-Pn dimers. The As-As bond lengths of the dimers in TaAs₂ and NbAs₂ are 2.42 Å and 2.44 Å, respectively [26,27], which are very close to the covalent radii 2.43 Å of the As-As dimer in tetramethyldiarsane [28]. Each tantalum and niobium atom is surrounded by six As atoms forming a trigonal prism with two additional As atoms outside the rectangular faces. The prisms are stacked along the crystallographic b direction through their trigonal faces to form the structure. A naive expectation for the electronic states of NbAs₂ and TaAs₂ is based on the simple Zintl concept [29,30]: a balanced state for valence electrons of $(Nb/Ta)_2^{5+}(As-As)^{4-}As_2^{6-}$ indicates an insulating state with a gap between the bands constructed by the bonding orbitals and antibonding orbitals of the As-As dimers. However, previous band structure calculations [30] suggested that NbAs₂ and TaAs₂ are semimetals instead of semiconductors. Although the Nb-Nb or Ta-Ta bonds with the bond length close to 3 Å are not considered in this oversimple

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^{*}gwljiashuang@pku.edu.cn



FIG. 1. (a) Crystal structure for Ta(Nb)As₂. (b) Photo of a single crystal of NbAs₂ on millimeter-grid paper. (c) Powder XRD spectrum for TaAs₂. The refinement results are close to those in Refs. [26] and [27]. (d) Laue backscattering diffraction pattern for a single crystal of NbAs₂.

Zintl concept, this rough estimation predicts the destitute density of states (DOS) for TaAs₂ and NbAs₂ near the Fermi level.

Physical properties for the MPn₂ family have not been extensively studied. MoAs₂ and WAs₂ were reported to be superconductors with the superconducting transition temperature (T_c) less than 1 K [30]. The semimetal behaviors and large, parabolic field-dependent MR at low temperatures in NbSb₂ were reported very recently [4]. The Sb atoms in NbSb₂ form the dimers with the Sb-Sb bond length of 2.71 Å as well. The band structure calculations for NbSb₂ show compensated, small electron and hole pockets near the Fermi level [4]. A electron hole compensation in NbSb₂ was suggested to be the reason for the large MR at low temperatures [4].

Due to highly volatile and toxic arsenic, single crystals of the transition metal diarsenides cannot be easily obtained via a regular flux growth as described in Refs. [4] and [2]. In this paper, we show that large single crystals of NbAs₂ and TaAs₂ can be grown via a chemical vapor transport (CVT) process. We found that these two compounds have large, parabolic-field-dependent MR at low temperatures. Based on the analysis of the electrical properties and band structure calculations, we believe that their large MR are due to an electron-hole compensation. After an analysis of the semimetals with large MR in previous reports, we found that



FIG. 2. (a) Measured "longitudinal MR" for the sample T1 of $TaAs_2$ when the contacts do NOT fully cross the sample as illustrated in the sketch. (b) Measured longitudinal MR for the same sample when the contacts fully cross the sample as illustrated in the sketch.



FIG. 3. (a) Temperature-dependent resistivity for different samples of TaAs₂ and NbAs₂ in a semilogarithmic plot (N for NbAs₂ and T for TaAs₂). Inset shows the $R(T)/R_{300K}$ versus T. (b) A double-logarithmic plot for the resistance and temperature. Inset shows the Resistivity versus T³ for N9. (c, d) Temperature-dependent transversal resistivity for the samples T4 and N5 in different magnetic fields, respectively.

the values of MR are strongly dependent on the samples' quality. We believe that most of the large parabolic-field-dependent MR for compensated semimetals can be explained as a result of electron-hole balance.

II. EXPERIMENT

Single crystals of TaAs₂ and NbAs₂ were found as byproducts when we tried to grow the single crystals of TaAs and NbAs via a CVT reaction method. Then we optimized the growth conditions in the previous report [27] for high-quality, large single crystals. Polycrystalline TaAs₂ and NbAs₂ was prepared by heating the stoichiometric amounts of elemental Ta/Nb and high-purity arsenic in evacuated silica ampules at 750°C for 2 days. Some redundant arsenic grains were always observed in the ampules after the synthesis. This phenomenon is consistent with the observation for the vacancy of arsenic atoms in TaAs₂ and NbAs₂ as described in Refs. [26] and [27]. The crystals of TaAs₂ were grown in a temperature gradient from 850°C (source) to 750°C (sink) for 7 days with the concentration of the agent TaBr₅ being 2.7mg/cm³, while the crystals of NbAs₂ were grown in a temperature gradient from 775°C (source) to 700°C (sink) for 7 days with the concentration of the agent NbI₅ being 2.7 mg/cm³. Blade-like crystals yielded in these conditions have a length of about 3–7 mm and a width of about 0.6 mm [Fig. 1(b)], while their residual resistivity ratio (RRR = ρ_{300K}/ρ_{2K}) is larger than 100 in transport measurements. CVT growth from the polycrystalline TaAs and NbAs as the sources instead of TaAs₂ and NbAs₂ and TeCl₄ (1.5 mg/cm³) or TeI₄ (1.9 mg/cm³) as the agent also yielded needle-like single crystals of TaAs₂ and NbAs₂ but their RRR is much smaller.

Powder x-ray diffraction (XRD) measurements were performed in a Rigaku MiniFlex 600 diffractometer. The



FIG. 4. (a, b) The MR versus magnetic field at different temperatures for the sample T4 and N5, respectively. Insets shows the MR at high temperatures. (c, d) The MR in double logarithmic plots for T4 and N5, respectively. MR $\propto H^m$ at low temperatures where m = 1.7 and 1.9 for T4 and N5, respectively.

refinements by using the Rietica Rietveld program confirm that the compounds crystallize in the structures of the space group C2/m, which is the same as previously reported [26]. The refined parameters are a = 9.3372(2) Å, b = 3.3847(1) Å, c = 7.7711(2) Å, and $\beta = 119.638(1)^{\circ}$ for TaAs₂ and a =9.4017(2) Å, b = 3.3947(1) Å, c = 7.8212(2) Å, and $\beta =$ 119.531(1)° for NbAs₂ [Fig. 1(c)]. Laue XRD measurements [Fig. 1(d)] on the single crystals were collected in a Photonic Science PSL-Laue backscattering system. The pattern shows that the long direction of the crystals is in the b direction, while the largest flat plane in the blade-like crystals is perpendicular to the (001) direction [Fig. 1(d)]. The transversal resistance and the Hall resistance were measured with a standard fourprobe ac apparatus [31] in which the current is parallel to the *b* axis and the magnetic field is parallel to the (001)direction. The resistance and Hall data have been symmetrized to remove the effect of the misalignment of voltage wires $(\rho_{xx}(H) = \frac{\rho_{xx}(H) + \rho_{xx}(-H)}{2}, \rho_{yx}(H) = \frac{\rho_{yx}(H) - \rho_{yx}(-H)}{2})$. All the physical property characterizations were performed in a Quantum Design physical property measurement system (PPMS-9) over the temperature range from 1.9 to 300 K and in the magnetic fields up to 9 T.

Band structure calculations were performed by using the WIEN2K package [32] based on the full potential linearized augmented plane wave (LAPW) method [33] within local density approximation (LDA) [34]. The lattice constants were chosen from Ref. [26]. The values of the smallest atomic sphere radii in the unit cell (R_{mt}) were set to 2.48 and 2.22 Bohr for Ta and As in TaAs₂ and 2.43 and 2.20 Bohr for Nb and As in NbAs₂ by considering the difference of their nearest neighbor distances. The general gradient approximation (GGA) proposed by Perdew *et al.* [35] was used for the



FIG. 5. MR for $TaAs_2$ and $NbAs_2$ in a pulsed magnetic field at 4.2 K.

exchange-correlation potential. A mesh of $21 \times 21 \times 21 \times 21$ k points was selected for Brillouin zone integration with $R_{mt} * K_{max} = 7.0$ and Fermi surface plot, where K_{max} is the magnitude of the largest reciprocal lattice vector (K).

III. RESULTS

Before starting to describe the transport properties for $TaAs_2$ and $NbAs_2$, we present a misleading measurement result due to an improper experimental setup. Figures 2(a) and 2(b) show the measured resistivity for a same sample T1 with differ-

ent contact setups as the insets, respectively, when the magnetic field is parallel with the current $[H \parallel I]$. The contacts attached the edge of the sample in the first setup, which is sometimes used for measuring Hall signal and transversal resistance at the same time. As shown in Fig. 2(a), a very large negative "longitudinal MR" occurs in the measurements for this setup. However, when the contacts fully cross the width in the second setup, the MR becomes a regular, small, positive profile [Fig. 2(b)]. We believe that the negative "longitudinal MR" is due to a magnetic-field-induced current jetting effect, which is commonly observed in high-mobility samples [16,36,37]. The current jetting effect must be seriously considered for measuring the longitudinal MR, especially for exploring the chiral anomaly in recently discovered Weyl semimetals [6,9].

Figure 3(a) shows a metallic profile for the temperature dependent resistivity for several representative samples for TaAs₂ and NbAs₂ in zero magnetic field. With a similar profile of linear-temperature-dependent resistivity above 100 K, the samples T3, T4, N5, and N9, which were grown by using TaBr₅ or NbI₅ as the transport agents have much smaller residual resistivity at 2 K than those of the samples T1 and N2, which were grown by using I₂ as the agent. Between 50 and 5 K, the samples with large RRR exhibit the *R*-*T* profiles with the relation $R(T) = R_0 + aT^n$ for TaAs₂ and NbAs₂. As shown in Fig. 3(b), *n* varies from 2.7 to 3.0 for five high-quality samples, which is different from a quadratic temperature dependent behavior for Fermi liquid. Due to the low resistance and relatively large measurement error, we cannot determine the power law below 5 K.



FIG. 6. (a) MR versus H at 2 K when H is tilted from (001) direction to *b* direction. Inset shows the sketch of the experimental setup. (b) Transversal MR versus H at 2 K when H is rotated in the plane perpendicular to *b* direction. Upper inset shows the angular dependent transversal MR at 2 K for H fixed in 3, 5, and 9 T. Lower inset shows the sketch of the transversal MR experimental setup. The angular dependence of the MR for NbAs₂ is similar as that for TaAs₂.



FIG. 7. (a, b) Field dependent Hall resistivity at different temperatures for the samples T4 and N5, respectively. (c, d) The fitting results for the electron and hole densities (*n* and *p*), and their mobilities (μ_e and μ_h) at different temperatures for T4 and N5, respectively. No hole information above 150 K is obtained in this measurement. The lines are guided by eyes.

When a magnetic field is applied to the samples T4 and N5 with large RRR, their resistivity at low temperatures significantly increases, while the increase at high temperatures is very limited [Figs. 3(c) and 3(d)]. The large difference of the MR at high and low temperatures leads $\rho(T)$ in a magnetic field showing a "transition" from a metallic profile at high temperatures to a semiconducting profile at low temperatures. This field-induced transition-like behavior is similar to the previously reported behavior in other semimetals [1,4,6,8,9,14,20].

Figures 4(a) and 4(b) show that the MR values decay three orders of magnitude from 2 to 100 K for T4 and N5. Below 5 K, the MR shows clear Shubnikov-de Haas (SdH) oscillations on the background of a parabolic profile. Although the MR changes significantly at different temperatures, it remains a clear power-law dependence with MR \propto H^m where m = 1.7and 1.9 for T4 and N5, respectively [Figs. 4(c) and 4(d)]. This power law is very close to a parabolic field dependence predicted by the two-band theory for identical compensated hole and electron densities [1,18,20]. The MR persists this power-law dependence but it is only 0.13 and 0.19 for N5 and T4 in 9 T at room temperature (300 K), respectively. We notice that some semimetals including bismuth and graphite [3,6,8,9,21,22,38,39] show their MR one to two orders of magnitude larger than those of TaAs₂ and NbAs₂ at room temperature. On the other hand, the MR values for T4 and N5 at room temperature are close to the reported values for WTe_2 [1], NbSb₂ [4], PdCoO₂ [15], and PtSn₄ [14].

In order to better understand the MR in higher magnetic fields, two samples were measured in a pulsed magnetic field as high as 55 T in National High Magnetic Lab in WuHan, China. Figure 5 shows the results for TaAs₂ and NbAs₂. The MR deviates from the quadratic field-dependent power law above 15 T. The MR for TaAs₂ shows strong, complicated SdH oscillations above 30 T and intends to be saturated at 45 T. Detail analysis on the saturations of MR in high magnetic field will be presented in the future.

Figure 6 shows the MR when the magnetic field is tilted in different directions while the current is always along *b* direction. In order to minimize the current jetting effect, the sample T17 was prepared to be long, thin bars with four contacts fully cross their width in this measurement. When the field direction is tilted from that parallel to the (001) direction to *b* direction, the MR drops rapidly [Fig. 6(a)]. Such large difference between the longitudinal and transversal MR is also observed in WTe₂ [1] and NbSb₂ [4]. On the other hand, the transversal MR changes small when the magnetic field is rotated in the plane perpendicular to *b* direction. Figure 6(b) shows that the minimal of the transversal MR occur when the field is close to the *a* direction (Θ = 280° and 100°). The polar plot for the MR in the inset of Fig. 6(b) shows a twofold-rotation symmetric pattern,



FIG. 8. (a, c) Oscillatory part of the resistivity ratio $\delta \rho_{xx}/\rho_{bg}$ versus $1/\mu_0 H$ for the samples T4 and N5 at 2 K, respectively, where ρ_{bg} is the nonoscillation background of ρ_{xx} . Insets show the (b, d) FFT spectra for T4 and N5 at different temperatures. Insets show the fitting of the amplitude of the SdH oscillations with respect to temperature yields the cyclotron electron mass for the main frequencies for T4 and N5, respectively.

which is consistent with the monoclinic crystal structure. The cusp in MR for $\Theta = 280^{\circ}$ and 100° may be due to special Fermi surface geometry [38] or relatively large sampling interval.

Hall measurements provide information about the carriers for TaAs₂ and NbAs₂. Figures 7(a) and 7(b) show that the Hall resistivity (ρ_{yx}) is negative and linear dependent to the magnetic field when T \ge 125 K and 200 K for T4 and N5, respectively. At low temperatures, ρ_{yx} apparently deviates from the linear-field dependence, which indicates two types of carriers. We found that the Hall conductivity (σ_{xy}) can be well-fitted by the two-carrier model from the two-band theory [40],

$$\sigma_{xy} = [n_h \mu_h^2 \frac{1}{1 + (\mu_h H)^2} - n_e \mu_e^2 \frac{1}{1 + (\mu_e H)^2}]eH, \quad (1)$$

where $\sigma_{xy} = \frac{\rho_{yx}}{\rho_{yx}^2 + \rho_{xx}^2}$, n_e (n_h), and μ_e (μ_h) denote the carrier concentrations and mobilities for electrons (holes), respectively. The Hall conductivity for NbAs₂ changes from positive to negative at low temperatures (the datum are not showing here). The fitting results show close *n* and *p* at low temperatures [Figs. 7(c) and 7(d)]. At high temperature

range, the hole contributions to the Hall signals are negligible due to its rapidly decreased mobility, and the data can be well fitted by an one-band model. The mobilities at 2 K are about 1×10^5 cm²V⁻¹s⁻¹ for electrons and holes for both TaAs₂ and NbAs₂. At room temperature, μ_e drops to the order of 5×10^2 cm²V⁻¹s⁻¹ for both samples. The Hall measurements reveal that TaAs₂ and NbAs₂ are compensated semimetals with large mobilities of electrons and holes at low temperatures.

The SdH oscillations in the field-dependent resistivity for TaAs₂ and NbAs₂ were extracted via subtracting the nonoscillation background (ρ_{bg}) and dividing by the background at low temperatures (Fig. 8). The SdH oscillations with respect to 1/H form complicated patterns which obviously come from multiple frequencies. Fast Fourier transform (FFT) analysis reveals that the samples T4 has two main frequencies about $\alpha = 39(3)$ T and $\beta = 54(3)$ T and N5 has two main frequencies about $\epsilon = 124(3)$ T and $\delta = 224(3)$ T, while the double and higher harmonic frequencies of the main frequencies were observed as well (Figs. 8(b) and 8(d)). The origin of frequency about 200 T for T4 is not clear. The complete identification of the entire Fermi surface from quantum oscillation is left for future work.



FIG. 9. (a, b) Fermi Surfaces of electron and hole pockets for $TaAs_2$ in its primitive Brillouin zone, respectively. (c) The total and partial DOS for $TaAs_2$.

The existence of multiharmonic frequencies indicates a long transport lifetime for both compounds in accordance with their high mobilities. In order to obtain the cyclotron mass for the main frequencies, we fitted FFT amplitude ratio amp/amp(2K) versus temperature for main frequencies of T4 and N5, respectively [Figs. 8(b) and 8(d)]. Using the fitting formula [41],

$$\frac{\mathrm{amp}}{\mathrm{amp}(2K)} \propto \frac{2\pi^2 k_B T / \hbar \omega_c}{\sinh(2\pi^2 k_B T / \hbar \omega_c)},\tag{2}$$

where k_B is Boltzmann constant, cyclotron frequency $\omega_c = e\bar{B}/m_{\rm cyc}$, $m_{\rm cyc}$ is cyclotron mass, and \bar{B} is the effective magnetic field in fitting range (from 1.6 T to 9 T), we got $m_{\rm cyc}$ equals to 0.08(2) m_e, 0.14(3) m_e, 0.24(3) m_e, and 0.21(3) m_e for α , β , ϵ , and δ , respectively. The $m_{\rm cyc}$ for frequency about 200 T for T4 equals to 0.18(3) m_e. Further experiments accompanied by accurate band structure calculations are needed to address the pockets related to these SdH oscillations.

IV. DISCUSSION

The calculated band structures for TaAs₂ and NbAs₂ are consistent with their compensated semimetal characteristics. Figures 9(a) and 9(b) show the Fermi surface mapping in which only one valence and conduction bands intersect the Fermi level, respectively. Small, ellipse-like electron and hole pockets appear at the boundary of the primitive Brillouin zone. The DOS at Fermi level is very close to the minimum in the spectrum [Fig. 9(c)]. The results for TaAs₂ and NbAs₂ are similar as those in Ref. [28] and the band structure for NbSb₂ [4]. Since the Fermi surface is complicated with several different electron and hole pockets, the identification of the frequencies for the SdH oscillations with the extremal surface areas needs further elaborations.

We next discuss the sample and temperature dependence for the large MR for TaAs₂. Table I shows that the values of RRR and MR are strongly dependent on the transport agents using in the growth. Although the MR changes four orders of magnitude for different samples, it follows the same power law of MR = bH^m , where $m = 1.7 \pm 0.1$ for all the samples (Fig. 10). On the other hand, the coefficient *b* seems to follow a

TABLE I. RRR and MR for different samples of TaAs₂ and NbAs₂ grown by using different transport agents.

Sample	Transport agent	RRR	MR at 2 K in 9 T
N2	I_2	53.6	332
N5	NbI_5 : 75 mg	317	8.8×10^{3}
N9		313	7.4×10^{3}
T1	I_2	24.0	95.7
T2	TaBr ₅ : 75 mg	224	3.2×10^{3}
Т3		198	2.0×10^{3}
T4		270	5.8×10^{3}
T17		218	3.6×10^{3}
T5	TaBr ₅ : 74 mg	44.0	315
T6	and TeBr ₄ : 4.7mg	42.4	285
T7		41.5	268
T8	TaBr ₅ : 74 mg	15.6	44.4
Т9	and TeBr ₄ : 11.4mg	17.3	53.8
T10	-	18.6	55.1
T11	TaBr ₅ : 74 mg	33.0	168
T13	and TeBr ₄ : 22.3 mg	38.4	208
T14	TeBr ₄ : 35 mg	2.30	0.093
T15	0	2.33	0.064
T16		2.42	0.258

All the samples come from nine batches with different transport agents. The growth conditions are the same as described in Sec. II.

power law of $b \propto RRR^q$ where q = 1.6 for all TaAs₂ samples. This power law relation for the RRR and *b* results in the giant difference for the MR for the samples with different qualities. The dependence of the MR and RRR on the transport agents is likely due to the doping effect of the impurity from the transport agents, such as tellurium atoms on the As site, or the arsenic deficiency described in Ref. [27]. Similar significant drop of large MR due to chemical doping has been observed in WTe₂ [42].

In order to better understand the electron scattering in different samples, the MR at 2 K for different samples is plotted in a modified Kohler plot [Fig. 10(c)]. Considering the large uncertainty of the dimensions for the samples, we chose $(\mu_0 H \times \text{RRR})^{1.7}$ instead of $(\mu_0 H / \rho_0)^{1.7}$ in the plot. We notice that the curves for the samples with relatively small MR (<100) cluster together, while the curves for the three samples of T2, T3 and T4 with large MR (>1000) clearly deviate from the cluster. A plausible explanation is that the impurity induced electron scattering dominates the transport behaviors in a magnetic field for those samples with small RRR. For high-quality samples such as T2, T3, and T4, the scattering process in a magnetic field is strongly dependent on the samples. A modified Kohler plot for the MR versus $(\mu_0 H/\rho_0)^{1.7}$ for the sample T4 at different temperatures is shown in Fig. 10(d). The curves do not fall into a same region



FIG. 10. (a) MR versus magnetic field for different samples of TaAs₂ at 2 K in a double-logarithmic plot. (b) Coefficient b (MR = $b \times H^{1.7}$) versus RRR for different samples of TaAs₂ in a double-logarithmic plot. The red line is the fitting line of the order of 1.6. (c) Modified Kohler plot for different samples of TaAs₂ at 2 K. Inset shows the Kohler plot for the samples T2, T3, and T4 at 2K in different range. (d) Kohler plot for the sample T4 at different temperatures using $MR = F[H/\rho(0)] \propto [H/\rho(0)]^{1.7}$. Inset shows the same Kohler plot of (d) in different range.

Semimetal	$\mu (\mathrm{cm}^2 \mathrm{V}^{-1} \mathrm{s}^{-1})$	RRR	MR	Reference
Bi	N.A	476	1.6×10^7 at 4.2 K 5 T	[17]
Bi	1.1×10^{7}	150	2×10^4 at 2 K 0.5 T	[38]
Bi	3×10^{6}	40	1.1×10^4 at 4.2 K and 1.8 T	[50]
Graphite	1.0×10^{6}	37	8.3×10^3 at 4.2 K and 2.3 T	[51]
WTe ₂	1×10^{5}	1256	3.1×10^3 at 1.2 K and 10 T	[2]
WTe ₂	5×10^{4}	370	4.5×10^3 at 4.5 K and 14.7 T	[1]
TaAs ₂	2.1×10^{5}	270	5.8×10^3 at 2 K and 9 T	Current paper
NbAs ₂	1.6×10^{5}	317	8.8×10^3 at 2 K and 9 T	Current paper
NbSb ₂	N.A	450	8.8×10^3 at 2 K and 9 T	[4]
Cd	N.A	$4.0 imes 10^4$	1×10^5 at 1.4 K and 2.5 T	[52]

The mobility data were collected from the fitting of two-carrier-model in the literatures.

at any temperatures. Since the mobilities of the electrons and holes have different temperature dependence, the violation of the Kohler's rule is not unexpected.

We list the parameters for the compensated semimetals with large, parabolic-field-dependent MR in Table II. All the semimetals with large MR have their main-carrier mobilities much larger than $1 \text{ m}^2 \text{V}^{-1} \text{s}^{-1}$. It seems that the MR for all the semimetals is positively correlated with their RRR and mobilities, despite the large difference of the band structures and carrier densities.

Finally, we discuss the saturation of the MR for the compensated semimetals. The parabolic MR for WTe_2 does not show any sign of saturation in a magnetic field up to 60 T, which is believed to be due to a prefect balance of the electron and hole pockets [1,2]. As far as we know, the saturations of the transversal MR for compensated semimetals have only been observed for bismuth [43] and graphite [44], which have small quantum limits less than 10 T. In such high field, the quantum effect may affect the MR significantly different from the classical behaviors. The MR for TaAs₂ intends to be saturated in 45 T while its quantum limit is about 50 T. Whether a quantum effect also affects the saturation of the MR

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for semimetal need further measurements for other semimetals in strong magnetic field.

V. CONCLUSION

To conclude, we grew large single crystals of NbAs₂ and TaAs₂ via a CVT process. These two compounds are compensated semimetals with large, parabolic-field-dependent MR at low temperatures. We believe that most of the large, parabolic MR for the compensated semimetals can be explained as a result of electron-hole balance.

Note: We noticed several related works in arXiv [45–49] when we prepared this draft.

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