# Accumulation, inversion, and depletion layers in SrTiO<sub>3</sub>

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We study potential and electron density depth profiles in accumulation, inversion, and depletion layers in crystals with a large and nonlinear dielectric response such as SrTiO<sub>3</sub>. We describe the lattice dielectric response using the Landau-Ginzburg free energy expansion. In accumulation and inversion layers we arrive at new nonlinear dependencies of the width *d* of the electron gas on an applied electric field  $D_0$ . Particularly important is the predicted electron density profile of accumulation layers (including the LaAlO<sub>3</sub>/SrTiO<sub>3</sub> interface)  $n(x) \propto (x + d)^{-12/7}$ , where  $d \propto D_0^{-7/5}$ . We compare this profile with available data and find satisfactory agreement. For a depletion layer we find an unconventional nonlinear dependence of the capacitance on voltage. We also evaluate the role of spatial dispersion in the dielectric response by adding a gradient term to the Landau-Ginzburg free energy.

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#### I. INTRODUCTION

In recent years there has been growing interest in the investigation of ABO<sub>3</sub> perovskite crystals, which are important for numerous technological applications and show intriguing magnetic, superconducting, and multiferroic properties [1]. Special attention [2,3] is paid to heterostructures involving SrTiO<sub>3</sub> which is a semiconductor with a band gap of  $\simeq 3.2 \text{ eV}$  [4] and a large dielectric constant  $\kappa$  ranging from  $2 \times 10^4$  at liquid helium temperatures to 350 at room temperature. As with conventional semiconductors, SrTiO<sub>3</sub> can be used as a building block for different types of devices, with reasonably large mobility [5,6].

Many devices are based on the accumulation layer of electrons near a heterojunction interface involving SrTiO<sub>3</sub>. For example, one can use modulation doping in the structure SrTiO<sub>3</sub>/SrZrO<sub>3</sub> to introduce electrons in the conduction band of SrTiO<sub>3</sub> from La donors within the wider-band-gap material SrZrO<sub>3</sub> [7]. Inside bulk SrTiO<sub>3</sub>  $\delta$  doping can be used to introduce two accumulation layers of electrons [8-10]. One can accumulate an electron gas using a field effect [11-13]instead of dopants. In Refs. [14,15] the authors accumulated up to  $10^{14}$  cm<sup>-2</sup> electrons on the surface of SrTiO<sub>3</sub> using ionic liquid gating. Finally, there is enormous interest [2,5,6] in LaAlO<sub>3</sub>/SrTiO<sub>3</sub> heterojunctions where electrons are accumulated by the electric field resulting from the "polar catastrophe" [16]. It is natural to think that the depth profiles of the potential and electron density inside SrTiO<sub>3</sub> have a universal origin in all these devices.

Another type of device based on *n*-doped SrTiO<sub>3</sub> is the Schottky diode. Due to the built-in Schottky barrier the region near the metal-semiconductor interface in doped SrTiO<sub>3</sub> is depleted. The large and nonlinear dielectric constant results in unconventional capacitance-voltage characteristics. Schottky diodes with different metals and bulk SrTiO<sub>3</sub> dopants have been studied: Au/Nb : SrTiO<sub>3</sub> [17], Ba<sub>1-x</sub>K<sub>x</sub>BiO<sub>3</sub>/Nb : SrTiO<sub>3</sub> [18], SrRuO<sub>3</sub>/Nb : SrTiO<sub>3</sub> [19], and Au/SrTiO<sub>3-x</sub> [20]. All of the devices cited above are based on accumulation and depletion layers. We do not know of any attempts to create a hole inversion layer in n-type SrTiO<sub>3</sub> or an electron inversion layer in p-type SrTiO<sub>3</sub> but they are likely to be of interest as well.

Interface properties determine characteristics of all these devices. Not surprisingly, the potential and electron density depth profiles in such devices have attracted attention from the experimental [21–24] and theoretical points of view [14,25–29]. For example, experimental data show that electrons are distributed in a layer of width  $\simeq$ 5–10 nm near the LaAlO<sub>3</sub>/SrTiO<sub>3</sub> interface. Theoretical works that attempt to explain such behavior are based on microscopic numerical calculations.

The goal of this paper is to create a simple, mostly phenomenological and analytical approach for describing the potential and the electron density depth profiles in  $SrTiO_3$ . To account for the nonlinear dielectric response in  $SrTiO_3$  we use the Landau-Ginzburg free energy expansion [30,31]. Electrons are almost everywhere described in the Thomas-Fermi approximation [32]. Although we mostly concentrate on  $SrTiO_3$ , the developed approach is applicable to  $KTaO_3$  [33] and  $CaTiO_3$  [34] as well.

Our main result is a new form for the potential and electron density depth profiles in accumulation, inversion, and depletion layers due to the nonlinear dielectric response. In particular, for an accumulation layer in SrTiO<sub>3</sub>, we find an electron concentration n(x) that depends on the distance from the surface x as  $n(x) \propto (x + d)^{-12/7}$ , where the width d decreases with the external electric field  $D_0$  as  $d \propto D_0^{-7/5}$ . These relations seem to agree with experimental data [22,23].

The remainder of this paper is organized as follows. In Sec. II we define the model based on the Landau-Ginzburg theory for calculating the lattice dielectric response and describe the parameters of SrTiO<sub>3</sub>. In Sec. III we use the Thomas-Fermi approach for calculating the self-consistent electric field to describe properties of electron accumulation layers. In the Sec. IV we apply our theory to the consideration of interfaces between SrTiO<sub>3</sub> and polar dielectrics. In particular, we pay attention to the case of an accumulation layer on the interface of LaAlO<sub>3</sub>/SrTiO<sub>3</sub> and compare our theory with experimental data. In Sec. V we calculate the quantum capacitance of

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the accumulation layer in SrTiO<sub>3</sub>. In Sec. VI we use a one subband approximation, in which electrons do not affect the electric field, to calculate properties of inversion layers. In Sec. VII we consider a depletion layer in SrTiO<sub>3</sub>, calculate the capacitance-voltage characteristics of a Schottky barrier for such systems, and compare our results with experimental data. In Sec. VIII we show that our results are not modified by the presence of spatial dispersion in the dielectric response. Section IX provides a summary and conclusion.

# **II. THE MODEL**

Bulk SrTiO<sub>3</sub> typically is an *n*-type semiconductor with a concentration of donors  $N > 10^{17}$  cm<sup>-3</sup>. Let us discuss the position of Fermi energy  $\varepsilon_F$  in such crystals. The electron spectrum near the bottom of the conduction band is complicated [35], and in order to make the problem of an accumulation layer tractable analytically we assume that it is isotropic and nondegenerate with the effective mass  $m^* \simeq 1.5m$ , [36] where m is free electron mass. Within the hydrogenic theory of shallow donors, the donor Bohr radius is equal to  $\kappa b$ , where  $b = \hbar^2/m^* e^2 \simeq 0.35$  Å, e is the electron charge, and  $\kappa$  is dielectric constant of the material. At room temperature when  $\kappa = 350$ , the Bohr radius  $\kappa b = 123$  Å is so large that the Mott criterion for the metal-insulator transition in doped semiconductors  $N_c b^3 = 0.02/\kappa^3$  leads to a very small critical concentration  $N_c = 1 \times 10^{16} \text{ cm}^{-3}$ . At helium temperatures  $\kappa = 2 \times 10^4$  and  $N_c = 6 \times 10^{10} \text{ cm}^{-3}$ . Thus, at the experimentally relevant concentration of donors N > $10^{17}$  cm<sup>-3</sup>, we are dealing with a heavily doped semiconductor in which the Fermi energy lies in the conduction band of SrTiO<sub>3</sub>. On the other hand, due to the relatively high effective mass, the bulk Fermi energy  $\varepsilon_F$  is smaller than the bending energy of the conduction band bottom near the interface (see Figs. 1 and 6). For example, for  $N = 10^{18} \text{ cm}^{-3}$ , the Fermi energy calculated from the bottom of the conduction band is  $\varepsilon_F \simeq 4$  meV, which can be up to 100 times smaller than the bending energy of the conduction band bottom in an accumulation layer for GdTiO<sub>3</sub>/SrTiO<sub>3</sub>. Therefore, we assume below that the Fermi energy coincides with the bottom of the conduction band.



We are interested in accumulation, inversion, and depletion layers near an interface of SrTiO<sub>3</sub>. We consider the case when the axis x is directed perpendicular to the interface (plane x =0) and lies along the [100] axis of a cubic crystal of SrTiO<sub>3</sub>. (In fact, SrTiO<sub>3</sub> changes symmetry from cubic to tetragonal at  $T \simeq 110$  K, but the distortion is small [37] and can be neglected.) An external electric field  $D_0$  applied from the left (see Figs. 1, 4, and 6) is directed along the x axis. In that case the problem is effectively one dimensional. If the charge density is denoted by  $\rho(x)$ , then the potential depth profile  $\varphi(x)$  in the system is determined by the equations

$$\frac{dD}{dx} = 4\pi\rho, \quad D = E + 4\pi P, \quad \frac{d\varphi}{dx} = -E, \quad (1)$$

where D(x), E(x), P(x) are electric induction, electric field, and polarization in SrTiO<sub>3</sub>. Equations (1) should be solved with proper boundary conditions. For example, for an accumulation layer the boundary conditions are  $D(0) = D_0$  and  $\varphi(\infty) = 0$ .

To solve the system (1) one needs to know two material relationships E(P) and  $\rho(\varphi)$ . Let us start from the lattice dielectric response E(P). SrTiO<sub>3</sub> is well known as a quantum paraelectric, where the onset of ferroelectric order is suppressed by quantum fluctuations [38].

A powerful approach to describe the properties of ferroelectriclike materials is based on the Landau-Ginzburg theory. For a continuous second-order phase transition the Landau-Ginzburg expression of the free energy density F is represented as a power series expansion with respect to the polarization P:

$$F = F_0 + \frac{\tau}{2}P^2 + \frac{1}{4}A\frac{1}{P_0^2}P^4 - EP,$$
 (2)

where  $F_0$  stands for the free energy density at P = 0 and  $\tau$ is the inverse susceptibility  $\tau = 4\pi/(\kappa - 1) \simeq 4\pi/\kappa$ . In this work  $0 < \tau \ll 1$ ,  $P_0 = e/a^2$  is the characteristic polarization and  $a \simeq 3.9$  Å [37] is the lattice constant. The coefficient *A* describes the nonlinear dielectric response. Analyzing the available data [18,39–41] in Sec. VII we find values of *A* between 0.5 and 1.5. For all estimates below we use A = 0.8following from Ref. [41]. The last term of Eq. (2) is responsible for the interaction between the polarization and the electric field *E*. In general *F* depends on the components of the vector *P*, but in the chosen geometry the problem is one dimensional, and all vectors are directed along the *x* axis. The crystal polarization *P* is determined by minimizing the free energy density *F* in the presence of the electric field *E*,  $\delta F/\delta P = 0$ . This condition relates *E* and *P*,

$$E = \frac{4\pi}{\kappa}P + \frac{A}{P_0^2}P^3.$$
 (3)

We note that  $E \ll 4\pi P$  and thus  $D = E + 4\pi P \simeq 4\pi P$ . The electric field  $D_c$  at which the transition from linear to nonlinear dielectric response occurs can be found by equating the first and second terms in expression (3):

$$D_c = P_0 \sqrt{\frac{(4\pi)^3}{\kappa A}}.$$
 (4)

If  $D \ll D_c$  the dielectric response of SrTiO<sub>3</sub> is linear and one can use the simplified expression for the electric field:

$$E = \frac{D}{\kappa}.$$
 (5)

For  $D \gg D_c$  the dielectric response of SrTiO<sub>3</sub> is nonlinear and one must instead use the expression

$$E = \frac{A}{(4\pi)^3 P_0^2} D^3.$$
 (6)

Next one should specify  $\rho(\varphi)$ , which depends on the specific device of interest.

#### **III. ACCUMULATION LAYER: THEORY**

In an accumulation layer the external electric field  $D_0$  attracts electrons with a three-dimensional concentration n(x) (see Fig. 1). Our goal is to find the electron depth profile n(x) and its characteristic width d.

Due to electric neutrality the number of accumulated electrons has to compensate the external field  $D_0$ , i.e.,

$$4\pi e \int_0^\infty n(x) dx = D_0. \tag{7}$$

To take into account the electron screening of the external field we use the Thomas-Fermi approach [32] in which the electron concentration n(x) and self-consistent potential profile  $\varphi(x)$ are related as  $e\varphi(x) + \mu(x) = \varepsilon_F = 0$ , where

$$\mu(x) = (3\pi^2)^{2/3} \frac{\hbar^2}{2m} [n(x)]^{2/3}$$
(8)

is the chemical potential of the electron gas. Thus, one can obtain the solution of Eqs. (1) by replacing  $\rho(x)$  with en(x) and using relations (5) and (6). For a linear dielectric response we obtain the equation for the potential:

$$\frac{d^2}{dx^2}\left(\frac{\varphi}{e/b}\right) = \frac{2^{3/2}}{3\pi^2} \frac{1}{b^2} \frac{1}{\kappa} \left(\frac{\varphi}{e/b}\right)^{3/2} \tag{9}$$

We use the boundary condition  $\varphi = 0$  at  $x \to \infty$  and get the solution

$$\varphi(x) = C_1 \frac{e}{b} \kappa^2 \left(\frac{b}{x+d}\right)^4, \tag{10}$$

$$n(x) = C_2 \frac{1}{b^3} \kappa^3 \left(\frac{b}{x+d}\right)^6,\tag{11}$$

where  $C_1 = (225/8)\pi^2 \simeq 278$  and  $C_2 = 1125\pi/8 \simeq 442$ . { $\varphi(0)$  was derived equivalently in Ref. [42] as a work function reduction for GaAs.} For a nonlinear dielectric response we obtain the equation for the potential:

$$\frac{d}{dx}\left[\left(\frac{d}{dx}\frac{\varphi}{e/b}\right)^{1/3}\right] = \frac{2^{3/2}}{3\pi^2}\frac{1}{b^{4/3}}A^{1/3}\left(\frac{e/b^2}{P_0}\right)^{2/3}\left(\frac{\varphi}{e/b}\right)^{3/2}(12)$$

With the same boundary condition we get the solution

$$\varphi(x) = C_3 \frac{e}{b} \left(\frac{b}{a}\right)^{8/7} \frac{1}{A^{2/7}} \left(\frac{b}{x+d}\right)^{8/7}, \quad (13)$$

$$n(x) = C_4 \frac{1}{b^3} \left(\frac{b}{a}\right)^{12/7} \frac{1}{A^{3/7}} \left(\frac{b}{x+d}\right)^{12/7}, \qquad (14)$$

TABLE I. Percentage of electrons in the TiO<sub>2</sub> layer *M* of SrTiO<sub>3</sub> for  $D_0 = 2\pi e/a^2$ , corresponding to a total surface density of  $0.5e/a^2$ .

М	1	2	3	4	5	6	7	8	9	10
Percent	27.9	14.4	9.0	6.2	4.6	3.5	2.8	2.3	1.9	1.6

where  $C_3 = [5^6 3^6 \pi^{12} / (7^8 2^3)]^{1/7} \simeq 5.8$  and  $C_4 = [5^9 3^2 \pi^4 2^6 / 7^{12}]^{1/7} \simeq 1.3$ .

The characteristic length d can be obtained using the neutrality condition [see Eq. (7)]. For a linear dielectric response this gives

$$d = C_5 b \left(\frac{a}{b}\right)^{2/5} \kappa^{3/5} \left(\frac{e/a^2}{D_0}\right)^{1/5},$$
 (15)

where  $C_5 = [\pi^2 225/2]^{1/5} \simeq 4$ . For a nonlinear dielectric response:

$$d = C_6 b \left(\frac{a}{b}\right)^{2/5} \left(\frac{e/a^2}{D_0}\right)^{7/5} \frac{1}{A^{3/5}},$$
 (16)

where  $C_6 = (16/7)(5^2 3^2 \pi^{11})^{1/5} \simeq 84$ . The electric field  $D_c$  at which the transition from linear to nonlinear dielectric response occurs can be found from equating Eqs. (15) and (16). This gives

$$D_c = \frac{C_7}{\sqrt{A}} \frac{e}{a^2} \sqrt{\frac{1}{\kappa}},\tag{17}$$

where  $C_7 = (2^{21}\pi^9/7^5)^{1/6} \simeq 12$ , consistent with Eq. (4). For SrTiO<sub>3</sub>, the critical field  $D_c$  depends on temperature  $D_c \simeq 0.1e/a^2$  for helium temperature and  $D_c \simeq 0.7e/a^2$  for room temperature.

The three-dimensional concentration profile n(x) for the nonlinear dielectric response Eq. (14) is the main result of our paper. Note that n(x) has a very long tail with a weak 12/7 power law dependence, which may lead to some arbitrariness in measurements of the width of the electron gas. Indeed, only 39% of electrons are located within the distance 0 < x < d near the interface and 68% of electrons are located within 0 < x < 4d. In the calculation above we used a space-continuous model. Actually, along the [100] axis SrTiO<sub>3</sub> is composed of alternating TiO<sub>2</sub> and SrO layers. The conduction band of SrTiO<sub>3</sub> corresponds to the bands composed of mainly 3d orbitals of Ti. Integrating n(x) over each lattice cell in Table I we get a percentage of electrons in each of the 10 first TiO<sub>2</sub> layers of SrTiO<sub>3</sub> for the case  $D = 2\pi e/a^2$ .

One can see from Eqs. (11) and (14) that the tails of the electron depth profiles n(x) at  $x \gg d$  do not depend on  $D_0$  and behave like

$$C_2 \frac{1}{b^3} \kappa^3 \left(\frac{b}{x}\right)^6$$

$$C_4 \frac{1}{b^3} \left(\frac{b}{a}\right)^{12/7} \frac{1}{A^{3/7}} \left(\frac{b}{x}\right)^{12/7}$$

for linear and nonlinear dielectric responses, respectively. Even for  $D_0 \gg D_c$ , when the electron distribution n(x) at moderately large x is described by dependence (14), at very

and

large distances the polarization becomes smaller and the linear dielectric response takes over so that the n(x) dependence switches from Eq. (14) to Eq. (11). This happens at the distance

$$x_0 = b \left(\frac{C_2}{C_4}\right)^{7/30} A^{1/10} \left(\frac{a}{b}\right)^{2/5} \kappa^{7/10}$$

 $(x_0 = 360 \text{ and } 20 \text{ nm for helium and room temperature,}$ respectively). Thus, the tail of n(x) is universal. For small  $D_0 < D_c$  the tail has the form  $n(x) \propto x^{-6}$ . For  $D_0 > D_c$  it has the form  $n(x) \propto x^{-12/7}$  for  $x < x_0$  and  $n(x) \propto x^{-6}$  for  $x > x_0$ .

On the other hand, one has to remember that our theory is correct only when n(x) is larger than the concentration of donors in the bulk of the material.

Let us verify whether the Thomas-Fermi approximation is applicable, i.e.,  $k_F d \gg 1$ . Here  $k_F = (3\pi^2)^{1/3} n(0)^{1/3}$  is the wave vector of an electron at the Fermi level. For  $D_0 \ll D_c$ 

$$k_F d = C_8 \kappa^{2/5} \left(\frac{b}{a}\right)^{2/5} \left(\frac{D_0}{e/a^2}\right)^{1/5},$$
 (18)

while for  $D_0 \gg D_c$ 

$$k_F d = C_9 \frac{1}{A^{2/5}} \left(\frac{b}{a}\right)^{2/5} \left(\frac{e/a^2}{D_0}\right)^{3/5},$$
 (19)

where  $C_8 = (5^3 3^3 \pi^3 / 2^4)^{1/5} \simeq 6$ ,  $C_9 = 4/7(15\pi^3)^{3/5} \simeq 23$ . One can see that  $k_F d > 1$  in the range of  $2 \times 10^{-7} e/a^2 < D_0 < 40e/a^2$  for room temperature. For lower temperatures this interval is even larger. Thus, the Thomas-Fermi approximation is applicable for practically all reasonable electric fields  $D_0$  [43].

## IV. ACCUMULATION LAYER: COMPARISON WITH EXPERIMENTAL DATA

It is widely believed that an electron gas emerges near polar/nonpolar interfaces such as LaAlO<sub>3</sub>/SrTiO<sub>3</sub>, GdTiO<sub>3</sub>/SrTiO<sub>3</sub> [13,24], LaVO<sub>3</sub>/SrTiO<sub>3</sub> [44], NdAlO<sub>3</sub>/SrTiO<sub>3</sub>, PrAlO<sub>3</sub>/SrTiO<sub>3</sub>, NdGaO<sub>3</sub>/SrTiO<sub>3</sub> [45], LaGaO<sub>3</sub>/SrTiO<sub>3</sub> [46], and LaTiO<sub>3</sub>/SrTiO<sub>3</sub> [47] due to the polar catastrophe [16]. For a large enough thickness of the polar crystal [27] the interface electron surface charge density

$$\sigma = e \int_0^\infty n(x) dx \tag{20}$$

is equal to  $0.5e/a^2$ , which corresponds to  $D_0 = 2\pi e/a^2$  [see Eq. (7)]. For any temperature this field is much larger than the critical field  $D_c$  and Eq. (19) gives  $k_F d = 3$ . Thus, we arrive at Eq. (14) for the electron concentration n(x) and Eq. (16) shows that  $d \simeq 6.7$  Å. In this case 68% of electrons are located within 2.7 nm. This result agrees with experimental estimates of the width of an electron gas near the GdTiO<sub>3</sub>/SrTiO<sub>3</sub> interface [24]. It also agrees with experimental data [48] for the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>/SrTiO<sub>3</sub> interface, where a 2DEG is formed due to the formation of oxygen vacancies on SrTiO<sub>3</sub>. In the case of  $\delta$ -doped SrTiO<sub>3</sub> with one layer of La [49] one has two accumulation layers each with  $\sigma = 0.5e/a^2$  and a similar width d.

To see how important the nonlinear dielectric response is, one can compare its prediction to the one obtained by assuming



FIG. 2. The experimental distribution of electrons n(x) from Ref. [23] is shown by circles in convenient dimensionless units. Fitting by Eq. (14) is shown by the solid line. The fitting parameter is  $\sigma = 0.13e/a^2$ , which gives  $d \simeq 142b \simeq 5$  nm.

the response to be linear, given by Eq. (15). For  $D = 2\pi e/a^2$ and helium temperature the linear dielectric response gives d = 167 nm. A similar result was obtained in Ref. [50], where the nonlinear dielectric response was not taken into account.

For the LaAlO<sub>3</sub>/SrTiO<sub>3</sub> interface the number of electrons accumulated is apparently smaller than what the "polar catastrophe" scenario predicts. For example, only  $\simeq 10\%$  of the electrons are seen in Hall measurements [11,51,52]. In order to describe the electron concentration n(x) for such a surface charge density,  $\sigma = 0.05e/a^2$ , one can still use Eqs. (14) and (16). As a result, we arrive at a much larger value of  $d \simeq 17$  nm.

We test our theory for the functional shape of n(x) by comparing to experimental data for the electron distribution n(x) near the interface of LaAlO<sub>3</sub>/SrTiO<sub>3</sub> at temperature  $\simeq 10$  K [23] (see Fig. 2). For such small temperatures the critical field  $D_c$  is small and we fit the experimental data by Eq. (14) with  $\sigma = D_0/4\pi$  as a fitting parameter and get  $\sigma = 0.12e/a^2$  with  $d \simeq 142b \simeq 5$  nm. Figure 2 shows satisfactory agreement between the data and the shape of n(x)described by Eq. (14).

Let us now dwell upon the experimental data for n(x) from Ref. [22]. The data are obtained via time-resolved photoluminescence spectroscopy of the LaAlO<sub>3</sub>/SrTiO<sub>3</sub> interface, where interface-induced electrons radiatively recombine with photoexcited holes in SrTiO<sub>3</sub>. Following the assumption from Ref. [22] that the photoexcited holes are immobile, the concentration of holes *p* decays with time *t* according to the equation

$$\frac{dp}{dt} = -r_1 p - r_2 [n(x)]^2 p, \qquad (21)$$

where  $r_1$  is the hole trapping rate and  $r_2$  is the three carrier nonradiative Auger recombination coefficient. The authors of Ref. [22] used the decay of the photoluminescence intensity to obtain n(x) in LaAlO<sub>3</sub>/SrTiO<sub>3</sub> with the help of the coefficients  $r_1, r_2$  from Ref. [53]. The resulting n(x) is shown in Fig. 3 by filled circles. We fit this data by the equation

$$n(x) = \frac{G}{(x+d)^{12/7}},$$
(22)

which is similar to Eq. (14), but G is a fitting parameter independent of d. We see from Fig. 3 that this fit is good,



FIG. 3. The experimental distribution of electrons n(x) (circles) from Ref. [22] obtained by time-resolved photoluminescence in convenient dimensionless units. Fitting with Eq. (22) is shown by the solid line. The fitting parameters are G = 0.16 and  $d = 250b \simeq 9$  nm.

however, the parameter *G* is seven times larger than the parameter  $C_4(b/a)^{12/7}/A^{3/7}$  entering Eq. (14). This can be explained by a 50-fold increase of the Auger coefficient  $r_2$  of Eq. (21) near the interface. A similar surface effect was observed in Ref. [54] for SrTiO<sub>3</sub> nanocrystals, where  $r_2$  is almost 150 times larger than for bulk SrTiO<sub>3</sub>.

From the fit we get  $d = 250b \simeq 9$  nm, which corresponds to an electron surface charge density  $\sigma = D_0/4\pi = 0.08e/a^2$  in agreement with other data [11,23,51]. One can check that this result is self-consistent, i.e., for such  $\sigma$  one can use Eq. (14) for the fitting of experimental data, because  $k_F d > 1$  [see Eq. (19)] and  $D_0 > D_c$  [see Eq. (17)].

# V. QUANTUM CAPACITANCE OF AN ACCUMULATION LAYER

In this section we address the capacitance of an accumulation layer, for example in the double junction metal/GdTiO<sub>3</sub>/SrTiO<sub>3</sub>. If the width of insulating GdTiO<sub>3</sub> layer is  $L \gg d$  we may view an accumulation layer as a conducting two-dimensional gas (2DEG). One can apply a positive voltage V between the metal and the 2DEG and measure the additional charge per unit area  $\sigma'$  and  $-\sigma'$ , which are induced on the metal and in the 2DEG, respectively. The capacitance per unit area of junction is  $C = d\sigma'/dV$ . If we imagine that the 2DEG is a perfect metal, the additional charge  $-\sigma'$  resides exactly in the plane of the  $GdTiO_3/SrTiO_3$  junction and the capacitance is equal to the geometric capacitance  $C = C_G = \kappa_G/4\pi L$ , where  $\kappa_G = 30$ is the dielectric constant of the GdTiO3 layer. Actually, the accumulation layer is not a perfect metal so that an additional negative charge  $-\sigma'$  is distributed in a layer of finite width d. As a result  $C^{-1} = C_G^{-1} + C_q^{-1}$ , where  $C_q$  is called the quantum capacitance [55]. Quantum capacitance is broadly studied for many 2DEGs such as silicon MOSFETs [56], GaAs/Ga<sub>x</sub> Al<sub>1-x</sub>As heterostructures [57], and graphene [58].

Concentrating on the case of  $D_0 \gg D_c$  and using Eqs. (12) and (16) for the potential difference between x = 0 and  $x = \infty$  we have at  $\sigma' = 0$ :

$$\varphi(0) = \frac{C_3}{C_6^{8/7}} \frac{e}{b} A^{2/5} \left(\frac{b}{a}\right)^{8/5} \left(\frac{D_0}{e/a^2}\right)^{8/5}.$$
 (23)

After the transfer of charge  $-\sigma'$  from the metal to the electron gas the potential changes as

$$\Delta \varphi = \frac{C_3}{C_6^{8/7}} \frac{e}{b} A^{2/5} \left(\frac{b}{a}\right)^{8/5} \left[ \left(\frac{D_0 + 4\pi \sigma'}{e/a^2}\right)^{8/5} - \left(\frac{D_0}{e/a^2}\right)^{8/5} \right].$$
(24)

Assuming that  $D_0 \gg 4\pi\sigma'$ , linearizing  $\Delta\varphi$  with respect of  $\sigma'$  and adding the voltage drop across the GdTiO<sub>3</sub> layer we get

$$V = \frac{e}{b} \frac{8}{5} \frac{C_3}{C_6^{8/7}} A^{2/5} \left(\frac{b}{a}\right)^{8/5} \frac{4\pi\sigma'}{e/a^2} \left(\frac{D_0}{e/a^2}\right)^{3/5} + \frac{4\pi L\sigma'}{\kappa_G}.$$

Using Eq. (16) for d we can write for the total capacitance:

$$C^{-1} = \frac{7}{5} \frac{4\pi d}{\kappa_{\text{eff}}} + \frac{4\pi L}{\kappa_G},\tag{25}$$

where

$$\kappa_{\rm eff} = \frac{(4\pi)^3}{A[D_0/(e/a^2)]^2}$$

is the effective nonlinear dielectric constant D/E at x = 0 [see Eq. (6)]. The first term of Eq. (25) is the inverse quantum capacitance  $C_q^{-1}$ , and the last term is the inverse geometric capacitance  $C_G^{-1}$ .

Due to the polar catastrophe in GdTiO<sub>3</sub> we get  $D_0 = 2\pi e/a^2$  and  $\kappa_{\text{eff}} = 63$ . The ratio of inverse capacitances is

$$\frac{C_q^{-1}}{C_G^{-1}} = \frac{7}{5} \frac{\kappa_G}{\kappa_{\text{eff}}} \frac{d}{L}.$$

For L = 40 Å we get  $C_q^{-1}/C_G^{-1} = 0.1$ . One can extend our calculation to the double junction  $SrTiO_3/GdTiO_3/SrTiO_3$ . In that case the inverse quantum capacitance is the sum of inverse quantum capacitances of both junctions.

#### VI. FROM ACCUMULATION TO INVERSION LAYER

In the previous section we considered an accumulation layer and showed that it can be described by the Thomas-Fermi approximation, when the Fermi level is near the bottom of the conduction band. But the Fermi level can be moved into the gap, for example, by a back gate. In a *p*-type inversion layer in SrTiO<sub>3</sub> (see Fig. 4) the Fermi level can be even deeper, at the top of the bulk valence band. In both cases, the charge of electrons compensates only a fraction of the external electric field  $D_0$ , i.e., Eq. (7) is violated. For example, the rest of the negative charge in the inversion layer is provided by negative acceptors (see Fig. 4) and the electron surface charge density  $\sigma < D_0/(4\pi)$ . To calculate the width of the electron density profile  $d(\sigma, D_0)$  below we use a scaling approach, i.e., we neglect all numerical coefficients.

If the surface electron density is high enough  $\sigma \gg e/d^2$ , the electron gas is three dimensional (3DEG), and one can use the Thomas-Fermi approach and the kinetic energy is

$$K = \frac{\hbar^2}{2m} \left(\frac{\sigma}{de}\right)^{2/2}$$

[see Fig. 5(a)]. On the other hand, at  $\sigma \ll e/d^2$  the electrons are confined to the first subband of the triangular potential



FIG. 4. (Color online) Inversion layer of a *p*-type semiconductor. Holes have been forced away by an external electric field  $D_0$ . The field is so strong that it attracts mobile electrons (blue region) in the inversion layer with width *d* and total surface charge density  $\sigma$ . Negatively charged acceptors are shown by blue minuses. The Fermi level  $\varepsilon_F$  is plotted by a dashed line.

well, so that the electron gas is two dimensional (2DEG) [59]. The electron kinetic energy is then

$$K = \frac{\hbar^2}{2md^2}.$$

In both cases, the characteristic potential energy of electrons is U = eEd [see Fig. 5(b)].

The dielectric response can be linear  $E = D_0/\kappa$  or nonlinear  $E = AD_0^3/P_0^2$  depending on whether the external electric field  $D_0$  is smaller or larger than  $D_c$ , respectively. This gives us four cases which correspond to high  $\sigma \gg e/d^2$  and low  $\sigma \ll e/d^2$  electron charge density, small field  $D_0 \ll D_c$ , and large field  $D_0 \gg D_c$ . The diagram in Fig. 5(c) shows the combined scaling results for  $d(\sigma, D)$  for all these four domains. We use  $\sigma$  as an independent coordinate because the three-dimensional concentration  $\sigma/d$  can be obtained by Hall effect measurements, while d can also be measured independently [22].

First, let us consider the domain of high electron charge density  $\sigma \gg e/d^2$  and small field  $D_0 \ll D_c$ . The kinetic energy is  $K = \hbar^2/2m(\sigma/ed)^{2/3}$  and the potential energy is  $U = edD_0/\kappa$ . From the condition K = U one obtains

$$d = b\kappa^{3/5} \frac{\sigma^{2/5}}{D_0^{3/5}} \left(\frac{e}{b^2}\right)^{1/5}$$
(26)

[see Fig. 5(c)]. When the neutrality relation (7) is satisfied:

$$\sigma = \frac{D_0}{4\pi},\tag{27}$$

we recover the previous accumulation layer result, Eq. (15). For low electron charge density  $\sigma \ll e/d^2$  and small field  $D_0 \ll D_c$  we get

$$d = b \left(\frac{\kappa e/b^2}{D_0}\right)^{1/3}.$$
 (28)

This is the classical result for the width of the inversion layer [60]. The critical electron density at which the transition from low to high electron charge density occurs can be obtained from the condition  $\sigma_c = e/d^2$  or from equating



FIG. 5. (Color online) Schematic energy diagram for conduction band electrons in the triangular potential well created by an external electric field  $D_0$  for two different cases: (a) 3DEG, when many subbands are filled by electrons and (b) 2DEG, when only the lowest subband is partially filled. This filling is shown by light yellow and dark green, respectively. (c) Schematic phase diagram for the dependence of the width of electron gas d in units of b on electron surface charge density  $\sigma$  and external electric field  $D_0$  both in the units  $e/b^2$  for SrTiO<sub>3</sub> with dielectric constant  $\kappa$ . The four domains of this diagram correspond to four different formulas for d. If the electric field is small,  $D_0 < 1/\sqrt{\kappa}$ , the linear relation between electric field and electric induction Eq. (5) is applicable. For  $D_0 > 1/\sqrt{\kappa}$  Eq. (5) is replaced by the nonlinear relation Eq. (6). The light and dark gray (yellow and green in color) regions distinguish 3DEG and 2DEG cases. The border lines between domains are marked by numbers of corresponding equations (27), (29), and (32).

expressions (26) and (28) for d, giving

$$\sigma_c = \frac{e}{b^2} \frac{1}{\kappa^{2/3}} \left(\frac{D_0}{e/b^2}\right)^{2/3}.$$
 (29)

We emphasize the difference between the two regimes of low [Eq. (28)] and high [Eq. (26)] electron charge densities. Sometimes [14,25] Eq. (28) for *d* of an inversion layer is used for an accumulation layer.

So far we described the two left, low  $D_0$  [ $D_0 < D_c = e/(a^2\sqrt{\kappa})$ ] domains of Fig. 5(c). For a large field  $D_0 \gg D_c$  and a high electron charge density  $\sigma \gg e/d^2$  we get

$$d = b \left(\frac{b}{a}\right)^4 \frac{\sigma^{2/5}}{D_0^{9/5}} \left(\frac{e}{b^2}\right)^{7/5}.$$
 (30)

At  $\sigma = D/4\pi$ , i.e., when the neutrality relation is satisfied, we get the previous result Eq. (16). At last, for a low electron charge density  $\sigma \ll e/d^2$  and large field  $D_0 \gg D_c$  we get

$$d = b \left(\frac{b}{a}\right)^{4/3} \frac{e/b^2}{D_0}.$$
(31)



FIG. 6. (Color online) Schematic energy diagram of a depletion layer of an n-doped semiconductor. Electrons have been forced away by an external electric field, which is created by negative charges on the surface of the semiconductor. As a result, the depletion layer with width W is filled by positively charged donors (red pluses).

The transition from low to high electron charge density at  $D_0 \gg D_c$  occurs at the critical electron density:

$$\sigma_c = \frac{e}{b^2} \left( \frac{D_0}{e/b^2} \right)^2. \tag{32}$$

All four results Eqs. (26), (28), (30), and (31) and all the border lines between different domains, Eqs. (27), (29), and (32), are shown in Fig. 5(c).

#### **VII. DEPLETION LAYER**

Schottky diodes are metal-*n*-type-SrTiO<sub>3</sub> junctions where the electron gas is depleted near the interface. Below we calculate the capacitance of this junction as a function of concentration of donors N and applied voltage V, which has been experimentally studied [17–20].

In order to calculate the capacitance of the Schottky diode let us consider Eqs. (1). We use the full depletion approximation, in which we assume that  $SrTiO_3$  is fully depleted from electrons over a distance *W* from the surface (Fig. 6). The charge density in that region is due to the ionized donors with 3D concentration *N*:

$$\rho = \frac{e}{a^3} (Na^3), \quad x < W,$$
  

$$\rho = 0, \quad x > W.$$
(33)

From Eqs. (33) and (1) one can determine the dependence of the width W on the potential drop across the depletion layer  $\Delta \varphi = \varphi(\infty) - \varphi(0) = V_0 - V > 0$ , where the negative voltage V is applied to the metal and  $V_0$  is the potential difference between the work functions of SrTiO<sub>3</sub> and the metal. If  $D_0 < D_c$  one can use the linear relation (5) to get

$$\Delta \varphi = \frac{2\pi}{\kappa} \frac{e}{a} (Na^3) \left(\frac{W}{a}\right)^2.$$
(34)

If  $D_0 > D_c$ , we use the nonlinear expression (6):

$$\Delta \varphi = A \frac{1}{4} \frac{e}{a} (Na^3)^3 \left(\frac{W}{a}\right)^4.$$
(35)



FIG. 7. (Color online) Experimental capacitance-voltage (*C*-*V*) characteristics at room temperature for Nb-doped SrTiO<sub>3</sub> for three values of the donor concentration *N*. The data for doping concentrations  $Na^3 = 9.5 \times 10^{-3}$  and  $Na^3 = 2.9 \times 10^{-5}$  are from Ref. [18], the one for concentration  $Na^3 = 1.4 \times 10^{-2}$  are from Ref. [41]. The capacitance and the voltage are measured in dimensionless units 1/a and e/a, where *a* is the lattice constant. Experimental results are shown by points. Solid lines correspond to fitting by Eq. (36) and by Eq. (37) for the smaller (upper line) and larger concentration (two lower lines), respectively.

Now one can calculate the capacitance per unit area of the Schottky diode

$$C = \frac{dQ}{d\Delta\varphi} = eN \left| \frac{dW}{d\Delta\varphi} \right|$$

This gives

$$\frac{1}{(Ca)^2} = \frac{8\pi}{\kappa} \left(\frac{V_0 - V}{e/a}\right) \frac{1}{Na^3}$$
(36)

for  $D_0 < D_c$ , and

$$\frac{1}{(Ca)^2} = 8\left(\frac{V_0 - V}{e/a}\right)^{3/2} \left(\frac{A}{Na^3}\right)^{1/2}$$
(37)

for  $D_0 > D_c$ . With growing V the crossover between Eqs. (36) and (37) happens at  $V = V_c$  where

$$V_c = V_0 - \frac{e}{a} \left(\frac{\pi}{\kappa}\right)^2 \frac{1}{ANa^3}.$$
 (38)

To test our theoretical predictions we consider the experimental data obtained at room temperature for lightly and heavier Nb-doped SrTiO<sub>3</sub> of Ref. [18],  $Na^3 = 2.9 \times 10^{-5}$  and  $Na^3 = 9.5 \times 10^{-3}$  as well as the more heavily doped sample  $Na^3 = 1.4 \times 10^{-2}$  of Ref. [41] (see Fig. 7). In the first case, one can expect  $D_0$  to be small and we use Eq. (36) to fit the experimental data. Using  $\kappa$  and  $V_0$  as fitting parameters we get  $V_0 = 0.5e/a, \kappa \simeq 320$ , which is close to the room temperature value  $\kappa \simeq 350$  for SrTiO<sub>3</sub>. For the heavier doped samples, one can expect that  $D_0$  is large and use Eq. (37) to find A and  $V_0$  from the experimental data. As a result, we get  $V_0 = 0.8e/a$ ,  $A \simeq 1.5$  and  $V_0 = 0.26e/a, A \simeq 0.8$  for the data of Refs. [18] and [41], respectively [61].

In Refs. [18,39–41,62] C(V) data are used to derive  $\kappa(E) \equiv dD/dE$ , which agrees with the Landau description [40]. Extracting the parameter A from  $\kappa(E)$  of Ref. [39] gives A = 0.5, while  $\kappa(E)$  in Refs. [18,41] lead to the above mentioned values of A. The scatter of values of A probably can

be explained by the effect of a noncontrollable "dead layer" of low dielectric constant between SrTiO<sub>3</sub> and the metal [18,41].

It is believed [41] the best interface was made in Ref. [41], whose data lead to  $A \simeq 0.8$ . Above we used this value for all numerical estimates [63].

## VIII. DOES SPATIAL DISPERSION OF DIELECTRIC RESPONSE AFFECT THE ACCUMULATION LAYER?

So far we assumed that the external field  $D_0$  and polarization P change abruptly at the interface, i.e., we have ignored the dispersion of the dielectric response. In this section we show that even without this assumption, the results of previous sections remain intact for SrTiO<sub>3</sub>. We concentrate on an accumulation layer which can be so narrow that the question of spatial dispersion arises. To take into account that the electric field and polarization cannot change abruptly, in the geometry chosen above we add the gradient term  $(1/2)a_0^2(dP/dx)^2$  to the Landau-Ginzburg free energy density Eq. (2). Here  $a_0^2 = g_{xxxx}$  in the general expression for the gradient term

$$\frac{1}{2}g_{iklm}\frac{\partial P_i}{\partial x_k}\frac{\partial P_l}{\partial x_m}$$

and *i*,*k*,*l*,*m* enumerate the three coordinates *x*,*y*,*z*. Adding such a term necessitates an additional boundary condition for P(x) at x = 0. The general form of this condition [64] is  $P + \lambda dP/dx = 0$ . It does not bring new physics into the problem if  $\lambda$  is large. Therefore, we explore the opposite case where  $\lambda$  can be ignored, P(0) = 0. From the condition  $\partial F/\partial P = 0$  we get

$$E = \frac{4\pi}{\kappa}P + \frac{A}{P_0^2}P^3 - a_0^2 \frac{d^2P}{dx^2}.$$
 (39)

Let us first find a solution to Eqs. (1) together with Eq. (39) for a system without electrons  $[\rho(x) = 0]$ . The electrical induction  $D_0 = E + 4\pi P$  is constant everywhere and we get an approximate solution for  $E = D_0 - 4\pi P$ :

$$E \simeq (D_0 - E_\infty) \exp\left(-\sqrt{4\pi}\frac{x}{a_0}\right) + E_\infty, \qquad (40)$$

where  $E_{\infty} = D_0/\kappa + (D_0/4\pi)^3/P_0^2$  is the electric field in the bulk.

Now, we take electrons into account and start from the case where in Eq. (39)  $\kappa = \infty$  and  $P_0 = \infty$ , so that the role of the gradient term is emphasized. From the resulting equation  $E = -a_0^2 d^2 P/dx^2$  we get  $\varphi = a_0^2 dP/dx$ . (We take into account that at  $x \to \infty$ ,  $\varphi = dP/dx = 0$ .) From the condition  $dE/dx + 4\pi dP/dx = 4\pi e\rho$  we get an equation for the potential:

$$\frac{d^2\varphi}{dx^2} = \frac{4\pi}{a_0^2}\varphi + \frac{e}{b^3}\frac{2^{7/2}}{3\pi}\left(\frac{\varphi}{e/b}\right)^{3/2}$$
(41)

[compare with Eq. (9)]. The solution to this equation, with boundary condition  $\varphi(\infty) = 0$ , is

$$\varphi(x) = \frac{15^2 \pi^4}{2^7} \frac{eb^3}{a_0^4} \frac{1}{\sinh^4 X},$$
(42)

where  $X = \sqrt{\pi}(x+d)/2a_0$  and *d* is still an unknown length, which can be found from the boundary condition. Let us

describe this solution. First, we see that the solution critically depends on the parameter X. If X > 1 or  $x > a_0 - d$  then one arrives at Eq. (40): the electric field and potential decay exponentially to zero as  $\propto \exp(-\sqrt{4\pi x/a_0})$ .

Let us consider the case when X < 1 or  $x < a_0 - d$ . In that case we get a result similar to Eq. (10), but with  $\kappa = 1$ ,

$$\varphi = \frac{225\pi^2}{8} \frac{eb^3}{(x+d)^4}.$$
(43)

Thus, at  $x < a_0 - d$  the electrons screen the external field much faster than the lattice, and hence the latter does not participate appreciably in the screening response.

One can try to estimate *d* assuming that the majority of electrons are in the region  $0 < x < a_0 - d$ . In that case the potential is given by Eq. (43) and *d* can be found from the condition of neutrality (7):

$$d_1 = C_5 b \left(\frac{e/b^2}{D_0}\right)^{1/5}.$$
 (44)

The condition  $d_1 < a_0$ , required for applicability of Eq. (43), is valid only when  $D_0 \gg D_d$ , where

$$D_d = C_5^5 b^3 \frac{e}{a_0^5} \simeq 1000 \left(\frac{a}{a_0}\right)^5 \left(\frac{b}{a}\right)^3 \frac{e}{a^2}.$$
 (45)

Let us now consider the opposite case  $D_0 \ll D_d$ , where most of the electrons are not located in the region  $x < a_0 - d$ . In fact, to find where they are located one has to introduce finite  $\kappa$  and  $P_0$ . The exponential decay of the electric field with x saturates at the level of  $E_{\infty}$ . At larger x we reach the electron gas with a larger width given by Eqs. (15) or (16) for linear and nonlinear dielectric responses, respectively. In this case, the electron gas resides at large distances and our considerations of the previous section are valid.

If we return to finite  $\kappa$  and  $P_0$  in Eq. (39) in the first case  $D_0 \gg D_d$ , we arrive to the following hybrid picture. An electron gas screens most of the external field at small  $x \simeq d_1$ . At larger distances we again get exponential decay described by Eq. (40), which results from the lattice response. After this exponential decay the electron concentration n(x) again follows Eqs. (11) or (14), with the bulk value of the dielectric constant  $\kappa$ . In other words, the electron density has a two component distribution. The electrons nearest to the interface are likely localized due to disorder. This could, in principle, create hope to explain why at the LaAlO<sub>3</sub>/SrTiO<sub>3</sub> interface only 10% of the electrons predicted by the polar catastrophe  $\sigma = 0.5 \cdot e/a^2$  are observed in transport measurements.

However, this hope does not survive for actual parameters of SrTiO<sub>3</sub>, b = 0.35 Å,  $a_0 = 1$  Å [65]. Although for such parameters the Thomas-Fermi approximation is still valid, the electric field at which all electrons are located within the distance  $a_0$  from the interface,  $D_d = 650e/a^2$  is so large that  $D_0 \ll D_d$  always. One may also worry that for such values of b and  $a_0$ , which are smaller than the lattice constant a of SrTiO<sub>3</sub>, the continuous theory that we use is not applicable.

However, the applicability of the continuous theory can be much better due to the so called background dielectric constant [65]. Indeed, until now we have described the entire dielectric response by the Landau-Ginzburg theory for a single order parameter, which can be identified with the displacement of the transversal optical soft mode of SrTiO<sub>3</sub>. This is a good approach when the dielectric response is very strong. However, near the interface the response of the soft mode is weakened due to the dispersion and the response of other optical modes as well as the polarization of ions must be included. This is done [65] by the addition of the linear nondispersive background dielectric constant  $10 < \kappa_b < 30$  [65,66]. To model this situation one can replace *P* by the soft mode contribution  $P_s$  in the free energy density (2) and add to it  $(2\pi/\kappa_b)P_b^2 - EP_b$  while keeping  $P = P_s + P_b$  in Eqs. (1). Here  $P_b$  is the background polarization.

As a result, the small distance dielectric constant becomes  $\kappa_b$  instead of 1 and the lengths b,  $a_0$ , and the characteristic field of Eq. (45)  $D_d$  are replaced by  $b\kappa_b$ ,  $a_0\sqrt{\kappa_b}$ , and  $D_d\kappa_b^{1/2}$ , respectively. For  $10 < \kappa_b < 30$  the characteristic lengths  $b\kappa_b$  and  $a_0\sqrt{\kappa_b}$  may reach the lattice constant a, thereby improving the applicability of our theory. At the same time the critical field  $D_d$  becomes even larger so that in all realistic situations,  $D_0 \ll D_d$ , all electrons are located at distances larger than the lattice constant a. This means that the dispersion does not play a substantial role and the dispersionless approach used in the previous sections is applicable for describing accumulation, inversion, and depletion layers in SrTiO<sub>3</sub>.

#### **IX. CONCLUSION**

In this paper we studied the potential and electron density depth profiles in accumulation, inversion, and depletion layers for materials with a very large dielectric constant and nonlinear dielectric response such as SrTiO<sub>3</sub>. In particular, we showed that in a depletion layer at a given donor concentration and for high enough voltage, the dependence of the capacitance on voltage decreases as  $C \propto V^{-3/4}$ , which is substantially different from the conventional result  $C \propto V^{-1/2}$  with linear dielectric response. For an inversion layer we found that the layer width depends on the external electric field  $D_0$  as  $d \propto$  $(\kappa/D_0)^{1/3}$  and  $d \propto 1/D_0$  for linear and nonlinear dielectric responses, respectively. In accumulation layers near interfaces like GdTiO<sub>3</sub>/SrTiO<sub>3</sub>, LaTiO<sub>3</sub>/SrTiO<sub>3</sub>, and LaAlO<sub>3</sub>/SrTiO<sub>3</sub> we obtained  $n(x) \propto (x+d)^{-12/7}$  with  $d \propto D_0^{-7/5}$ , due to the nonlinearity of the dielectric response. We found that 70% of electrons are located within 2.6 nm of the interfaces GdTiO<sub>3</sub>/SrTiO<sub>3</sub> and LaTiO<sub>3</sub>/SrTiO<sub>3</sub> (where the electron surface charge density is  $\sigma = 0.5e/a^2$ ) in agreement with experimental data. The predicted functional shape of the electron depth profile n(x) also shows satisfactory agreement with the experimental data. Spatial dispersion in the dielectric response was shown to be negligible for the description of potential and electron density depth profiles in SrTiO<sub>3</sub> devices. This paper uses a simplified isotropic electron spectrum, while the electronic structure of SrTiO<sub>3</sub> is multiorbital in nature. In future work we plan to go beyond our simplified description.

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