

Spin-density functional theories and their $+U$ and $+J$ extensions: A comparative study of transition metals and transition metal oxides

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Previous work on the physical content of exchange-correlation functionals that depend on both charge and spin densities is extended to elemental transition metals and a wider range of perovskite transition metal oxides. A comparison of spectra and magnetic moments calculated using charge-only and spin-dependent exchange-correlation functionals as well as their $+U$ and $+J$ extensions confirms previous conclusions that the spin-dependent part of the exchange-correlation functional provides an effective Hund's interaction acting on the transition metal d orbitals. For the local spin density approximation and spin-dependent generalized gradient approximation in the Perdew-Burke-Ernzerhof parametrization, the effective Hund's exchange implied by the spin dependence of the exchange correlation functional is found to be larger than 1 eV. The results indicate that at least as far as applications to transition metals and their oxides are concerned, $+U$, $+J$, and +dynamical-mean-field-theory extensions of density functional theory should be based on charge-only exchange-correlation functionals.

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I. INTRODUCTION

Density functional theory (DFT) is an enormously successful and powerful method for treating the properties of interacting electrons in atoms, molecules, and solids [1]. In its original form, DFT was based on the minimization of a functional of the space-dependent electronic charge density [2], but soon after, extensions to functionals depending on the spin density as well as the charge density were introduced [3,4]. These functionals are not exactly known, but current approximations to the charge-density-only functional such as the local-density approximation (LDA) [3] and the generalized gradient approximation (GGA) [5] provide a very good representation of the electronic properties of many materials. Spin-dependent extensions of the local-density approximation (LSDA) [4,6,7] and of the generalized gradient approximation [8,9] provide important insights into the magnetic properties of many materials. However, the currently available implementations of DFT have difficulty dealing with phenomena such as magnetism and metal-insulator transition [10], which are associated with strong electronic correlations arising from partly filled transition metal d shells or partly filled lanthanide f shells. These difficulties have motivated extensions of the original density functional idea that explicitly include additional interaction terms among physically relevant orbitals [11,12]. Loosely speaking, the extra interactions consist of a term, typically referred to as U , that couples to the square of the total occupancy of the selected orbitals and a set of terms, typically referred to as J , that distinguish different multiplets at fixed total occupancy of the d shells. When the interaction effects are treated within a Hartree-Fock approximation, the extensions are typically referred to as $+U$ and $+J$ methods. When the interaction physics is solved via the dynamical mean-field method, the extension [13–15] is referred to as +DMFT.

A key aspect of correlation physics in transition metals and their oxides is the formation and dynamics of local moments arising from electrons in partially filled transition metal d

shells. Both the spin-dependent DFT (sDFT) methods and the $+U/+J$ extensions of DFT express important aspects of this physics, and a combined sDFT+ $U+J$ methodology seems an attractive approach to strong correlation physics. However, recent studies indicate that this combination produces seemingly unphysical behavior, including an unreasonable J dependence of structural parameters in nickelates [16,17] and of the high-spin/low-spin energy difference in a spin-crossover molecule [18]. A study by Park, Marianetti, and one of us [17] on the rare-earth nickelates led to the conclusion that a source of the difficulty was that the sDFT theories contain an effective J acting on the Ni d states that is already larger than the value considered to be reasonable for transition metals.

In this paper we extend the analysis of Ref. [17] to wider classes of materials and additional observables. We study SrMnO₃ (an antiferromagnetic insulator with a d^3 formal valence that is of current interest for potential multiferroic behavior [19,20]), SrVO₃ (a moderately correlated metal with formal transition metal valence d^1), and elemental Fe. For completeness we also present results for the case of LaNiO₃ previously studied in the literature [17]. We restrict our attention to a Hartree-Fock treatment of the additional correlations, i.e., consider only $+U/+J$ extensions but not +DMFT, although we expect our conclusions will apply to that case also. We compute energy differences between ferromagnetic and antiferromagnetic states as well as magnetic moments. Further, we display the spin-dependent density of states, which provides insight into the issues. Following Ref. [17], we compare results obtained from sDFT theories to results of sDFT+ $U+J$ and DFT+ $U+J$ theories. We find that DFT+ $U+J$ with $J \sim 1\text{--}1.5$ eV reproduces most aspects of sDFT+ U ($J = 0$) calculations, confirming that the conclusions of Ref. [17] apply to a wide range of transition-metal-based materials. We show explicitly that in these systems, the $+U/+J$ extensions of charge-density-only DFT provide a better description of the physical properties than $+U/+J$ extensions of sDFT.

The rest of this paper is organized as follows. Section II presents the formalism we use. Section III presents energy differences between different magnetic states and magnetic moments for ferromagnetic and antiferromagnetic states. Section IV presents an analysis of calculated densities of states. Section V is a summary and conclusion.

II. FORMALISM

A. Theoretical approach

DFT and sDFT and their $+U$ and $+J$ extensions are based on extremization of functionals of charge density $n(\mathbf{r})$, spin density $m(\mathbf{r})$, and the reduced density matrix describing the charge n_a and spin m_a state of designated correlated orbitals labeled by a . The extremization is actually accomplished by solving a Schrödinger equation involving an exchange-correlation potential V_{XC} which depends on $n(\mathbf{r})$ (in the case of DFT) or on $n(\mathbf{r})$ and $m(\mathbf{r})$ (in the case of sDFT) and an additional potential that depends on the orbital occupancies and on the interaction parameters (local d and intra- d orbitals in the usual applications to transition metals and their oxides). Schematically, this additional potential can be written as $V_U(n_a, m_a; U, J)$. An important part of the additional potential is a double-counting correction V_{DC} that removes from V_{XC} the terms that are present in V_U . In the DFT+ methodologies V_{DC} does not have spin dependence, whereas in the sDFT+ methodologies it does.

The known exchange-correlation functionals depend on the full charge (spin) density. The portion pertaining to the designated correlated orbitals cannot be extracted, and the double-counting correction thus cannot be rigorously derived [21]. The double-counting term must be specified by approximate, phenomenologically based arguments. Different forms have been introduced [22,23]. In this study, we use the widely adopted fully localized limit (FLL) form. However, our basic conclusions are independent of the precise form chosen for V_{DC} .

For the case of DFT+ $U+J$, the FLL double-counting correction reads

$$V_{\text{DC}} = U(N_d - \frac{1}{2}) - J(\frac{1}{2}N_d - \frac{1}{2}), \quad (1)$$

where N_d is the total occupancy of designated correlated orbitals (here transition metal d orbitals). U is the Hubbard U , and J is Hund's coupling; they are the standard inputs of DFT+ $U+J$ calculations.

For the case of sDFT+ $U+J$, V_{DC} is spin dependent, and the explicit FLL double-counting form reads

$$V_{\text{DC}}^\sigma = U(N_d^\sigma - \frac{1}{2}) - J(N_d^\sigma - \frac{1}{2}), \quad (2)$$

where N_d^σ is the total occupancy of designated orbitals with spin σ . $N_d = \sum_\sigma N_d^\sigma$. U and J have the same meaning as in Eq. (1). For nonmagnetic cases, $N_d^\sigma = \frac{1}{2}N_d$, and Eq. (2) reduces to Eq. (1).

In our studies we compare two forms of $V_{\text{XC}}(n(\mathbf{r}))$: the LDA [24,25] and the GGA with the Perdew-Burke-Ernzerhof (PBE) parametrization [26]. Correspondingly, for the spin-dependent density functionals, we use the LSDA [24,25] and the spin-dependent GGA with the PBE parametrization (sPBE) [26]. For the $+U$ and $+J$ extensions, we use the

rotationally invariant Hubbard/Hund's corrections introduced by Liechtenstein *et al.* [12].

We note that DFT+ $U+J$ and sDFT+ $U+J$ methods become equivalent if applied to nonmagnetic states [$m(\mathbf{r}) = m_a = 0$]. For magnetic materials, the two methods differ in principle because in the DFT+ $U+J$ case only the spin dependence of the correlated orbitals (here transition metal d orbitals) contributes to the spin dependence of the self-consistent potential felt by electrons. This is because the exchange-correlation potential depends only on the total charge density, so it yields a spin-independent contribution to the potential. In contrast, in the sDFT+ $U+J$ case the spin dependence of the exchange-correlation potential means that the spin polarization of the non- d orbitals also contributes to the spin dependence of the self-consistent potential. However, we shall see that for the situations we consider, this difference is unimportant in practice, probably because the polarization of the noncorrelated orbitals is small. The key difference between different choices of exchange-correlation functionals will be seen to be the magnitude of the spin-dependent term acting on the correlated orbitals.

B. Computational details

We present results for three representative transition metal oxides, cubic SrMnO₃, cubic SrVO₃, and pseudocubic LaNiO₃ (the last compound was previously studied in Ref. [17], and we reproduce the results for comparison), and one representative transition metal, iron. The simulation cell is illustrated in Fig. 1. For transition metal oxides, it consists of two perovskite primitive cells (10 atoms in total) stacked along the [111] direction [Figs. 1(a1) and 1(a2)]. For transition metal, we study body-centered iron [Figs. 1(b1) and 1(b2)]. For both transition metal oxides and transition metals, the computational cell

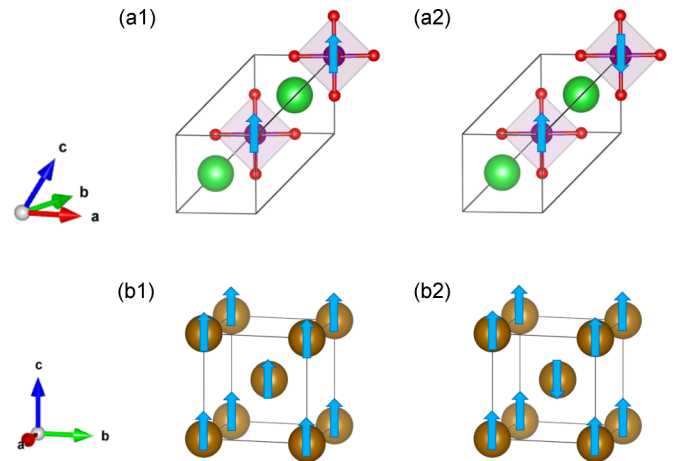


FIG. 1. Computational unit cells showing atoms (balls) and spin alignments (arrows). (a1) and (a2) Simulation cell for transition metal oxides AMO_3 . The two perovskite unit cells are stacked along the [111] direction. The A-site ion ($A = \text{La}$ or Sr in the current study) is the large green ball, and the intermediate-sized purple ball represents the transition metal ($M = \text{Mn}$, V , or Ni in the current study). The small red balls represent oxygen atoms. (b1) and (b2) Simulation cell for body-centered iron. Left: ferromagnetic ordering; right: G-type antiferromagnetic ordering.

can accommodate both ferromagnetic ordering and G -type (two-sublattice Néel) antiferromagnetic ordering. We use experimental lattice constants 3.80 Å (SrMnO₃ [27]), 3.84 Å (SrVO₃ [28]), 3.86 Å (LaNiO₃ [29]), and 2.86 Å (Fe [30]).

The density functional theory calculations [2,3] are performed within the *ab initio* plane-wave approach [31], as implemented in the Vienna Ab initio Simulation Package (VASP) [32]. We employ projector augmented-wave (PAW) pseudopotentials [33,34]. We use an energy cutoff of 600 eV and a $10 \times 10 \times 10$ Monkhorst-Pack grid. A higher energy cutoff (800 eV) and a denser k -point sampling ($12 \times 12 \times 12$) are used to test the convergence, and no significant difference is found. All the calculations allow for the possibility of spin polarization to study different types of long-range magnetic orderings (if they can be stabilized). LDA+ U + J and PBE+ U + J are implemented in VASP as LDAUTYPE = 4, and LSDA+ U + J and sPBE+ U + J are implemented in VASP as LDAUTYPE = 1.

III. MAGNETIZATION AND ENERGY DIFFERENCES

In this section, we consider the ferromagnetic-antiferromagnetic energy differences and local magnetic moments in the ferromagnetic and antiferromagnetic states obtained using different methods. We begin with SrMnO₃, a cubic perovskite antiferromagnetic insulator known experimentally [27,35] to exhibit an approximately high-spin d^3 configuration with a fully spin-polarized t_{2g} shell and a nearly empty e_g shell.

Figure 2(a1) presents the energy difference between G -type (two sublattice Néel) antiferromagnetic and ferromagnetic

states calculated using the DFT+ U + J method with two choices of exchange-correlation potential: the LDA and the generalized gradient approximation in the PBE. Figures 2(a2) and 2(a3) present the local magnetic moments of the ferromagnetic and antiferromagnetic states, respectively. We require that the net on-site interaction is repulsive: this imposes the constraint that $U > 3J$. Therefore, for $J = 1$ eV, we consider only $U > 3$ eV. In the DFT+ U + J method, there is no intrinsic exchange splitting in the exchange-correlation functionals.

We see immediately that the two density functionals, LDA and PBE, give essentially identical results. For pure LDA and PBE ($U = J = 0$), SrMnO₃ is predicted to be nonmagnetic. For moderate $U = 2$ eV, the ground state is antiferromagnetic, and a ferromagnetic state could not be stabilized. For larger $U \gtrsim 4$ eV, the ferromagnetic state is locally stable. For sufficiently large U , the ground state is ferromagnetic. Increasing J favors ferromagnetism. For $U < 4$ eV, the calculated moments are substantially below the experimental value of $2.6 \mu_B/\text{Mn}$ [36]. We therefore believe that to adequately represent the physics of SrMnO₃ within the DFT+ U + J method $U \gtrsim 4$ eV is required. For sufficiently large U and J ($J \gtrsim 1$ eV for $U = 6$ eV or $J \gtrsim 0.8$ eV for $U = 8$ eV), the calculated ground state of SrMnO₃ is ferromagnetic instead of the experimentally observed G -type antiferromagnetic. We therefore believe that $U \lesssim 8$ eV is required within this method.

Magnetism arises from a Hartree treatment of the U interaction, supplemented by the tendency of the J term to favor high-spin states. As U is increased above $U = 4$ eV or J is increased from $J = 0$, the energy of the ferromagnetic state decreases relative to that of the antiferromagnetic state. The change in energy of the two states can be explained in terms of the energy dependence of the relevant exchange processes. To quantitatively describe exchange in cubic SrMnO₃, we use the Hartree-Fock approximation of Slater-Kanamori interaction with three symmetric t_{2g} orbitals:

$$E = \sum_{\alpha} U n_{\alpha\uparrow} n_{\alpha\downarrow} + \frac{1}{2} \sum_{\alpha \neq \beta, \sigma} (U - 2J) n_{\alpha\sigma} n_{\beta\bar{\sigma}} + \frac{1}{2} \sum_{\alpha \neq \beta, \sigma} (U - 3J) n_{\alpha\sigma} n_{\beta\sigma}, \quad (3)$$

where α labels a t_{2g} orbital and σ labels spin and $\bar{\sigma}$ is the opposite spin of σ . Antiferromagnetism results from an inter- t_{2g} superexchange. The superexchange process involves two states: the ground state (two Mn sites, each with three t_{2g} electrons in the high-spin configuration) and the intermediate state (one Mn site with two t_{2g} electrons in a high-spin configuration and the other Mn site with four t_{2g} electrons, three with a majority spin and one with a minority spin). The energy difference between the ground state and the intermediate state is $U + 2J$, and the exchange is therefore $\sim t^2/(U + 2J)$, where t is the inter- t_{2g} hopping. Ferromagnetism comes from double exchange mediated by virtual occupancy of e_g , which is proportional to inter- e_g hopping t' and is favored by on-site Hund's coupling J . The U dependence of double exchange is weak because of the double counting. As U and J are increased, the antiferromagnetic interaction thus weakens, and

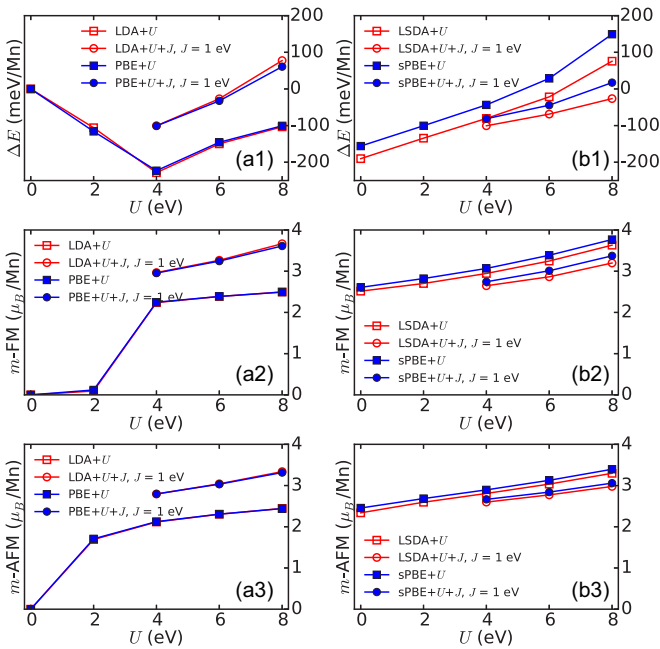


FIG. 2. Comparison of predictions from LDA+ U + J /PBE+ U + J (left) and LSDA+ U + J /sPBE+ U + J (right) methods for ground-state properties of cubic SrMnO₃. (a1) and (b1) Energy difference $\Delta E = E(G) - E(F)$ between ferromagnetic (F) and G -type antiferromagnetic (G) ordering. (a2) and (b2) Magnetic moment per Mn of ferromagnetic state. (a3) and (b3) Magnetic moment per Mn of G -type antiferromagnetic state.

above some critical U_c and J_c , the ferromagnetic interaction dominates.

We next consider the predictions of the spin-dependent density functionals, shown in Figs. 2(b1)–2(b3). We first observe that $\text{LSDA}+U+J$ and $\text{sPBE}+U+J$ produce different results, with $\text{sPBE}+U+J$ favoring ferromagnetism more than $\text{LSDA}+U+J$ and predicting slightly larger moments. Even without the $+U/+J$ corrections, pure LSDA and sPBE stabilize both ferromagnetic and antiferromagnetic states with local magnetic moments close to the experimental values. We interpret this result as indicating that the spin-dependent functionals possess an intrinsic exchange splitting that is large enough to separate the lower and upper Hubbard bands of Mn d states, consistent with previous findings of Ref. [17] in the context of rare-earth nickelates. We also comment that it is widely known [37–40] that sPBE gives a reasonable description of magnetic properties of $\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$ (in particular, the magnetic transition point around $x = 0.5$), while adding U to sPBE impairs the agreement between theory and experiment. However, the physical U on Mn d orbitals is believed to be nonzero (around 4 eV from constrained random-phase-approximation calculations, cRPA [41]). Our results provide a natural explanation: the intrinsic “ J ” in the sPBE already produces a large enough spin splitting, and adding U further splits spin channels, which thus leads to some unphysical results. Using $\text{LDA}+U+J/\text{PBE}+U+J$, we find that a physical range of U is between 4 and 8 eV, which is more consistent with previous cRPA calculations.

As was found in $\text{DFT}+U+J$ calculations, increasing U in $\text{sDFT}+U+J$ decreases the energy difference between the antiferromagnetic and ferromagnetic states, so that for large enough U the ferromagnetic state becomes favored. However, in contrast to $\text{DFT}+U+J$, increasing J in $\text{sDFT}+U+J$ destabilizes the ferromagnetic state. This counterintuitive result is similar to the previous finding of J dependence of the high-spin/low-spin transition point in a spin crossover molecule [18] and is discussed in more detail in the next section.

To further investigate the differences between $\text{DFT}+U+J$ and $\text{sDFT}+U+J$ methods and to understand the robustness of our results across the perovskite family of materials, we present in Fig. 3 the ferromagnetic-antiferromagnetic energy difference $E(G) - E(F)$ of different transition metal oxides, calculated using $\text{sDFT}+U$ (with $J = 0$) and $\text{DFT}+U+J$ (with $J = 1$ eV). We compare SrMnO_3 (an antiferromagnetic insulator with a half-filled t_{2g} shell), SrVO_3 (a moderately correlated metal), and LaNiO_3 (a negative charge-transfer metal).

Figure 3 shows clearly that increasing J brings the $\text{DFT}+U+J$ results into closer agreement with the results of $\text{sDFT}+U$ ($J = 0$) calculations, indicating that in transition metal perovskites the main physical content of the spin-dependent density functionals is an effective J acting on the transition metal d levels. We may define the size of the effective J of the sDFT functionals as the J that needs to be added to make the $\text{DFT}+U+J$ results coincide with the $\text{sDFT}+U$ ($J = 0$) results. The effective J is $\gtrsim 1$ eV and is seen to depend on materials and functionals, being larger for sPBE than for LSDA and larger for LaNiO_3 than for SrMnO_3 .

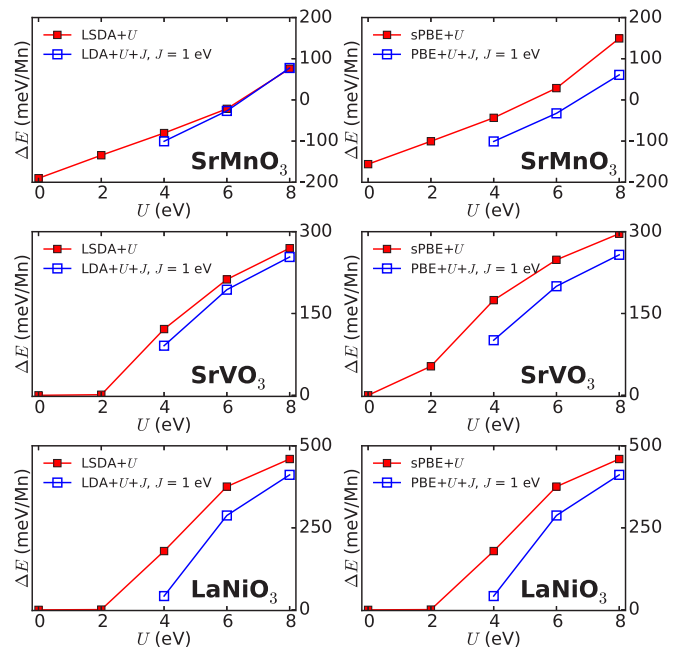


FIG. 3. Comparison of antiferromagnetic-ferromagnetic ground-state energy differences obtained using $\text{sDFT}+U$ (red solid symbols) and $\text{DFT}+U+J$ with $J = 1$ eV (blue open symbols) for the materials indicated. Left: $\text{LSDA}+U$ and $\text{LDA}+U+J$ (with $J = 1$ eV). Right: $\text{sPBE}+U$ and $\text{PBE}+U+J$ (with $J = 1$ eV).

Figure 4 shows the magnetic moments of different transition metal oxides, calculated as in Fig. 3 and presented using

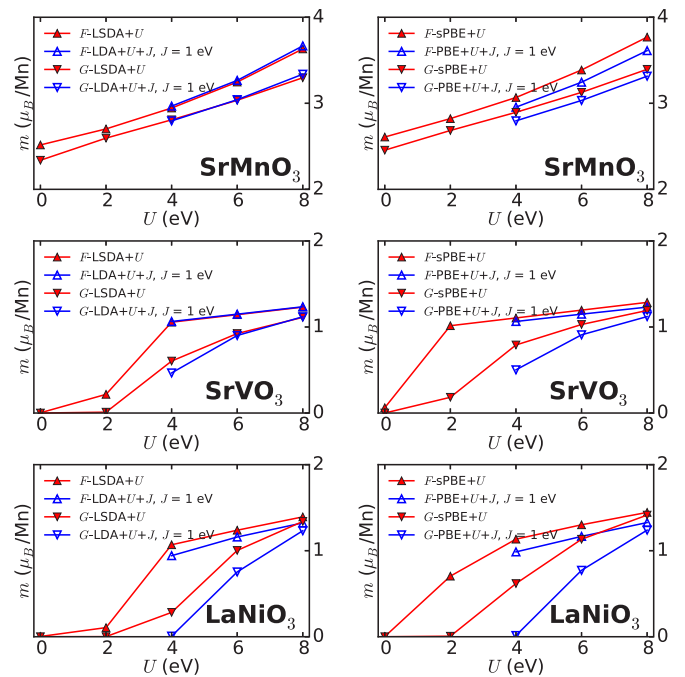


FIG. 4. Comparison of magnetic moments obtained using $\text{sDFT}+U$ (red solid symbols) and $\text{DFT}+U+J$ with $J = 1$ eV (blue open symbols) for the materials indicated. Left: $\text{LSDA}+U$ and $\text{LDA}+U+J$ (with $J = 1$ eV). Right: $\text{sPBE}+U$ and $\text{PBE}+U+J$ (with $J = 1$ eV). The upward triangles are for ferromagnetism. The downward triangles are for G -type antiferromagnetism.

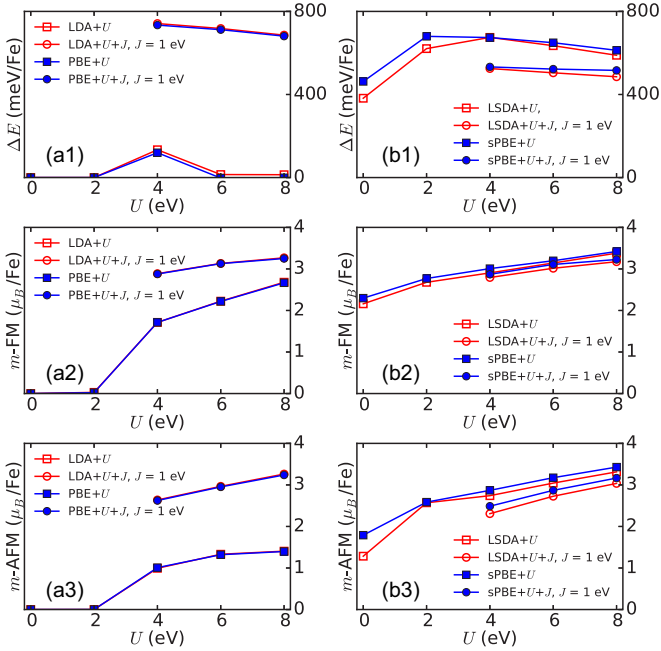


FIG. 5. Comparison of predictions from LDA+ $U+J$ /PBE+ $U+J$ (left) and LSDA+ $U+J$ /sPBE+ $U+J$ (right) methods for ground-state properties of body-centered Fe. (a1) and (b1) Energy difference $\Delta E = E(G) - E(F)$ between ferromagnetic (F) and G -type antiferromagnetic (G) ordering. (a2) and (b2) Magnetic moment per Fe of ferromagnetic state. (a3) and (b3) Magnetic moment per Fe of G -type antiferromagnetic state.

the same conventions. Consistent with Fig. 3, a J equal to or slightly larger than 1 eV must be added in the spin-independent DFT+ $U+J$ calculations to reproduce the magnetic moments calculated from the sDFT+ U ($J = 0$) method.

We next consider Fe, which we study as a representative elemental transition metal. We investigate the extent to which the previous results we obtain from perovskite oxides may apply to transition metals. Figure 5, which uses the same convention as Fig. 2, presents the antiferromagnetic-ferromagnetic energy difference as well as the local moments in ferromagnetic and antiferromagnetic states for body-centered Fe. Experimentally, body-centered iron is ferromagnetic with a magnetic moment of $2.2\mu_B/\text{Fe}$ atom [42]. The left panels of Fig. 5 show that a Hubbard U less than 4 eV in the DFT+ $U+J$ method does not produce a magnetic ground state for iron, which is inconsistent with experiment. For $U \geq 4$ eV, a magnetic ground state is produced with a sizable magnetic moment on Fe ($> 2\mu_B/\text{Fe}$). At $J = 0$, the ferromagnetic and antiferromagnetic states are almost degenerate for a wide range of Hubbard U . As J is increased from $J = 0$, the ferromagnetic state becomes substantially favored in energy. Similar results were also found for SrMnO₃ and other perovskite oxides. The right panels of Fig. 5 show (also as found in perovskite oxides) that LSDA/sPBE alone ($U = J = 0$) suffices to split the spin and yield a sizable magnetic moment ($\sim 2.2\mu_B/\text{Fe}$ for ferromagnetism and $\sim 1.5\mu_B/\text{Fe}$ for antiferromagnetism), which agrees well with the experiment [42]. Increasing U

impairs the agreement, and increasing J in sDFT+ $U+J$ destabilizes ferromagnetism.

IV. DENSITY OF STATES

In this section, we study the density of states (DOS) obtained using different exchange-correlation functionals at $U = 0$ and 6 eV and Hund's coupling $J = 0$ and 1 eV. For ease of interpretation, we present results obtained in the ferromagnetic state. It is useful to analyze the results in terms of the standard phenomenological Slater-Kanamori interaction, which for simplicity we discuss for the simple case of a half-filled fully spin-polarized orbitally symmetric t_{2g} shell treated in the Hartree-Fock (+ $U+J$) approximation (this is a simple model for cubic SrMnO₃). In this case the spin-up/spin-down potential for each t_{2g} orbital arising from this interaction is

$$V_\alpha^\uparrow = 2U - 6J, \quad (4)$$

$$V_\alpha^\downarrow = 3U - 4J, \quad (5)$$

where α labels a t_{2g} orbital (the derivation is in the Appendix). Taking into account the double-counting terms, in the DFT+ $U+J$ method, we have

$$V_\alpha^\uparrow = 2U - 6J - V_{\text{DC}}, \quad (6)$$

$$V_\alpha^\downarrow = 3U - 4J - V_{\text{DC}}, \quad (7)$$

where the double-counting correction V_{DC} is spin independent. Therefore, in the DFT+ $U+J$ case the energy difference between the spin-up and spin-down potentials is

$$|V_\alpha^\uparrow - V_\alpha^\downarrow| = U + 2J. \quad (8)$$

However, in the sDFT+ $U+J$ method, we have

$$V_\alpha^\uparrow = 2U - 6J - V_{\text{DC}}^\uparrow, \quad (9)$$

$$V_\alpha^\downarrow = 3U - 4J - V_{\text{DC}}^\downarrow, \quad (10)$$

where the double-counting correction V_{DC}^σ is spin dependent. Therefore, in the sDFT+ $U+J$ case, using the FLL double-counting scheme (2), the energy difference between the spin-up and spin-down potentials is

$$|V_\alpha^\uparrow - V_\alpha^\downarrow| = |U + 2J - Jm| = |U - J|, \quad (11)$$

where $m = N_d^\uparrow - N_d^\downarrow$ is the magnetization of the d orbitals and in this simple t_{2g} model $m = 3$. This shows that adding J in sDFT+ $U+J$ reduces the spin splitting.

Next, we present the DFT-computed densities of states in which we find the peaks that are attributable to the d levels. The energy differences between the majority and minority spin channels then reflect the values of U and J . Figure 6 presents the pure LSDA densities of states for ferromagnetic SrMnO₃ and elemental Fe. We see that both materials exhibit a DOS peak at an energy of ~ 2 eV above the Fermi level in the minority-spin channel and a d -related peak in the majority-spin channel slightly below the Fermi level. In SrMnO₃ the d states visible at energies ~ -4 to -6 eV arise from admixture with oxygen orbitals. We define the spin

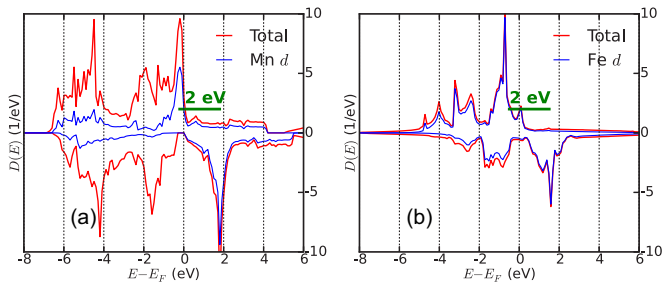


FIG. 6. Density of states of ferromagnetic ordering calculated using the LSDA ($U = J = 0$) method. (a) Cubic SrMnO₃; (b) iron. Positive and negative y -axis curves show majority and minority densities of states, respectively. The horizontal green line and the number provide estimates of the spin splitting.

splitting as the peak-to-peak energy difference between the majority and minority spin d contributions to the densities of states and indicate it by the thick green line. For SrMnO₃ the peak-to-peak splitting of the d bands provides an estimate of intrinsic Hund's coupling implied by the spin dependence of the exchange-correlation potential. From the Slater-Kanamori model, with $U = 0$ the spin splitting is $2J$ [Eq. (8)], so the LSDA-calculated separation indicates an effective J of about 1 eV. In Fe the interpretation is complicated by the higher occupancy of the d band (the calculated N_d of Fe is very close to the nominal occupancy of 6).

Figure 7 presents the density of states for ferromagnetic SrMnO₃ calculated using different exchange-correlation functionals. The majority-spin density of states has a significant peak between -2 and 0 eV, but this peak has only modest d content. It arises from oxygen p states, with modest p - d hybridization. The main portion of the occupied majority-spin

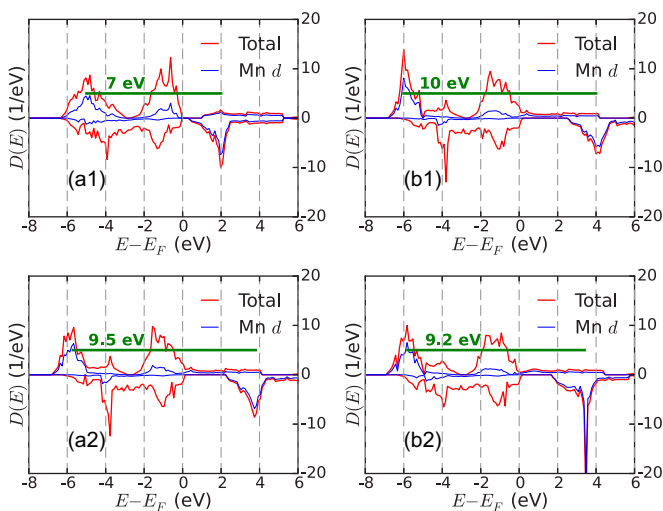


FIG. 7. Total density of states (thick red line) and Mn d -projected density of states (thin blue line) of cubic SrMnO₃ in the ferromagnetic state. Positive and negative y -axis curves show the majority and minority densities of states, respectively. (a1) LDA+ U , (a2) LDA+ U + J , (b1) LSDA+ U , and (b2) LSDA+ U + J ; $U = 6$ eV and $J = 1$ eV. Horizontal green lines and numbers provide estimates of spin splitting obtained from peak-to-peak separation of majority- and minority-spin d density of states peaks.

d states occurs much further below the Fermi level, at an energy of -5 to -6 eV, with the precise energy depending on the exchange-correlation functional. In mathematical terms the double-counting correction shifts the mean energy of the d states down to this low energy (a level repulsion due to hybridization with the oxygen p states also plays a role).

The spin splitting is defined as in the previous case and is again shown as a horizontal green bar in Fig. 7. Comparison of Figs. 7(a1) (LDA+ U) and 7(a2) (LDA+ U + J) shows that adding a J to LDA+ U calculations increases the spin splitting by 2.5 eV, slightly larger than $2J$. The difference arises from a small occupancy of e_g states.

Comparing Figs. 7(a1) (LDA+ U) and 7(a2) (LDA+ U + J) to Fig. 7(b1) (LSDA+ U) reveals that even with no added J , the LSDA+ U method produces a larger spin splitting than the LDA+ U + J method with $J = 1$ eV: in other words, the spin dependence of the exchange-correlation functional corresponds to an effective $J \gtrsim 1$ eV on the transition metal d orbitals. This is consistent with the estimate of intrinsic J from the pure LSDA spectrum (Fig. 6).

Inspection of Fig. 7(b2) (LSDA+ U + J) reveals that adding a J to the LSDA+ U reduces the spin splitting, in contrast to the effect of adding a J to the LDA+ U calculation. This is consistent with the analysis of our simple t_{2g} model [see Eq. (11)]. We believe this counterintuitive J dependence is a general feature of sDFT+ U + J methods, as was previously noted in the study of a spin-crossover molecule [18]. The underlying origin is that the spin dependence of the double-counting correction overcompensates for the Hartree shift produced by J , which is consistent with the trend that increasing J in sDFT+ U + J destabilizes ferromagnetism [see Eq. (11)].

Figure 8 presents the density of states of body-centered Fe. Qualitatively, the variation of spin splittings predicted by different exchange-correlation functionals (LDA+ U ,

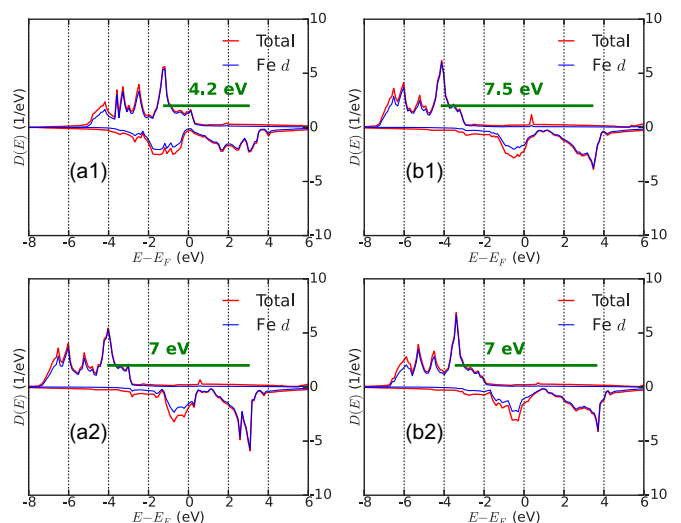


FIG. 8. Density of states of body-centered iron with ferromagnetic ordering, calculated using different exchange-correlation functional approximations: (a1) LDA+ U , (a2) LDA+ U + J , (b1) LSDA+ U and (b2) LSDA+ U + J ; $U = 6$ eV and $J = 1$ eV. Horizontal green lines and numbers provide estimates of spin splitting.

LDA+ U + J , LSDA+ U , and LSDA+ U + J) is very similar to that found for cubic ferromagnetic SrMnO₃.

V. CONCLUSIONS

In this paper we have studied energetics and local magnetic moments of representative transition metal oxides (the antiferromagnetic Mott insulator SrMnO₃, the moderately correlated metal SrVO₃, and the negative charge-transfer insulator LaNiO₃) and an elemental transition metal (Fe) to gain further insight into the physics of spin-dependent density functional theories and their + U and + J extensions previously noted in Ref. [17]. In these materials, the only states with significant spin polarization are the transition metal d states, and important aspects of the physics are controlled by an exchange splitting of spin configurations of these states. For a transition metal ion in free space, the exchange splitting is conventionally described by a Hund's coupling parameter J , and we interpret the exchange splitting found in our calculations as an effective J_{eff} , which may have contributions from the spin dependence of the density functional and from an explicitly added interaction term.

The results are similar for all materials studied. The spin-dependent density functionals are found to encode an exchange splitting between different spin states of the transition metal d orbitals. This spin splitting is larger in the spin-dependent PBE functional than that in the local spin density functional but in both cases is at least 2 eV. Comparison to results of the + U + J methods suggests that the J_{eff} corresponding to the spin-dependent density functional is about 1 eV. This value is larger than the range of 0.6–1 eV which is generally accepted as a reasonable estimation for transition metals and their oxides, suggesting that the present implementations of the spin-polarized DFT methods may overestimate the effects of spin polarization in transition metal d orbitals.

We also found that adding an explicit Hund's coupling J to the spin-dependent DFT functional (sDFT+ U + J) reduces the calculated exchange splitting below its $J = 0$ value, whereas adding a J to the charge-density-only DFT functional (DFT+ U + J) increases the splitting as expected. This counterintuitive J dependence in the sDFT+ U + J method arises from the spin dependence of the double-counting correction. The effect was previously noted in the study of LaNiO₃ [17] and was carefully documented in the study of a spin-crossover molecule [18]. Our results provide further support for the previous conclusions [17,18] that while spin-dependent density functionals provide successful descriptions of many materials, caution is needed in their applications to transition metals and their oxides. In particular, for these compounds it is advantageous to base beyond-density-functional-analyses such as the + U + J and +DMFT on spin-independent density functionals (LDA or the PBE-parametrized GGA functional) because the physical meaning of U and J in the parametrization is more clear and the value of J implicit in present implementations of the spin-dependent exchange-correlation functionals is likely to be too large. Whether the value of J implicit in the spin-dependent exchange-correlation functionals increases or decreases with correlated orbitals changing from $3d$ (considered in this study) to $4d$ and $5d$ is a very interesting question which deserves further investigation in future work.

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APPENDIX: DERIVATION OF EQUATIONS (4) AND (5)

In this Appendix, we derive Eqs. (4) and (5). The rotationally invariant Slater-Kanamori (SK) interaction is

$$\begin{aligned} \hat{H}_{\text{SK}} = & \sum_{\alpha} U \hat{n}_{\alpha\uparrow} \hat{n}_{\alpha\downarrow} + \frac{1}{2} \sum_{\alpha \neq \beta, \sigma} (U - 2J) \hat{n}_{\alpha\sigma} \hat{n}_{\beta\bar{\sigma}} \\ & + \frac{1}{2} \sum_{\alpha \neq \beta, \sigma} (U - 3J) \hat{n}_{\alpha\sigma} \hat{n}_{\beta\sigma} \\ & - \sum_{a \neq b} J (d_{a\downarrow}^{\dagger} d_{b\uparrow}^{\dagger} d_{b\downarrow} d_{a\uparrow} + d_{b\uparrow}^{\dagger} d_{b\downarrow}^{\dagger} d_{a\uparrow} d_{a\downarrow} + \text{H.c.}), \end{aligned} \quad (\text{A1})$$

where α labels a d orbital and σ labels a spin and $\bar{\sigma}$ is the opposite spin of σ . Within the Hartree-Fock approximation and for collinear spin states, the spin-flip and pair-hopping terms (proportional to J) in Eq. (A1) vanish, and the operator $\hat{n}_{\alpha\sigma}$ is reduced to an occupancy $n_{\alpha\sigma}$. Then we obtain an energy functional:

$$\begin{aligned} E = & \sum_{\alpha} U n_{\alpha\uparrow} n_{\alpha\downarrow} + \frac{1}{2} \sum_{\alpha \neq \beta, \sigma} (U - 2J) n_{\alpha\sigma} n_{\beta\bar{\sigma}} \\ & + \frac{1}{2} \sum_{\alpha \neq \beta, \sigma} (U - 3J) n_{\alpha\sigma} n_{\beta\sigma}. \end{aligned} \quad (\text{A2})$$

The potential associated with a given orbital α and a given spin σ is $V_{\alpha}^{\sigma} = \frac{\partial E}{\partial n_{\alpha\sigma}}$. Therefore we have

$$\begin{aligned} V_{\alpha}^{\uparrow} = & \frac{\partial E}{\partial n_{\alpha\uparrow}} = U n_{\alpha\downarrow} + \sum_{\beta \neq \alpha} (U - 2J) n_{\beta\downarrow} \\ & + \sum_{\beta \neq \alpha} (U - 3J) n_{\beta\uparrow}, \end{aligned} \quad (\text{A3})$$

$$\begin{aligned} V_{\alpha}^{\downarrow} = & \frac{\partial E}{\partial n_{\alpha\downarrow}} = U n_{\alpha\uparrow} + \sum_{\beta \neq \alpha} (U - 2J) n_{\beta\uparrow} \\ & + \sum_{\beta \neq \alpha} (U - 3J) n_{\beta\downarrow}. \end{aligned} \quad (\text{A4})$$

For the simple model of a half-filled fully spin-polarized orbitally symmetric t_{2g} shell, we have three orbitals, $\alpha = d_{xy}, d_{xz}, d_{yz}$, and the occupancy is $n_{\alpha\uparrow} = 1, n_{\alpha\downarrow} = 0$ for each orbital α . Therefore we have

$$V_{\alpha}^{\uparrow} = 0 + 0 + 2(U - 3J) = 2U - 6J, \quad (\text{A5})$$

$$V_{\alpha}^{\downarrow} = U + 0 + 2(U - 2J) = 3U - 4J, \quad (\text{A6})$$

which are Eqs. (4) and (5).

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