

Nonadiabatic effects in the phonon dispersion of $\text{Mg}_{1-x}\text{Al}_x\text{B}_2$

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Superconducting MgB_2 shows an E_{2g} zone center phonon, as measured by Raman spectroscopy, that is very broad in energy and temperature dependent. The Raman shift and lifetime show large differences with the values elsewhere in the Brillouin zone measured by inelastic x-ray scattering (IXS), where its dispersion can be accounted for by standard harmonic phonon theory, adding only a moderate electron-phonon coupling. Here we show that the effects rapidly disappear when electron-phonon coupling is switched off by Al substitution on the Mg sites. Moreover, using IXS with very high wave-vector resolution in MgB_2 , we can follow the dispersion connecting the Raman and the IXS signal, in agreement with a theory using only electron-phonon coupling but without strong anharmonic terms. The observation is important in order to understand the effects of electron-phonon coupling on zone center phonon modes in MgB_2 , and also in all metals characterized by a small Fermi velocity in a particular direction, typical for layered compounds.

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MgB_2 is a superconductor [1] where the mechanism for pairing is conventional electron-phonon coupling. It shows an unexpectedly high transition temperature of 39 K, almost twice that of most conventional systems at ambient pressure [2], and close to those of high-temperature superconductors such as cuprates and iron pnictides. The phonon mode involved in the coupling is the transverse E_{2g} mode, propagating along the c^* direction with the atomic displacement parallel to the ab plane (Refs. [3–5] and references therein). Understanding the physics in MgB_2 is fundamental because it is the first and archetypal electron-phonon mediated superconductor with high-frequency phonons at ambient pressure [6]. This is very important in the search of high-temperature superconductivity, since these systems can be well modeled [6,7] and have the potential for extremely high transition temperatures [8].

MgB_2 shows an anomalously low isotopic effect, surprising for a conventional pairing mechanism [9]. It has been suggested that a strong anharmonic term, in the E_{2g} mode energy, may explain this anomaly [10]. The proposed strong anharmonicity seems in good agreement with the Raman data [3,11–15], but it is in contrast with inelastic x-ray scattering (IXS) phonon dispersion results [4,5], which are in good agreement with *ab initio* models employing the quasiharmonic approximation. However, a direct comparison is not possible because Raman spectroscopy only probes phonons close to the Brillouin zone center, Γ , where the IXS signal is dominated by elastic scattering.

In a previous work [5], it has been shown, from a close comparison of the temperature dependence of the IXS and Raman shift, that the apparent dichotomy between the

two measurements could be reconciled taking into account nonadiabatic effects [16] appearing only close to Γ . In that framework, anomalies in the Raman spectra have their origin in a very steep dispersion where the Landau damping of phonons, induced by the electrons, disappears, because the phonon wave vector is shorter than the Landau-damping threshold [5]. This effect, which is very small and difficult to detect in most metals [17,18], would induce spectacular, temperature-dependent, changes of the Raman width and shift in MgB_2 .

Although the above explanation is quite reasonable, there is no complete proof of it, since the calculations are only qualitative, and we lack a direct observation of the suggested steep, anomalous dispersion close to Γ . Here we show, using high Q resolution inelastic x-ray scattering, that a very steep phonon dispersion can be detected in $\text{Mg}_{1-x}\text{Al}_x\text{B}_2$ at very small x substitution, and rapidly disappears at higher Al content. The result is important for the theory of superconductivity of MgB_2 , because an alternative interpretation of the Raman shift and broadening, in terms of anharmonicity, could explain the anomalous isotopic effect of MgB_2 [19], but not the IXS shift and broadening [5]. Moreover, there are very few experimental confirmations of nonadiabatic effects in solids, in particular their effect in the phonon dispersion, even if they can be relevant in the interpretation of vibrational data and the electron-phonon coupling of various layered metals such as intercalated graphite and graphene and, in general, for metals characterized by a small Fermi velocity in a particular direction in the Brillouin zone [17].

For crystal growth and characterization, see Ref. [20] as well as Ref. [5] for the MgB_2 samples. Note that in the text we rounded the precise stoichiometry given in Ref. [20] as follows for the Al content: $0.93 \rightarrow 1$ (note that there is no Mg in that sample); $0.71 \rightarrow 0.7$ and $0.37 \rightarrow 0.4$.

The IXS experiments were performed on the ID28 beam line of the European Synchrotron Radiation Facility. The

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$\text{Mg}_{1-x}\text{Al}_x\text{B}_2$ samples ($0.37 \leq x \leq 1.0$) were mounted in a vacuum chamber at room temperature such that the 110 and 001 directions were contained in the scattering plane. This orientation allowed measurements of transverse and longitudinal branches along the Γ -A and Γ -M direction. For these samples, the backscattering monochromator was set at the Si (8 8 8) reflection order [21,22], corresponding to a wavelength $\lambda = 0.7839 \text{ \AA}$ (photon energy: 15 817 eV), and a resolution of ΔE of 5.5 meV.

IXS measurements in stoichiometric MgB_2 crystal were realized using higher energy resolution ($\Delta E = 3 \text{ meV}$) at the Si (9 9 9) reflection order of the backscattering monochromator, in a very similar instrumental configuration of a previous experiment [5] with the notable difference that we benefited from using nine instead of five analyzers, thus allowing finer steps in the reciprocal space in particular for the *in-plane* direction Γ -M, and taking particular care of adjusting the slit opening in front of the analyzers to maximize the spectrometer brilliance while optimizing the Q resolution as detailed in Ref. [20].

All measurements were guided by calculations of inelastic structure factors resulting from first principles calculations. The spectra were analyzed by a fitting program which automatically accounted for the experimental resolution taken from measurements or a model as described in Ref. [5]. Typical scan times were about 2 h. Details of the *ab initio* simulation in the virtual crystal approximation (VCA) used in the present work are described in Ref. [23].

Most of our data have been collected for the MgB_2 and $\text{Mg}_{0.6}\text{Al}_{0.4}\text{B}_2$ single crystals and the Γ -A direction which is of particular interest. In Fig. 1, left panel, we show typical IXS spectra along the Γ -A line, at a reduced wave vector $\mathbf{q} = (0\ 0\ 0.2)$, for several Al contents spanning from 0 to 1. In Fig. 1, right panel, the energy of the E_{2g} phonon as determined from the fit of these data is compared to the ones calculated for the same phonon using the VCA approach (from Ref. [23] and present work for $x = 0.4$). In Fig. 2 we show a complete

dispersion calculated for Al content $x = 0.4$ along Γ -A and Γ -M high symmetry directions, in the VCA approach. The dispersion is compared with our IXS and Raman data for the sample with $x = 0.37$ ($\text{Mg}_{0.6}\text{Al}_{0.4}\text{B}_2$).

In Fig. 3, left panel, we show the results of our present IXS measured dispersion for Al content $x = 0$ (blue markers) and 0.4 (white filled markers) along Γ -A and Γ -M directions. We also add Raman data at $x = 0$ (cyan and yellow filled symbols, from Ref. [5]) and 0.4 (black filled symbols). We compare with DFT calculations (red lines) for $x = 0$ only, taken from Ref. [5]. The agreement is quite well overall, except for the zone center Raman data at $x = 0$. Calculation in Refs. [5,23] are quite similar for $x = 0$. An important difference, apart from the virtual crystal approximation that allows Ref. [23] to change x , is that the authors of Ref. [23] used relaxed lattice parameters from the DFT calculation itself, while calculations in Ref. [5] used experimental lattice parameters, which give in general energies closer to the experimental ones, while relaxed lattice parameters tend to push some modes to higher energies, about 6 meV in the present case. This made the E_{2g} mode energy calculated in Ref. [23] to match well the zone center energy but is about 6 meV higher at the zone boundary (point A). Instead the E_{2g} mode energy calculated in [5] with experimental lattice parameters matches well at the zone boundary (point A), as shown in Fig. 3. To them, we added also an empirically renormalized dispersion (black lines) as guide to the eye of the zone center anomaly.

The overall evolution of the phonon modes along Γ -A and Γ -M as a function of x , going from $x = 0.0$ to 0.4 qualitatively follows what is expected according to Ref. [23] for Al substitution in the $x = 0.3 - 0.5$ range: a large hardening for the E_{2g} mode ($\sim +35 \text{ meV}$) and a reduction for the A_{2u} and longitudinal acoustic ones (few meV energy increase) with a concurrent moderate softening of the B_{1g} mode (few meV decrease), with the E_{1u} staying approximately at the same energy. A similar behavior is also experimentally observed for the E_{2g} mode in carbon-doped $\text{Mg}(\text{B}_{1-x}\text{C}_x)_2$ by IXS [24].

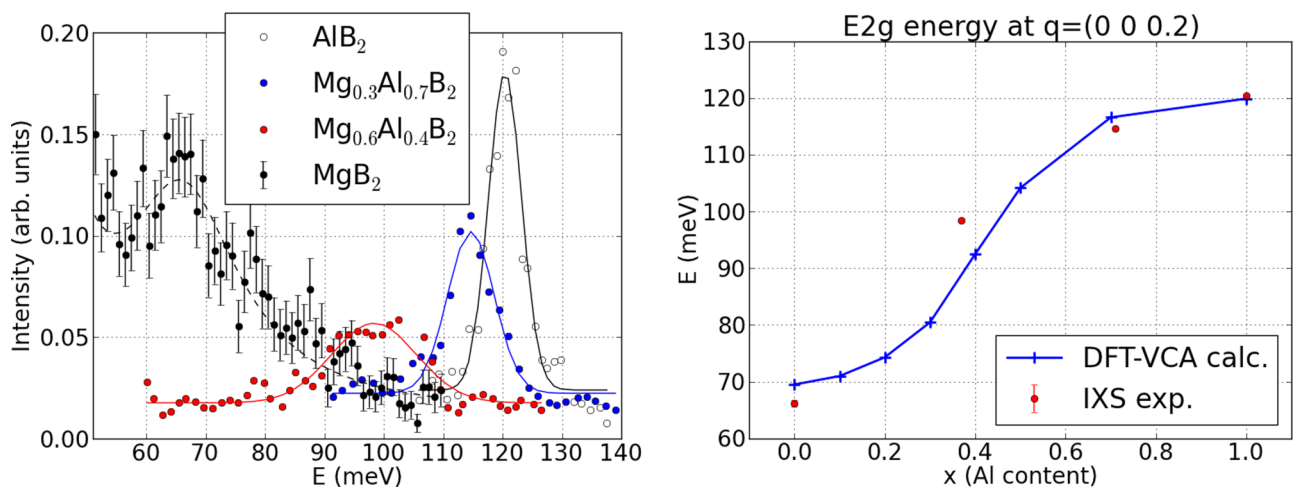


FIG. 1. Left panel: IXS spectra (circle) and fit (lines) for $\text{Mg}_{1-x}\text{Al}_x\text{B}_2$ at the reduced wave vector $q = (0\ 0\ 0.2)$, with colors coding the content x . Error bars can be inferred from the data dispersion. Lines are models of the data, with pseudo-Voigt fits (dashed, as in Ref. [5]) or simple Gaussian (continuous lines) profiles. Right panel: Energy shift for the phonon at $q = (0\ 0\ 0.2)$ as a function of the Al content x , from the fit of the data in the left panel (error bars from fit, mostly smaller than symbols), as well as from calculations in the virtual crystal approximation (VCA).

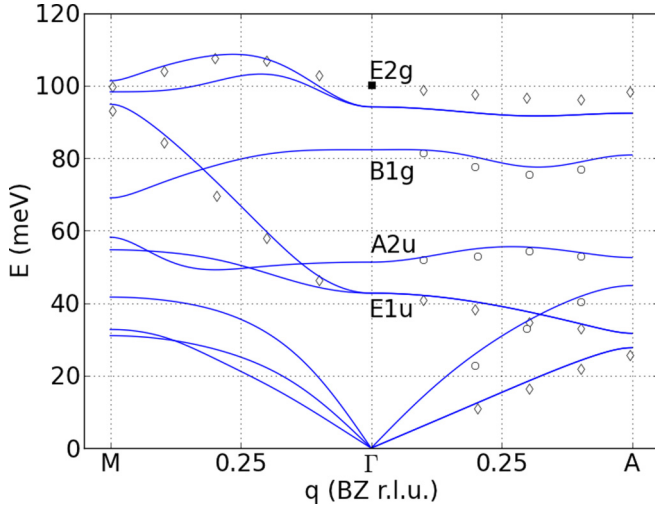


FIG. 2. Calculated and measured dispersion of $\text{Mg}_{0.6}\text{Al}_{0.4}\text{B}_2$. q is given in Brillouin zone (BZ) reciprocal lattice units (r.l.u.).

Dispersion calculations for $x = 0.4$ show a good quantitative agreement with our data; in particular we note that the frequency for the E_{2g} mode at the zone center measured by Raman lies on an almost straight line between the point at small q in the Γ -A and Γ -M directions, as in the VCA simulation.

As mentioned, this is not the case for $x = 0.0$, for which there is a disagreement between the DFT calculations extrapolated to the zone center and the Raman data for the E_{2g} mode, as can be appreciated in Fig. 3, left panel. This was first ascribed to anharmonic effects [10], and seemed to be corroborated by the larger width of the Raman peak at room temperature, a factor of 2 compared to the mode width measured by IXS along Γ -A, and with a large temperature effect that was not observed

in IXS measurements [4]. In turn, IXS data seemed to fit quite well with the DFT calculation. Alternatively, it was later suggested that the effect could be ascribed to the crossing of the Landau damping threshold [25], meaning that below a certain q wave vector q_0 in the Brillouin zone, the electron-phonon coupling cannot take place anymore, so that the phonon energy recovers from its value with electron-phonon coupling E to the bare one at $q = 0$, E_0 , which would be about 93 meV for the E_{2g} mode. It follows that in the region $0 < q < q_0$, where the phonon modes decouple from the electrons, the system is not adiabatic anymore. This leads to a strong modification of the Raman spectral function, which probes the polarizability variation very close to (but not exactly at) $q = 0$ as shown by Ref. [16], with a larger broadening compared to the intrinsic one of the phonon modes, and a temperature dependence of such broadening in qualitative agreement with the measured one. The latter explanation also predicts an energy hardening with temperature, which was experimentally confirmed in Ref. [5]. The reverse behavior would be expected from an anharmonic effect.

However, the proof was only qualitative, lacking a direct numerical comparison with data. The above mentioned recovery of E_0 would also induce a very steep dispersion as q_0 should be small (in the order of 0.1 of the BZ or less) while the energy shift is as large as $\Delta E = E - E_0 \sim 35$ meV as shown in Ref. [26]. Previous attempts to complete the dispersion close to the zone center Γ were unsuccessful (see Fig. 5 in Ref. [5]). In the present work we came back to perform measurements with a very high Q resolution and very fine step in Q close to Γ , from both the Γ -M and Γ -A directions. The data clearly bend up toward Γ from both sides departing from the calculated dispersion. We stress again that this is not the case at all for the sample with $x = 0.4$, where the Raman data fall exactly on a line going from Γ -M to Γ -A at small q , as expected from

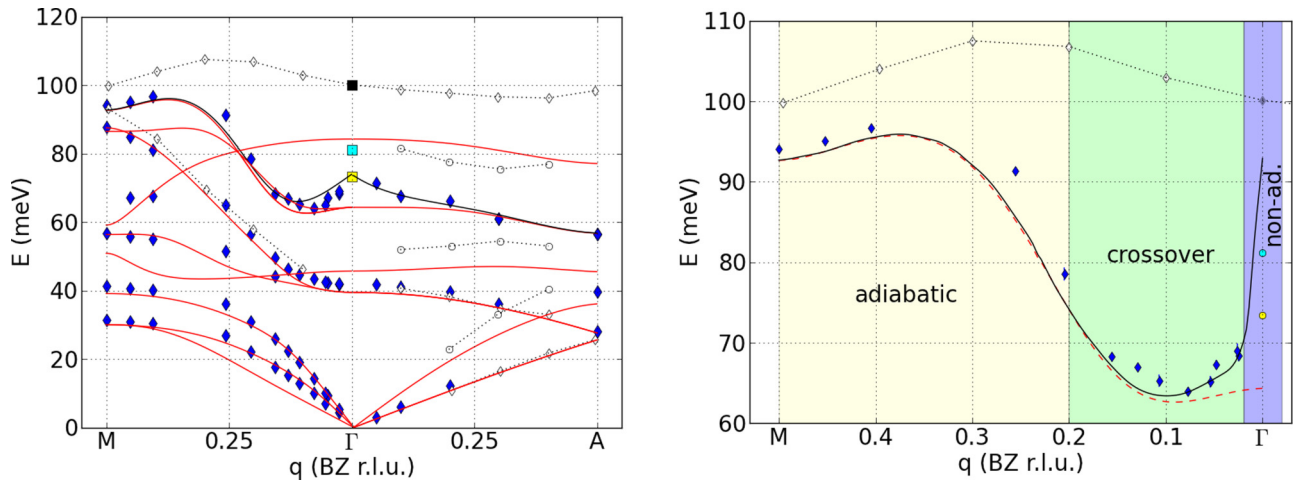


FIG. 3. Left panel: $\text{Mg}_{1-x}\text{Al}_x\text{B}_2$ phonon dispersion along Γ -A and Γ -M directions: IXS data for $x = 0$ (blue diamonds symbols) and 0.4 (white filled diamonds); Raman data for $x = 0$ (filled squares, cyan, average values for $T \leq 100$ K, and yellow, at ambient temperature, from Ref. [5]) and $x = 0.4$ (black square). Error bars from the fit are smaller or of the same order as the symbol size. DFT calculation (red lines) for $x = 0$ only (taken from Ref. [5]). The black line represents an empirical renormalization of the DFT calculation (see text). Right panel: zoom on the E_{2g} dispersion along the Γ -M direction. The dashed red line corresponds to the DFT calculation (red continuous line in Fig. 3). The black continuous line represents the nonadiabatic correction to the DFT calculations as explained in the text. We approximately indicate empirical adiabatic (pale yellow), and nonadiabatic (pale blue) region, separated by a crossover region in pale green, estimated from the difference between the two calculated dispersion curves.

calculations at $x \sim 0.4$ in Fig. 2. To highlight the effect on E_{2g} at $x = 0$, in Fig. 3 we modified the DFT calculated dispersion, multiplying it with an empirical exponential function $c^*(1 + e^{(-q/q_0)})$ (black lines, with $q_0 \approx 0.1$), as guide to the eye. We note here that the function is symmetric in q units of the Brillouin zone, but appears bending faster on the Γ -M side. This is due to the fact that the bare DFT calculated dispersion itself is already bending upward close to Γ . We also note that a very steep phonon dispersion, known as the waterfall effect, was reported for relaxor perovskite materials [27] and explained in terms of mode-mode interference close to the zone center [28]. However, in this scenario the two phonon modes involved have to have the same symmetry. As the only energy phonon branch in MgB_2 above the E_{2g} has B_{1g} symmetry, we can exclude a similar origin of the steep dispersion near Γ in MgB_2 . Moreover in $\text{Mg}_{0.6}\text{Al}_{0.4}\text{B}_2$ there is no steep dispersion at all, although the two modes are even closer in energy.

To see if our observations are compatible with crossing the Landau damping threshold, and the corresponding modifications of the dispersion, we compare with the calculations in Ref. [26]. In that work a modified dispersion along Γ -M is shown, including nonadiabatic effects mentioned above (see Fig. 3 left panel of Ref. [26]). But the E_{2g} dispersion was a simplified, analytical one, and, although quite close to the E_{2g} dispersion calculated by DFT, not accurate enough for a direct comparison with data, being, e.g., about 5 meV below the data for $q \sim 0.3 - 0.4 \times \Gamma$ -M, a shift about as large as the one we are looking for at Γ . This approximate dispersion was used because DFT can be performed neither exactly at the zone center nor with enough q resolution to estimate the corrections of the self-energy for $0 < q < q_0$. Conversely, the DFT calculated dispersion is much closer to the data if we exclude the E_{2g} mode in the q region close to the zone center. In Fig. 3, right panel, we zoom on the same E_{2g} data as in the left panel, along Γ -M, with the red dashed line corresponding to the E_{2g} dispersion calculated by DFT as in the left panel. We extract then the correction due to a frequency dependence of the phonon self-energy from the dispersion in Ref. [26], as the ratio between the adiabatic ($\epsilon_{qE_{2g}}^{\text{adiabatic}}$) and nonadiabatic ($\epsilon_{qE_{2g}}$) dispersion $\frac{\epsilon_{qE_{2g}}}{\epsilon_{qE_{2g}}^{\text{adiabatic}}}$. This is a rather crude approximation in principle, but the corrections are rather small in absolute values, so *a posteriori* valid to first order.

We apply the corrections to the DFT calculated dispersion (black line in Fig. 3, right panel). This is justified by the fact that the analytic calculated $\epsilon_{qE_{2g}}^{\text{adiabatic}}$ in Ref. [26] is already a rather good approximation of the real one. The result is astonishingly close to the data, hence supporting the view that the effect comes indeed from the inclusion of nonadiabatic effects, by computing the phonon self-energy at finite phonon frequency when including the electron-phonon coupling as explained in Ref. [26]. Note that while the zero-temperature calculation recovers the unperturbed frequency (without electron-phonon coupling) at the zone center, the finite temperature and the resolution shift the Raman energy to lower values, with a positive temperature dependence that cannot be explained by anharmonic effects, as first pointed out in Refs. [5,16].

In conclusion, we experimentally reconciled the IXS dispersion for the E_{2g} mode in MgB_2 to the Raman data at the zone center, showing that the apparent difference can be explained without invoking any experimental artifact, but is connected to a very steep dispersion, as foreseen by theory based on anomalous effects of the electron-phonon coupling near the zone center [16,17,26], and does not require any strong contribution from phonon-phonon scattering (anharmonicity). This zone center anomaly disappears when the electron density of states at the Fermi level is depleted, by Al substitution $x \sim 0.4$, concurrently with the loss of superconducting properties [29], a further indication that this anomaly arises from electron-phonon coupling effects.

The observation is important to understand the effects of electron-phonon coupling on zone center phonon modes in MgB_2 , and its relevance to its superconducting properties. Moreover, this is relevant in all metals characterized by a small Fermi velocity in a particular direction, and for which the temperature dependence of the Raman shift is often measured in order to detect electron-phonon coupling effects [17].

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