Macroscopic response and directional disorder dynamics in chemically substituted ferroelectrics

Jacopo Parravicini,^{1,*} Eugenio DelRe,¹ Aharon J. Agranat,² and Gianbattista Parravicini^{3,4}

¹Dipartimento di Fisica, "Sapienza" Università di Roma, 00185 Roma, Italy

³Dipartmento di Fisica, Università di Pavia, 27100 Pavia, Italy

⁴IPCF-CNR, "Sapienza" Università di Roma, 00185 Roma, Italy

(Received 1 August 2015; revised manuscript received 7 December 2015; published 21 March 2016)

Using temperature-resolved dielectric spectroscopy in the range 25–320 K we investigate the macroscopic response, phase symmetry, and order/disorder states in bulk ferroelectric $K_{1-y}Li_yTa_{1-x}Nb_x$ (KLTN). Four long-range symmetry phases are identified with their relative transitions. Directional analysis of the order/disorder states using Fröhlich entropy indicates global symmetry breaking along the growth axis and an anisotropic dipolar effective thermodynamic behavior, which ranges from disordered to ordered at the same temperature for different directions in the sample. Results indicate that the macroscopic polarization, driven by nanosized polar regions, follows a microscopic perovskite eight-sites lattice model.

DOI: 10.1103/PhysRevB.93.094203

I. INTRODUCTION

Many promising functional materials with remarkably strong and malleable magnetic, electric, and mechanical properties, are nominally pure crystals that actually host compositional disorder at the nanometric scale [1-4]. Although the presence of disorder does appear to be a key ingredient, its role in response is still largely unknown and subject of intense research efforts [5–7]. One line of research focuses on nanodisordered ferroelectrics, with the aim of developing innovative materials for capacitors, memories, and photonics [8–10]. Among these, an important role is played by solidsolution perovskites, which are commonly accepted as a paradigm of disordered ferroelectric crystals [5,11,12]. The behavior of these compounds can continuously span from that of a canonical relaxor, diffuse-transition relaxor, sharptransition relaxor, to a true ferroelectric [5,12], depending on the composition and stoichiometry [5,13]. A key feature of disordered ferroelectrics is that they can manifest many of the properties of standard crystals, in terms of electrical, optical, and mechanical response, and still host a population of nanosized reconfigurable polar regions, the so-called polar nanoregions (PNRs). These are correlated mesoscopic regions of polarizations that are widely considered responsible for the giant electro-optic and piezoelectric effect [5,14,15]. Moreover, these PNRs can behave as liquid, solid, and even glassy systems, in the so-called nonergodic dipolar regimes, and all this in one and the same crystal structure, typically a large scale perovskite [16-18]. How exactly a crystal is able to host a dynamic population of dipoles is still an open question, and its understanding evidently lies at the heart of future applications [5,11,12]. In fact, it is well known that in a perovskite structure the ion disposition in the ABO₃ lattice may be distorted, causing local inherent or internal dipolar disorder [19,20]. When a substitutional disorder is simultaneously present, below the so-called Burns temperature T_B , its combination with this internal disorder can give rise to lattice distortions, strains, and local electric fields [21], which then lead to PNRs [3,5,22-24]. Moreover, the growth process along a specific direction (say c) is known to strongly influence the defects and the distributions of strains and charges even though the crystal is never poled [25,26]. Altogether these phenomena are able to cause a local symmetry breaking in what would be normally considered a high-symmetry paraelectric phase, a breaking that generates a wide landscape of PNR-driven properties whose nature and temperature dependence is still under study. The local ordering and crystalline symmetry of PNRs have been extensively investigated using various techniques, the principal being diffractometric [20,27–35]. Results indicate that the behavior and response of PNR-hosting samples is strongly affected by disorder [5,13,21,36,37]. Several works have been devoted to investigating PNR dynamics in correspondence to the cubic-to-tetragonal phase transition using NMR and EXAFS techniques [29–31]. Moreover, also in the lower temperature orthorhombic and rhombohedral phases, the internal symmetry of the PNRs was recently explored [27,28]. Nonetheless the features of the PNRs in the low-temperature crystalline phases remain, at present, only partially explored for a very restricted number of compounds (e.g., BaTiO₃, SrTiO₃, KNbO₃, and PbTiO₃) [5,11,12,23].

Here we study the dielectric response of a solid solution single crystal KLTN, a disordered ferroelectric with established out-of-equilibrium properties and giant response [25,38-40]. Directional components of entropy variations are inspected using dielectric spectroscopy and Fröhlich entropy [41-43] (Fig. 1). We focus on the correlation between the degrees of order (disorder) and four long-range symmetry phases in the temperature range 25-320 K. The anisotropic features are specifically investigated by considering the structural phases along the three main crystalline directions. Remarkably, the distribution of the macroscopic order/disorder state is found to be in agreement with the microscopic so-called eight-sites model developed by Comès et al. [19,20]. This model was introduced to account for diffused photon scattering in xray studies of BaTiO₃ and KNbO₃ and implies that some perovskites never actually form high-temperature crystalline phases, have order/disorder transitions, and have a built-in

²Applied Physics Department, Hebrew University of Jerusalem, 91904, Israel

^{*}Corresponding author: jacopo.parravicini@roma1.infn.it; Currently with Dipartimento di Scienza dei Materiali, Università di Milano-Bicocca, 20125 Milano, Italy.



Temperature

FIG. 1. From [41]: Schematic temperature dependence of the dielectric constant ϵ_s (blue curve) and of the related entropy change as defined by Eqs. (1) and (2): $S \propto \partial \epsilon_s / \partial T$ (red curve) due to the field-induced polarization. Fröhlich stated that if S > 0 the field creates disorder, if S < 0 it creates order.

capability of hosting disorder. Our results bring us to conclude that it is these extra degrees of freedom that support the polar dynamics of the PNRs within a specific crystal phase, a basic tassel in our understanding and harnessing of functional ferroelectrics and relaxors.

II. EXPERIMENTS

We have grown a solid solution of lithium-enriched potassium-tantalate-niobate ($K_{1-y}Li_yTa_{1-x}Nb_xO_3$) with x =0.36, y = 0.003, single crystal using the top-seeded-solution method and we have extracted a *c*-direction-pulled zero-cut $3.87^{(a)} \times 3.55^{(b)} \times 2.30^{(c)}$ mm sample [25,44]. The concentration of potassium, tantalum, and niobium in the KLTN sample was determined by electron microprobe analysis. The concentration of Cu and Li were measured using flame atomic absorption spectroscopy employing a procedure that was specially developed for KLTN and described in [45]. The permittivity measurements versus temperature are carried out by a precision LCR meter (Agilent-4284A) for different frequencies, applying a probing 1 V/cm field to planeparallel electrodes painted on the crystal facets. Temperature variation in the 25–320 K range is achieved using a Tcontrolled closed two-stage helium cryostat (in a vacuum configuration). The cooling rate is 0.5 K/min and temperature is monitored through a calibrated silicon diode sensor (0.01 K in precision). The efficient thermal contact of the sample with the temperature source and the adopted low cooling rate render each single permittivity ϵ measurement isothermal. The dielectric response is measured along the three crystalline main directions, a, b, c, in three different consecutive scan stages, keeping all experimental parameters constant, in particular thermal cooling rate, applied field value, and thermal contact. After the experimental stage in one direction is accomplished, the pair of parallel electrodes is removed and placed on the facets corresponding to the next direction. Each set of measurements was carried out for the frequencies: 1, 10, 100, 1000 kHz.

A. Dielectric response

In Figs. 2(a)-2(c) the relative permittivity as a function of temperature $\epsilon(T)$ is reported for the employed frequencies along the crystalline axes a, b, and c, respectively, in the 25–320 K range, on cooling. The $\epsilon(T)$ curves, for all the investigated directions, give evidence of the occurring of three phase transitions. In particular, considering the a direction [Fig. 2(a)], three marked peaks are displayed with their maxima T_m , distributed in frequency. Below the maxima, the overlapping of the permittivity curves for the different frequencies, in correspondence with a sharp decrease, marks



FIG. 2. Dependencies on temperature of the relative dielectric permittivity measured at 10^3-10^6 Hz frequencies along *a* (a), *b* (b), and *c* (c) directions for the cooling stage. For each plot are indicated the nominal crystalline phases $P_{\rm rho}$ (rhombohedral), $P_{\rm ort}$ (orthorhombic), $P_{\rm tet}$ (tetragonal), and $P_{\rm cub}$ (cubic), with the relative transition temperatures T'_C (rhombohedral-to-orthorhombic). For each transition, the characteristic temperatures are found to be the same for all directions. Note that in main plot (c) the permittivity scale is one order of magnitude smaller than in (a) and (b) plots, while the insets with the same scale show that the differences between permittivities tend to vanish as the temperature approaches 0 K.

the critical temperature T_C of a long-range symmetry transition in the sample [5,11,12]. The shift of the permittivity maxima with respect to the corresponding T_C and the dispersion in frequency of the $\epsilon(T)$ curves around the maxima signal a broad-type transition. This kind of frequency dependence is found in agreement with that reported in the literature on lead-free relaxor ferroelectrics [5,13]: We note that it does not manifest the features of a *canonical relaxor* response but instead those of a *sharp-transition relaxor* [5] (or "incipient relaxor" of [46]).

We associate the detected temperatures $T_C^{'''} = 282$ K, $T_C^{''} = 202$ K, and $T_C^{'} = 156$ K with the cubic/tetragonal, tetragonal/orthorhombic, and orthorhombic/rhombohedral transitions, respectively, in agreement with other well established disordered perovskite structures [26–28] and with previous studies into different compositions of KLTN [17,25]. We note that dielectric dispersion is detected along the entire range of the four observed structural phases, here indicated as P_{cub} , P_{tet} , P_{ort} , P_{rho} for decreasing temperatures.

Comparing $\epsilon(T)$ in *a*, *b*, *c* directions in the (nominally isotropic) cubic phase (T > 281 K, Fig. 2), we observe different values of the permittivity. Especially along the *c* axis, i.e., the growth axis, $\epsilon(T)$ is found almost one order of magnitude smaller than in the other directions [Fig. 2(c)]. In the tetragonal phase, which should be symmetric for fields orthogonal to the *c* axis, i.e., *a* and *b* directions should be equivalent, we observe these to have noticeably different permittivity. A similar response is found also for the other lower-temperature crystalline phases and the related directions, except for the lowest (rhombohedral) phase ($T \leq 50$ K), where anisotropy tends to vanish (insets of Fig. 2). An analogous behavior (a freezing effect of local domain) can be found, e.g., in [46].

B. Directional analysis using Fröhlich entropy

The anisotropy in the response is further analyzed in terms of directionally resolved order-disorder states using Fröhlich entropy. This is based on the scheme developed by Fröhlich who formulated the relationship between the behavior of the real part of the dielectric function versus temperature $\epsilon(T)$ and the entropy variation in the system S = S(T). Under suitable conditions the entropy of a dielectric subject to an external electric field *E* is given, per unit volume, by the relationship [41,42]

$$S(T) = S_0(T) + \frac{E^2 \epsilon_0}{2} \frac{\partial \epsilon_s}{\partial T} = S_0(T) + S_E(T), \qquad (1)$$

where S_0 is the entropy of the system at zero field, ϵ_0 is the vacuum permittivity, ϵ_s is the real part of the dielectric function in the low-frequency limit, and *T* is the absolute temperature. The Fröhlich entropy S_E/E^2 is given by

$$s(T) \equiv \frac{S_E}{E^2} = \frac{\epsilon_0}{2} \frac{\partial \epsilon_s}{\partial T},$$
(2)

which is the field-induced entropy per unit volume and field. Since the electric field selects a specific direction relative to the crystal, s(T) inspects the specific component to the global entropy along the specific direction of the electric field. Equation (2) shows how the application of an electric field increases the

entropy if $\partial \epsilon_s / \partial T$ is positive, while it decreases entropy when it is negative. As pointed out by Fröhlich, in disordered dipolar systems (which we call "dipolar liquidlike" in analogy with [17,41]) an applied electric field creates order since it orients the dipoles otherwise randomly arranged; as a consequence the *s* entropy variation induced by the field ("Fröhlich entropy") is negative. On the contrary, in dipolar ordered systems (which, similarly to the previous term, we call "dipolar solidlike"), the field perturbs the established ordered state and the fieldinduced entropy *s* is positive (see Fig. 1). Therefore, the condition $\partial \epsilon_s / \partial T > 0$ signals a system globally disordered, while when $\partial \epsilon_s / \partial T < 0$ the system is globally ordered.

When phase transitions are considered, on the other hand, an order/order transition manifests itself as a variation of the positive level of S_E ; on the contrary, an order/disorder transition will manifest itself in the permittivity curve as slope reversal, and in $\partial \epsilon_s / \partial T$ as a change from positive to negative sign (Fig. 1) [41-43,47,48]. In Fig. 3 we report the entropy s = s(T) for the *a*, *b*, *c* crystalline directions. In the cubicphase T range [Figs. 3(a)-3(c)], the consistently negative entropy indicates a disorder state for all three measured directions. In turn, in the tetragonal phase a wide interval (206-259 K) of highly disordered state, characterized by the negative value of s(T), is found only in the *a* and *b* directions [Figs. 3(a) and 3(b)], while in the c direction [Fig. 3(c)] the induced entropy is positive for the whole range of the phase. Lowering the temperature, in the orthorhombic phase, disorder is found only in the *a* direction (162–193 K). Finally, in the lowest-temperature rhombohedral phase, the s(T) curve still exhibits a positive sign in the whole T range, for all directions, where it manifests a regular slow decrease with temperature: this response is the signature of a stable ordered phase and characterizes only the rhombohedral symmetry.

III. DISCUSSION

A. Permittivity anisotropy

Permittivity results indicate an anisotropic response with peculiar features. In the highest-temperature region, values of $\epsilon(T)$ as a function of direction evidence anisotropy along the growth axis c also in the nominally centrosymmetric cubic phase (Fig. 2). This suggests a growth-induced alteration in polarization [25], a phenomenon that is commonly associated with internal strain in the pull direction [12]. In specific perovskites this is known to cause the flexoelectric effect, i.e., a macroscopic strain-induced spontaneous polarization which produces the coupling between the strain gradient and polarization [49–51]. Because in a disordered ferroelectric, the strains are inversely proportional to the size of the polar regions [12,52,53], which are nanometric in our case, and the susceptibility has very high values ($\epsilon \sim 10^4$ and more), we expect this effect be relevant. Congruently, the marked macroscopic breaking of centrosymmetry in the cubic phase (Fig. 2), which was recently demonstrated in perovskite materials caused by flexoelectric dynamics [54], can be recognized as a consistent signature of the presence of this phenomenon. Remarkably, in the lowest region of the rhombohedral phase ($T \lesssim 50$ K), when the temperature approaches 0 K, anisotropy tends to vanish (insets of Fig. 2).



FIG. 3. Frölich entropy [following Eq. (2)] s_a , s_b , s_c for, respectively, a (a), b (b), c (c) axes direction, for unit of volume and field square (cooling stage) in the 100–320 K temperature range. The indicated temperatures T'_C , T''_C , T''_C are the transition temperatures shown in Fig. 2. The insets show the temperature range 0–100 K for each respective direction and, congruently with the permittivity behavior, they point out that the anisotropy tend to vanish as the temperature approaches 0 K.

B. Order/disorder state anisotropy

The exploration of the macroscopic order/disorder state of each phase, along the three directions, using Fröhlich entropy differs from the standard diffraction techniques because (i) it is able to give a macroscopic account of the ordering state; (ii) being directly derived from dielectric data, it yields a nearly continuous description of each single phase evolution in temperature, as well as of the transition dynamics [43,47,48]; and (iii) it is highly directional, in that it measures entropy in response to an electric field applied in a specific direction relative to the lattice. Here it is worth underlining that, due to the applied probing field strength (1 V/cm), entropy data describes the thermal evolution of the system without any coercive (several kV/cm) field effects on the PNR size and local structure, as reported by many authors [36,55,56]. The Fröhlich entropy analysis reported in Fig. 3 demonstrates that in the KLTN sample the macroscopic ordering state depends on the long-range symmetry of the phases and, for each phase, on the crystalline directions. Specifically, different (order or disorder) nanosized dipole states can coexist at the same temperature for different directions. Referring to the nominally cubic phase, Fröhlich entropy establishes that the macroscopic PNR-driven dielectric response has features analogous to those of a dipolar liquid (disordered), indicating that PNRs are in a weakly interacting ergodic state, i.e., they act as a liquidlike system being free to reorient in the parent paraelectric matrix [5,12,13].

C. Eight-sites-model macroscopic response

Local symmetries and order transitions of the crystal cell have been studied with x-ray diffuse scattering measurements on prototypical perovskite crystals BaTiO₃ and KNbO₃ by Comès et al. [19,20]. These authors showed that in these compounds an intrinsic disorder is a consequence of the dynamic displacement of the *B* cation on eight energetically equivalent sites along the diagonals of the cubic cell, whose allowed directions of the displacements are dependent on the given crystalline phase (see the scheme in Fig. 4). Specifically, they assessed that: in the cubic phase there are three planes of diffusive scattering normal to the three cubic axes (i.e., disorder is manifested in all directions); in the tetragonal phase there are only two (disorder is manifested in the directions perpendicular to the tetragonal c axis); in the *orthorhombic phase* there is only one plane of diffusion. Finally, in the rhombohedral phase there is no diffusion (only this symmetry shows a behavior of a true crystal): in this phase the displacements are correlated and the B cations are displaced in the same direction. Therefore, the *rhombohedral phase* is found to be the sole ordered and stable phase for the cited perovskites [19,20,27,28]. In our specimen, when the order/disorder states are characterized using Fröhlich entropy (Fig. 3), we observe that the distribution of the macroscopic order/disorder states, in relation both with each single phase and crystal directions, is found to be in remarkable agreement with the microscopic so-called eight-sites model developed by Comès and colleagues. From our results we congruently conclude that the macroscopic ordering state throughout all the four phases is essentially driven by this microscopic eight-sites displacements of the B cations through the mesoscopic response of the system of PNRs: these evolve dynamically and occupy the inner states dictated by the Comès model. In particular, our findings show that also in a complex chemical-substituted disordered ferroelectric, similarly to the nominally pure crystals of Comès, a fully ordered state is achieved at the rhombohedral phase. Recent experiments using neutron scattering in

Crystalline Crystal- phase line direction	Rhombo hedral	Ortho rhombic	Tetra gonal	Cubic
(a)	Order	Disorder	Disorder	Disorder
(b)	Order	Order	Disorder	Disorder
(c)	Order	Order	Order	Disorder

FIG. 4. Scheme of the 12 order/disorder combinations, as a function of the crystalline directions and the long-range symmetry phases, allowed by the model of Comès *et al.* [19,20]. This very same diagram is found to hold also for the KLTN macroscopic response (see Fig. 3).

compound ferroelectrics [46] suggest methods to evaluate the energy associated with the reorientation of the dipolar structures specifically associated with disorder. In this vein it may be possible to shed light on the mechanisms linking the inherent dipoles and the larger scale dipolar systems by analyzing the scaling of the reorientation energy as a function of structure size.

D. Order/disorder transitions

The features of s(T) in correspondence of the phase transitions give information on their nature; specifically the typical order-disorder two-opposite-peaks line shape (as schematically depicted in Fig. 1) is exhibited for the three considered transitions: cubic/tetragonal (*a*, *b*, *c* directions), tetragonal/orthorhombic (*a*, *b* directions), and orthorhombic/rhombohedral (*a* direction). We may therefore conclude that all the long-range symmetry transformations have an order/disorder component. We observe that on the contrary, and in agreement with the Fröhlich treatment, in the case of order/order transformations the s(T) curve exhibits only a positive peak (Fig. 3, *b* and *c* directions).

Significantly, when a single global phase is examined, the comparison of the disorder (s < 0) levels between the three directions shows significant differences. This behavior, in spite of the considerable qualitative agreement, quantitatively does not agree with the eight-sites model, which in fact is solely referred to an ideal ABO₃ lattice. The Fröhlich entropy evolution reported in Fig. 3 shows that, for the KLTN, in the cubic phase, for a given temperature T, the negative amount of s(T) along the *a* direction is higher than that along the b direction and even higher than along the c direction. This contrasts with the nominally centrosymmetric nature of this phase, where the local eight-sites model, assigning to the three directions the same probability for the *B*-cation displacements, predicts the same amount of disorder along all crystalline axes. Analogous considerations hold for the tetragonal phase (disorder levels in a and b directions). Cubic phase anisotropy (Fig. 3) itself gives evidence of a strong symmetry breaking and supports the interpretation of this phenomenon as due to a strain-triggered flexoelectric effect [54]. On the other hand, in the tetragonal phase, the different level of disorder in the a and b directions, perpendicular to the preferential c axis, signals a breaking of the nominal tetragonal symmetry. This symmetry breaking indicates that the crystal responds as a single domain, even though it has not been previously poled nor subjected to a coercive field during measurement.

IV. CONCLUSIONS

Dielectric measurements on a solid-solution disordered ferroelectric single-crystal KLTN are performed in the temperature range 25–320 K, along the three main crystalline directions for different frequencies. Permittivity measurements as a function of temperature $\epsilon(T)$ identify three phase

transitions of a diffusive type at the critical temperatures $T_C = 282$, 202, and 156 K, indicating the transformations of the (nominal) phases cubic-to-tetragonal, tetragonal-to-orthorhombic, and orthorhombic-to-rhombohedral, respectively, upon cooling. Summarizing, we here provide:

(1) An experimental investigation of the macroscopic order-disorder features performed on a chemical substituted ferroelectric solid solution (KLTN single crystals), on the basis of the relationship between the permittivity behavior vs temperature and the entropy variation due to applied probing field (Fröhlich entropy).

(2) The analysis of the macroscopic order/disorder states with respect to the four phases and the three considered crystalline directions, which agree with the 12 combinations predicted by the Comès pattern, finds that behavior agrees with the microscopic eight-sites model, which holds for nominally pure perovskite crystals. Congruently Fröhlich entropy analysis signals that the lowest-temperature rhombohedral phase is the only ordered stable symmetry of our system. Furthermore the analysis of the Fröhlich entropy evolution in correspondence of the transitions fully confirms the orderdisorder character of all the phase transformations.

(3) The analysis of the permittivity response and Fröhlich entropy for the four long-range symmetry phases in relation to the crystalline directions, which points out marked anisotropic characteristics: in a bulk crystal the coexistence of macroscopic states having different ordering in different directions for a single crystalline phase is evidenced in an unpoled sample.

(4) Evidence of the breaking of the macroscopic centric symmetry in the nominally cubic phase in absence of strains due to external electric field or forces. The breaking of macroscopic symmetry is found also in the tetragonal phase with respect to its axis. This "self-poling" is attributed to a microscopic strain-driven flexoelectric effect that is known to play a relevant role in ferroelectrics.

From a methodological point of view, our study underlines how Fröhlich entropy forms a powerful method complementary to the usual experimental techniques based on local investigation at fixed temperatures, e.g., x ray, electron, neutron diffraction, and EXAFS, XANES. Specifically, the ability to analyze response in different macroscopic directions makes it suitable to investigate systems which are inhomogeneous in chemical composition and strain disposition (including anisotropic features), as occurs, e.g., in complex relaxors and ferroelectrics, where order-disorder states typically emerge as a function of temperature, crystal phases, and directions.

ACKNOWLEDGMENTS

This work was partially founded by Grants PHOCOS-RBFR08E7VA and PRIN 2014BFNWZ2 of the Italian Ministry of Research. We thank L. Cattaneo for technical assistance.

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