



(Li_{0.84}Fe_{0.16})OHFe_{0.98}Se superconductor: Ion-exchange synthesis of large single-crystal and highly two-dimensional electron properties

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A large and high-quality single crystal (Li_{0.84}Fe_{0.16})OHFe_{0.98}Se, the optimal superconductor of a reported (Li_{1-x}Fe_x)OHFe_{1-y}Se system, has been successfully synthesized via a hydrothermal ion-exchange technique. The superconducting transition temperature (T_c) of 42 K is determined by magnetic susceptibility and electric resistivity measurements, and the zero-temperature upper critical magnetic fields are evaluated as 79 and 313 T for the field along the c axis and the ab plane, respectively. The ratio of out-of-plane to in-plane electric resistivity ρ_c/ρ_{ab} is found to increase with decreasing temperature and to reach a high value of 2500 at 50 K, with an evident kink occurring at a characteristic temperature $T^* = 120$ K. The negative in-plane Hall coefficient indicates that electron carriers dominate in the charge transport, and the hole contribution is significantly reduced as the temperature is lowered to approach T^* . From T^* down to T_c we observe the linear temperature dependencies of the in-plane electric resistivity and the magnetic susceptibility for the FeSe layers. Our findings thus reveal that the normal state of (Li_{0.84}Fe_{0.16})OHFe_{0.98}Se becomes highly two dimensional and anomalous prior to the superconducting transition, providing an insight into the mechanism of high- T_c superconductivity.

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The binary Fe_{1+ δ} Se compound [1,2] structured with FeSe-tetrahedron layers of a maximum interlayer compactness exhibits a superconductivity only at 8.5 K under ambient pressure. However, in a highly two-dimensional single unit-cell FeSe film on a SrTiO₃ substrate, the superconducting transition temperature T_c can be increased up to a value as high as 65 K [3–6]. In other more complicated FeSe-based superconductor compounds, where the FeSe-tetrahedron layers with a larger interlayer spacing serve all the same as basic superconducting units, the superconductivity of a higher T_c than binary bulk Fe_{1+ δ} Se is always realized. In A_yFe_{2-x}Se₂ compounds, for example, the alkali metal ions A are intercalated into between the adjacent FeSe-tetrahedron layers, expanding the interlayer separation and yielding superconductivity with $T_c \sim 30$ K. It seems that the superconductivity is related to the structural and/or electronic two dimensionality to some extent. However, the superconducting phase in A_yFe_{2-x}Se₂ is always intergrown with an insulating antiferromagnetic (AFM) phase resulting from a $\sqrt{5} \times \sqrt{5}$ superstructure of Fe vacancies [7]. This complexity obstructs further experimental studies on its intrinsic properties of both the normal and superconducting states.

Recently, (Li_{1-x}Fe_x)OHFe_{1-y}Se (FeSe1111) superconductors with T_c up to 40 ~ 43 K have been synthesized by hydrothermal approaches [8–11]. The troublesome $\sqrt{5} \times \sqrt{5}$ Fe vacancy ordered phase is *absent* [11], but a canted AFM

order in the intercalated (Li/Fe)OH layers is reported to coexist with the superconductivity [8]. Since FeSe1111 has a larger interlayer spacing than that of A_yFe_{2-x}Se₂, the correlation between the adjacent FeSe layers becomes even weaker. Such a quasi-two-dimensional structure is propitious for unveiling the interplay of electronic anisotropy and high- T_c superconductivity in the FeSe-based superconductors. For in-depth investigations on the intrinsic electronic properties of FeSe1111, high-quality large single crystals are indispensable.

In this paper we report the successful synthesis of a large and high-quality single crystal (Li_{0.84}Fe_{0.16})OHFe_{0.98}Se via a hydrothermal ion-exchange technique using a big insulating K_{0.8}Fe_{1.6}Se₂ crystal as a matrix. Its superconducting transition is confirmed by magnetic susceptibility and electric resistivity measurements, and the zero-temperature upper critical magnetic fields are estimated to be 79 and 313 T for the field along the c axis and the ab plane, respectively. The temperature dependencies of out-of-plane and in-plane electric resistivity are measured and their ratio ρ_c/ρ_{ab} , as a measure of the charge transport anisotropy, is found to increase with decreasing temperature and to reach its large value of 2500 at 50 K, showing an evident kink at a characteristic temperature $T^* = 120$ K. Meanwhile, the negative in-plane Hall coefficient indicates that electron carriers dominate the charge transport and the hole contribution is significantly reduced as the temperature is lowered to approach T^* . From T^* down to T_c , there emerge the linearly temperature dependent in-plane electric resistivity and the magnetic susceptibility deduced for the FeSe layers. Hence, our results reveal that the electron carriers of (Li_{0.84}Fe_{0.16})OHFe_{0.98}Se superconductor are getting highly two dimensional prior to the superconducting transition, resulting in the anomalous normal state properties.

The current hydrothermal methods using commercially available reagents together with, in a particular case,

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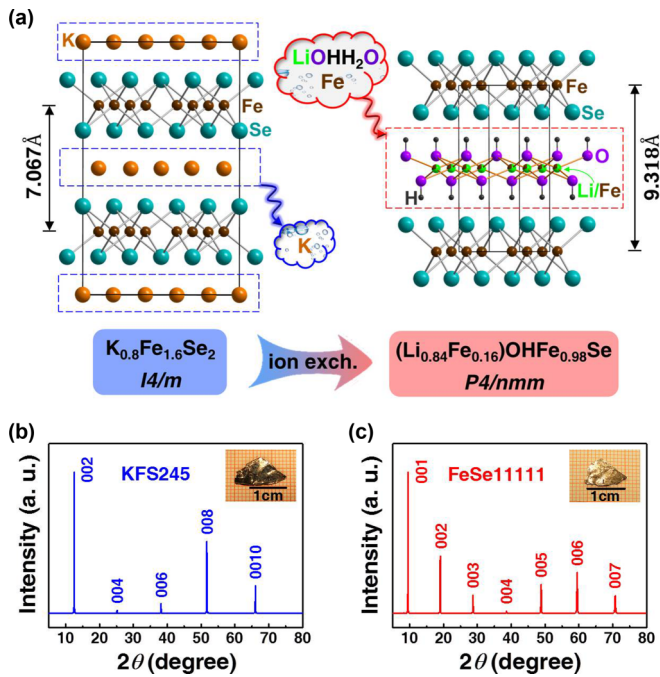


FIG. 1. (Color online) Illustration of hydrothermal ion-exchange process. (a) A schematic illustration of the hydrothermal ionic exchange reaction with the starting materials of big matrix crystals of $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$, $\text{LiOH}\cdot\text{H}_2\text{O}$, Fe , and $\text{CH}_4\text{N}_2\text{Se}$. The spacing between the adjacent FeSe layers of thus synthesized $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ crystal with the space group $P4/nmm$ is consequently enlarged by $\sim 32\%$ [from ~ 7.067 Å in $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ to ~ 9.318 Å in $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$]. For clarity, the H-Se and K-Se bondings in the structures are not shown. (b) and (c) The XRD patterns of (00*l*) type for the $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ and the $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ crystal pieces, respectively, demonstrating their crystal orientations along (001) planes. The insets show the corresponding photographs of the crystals.

synthesized binary FeSe [10] as starting materials have only produced powder samples of FeSe1111 [8–11]. Our recipe for preparing the large and high quality crystals is that large crystals of $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ (nominal 245 phase, KFS245 for short) are specially grown and used as a kind of *matrix* for a hydrothermal ionic exchange reaction. The KFS245 structure consists of an alternative stacking with layers of K ions and FeSe tetrahedra similar to our target compound, so as to be the best precursor with an ideal matrix structure. Moreover, the K ions in KFS245 are *completely* released into solution after the hydrothermal reaction process. Simultaneously, (Li/Fe)OH layers constructed by ions from the solution are squashed into the matrix, linking the adjacent edge-sharing FeSe tetrahedra via a weak hydrogen bonding. The separation between the neighboring FeSe-tetrahedron layers is consequently enlarged by $\sim 32\%$ (from ~ 7.067 to ~ 9.318 Å), resulting in an enhanced two dimensionality of the electronic structure compared with both the bulk FeSe and the intercalated $\text{A}_y\text{Fe}_{2-x}\text{Se}_2$ superconductors.

As-derived high-quality FeSe1111 single crystals are of a size over 10 mm in length and about 0.4 mm in thickness. The cartoon in Fig. 1(a) illustrates such an ionic exchange process. The experimental details of crystal synthesis are described in Supplemental Material [12]. The insets of Figs. 1(b) and 1(c) are the photographs of two typical crystal pieces for KFS245

precursor and its subsequently derived FeSe1111, respectively. Interestingly, the FeSe1111 crystal roughly inherits the original shape of its precursor. The room temperature x-ray diffraction (XRD) patterns of (00*l*) type on the two crystal pieces demonstrate their crystal orientations along (001) planes, with $l = 2n$ ($n = \text{integer}$) for KFS245 [Fig. 1(b)] and without the systematic extinction for FeSe1111 [Fig. 1(c)]. Their corresponding powder XRD patterns are given in Figs. S1a and S1b of the Supplemental Material [12], showing no detectable impurity phases within experimental resolution. All the reflections in each powder pattern can be well indexed on known tetragonal structures with the space group of $I4/m$ [7] for KFS245 and $P4/nmm$ [8–10] for FeSe1111. The least-squares refined unit cell dimensions are $a = 8.7248(6)$ Å and $c = 14.1339(12)$ Å for KFS245, and $a = 3.7827(4)$ Å and $c = 9.3184(7)$ Å for FeSe1111. Single crystal XRD data of FeSe1111 are also collected at 180 K and the structure refinement is performed in light of the reported structure model [8,10]. The resulting crystallographic data are listed in Table S1 of the Supplemental Material [12]. Finally the chemical formula of our single crystal is expressed as $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ based on the structural refinement. While the $\sqrt{5} \times \sqrt{5}$ ordered vacant Fe sites in the FeSe layers of $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ are of $\sim 20\%$ in amount [7], most of them are occupied by Fe ions imported during the ionic exchange process, leaving over there only 2% disordered Fe vacancies in the end crystal of FeSe1111. The atomic ratio of Li:Fe:Se (0.838:1.142:1, Supplemental Material Table S1 [12]) determined from the structural refinement is in perfect agreement with the result (0.82:1.17:1) from inductively coupled plasma atomic emission spectroscopy (ICP-AES) analysis. Importantly, no trace of K ions in our FeSe1111 crystals is detectable by energy dispersive x-ray spectrometry (EDX) and ICP-AES, verifying a complete release of K ions after the hydrothermal ionic exchange.

Figure 2(a) displays the magnetic susceptibility data for the KFS245 precursor, exhibiting an AFM phase transition at

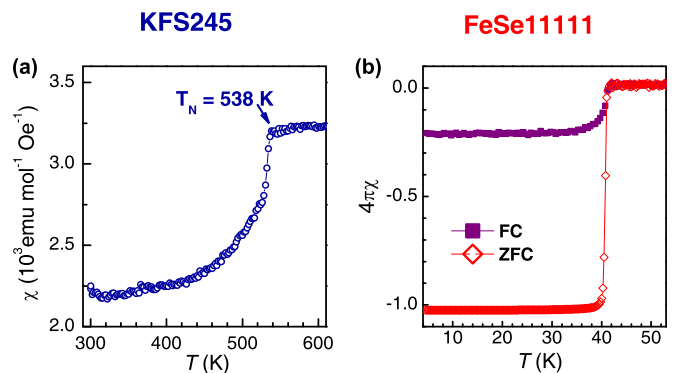


FIG. 2. (Color online) Magnetic properties of $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ and $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ crystals. (a) The temperature dependence of the static magnetic susceptibility of $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ exhibits an antiferromagnetic transition at 538 K. (b) The magnetic susceptibilities corrected for demagnetization factor under zero-field cooling (ZFC) and field cooling (FC, 1 Oe along the *c* axis) for $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ show that a sharp diamagnetic transition occurs at 42 K. The ZFC curve shows a 100% superconducting shielding and the FC one a Meissner signal up to $\sim 21\%$.

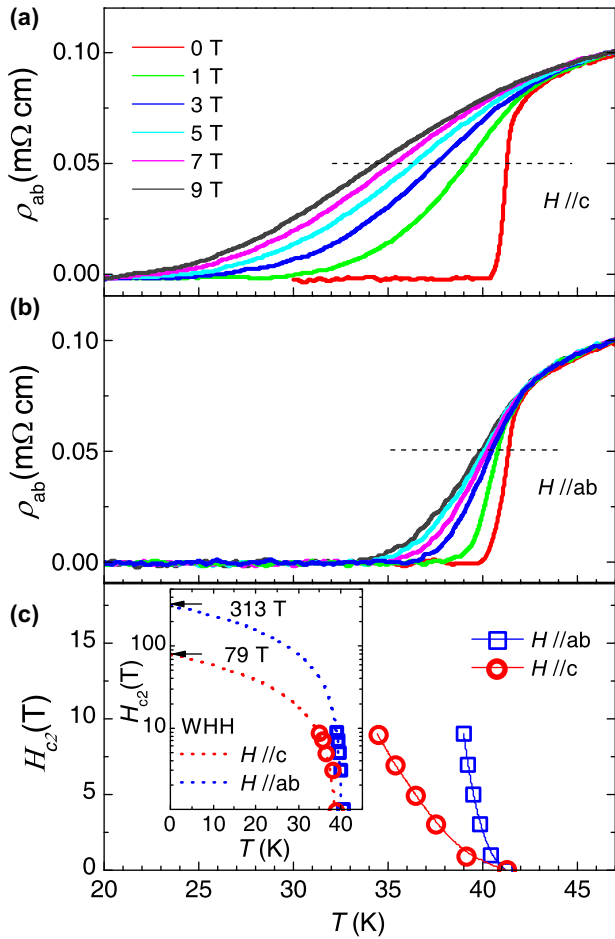


FIG. 3. (Color online) The in-plane electric resistivity under different magnetic fields and the upper critical magnetic fields. (a) The in-plane electric resistivity of (Li_{0.84}Fe_{0.16})OHFe_{0.98}Se displays the superconducting transitions under different applied magnetic fields along the c axis. (b) The in-plane electric resistivity under different applied magnetic fields along the ab plane. Up to 9 T, the onset superconducting transition temperature remains almost unchanged in both (a) and (b), but the temperature of zero resistance state shifts quickly from 40 K to lower temperatures with increasing field. (c) The anisotropic upper critical magnetic field H_{c2} as functions of temperature deduced from (a) and (b), from which the zero-temperature upper critical fields can be extrapolated according to the Werthamer-Helfand-Hohenberg (WHH) formula (the inset).

538 K consistent with the previous reports [7]. Its insulating behavior is checked up by electric resistivity measurement (Supplemental Material Fig. S2 [12]). By contrast, the ion-exchange synthesized FeSe1111 crystal exhibits bulk superconductivity, as evidenced by its magnetic susceptibilities drop at 42 K [Fig. 2(b)], with a narrow temperature range $\Delta T \sim 1$ K between 10% and 90% shielding signals. The sharp diamagnetic transition along with its 100% diamagnetic shielding demonstrates a high quality of our FeSe1111 single crystal.

Displayed in Figs. 3(a) and 3(b) are the temperature dependencies of the in-plane electric resistivity $\rho_{ab}(T)$ under the magnetic fields along the c axis ($H//c$) and the ab plane ($H//ab$), respectively. The onset superconducting transition temperature remains almost the same under fields up to 9 T,

no matter $H//c$ or $H//ab$. However, the zero resistance state shifts to lower temperatures with increasing magnetic field, but more slowly in $H//ab$ than in $H//c$, implying the formation of complex vortex states. Such a behavior resembles the features observed in copper-oxide high temperature superconductors [13,14]. By taking the temperature at the midpoint of the electric resistivity drop as that of the upper critical field H_{c2} [Fig. 3(c)], the zero-temperature upper critical magnetic fields can be estimated as 79 T for field along the c axis and 313 T for field along the ab plane from the Werthamer-Helfand-Hohenberg (WHH) formula [15] [inset of Fig. 3(c)]. So the superconducting state of the FeSe1111 sample is anisotropic.

In the normal state, the in-plane electric resistivity ρ_{ab} displays a metallic behavior in the whole measuring temperature range [Fig. 4(a)]. As the temperature is lowered to the characteristic temperature $T^* = 120$ K, the slope of the resistivity curve undergoes a prominent reduction. Subsequently, there emerges the linear temperature dependence of ρ_{ab} from 80 K down to T_c [inset of Fig. 4(a)]. Due to the large size of our sample, the out-of-plane electric resistivity of FeSe1111 single crystal is readily measured. The resulting ρ_c shows a roughly similar behavior to ρ_{ab} . However, their ratio ρ_c/ρ_{ab} , as a measure of the carrier anisotropy, increases with decreasing temperature and reaches to a large value of 2500 at 50 K [Fig. 4(a)]. Interestingly, a kink in the ratio also appears at the characteristic temperature T^* , due to the significant slope change in ρ_{ab} compared to ρ_c (Supplemental Material Fig. S3 [12]). It is clear that, prior to the superconducting transition, the normal state electronic properties of the FeSe1111 crystal turns out to be highly two dimensional.

Actually, other normal state properties are also readily measured on our large FeSe1111 crystal. The in-plane Hall voltage has been obtained under the magnetic fields along the c axis by sweeping the field up to 9 T at fixed temperatures. The extracted Hall resistivity (ρ_{xy}) is proportional to the magnetic field at all measuring temperatures (Supplemental Material Fig. S4 [12]), so the Hall coefficient $R_H = \rho_{xy}/H$ is field independent. As shown in Fig. 4(b), R_H is negative in the whole measuring temperature, but displays a dip feature around the characteristic temperature $T^* = 120$ K. We propose to use a compensated metal picture of multiple electron and hole bands to explain our observation. With the equal total electron and hole concentrations ($n_h = n_e$), the Hall coefficient satisfies $R_H = (\mu_h - \mu_e)/[(\mu_h + \mu_e)n_e e]$, where $\mu_e(\mu_h)$ stands for the mobility of electron (hole) charge carriers. The negative Hall coefficient indicates that the contribution of electrons dominates the charge transport of the sample ($\mu_e > \mu_h$). Above the characteristic temperature T^* , the mobility of holes μ_h is reduced as the temperature is getting lower, leading to a decreasing Hall coefficient. Below T^* , the electron carriers govern the charge transport ($\mu_e \gg \mu_h$). The Hall coefficient can thus be simplified as $R_H \sim -1/(n_e e)$, and its upturn with the further lowering temperature can be attributed to an enhanced carrier density.

The static magnetic susceptibility $\chi = M/H$ is carefully measured and analyzed under different magnetic fields H along the c axis in the region with linear field dependencies of magnetization M (Supplemental Material Fig. S5 [12]). There exist two different contributions to the magnetic susceptibility: one part arises from the intercalated (Li/Fe)OH layers and

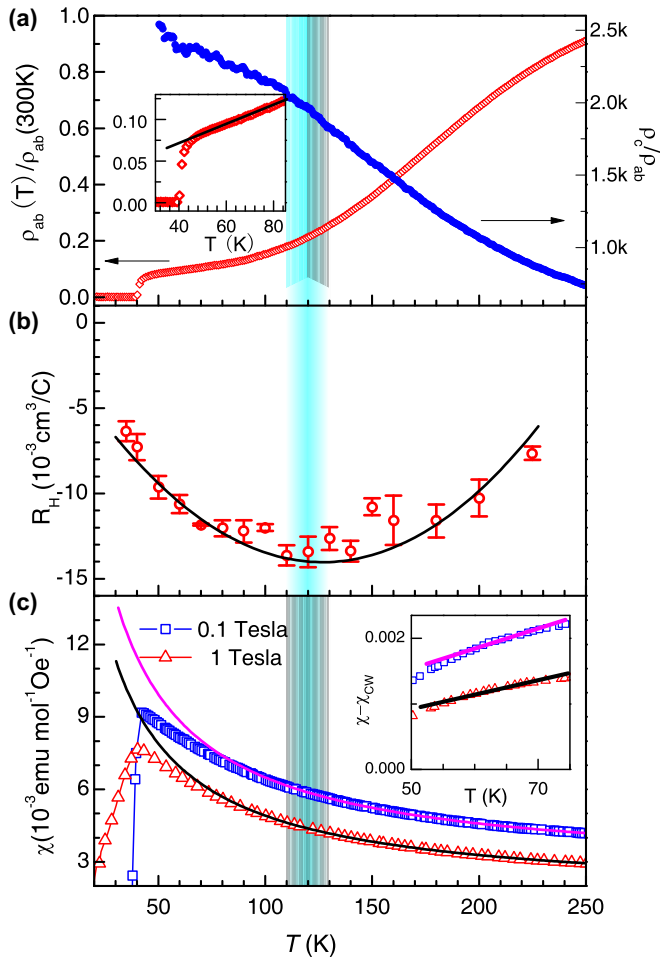


FIG. 4. (Color online) The electric resistivity, Hall coefficient, and static magnetic susceptibility of $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ single crystal. (a) The in-plane electric resistivity and the ratio of the out-of-plane resistivity to the in-plane one as functions of temperature. Around $T^* = 120$ K, the slope of the resistivity curve undergoes a prominent reduction and a corresponding kink occurs in the curve of ρ_c/ρ_{ab} . The inset figure shows the linearly temperature dependent range from 80 K down to T_c for ρ_{ab} . (b) The in-plane Hall coefficient R_H as a function of temperature exhibits a dip feature around $T^* = 120$ K. The black curve is a guide to the eye. (c) The temperature dependencies of static magnetic susceptibility $\chi = M/H$ under magnetic fields along the c axis. The results *slightly* depend on the magnitude of the applied field. The sudden drops in the magnetic susceptibility are due to the appearance of the superconductivity. In the high temperature range, all the data can be fitted to a modified Curie-Weiss law $\chi_m = \chi_0 + \chi_{\text{CW}}$. But a deviation is clearly visible below $T^* = 120$ K as well. Further lowering temperature, the rest magnetic susceptibility in which the Curie-Weiss term is subtracted shows the nearly linear temperature dependence in a narrow temperature range (the inset).

another part from the FeSe layers. The total susceptibility slightly depends on the magnitude of the applied field [Fig. 4(c)]. In the higher temperature range, a modified Curie-Weiss law is obeyed: $\chi_m = \chi_0 + \chi_{\text{CW}}$, where $\chi_{\text{CW}} = C/(T - \theta)$. The effective magnetic moment in the (Li/Fe)OH layers is fitted as $\mu_{\text{Fe}} \approx 4.2 \sim 4.6 \mu_B$, consistent with the previous results [10]. These local magnetic moments are nearly free.

But the fitted θ is of a small negative value (-6.6 to -7.5 K), implying an AFM long-range order will eventually form in the (Li/Fe)OH layers in the lower temperature regime [8]. A constant term χ_0 signifies a Pauli paramagnetic contribution from itinerant charge carriers. But a deviation from the modified Curie-Weiss law is evident again below the characteristic temperature $T^*(=120$ K). When the Curie-Weiss term χ_{CW} is subtracted, the rest part of magnetic susceptibility depends almost linearly on temperature in the range of $55 \sim 75$ K [inset of Fig. 4(c)], suggesting the presence of two-dimensional AFM spin fluctuations in the FeSe layers. Such a linear behavior has been previously seen in the paramagnetic metallic states of iron arsenide superconductors [16,17]. Therefore, the prominent slope reduction in the ρ_{ab} - T curve around T^* is likely due to the additional scattering mechanism brought about by the emergent two-dimensional AFM spin fluctuations and the vanishing mobility of holes below T^* .

Finally, we have established a common temperature scale ($T^* = 120$ K) in $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ superconductor, which characterizes the prominent reduction in the slope of the in-plane electric resistivity, the kink in the ratio ρ_c/ρ_{ab} , the dip in the in-plane Hall coefficient, and the deviation in the magnetic susceptibility from the Curie-Weiss law. Below this characteristic T^* , normal state electronic behaviors are getting highly two dimensional and the AFM spin fluctuations in the FeSe layers set in, manifested by the anomalous linear in-plane electric resistivity and nearly linear magnetic susceptibility deduced for the FeSe layers. The two-dimensional electron-electron interaction is crucial to the electron pairing of superconductivity in the FeSe layers, as commonly believed in optimal iron arsenide [18–20] and copper-oxide high temperature superconductors [21–23], as well as in heavy fermion [24,25] and organic superconductors [26]. Hence our results suggest that the $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ superconductor falls under the same universality class.

In conclusion, we have successfully synthesized large and high quality superconducting $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ single crystal out of insulating $\text{K}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ matrix crystal via hydrothermal ion exchange. Its in-plane lattice constants a and b determined as 3.78 \AA are comparable to the primary bulk FeSe (3.77 \AA) and $\text{A}_y\text{Fe}_{2-x}\text{Se}_2$ (3.90 \AA) superconductors. In contrast, its FeSe-tetrahedron layers weakly bonded to the (Li/Fe)OH layers via hydrogen ions have a larger interlayer spacing (9.32 \AA) compared with the bulk FeSe (5.52 \AA) and $\text{A}_x\text{Fe}_{2-y}\text{Se}_2$ (7.02 \AA) superconductors. Consequently, $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ exhibits an enhanced two dimensionality of the electronic structure and the higher superconducting transition at 42 K than the bulk FeSe (~ 8.5 K) and $\text{A}_y\text{Fe}_{2-y}\text{Se}_2$ (~ 30 K). Hence our findings provide an insight into the pairing mechanism of high- T_c superconductivity.

Furthermore, a series of large single crystals of $(\text{Li}_{1-x}\text{Fe}_x)\text{OHFe}_{1-y}\text{Se}$ system are expected by duplicating the hydrothermal ion-exchange technique. The breakthrough in crystal preparation reported here will greatly promote the in-depth investigations on this system. In addition, this very successful technique developed for synthesizing $(\text{Li}_{0.84}\text{Fe}_{0.16})\text{OHFe}_{0.98}\text{Se}$ crystals may be applicable in other layer-structured materials, opening up a new route for producing and exploring high-quality bulk functional crystals desired for both basic research and applications.

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