From soft harmonic phonons to fast relaxational dynamics in CH₃NH₃PbBr₃

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The lead-halide perovskites, including $CH_3NH_3PbBr_3$, are components in cost effective, highly efficient photovoltaics, where the interactions of the molecular cations with the inorganic framework are suggested to influence the electronic and ferroelectric properties. $CH_3NH_3PbBr_3$ undergoes a series of structural transitions associated with orientational order of the CH_3NH_3 (methylammonium) molecular cation and tilting of the $PbBr_3$ host framework. We apply high-resolution neutron scattering to study the soft harmonic phonons associated with these transitions, and find a strong coupling between the $PbBr_3$ framework and the quasistatic CH_3NH_3 dynamics at low energy transfers. At higher energy transfers, we observe a $PbBr_6$ octahedra soft mode driving a transition at 150 K from bound molecular excitations at low temperatures to relatively fast relaxational excitations that extend up to $\sim 50-100$ meV. We suggest that these temporally overdamped dynamics enables possible indirect band gap processes in these materials that are related to the enhanced photovoltaic properties.

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Organic-inorganic, hybrid perovskites (OIPs) are materials based upon an inorganic perovskite host framework with an organic molecular cation occupying the interstitial space. These materials have been studied for quite some time [1], but interest has recently surged owing to their use in photovoltaic devices and to possible ferroelectricity [2]. While earlier work centered on Sn-based OIPs as possible sensor materials [3,4], recent interest has been focused towards Pb-based OIPs, due to their potential advantages in inexpensive photovoltaic devices, with efficiencies of the order of 20% [5]. We apply neutron scattering to study the coupled dynamics of the host PbBr₃ framework and the methylamonium (MA) cation in CH₃NH₃PbBr₃, and discuss the possible relation with the photovoltaic properties. We map out the soft phonons and low-temperature quasistatic molecular rotations, and show that the low-temperature, harmonic fluctuations cross over to temporally overdamped dynamics at high temperature.

The OIPs are composed of two sublattices [Fig. 1(a) for CH₃NH₃PbBr₃]: the inorganic sublattice, consisting of a fully corner-bonded framework of octahedra (PbBr₆⁻); and the organic sublattice consisting of the MA molecular cation, (CH₃NH₃⁺). At high temperatures, the structure of $CH_3NH_3PbBr_3$ is cubic (space group Pm3m), and below 235 K has a tetragonal structure in symmetry I4/mcm [6,7]. At 150 K, a transition to an unknown structure (believed to be incommensurate [8]) occurs followed by further distortion to orthorhombic (Pnma) at 148 K [9–12]. Neutron diffraction measurements have suggested that these transitions originate from tilting of the PbBr₆ octahedra and orientational ordering of the MA cation [6,13]. We note that despite minor differences in the phase diagrams, for all MAPb X_3 (X = Cl, Br, I), a transition to an ordered phase, in which the octahedra are tilted and the MA cations have relatively well-defined orientations, occurs in the temperature range 150–175 K [6,9,13,14].

The MA orientations have been implicated as influencing charge separation through a bulk polarization [15]. This, in turn, has been suggested to enhance the photovoltaic properties [16]. To this end, we apply neutron inelastic scattering to study the dynamics of the MA molecular cation, the PbBr₃ framework, and the coupling between the two sets of dynamics. The experiments were performed on the MARI (ISIS, UK), NEAT (HMI, Germany), IN4, and IN10 (ILL, France) spectrometers. Samples of CH₃NH₃PbBr₃ and CD₃ND₃PbBr₃ were synthesized as discussed elsewhere. [6] The scattering measurements of the harmonic vibrations were compared against Castep/Aclimax first-principles calculations [17,18].

The low-temperature phase of CH₃NH₃PbBr₆ consists of a PbBr₃ framework with a MA molecular cation hydrogen bonded to the Br atoms [Fig. 1 (a)] [6]. Because the neutron incoherent cross section of H is over an order of magnitude larger than those of the non-H atoms, the angular-independent neutron scattering intensity is dominated by H dynamics. We exploit this in Fig. 1(b) which plots the average meansquared displacement of the hydrogen atoms. This is extracted from an energy integrating "fixed window" scan integrating over energy transfers of $\pm 1 \ \mu eV$ (further details in the Supplemental Material [19]). The temperature scans illustrate an onset of fluctuations near 50 K where dynamic processes enter the resolution of the IN10 spectrometer. With increasing temperature, this is followed by an abrupt change at the I-II and II-III structural transitions of 148 and 154 K, respectively, from where the mean-squared displacement evolves continuously with increasing temperature.

The discontinuity in $\langle u^2 \rangle$ at 148 K (phases I and II) has been previously noted with neutron diffraction [6], calorimetry, and NMR [9]. Based on comparisons to MAPbCl₃ and TMAGeCl₃ (TMA = tetramethylammonium), it is likely that this is a transition to an intermediate incommensurate phase



FIG. 1. (Color online) (a) The structure of CH₃NH₃PbBr₃ illustrating the MA molecular cations in the PbBr₃ framework at 5 K. (b) shows the mean-squared displacement of the hydrogen atoms $(\langle u^2 \rangle)$ extracted from an IN10 elastic window scan as a function of temperature and (c) plots a representative quasielastic scan taken on IN10 at 45 K.

separating orthorhombic (phase I) and tetragonal (phase III) states [8,20,21]. There is no observable discontinuity in the mean-squared displacement of the hydrogen atoms associated with the III-IV (tetragonal to cubic) transition on this time scale $(1/\tau \sim \delta E = \pm 1 \ \mu eV)$. This indicates the low-energy molecular dynamics are similar over this temperature range.

The time scale of the dynamics of the MA ion can be probed from energy analyzing neutron quasielastic scattering (QENS). Figure 1(c) shows a representative QENS scan taken on IN10 at 45 K displaying a two-component line shape with a sharp, resolution-limited elastic component (dashed line) and a broader-than-resolution relaxational component, fit by a Lorentzian (solid curve). The broader Lorentzian is indicative of molecular dynamics from which a time scale (τ) can be extracted from the energy width ($\gamma \sim 1/\tau$). We now discuss the quasielastic components as a function of both momentum and temperature.

Figure 2 plots the energy (*E*), momentum (*Q*), and temperature (*T*) dependence of the quasielastic scattering in the CH₃NH₃PbBr₃. *T*- and *Q*-dependent parameters were extracted by fitting each energy scan to the sum of an elastic and dynamic component $[S(Q, E) = I_{el} + I_{qe} = I_{el}\delta(E) + I_{dyn}/[1 + (E/\gamma)^2]]$ convolved with the resolution. Following previous studies, we define the elastic incoherent structure factor (EISF) from the normalized intensity $I_{EISF} = I_{el}/(I_{el} +$



FIG. 2. The quasielastic linewidth (a), momentum dependence (b), and the elastic fraction (c) are plotted for $CH_3NH_3PbBr_3$. Sample quasielastic scans and fits are shown from the IN10 backscattering spectrometer (d) and the NEAT direct-geometry time-of-flight spectrometer (e).

 I_{dyn} [22]. Example fits are displayed in Figs. 1(c), 2(d), and 2(e).

The Q dependence of the EISF is shown in Fig. 2(b), and decays monotonically with O at all temperatures studied [shown in Fig. 2(e)]. As noted in Ref. [23], the Q dependence of the EISF is sensitive to the real space nature of the equilibrium molecular motions; however, distinguishing features in the models are only present at momentum transfers beyond the range accessible in this experiment, limited by strong nuclear Bragg peaks, resolution, and kinematics of neutron spectroscopy. We have therefore followed Ref. [22] and fit the results to a model for isotropic molecular reorientations with the temperature dependence given by two components: one where the molecular cation is bound, and a second fraction (α) that is fluctuating and contributes to the inelastic scattering component. The cross section, therefore, has the form $I_{\text{total}} =$ $(1 - \alpha)I_{\text{bound}} + \alpha(I_{\text{el}} + I_{\text{qe}})$. "Isotropic" molecular reorientations give a dependence of $I_{\text{EISF}}(Q) = \frac{\sin(Qr)}{Qr}$, with r the radius of the molecular cation and for simplicity we have taken r to be the C-H distance of 1.05 Å from the methyl group. We do not consider whole body rotations which would require a larger effective radius including the C-N distance [24] which is inconsistent with the momentum dependence. The bound component (I_{bound}) is taken to be a constant independent of momentum transfer. Therefore, we consider rigid motions around a fixed equilibrium position where the molecular cations are orientationally ordered as expected based on previous structural analysis [6]. The fraction of MA ions in motion is given by α in Fig. 2(c). A fit to an activation form $\alpha \propto e^{-E_{\rm B}/T}$ gives $E_{\rm B} = 51 \pm 5$ K which characterizes the energy required to activate molecular reorientation which matches the onset temperature of observable displacements of the hydrogen atoms in Fig. 1(a). The *T* dependence of α [Fig. 2(c)] implies a phase transition at ~150 K when approximately 50% of the molecules become "unbound."

The time scale of the molecular reorientations is presented in Fig. 2(a) plotting γ vs *T* below 150 K. The molecular motions above 150 K were too fast for the dynamic ranges on both spectrometers and are discussed further below. The solid curve is a fit to an Arrhenius form $\gamma = \gamma_0 e^{-E_0/T}$ with $\gamma_0 = 323 \pm 20$ K and $E_0 = 2.3 \pm 0.1$ meV. The deviation at low temperatures is due to the resolution limit of $\sim 2 \mu eV$ on IN10 from which all data appear resolution limited within error as the dynamics leave the spectrometer window. Our analysis reveals two energy scales: one lower energy E_B characterizing the energy scale to allow molecular motion and a second higher activation energy γ_0 associated with the energy scale of the molecular reorientation.

An unusual feature of the data, shown in Fig. 2(d), is the slow quasistatic component which enters into the IN10 time window near the structural transition at ~150 K. Panel (d) shows quasielastic data at 140 and 165 K. The 140 K data shows no inelastic contribution and is well described by the resolution function (dashed line). The T = 165 K shows a quasistatic contribution with a linewidth of ~5 μ eV with slightly higher temperatures showing this component to be out of the time window on IN10. Based on infrared work on OIPs [25] and also quasielastic scattering in nitromethane [26], we expect the lower-temperature, quasistatic dynamics that is onset above ~50 K to be dominated by slowing methyl-group rotations, whereas the dynamics observed just below the 150 K transition are likely due to the NH₃⁺ group, which are hydrogen bonded to the Br framework at lower temperatures.

Having discussed the molecular dynamics, we now investigate the coupling between the molecular rotations and the inorganic framework. Figure 3(a) shows *Q*-integrated and powder-averaged data taken on the IN4 spectrometer mapping out the *T* dependence of the low-energy modes. The plot shows a series of intense modes above an energy transfer of ~ 10 meV which disappear at the structural transition at 150 K. The modes, which are sharp in energy, are replaced by a broad relaxational line shape centered at the elastic position. This onset of fast relaxational dynamics may be the origin of the short-range distortions of the cubic framework observed in pair distribution function studies of bulk and thin film materials [10,27].

Concomitant with this rapid decay, a softening of a weaker band of excitations is observed at \sim 5 meV which is tracked by the solid points in Fig. 3(a) obtained by fitting each spectra to a *T*-dependent Gaussian. To assign this band to either molecular motions (MA) or the host lattice (PbBr₆ octahedra), we have performed a scan on the MARI spectrometer on both

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FIG. 3. (Color online) (a) displays inelastic scans taken on the IN4 spectrometer. (b) and (c) show momentum integrated scans on MARI on fully protonated and deuterated samples. (d) shows a plot of the square of the PbBr₆ octahedra phonon frequency as a function of T.

protonated and deuterated variants with the results displayed in *Q*-integrated energy scans in panels (b) and (c). On deuteration, two effects are observed; first, a comparative reduction of peak intensities at ~10 meV, due to the lack of H atoms and the corresponding smaller cross section; second, a significant shift to lower energies owing to the heavier mass of D over H. For these two reasons, we assign the 10-meV modes to molecular motions and the 5-meV band to PbBr₃ framework motions (not involving MA motion). We corroborate this conclusion below by comparing MARI spectra with results from first-principles calculations.

Figure 3(d) illustrates the frequency squared of the PbBr₆ octahedra mode as a function of temperature with the solid line a fit to the "Cochran law" [28] $E_0^2 \propto (T - T_c)$, where T_c is the structural transition. This follows other known zone-boundary transitions such as KMnF₃ [29] and SrTiO₃ [30], except that we do not observe a high-temperature recovery; instead, the spectra are dominated by the relaxational dynamics discussed above which masks any observable recovery of the soft mode. A strong deviation from linearity is observed at ~50 K, which is the temperature scale where molecular motion is onset (Fig. 1). The IN4 results illustrate a coupling between



FIG. 4. (Color online) High-energy data taken on MARI in several configurations with different energy resolutions, compared to calculations. (a)–(c) CASTEP calculations showing where a finite phonon cross section exists. (d)–(f) MARI data taken at low temperatures in the orthorhombic phase. (g)–(i) show MARI data taken in the tetragonal phase at 200 K. The sharp, harmonic modes are replaced by quasistatic fluctuations extending up to high energies.

the dynamics of both molecular (organic) and framework (inorganic) sublattices.

The MARI data at T = 5 and 200 K is compared to Castep first-principles calculations in Figs. 4(a)–4(c). These calculations confirm the deuteration analysis (Fig. 3) that the low-energy mode that softens is associated with Pb-Br motions, and in particular, octahedra tilting. The result is also consistent with previous work on the phase transitions CsPbCl₃ which were found to result from soft tilt modes of the PbCl₆ octahedra [19,31–34]. The more intense, higher-energy modes that disappear at T_c are related to H motion. CH₃NH₃PbBr₃ therefore represents a case where a displacive PbBr₆ octahedra

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transition is coupled to an order-disorder MA transition, corresponding to melting of the orientational order.

We now discuss possible relations with electronic properties important for photoluminescence. Measurements and calculations of the frequency-dependent dielectric response and electronic gap energies provide evidence that molecular vibrations are coupled to the electronic properties [16,35] and this has been further suggested by recent QENS results on the PbI₃ framework based compounds [36,37]. Further indications of the importance of the molecular vibrations come from the temperature-dependent lifetime of the photoluminescence that shows an emission at 1.62 eV with a FWHM of 103 meV, sharpening at \sim 150 K where molecular orientational order occurs [38]. While the energy scale of the molecular dynamics would be expected to be too low (less than meV displayed in Fig. 1) to influence the properties of the photoluminescence at this energy scale, the broad relaxational dynamics displayed in Fig. 4 do have a similar energy scale to the changes observed in photoluminescence. While the molecular cation is not expected to directly contribute to the electronic band structure [39], it is thought to indirectly influence the electronic band gap through coupling to the PbBr₃ framework. As in many Pb perovksites, the lone electron pair on the Pb ion is strongly influential to the electronic band properties [40]. The strong coupling between the PbBr₃ framework and the MA molecular motions may also affect band structure details as predicted in Ref. [16].

The potential impact on the electronic properties is further suggested in Ref. [41] where it is noted that MA orientational driven indirect band gap processes may exist introducing a "dynamical band gap" discouraging carrier recombination. The coupled relaxational dynamics observed here between the PbBr₃ framework and the MA molecule cation may provide a channel for such processes. We note that a similar multiphonon process has been proposed for the luminescence in KCl [42].

In summary, we have reported a neutron inelastic scattering study of the coupled molecular and framework dynamics in $CH_3NH_3PbBr_3$. We observe quasistatic molecular fluctuations coupled to a soft mode associated with tilting of the $PbBr_3$ framework. The low-temperature harmonic fluctuations cross over to fast overdamped relaxational dynamics at higher temperatures. We have discussed possible coupling between these dynamics and the photoluminescence spectra.

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- [1] R. Wyckoff, Am. J. Sci. 16, 349 (1928).
- [2] A. Kojima, K. Teshima, Y. Shirai, and T. Miyasaka, J. Am. Chem. Soc. 131, 6050 (2009).
- [3] D. B. Mitzi, K. Chondroudis, and C. R. Kagan, IBM J. Res. Dev. 45, 29 (2001).
- [4] D. B. Mitzi, Prog. Inorg. Chem. 48, 1 (2007).
- [5] N. G. Park, Phys. Chem. Lett. 4, 2423 (2013).

- [6] I. P. Swainson, R. P. Hammond, C. Souliere, O. Knop, and W. Massa, J. Solid State Chem. 176, 97 (2003).
- [7] A. Maalej, M. Bahri, Y. Abid, and N. Jaidane, J. Phys. Chem. Lett. 6, 681 (2015).
- [8] K. Futterer, R. L. Withers, T. R. Welberry, and W. Depmeier, J. Phys.: Condens. Matter 7, 4983 (1995).

- [9] O. Knop, R. E. Wasylishen, M. A. White, T. S. Cameron, and M. J. M. V. Oort, Can. J. Chem. 68, 412 (1990).
- [10] R. J. Worhatch, H. Kim, I. Swainson, A. L. Yonkeu, and S. J. Billinge, Chem. Mater. 20, 1272 (2008).
- [11] Y. Kawamura, H. Mashiyama, and K. Hasebe, J. Korean Phys. Soc. 71, 1694 (2002).
- [12] H. Mashiyama, Y. Kawamura, H. Kasano, T. Asahi, Y. Noda, and H. Kimura, Ferroelectrics 348, 182 (2007).
- [13] L. Chi, I. P. Swainson, L. Cranswick, J. H. Her, P. Stephens, and O. Knop, J. Solid State Chem. **178**, 1376 (2005).
- [14] H. Mashiyama, Y. Kawamura, and Y. Kubota, J. Korean Phys. Soc. 51, 850 (2007).
- [15] C. Quarti, E. Mosconi, and F. D. Angelis, Chem. Mater. 26, 6557 (2014).
- [16] F. Zheng, H. Takenaka, F. Wang, N. Z. Koocher, and A. M. Rappe, J. Phys. Chem. Lett. 6, 31 (2015).
- [17] M. Segall, P. Lindan, M. Probert, C. Pickard, P. Hasnip, S. Clark, and M. Payne, J. Phys.: Condens. Matter 14, 2717 (2002).
- [18] A. Ramirez-Cuesta, Comput. Phys. Commun. 157, 226 (2004).
- [19] See Supplemental Material at http://link.aps.org/supplemental/ 10.1103/PhysRevB.92.100303 for additional information regarding analysis, mode assignment, and open access data.
- [20] Y. Kawamura and H. Mashiyama, J. Korean Phys. Soc. 35, S1437 (1999).
- [21] I. P. Swainson, Acta Crystallogr., Sect. B: Struct. Sci. 61, 616 (2005).
- [22] C. M. B. Line, B. Winkler, and M. T. Dove, Phys. Chem. Miner. 21, 451 (1994).
- [23] M. Bee, Quasielastic Neutron Scattering: Principles and Applications in Solid State Chemistry, Biology, and Materials Science (Adam Hilger, Bristol, 1988).
- [24] G. Kieslich, S. Sun, and A. K. Cheetham, Chem. Sci. 5, 4712 (2014).
- [25] Y. Varma, R. Bhattacharjee, H. N. Vasan, and C. N. R. Rao, Spectrochim. Acta, Part A 48, 1631 (1992).

[26] S. F. Trevino and W. H. Rymes, J. Chem. Phys. 73, 3001

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- (1980).
- [27] J. Choi, X. Yang, Z. Norman, S. Billinge, and J. Owen, Nano. Lett. 14, 127 (2014).
- [28] W. Cochran, Adv. Phys. 9, 387 (1960).
- [29] K. Gesi, J. D. Axe, G. Shirane, and A. Linz, Phys. Rev. B 5, 1933 (1972).
- [30] G. Shirane and Y. Yamada, Phys. Rev. 177, 858 (1969).
- [31] Y. Fujii, S. Hoshino, Y. Yamada, and G. Shirane, Phys. Rev. B 9, 4549 (1974).
- [32] C. Carabatos-Nedelec, M. Oussaid, and K. Nitsch, J. Raman Spectrosc. 34, 388 (2003).
- [33] J. R. Shi, Y. Kume, J. Pelzl, Y. C. Xu, and X. Wu, J. Raman Spectrosc. 29, 149 (1998).
- [34] D. M. Calistru, L. Mihut, S. Lefrant, and I. Baltog, J. Appl. Phys. 82, 5391 (1997).
- [35] J. Even, L. Pedesseau, and C. Katan, J. Phys. Chem. C 118, 11566 (2014).
- [36] T. Chen, B. J. Foley, B. Ipek, M. Tyagi, J. R. D. Copley, C. M. Brown, J. J. Choi, and S.-H. Lee, arXiv:1506.02205.
- [37] A. M. A. Leguy, J. M. Frost, A. P. McMahon, V. G. Sakai, W. Kockelmann, C. Law, X. Li, F. Foglia, A. Walsh, B. C. O'Reagan, J. Nelson, J. T. Cabral, and P. R. F. Barnes, Nat. Commun. 6, 7124 (2015).
- [38] C. Wehrenfennig, M. Liu, H. J. Snaith, M. B. Johnston, and L. M. Herz, J. Phys. Chem. Lett. 5, 1300 (2014).
- [39] I. Borriello, G. Cantele, and D. Ninno, Phys. Rev. B 77, 235214 (2008).
- [40] D. J. Payne, R. G. Egdell, A. Walsh, G. W. Watson, J. Guo, P. A. Glans, T. Learmonth, and K. E. Smith, Phys. Rev. Lett. 96, 157403 (2006).
- [41] C. Motta, F. El-Mellouhi, S. Kais, N. Tabet, F. Alharbi, and S. Sanvito, Nat. Commun. 6, 7026 (2015).
- [42] J. Hlinka, E. Mihokova, and M. Nikl, Phys. Status Solidi B 166, 503 (1991).